

The Chemistry of Radical Polymerization Second fully revised edition

Graeme Moad & David H. Solomon



THE CHEMISTRY OF RADICAL POLYMERIZATION

SECOND FULLY REVISED EDITION

This Page is Intentionally Left Blank

THE CHEMISTRY OF RADICAL POLYMERIZATION

SECOND FULLY REVISED EDITION

GRAEME MOAD

CSIRO Molecular and Health Technologies Bayview Ave, Clayton, Victoria 3168, AUSTRALIA

and

DAVID H. SOLOMON

Department of Chemical and Biomolecular Engineering University of Melbourne, Victoria 3010, AUSTRALIA

2006



Amsterdam – Boston – Heidelberg – London – New York – Oxford – Paris – San Diego San Francisco – Singapore – Sydney – Tokyo ELSEVIER B.V.ELSEVIER Inc.ELSEVIER LtdELSEVIER LtdRadarweg 29525 B Street, Suite 1900The Boulevard, Langford Lane84 Theobalds RoadP.O. Box 211, 1000 AE AmsterdamSan Diego, CA 92101-4495Kidlington, Oxford OX5 1GBLondon WC1X 8RRThe NetherlandsUSAUKUK

© 2006 Elsevier Ltd. All rights reserved.

This work is protected under copyright by Elsevier Ltd, and the following terms and conditions apply to its use:

Photocopying

Single photocopies of single chapters may be made for personal use as allowed by national copyright laws. Permission of the Publisher and payment of a fee is required for all other photocopying, including multiple or systematic copying, copying for advertising or promotional purposes, resale, and all forms of document delivery. Special rates are available for educational institutions that wish to make photocopies for non-profit educational classroom use.

Permissions may be sought directly from Elsevier's Rights Department in Oxford, UK: phone (+44) 1865 843830, fax (+44) 1865 853333, e-mail: permissions@elsevier.com. Requests may also be completed on-line via the Elsevier homepage (http://www.elsevier.com/locate/permissions).

In the USA, users may clear permissions and make payments through the Copyright Clearance Center, Inc., 222 Rosewood Drive, Danvers, MA 01923, USA; phone: (+1) (978) 7508400, fax: (+1) (978) 7504744, and in the UK through the Copyright Licensing Agency Rapid Clearance Service (CLARCS), 90 Tottenham Court Road, London W1P 0LP, UK; phone: (+44) 20 7631 5555; fax: (+44) 20 7631 5500. Other countries may have a local reprographic rights agency for payments.

Derivative Works

Tables of contents may be reproduced for internal circulation, but permission of the Publisher is required for external resale or distribution of such material. Permission of the Publisher is required for all other derivative works, including compilations and translations.

Electronic Storage or Usage

Permission of the Publisher is required to store or use electronically any material contained in this work, including any chapter or part of a chapter.

Except as outlined above, no part of this work may be reproduced, stored in a retrieval system or transmitted in any form or by any means, electronic, mechanical, photocopying, recording or otherwise, without prior written permission of the Publisher. Address permissions requests to: Elsevier's Rights Department, at the fax and e-mail addresses noted above.

Notice

No responsibility is assumed by the Publisher for any injury and/or damage to persons or property as a matter of products liability, negligence or otherwise, or from any use or operation of any methods, products, instructions or ideas contained in the material herein. Because of rapid advances in the medical sciences, in particular, independent verification of diagnoses and drug dosages should be made.

First edition 2006

Library of Congress Cataloging in Publication Data A catalog record is available from the Library of Congress.

British Library Cataloguing in Publication Data A catalogue record is available from the British Library.

ISBN-10: 0-08-044288-9 (Hardbound) ISBN-13: 978-0-08-044288-4 (Hardbound) ISBN-10: 0-08-044286-2 (Paperback) ISBN-13: 978-0-08-044286-0 (Paperback)

☺ The paper used in this publication meets the requirements of ANSI/NISO Z39.48-1992 (Permanence of Paper). Printed in The Netherlands.



| Cor | iten | ts | v |
|--------------------|---------------------|--|----------|
| Index to Tablesxvi | | | |
| Ind | Index to Figures | | |
| Pre | face | to the First Editionxx | kiii |
| Pre | face | to the Second Editionx | xv |
| Ack | Acknowledgmentsxxvi | | |
| 1 | IN | TRODUCTION | 1 |
| 1.1 | - · · · | leferences | |
| | р. | DICAL DEACTIONS | 11 |
| 2 | KA | TOTCAL REACTIONS | 11 |
| 2.1 | 1 | | 11 |
| 2.2 | 2 1 | | 12 |
| 2. | .2.1 | | 12 |
| 2. | .2.2 | Stabilities of Radicals. | 14 |
| 2. | .2.3 | Detection of Radicals | 14 |
| 2.3 | A | Addition to Carbon-Carbon Double Bonds | 16 |
| 2, | .3.1 | Steric Factors | 19 |
| 2. | .3.2 | Polar Factors | 21 |
| 2. | .3.5 | Bond Strengths | 22 |
| 2. | .3.4 | Stereoelectronic Factors | 23 |
| 2. | .3.3 | Entropic Considerations | 24 |
| 2. | .3.0 | Reaction Conditions | 24 |
| | 2.3 | .0.1 Temperature | 24 |
| h | 2.3 | .0.2 Solvent | 20 |
| 2. | .3.7 20 | Summary | 20 |
| 2. | | | 20 |
| 2.4 | 4 1 | Hydrogen Atom Transfer | 29 |
| 2. | .4.1 | Bond Dissociation Energies | 30 |
| - 2. | .4.2 | Steric Factors. | 30 |
| 2. | .4.5 | Polar raciors | 31 22 |
| 2. | .4.4 | Stereoelectronic Factors. | 32 |
| | .4.3 7 4 | A batraction we Addition | 21 |
| 2. | .4.0 | | 24 |
| 2. | .4./ | Summary | 30 |

| 2.5 Radical-Radical Reactions | 36 |
|--|-----------|
| 2.5.1 Pathways for Combination | 37 |
| 2.5.2 Pathways for Disproportionation | 38 |
| 2.5.3 Combination vs Disproportionation | 39 |
| 2.5.3.1 Statistical factors | 39 |
| 2.5.3.2 Steric factors | 40 |
| 2.5.3.3 Polar factors | 41 |
| 2.5.3.4 Stereoelectronic and other factors | 41 |
| 2.5.3.5 Reaction conditions | 42 |
| 2.5.4 Summary | 43 |
| 2.6 References | 44 |
| 3 INITIATION | 49 |
| 3.1 Introduction | 49 |
| 3.2 The Initiation Process | 50 |
| 3.2.1 Reaction with Monomer | 51 |
| 3.2.2 Fragmentation | 54 |
| 3.2.3 Reaction with Solvents, Additives, or Impurities | 55 |
| 3.2.4 Effects of Temperature and Reaction Medium on Radical Reactivity | 55 |
| 3.2.5 Reaction with Oxygen | 56 |
| 3.2.6 Initiator Efficiency in Thermal Initiation | 57 |
| 3.2.7 Photoinitiation | 58 |
| 3.2.8 Cage Reaction and Initiator-Derived Byproducts | 60 |
| 3.2.9 Primary Radical Termination | 61 |
| 3.2.10 Transfer to Initiator | 62 |
| 3.2.11 Initiation in Heterogeneous Polymerization | 63 |
| 3.3 The Initiators | 64 |
| 3.3.1 Azo-Compounds | 68 |
| 3.3.1.1 Dialkyldiazenes | 68 |
| 3.3.1.1.1 Thermal decomposition | 72 |
| 3.3.1.1.2 Photochemical decomposition | 74 |
| 3.3.1.1.3 Initiator efficiency | 74 |
| 3.3.1.1.4 Transfer to initiator | 77 |
| 3.3.1.2 Hyponitrites | 78 |
| 3.3.2 Peroxides | 79 |
| 3.3.2.1 Diacyl or diaroyl peroxides | 82 |
| 3.3.2.1.1 Thermal decomposition | 82 |
| 3.3.2.1.2 Photochemical decomposition | 83 |
| 3.3.2.1.3 Initiator efficiency. | 84 |
| 3.3.2.1.4 Transfer to initiator and induced decomposition | 85 |
| 3.3.2.1.3 Kedox reactions | כא דיס |
| 2.2.2.2 Dialkyi peroxydicarbonates | 10 00 |
| 5.5.2.5 FEIOXYESTEIS | 00 |

vi

| 3.3.2. | 3.1 Thermal decomposition | 88 |
|---|--|---|
| 3.3.2. | 3.2 Photochemical decomposition | |
| 3.3.2.4 | Dialkyl peroxides | |
| 3.3.2.5 | Alkyl hydroperoxides | |
| 3.3.2.6 | Inorganic peroxides | 93 |
| 3.3.2. | 5.1 Persulfate | |
| 3.3.2. | 6.2 Hydrogen peroxide | |
| 3.3.3 Mu | ltifunctional Initiators | |
| 3.3.3.1 | Concerted decomposition | 97 |
| 3.3.3.2 | Non-concerted decomposition | 97 |
| 3.3.4 Pho | tochemical Initiators | |
| 3.3.4.1 | Aromatic carbonyl compounds | |
| 3.3.4. | 1.1 Benzoin and related compounds | |
| 3.3.4. | 1.2 Carbonyl compound-tertiary amine systems | |
| 3.3.4.2 | Sulfur compounds | |
| 3.3.5 Rec | lox Initiators | 104 |
| 3.3.5.1 | Metal complex-organic halide redox systems | 104 |
| 3.3.5.2 | Ceric ion systems | |
| 3.3.6 The | rmal Initiation | 106 |
| 3.3.6.1 | Styrene homopolymerization | |
| 3.3.6.2 | Acrylate homopolymerization | |
| 3.3.6.3 | Copolymerization | 110 |
| 24 The I | | |
| 3.4 Ine E | tadicals | 111 |
| 3.4 Ine E 3.4.1 Car | tadicalsbon-Centered Radicals | |
| 3.4 The F 3.4.1 Car 3.4.1.1 | tadicals bon-Centered Radicals Alkyl radicals | |
| 3.4.1 Car 3.4.1.1 3.4.1.1 3.4.1. | adicals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals | |
| 3.4 The F 3.4.1 Car 3.4.1.1 3.4.1. 3.4.1.2 | adicals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals Aryl radicals | 111 |
| 3.4.1 Car 3.4.1.1 3.4.1.1 3.4.1.2 3.4.1.2 3.4.1.3 | adicals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals Aryl radicals Acyl radicals | |
| 3.4.1 Car 3.4.1.1 3.4.1.1 3.4.1.2 3.4.1.2 3.4.1.3 3.4.2 Ox | tadicals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals Aryl radicals Acyl radicals | |
| 3.4.1 Car 3.4.1.1 3.4.1.1 3.4.1.2 3.4.1.2 3.4.1.3 3.4.2 Oxy 3.4.2.1 | adicals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals Aryl radicals Acyl radicals ygen-Centered Radicals Alkoxy radical | |
| 3.4.1 Car 3.4.1.1 3.4.1.1 3.4.1.2 3.4.1.2 3.4.1.3 3.4.2 Oxy 3.4.2.1 3.4.2.1 3.4.2. | tadicals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals Aryl radicals Acyl radicals gen-Centered Radicals Alkoxy radical 1.1 <i>t</i> -Butoxy radicals | |
| 3.4.1 Car 3.4.1.1 3.4.1.1 3.4.1.2 3.4.1.2 3.4.1.3 3.4.2 Oxy 3.4.2.1 3.4.2. 3.4.2. 3.4.2. | tableals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals Aryl radicals Aryl radicals Acyl radicals /gen-Centered Radicals Alkoxy radicals 1.1 t-Butoxy radicals 1.2 Other t-alkoxy radicals | |
| 3.4.1 Car 3.4.1.1 3.4.1.1 3.4.1.2 3.4.1.3 3.4.2 Oxy 3.4.2.1 3.4.2. 3.4.2. 3.4.2. 3.4.2. 3.4.2. 3.4.2. | tableals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals Aryl radicals Acyl radicals Acyl radicals Acyl radicals Agen-Centered Radicals Alkoxy radical 1.1 t-Butoxy radicals 1.2 Other t-alkoxy radicals 1.3 Primary and secondary alkoxy radical | 111 |
| 3.4.1 Car 3.4.1.1 3.4.1.1 3.4.1.2 3.4.1.3 3.4.2 Oxy 3.4.2.1 3.4.2. 3.4.2. 3.4.2. 3.4.2. 3.4.2. 3.4.2. 3.4.2. | tableals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals Aryl radicals Acyl radicals /gen-Centered Radicals /gen-Centered Radicals Alkoxy radical 1.1 t-Butoxy radicals 1.2 Other t-alkoxy radicals 1.3 Primary and secondary alkoxy radicals Acyloxy and alkoxycarbonyloxy radicals | 111 112 112 113 113 117 117 117 118 118 119 124 125 125 |
| 3.4.1 Car 3.4.1.1 3.4.1.1 3.4.1.2 3.4.1.3 3.4.2 Oxy 3.4.2.1 3.4.2. | tableals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals Aryl radicals Acyl radicals /gen-Centered Radicals /gen-Centered Radicals Alkoxy radical 1.1 t-Butoxy radicals 1.2 Other t-alkoxy radicals 1.3 Primary and secondary alkoxy radical Acyloxy and alkoxycarbonyloxy radicals 2.1 Benzovloxy radicals | 111 112 113 113 117 117 117 118 118 119 124 125 125 126 |
| 3.4.1 Car 3.4.1.1 3.4.1.1 3.4.1.2 3.4.1.3 3.4.2 Oxy 3.4.2.1 3.4.2. | tableals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals Aryl radicals Acyl radicals Acyl radicals /gen-Centered Radicals /gen-Centered Radicals 1.1 t-Butoxy radicals 1.2 Other t-alkoxy radicals 1.3 Primary and secondary alkoxy radical Acyloxy and alkoxycarbonyloxy radicals 2.1 Benzoyloxy radicals 2.2 Alkoxycarbonyloxy radicals | 111 |
| 3.4 The F 3.4.1 Car 3.4.1.1 3.4.1.2 3.4.1.3 3.4.2 Ox 3.4.2.1 3.4.2. | tableals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals Aryl radicals Acyl radicals Acyl radicals /gen-Centered Radicals /gen-Centered Radicals /l t-Butoxy radicals 1.1 t-Butoxy radicals 1.2 Other t-alkoxy radicals | 111 112 112 113 117 117 117 118 119 124 125 125 125 126 127 128 |
| 3.4 The F 3.4.1 Car 3.4.1.1 3.4.1.2 3.4.1.3 3.4.2 Oxy 3.4.2.1 3.4.2.3 3.4.2.4. | tableals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals Aryl radicals Acyl radicals Acyl radicals ygen-Centered Radicals Alkoxy radical 1.1 t-Butoxy radicals 1.2 Other t-alkoxy radicals 1.3 Primary and secondary alkoxy radicals 2.1 Benzoyloxy radicals 2.2 Alkoxycarbonyloxy radicals Hydroxy radicals Sulfate radical anion | 111 112 112 113 117 117 117 118 119 124 125 125 126 127 128 129 |
| 3.4 The F 3.4.1 Car 3.4.1.1 3.4.1.2 3.4.1.3 3.4.2 Ox 3.4.2.1 3.4.2.3 3.4.2.4. 3.4.2.4. 3.4.2.5 | tableals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals Aryl radicals Acyl radicals Acyl radicals /gen-Centered Radicals /gen-Centered Radicals Alkoxy radical 1.1 t-Butoxy radicals 1.2 Other t-alkoxy radicals 1.3 Primary and secondary alkoxy radical Acyloxy and alkoxycarbonyloxy radicals 2.1 Benzoyloxy radicals 2.2 Alkoxycarbonyloxy radicals Sulfate radical anion Alkylperoxy radicals | 111 112 112 113 113 117 117 117 118 119 124 125 126 127 128 129 130 |
| 3.4 The F 3.4.1 Car 3.4.1.1 3.4.1.2 3.4.1.3 3.4.2 Oxy 3.4.2.1 3.4.2. 3.4.3. Oth | tableals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals Aryl radicals Acyl radicals Acyl radicals Acyl radicals Agen-Centered Radicals Alkoxy radicals 1.1 t-Butoxy radicals 1.2 Other t-alkoxy radicals 1.3 Primary and secondary alkoxy radicals 2.1 Benzoyloxy radicals 2.2 Alkoxycarbonyloxy radicals Benzoyloxy radicals Sulfate radical anion Alkylperoxy radicals Benzoyloxy radicals | 111 112 112 113 117 117 117 117 118 119 124 125 126 126 127 128 129 130 131 |
| 3.4.1 Car 3.4.1.1 3.4.1.1 3.4.1.2 3.4.1.3 3.4.2 Oxy 3.4.2.1 3.4.2.3 3.4.2.3 3.4.2.3 3.4.2.4 3.4.2.5 3.4.3.0th 3.4.3.1 | tableals bon-Centered Radicals Alkyl radicals Aryl radicals Aryl radicals Acyl radicals Acyl radicals ygen-Centered Radicals Alkoxy radical 1.1 t-Butoxy radicals 1.2 Other t-alkoxy radicals 1.3 Primary and secondary alkoxy radicals 2.1 Benzoyloxy radicals 2.2 Alkoxycarbonyloxy radicals Butyr adicals Sulfate radical anion Alkylperoxy radicals Sulfate radical anion Alkylperoxy radicals Silicon-centered radicals | 111 112 112 113 117 117 117 117 117 118 119 124 125 126 127 128 129 130 131 131 |
| 3.4.1 Car 3.4.1.1 3.4.1.1 3.4.1.2 3.4.1.3 3.4.2 Ox 3.4.2.1 3.4.2. 3.4.3.1 3.4.3.2 | tableals bon-Centered Radicals Alkyl radicals 1.1 α-Cyanoalkyl radicals Aryl radicals Acyl radicals Acyl radicals ygen-Centered Radicals ygen-Centered Radicals Alkoxy radical 1.1 t-Butoxy radicals 1.2 Other t-alkoxy radicals 1.3 Primary and secondary alkoxy radical 2.1 Benzoyloxy radicals 2.2 Alkoxycarbonyloxy radicals 2.2 Alkoxycarbonyloxy radicals Sulfate radical anion Alkylperoxy radicals sulfur- and selenium-centered radicals | 111 |
| 3.4.1 Car 3.4.1.1 3.4.1.2 3.4.1.3 3.4.2 Ox 3.4.2.1 3.4.2. 3.4.3.1 3.4.3.2 3.4.3.3 | tableals bon-Centered Radicals Alkyl radicals Aryl radicals Acyl radicals Acyl radicals ygen-Centered Radicals Alkoxy radical 1.1 t-Butoxy radicals 1.2 Other t-alkoxy radicals 1.3 Primary and secondary alkoxy radicals 2.1 Benzoyloxy radicals 2.2 Alkoxycarbonyloxy radicals 2.3 Primary and secondary alkoxy radicals 2.4 Roxycarbonyloxy radicals 2.5 Alkoxycarbonyloxy radicals 2.6 Alkoy radicals 2.7 Benzoyloxy radicals 2.8 Alkoxycarbonyloxy radicals 2.9 Alkoxycarbonyloxy radicals 2.1 Benzoyloxy radicals 2.2 Alkoxycarbonyloxy radicals 2.3 Primary and secondary alkoxy radicals 2.4 Roxycarbonyloxy radicals 2.5 Alkoxycarbonyloxy radicals 2.6 Alkylperoxy radicals Sulfate radical anion Alkylperoxy radicals Silicon-centered radicals Sulfur- and selenium-centered radicals Phosphorus-centered radicals | 111 112 113 113 113 117 117 117 117 117 117 118 119 124 125 126 127 128 129 130 131 131 132 132 |

vii

| 3.5 Tec | hniques | |
|--------------------|--|------------|
| 3.5.1 K | inetic Studies | |
| 3.5.2 R | adical Trapping | |
| 3.5.2.1 | Spin traps | |
| 3.5.2.2 | 2 Transition metal salts | |
| 3.5.2.3 | Metal hydrides | |
| 3.5.2.4 | Nitroxides | |
| 3.5.2.5 | α-Methystyrene dimer | |
| 3.5.3 D | irect Detection of End Groups | 141 |
| 3.5.3.1 | Infra-red and UV-visible spectroscopy | |
| 3.5.3.2 | 2 Nuclear magnetic resonance spectroscopy | |
| 3.5.3.3 | B Electron paramagnetic resonance spectroscopy | |
| 3.5.3.4 | Mass spectrometry | |
| 3.5.3.5 | 5 Chemical methods | |
| 3.5.4 La | abeling Techniques | |
| 3.5.4.1 | Radiolabeling | |
| 3.5.4.2 | 2 Stable isotopes and nuclear magnetic resonance | |
| 3.6 Refe | erences | 149 |
| 4 PROP | PAGATION | |
| 4.1 Intr | oduction | |
| 4.2 Ster | easequence Isomerism - Tacticity | 168 |
| 421 T | erminology and Mechanisms | 168 |
| 422 F | xperimental Methods for Determining Tacticity | 173 |
| 423 T | acticities of Polymers | 173 |
| 12.5 I | issequence Isomerium II and us Tail Addition | 176 |
| 4.3 Keg | losequence Isomensin - neau vs Tan Audulon | 170 176 |
| 4.3.1 1 | Delv(vinv) acetata) | 170 178 |
| 4313 | Poly(vinyl acciace). | |
| 4313 | Fluoro-olefin polymers | |
| 4314 | 1 Allyl polymers | |
| 4315 | Acrylic polymers | |
| 432 C | onjugated Diene Polymers | 187 |
| 4321 | Polybutadiene | 184 |
| 4322 | Polychloronrene_polyisonrene | 184 |
| A A Star | totural Isomorism - Doarrongomont | |
| 4.4 500 | velopolymerization | |
| 4.4.1 0 | | 105 196 |
| 4.4.1.1 | Triene monomers | |
| н.н.1.2 ЛЛ12 | 1 1 and 1.5 dienes | 191 102 |
| 4.4.1.3 4 4 1 4 | 1. 1.7 and higher 1 n dianac | ۲۶۲ ۱۵۶ |
| 4.4.1.4 ////14 | r 1,7- and inghet 1,7-denes | |
| 4.4.1.J 447 D | Sycio-coporymenization ing-Opening Polymerization | 194 104 |
| 7.7.4 K | mg-opening rotymenzation | |

viii

| 4.4.2.1 | Vinyl substituted cyclic compounds | 196 |
|----------------|--|-----------------------|
| 4.4.2.2 | Methylene substituted cyclic compounds | 199 |
| 4.4.2.3 | Double ring-opening polymerization | 205 |
| 4.4.3 Int | ramolecular Atom Transfer | 208 |
| 4.4.3.1 | Polyethylene and copolymers | 208 |
| 4.4.3.2 | Vinyl polymers | 211 |
| 4.4.3.3 | Acrylate esters and other monosubstituted monomers | 211 |
| 4.4.3.4 | Addition-abstraction polymerization | 212 |
| 4.5 Prop | agation Kinetics and Thermodynamics | 213 |
| 4.5.1 Po | lymerization Thermodynamics | 213 |
| 4.5.2 Me | asurement of Propagation Rate Constants | 216 |
| 4.5.3 De | pendence of Propagation Rate Constant on Monomer Structure | 218 |
| 4.5.4 Ch | ain Length Dependence of Propagation Rate Constants | 220 |
| 4.6 Refe | rences | 221 |
| 5 TERM | INATION | 233 |
| 5.1 Intro | duction | 233 |
| 5.1 Inde | al Dadical Tarmination | 324 |
| 5.2 Kau | mination K instice | 234 225 |
| 5211 | Classical kinetics | 235 |
| 5 2 1 2 | Molecular weights and molecular weight averages | 238 |
| 5 2 1 3 | Molecular weight distributions | . 240 |
| 5.2.1.4 | Diffusion controlled termination | 242 |
| 5.2.1 | 4.1 Termination at low conversion | 244 |
| 5.2.1 | 4.2 Termination at medium to high conversions | 248 |
| 5.2.1.5 | Termination in heterogeneous polymerization | 249 |
| 5.2.1.6 | Termination during living radical polymerization | 250 |
| 5.2.2 Dia | sproportionation vs Combination | 251 |
| 5.2.2.1 | Model studies | 252 |
| 5.2.2 | 1.1 Polystyrene and derivatives | 253 |
| 5.2.2 | 1.2 Poly(alkyl methacrylates) | 255 |
| 5.2.2 | 1.3 Poly(methacrylonitrile) | 256 |
| 5.2.2 | 1.4 Polyethylene | 258 |
| 5.2.2.2 | Polymerization | 258 |
| 5.2.2 | 2.1 Polystyrene | 260 |
| 5.2.2 | 2.2 Poly(alkyl methacrylates) | 261 |
| 5.2.2 | 2.3 Poly(methacrylonitrile) | 262 |
| 5.2.2 | 2.5 Dely(alkyl acrylates) | 262 |
| 5.2,2 | 2.6 Poly(acrytonitrile) | 262 |
| 5.2.2 | 2.0 roiy(vinyi acetate) | 203 |
| 5.2.2 5.2.2 | Summary | 203 |
| J.2.2.3 | Summary | CU2 |
| 5.5 Inhil | oltion and Retardation | 204 |

| | 5.3.1 " | Stable' Radicals | 267 |
|----|---------|--|------|
| | 5.3.2 (| Oxygen | 268 |
| | 5.3.3 N | Monomers | 269 |
| | 5.3.4 F | Phenols | 270 |
| | 5.3.5 (| Quinones | 271 |
| | 5.3.6 F | Phenothiazine | 272 |
| | 5.3.7 N | Nitrones, Nitro- and Nitroso-Compounds | 272 |
| | 5.3.8 7 | Fransition Metal Salts | 273 |
| 5. | .4 Re | ferences | 273 |
| 5. | .4 Re | ferences | 279 |
| 6 | СНА | IN TRANSFFR | 279 |
| 6 | 1 Int | raduction | |
| 6 | 2 Ch | ain Tranefar | |
| v | 621 M | Measurement of Transfer Constants | 283 |
| | 621 | 1 Addition-fragmentation | 285 |
| | 621 | 2 Reversible chain transfer | 288 |
| | 622.1 | Homolytic Substitution Chain Transfer Agents | 280 |
| | 622 | 1 Thiols | 200 |
| | 622 | 2 Disulfides | 201 |
| | 622 | 3 Monosulfides | 202 |
| | 622 | A Halocathons | 202 |
| | 622 | 5 Solvents and other reagents | 295 |
| | 623 | Addition-Fragmentation Chain Transfer Agents | 206 |
| | 623 | 1 Vinyl ethers | 208 |
| | 623 | Allyl sulfides sulfonates halides phosphonates silanes | 200 |
| | 623 | 3 Allyl perovides | 303 |
| | 623 | A Macromonomers | 305 |
| | 623 | 5 Thionoester and related transfer agents | 308 |
| | 624 | Abstraction-Fragmentation Chain Transfer | 309 |
| | 625 (| Catalytic Chain Transfer | 310 |
| | 625 | 1 Mechanism | 310 |
| | 625 | 2 Catalysts | 313 |
| | 6.2 | 5.2.1 Cobalt porphyrin and related complexes | 313 |
| | 6.2 | 5.2.2 Cobalt (II) cobaloximes | .313 |
| | 6.2 | 5.2.3 Cobalt (III) cobaloximes | 314 |
| | 6.2 | 5.2.4 Other catalysts | 315 |
| | 625 | 3 Reaction conditions | 315 |
| | 6.2.6 7 | Fransfer to Monomer | |
| | 6.2.6 | 1 Styrene | |
| | 6.2.6 | 2 Vinvl acetate | 318 |
| | 6.2.6 | .3 Vinyl chloride | 318 |
| | 6.2.6 | 4 Allyl monomers | |
| | | - | |

x

| 6.2.7 Tra | ansfer to Polymer | 320 |
|-----------|--|-----|
| 6.2.7.1 | Polyethylene | 321 |
| 6.2.7.2 | Poly(alkyl methacrylates) | 321 |
| 6.2.7.3 | Poly(alkyl acrylates) | 322 |
| 6.2.7.4 | Poly(vinyl acetate) | 323 |
| 6.2.7.5 | Poly(vinyl chloride) | 325 |
| 6.2.7.6 | Poly(vinyl fluoride) | 325 |
| 6.2.8 Tra | insfer to Initiator | 325 |
| 6.3 Refe | rences | 325 |
| 7 COPO | LYMERIZATION | 333 |
| 7.1 Intro | duction | 333 |
| 7.2 Cope | lymer Depiction | 335 |
| 7.3 Prop | agation in Statistical Copolymerization | 335 |
| 7.3.1 Pro | pagation Mechanisms in Copolymerization | 337 |
| 7.3.1.1 | Terminal model | 337 |
| 7.3.1.2 | Penultimate model | 342 |
| 7.3.1 | .2.1 Model description | 342 |
| 7.3.1 | .2.2 Remote substituent effects on radical addition | 344 |
| 7.3.1 | .2.3 MMA-S copolymerization | 347 |
| 7.3.1 | .2.4 Other copolymerizations | 348 |
| 7.3.1 | .2.5 Origin of penultimate unit effects | 349 |
| 7.3.1.3 | Models involving monomer complexes | 350 |
| 7.3.1.4 | Copolymerization with depropagation | 353 |
| 7.3.2 Ch | ain Statistics | 354 |
| 7.3.2.1 | Binary copolymerization according to the terminal model | 354 |
| 7.3.2.2 | Binary copolymerization according to the penultimate model | 355 |
| 7.3.2.3 | Binary copolymerization according to other models | 356 |
| 7.3.2.4 | Terpolymerization | 357 |
| 7.3.3 Est | timation of Reactivity Ratios | 359 |
| 7.3.3.1 | Composition data | 360 |
| 7.3.3.2 | Monomer sequence distribution | 362 |
| 7.3.4 Pre | ediction of Reactivity Ratios | 363 |
| 7.3.4.1 | <i>Q-e</i> scheme | 363 |
| 7.3.4.2 | Patterns of reactivity scheme | 365 |
| 7.4 Tern | nination in Statistical Copolymerization | 366 |
| 7.4.1 Ch | emical Control Model | 366 |
| 7.4.2 Di | ffusion Control Models | 368 |
| 7.4.3 Co | mbination and Disproportionation during Copolymerization | 370 |
| 7.4.3.1 | Poly(methyl methacrylate-co-styrene) | 371 |
| 7.4.3.2 | Poly(methacrylonitrile-co- styrene) | 373 |
| 7.4.3.3 | Poly(butyl methacrylate-co-methacrylonitrile) | 374 |
| 7.4.3.4 | Poly(butyl methacrylate-co-methyl methacrylate) | 374 |

xi

| 7.4.3.5 Poly(ethylene-co-methacrylonitrile) | |
|---|--|
| 7.5 Functional and End-Functional Polymers | |
| 7.5.1 Functional Initiators | |
| 7.5.2 Functional Transfer Agents | |
| 7.5.3 Thiol-ene Polymerization | 378 |
| 7.5.4 Functional Monomers | |
| 7.5.5 Functional Inhibitors | |
| 7.5.6 Compositional Heterogeneity in Functional Copolymers | |
| 7.6 Block & Graft Copolymerization | |
| 7.6.1 Polymeric and Multifunctional Initiators | |
| 7.6.2 Transformation Reactions | |
| 7.6.3 Radiation-Induced Grafting Processes | |
| 7.6.4 Radical-Induced Grafting Processes | |
| 7.6.4.1 Maleic anhydride graft polyolefins | |
| 7.6.4.2 Maleate ester and maleimide graft polyolefins | |
| 7.6.4.3 (Meth)acrylate graft polyolefins | |
| 7.6.4.4 Styrenic graft polyolefins | |
| 7.6.4.5 Vinylsilane graft polyolefins | |
| 7.6.4.6 Vinyl oxazoline graft polyolefins | 400 |
| 7.6.5 Polymerization and Copolymerization of Macromonomers | 400 |
| 7.7 References | |
| | |
| 8 CONTROLLING POLYMERIZATION | 413 |
| 8 CONTROLLING POLYMERIZATION8.1 Introduction | 413 413 |
| 8 CONTROLLING POLYMERIZATION | 413 413 414 |
| 8 CONTROLLING POLYMERIZATION | 413 413 414 414 |
| 8 CONTROLLING POLYMERIZATION | 413 413 414 414 417 |
| 8 CONTROLLING POLYMERIZATION | 413 413 414 414 417 420 |
| 8 CONTROLLING POLYMERIZATION | 413 413 414 414 417 420 421 |
| 8 CONTROLLING POLYMERIZATION | 413 413 414 414 417 420 421 425 |
| 8 CONTROLLING POLYMERIZATION | 413 413 414 414 417 420 421 425 426 |
| 8 CONTROLLING POLYMERIZATION | |
| 8 CONTROLLING POLYMERIZATION | |
| 8 CONTROLLING POLYMERIZATION | 413 414 414 414 417 420 420 421 425 425 426 429 432 432 |
| 8 CONTROLLING POLYMERIZATION | 413 414 414 414 417 420 421 425 425 426 429 432 432 433 |
| 8 CONTROLLING POLYMERIZATION | 413 413 414 414 414 417 420 425 426 426 429 432 433 433 |
| 8 CONTROLLING POLYMERIZATION | 413 413 414 414 414 417 420 420 421 425 426 429 432 432 433 433 433 |
| 8 CONTROLLING POLYMERIZATION | 413 414 414 414 417 420 420 421 425 426 429 432 432 433 433 433 435 437 |
| 8 CONTROLLING POLYMERIZATION | 413 414 414 414 417 420 420 421 425 425 425 426 429 432 432 432 433 433 433 435 437 |
| 8 CONTROLLING POLYMERIZATION | 413 413 414 414 417 420 421 425 425 426 429 432 432 432 433 433 435 437 437 438 |
| 8 CONTROLLING POLYMERIZATION | 413 413 414 414 414 417 420 420 420 421 425 426 429 432 432 433 433 433 435 437 437 437 438 440 |

xii

| 8.4 References | |
|---|--------------------|
| 9 LIVING RADICAL POLYMERIZATION | |
| 9.1 Introduction | |
| 9.1.1 Living? Controlled? Mediated? | |
| 9.1.2 Tests for Living (Radical) Polymerization | |
| 9.2 Agents Providing Reversible Deactivation | |
| 9.3 Deactivation by Reversible Counting and Unimolec | ular Activation457 |
| 9.3.1 Kinetics and Mechanism | |
| 9.3.1.1 Initiators, iniferters, initers | |
| 9.3.1.2 Molecular weights and distributions | |
| 9.3.1.3 Polymerization kinetics | |
| 9.3.2 Sulfur-Centered Radical Mediated Polymerization | |
| 9.3.2.1 Disulfide initiators | |
| 9.3.2.2 Monosulfide initiators | |
| 9.3.2.3 Monomers, mechanism, side reactions | |
| 9.3.3 Selenium-Centered Radical Mediated Polymerizatio | n466 |
| 9.3.4 Carbon-Centered Radical Mediated Polymerization. | |
| 9.3.4.1 Monomers, mechanism, side reactions | |
| 9.3.5 Reversible Addition-Fragmentation | |
| 9.3.6 Nitroxide Mediated Polymerization | |
| 9.3.6.1 Nitroxides | |
| 9.3.6.2 Initiation | |
| 9.3.6.3 Side reactions | |
| 9.3.6.4 Rate enhancement. | |
| 9.3.6.5 Monomers | |
| 9.3.6.5.1 Styrene, vinyl aromatics | |
| 9.3.6.5.2 Acrylates | 480 |
| 9.3.6.5.5 Wielnacrylates | 401 |
| 9.3.6.5.4 Diene monomers | |
| 9.3.7 Other Oxygen Centered Padical Mediated Polymeri | |
| 9.3.8 Nitrogen-Centered Radical Mediated Polymerization | 1 483 |
| 9.3.9 Metal Complex-Mediated Radical Polymerization | 484 |
| 9.4 Atom Transfor Dadical Polymerization | 186 |
| 9.4.1 Initiators | 488 |
| 9.4.1.1 Molecular weights and distributions | 400 400 |
| 9.4.1.2 Reverse ATRP | 491 |
| 9413 Initiator activity | 492 |
| 9.4.2 Catalysts | |
| 9.4.2.1 Copper complexes | |
| 9.4.2.2 Ruthenium complexes | |
| 9.4.2.3 Iron complexes | |

xiii

| 9.4.2.4 | Nickel complexes | 496 |
|----------|------------------------------------|-----|
| 9.4.3 M | onomers and Reaction Conditions | 497 |
| 9.4.3.1 | Solution polymerization | 497 |
| 9.4.3.2 | Heterogeneous polymerization | 497 |
| 9.5 Reve | ersible Chain Transfer | 498 |
| 9.5.1 M | olecular Weights and Distributions | |
| 9.5.2 M | acromonomer RAFT. | 501 |
| 9.5.3 Th | iocarbonylthio RAFT | 502 |
| 9.5.3.1 | Mechanism | 503 |
| 9.5.3.2 | RAFT agents | 505 |
| 9.5.3.3 | RAFT agent synthesis | 515 |
| 9.5.3.4 | Side Reactions | 517 |
| 9.5.3.5 | Reaction conditions | 518 |
| 9.5.3.6 | Heterogeneous polymerization | 520 |
| 9.5.4 Io | dine-Transfer Polymerization | 521 |
| 9.5.5 Te | Iluride-Mediated Polymerization | 522 |
| 9.5.6 St | bine-Mediated Polymerization | 524 |
| 9.6 Livi | ng Radical Copolymerization | |
| 9.6.1 Re | activity Ratios | |
| 9.6.2 Gr | adjent Copolymers | |
| 9.6.3 NI | ИР | |
| 9.6.4 A | Γ R Ρ | |
| 9.6.5 RA | ١FT | 529 |
| 9.7 End | -Functional Polymers | 531 |
| 9.7.1 NI | MP | |
| 9.7.1.1 | ω-Functionalization | |
| 9.7.1.2 | a-Functionalization | |
| 9.7.2 A | Г R Р | |
| 9.7.2.1 | ω-Functionalization | |
| 9.7.2.2 | α -Functionalization | 536 |
| 9.7.3 RA | λFΤ | 538 |
| 9.7.3.1 | ω-Functionalization | 538 |
| 9.7.3.2 | α-Functionalization | 539 |
| 9.8 Bloc | k Conolymers | |
| 9.8.1 Di | rect Diblock Synthesis | |
| 9.8.1.1 | NMP | |
| 9.8.1.2 | ATRP | |
| 9.8.1.3 | RAFT | 543 |
| 9.8.2 Tr | ansformation Reactions | |
| 9.8.2.1 | Second step NMP | 545 |
| 9.8.2.2 | Second step ATRP | |
| 9.8.2.3 | Second step RAFT | 546 |

xiv

XV

Index to Tables

| Table 2.1 | Carbon-Hydrogen and Heteroatom-Hydrogen Bond Dissociation | 15 |
|------------|---|-----|
| Energ | gies (D in KJ mol.) | 15 |
| Table 2.2 | Relative Rate Constants and Regiospecificities for Addition of | 17 |
| | als to Halo-Oletins | L / |
| Table 2.3 | Relative Rate Constants for Reactions of Radicals with Alkyl- | 10 |
| Subst | ituted Aerylate Esters | 18 |
| Table 2.4 | Hammett ρ and ρ' Parameters for Reactions of Radicals | |
| Table 2.5 | Specificity of Intramolecular Hydrogen Abstraction | 32 |
| Table 2.6 | Bond Dissociation Energies | 34 |
| Table 2.7 | Values of k_{td}/k_{tc} for the Cross-Reaction between Fluoromethyl | |
| and E | Ethyl Radicals at 25 °C | 41 |
| Table 2.8 | Values of k_{td}/k_{tc} for <i>t</i> -Butyl Radicals at 25 °C | 43 |
| Table 3.1 | Guide to Properties of Polymerization Initiators | 66 |
| Table 3.2 | Selected Kinetic Data for Decomposition of Azo-Compounds | 70 |
| Table 3.3 | Solvent Dependence of Rate Constants for AIBMe | |
| Deco | mposition | 73 |
| Table 3.4 | Zero-Conversion Initiator Efficiency for AIBMe under Various | |
| React | tion Conditions | 76 |
| Table 3.5 | Selected Kinetic Data for Decomposition of Peroxides | 80 |
| Table 3.6 | Kinetic Data for Reactions of Carbon-Centered Radicals | 114 |
| Table 3.7 | Selected Rate Data for Reactions of Oxygen-Centered Radicals | 119 |
| Table 3.8 | Specificity Observed in the Reactions of Oxygen-Centered | |
| Radio | als with Various Monomers at 60 °C | 120 |
| Table 3.9 | Kinetic Data for Reactions of t-Butoxy Radicals in Various | |
| Solve | ents | 124 |
| Table 3.10 |) Selected Rate Data for Reactions of Heteroatom-Centered | |
| Radio | cals | 131 |
| Table 3.1 | Radical Trapping Agents for Studying Initiation | 134 |
| Table 3.12 | 2 Application of MALDI-TOF or ESI Mass Spectrometry to | |
| Polvr | ners Prepared by Radical Polymerization | 144 |
| Table 3.13 | 3 Radical Polymerizations Performed with Initiators Labeled with | |
| Stabl | e Isotopes | 147 |
| Table 4.1 | Tacticities of Selected Homonolymers. | |
| Table 4.2 | Temperature Dependence of Head vs Tail Addition for Eluoro- | |
| olefir | Monomers | 181 |
| Table 4.3 | Microstructure of Poly(chloronrene) vs Temperature | 185 |
| 14010 1.0 | interocative of Forgemoroprone, to Temperature management | |

| Index | to | Та | b | es |
|-------|----|----|---|----|
|-------|----|----|---|----|

| Table 4.4 Ring Sizes Formed in Cyclopolymerization of Symmetrical 1,6- | 100 |
|---|-----|
| Table 4.5 Extent of Ping opening During Polymerizations of | 190 |
| 2 Mothylene 1.3 dioxolano and Related Species | 200 |
| Table 4.6 Extent of Ring-Opening During Polymerizations of | 200 |
| 4.Methylene-1 3-dioxolane and 2-Methylene-1 4-dioxane Derivatives | 203 |
| Table 4.7 Extent of Ring-Opening During Polymerizations of | |
| 2-Methylenetetrahydrofuran and Related Compounds | 204 |
| Table 4.8 Extent of Double Ring-Opening During Polymerization of | |
| Polycyclic Monomers | 207 |
| Table 4.9 Structures Formed by Backbiting in Ethylene Copolymerization | |
| Table 4.10 Heats of Polymerization for Selected Monomers | 215 |
| Table 4.11 Kinetic Parameters for Propagation in Selected Radical | |
| Polymerizations in Bulk Monomer | 219 |
| Table 4.12 Rate Constants and Arrhenius Parameters for Propagation of | |
| Monomers Compared with Rate Constants for Addition of Small | |
| Radicals | 221 |
| Table 5.1 Parameters Characterizing Chain Length Dependence of | |
| Termination Rate Coefficients in Radical Polymerization of Common | |
| Monomers | 247 |
| Table 5.2 Values of k_{td}/k_{tc} for Polystyryl Radical Model Systems | 254 |
| Table 5.3 Values of k_{td}/k_{tc} for Methacrylate Ester Model Systems | 255 |
| Table 5.4 Values of k_{td}/k_{te} for Reactions involving Cyanoisopropyl Radicals | 257 |
| Table 5.5 Determinations of k_{td}/k_{te} for MMA Polymerization | 261 |
| Table 5.6 Kinetic Data for Various Inhibitors with Some Common | |
| Monomers | 265 |
| Table 5.7 Absolute Rate Constants for the Reaction of Carbon-Centered | |
| Radicals with Some Common Inhibitors | 266 |
| Table 6.1 Chain Length Dependence of Transfer Constants (C _n) | 283 |
| Table 6.2 Transfer Constants (60 °C, bulk) for Thiols (RSH) with Various | |
| Monomers | 290 |
| Table 6.3 Transfer Constants for Disulfides (R-S-S-R) With Various | |
| Monomers | 292 |
| Table 6.4 Transfer Constants (80 °C, bulk) for Halocarbons with Various | |
| Monomers | 293 |
| Table 6.5 Transfer Constants (60 °C, bulk) for Selected Solvents and | |
| Additives with Various Monomers | 295 |
| Table 6.6 Transfer Constants for Vinyl Ethers at 60 °C | 299 |
| Table 6.7 Transfer Constants for Allyl Sulfides at 60 °C | 300 |
| Table 6.8 Transfer Constants for Allyl Sulfonates and Sulfoxides at 60 °C | 302 |
| Table 6.9 Transfer Constants for Allyl Halides, Phosphonates, Silanes and | |
| Stannanes at 60 °C | 303 |

xvii

Index to Tables

| Table 6.10 Transfer Constants for Allyl Peroxide and Related Transfer | |
|---|-------|
| Agents at 60 °C | 304 |
| Table 6.11 Transfer Constants for Macromonomers | 307 |
| Table 6.12 Transfer Constants for Thionoester and Related Transfer | |
| Agents at 60 °C | 309 |
| Table 6.13 Transfer Constants for Cobalt Complexes | 316 |
| Table 6.14 Transfer Constants to Monomer | 317 |
| Table 6.15 Transfer Constants to Polymer | 320 |
| Table 7.1 Reactivity Ratios for Some Common Monomer Pairs | 339 |
| Table 7.2 Relative Rates for Addition of Substituted Propyl Radicals to AN | |
| and S | .,345 |
| Table 7.3 Relative Rates for Addition of Substituted Methyl Radicals | |
| $(R^{3}R^{2}R^{1}C^{\bullet})$ to MMA and S at ~25 °C | 346 |
| Table 7.4 Rate Constants (295 K) for Addition of Substituted Propyl | |
| Radicals to (Meth)acrylate Esters | 347 |
| Table 7.5 Implicit Penultimate Model Reactivity Ratios | 348 |
| Table 7.6 List of Donor and Acceptor Monomers | 351 |
| Table 7.7 Q-e and Patterns Parameters for Some Common Monomers | 365 |
| Table 7.8 Identity of Chain End Units Involved in Radical-Radical | |
| Termination in MMA-S Copolymerization | 372 |
| Table 8.1 Solvent Effect on Homopropagation Rate Constants for VAc | |
| at 30°C | 427 |
| Table 8.2 Effect of Solvent on Tacticity of Poly(alkyl methacrylate) | |
| at -40 °C | 428 |
| Table 8.3 Effect of Amines on Tacticity of Poly(methacrylic acid) at 60 °C | 429 |
| Table 8.4 Solvent Dependence of Reactivity Ratios for MMA-MAA | |
| Copolymerization at 70°C | 430 |
| Table 8.5 Solvent Dependence of Penultimate Model Reactivity Ratios for | |
| S-AN Copolymerization at 60°C | 430 |
| Table 8.6 Effect of Lewis Acids on Tacticity of Polymers Formed in High | |
| Conversion Radical Polymerizations at 60 °C | 435 |
| Table 9.1 Five-Membered Ring Nitroxides for NMP | 473 |
| Table 9.2 Six-Membered Ring Nitroxides for NMP | 474 |
| Table 9.3 Open-Chain Nitroxides for NMP | 475 |
| Table 9.4 Seven-Membered Ring Nitroxides for NMP | 475 |
| Table 9.5 Structures of Ligands for Copper Based ATRP Catalysts | 494 |
| Table 9.6 Structures of Ruthenium Complexes Used as ATRP Catalysts | 495 |
| Table 9.7 Structures of Iron Complexes Used as ATRP Catalysts | 496 |
| Table 9.8 Structures of Nickel Complexes Used as ATRP Catalysts | 496 |
| Table 9.9 Block Copolymers Prepared by Macromonomer RAFT | |
| Polymerization | 502 |
| Table 9.10 Tertiary Dithiobenzoate RAFT Agents | 508 |
| Table 9.11 Other Aromatic Dithioester RAFT Agents | 509 |

xviii

Index to Tables

| Table 9.12 | Primary and Secondary Dithiobenzoate RAFT Agents | 510 |
|------------|---|-----|
| Table 9.13 | Bis-RAFT Agents | 511 |
| Table 9.14 | Dithioacctate and Dithiophenylacetate RAFT Agents | 511 |
| Table 9.15 | Symmetrical Trithiocarbonate RAFT Agents | 512 |
| Table 9.16 | Non-Symmetrical Trithiocarbonate RAFT Agents | 512 |
| Table 9.17 | Xanthate RAFT Agents | 513 |
| Table 9.18 | Dithiocarbamate RAFT Agents | 514 |
| Table 9.19 | Initiators for Telluride-Mediated Polymerization | 524 |
| Table 9.20 | Statistical/Gradient Copolymers Synthesized by NMP | 528 |
| Table 9.21 | Statistical/Gradient Copolymers Synthesized by ATRP | 529 |
| Table 9.22 | Statistical/Gradient Copolymers Synthesized by RAFT | |
| Polym | erization | 529 |
| Table 9.23 | Methods for End Group Transformation of Polymers Formed | |
| by NM | IP | 532 |
| Table 9.24 | Methods for End Group Transformation of Polymers Formed | |
| by AT. | RP by Addition or Addition-Fragmentation | 534 |
| Table 9.25 | End Group Transformations for Polymers Formed by ATRP | 535 |
| Table 9.26 | Methods for End Group Removal from Polymers Formed by | |
| RAFT | Polymerization | 539 |
| Table 9.27 | Diblock Copolymers Prepared by ATRP | 543 |
| Table 9.28 | Diblock Copolymers Prepared by RAFT Polymerization | 543 |
| Table 9.29 | Star Precursors for NMP | 550 |
| Table 9.30 | Star Precursors for ATRP | 550 |
| Table 9.31 | Star Precursors for RAFT Polymerization | 551 |
| | | |

Index to Figures

| Figure 1.1 Publication rate of papers on radical polymerization and on living, controlled or mediated radical polymerization for period 1975- 2002 based on SciEinder [™] search | 7 |
|---|---------|
| Figure 2.1 Transition state for methyl radical addition to ethylene | |
| Geometric parameters are from <i>ab initio</i> calculation with OCISD(T)/6- | |
| 31GT(d) basis set | 20 |
| Figure 2.2 Effect of polar factors on regiospecificity of radical addition | 22 |
| Figure 2.3 Relative rate constants for addition of alkyl radicals to | |
| fumarodinitrile (k_1) and methyl α -chloroacrylate (k_2) as a function of | |
| temperature | 25 |
| Figure 2.4 SOMO-HOMO and SOMO-LUMO orbital interaction diagrams | 27 |
| Figure 2.5 Schematic state correlation diagram for free radical addition to a | |
| carbon-carbon double bond showing configuration energies as a | |
| function of the reaction coordinate | 28 |
| Figure 2.6 Transition state for hydrogen atom abstraction | 29 |
| Figure 2.7 Predicted order of reactivity of X-II compounds | 30 |
| Figure 2.8 Preferred site of attack in hydrogen abstraction by various | |
| radicals | 32 |
| Figure 2.9 Relative reactivity per hydrogen atom of indicated site towards | |
| <i>t</i> -butoxy radicals | 33 |
| Figure 2.10 Dependence of abstraction:addition ratio on nucleophilicity for | |
| oxygen-centered radicals | 35 |
| Figure 2.11 Dependence of abstraction:addition ratio on nucleophilicity for | |
| carbon-centered radicals | 35 |
| Figure 2.12 Trend in k_{id}/k_{ic} for radicals (CH ₃) ₂ C(•)-X | 42 |
| Figure 2.13 Temperature dependence of k_{td}/k_{tc} values for <i>t</i> -butyl radicals | 10 |
| with dodecane or 3-methyl-3-pentanol as solvent | 43 |
| Figure 3.1 Temperature dependence of rate constants for reactions of | |
| cumyloxy radicals (a) β -scission to methyl radicals (b) abstraction | |
| from cumene and (c) addition to styrene. Data are an extrapolation | 54 |
| based on literature Armenius parameters | 30 |
| Figure 3.2 Jabionski diagram describing photoexcitation process | 39 |
| righter 5.5 Cumulative and instantaneous initiator efficiency (7) of AIBN as | 76 |
| Figure 2.4. Deletive regarizity of indicated site towards t bytem redicate for | 70 |
| rigure 5.4 Relative reactivity of indicated site towards <i>t</i> -butoxy fadicals for allul mothesized and allul condition | 122 |
| anyi memaciyiate anu anyi actylate | . 1 2 2 |

Index to Figures

| Figure 3.5 Relative reactivity of indicated site towards <i>t</i> -butoxy radicals for | |
|--|------|
| BMA | 123 |
| Figure 3.6 Relative rate constants for β -scission of <i>t</i> -alkoxy radicals at 60 °C | 124 |
| Figure 4.1 Representation of meso (m) and racemic (r) dyads with polymer chains | 160 |
| Eigure 4.2 Depresentation of mass (w) and meaning (w) directoresis of | 109 |
| Figure 4.2 Representation of meso (m) and racemic (r) diastereoisomers of | 170 |
| low molecular weight compounds | 170 |
| Figure 4.3 Representation of mrrrmr heptad | 170 |
| Figure 4.4 Dependence of K_{eq} on temperature for selected monomers | 214 |
| Figure 4.5 Experimental molecular weight distribution obtained by GPC | |
| and its first derivative with respect to chain length for PS prepared by | |
| PLP | 218 |
| Figure 5.1 (a) Number and (b) GPC distributions for two polymers both | |
| with $\overline{X}_{n} - 100$ | 241 |
| Figure 5.2 Dispersity(D) as a function of \overline{X}_n for polymers formed by (a) | |
| disproportionation or chain transfer and (b) combination | 242 |
| Figure 5.3 Conversion-time profile for bulk MMA polymerization at 50 °C | |
| with AIBN initiator illustrating the three conversion regimes | 243 |
| Figure 5.4 Chain length dependence of k^{ij} predicted by (a) the geometric | |
| mean or (b) the harmonic mean approximation | 246 |
| Figure 5.5. Chain length dependence of k^{ij} predicted by the Smoluchowski | |
| mean with $\alpha=0.5$ and $k=10^9$ and the geometric mean with $\alpha=0.2$ and | |
| $k = 10^8$; i and i are the lengths of the reacting chains | 248 |
| x_{to} = 10°, 1 and 1 are the lengths of the colculated limiting change. (lost 1.0%) | |
| alonged and 'top 200' alonged are graphed as a function of [T]/[M] | 205 |
| Slopes and top 20% slopes are graphed as a function of $[1]/[M]$ | 283 |
| Figure 7.1 Plot of the instantaneous copolymer composition (PA) vs | |
| monomer feed composition (<i>f</i> A) for the situation where (a) $r_A = r_B = 1.0$, | 0.40 |
| (b) $r_{\rm A} = r_{\rm B} = 0.5$, (c) $r_{\rm A} = r_{\rm B} = 0.01$, (d) $r_{\rm A} = 0.5$, $r_{\rm B} = 2.0$ | 340 |
| Figure 7.2 Chain end terminology | 344 |
| Figure 7.3 Molecular weight distributions for HEA:BA:S copolymer | |
| prepared with butanethiol chain transfer agent | 382 |
| Figure 7.4 Molecular weight distributions for HEA:BA:S copolymer | |
| prepared with butanethiol chain transfer agent | 383 |
| Figure 9.1 Predicted evolution of molecular weight with monomer | |
| conversion for a conventional radical polymerization with constant rate | |
| of initiation and a living polymerization | 453 |
| Figure 9.2 (a) Number and (b) GPC distributions for three polymers each | |
| with $\overline{X}_{n}=100$. | 454 |
| Figure 9.3 General description of macromonomer and thiocarbonylthio | |
| RAFT agents. | 501 |
| Figure 9.4 Effect of Z substituent on effectiveness of RAFT agents | 505 |
| Figure 9.5. Canonical forms of thiocarbonylthio compounds | 506 |
| x Seve > 10 - Sumonious round or anoour bonyruno compounds mananinini | |

Index to Figures

| Figure 9.6 Effect of R substituent on effectiveness of RAFT agents | 507 |
|---|-----|
| Figure 9.7 Evolution of molecular weight and dispersity with conversion | |
| for MMA polymerizations in the presence of RAFT agent | 519 |
| Figure 9.8 Comparison of molecular weight distributions for a | |
| conventional and RAFT polymerization | 520 |
| Figure 9.9 General description of organochalcogenide transfer agents | 524 |
| Figure 9.10 General description of organostibine transfer agents | 525 |
| Figure 9.11 Star Architectures | 548 |
| Figure 9.12 GPC distributions obtained during bulk thermal polymerization | |
| of styrene at 110 °C with tetrafunctional RAFT agents | 553 |

xxii

Preface to the First Edition

In recent years, the study of radical polymerization has gone through something of a renaissance. This has seen significant changes in our understanding of the area and has led to major advances in our ability to control and predict the outcome of polymerization processes. Two major factors may be judged responsible for bringing this about and for spurring an intensified interest in all aspects of radical chemistry:

Firstly, the classical theories on radical reactivity and polymerization mechanism do not adequately explain the rate and specificity of simple radical reactions. As a consequence, they can not be used to predict the manner in which polymerization rate parameters and details of polymer microstructure depend on reaction conditions, conversion and molecular weight distribution.

Secondly, new techniques have been developed which allow a more detailed characterization of both polymer microstructures and the kinetics and mechanism of polymerizations. This has allowed mechanism-structure-property relationships to be more rigorously established.

The new knowledge and understanding of radical processes has resulted in new polymer structures and in new routes to established materials; many with commercial significance. For example, radical polymerization is now used in the production of block copolymers, narrow polydispersity homopolymers, and other materials of controlled architecture that were previously available only by more demanding routes. These commercial developments have added to the resurgence of studies on radical polymerization.

We believe it is now timely to review the recent developments in radical polymerization placing particular emphasis on the organic and physical-organic chemistry of the polymerization process. In this book we critically evaluate the findings of the last few years, where necessary reinterpreting earlier work in the light of these ideas, and point to the areas where current and future research is being directed. The overall aim is to provide a framework for further extending our understanding of free radical polymerization and create a definable link between synthesis conditions and polymer structure and properties. The end result should be polymers with predictable and reproducible properties.

The book commences with a general introduction outlining the basic concepts. This is followed by a chapter on radical reactions that is intended to lay the theoretical ground-work for the succeeding chapters on initiation, propagation, and termination. Because of its importance, radical copolymerization is treated in a separate chapter. We then consider some of the implications of these chapters by

Preface to the First Edition

discussing the prospects for controlling the polymerization process and structureproperty relationships. In each chapter we describe some of the techniques that have been employed to characterize polymers and polymerizations and which have led to breakthroughs in our understanding of radical polymerization. Emphasis is placed on recent developments.

This book will be of major interest to researchers in industry and in academic institutions as a reference source on the factors which control radical polymerization and as an aid in designing polymer syntheses. It is also intended to serve as a text for graduate students in the broad area of polymer chemistry. The book places an emphasis on reaction mechanisms and the organic chemistry of polymerization. It also ties in developments in polymerization kinetics and physical chemistry of the systems to provide a complete picture of this most important subject.

Graeme Moad David H Solomon

xxiv

Preface to the Second Edition

In the ten years since the first edition appeared, the *renaissance* in Radical Polymerization has continued and gained momentum. The period has seen the literature with respect to controlled and, in particular, living radical polymerization expand dramatically. The end of 1995, saw the first reports on atom transfer radical polymerization (ATRP) and in 1998 polymerization with reversible addition fragmentation chain transfer (RAFT) was introduced. The period has also seen substantial development in nitroxide-mediated polymerization (NMP) first reported in 1987 and discussed in the first edition. A new generation of control agents has added greater versatility and new applications. The area of living radical polymerization is now responsible for a very substantial fraction of the papers in the field. In this edition, we devote a new chapter to living radical polymerization.

The initial thrust of work in the area of living radical polymerization was aimed at capitalizing on the versatility of radical polymerization with respect to reaction conditions and the greater range of suitable monomers as compared to anionic systems. Anionic polymerizations were seen as the standard. This has now changed, and living radical polymerizations are now seen as offering polymers with unique compositions and properties not achievable with other methodologies. Living radical polymerization has also been combined with other processes and mechanisms to give structures and architectures that were not previously thought possible. The developments have many applications particularly in the emerging areas of electronics, biotechnology and nanotechnology.

A small change has been made to the title and the text of this edition to reflect the current IUPAC recommendation that radicals are no longer 'free'. Of the classical steps of a radical polymerization, while there remains some room for improvement, it can be stated that we now have methodologies that give control over the termination and initiation steps to the extent that specific structures, molecular weight distributions, and architectures can be confidently obtained. The remaining 'holy grail' in the field of radical polymerization is control over the stereochemistry and regiospecificity in the propagation step. Although some small steps have been taken towards achieving this goal, much remains to be done.

The last ten years have also seen significant advances in other areas of radical polymerization. Chapters one through eight have been updated and many new references added to reflect these developments.

Graeme Moad David H Solomon

Acknowledgments

We gratefully acknowledge the contribution of the following for their assistance in the preparation and proof reading of the manuscript.

Dr Agnes Ho Dr Catherine L. Moad Dr Almar Postma Dr Greg Qiao Dr Tiziana Russo

In addition, we thank again those who contributed to the production of the first edition.

We also thank Max McMaster of McMaster Indexing for his efforts in producing the index for this volume.

1 Introduction

From an industrial stand-point, a major virtue of radical polymerizations is that they can often be carried out under relatively undemanding conditions. In marked contrast to ionic or coordination polymerizations, they exhibit a tolerance of trace impurities. A consequence of this is that high molecular weight polymers can often be produced without removal of the stabilizers present in commercial monomers, in the presence of trace amounts of oxygen, or in solvents that have not been rigorously dried or purified. Indeed, radical polymerizations are remarkable amongst chain polymerization processes in that they can be conveniently conducted in aqueous media.

It is this apparent simplicity of radical polymerization that has led to the technique being widely adopted for both industrial and laboratory scale polymer syntheses. Today, a vast amount of commercial polymer production involves radical chemistry during some stage of the synthesis, or during subsequent processing steps. These factors have, in turn, provided the driving force for extensive research efforts directed towards more precisely defining the kinetics and mechanisms of radical polymerizations. The aim of these studies has been to define the parameters necessary for predictable and reproducible polymer syntheses and to give better understanding of the properties of the polymeric materials produced. With understanding comes control. Most recently, we have seen radical polymerization move into new fields of endeavor where control and precision are paramount requirements. Indeed, these aspects now dominate the literature.

The history of polymers, including the beginnings of addition and of radical polymerization, is recounted by Morawetz.¹ The repeat unit structure (1) of many common polymers, including PS, PVC and PVAc, was established in the latter half of the 19^{th} century. However, the concept that these were materials of high molecular weight took longer to be accepted. Staudinger was one of the earliest and most strident proponents of the notion that synthetic polymers were high molecular weight compounds with a chain structure and he did much to dispel the then prevalent belief that polymers were composed of small molecules held together by colloidal forces.² Staudinger and his colleagues are also often credited with coming up with the concept of a chain polymerization. In an early paper in 1920, he proposed that polymer chains might retain unsatisfied valencies at the chain ends (2).³ In 1929, it was suggested that the monomer units might be

The Chemistry of Free Radical Polymerization

connected by covalent linkages in large cyclic structures (3) to solve the chain end problem.⁴ In 1910, Pickles⁵ had proposed such a structure for natural rubber. However, by 1935 it was recognized that polymers have discrete functional groups at the chain ends formed by initiation and termination reactions.⁶



In the period 1910-1950 many contributed to the development of free-radical polymerization.¹ The basic mechanism as we know it today (Scheme 1.1), was laid out in the 1940s and 50s.⁷⁻⁹ The essential features of this mechanism are initiation and propagation steps, which involve radicals adding to the less substituted end of the double bond ("tail addition"), and a termination step, which involves disproportionation or combination between two growing chains.

In this early work, both initiation and termination were seen to lead to formation of structural units different from those that make up the bulk of the chain. However, the quantity of these groups, when expressed as a weight fraction of the total material, appeared insignificant. In a polymer of molecular weight 100,000 they represent only ca 0.2% of units.[†] Thus, polymers formed by radical polymerization came to be represented by, and their physical properties and chemistry interpreted in terms of, the simple formula 1.

However, it is now quite apparent that the representation 1 while convenient, and useful as a starting point for discussion, has serious limitations when it comes to understanding the detailed chemistry of polymeric materials. For example, how can we rationalize the finding that two polymers with nominally the same chemical and physical composition have markedly different thermal stability? PMMA (1, $X=CH_3$, $Y=CO_2CH_3$) prepared by anionic polymerization has been reported to be more stable by some 50 °C than that prepared by a radical process.¹⁰ The simplified representation, (1), also provides no ready explanation for the discrepancy in chemical properties between low molecular weight model compounds and polymers even though both can be represented ostensibly by the same structure (1). Consideration of the properties of simple models indicates that the onset of thermal degradation of PVC (1, X=H, Y=CI) should occur at a temperature 100 °C higher than is actually found.¹¹

^{*} IUPAC recommendations suggest that polymers derived from 1,1-disubstituted monomers CXY=CH₂ (or CH₂=CXY) be drawn as 1b rather than as 1a. However, formula 1a follows logically from the traditional way of writing the mechanism of radical addition (*e.g.* Scheme 1.1). Because of our focus on mechanism, the style 1a has been adopted throughout this book.

Based on a monomer molecular weight of 100.

Introduction

Such problems have led to a recognition of the importance of defect groups^{*} or structural irregularities.¹²⁻¹⁶ If we are to achieve an understanding of radical polymerization, and the ability to produce polymers with optimal, or at least predictable, properties, a much more detailed knowledge of the mechanism of the polymerization and of the chemical microstructure of the polymers formed is required.¹⁶



Scheme 1.1

Structural irregularities are introduced into the chain during each stage of the polymerization and we must always question whether it is appropriate to use the generalized formula (1) for representing the polymer structure. Obvious examples of defect structures are the groups formed by chain initiation and termination. Initiating radicals[†] are not only formed directly from initiator decomposition (Scheme 1.1) but also indirectly by transfer to monomer, solvent, transfer agent, or impurities (Scheme 1.2).

^{* &#}x27;Defect groups' or 'structural irregularities' need not impair polymer properties, they are simply units that differ from those described by the generalized formula 1

⁺ Initiating radicals are formed from those initiator- or transfer agent-derived radicals that add monomer so as to form propagating radicals (see 3.1).

The Chemistry of Free Radical Polymerization

In termination, unsaturated and saturated ends are formed when the propagating species undergo disproportionation, head-to-head linkages when they combine, and other functional groups may be introduced by reactions with inhibitors or transfer agents (Scheme 1.2). In-chain defect structures (within the polymer molecule) can also arise by copolymerization of the unsaturated byproducts of initiation or termination.



Scheme 1.2

The generalized structure (1) also overestimates the homogeneity of the repeat units (the specificity of propagation). The traditional explanation offered to rationalize structure 1, which implies exclusive formation of head-to-tail linkages in the propagation step, is that the reaction is under thermodynamic control. This explanation was based on the observation that additions of simple radicals to mono- or 1,1-disubstituted olefins typically proceed by tail addition to give secondary or tertiary radicals respectively rather than the less stable primary radical (Scheme 1.3) and by analogy with findings for ionic reactions where such thermodynamic considerations are of demonstrable importance.



Scheme 1.3

Until the early 1970s, the absence of suitable techniques for probing the detailed microstructure of polymers or for examining the selectivity and rates of radical reactions prevented the traditional view from being seriously questioned. In more recent times, it has been established that radical reactions, more often than not, are under kinetic rather than thermodynamic control and the preponderance of

Introduction

head-to-tail linkages in polymers is determined largely by steric and polar influences (see 2.2).¹⁷

It is now known that a proportion of "head" addition occurs during the initiation and propagation stages of many polymerizations (see 4.3). For example, poly(vinyl fluoride) chains contain in excess of 10% head-to-head linkages.¹⁸ Benzoyloxy radicals give *ca* 5% head addition with styrene (see 3.4.2.2).^{19,20} However, one of the first clear-cut examples demonstrating that thermodynamic control is not of overriding importance in determining the outcome of radical reactions is the cyclopolymerization of diallyl compounds (see 4.4.1).²¹⁻²⁴

Monomers containing multiple double bonds might be anticipated to initially yield polymers with pendant unsaturation and ultimately crosslinked structures. The pioneering studies of Butler and coworkers^{23,24} established that diallyl compounds, of general structure (4), undergo radical polymerization to give linear saturated polymers. They proposed that the propagation involved a series of interand intramolecular addition reactions. The presence of cyclic units in the polymer structure was rigorously established by chemical analysis.²⁵ Addition of a radical to the diallyl monomer (4) could conceivably lead to the formation of 5-, 6- or even 7-membered rings as shown in Scheme 1.4. However, application of the then generally accepted hypothesis, that product radical stability was the most important factor determining the course of radical addition, indicated that the intermolecular step should afford a 6-membered ring and a secondary radical (7). On the basis of this theory, it was proposed that the cyclopolymer was composed of 6-membered rings (9) rather than 5-membered rings (8).



Scheme 1.4

It was established in the early 1960s that hexenyl radicals and simple derivatives gave 1,5- rather than 1,6-ring closure under conditions of kinetic

The Chemistry of Free Radical Polymerization

control.²⁶ However, it was not until 1976 that the structures of cyclopolymers formed from 1,6-dienes (4) were experimentally determined and Hawthorne *et al.*²⁷ showed that the intramolecular cyclization step gives preferentially the less stable radical (6) (5- *vs* 6-membered ring, primary *vs* secondary radical) - *i.e.* \geq 99% head addition. Over the last two decades, many other examples of radical reactions which preferentially afford the thermodynamically less stable product have come to light. A discussion of various factors important in determining the course and rate of radical additions will be found in Chapter 2.

The examples described in this chapter serve to illustrate two well-recognized, though often overlooked, principles, which lie at the heart of polymer, and, indeed, all forms of chemistry. These are:

- (a) The dependence of a reaction (polymerization, polymer degradation, etc.) on experimental variables cannot be understood until the reaction mechanism is established.
- (b) The reaction mechanism cannot be fully defined, when the reaction products are unknown.

The recent development of radical polymerizations that show the attributes of living polymerization is a prime example of where the quest for knowledge on polymerization mechanism can take us (Chapter 9). Living radical polymerization relies on the introduction of a reagent that undergoes reversible termination with the propagating radicals thereby converting them to a dormant form (Scheme 1.5). This enables control of the active species concentration allowing conditions to be chosen such that all chains are able to grow at a similar rate (if not simultaneously) throughout the polymerization. This has, in turn, enabled the synthesis of polymers with low dispersity and a wide variety of block, stars and other structures not hitherto accessible by any mechanism. Specificity in the reversible initiation-termination step is of critical importance in achieving living characteristics.



Scheme 1.5

The first steps towards living radical polymerization were taken by Otsu and colleagues^{28,29} in 1982. In 1985, this was taken one step further with the development by Solomon *et al.*³⁰ of nitroxide-mediated polymerization (NMP). This work was first reported in the patent literature³⁰ and in conference papers but was not widely recognized until 1993 when Georges *et al.*³¹ applied the method in

Introduction

the synthesis of narrow polydispersity polystyrene. NMP was described in detail in a small section in the first edition of this book. Since that time the area has expanded dramatically. The scope of NMP has been greatly extended³² and new, more versatile, methods have appeared. The most notable are atom transfer radical polymerization (ATRP)^{33,34} and polymerization with reversible addition fragmentation (RAFT).^{35,36} From small beginnings pre-1995, this area now accounts for a third of all papers in the field of radical polymerization. Moreover, the growth in the field since 1995 is almost totally attributable to developments in this area (Figure 1.1).



Figure 1.1 Publication rate of journal papers on radical polymerization and on living, controlled or mediated radical polymerization^{*} for period 1975-2002 based on SciFinder[™] search (as of Mar 2005).

In the succeeding chapters we detail the current state of knowledge of the chemistry of each stage of polymerization. We consider the details of the mechanisms, the specificity of the reactions, the nature of the group or groups incorporated in the polymer chain, and any byproducts. The intention is to create an awareness of the factors that must be borne in mind in selecting the conditions for a given polymerization and provide the background necessary for a more thorough understanding of polymerizations and polymer properties. In the final chapters, we examine the current status of efforts to control polymerization using either conventional technology or using the various approaches to living radical polymerization.

^{*} Does not distinguish forms of controlled radical polymerization. Includes most papers on ATRP, RAFT and NMP and would also include conventional (non-living) but controlled radical polymerizations. It would not include papers, which do not mention the terms 'living', 'controlled' or 'mediated'.

The Chemistry of Free Radical Polymerization

1.1 References

- 1. Morawetz, II. *Polymers. The Origins and Growth of a Science*; Dover: New York, 1995.
- 2. Furukawa, Y. Inventing Polymer Science: Staudinger, Carothers, and the Emergence of Macromolecular Chemistry; University of Pennsylvania Press: Philadelphia, 1998.
- 3. Staudinger, H. Chem. Ber. 1920, 53, 1073.
- 4. Staudinger, H.; Signer, R.; Johner, H.; Lüthy, M.; Kern, W.; Russidis, D.; Schwetzer, O. Ann. 1929, 474, 145.
- 5. Pickles, S.S. J. Chem. Soc. 1910, 97, 1085.
- 6. Staudinger, H.; Steinhofer, A. Ann. 1935, 517, 35.
- 7. Flory, P.J. *Principles of Polymer Chemistry*; Cornell University Press: Ithaca, New York, 1953.
- 8. Walling, C. Free Radicals in Solution; Wiley: New York, 1957.
- Bamford, C.II.; Barb, W.G.; Jenkins, A.D.; Onyon, P.F. *The Kinetics of Vinyl* Polymerization by Radical Mechanisms; Butterworths: London, 1958.
- 10. McNeill, I.C. Eur. Polym. J. 1968, 4, 21.
- 11. Mayer, Z. J. Macromol. Sci., Rev. Macromol. Chem. 1974, C10, 263.
- 12. Solomon, D.H.; Cacioli, P.; Moad, G. Pure Appl. Chem. 1985, 57, 985.
- 13. Hwang, E.F.J.; Pearce, E.M. Polym. Eng. Rev. 1983, 2, 319.
- 14. Mita, I. In *Aspects of Degradation and Stabilization of Polymers*; Jellineck, H.H.G., Ed.; Elsevier: Amsterdam, 1978; p 247.
- 15. Solomon, D.H. J. Macromol. Sci., Chem. 1982, A17, 337.
- 16. Moad, G.; Solomon, D.H. Aust. J. Chem. 1990, 43, 215.
- 17. Tedder, J.M. Angew. Chem., Int. Ed. Engl. 1982, 21, 401.
- 18. Cais, R.E.; Kometani, J.M. ACS Symp. Ser. 1984, 247, 153.
- 19. Moad, G.; Rizzardo, E.; Solomon, D.H. Macromolecules 1982, 15, 909.
- 20. Moad, G.; Rizzardo, E.; Solomon, D.H.; Johns, S.R.; Willing, R.I. Makromol. Chem., Rapid Commun. 1984, 5, 793.
- 21. Butler, G.B. Acc. Chem. Res. 1982, 15, 370.
- 22. Solomon, D.H.; Hawthorne, D.G. J. Macromol. Sci., Rev. Macromol. Chem. 1976, C15, 143.
- Butler, G.B. In *Encyclopedia of Polymer Science and Engineering*, 2nd ed.; Mark, II.F.; Bikales, N.M.; Overberger, C.G.; Menges, G., Eds.; Wiley: New York, 1986; Vol. 4, p 543.
- 24. Butler, G.B. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 4, p 423.
- 25. Butler, G.B.; Crawshaw, A.; Miller, W.L. J. Am. Chem. Soc. 1958, 80, 3615.
- 26. Beckwith, A.L.J.; Ingold, K.U. In *Rearrangements in Ground and Excited States*; de Mayo, P., Ed.; Academic Press: New York, 1980; Vol. 1, p 162.
- Hawthorne, D.G.; Johns, S.R.; Solomon, D.H.; Willing, R.I. Aust. J. Chem. 1976, 29, 1955.
- 28. Otsu, T.; Yoshida, M. Makromol. Chem., Rapid Commun. 1982, 3, 127.
- 29. Otsu, T.; Yoshida, M.; Tazaki, T. Makromol. Chem., Rapid Commun. 1982, 3, 133.
- 30. Solomon, D.H.; Rizzardo, E.; Cacioli, P. US 4581429, 1986 (Chem. Abstr. 1985, 102, 221335q).
- Georges, M.K.; Veregin, R.P.N.; Kazmaier, P.M.; Hamer, G.K. Macromolecules 1993, 26, 2987.
- 32. Hawker, C.J.; Bosman, A.W.; Harth, E. Chem. Rev. 2001, 101, 3661.

8

Introduction

- 33. Matyjaszewski, K.; Xia, J. Chem. Rev. 2001, 101, 2921.
- 34. Kamigaito, M.; Ando, T.; Sawamoto, M. Chem. Rev. 2001, 101, 3689.
- Chiefari, J.; Chong, Y.K.; Ercole, F.; Krstina, J.; Jeffery, J.; Le, T.P.T.; Mayadunne, R.T.A.; Meijs, G.F.; Moad, C.L.; Moad, G.; Rizzardo, E.; Thang, S.H. Macromolecules 1998, 31, 5559.
- 36. Moad, G.; Rizzardo, E.; Thang, S. Aust. J. Chem. 2005, 58, 379.
This Page is Intentionally Left Blank

1.1 Introduction

The intention of this chapter is to discuss in some detail the factors that determine the rate and course of radical reactions. Emphasis is placed on those reactions most frequently encountered in radical polymerization:

(a) Addition to carbon-carbon double bonds (e.g. initiation - Chapter 3, propagation - Chapter 4).

$$\mathbf{x} \cdot \mathbf{y} = \left\{ \begin{array}{c} \mathbf{x}_{\mathrm{T}} \text{ or } \mathbf{x}_{\mathrm{H}} \\ \hline \end{array} \right. \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{y} = \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} \cdot \mathbf{x} + \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{y} \right\} \cdot \mathbf{x} + \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{x} + \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{x} \right\} \cdot \mathbf{x} + \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{x} + \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{x} \right\} \cdot \mathbf{x} + \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{x} + \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{x} \right\} \cdot \mathbf{x} + \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{x} + \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{x} + \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{x} \right\} \right\} \cdot \mathbf{x} + \left\{ \mathbf{x}_{\mathrm{T}} \mathbf{x}$$

(a) The self-reaction of carbon-centered radicals (e.g. termination - Chapter 5).



(a) Hydrogen atom transfer (e.g. chain transfer - Chapter 6).

Other radical reactions not covered in this chapter are mentioned in the chapters that follow. These include additions to systems other than carbon-carbon double bonds [*e.g.* additions to aromatic systems (Section 3.4.2.2.1) and strained ring systems (Section 4.4.2)], transfer of heteroatoms [*e.g.* chain transfer to disulfides (Section 6.2.2.2) and halocarbons (Section 6.2.2.4)] or groups of atoms [*e.g.* in RAFT polymerization (Section 9.5.3)], and radical-radical reactions involving heteroatom-centered radicals or metal complexes [*e.g.* in inhibition (Sections 3.5.2 and 5.3), NMP (Section 9.3.6) and ATRP (Section 9.4)].

Until the early 1970s, views of radical reactions were dominated by two seemingly contradictory beliefs: (a) that radical reactions, in that they involve highly reactive species, should not be expected to show any particular selectivity, and (b) that (as is often possible with ionic reactions) the outcome could be predicted purely on the basis of the relative thermochemical stability of the product radicals. For condition (a) to apply, a reaction should have an early reactant-like transition state and near-zero activation energy. For condition (b) to apply the transition state should be late (or product-like) or the reaction leading to products must be under thermodynamic control by virtue of being rapidly reversible. While either of the above conditions may apply in specific cases, for radical reactions in general, neither need apply.

It is now recognized that radical reactions are, more often than not, under kinetic rather than thermodynamic control. The reactions can nonetheless show a high degree of specificity which is imposed by steric (non-bonded interactions), polar (relative electronegativities), stereoelectronic (requirement for overlap of frontier orbitals), bond-strength (relative strengths of bonds formed and broken) and perhaps other constraints.¹⁻⁴ In the following sections we discuss these factors, consider their relative importance in specific reactions and suggest guidelines for predicting the outcome of radical reactions.

1.1 Properties of Radicals

Radicals are chemical species that possess an unpaired electron sometimes called a free spin. The adjective "free", often used to designate radicals, relates to the state of the unpaired electron; it is not intended to indicate whether the compound bearing the free spin is complexed or uncomplexed. In this section we provide a brief overview of the structure, energetics and detection of radicals.

1.1.1 Structures of Radicals

Most radicals located on saturated bonds are π -radicals with a planar configuration and may be depicted with the free spin located in a *p*-orbital (1). Because such radical centers are achiral, stereochemical integrity is lost during radical formation. A new configuration will be assumed (or a previous configuration resumed) only upon reaction. Stereoselectivity in radical reactions is therefore dependent on the environment and on remote substituents.



Radicals with very polar substituents (*e.g.* trifluoromethyl radical 2), and radicals that are part of strained ring systems (*e.g.* cyclopropyl radical 3) are σ -radicals. They have a pyramidal structure and are depicted with the free spin resident in an sp^3 hybrid orbital. σ -Radicals with appropriate substitution are potentially chiral, however, barriers to inversion are typically low with respect to the activation energy for reaction.

Most radicals located on double bonds (*e.g.* 4, 5) or aromatic systems (*e.g.* 6) are σ -radicals. The free spin is located in an orbital orthogonal to the π -bond system and it is not delocalized. The orbital of the vinyl radical (4) containing the free spin can be *cis*- or *trans*- with respect to substituents on the double bond. The barrier for isomerization of vinyl radicals can be significant with respect to the rate of reaction.



Radicals with adjacent π -bonds [*e.g.* allyl radicals (7), cyclohexadienyl radicals (8), acyl radicals (9) and cyanoalkyl radicals (10)] have a delocalized structure. They may be depicted as a hybrid of several resonance forms. In a chemical reaction they may, in principle, react through any of the sites on which the spin can be located. The preferred site of reaction is dictated by spin density, steric, polar and perhaps other factors. Maximum orbital overlap requires that the atoms contained in the delocalized system are coplanar.



Radicals with adjacent heteroatoms bearing lone pairs (N, O, Cl, *etc.*), *e.g.* **11**, **12** can also be depicted as a resonance hybrid involving charged structures. The free spin may also be delocalized into adjacent C-H and C-C single bonds through a phenomenon known as hyperconjugation. Maximal hyperconjugative interaction requires coplanarity of the *p*-orbital containing the unpaired electron and the C-H and C-C bonds. Hyperconjugation is used to rationalize the relative stability and the nucleophilicity of alkyl radicals (tertiary > secondary > primary).

1.1.2 Stabilities of Radicals

Most radicals are transient species. They (e.g. 1-10) decay by self-reaction with rates at or close to the diffusion-controlled limit (Section 1.4). This situation also pertains in conventional radical polymerization. Certain radicals, however, have thermodynamic stability, kinetic stability (persistence) or both that is conferred by appropriate substitution. Some well-known examples of stable radicals are diphenylpicrylhydrazyl (DPPH), nitroxides such as 2,2,6,6tetramethylpiperidin-N-oxyl (TEMPO), triphenylmethyl radical (13) and galvinoxyl (14). Some examples of carbon-centered radicals which are persistent but which do not have intrinsic thermodynamic stability are shown in Section 1.4.3.2. These radicals (DPPH, TEMPO, 13, 14) are comparatively stable in isolation as solids or in solution and either do not react or react very slowly with compounds usually thought of as substrates for radical reactions. They may, nonetheless, react with less stable radicals at close to diffusion controlled rates. In polymer synthesis these species find use as inhibitors (to stabilize monomers against polymerization or to quench radical reactions - Section 5.3.1) and as reversible termination agents (in living radical polymerization - Section 9.3).



Hydrogen-other atom/group bond dissociation energies are often used as an indication of radical stability. Substitution at a radical center almost invariably increases stability as indicated by a reduced bond dissociation energy. Thus, for alkyl radicals, stability increases in the order primary<secondary<tertiary. Fluorine substitution provides the exception to this rule. Radicals are inductively destabilized by fluorine substituents α - or β - to the radical center.⁵ The greatest stabilizing effect is observed with substituents that are able to delocalize the free spin (Ph, CN, C=C). Experimental gas phase bond dissociation energies are tabulated in Table 1.1.⁶ Bond dissociation energies can often be estimated with reasonable accuracy using group additivity rules.⁷

While it is desirable and important to have some knowledge of radical stabilities, the following sections will show that this is only one, and often not the major, factor in determining the outcome of radical reactions.

1.1.3 Detection of Radicals

In radical polymerization and in most radical reactions the radical species are present only in low concentrations (total concentration $\sim 10^{-8}$ - 10^{-7} M). Radicals are

14

either generated in a chain reaction in which the radical species attain a low steady state concentration or they are generated reversibly and their concentration is controlled by an equilibrium process.

Largely for these reasons, radicals are most often characterized indirectly by examining the products of their reaction. Many of the methods used to study radical reactions have been applied to study initiation of polymerization. Some of these techniques are detailed in Section 3.5.

| · · · · | | · · · | |
|--|-----|--|-----|
| C-H Bond | D | X-H Bond | D |
| CF ₃ -H | 450 | HO-H | 497 |
| CH ₃ -H | 439 | CH ₃ C(=O)O-H | 442 |
| C ₂ H ₅ -H | 423 | CH ₃ O-H | 436 |
| <i>i</i> -С ₃ Н ₇ -Н | 409 | (CH ₃) ₃ CO-H | 440 |
| <i>t-</i> C₄H ₉ -H | 404 | (CH ₃) ₃ COO-H | 374 |
| HOCH ₂ -H | 402 | HOO-H | 369 |
| H(C=O)CH ₂ -H | 394 | PhO-H | 362 |
| CH ₂ (CN)-H | 393 | | |
| CCl ₃ -H | 393 | CH₃S-H | 365 |
| PhCH ₂ -H | 376 | PhS-H | 349 |
| CH ₂ =CHCH ₂ -H | 362 | PhSe-H | 326 |
| (CH ₃) ₂ C(CN)-H | 362 | | |
| CH ₃ CH(Ph)-H | 357 | NH ₂ -H | 453 |
| $(CH_3)_2C(Ph)-H$ | 353 | CH ₃ NH-H | 418 |
| | | PhNH-H | 368 |
| CH=C-H | 556 | NH ₂ NH-H | 366 |
| Ph-H | 473 | | |
| CH ₂ =CH-H | 465 | (CH ₃) ₃ Si-H | 378 |
| $cC_{3}H_{5}$ -H | 445 | (CH ₃) ₃ Ge-H | 339 |
| O=CH-H | 369 | (C ₄ H ₉) ₃ Sn-H | 308 |

Table 1.1 Carbon-Hydrogen and Heteroatom-Hydrogen Bond DissociationEnergies $(D \text{ in } kJ \text{ mol}^{-1})^{a,6}$

a All values rounded to the nearest integer.

Electron paramagnetic resonance spectroscopy (EPR), also called electron spin resonance spectroscopy (ESR), may be used for direct detection and conformational and structural characterization of paramagnetic species. Good introductions to EPR have been provided by Fischer⁸ and Leffler⁹ and most books on radical chemistry have a section on EPR. EPR detection limits are dependent on radical structure and the signal complexity. However, with modern instrumentation, radical concentrations >10⁻⁹ M can be detected and concentrations >10⁻⁷ M can be reliably quantified.

UV-visible spectrophotometry and fluorescence spectrophotometry are also used for the direct observation of radical species and their reactions in some

circumstances. Radical species typically absorb at significantly higher wavelengths than similar saturated compounds (bathochromic shift).

Molecular orbital calculations (*ab initio* or semiempirical methods) are also often used to provide a description of radical species and their reactions. High levels of theory are required to provide reliable data. However, rapid advances in computer power and computational methods are seeing these methods more widely used and with greater success (for leading references on the application of theory to describe radical addition reactions, see Section 1.2.7).

1.2 Addition to Carbon-Carbon Double Bonds

With few exceptions, radicals are observed to add preferentially to the less highly substituted end of unsymmetrically substituted olefins (*i.e.* give predominantly tail addition^{*} - Scheme 1.1).

For a long time, this finding was correlated with the observation that substituents at a radical center tend to enhance its stability (Section 1.1.2). This in turn led to the belief that the degree of stabilization conferred on the product radical by the substituents was the prime factor determining the orientation and rate of radical addition to olefins. That steric, polar, or other factors might favor the same outcome was either considered to be of secondary importance or simply ignored.^{*}



Indeed, while alternative hypotheses were entertained by some,¹⁰ there was no serious questioning of the dominant role of thermochemistry in the wider community until the 1970s. Many factors were important in bringing about this change in thinking. Three of the more significant were:

(a) A few isolated examples appeared where "wrong way" addition (formation of the less thermodynamically stable radical) was a significant, or even the major, pathway. Notable examples are predominantly head addition in the intramolecular step of cyclopolymerization of 1,6-dienes (Scheme 1.2)¹¹ and in the reaction of *t*-butoxy radicals with difluoroethylene (Scheme 1.3).¹²

^{*} The term tail addition is used to refer to addition to the less highly substituted end of the double bond.

[†] To this day some texts put forward product stability as the sole explanation for preferential tail addition.



- (b) Dependable measurements of rate constants for radical reactions became available which allowed structure-reactivity relationships to be reliably assessed.¹³
- (c) Data on bond dissociation energies were evaluated to demonstrate that the amount of stabilization provided to a radical center by adjacent alkyl substituents is small. The relative stability of primary vs secondary vs tertiary radicals, even if fully reflected in the transition state, is not sufficient to account for the degree of regioselectivity observed in additions to alkenes.¹⁴

It is now established that product radical stability is a consideration in determining the outcome of radical addition reactions only where a substituent provides substantial delocalization of the free spin into a π -system. Even then, because these reactions are generally irreversible and exothermic (and consequently have early transition states), resonance stabilization of the incipient radical center may play only a minor role in determining reaction rate and specificity.^{2,15-19} Thermodynamic factors will be the dominant influence only when polar and steric effects are more or less evenly balanced.^{20,21}

The importance of the various factors determining the rate and regiospecificity of addition is illustrated by the data shown in Table 1.2 and Table 1.3.

| Olefin | (CH ₃) |)₃CO• ^b | CI | H ₃ •° | CF | 3• ^c | cc | ¦l₃• ^c |
|----------------------------------|--------------------|---------------------|---------------|-----------------------|---------------|-----------------------|---------------|-----------------------|
| | $k_{\rm rel}$ | $k_{ m H}/k_{ m T}$ | $k_{\rm rel}$ | $k_{\rm H}/k_{\rm T}$ | $k_{\rm rel}$ | $k_{\rm H}/k_{\rm T}$ | $k_{\rm rel}$ | $k_{\rm H}/k_{\rm T}$ |
| CH ₂ =CH ₂ | 1.0 | - | 1.0 | - | 1.0 | - | 1.0 | - |
| CH ₂ =CHF | 0.7 | 0.35 | 1.1 | 0.2 | 0.5 | 0.12 | 0.62 | 0.11 |
| $CH_2=CF_2$ | 1.1 | 4.0 | - | 1 | 0.2 | 0.04 | 0.25 | 0.016 |
| CHF=CF ₂ | 6.6 | 4.5 | 5.8 | 2.1 | 0.05 | 0.55 | 0.29 | 0.32 |

 Table 1.2 Relative Rate Constants and Regiospecificities for Addition of Radicals to Halo-Olefins^a

a $k_{\rm rel}$ is overall rate constant for addition $(k_{\rm H}+k_{\rm T})$ relative to that for addition to ethylene (=1.0). All values have been rounded to 2 significant figures. b At 60 °C.²² c At 164 °C.¹³

Relative rate constants for reaction of methyl, trifluoromethyl, trichloromethyl,¹³ and *t*-butoxy radicals^{22,23} with the fluoro-olefins are summarized in Table 1.2. Note the following points:

- (a) Overall rates of addition for methyl and *t*-butoxy radicals are accelerated by fluorine substitution. In contrast, rates for trifluoromethyl and trichloromethyl radicals are reduced by fluorine substitution.
- (b) Trifluoromethyl and trichloromethyl radicals preferentially add to the less substituted end of trifluoroethylene. Methyl and *t*-butoxy radicals add preferentially to the more substituted end.
- (c) Trifluoromethyl and trichloromethyl radicals give predominantly tail addition to vinylidene fluoride, methyl radicals give both tail and head addition, *t*-butoxy radicals give predominantly head addition.

The overall trend of reactivities for *t*-butoxy radicals with the fluoro-olefins more closely parallels that for methyl radicals than that for the electrophilic trifluoromethyl or trichloromethyl radicals.

Table 1.3 Relative Rate Constants for Reactions of Radicals with Alkyl-
Substituted Acrylate Esters $CHR^1=CR^2CO_2CH_3^a$

| Monomer | R ^ſ | R^2 | PhCO; | PhCO ₂ • ^b Ph• ^b | | (CH ₃) ₃ CO+ ^b | | <i>c</i> -C ₆ H ₁₁ •° | | |
|-----------------|----------------|-------|----------------|---|----------------|--|-------------|---|----------------|-------|
| | | | k _H | kт | k _н | k _T | $k_{\rm H}$ | kт | k _H | kτ |
| MA | Н | Н | 0.2 | 1.0 | 0.03 | 1.0 | 0.02 | 1.0 | 0.002 | 1.0 |
| MMA | н | CH₃ | 0.35 | 4.5 | ≤0.01 | 1.6 | 0 | 2.9 | ≤0.001 | 0.71 |
| MC ^d | CH_3 | н | 1.6 | 1.3 | 0.07 | 0.12 | ≤0.03 | 0.3 | 0.001 | 0.011 |

a Rate constants relative to that for tail addition to MA (-1.0). All data have been rounded to 2 significant figures. b At 60 °C.²⁴ c At 20 °C.²⁵ d Methyl *trans*-2-butenoate (methyl crotonate).



Scheme 1.4

Outcomes from the reactions of radicals with substituted acrylate esters depend on the attacking radical (refer Table 1.3 and Scheme 1.4). The results may be summarized as follows (the methyl substituent is usually considered to be electron donating – Section 1.2.2):

- (a) Irrespective of the attacking radical, there is preferential addition to the tail of the double bond (to the end remote from the carbomethoxy group).
- (b) For the nucleophilic cyclohexyl radicals (c-C₆H₁₁•), the rate of addition to the unsubstituted end of the double bond is slightly retarded by alkyl substitution (ca 30% for MMA vs MA). The rate of addition to the substituted end of the double bond is dramatically retarded by alkyl substitution (ca 90-fold for MC vs MA).²⁵
- (c) For the slightly electrophilic phenyl and *t*-butoxy radicals [Ph•, $(CH_3)_3CO$ •]: the rate of addition to the unsubstituted end of the double bond is enhanced (2-3-fold) by alkyl substitution; the rate of addition to the substituted end of the double bond is retarded (>3-fold for MC vs MA) by alkyl substitution.^{24,26}
- (d) For the electrophilic benzoyloxy radicals (PhCO₂•): the rate of addition to the unsubstituted (tail) end of the double bond is enhanced (4.5-fold for MMA vs MA) by alkyl substitution; the rate of addition to the substituted (head) end of the double bond is slightly enhanced (75% for MMA vs MA) by alkyl substitution.²⁴

The data of Table 1.2 and Table 1.3 clearly cannot be rationalized purely in terms of the relative stabilities of the product radicals. Rather, "a complex interplay of polar, steric, and bond strength terms" must be invoked.¹³ In the following sections, each of these factors will be examined separately to illustrate their role in determining the outcome of radical addition.

1.2.1 Steric Factors

A clear demonstration of the relative importance of steric and resonance factors in radical additions to carbon-carbon double bonds can be found by considering the effect of (non-polar) substituents on the rate of attack of (non-polar) radicals. Substituents on the double bond strongly retard addition at the substituted carbon while leaving the rate of addition to the other end essentially unaffected (for example, Table 1.3). This is in keeping with expectation if steric factors determine the regiospecificity of addition, but contrary to expectation if resonance factors are dominant.

It is possible to resolve steric factors into several terms:

- (a) B-strain engendered by the change from sp^2 towards sp^3 hybridization at the site of attack.^{2,14} B-strain is a consequence of the substituents on the (planar) α -carbon of the double bond being brought closer together on moving towards a tetrahedral disposition (Figure 1.1). This term is important in all radical additions and is thought to be the main factor responsible for preferential attack at the less substituted end of the double bond.
- (b) Steric hindrance to approach of the attacking radical to the site of attack on the olefin. This term is usually only a minor factor except where substituents on the radical or on the olefin are very bulky.^{14,27}

(c) Steric hindrance to adoption of the required transition state geometry. This is not usually a determining factor in intermolecular addition of small radicals, but is extremely important in intramolecular addition where the approach of the reacting centers is constrained by the molecular geometry (Section 1.2.4).²⁸



Figure 1.1 Transition state for methyl radical addition to ethylene. Geometric parameters are from *ab initio* calculation with QCISD(T)/6-31GT(d) basis set.²⁹

Radical additions are typically highly exothermic and activation energies are small for carbon^{30,31} and oxygen-centered^{32,33} radicals of the types most often encountered in radical polymerization. Thus, according to the Hammond postulate,^{*} these reactions are expected to have early reactant-like transition states in which there is little localization of the free spin on C_{β} . However, for steric factors to be important at all, there must be significant bond deformation and movement towards sp^3 hybridization at C_{α} .

Various *ab initio* and semi-empirical molecular orbital calculations have been carried out on the reaction of radicals with simple alkenes with the aim of defining the nature of the transition state (Section 1.2.7).^{29,35,36} These calculations all predict an unsymmetrical transition state for radical addition (*i.e.* Figure 1.1) though they differ in other aspects. Most calculations also indicate a degree of charge development in the transition state.

The rate of radical addition is most dramatically affected by substituents either at the site of attack or at the radical center. Remote substituents generally have only a small influence on the stereochemistry and regiospecificity of addition unless these groups are very bulky or the geometry of the molecules is constrained (*e.g.* intramolecular addition – Section 1.2.4).

It is a common assumption that the influence of steric factors will be manifested mainly as a higher activation energy. In fact, there is good evidence³⁷ to show that steric factors are mainly reflected in a less favorable entropy of activation or Arrhenius frequency factor. This is due to the degrees of freedom

^{*} A highly exothermic (low activation energy) reaction will generally have a transition state that resembles the reactants³⁴

that are lost as the radical center approaches the terminus of the double bond and the α -substituents on the double bond are brought closer together on rehybridization.

1.2.2 Polar Factors

The rates of addition to the unsubstituted terminus of monosubstituted and 1,1disubstituted olefins (this includes most polymerizable monomers) are thought to be determined largely by polar factors.^{2,16} Polymer chemists were amongst the first to realize that polar factors were an important influence in determining the rate of addition. Such factors can account for the well-known tendency for monomer alternation in many radical copolymerizations and provide the basis for the Q-e, the Patterns of Reactivity, and many other schemes for estimating monomer reactivity ratios (Section 7.3.4).

The traditional means of assessment of the sensitivity of radical reactions to polar factors and establishing the electrophilicity or nucleophilicity of radicals is by way of a Hammett $\sigma\rho$ correlation. Thus, the reactions of radicals with substituted styrene derivatives have been examined to demonstrate that simple alkyl radicals have nucleophilic character^{38,39} while haloalkyl radicals⁴⁰ and oxygen-centered radicals²³ have electrophilic character (Table 1.4). It is anticipated that electron-withdrawing substituents (*e.g.* Cl, F, CO₂R, CN) will enhance overall reactivity towards nucleophilic radicals and reduce reactivity towards electrophilic radicals. Electron-donating substituents (alkyl) will have the opposite effect.

Many researchers have applied similar approaches to develop or apply linear free energy relationships, when the substituent is directly attached to the double bond, with some success. Two of the more notable examples can be found in the Patterns of Reactivity Scheme (Section 7.3.4) and the works of Giese and coworkers.^{16,19}

While steric terms may be the most significant factor in determining that tail addition is the predominant pathway in radical addition, polar factors affect the overall reactivity and have a significant influence on the degree of regiospecificity. In the reaction of benzoyloxy radicals with MMA, even though there is still a marked preference for tail addition, the methyl substituent enhances the rate constants for attack at both head and tail positions over those seen for MA (Table 1.3). With cyclohexyl radicals the opposite behavior is seen. Relative rate constants are reduced and the preference for tail addition is reinforced. For olefins substituted with electron-donor substituents, nucleophilic radicals give the greatest tail *vs* head specificity. The converse generally also applies.

In the reactions of the fluoro-olefins, steric factors are of lesser importance because of the relatively small size of the fluoro-substituent.⁵ Fluorine and hydrogen are of similar bulk. In these circumstances, it should be expected that polar factors could play a role in determining regiospecificity. Application of the usual rules to vinylidene fluoride leads to a prediction that, for nucleophilic radicals, the rate of head addition will be enhanced. Similarly, for electrophilic radicals, the rate of tail addition will be enhanced (Figure 1.2).

| | | Addition to styrenes | | Addition to styrenes H Abstrac | | | | toluenes |
|------------|---|-----------------------|-----------------------|--------------------------------|-----------------------|------------|--|----------|
| | radical | ρ | ρ | ρ^+ | ρ | ρ^{d} | | |
| | (CH ₃) ₃ C• | 1.1 ^{a,38} | - | 0.49 ^{b.41} | - | - | | |
| ſ | <i>c</i> -C ₆ H ₁₁ ∙ | $0.68^{a,38}$ | - | - | - | - | | |
| | <i>n</i> -C ₆ H ₁₃ • | - | - | $0.45^{a,38}$ | - | - | | |
| <u>i</u> j | <i>n</i> -C ₁₁ H ₂₃ • | - | - | - | $0.45^{a.42}$ | - | | |
| lir | CH3• | - | - | -0.1 ^{c.43} | -0.12 ^{c,43} | -0.21 | | |
| eo | (CH ₃) ₃ CO- | -0.27 ^{e.23} | -0.31 ^{e.23} | -0.32 ^{f.44} | -0.36 ^{f,44} | -0.36 | | |
| | (CH ₃) ₃ COO+ | - | - | $-0.56^{2.45}$ | -0.78 ^{2.45} | -0.73 | | |
| <u> </u> | (CH ₃) ₂ N• | - | - | $-1.08^{h.46}$ | -1.66 ^{h,46} | -0.96 | | |
| | CCI3• | $-0.42^{i.40}$ | $-0.43^{i.40}$ | $-1.46^{1.47}$ | -1.46 ^{j.47} | -1.67 | | |
| I | <i>n</i> -C ₈ F ₁₇ • | - | -0.53 ⁴⁸ | - | - | - | | |

Table 1.4 Hammett ρ and ρ' Parameters for Reactions of Radicals

a 42 °C. b 80 °C. c 100 °C. d ρ values recalculated by Pryor *et al.*⁴⁹ based on *m*-substituted derivatives only. e 60 °C, benzene. f 45 °C, chlorobenzene. Value shows solvent dependence. g 40 °C. h 136 °C. i 70 °C. j 50 °C.



Figure 1.2 Effect of polar factors on regiospecificity of radical addition.

The behavior of methyl and halomethyl radicals in their reactions with the fluoro-olefins (Table 1.2), can thus be rationalized in terms of a more dominant role of polar factors and the nucleophilic or electrophilic character of the radicals involved.² Methyl radicals are usually considered to be slightly nucleophilic, trifluoromethyl and trichloromethyl radicals are electrophilic (Table 1.4).

However, consideration of polar factors in the traditional sense does not provide a ready explanation for the regiospecificity shown by the *t*-butoxy radicals (which are electrophilic, Table 1.3) in their reactions with the fluoro-olefins (Table 1.2).^{22,23} Apparent ambiphilicity has been reported²¹ for other "not very electrophilic radicals" in their reactions with olefins and has been attributed to the polarizability of the radical.

1.2.3 Bond Strengths

The overriding importance of polar factors in determining rates of addition has recently been questioned by Fischer and Radom⁴ who argue that reaction enthalpy

should be considered the dominant factor in determining the rate of tail addition. Tedder and Walton¹³ have stated: "If an experimentalist requires a simple qualitative theory, he should seek to estimate the strength of the new bond formed during the initial addition step...". Historically, a perceptual problem has been that the bond strength or reaction enthalpy term cannot be separated rigorously from the polar and steric factors discussed above since the latter both play an important role in determining the strength of the new bond. Fischer and Radom's⁴ rationale is discussed below (1.2.7).

Just as steric factors may in some cases retard addition, factors that favor bond formation should be anticipated to facilitate addition. A pertinent example is the influence of α -fluorine substitution on C-X bond strength.⁵⁰ The C-C bond in CH₃-CF₃ is 46 kJ mol⁻¹ stronger than that in CH₃-CH₃. Further fluorine substitution leads to a progressive strengthening of the bond. The effect is even greater for C-O bonds. The C–O bond dissociation energies in CF₃–O–CF₃ and CF₃–OH are greater by 92 and 75 kJ mol⁻¹, respectively, than those in CH₃–O–CH₃ and CH₃–OH. This effect offers an explanation for the differing specificity shown by oxygen- and carbon-centered radicals in their reactions with the fluoro-olefins (Table 1.2).^{23,51-53} The finding, that *t*-butoxy radicals give predominantly head addition with vinylidene fluoride (Scheme 1.3), can therefore be understood in terms of the relative strengths of the CF₂–O and CH₂–O bonds.²³

1.2.4 Stereoelectronic Factors

A stereoelectronic requirement in radical addition to carbon-carbon double bonds first became apparent from studies on radical cyclization and the reverse (fragmentation) reactions.⁵⁴⁻⁵⁶ It provides a rationalization for the preferential formation of the less thermodynamically stable *exo*-product (*i.e.* head addition) from the cyclization of ω -alkenyl radicals (**16** - Scheme 1.5).^{18,57-64}



It was proposed that the transition state requires approach of the radical directly above the site of attack and perpendicular to the plane containing the carbon-carbon double bond. An examination of molecular models shows that for the 3-butenyl and 4-pentenyl radicals (16, n=1,2) such a transition state can only be reasonably achieved in *exo*-cyclization (*i.e.* $16\rightarrow15$). With the 5-hexenyl and 6-heptenyl radicals (16, n=3,4), the transition state for *exo*-cyclization $(16\rightarrow15)$ is more easily achieved than that for *endo*-cyclization (*i.e.* $16\rightarrow17$).

The mode and rate of cyclization can be modified substantially by the presence of substituents at the radical center, on the double bond, and at positions on the

connecting chain. As with intermolecular addition, substituents at the site of attack on the double bond strongly retard addition. For the 5-hexenyl system (16, n=3) the magnitude of the effect is such that methyl substitution at the 5-position causes *endo*-cyclization to be favored. For the 5,6-disubstituted radical the rates for both *exo-* and *endo*-addition are slowed and *exo*-cyclization again dominates. A full discussion of substituent effects on intramolecular addition can be found in the reviews cited above.

Stereoelectronic factors may also become important in polymerization when bulky substituents may hinder adoption of the required transition state. They may help explain why rate constants for addition of monomeric radicals may be very different from those for addition of dimeric or higher radicals.⁴

1.2.5 Entropic Considerations

The Arrhenius frequency factors $[\log(A/M^{-1}s^{-1})]$ for addition of carbon centered radicals to the unsubstituted terminus of monosubstituted or 1,1disubstituted olefins cover a limited range (6.0-9.0), depend primarily on the steric demand of the attacking radical and are generally unaffected by remote alkene substituents. Typical values of $\log(A/M^{-1}s^{-1})$ are *ca* 6.5 for tertiary polymeric (*e.g.* PMMA•), *ca* 7.0 for secondary polymeric (PS•, PMA•), and *ca* 7.5, 8.0 and 8.5 for small tertiary (*e.g. t*-C₄H₉•), secondary (*i*-C₃H₇•) and primary (CH₃•, C₂H₅•) radicals respectively (Section 4.5.4).⁴ For 1,2,2-trisubstituted alkenes the frequency factors are about an order of magnitude lower.⁴ The trend in values is consistent with expectation based on theoretical calculations.

Frequency factors are often determined from data obtained within a narrow temperature window. For this reason, it has been recommended⁴ that when extrapolating rate constants less error might be introduced by adopting the standard values for frequency factors (above) than by using experimentally measured values. The standard values may also be used to estimate activation energies from rate constants measured at a single temperature.

1.2.6 Reaction Conditions

There is ample evidence to show that the outcome of radical addition is dependent on reaction conditions and, in particular, the reaction temperature and the reaction medium.

1.2.6.1 Temperature

Radical additions to double bonds are, in general, highly exothermic processes and rates increase with increasing temperature. The regiospecificity of addition to double bonds and the relative reactivity of various olefins towards radicals are also temperature dependent. Typically, specificity decreases with increasing temperature (the Reactivity-Selectivity Principle applies). However, a number of exceptions to this general rule have been reported.^{38,65}

Giese and Feix⁶⁵ examined the temperature dependence of the relative reactivity of fumarodinitrile and methyl α -chloroacrylate towards a series of alkyl radicals (Scheme 1.6). The temperature dependence was such that they predicted that the order of reactivity of the radicals would be reversed for temperatures above 280 K (the isoselective temperature - Figure 1.3). This finding clearly indicates the need for care when comparing relative reactivity data.⁶⁶



Figure 1.3 Relative rate constants for addition of alkyl radicals to fumarodinitrile (k_1) and methyl α -chloroacrylate (k_2) as a function of temperature (Scheme 1.6).⁶⁵

1.2.6.2 Solvent

It is established that rates of propagation in radical polymerization and reactivity ratios in copolymerization can show significant variation according to the solvent employed (Section 8.3.1).⁶⁷⁻⁷¹ For polymerizations of ethylene and vinyl acetate, effects on low conversion values of k_p in excess of an order of magnitude have been reported.^{68,72} Smaller though measurable solvent effects on k_p are seen for other monomers. However, conventional wisdom has it that, except for those reactions involving charged intermediates, solvent effects on the rate and regioselectivity of radical addition to olefins are small and, consequently, they have not been widely studied. Nonetheless, reports of measurable solvent effects continue to appear.

Gas phase rate constants are typically an order of magnitude higher than solution phase rate constants. Fischer and Radom⁴ have postulated that gas phase

frequency factors should be similar to liquid phase numbers and the higher rate constants should therefore be largely attributed to lower activation energies (by ca 6.5 kJ mol⁻¹).

Giese and Kretzschmar⁷³ found the rate of addition of hexenyl radicals to methyl acrylate increased 2-fold between aqueous tetrahydrofuran and aqueous ethanol. Salikhov and Fischer⁷⁴ reported that the rate constant for *t*-butyl radical addition to acrylonitrile increased 3.6-fold between tetradecane and acetonitrile. Bednarek *et al.*⁷⁵ found that the relative reactivity of S *vs* MMA towards phenyl radicals was *ca* 20% greater in ketone solvents than it was in aromatic solvents.

More pronounced solvent effects have been observed in special cases where substrates or products possess ionic character. Ito and Matsuda⁷⁶ found a 35-fold reduction in the rate of addition of the arenethiyl radical **18** to α -methylstyrene when the solvent was changed from dimethylsulfoxide to cyclohexane. Rates for addition of other arenethiyl radicals do not show such a marked solvent dependence. The different behavior was attributed to the radical **18** existing partly in a zwitterionic quinonoid form (Scheme 1.7).⁷⁷



Scheme 1.7

1.2.7 Theoretical Treatments

There have been many theoretical studies of radical addition reactions using *ab initio* methods,^{4,35,36,53,78-88} semi-empirical calculations,^{89,90} molecular mechanics^{54,55} and other procedures. While geometries do not vary substantially with the level of theory, to obtain meaningful activation parameters with *ab initio* methods, a very high level of theory is required.^{4,36} Such calculations are, at this stage, only practicable for small systems. However, computational power and method efficiency have improved substantially over the past few years and there is no evidence that this trend is leveling off. Heuts *et al.*^{82,84} have argued that reliable Arrhenius Λ factors may be available using lower levels of theory.

The calculations using semi-empirical and low level *ab initio* methods do not give good values of reaction enthalpies or activation parameters and appear to fail dismally in some circumstances.⁴ However, they have been shown to be useful in predicting relative energies for structurally similar systems and can give useful insights into mechanism. Methods for obtaining estimates of relative activation energies by molecular mechanics have also been devised.⁵⁵

Various empirical schemes have also been proposed as predictive tools with respect to the outcome of radical addition reactions.^{91,92} Two-parameter schemes, including the Q-e scheme (Section 7.3.4.1), Patterns of Reactivity (Section 7.3.4.2)

and another developed by Ito and Matsuda⁹³ have been used with some success. Bakken and Jurs have used an approach based on multiple regression and neural networks and tested it by predicting rate constants for addition of methyl⁹² and hydroxyl radicals⁹¹ to various substrates. Denisov⁹⁴ proposed the "parabolic model" which involves eight descriptors.

Frontier Molecular Orbital (FMO) theory⁹⁵ may also be applied to provide qualitative understanding.^{16,19} The frontier orbital of the radical is that bearing the free spin (the SOMO) and during radical addition this will interact with both the π^* antibonding orbital (the LUMO) and the π -orbital (the HOMO) of the olefin. Both the SOMO-HOMO and the SOMO-LUMO interactions lead to a net drop in energy [*i.e.* $2(E_2)$ - E_3 or E_1 respectively - Figure 1.4]. The dominant interaction and the reaction rates depend on the relative energies of these orbitals. Most radicals have high energy SOMO's and the SOMO-LUMO interaction is likely to be the most important. However, with highly electron deficient radicals, the SOMO may be of sufficiently low energy for the SOMO-HOMO interaction to be dominant.



Figure 1.4 SOMO-HOMO and SOMO-LUMO orbital interaction diagrams.

For olefins with π -substituents, whether electron-withdrawing or electrondonating, both the HOMO and LUMO have the higher coefficient on the carbon atom remote from the substituent. A predominance of tail addition is expected as a consequence. However, for non-conjugated substituents, or those with lone pairs (*e.g.* the halo-olefins), the HOMO and LUMO are polarized in opposite directions. This may result in head addition being preferred in the case of a nucleophilic radical interacting with such an olefin. Thus, the data for attack of alkyl and fluoroalkyl radicals on the fluoro-olefins (Table 1.2) have been rationalized in terms of FMO theory.¹⁶ Where the radical and olefin both have near "neutral" philicity, the situation is less clear.²¹

The State Correlation Diagram (SCD) approach introduced by Shaik and Pross⁹⁶ appears similar in some respects. However, the LUMO, HOMO and the first two excited states are considered. (refer Figure 1.5)^{4,53} Thus, if we consider the interaction of the radical with the olefin in its ground (singlet) state ($\mathbf{R} + \mathbf{C}=\mathbf{C}^1$) and excited (triplet) state ($\mathbf{R} + \mathbf{C}=\mathbf{C}^3$) and two charge transfer

configurations ($R^+ + C^-C^-$) and ($R^- + C^-C^+$), the energy of ground state configuration increases while those of the excited state configurations decrease as the reactants approach. In the transition state, the various configurations mix according to their relative energies. A lucid description of the application of this approach to rationalize rate constants to addition of carbon centered radicals to olefins has recently been provided by Fischer and Radom.⁴ Guided by the SCD analysis, they devised a scheme to predict absolute rate constants of radical addition based on knowledge of the reaction enthalpy, the singlet-triplet energy gap, the ionization potential and electron affinity of the olefin and the radical and the Coulomb interaction energy.



Figure 1.5 Schematic state correlation diagram for radical addition to a carboncarbon double bond showing configuration energies as a function of the reaction coordinate.

1.2.8 Summary

No single factor can be identified as determining the outcome of radical addition. Nonetheless, there is a requirement for a set of simple guidelines to allow qualitative prediction. This need was recognized by Tedder and Walton,^{2,17} Beckwith *et al.*,⁵⁹ Giese,¹⁶ and, most recently, Fischer and Radom.⁴ With the current state of knowledge, any such rules must be partly empirical and, therefore, it is to be expected that they may have to be revised from time to time as more results become available and further theoretical studies are carried out. However, this does not diminish their usefulness.

The following set of guidelines is a refinement of those suggested by Tedder:²

(a) For mono- or 1,1-disubstituted olefins, there is usually preferential addition to the unsubstituted (tail) end of the double bond. This selectivity can be largely

attributed to the degree of steric compression associated with the formation of the new bond which usually overrides other influences on the regioselectivity.

- (b) Substituents with π -orbitals (*e.g.* -CH=CH₂, -Ph) that can overlap with the half-filled atomic orbital of the incipient radical center may enhance the rate of addition at the remote end of the double bond. However, substituents with non-bonding pairs of electrons (*e.g.* -F, -Cl, -OR) have only a very small resonance effect. Most radical additions are exothermic and have early transition states and delocalization of the unpaired electron in the adduct radical is of small importance.
- (c) Polarity can have a major effect on the overall rate of addition. Electron withdrawing substituents facilitate the addition of nucleophilic radicals while electron donating substituents enhance the addition of electrophilic radicals.
- (d) The regioselectivity of addition to polysubstituted olefins is primarily controlled by the degree of steric compression associated with forming the new bond. However, if steric effects are small or mutually opposed, polarity can be the deciding factor.
- (e) Even though the regioselectivity of addition to polysubstituted olefins is governed mainly by steric compression, polarity can influence the magnitude of the regioselectivity, making it larger or smaller depending on the relative electronegativity of the radical and the substituents on the olefin. The net result may be that the more reactive radical is the more selective.

1.3 Hydrogen Atom Transfer

Atom or radical transfer reactions generally proceed by a S_H^2 mechanism (substitution, homolytic, bimolecular) that can be depicted as shown in Figure 1.6. This area has been the subject of a number of reviews.^{1-3,27,97,99} The present discussion is limited, in the main, to hydrogen atom abstraction from aliphatic substrates and the factors which influence rate and specificity of this reaction.



Figure 1.6 Transition state for hydrogen atom abstraction.

1.3.1 Bond Dissociation Energies

30

Simple thermochemical criteria can often be used to predict the relative facility of hydrogen atom transfer reactions. Evans and Polanyi¹⁰⁰ recognized this and suggested the following relationship (the Evans-Polanyi equation, eq. 1) between the activation energy for hydrogen atom abstraction (E_a) and the difference between the bond dissociation energies for the bonds being formed and broken (ΔH°):

$$E_{\rm a} = \alpha \,\Delta H^{\rm o} + \beta \tag{1}$$

where α and β are constants. It follows that for hydrogen abstraction by a given radical from a compound X-H, since the strength of the bond being formed is a constant, there should be a straight line relationship between the activation energy and the strength of the bond being broken [D(X-H)] (eq. 2):

$$E_{a} = \alpha' [D(X-H)] + \beta'$$
⁽²⁾

where α' and β' are constants. Examples of the application of the Evans-Polanyi equation can be found in reviews by Russell⁹⁷ and Tedder.^{2,3} In the absence of severe steric constraints, straight line correlations between the relative reactivity of substrates towards a given radical can be found for systems: (a) where there is little polarity in the transition state, or (b) when the transition states are of like polarity. Tedder^{2,3} has also stressed that, in these reactions, it is important to take note of the strength of the bond being formed. If there is no polarity in the transition state, the more exothermic reaction will generally be the less selective.

Bond dissociation energies qualitatively predict the order of reactivity of X-H bonds shown in Figure 1.7 (for examples see Table 1.1). However, as will become apparent, a variety of factors may perturb this order.



 $R_2N-H \sim RO-H < R_3C-H < R_3Si-H < ROO-H < R_2P-H < RS-H < RSe-H$

Figure 1.7 Predicted order of reactivity of X-H compounds.

1.3.2 Steric Factors

Steric factors fall into four main categories:²⁷

(a) The release or occurrence of steric compression due to rehybridization in the transition state where the attacking radical and site of attack are each undergoing rehybridization (from $sp^2 \rightarrow sp^3$ and $sp^3 \rightarrow sp^2$ respectively for aliphatic carbons – refer Figure 1.6). As a consequence, substituents on the attacking radical are brought closer together while those at the site of attack

move apart. Thus, depending on the nature of the substituents at these centers, steric retardation or acceleration may accompany rehybridization.

- (b) Steric hindrance of the approach of the attacking radical to the point of reaction in the substrate. This is important for the attack of very bulky radicals on hindered substrates.
- (c) Steric inhibition of resonance important in conformationally constrained molecules (Section 1.3.4).
- (d) Steric hindrance to adoption of the required co-linear arrangement of atoms in the transition state. This is important in intramolecular reactions (Section 1.3.4).

The first term is of importance in all atom abstraction reactions, however, since the reactions are often highly exothermic with consequent early transition states, the effect may be small.

1.3.3 Polar Factors

Polar factors can play an extremely important role in determining the overall reactivity and specificity of homolytic substitution.⁹⁷ Theoretical studies on atom abstraction reactions support this view by showing that the transition state has a degree of charge separation.^{101,102}

The traditional method of assessing the polarity of reactive intermediates is to examine the effect of substituents on rates and establish a linear free energy relationship (*e.g.* the Hammett relationship). The reactions of numerous radicals with substituted toluenes have been examined in this context. The value of the Hammett ρ parameter provides an indication of the sensitivity of the reaction to polar factors and gives a measure of the electrophilic or nucleophilic character of the attacking radical. For example, methyl radicals, usually considered to be slightly nucleophilic, have a slightly negative ρ value with respect to abstraction of benzylic hydrogens (Table 1.4).⁴³ Other simple alkyl radicals typically have positive ρ values.^{41,42,103,104} Heteroatom-centered radicals (*e.g.* R₂N•, RO•, Cl•) generally have negative ρ values.^{44,46,105,106} However, care must be taken in interpreting the results purely in terms of polar effects since electron withdrawing substituents typically also increase bond dissociation energies.^{41,49,102,105}

The basic Hammett scheme often does not offer a perfect correlation and a number of variants on this scheme have been proposed to better explain reactivities in radical reactions.²³ However, none of these has achieved widespread acceptance. It should also be noted that linear free energy relationships are the basis of the Q-e and Patterns of Reactivity schemes for understanding reactivities of propagating species in chain transfer and copolymerization.

A striking illustration of the influence of polar factors in hydrogen abstraction reactions can be seen in the following examples (Figure 1.8) where different sites on the molecule are attacked preferentially according to the nature of the attacking radical.⁹⁷



Figure 1.8 Preferred site of attack in hydrogen abstraction by various radicals.

1.3.4 Stereoelectronic Factors

There is a demonstrated requirement for a near co-linear arrangement of the orbital bearing the unpaired electron and the breaking C-H bond in the transition state for hydrogen atom transfer.^{28,60,107,108} This becomes of particular importance for intramolecular atom transfer and accounts for the well-known preference for these reactions to occur by way of a six-membered transition state. The adoption of the chair conformation in the transition state for 1,5-atom transfer allows the requisite arrangement of atoms to be adopted readily. Such a transition state cannot be as readily achieved in smaller rings without significant strain being incurred, or in larger rings due to the severe non-bonded interactions and/or a less favorable entropy of activation.¹⁰⁷⁻¹¹⁰

Thus, for radicals **19**, there is a strong preference for 1,5-hydrogen atom transfer (Table 1.5).¹¹¹ Although 1,6-transfer is also observed, the preference for 1,5-hydrogen atom transfer over 1,6-transfer is substantial even where the latter pathway would afford a resonance stabilized benzylic radical.^{111,112} No sign of 1,2-, 1,3-, 1,4-, or 1,7-transfer is seen in these cases. Similar requirements for a co-linear transition state for homolytic substitution on sulfur and oxygen have been postulated.^{18,60}

 Table 1.5
 Specificity of Intramolecular Hydrogen Abstraction¹¹¹



It is expected from simple thermochemical considerations that adjacent π -, σ or lone pair orbitals should have a significant influence over the facility of atom transfer reactions. Thus, the finding that *t*-butoxy radicals show a marked preference for abstracting hydrogens α to ether oxygens (Figure 1.9) is not

surprising. The reduced reactivity of the hydrogens β to oxygen in these compounds is attributed to polar influences.^{113,114}

The most direct evidence that stereoelectronic effects are also important in these reactions follows from the specificity observed in hydrogen atom abstraction from conformationally constrained compounds.^{18,60} C-H bonds adjacent to oxygen¹¹³⁻¹¹⁸ or nitrogen¹¹⁹ and which subtend a small dihedral angle with a lone pair orbital ($<30^\circ$) are considerably activated in relation to those where the dihedral angle is or approaches 90°. Thus, the equatorial H in **20** is reported to be 12 times more reactive towards *t*-butoxy radicals than the axial H in **21**.¹¹⁵



Figure 1.9 Relative reactivity per hydrogen atom of indicated site towards *t*-butoxy radicals.^{113,114}



A further example of the importance of this type of stereoelectronic effect is seen in the reactions of *t*-butoxy radicals with spiro[2,n]alkanes (**22**) where it is found that hydrogens from the position α - to the cyclopropyl ring are specifically abstracted. This can be attributed to the favorable overlap of the breaking C-H bond with the cyclopropyl σ bonds.^{120,121} No such specificity is seen with bicyclo[n,1,0]alkanes (**23**) where geometric constraints prevent overlap.



1.3.5 Reaction Conditions

Even though dissociation energies for X-H bonds appear insensitive to solvent changes, 122,123 the nature of the reaction medium 70,71,124 and the reaction

temperature⁶⁶ can significantly affect the specificity and rate of atom abstraction reactions. One of the more controversial cases concerns the effect of aromatic solvents on hydrogen abstraction by atomic chlorine.

It has been proposed that aromatic solvents, carbon disulfide, and sulfur dioxide form a complex with atomic chlorine and that this substantially modifies both its overall reactivity and the specificity of its reactions.¹²⁵ For example, in reactions of Cl• with aliphatic hydrocarbons, there is a dramatic increase in the specificity for abstraction of tertiary or secondary over primary hydrogens in benzene as opposed to aliphatic solvents. At the same time, the overall rate constant for abstraction is reduced by up to two orders of magnitude in the aromatic solvent.¹²⁶ The exact nature of the complex responsible for this effect, whether a π -complex (24) or a chlorocyclohexadienyl radical (25), is not yet resolved.¹²⁶⁻¹³²



Significant, though smaller, solvent effects have also been reported for alkoxy radical reactions (Section 3.4.2.1).¹³³⁻¹³⁷

1.3.6 Abstraction vs Addition

The relative propensity of radicals to abstract hydrogen or add to double bonds is extremely important. In radical polymerization, this factor determines the significance of transfer to monomer, solvent, *etc.* and hence the molecular weight and end group functionality (Chapter 6). It also provides one basis for initiator selection (Section 3.2.1).

| Bond | D | Bond | D | Bond | D | Bond | D | |
|------------------|----------------|---|-----|-------------------------------|-----|-------------------------------------|-----|---|
| (a) C-R bonds | | | | | | | | |
| $C_2H_5-C_2H_5$ | - 343 <i>i</i> | -C ₃ H ₇ -C ₂ H ₅ | 335 | $t-C_4H_9-C_2H_5$ | 326 | allyl-C ₂ H ₅ | 299 | |
| C_2H_5-H | -410 <i>i</i> | -C ₃ H ₇ -H | 395 | <i>t-</i> C₄H ₉ -H | 384 | allyl-H | 364 | |
| $\Delta^{\rm b}$ | 67 | | 60 | | 58 | | 65 | |
| (b) X-R bonds | | | | | | | | |
| $H_2N-C_2H_5$ | 351 H | HO-C₂H₅ | 381 | $C_2H_5O-C_2H_5$ | 339 | CI-C₂H₅ | 339 | |
| H₂N-H | 460 H | HO-H | 498 | C₂H₅O-H | 435 | CI-H | 431 | |
| Δ^{b} | 109 | | 117 | | 96 | | 92 | |
| | | 1 | | | | | | - |

Table 1.6 Bond Dissociation Energies $(D \text{ in } kJ \text{ mol}^{-1})^{a.7}$

a Values rounded to nearest kJ mol⁻¹. b Difference between $D(C-C_2H_5)$ and D(C-H) c Difference between $D(X-C_2H_5)$ and D(X-H).

The hydrogen abstraction:addition ratio is generally greater in reactions of heteroatom-centered radicals than it is with carbon-centered radicals. One factor is the relative strengths of the bonds being formed and broken in the two reactions (Table 1.6). The difference in exothermicity (Δ) between abstraction and addition reactions is much greater for heteroatom-centered radicals than it is for carbon-centered radicals. For example, for an alkoxy as opposed to an alkyl radical, abstraction is favored over addition by *ca* 30 kJ mol⁻¹. The extent to which this is reflected in the rates of addition and abstraction will, however, depend on the particular substrate and the other influences discussed above.

A number of studies have found that increasing nucleophilicity of the attacking radical favors abstraction over addition to an unsaturated system (benzene ring or double bond).^{41,138,139} Bertrand and Surzur¹³⁹ surveyed the literature on the reactions of oxygen-centered radicals and observed that the ratio of abstraction to addition increased as shown in Figure 1.10.



Figure 1.10 Dependence of abstraction:addition ratio on nucleophilicity for oxygen-centered radicals.

They, and later Houk,¹⁴⁰ attempted to establish a theoretical basis for this trend in terms of FMO theory. Pryor *et al.*⁴¹ have found a similar trend for a series of aryl and alkyl radicals (Figure 1.11).



Figure 1.11 Dependence of abstraction:addition ratio on nucleophilicity for carbon-centered radicals.

However, the situation is not as clear-cut as it might at first seem since a variety of other factors may also contribute to the above-mentioned trend. Abuin *et al.*¹⁴¹ pointed out that the transition state for addition is sterically more demanding than that for hydrogen-atom abstraction. Within a given series (alkyl or alkoxy), the more nucleophilic radicals are generally the more bulky (*i.e.* steric factors favor the same trends). It can also be seen from Table 1.6 that, for alkyl radicals, the values of *D* decrease in the series primary>secondary>tertiary (*i.e.* relative bond strengths favor the same trend).

1.3.7 Summary

A simple unifying theory to explain rate and specificity in atom abstraction reactions has yet to be developed. However, as with addition reactions, it is possible to devise a set of guidelines to predict qualitatively the rate and outcome of radical transfer processes. The following are based on those suggested by Tedder:²

- (a) When there is little polarity in the transition state (or where the polarity is constant in a reaction series), the relative rates of atom transfer by a particular radical (selectivity) will correlate with the strengths of the bonds being broken.
- (b) The strength of the bond being formed will be important in determining the absolute rate and the degree of selectivity.
- (c) Steric strain relieved or incurred with formation of the new radical center may be important particularly for endothermic or near thermoneutral reactions.
- (d) Nucleophilic radicals will prefer to attack electron rich sites. Electrophilic radicals will prefer to attack electron poor sites. If ΔII is small, polar factors may override thermodynamic considerations.

1.4 Radical-Radical Reactions

The last comprehensive review of reactions between carbon-centered radicals appeared in 1973.¹⁴² Rate constants for radical-radical reactions in the liquid phase have been tabulated by Griller.¹⁴³ The area has also been reviewed by Alfassi¹⁴⁴ and Moad and Solomon.¹⁴⁵ Radical-radical reactions are, in general, very exothermic and activation barriers are extremely small even for highly resonance-stabilized radicals. As a consequence, reaction rate constants often approach the diffusion-controlled limit (typically ~10⁹ M⁻¹ s⁻¹).

The reaction may take several pathways:

(a) Combination, which usually but not invariably (Section 1.4.1), takes place by a simple head-to-head coupling of radicals.

(b) Disproportionation, which involves the transfer of a β -hydrogen from one radical of the pair to the other (Section 1.4.2).

(c) Electron transfer, in which the product is an ion pair.

$$-c\langle \cdot + \cdot \rangle c - - -c\langle \oplus + \odot \rangle c -$$

36

The latter pathway is rare for reactions involving only carbon-centered radicals and will not be considered further in this chapter.

1.4.1 Pathways for Combination

The combination of carbon-centered radicals usually involves head-to-head $(\alpha, \alpha$ -) coupling. Exceptions to this general rule occur where the free spin can be delocalized into a π -system. The classic example involves the triphenylmethyl radical (13) which combines to give exclusively the α -para coupling product (26), Scheme 1.8).²⁷ This chemistry is also seen in cross reactions of 13 with other tertiary radicals.¹⁴⁶



Scheme 1.8

Other benzyl radicals, including the parent benzyl radical, give reversible formation of quinonemethide derivatives (typically a mixture of α,p - and α,o -coupling products) in competition with α,α -coupling (see also Section 5.2.2.1.1).¹⁴⁷⁻¹⁵¹ The kinetic product distribution appears to be determined by steric factors: α -substitution favors quinonemethide formation; ring substitution favors α,α -coupling. However, since quinonemethide formation is reversible, the only isolable product is often that from α,α -coupling.

For combination processes involving cyanoalkyl radicals, reversible C,Ncoupling occurs in competition with C,C-coupling. Steric factors appear to be important in determining the relative amounts of C,C- and C,N-coupling¹⁵² and exclusive C,N-coupling is observed when two bulky radicals combine.¹⁵³ For cyanoisopropyl radicals, C,N-coupling is the kinetically preferred pathway (Scheme 1.9).¹⁰⁵⁻¹⁰⁷ However, since the formation of the ketenimine is thermally reversible, the C,C-coupling product is usually the major isolated product (Section 5.2.2.1.3).

$$N \equiv C - \stackrel{i}{C} - \stackrel{i}{C} = C \equiv N - 2 \left[\stackrel{i}{\downarrow} C - C \equiv N - 2 \left[\stackrel{i$$

Ē

Scheme 1.9

An example of C,O-coupling of α -ketoalkyl radicals with reversible formation of an enol ether has also been reported for a system where C,C-coupling is very hindered (Scheme 1.10).¹⁵⁴ However, this pathway is not observed for simpler species (Section 5.2.2.1.2).



Scheme 1.10

1.4.2 Pathways for Disproportionation

For simple alkyl radicals, the product distribution appears to be predictable using statistical arguments.



Scheme 1.11

For example, disproportionation of but-2-yl radicals produces a mixture of butenes as shown (Scheme 1.11).¹³⁸ Thermodynamic considerations suggest that but-1-ene and but-2-enes should be formed in a ratio of *ca* 2:98. However, the observed 5:4 ratio of but-1-ene:but-2-enes is little different from the 3:2 ratio that is expected on statistical grounds (*i.e.* ratio of β -hydrogens in the 1- and 3-positions).



For more highly substituted examples, it is clear that other factors are also important. Substitution at the radical center has a profound effect. For example, in disproportionation, radicals 27^{155} and 28^{156} show a marked preference for loss of a hydrogen from the α -methyl substituent.



With the radical **29**, even though loss of an equatorial hydrogen should be sterically less hindered and is favored thermodynamically (by relief of 1,3 interactions of the axial methyl), there is an 8-fold preference for loss of the axial hydrogen (at 100 °C). The selectivity observed in the disproportionation of this and other substituted cyclohexyl radicals led Beckwith¹⁸ to propose that disproportionation is subject to stereoelectronic control which results in preferential breaking of the C–H bond which has best overlap with the orbital bearing the unpaired spin.

1.4.3 Combination vs Disproportionation

Reactions between carbon-centered radicals generally give a mixture of disproportionation and combination. Much effort has been put into establishing the relative importance of these processes. The ratio of disproportionation to combination (k_{td}/k_{tc}) is dependent on the structural features of the radicals involved and generally shows only minor variation with solvent, pressure, temperature, *etc.*



Scheme 1.12

Early workers in the area^{157,158} suggested the involvement of a single 4-center transition state or intermediate which could lead to either disproportionation or combination (Scheme 1.12). The hypothesis fell from favor when it was established that k_{td}/k_{tc} showed a small though measurable dependence on temperature and pressure.¹⁴² It is now generally recognized that combination and disproportionation should be considered as two separate reactions with distinct transition states. This view is supported by theoretical studies.¹⁵⁹⁻¹⁶³

1.4.3.1 Statistical factors

For a given series of radicals, the ratio k_{td}/k_{tc} increases with the number of β -hydrogen atoms. However, in general, there is no straight-forward relationship between k_{td}/k_{tc} and the number of β -hydrogens and it is clear that other factors are involved.^{27,142} It is usually observed that even after allowing for the different

number of β -hydrogens, the importance of disproportionation increases with increasing substitution at the radical center. For example, in the self-reaction of simple primary, secondary, and tertiary alkyl radicals, the values of $k_{td}/k_{tc}n$ are *ca* 0.06, 0.2, and 0.8 respectively, where *n* is the number of β -hydrogens.^{27,142}

1.4.3.2 Steric factors

It has been suggested that the discrepancies between the value of k_{td}/k_{tc} observed and that predicted on the basis of simple statistics may reflect the greater sensitivity of combination to steric factors. Beckhaus and Rüchardt¹⁶⁴ reported a correlation between $\log(k_{td}/k_{tc})$ (after statistical correction) and Taft steric parameters for a series of alkyl radicals.

A graphic demonstration of the importance of steric factors on k_{td}/k_{tc} is provided by the contrasting behavior of radicals 30 and 31. The self-reaction of cumyl radicals (30) affords predominantly combination while the radical 31, in which an α -methyl is replaced by a neopentyl group, gives predominantly disproportionation.¹⁶⁵



In extreme cases, suitably bulky substituents at the radical center can render a radical persistent [*e.g.* di-*t*-butyl methyl radical (32)].^{166,167} This radical (32) possesses no hydrogens on the α -carbon and therefore cannot decay by the normal disproportionation mechanism.



The triisopropylmethyl radical (33) is another example of a persistent radical. In this case, both disproportionation and combination are substantially retarded by steric factors.^{168,169} In the preferred conformation shown, the β -hydrogens lie in a plane orthogonal to the orbital bearing the free spin.

The examples considered in this section lead to three conclusions:

(a) Disproportionation and combination can both be dramatically slowed by large β - or γ -substituents.

^{*} In the original work¹⁶⁵ the neopentyl substituent is incorrectly shown as a *t*-butyl substituent.

- (b) Combination is more sensitive to the presence of bulky β -substituents than disproportionation (*i.e.* k_{to}/k_{tc} is enhanced).
- (c) Steric factors can outweigh simple statistical factors (*e.g.* even though **31** has fewer β -hydrogens, it gives more disproportionation than **30**).

Two quite separate influences are important in determining the rate of disproportionation:

- (a) Steric hindrance to approach of the attacking radical (important for combination and disproportionation).
- (b) Steric hindrance to rotation about the α , β -bond (important for disproportionation).

This latter term is considered in more detail under stereoelectronic factors (Section 1.4.3.4).

1.4.3.3 Polar factors

Minato *et al.*¹⁶² proposed that the transition state for disproportionation has polar character while that for combination is neutral. The finding that polar solvents enhance k_{td}/k_{tc} for ethyl¹⁷⁰ and *t*-butyl radicals (Section 2.5.3.5), the very high k_{td}/k_{tc} seen for alkoxy radicals with α -hydrogens,¹⁷¹ and the trend in k_{td}/k_{tc} observed for reactions of a series of fluoroalkyl radicals (Scheme 1.13, Table 1.7) have been explained in these terms.^{144,162}

$$CH_{3-x}F_{x}^{-}C_{2}H_{5} \stackrel{k_{tc}}{\longleftarrow} CH_{3-x}F_{x}^{\bullet} + C_{2}H_{5}^{\bullet} \stackrel{k_{td}}{\longrightarrow} CH_{4-x}F_{x} + C_{2}H_{4}$$

Scheme 1.13

Table 1.7 Values of k_{td}/k_{tc} for the Cross-Reaction between Fluoromethyl and
Ethyl Radicals (25 °C) $^{172-174}$

| Radical | $k_{\rm td}/k_{\rm tc}$ | Radical | $k_{\rm td}/k_{\rm tc}$ | Radical | $k_{\rm td}/k_{\rm tc}$ |
|--------------------|-------------------------|---------|-------------------------|---------------------------------|-------------------------|
| CH3• | 0.039 | CHF2. | 0.068 | C ₂ F ₅ • | 0.24 |
| CH ₂ F• | 0.038 | CF3• | 0.11 | | |

1.4.3.4 Stereoelectronic and other factors

The transition state for disproportionation requires overlap of the β C—H bond undergoing scission and the *p*-orbital containing the unpaired electron.¹⁸ This requirement rationalizes the specificity observed in disproportionation of radicals **29** (Section 1.4.2) and provides an explanation for the persistency of the triisopropylmethyl radical (**33**) and related species (Section 1.4.3.2).¹⁶⁶ In the case of **33**, the β -hydrogens are constrained to lie in the nodal plane of the *p*-orbital due to steric buttressing between the methyls of the adjacent isopropyls.

It has been noted by a number of workers that the presence of α -substituents which delocalize the free spin favors combination over disproportionation.^{127,148,175} For radicals of structure (CH₃)₂C(•)-X, k_{td}/k_{tc} increases as shown in Figure 1.12. A correlation between the degree of exothermicity and the value of k_{td} has also been found but only for the case of resonance stabilized radicals.^{144,176,177}

| X = alkynyl ~ alkeny | l < aryl ~ nitrile < | keto < ester << alkyl |
|----------------------|----------------------|-----------------------|
|----------------------|----------------------|-----------------------|

| exothermicity | _ |
|-------------------------|---|
| delocalization | |
| $k_{\rm td}/k_{\rm tc}$ | _ |

Figure 1.12 Trend in k_{td}/k_{tc} for radicals (CH₃)₂C(•)-X.

It has been suggested that benzylic radicals may form a dimeric association complex which may easily collapse to the combination product but be geometrically unfavorable for disproportionation.^{178,179} Even if this applies for the aralkyl radicals, it cannot account for the behavior of systems with other p-substituents.

Another explanation follows from the above discussion on stereoelectronic factors.¹⁴⁵ If overlap between the semi-occupied orbital and the breaking C-H bond favors disproportionation, then substituents which delocalize the free spin will serve to reduce this interaction and disfavor disproportionation. A proposal along these lines was made originally by Nelson and Bartlett¹⁴⁸ who also noted that diminishment of the spin density at C_{α} could retard combination. However, it is not necessary that the two effects should cancel one another.

1.4.3.5 Reaction conditions

Values of k_{td}/k_{tc} for simple alkyl radicals are sensitive to reaction conditions (solvent, temperature, pressure). However, the effects appear to be generally small (<2-fold).^{142,144} Values of k_{td}/k_{tc} for *t*-butyl radicals in solution decrease with increasing temperature (the magnitude of the dependence increases with increasing solvent polarity - Figure 1.13) indicating a difference in activation energy of 3-12 kJ mol⁻¹. Smaller differences (1-2 kJ mol⁻¹) are seen for ethyl radicals.¹⁴² For a given solvent type (alkane or alcohol), a very small dependence on the viscosity of the medium is also observed (Table 1.8). The temperature dependence of k_{td}/k_{tc} has been related to the rate of molecular reorientation and the dependence on viscosity.¹⁸⁰ A very small decrease in k_{td}/k_{tc} with temperature is observed for **28** (Section 5.2.2.1.2).^{156,181} This small dependence of k_{td}/k_{tc} on temperature appears in marked contrast with the significant increase in k_{td}/k_{tc} with temperature reported for polymeric species (Section 5.2.2.2.2).

In studies of radical-radical reactions, radicals are typically generated pairwise and the products come from both cage and encounter (non-cage) reactions. Several studies have indicated that cage vs encounter product distributions are the same.¹⁵⁶ However, it has been suggested that influences of pressure and viscosity on k_{td}/k_{tc} are more substantial for radicals which undergo self-reaction within the solvent cage.¹⁴⁹

| Solvent | Temperature (°C) | η cP | $k_{\rm td}/k_{\rm tc}$ |
|---------------------|------------------|-----------|-------------------------|
| $n-C_8H_{18}$ | 25 | 0.51 | 5.4 |
| $n-C_{10}H_{22}$ | 25 | 0.86 | 5.7 |
| $n-C_{12}H_{26}$ | 25 | 1.37 | 5.9 |
| $n-C_{14}H_{30}$ | 25 | 2.10 | 6.4 |
| $n-C_{16}H_{34}$ | 25 | 3.09 | 6.9 |
| CH ₃ CN | 16.5 | 7.3 | 7.5 |
| t-BuOH-pinacol(1:2) | 25 | - | 10.1 |
| 3-methyl-3-pentanol | 24.5 | - | 7.5 |

Table 1.8 Values of k_{td}/k_{tc} for *t*-Butyl Radicals at 25 °C¹⁸⁰



Figure 1.13 Temperature dependence of k_{td}/k_{tc} values for *t*-butyl radicals with dodecane (—) or 3-methyl-3-pentanol (- - -) as solvent.

1.4.4 Summary

The relative importance of combination and disproportionation may be predicted using the following guidelines:¹⁴⁵

- (a) Radical-radical reactions involving carbon-centered radicals give products from both combination and disproportionation.
- (b) Simple primary and secondary radicals give predominantly combination. Tertiary radicals give some disproportionation.

(c) The importance of combination is increased by π -substituents at the radical center and decreased by bulky groups at or near the radical center.

1.5 References

- 1. Bamford, C.H. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 219.
- 2. Tedder, J.M. Angew. Chem., Int. Ed. Engl. 1982, 21, 401.
- 3. Tedder, J.M. Tetrahedron 1982, 38, 313.
- 4. Fischer, H.; Radom, L. Angew. Chem., Int. Ed. Engl. 2001, 40, 1340.
- 5. Dolbier, W.R., Jr. Chem. Rev. 1996, 96, 1557.
- 6. Kerr, J.A.; Stocker, D.W. In *CRC Handbook of Chemistry and Physics*; Lide, D.R., Ed.; CRC Press: Boca Raton, Florida, 2002; p (9)53.
- 7. Benson, S.W. Thermochemical Kinetics; Wiley: New York, 1976.
- Fischer, H. In Free Radicals; Kochi, J.K., Ed.; Wiley: New York, 1973; Vol. 2, p 435.
- 9. Leffler, J.E. An Introduction to Free Radicals; John Wiley & Sons: New York, 1993.
- 10. Walling, C. Free Radicals in Solution; Wiley: New York, 1957.
- 11. Hawthorne, D.G.; Johns, S.R.; Solomon, D.H.; Willing, R.I. Aust. J. Chem. 1976, 29, 1955.
- 12. Elson, I.H.; Mao, S.W.; Kochi, J.K. J. Am. Chem. Soc. 1975, 97, 335.
- 13. Tedder, J.M.; Walton, J.C. Acc. Chem. Res. 1976, 9, 183.
- 14. Rüchardt, C. Angew. Chem., Int. Ed. Engl. 1970, 9, 830.
- 15. Giese, B. Angew. Chem., Int. Ed. Engl. 1989, 28, 969.
- 16. Giese, B. Angew. Chem., Int. Ed. Engl. 1983, 22, 753.
- 17. Tedder, J.M.; Walton, J.C. Tetrahedron 1980, 36, 701.
- 18. Beckwith, A.L.J. Tetrahedron 1981, 37, 3073.
- 19. Ghosez-Giese, A.; Giese, B. ACS Symp. Ser. 1998, 685, 50.
- 20. Giese, B.; He, J.; Mehl, W. Chem. Ber. 1988, 121, 2063.
- 21. Beranek, I.; Fischer, H. In *Free Radicals in Synthesis and Biology*; Minisci, F., Ed.; Kluwer: Dordrecht, 1989; p 303.
- 22. Cuthbertson, M.J.; Rizzardo, E.; Solomon, D.H. Aust. J. Chem. 1985, 38, 315.
- 23. Jones, M.J.; Moad, G.; Rizzardo, E.; Solomon, D.H. J. Org. Chem. 1989, 54, 1607.
- 24. Moad, G.; Rizzardo, E.; Solomon, D.H. Aust. J. Chem. 1983, 36, 1573.
- 25. Giese, B.; Lachhein, S. Angew. Chem., Int. Ed. Engl. 1981, 20, 967.
- 26. Citterio, A.; Minisci, F.; Vismara, E. J. Org. Chem. 1982, 47, 81.
- 27. Rüchardt, C. Top. Curr. Chem. 1980, 88, 1.
- 28. Beckwith, A.L.J.; Ingold, K.U. In *Rearrangements in Ground and Excited States*; de Mayo, P., Ed.; Academic Press: New York, 1980; Vol. 1, p 162.
- 29. Wong, M.W.; Radom, L. J. Phys. Chem. 1995, 99, 8582.
- 30. Lorand, J.P. In Landoldt-Bornstein, New Series, Radical Reaction Rates in Solution; Fischer, II., Ed.; Springer-Verlag: Berlin, 1984; Vol. II/13a, p 135.
- 31. Roduner, E.; Crockett, R. In Landoldt-Bornstein, New Series, Radical Reaction Rates in Solution; Fischer, H., Ed.; Springer-Verlag: Berlin, 1995; Vol. II/18a, p 79.
- 32. Howard, J.A.; Scaiano, J.C. In Landoldt-Bornstein. New Series, Radical Reaction Rates in Solution; Fischer, H., Ed.; Springer-Verlag: Berlin, 1984; Vol. II/13d, p 5.
- 33. Lusztyk, J. In Landoldt-Bornstein. New Series, Radical Reaction Rates in Solution; Fischer, H., Ed.; Springer-Verlag: Berlin, 1995; Vol. II/18d1, p 1.

- 34. Hammond, G.S. J. Am. Chem. Soc. 1955, 77, 334.
- 35. Gonzalez, C.; Sosa, C.; Schlegel, H.B. J. Phys. Chem. 1989, 93, 2435.
- 36. Wong, M.W.; Radom, L. J. Phys. Chem. A 1998, 102, 2237.
- 37. Owen, G.E., Jr.; Pearson, J.M.; Szwarc, M. Trans. Faraday Soc. 1965, 61, 1722.
- 38. Giese, B.; Meister, J. Angew. Chem., Int. Ed. Engl. 1977, 16, 178.
- 39. Giese, B.; Meixner, J. Chem. Ber. 1981, 114, 2138.
- 40. Sakurai, H.; Hayashi, S.; Hosomi, A. Bull. Chem. Soc. Japan 1971, 44, 1945.
- 41. Pryor, W.A.; Tang, F.Y.; Tang, R.H.; Church, D.F. J. Am. Chem. Soc. 1982, 104, 2885.
- 42. Henderson, R.W.; Ward, R.D., Jr. J. Am. Chem. Soc. 1974, 96, 7556.
- 43. Pryor, W.A.; Tonellato, U.; Fuller, D.L.; Jumonville, S. J. Org. Chem. 1969, 34, 2018.
- 44. Sakurai, H.; Hosomi, A. J. Am. Chem. Soc. 1967, 89, 458.
- 45. Howard, J.A.; Chenier, J.H.B. J. Am. Chem. Soc. 1973, 95, 3054.
- 46. Michejda, C.J.; Hoss, W.P. J. Am. Chem. Soc. 1970, 92, 6298.
- 47. Huyser, E.S. J. Am. Chem. Soc. 1960, 82, 394.
- Avila, D.V.; Ingold, K.U.; Lusztyk, J.; Dolbier, W.R., Jr.; Pan, H.-Q.; Muir, M. J. Am. Chem. Soc 1994, 116, 99.
- 49. Pryor, W.A.; Lin, T.H.; Stanley, J.P.; Henderson, R.W. J. Am. Chem. Soc. 1973, 95, 6993.
- 50. Smart, B.E. In *Molecular Structure and Energetics*; Liebman, J.F.; Greenberg, A., Eds.; VCH: Deerfield Beach, Florida, 1976; Vol. 3, p 141.
- 51. Arnaud, R.; Subra, R.; Barone, V.; Lelj, F.; Olivella, S.; Solé, A.; Russo, N. J. Chem. Soc., Perkin Trans. 2 1986, 1517.
- 52. Canadell, E.; Eisenstein, O.; Ohanessian, G.; Poblet, J.M. J. Phys. Chem. 1985, 89, 4856.
- 53. Shaik, S.S.; Canadell, E. J. Am. Chem. Soc. 1990, 112, 1446.
- 54. Spellmeyer, D.C.; Houk, K.N. J. Org. Chem. 1987, 52, 959.
- 55. Beckwith, A.L.J.; Schiesser, C.H. Tetrahedron 1985, 41, 3925.
- 56. Beckwith, A.L.J.; Moad, G. J. Chem. Soc., Perkin Trans. 2 1980, 1083.
- 57. Beckwith, A.L.J. In Chem. Soc. Spec. Publ. Essays on Free Radical Chemistry; Chem. Soc.: London, 1970; Vol. 24, p 239.
- 58. Wilt, J.W. In *Free Radicals*; Kochi, J.K., Ed.; Wilcy: New York, 1973; Vol. 1, p 333.
- 59. Beckwith, A.L.J.; Easton, C.J.; Serelis, A.K. J. Chem. Soc., Chem. Commun. 1980, 482.
- 60. Beckwith, A.L.J. Chem. Soc. Rev. 1993, 143.
- 61. Curran, D.P.; Porter, N.A.; Giese, B. Stereochemistry of Radical Reactions; VCH: Weinheim, 1996.
- 62. Giese, B.; Kopping, B.; Göbel, T.; Dickhaut, J.; Thoma, G.; Kulicke, K.J.; Trach, F. *Organic Reactions* **1996**, *48*, 301.
- 63. Julia, M. Pure Appl. Chem. 1974, 40, 553.
- 64. Julia, M. Acc. Chem. Res. 1971, 4, 386.
- 65. Giese, B.; Feix, C. Isr. J. Chem. 1985, 26, 387.
- 66. Giese, B. Acc. Chem. Res. 1984, 17, 438.
- 67. Spirin, Y., L. Russ. Chem. Rev. (Engl. Transl.) 1969, 38, 529.
- 68. Kamachi, M. Adv. Polym. Sci. 1981, 38, 55.
- 69. Gromov, V.F.; Khomiskovskii, P.M. Russ. Chem. Rev. (Engl. Transl.) 1979, 48, 1040.
- 70. Huyser, E.S. Adv. Free Radical Chem. 1965, 1, 77.
- 71. Martin, J.C. In *Free Radicals*; Kochi, J.K., Ed.; Wiley: New York, 1973; Vol. 2, p 493.
- 72. Shostenko, A.G.; Myshkin, V.E. Dokl. Phys. Chem. (Engl. Transl.) 1979, 246, 569.
- 73. Giese, B.; Kretzschmar, G. Chem. Ber. 1984, 117, 3160.
- 74. Salikhov, A.; Fischer, H. Appl. Magn. Reson. 1993, 5, 445.
- Bednarek, D.; Moad, G.; Rizzardo, E.; Solomon, D.H. Macromolecules 1988, 21, 1522.
- 76. Ito, O.; Matsuda, M. J. Phys. Chem. 1984, 88, 1002.
- 77. Fong, C.W.; Kamlet, M.J.; Taft, R.W. J. Org. Chem. 1983, 48, 832.
- 78. Houk, K.N.; Padden-Row, M.N.; Spellmeyer, D.C.; Rondan, N.G.; Nagase, S. J. Org. Chem. 1986, 51, 2874.
- 79. Sosa, C.; Schlegel, H.B. J. Am. Chem. Soc. 1987, 109, 4193.
- Arnaud, R.; Barone, V.; Olivella, S.; Russo, N.; Solé, A. J. Chem. Soc., Chem. Commun. 1985, 1331.
- Delbecq, F.; Ilavsky, D.; Nguyen, T.A.; Lefour, J.M. J. Am. Chem. Soc. 1985, 107, 1623.
- 82. Heuts, J.P.A.; Gilbert, R.G.; Radom, L. Macromolecules 1995, 28, 8771.
- 83. Coote, M.L.; Davis, T.P.; Radom, L. Macromolecules 1999, 32, 2935.
- 84. Heuts, J.P.A.; Gilbert, R.G.; Maxwell, I.A. Macromolecules 1997, 30, 726.
- 85. Coote, M.L.; Davis, T.P.; Radom, L. Macromolecules 1999, 32, 5270.
- 86. Arnaud, R.; Vetere, V.; Barone, V. J. Comput. Chem. 2000, 21, 675.
- 87. Arnaud, R.; Vetere, V.; Barone, V. Chem. Phys. Lett. 1998, 293, 295.
- 88. Radom, L.; Wong, M.W.; Pross, A. ACS Symp. Ser. 1998, 685, 31.
- 89. Dewar, M.J.S.; Olivella, S. J. Am. Chem. Soc. 1978, 17, 5290.
- 90. Arnaud, R.; Douady, J.; Subra, R. Nouv. J. Chim 1981, 5, 181.
- 91. Bakken, G.A.; Jurs, P.C. J. Chem. Inf. Comput. Sci. 1999, 39, 1064.
- 92. Bakken, G.A.; Jurs, P.C. J. Chem. Inf. Comput. Sci. 1999, 39, 508.
- 93. Ito, O.; Matsuda, M. Prog. Polym. Sci. 1992, 17, 827.
- 94. Denisov, E.T. Russ. Chem. Rev. 2000, 69, 153.
- 95. Flemming, I. Frontier Orbitals and Organic Chemical Reactions; Wiley: Chichester, 1976.
- 96. Shaik, S.S.; Pross, A. Acc. Chem. Res. 1983, 16, 363.
- 97. Russell, G.A. In *Free Radicals*; Kochi, J.K., Ed.; Wiley: New York, 1973; Vol. 1, p 275.
- Poutsma, M.L. In Free Radicals; Kochi, J.K., Ed.; Wiley: New York, 1973; Vol. 2, p 113.
- 99. Hendry, D.G.; Mill, T.; Piszkiewicz, L.; Howard, J.A.; Eigenmann, H.K. J. Phys. Chem. Ref., Data 1974, 3, 937.
- 100. Evans, M.G.; Polanyi, M. Trans. Faraday Soc. 1938, 34, 11.
- 101. Pross, A.; Yamataka, H.; Nagase, S. J. Phys. Org. Chem. 1991, 4, 135.
- 102. Gilliom, R.D. J. Mol. Struct. 1986, 138, 157.
- 103. Pryor, W.A.; Davis, W.H., Jr. J. Am. Chem. Soc. 1974, 96, 7557.
- 104. Zavitsas, A.A.; Hanna, G.M. J. Org. Chem. 1975, 40, 3782.
- 105. Zavitsas, A.A.; Pinto, J.A. J. Am. Chem. Soc. 1972, 94, 7390.
- 106. Walling, C.; McGuinness, J.A. J. Am. Chem. Soc. 1969, 91, 2053.
- 107. Huang, X.L.; Dannenberg, J.J. J. Org. Chem. 1991, 56, 5421.
- 108. Houk, K.N.; Tucker, J.A.; Dorigo, A.E. Acc. Chem. Res. 1990, 23, 107.
- 109. Toh, J.S.S.; Huang, D.M.; Lovell, P.A.; Gilbert, R.G. Polymer 2001, 42, 1915.
- 110. Filley, J.; McKinnon, J.T.; Wu, D.T.; Ko, G.H. Macromolecules 2002, 35, 3731.
- 111. Walling, C.; Padwa, A. J. Am. Chem. Soc. 1963, 85, 1597.

Radical Reactions

- 112. Neale, R.S.; Walsh, M.R.; Marcus, N.L. J. Org. Chem. 1965, 30, 3683.
- 113. Busfield, W.K.; Grice, I.D.; Jenkins, I.D. J. Chem. Soc., Perkin Trans. 2 1994, 1079.
- 114. Busfield, W.K.; Grice, I.D.; Jenkins, I.D.; Monteiro, M.J. J. Chem. Soc., Perkin Trans. 2 1994, 1071.
- 115. Beckwith, A.L.J.; Easton, C.J. J. Chem. Soc., Perkin Trans. 2 1983, 661.
- 116. Malatesta, V.; Scaiano, J.C. J. Org. Chem. 1982, 47, 1455.
- 117. Beckwith, A.L.J.; Easton, C.J. J. Am. Chem. Soc. 1981, 103, 615.
- 118. Malatesta, V.; Ingold, K.U. J. Am. Chem. Soc. 1981, 103, 609.
- 119. Griller, D.; Howard, J.A.; Marriott, P.R.; Scaiano, J.C. J. Am. Chem. Soc. 1981, 103, 619.
- 120. Roberts, C.; Walton, J.C. J. Chem. Soc., Perkin Trans. 2 1985, 841.
- 121. Roberts, C.; Walton, J.C. J. Chem. Soc., Chem. Commun. 1984, 1109.
- 122. Bausch, M.J.; Gostowski, R.; Guadalupe-Fasano, C.; Selmarten, D.; Vaughn, A.; Wang, L.-H. J. Org. Chem. 1991, 56, 7191.
- 123. Kanabus-Kaminske, J.M.; Gilbert, B.C.; Griller, D. J. Am. Chem. Soc. 1989, 111, 3311.
- 124. Reichardt, C. Solvent Effects in Organic Chemistry; Verlag Chemie: Weinheim, 1978.
- 125. Russell, G.A. J. Am. Chem. Soc. 1958, 80, 4987.
- 126. Bunce, N.J.; Ingold, K.U.; Landers, J.P.; Lusztyk, J.; Scaiano, J.C. J. Am. Chem. Soc. 1985, 107, 5464.
- 127. Walling. J. Org. Chem. 1988, 53, 305.
- 128. Tanko, J.M.; Anderson, F.E., III. J. Am. Chem. Soc. 1988, 110, 3525.
- 129. Taylor, C.K.; Skell, P.S. J. Am. Chem. Soc. 1983, 105, 120.
- Skell, P.S.; Baxter, H.N.I., Tanko, J.M.; Chebolu, V. J. Am. Chem. Soc. 1986, 108, 6300.
- 131. Ponec, R.; Hajeck, J. Z. Phys. Chem. (Leipzig) 1987, 268, 1233.
- 132. Ingold, K.U.; Lusztyk, J.; Raner, K.D. Acc. Chem. Res. 1990, 23, 219.
- 133. Walling, C.; Wagner, P.J. J. Am. Chem. Soc. 1964, 86, 3368.
- 134. Mendenhall, G.D.; Stewart, L.C.; Scaiano, J.C. J. Am. Chem. Soc. 1982, 104, 5109.
- 135. Grant, R.D.; Griffiths, P.G.; Moad, G.; Rizzardo, E.; Solomon, D.H. Aust. J. Chem. 1983, 36, 2447.
- 136. Grant, R.D.; Rizzardo, E.; Solomon, D.H. Makromol. Chem. 1984, 185, 1809.
- Avila, D.V.; Brown, C.E.; Ingold, K.U.; Lusztyk, J. J. Am. Chem. Soc 1993, 115, 466.
- 138. Sheldon, R.A.; Kochi, J.K. J. Am. Chem. Soc. 1970, 92, 4395.
- 139. Bertrand, M.P.; Surzur, J.-M. Tetrahedron Lett. 1976, 17, 3451.
- 140. Houk, K.N. In *Frontiers in Free Radical Chemistry*; Pryor, W.A., Ed.; Academic Press: New York, 1980; p 43.
- 141. Abuin, E.; Mujica, C.; Lissi, E. Rev. Latinoamer. Quim. 1980, 11, 78.
- 142. Gibian, M.J.; Corley, R.C. Chem. Rev. 1973, 73, 441.
- Griller, D. In Landoldt-Bornstein, New Series, Radical Reaction Rates in Solution; Fischer, H., Ed.; Springer-Verlag: Berlin, 1984; Vol. II/13a, p 5.
- Alfassi, Z.B. In Chemical Kinetics of Small Organic Radicals; Alfassi, Z.B., Ed., CRC Press: Boca Raton, Fla., 1988; Vol. 1, p 129.
- 145. Moad, G.; Solomon, D.H. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 147.
- 146. Engel, P.S.; Chen, Y.; Wang, C. J. Org. Chem. 1991, 56, 3073.
- 147. Gleixner, G.; Olaj, O.F.; Breitenbach, J.W. Makromol. Chem. 1979, 180, 2581.

- 148. Nelsen, S.F.; Bartlett, P.D. J. Am. Chem. Soc. 1966, 88, 137.
- 149. Neuman, R.C., Jr.; Amrich, M.J., Jr. J. Org. Chem. 1980, 45, 4629.
- 150. Skinner, K.J.; Hochster, H.S.; McBride, J.M. J. Am. Chem. Soc. 1974, 96, 4301.
- 151. Langhals, H.; Fischer, H. Chem. Ber. 1978, 111, 543.
- 152. Barbc, W.; Rüchardt, C. Makromol. Chem. 1983, 184, 1235.
- 153. Zarkadis, A.K.; Neumann, W.P.; Dünnebacke, D.; Penenory, A.; Stapel, R.; Stewen, U. Chem. Ber. 1993, 126, 1179.
- 154. Neumann, W.P.; Stapel, R. Chem. Ber. 1986, 119, 3422.
- 155. Bartlett, P.D.; McBride, J.M. Pure Appl. Chem. 1967, 15, 89.
- 156. Bizilj, S.; Kelly, D.P.; Serelis, A.K.; Solomon, D.H.; White, K.F. Aust. J. Chem. 1985, 38, 1657.
- 157. Bradley, J.N.; Rabinovitch, B.S. J. Chem. Phys. 1962, 36, 3498.
- 158. Kerr, J.A.; Trotman-Dickenson, A.F. Prog. React. Kinet. 1961, 1, 107.
- 159. Benson, S.W. Acc. Chem. Res. 1986, 19, 335.
- 160. Dannenberg, J.J.; Baer, B. J. Am. Chem. Soc. 1987, 109, 292.
- 161. Imoto, M.; Sakai, S.; Ouchi, T. J. Chem. Soc. Japan 1985, 1, 97.
- 162. Minato, T.; Yamabe, S.; Fujimoto, H.; Fukui, K. Bull. Chem. Soc. Japan 1978, 51, 1.
- 163. Smith, W.B. Struct. Chem. 2001, 12, 213.
- 164. Beckhaus, H.D.; Rüchardt, C. Chem. Ber. 1977, 110, 878.
- 165. Fraenkel, G.; Geckle, M.J. J. Chem. Soc., Chem. Commun. 1980, 55.
- 166. Griller, D.; Ingold, K.U. Acc. Chem. Res. 1976, 9, 13.
- 167. Griller, D.; Marriot, P.R. Int. J. Chem. Kinet. 1979, 11, 1163.
- 168. Schlüter, K.; Berndt, A. Tetrahedron Lett. 1979, 20, 929.
- Griller, D.; Içli, S.; Thankachan, C.; Tidwell, T. J. Chem. Soc., Chem. Commun. 1974, 913.
- 170. Stefani, A.P. J. Am. Chem. Soc. 1968, 90, 1694.
- 171. Druliner, J.D.; Krusic, P.D.; Lehr, G.F.; Tolman, C.A. J. Org. Chem. 1985, 50, 5838.
- 172. Pritchard, G.O.; Johnson, K.A.; Nilsson, W.B. Int. J. Chem. Kinet. 1985, 17, 327.
- 173. Pritchard, G.O.; Nilsson, W.B.; Kirtman, B. Int. J. Chem. Kinet. 1984, 16, 1637.
- 174. Pritchard, G.O.; Kennedy, V.H.; Heldoorn, G.M.; Piasecki, M.L.; Johnson, K.A.; Golan, D.R. Int. J. Chem. Kinet. 1987, 19, 963.
- 175. Ingold, K.U. In *Free Radicals*; Kochi, J.K., Ed.; Wiley: New York, 1973; Vol. 1, p 37.
- 176. Manka, M.J.; Stein, S.E. J. Phys. Chem. 1984, 88, 5914.
- 177. Engel, P.S.; Wu, W.-X. J. Org. Chem. 1990, 55, 2720.
- 178. Neuman, R.C.; Alhadeff, E.S. J. Org. Chem. 1970, 35, 3401.
- 179. Kopecky, K.R.; Yeung, M.-Y. Can. J. Chem. 1988, 66, 374.
- 180. Schuh, H.; Fischer, H. Helv. Chim. Acta 1978, 61, 2463.
- 181. Trecker, D.J.; Foote, R.S. J. Org. Chem. 1968, 33, 3527.

3.1 Introduction

Initiation is defined as the series of reactions that commences with generation of *primary* radicals^{*} and culminates in addition to the carbon-carbon double bond of the monomer so as to form *initiating* radicals (Scheme 3.1).^{1,2}



Scheme 3.1

Classically, initiation was only considered as the first step in the chain reaction that constitutes radical polymerization. Although the rate and efficiency of initiation were known to be extremely important in determining the kinetics of polymerization, it was generally thought that the detailed mechanism of the process could be safely ignored when interpreting polymer properties. Furthermore, while it was recognized that initiation would lead to formation of structural units different from those which make up the bulk of the chain, the proportion of initiator-derived groups seemed insignificant when compared with total material.[†] This led to the belief that the physical properties and chemistry of polymers could be interpreted purely in terms of the generalized formula - *i.e.* (CH₂-CXY)_n (see Chapter 1).

This view prevailed until the early 1970s and can still be found in some current-day texts. It is only in recent times that we have begun to understand the complexities of the initiation process and can appreciate the full role of initiation

^{*} The term primary radical used in this context should be distinguished from that used when describing the substitution pattern of alkyl radicals.

[†] For example, in PS the initiator-derived end groups will account for *ca* 0.2% of units in a sample of molecular weight 100,000 (termination is mainly by combination).

in influencing polymer structure and properties. Four factors may be seen as instrumental in bringing about a revision of the traditional view:

- (a) The realization that polymer properties (*e.g.* resistance to weathering, thermal or photochemical degradation) are often not predictable on the basis of the repeat unit structure but are in many cases determined by the presence of "defect groups".³⁻⁶
- (b) The development of techniques whereby details of the initiation and other stages of polymerization can be studied in depth (Section 3.5).
- (c) The finding that radical reactions are typically under kinetic rather than thermodynamic control (Section 2.1). Many instances can be cited where the less thermodynamically favored pathway is a significant, or even the major, pathway.
- (d) The development of living or controlled radical polymerization (NMP, ATRP, RAFT, see Chapter 9). Lack of specificity in initiation can lead to dead chains and in turn to impure block copolymers or defects in complex architectures (stars, dendrimers, *etc.*).

It is the aim of this chapter to describe the nature, selectivity, and efficiency of initiation. Section 3.2 summarizes the various reactions associated with initiation and defines the terminology used in describing the process. Section 3.3 details the types of initiators, indicating the radicals generated, the byproducts formed (initiator efficiency), and any side reactions (*e.g.* transfer to initiator). Emphasis is placed on those initiators that see widespread usage. Section 3.4 examines the properties and reactions of the radicals generated, paying particular attention to the specificity of their interaction with monomers and other components of a polymerization system. Section 3.5 describes some of the techniques used in the study of initiation.

The intention is to create a greater awareness of the factors that must be borne in mind by the polymer scientist when selecting an initiator for a given polymerization.

3.2 The Initiation Process

The simple initiation process depicted in many standard texts is the exception rather than the rule. The yield of primary radicals produced on thermolysis or photolysis of the initiator is usually not 100%. The conversion of primary radicals to initiating radicals is dependent on many factors and typically is not quantitative. The primary radicals may undergo rearrangement or fragmentation to afford new radical species (secondary radicals) or they may interact with solvent or other species rather than monomer.

The reactions of the radicals (whether primary, secondary, solvent-derived, *etc.*) with monomer may not be entirely regio- or chemoselective. Reactions, such as head addition, abstraction or aromatic substitution, often compete with tail

addition. In the sections that follow, the complexities of the initiation process will be illustrated by examining the initiation of polymerization of two commercially important monomers, styrene (S) and methyl methacrylate (MMA), with each of three commonly used initiators, azobisisobutyronitrile (AIBN), dibenzoyl peroxide (BPO), and di-*t*-butyl peroxyoxalate (DBPOX). The primary radicals formed from these three initiators are cyanoisopropyl, benzoyloxy, and *t*-butoxy radicals respectively (Scheme 3.2). BPO and DBPOX may also afford phenyl and methyl radicals respectively as secondary radicals (see 3.2.2).





3.2.1 Reaction with Monomer

First consider the interaction of radicals with monomers. Some behave as described in the classic texts and give tail addition as the only detectable pathway (Scheme 3.3). However, tail addition to the double bond is only one of the pathways whereby a radical may react with a monomer. The outcome of the reaction is critically dependent on the structure of both radical and monomer.



Scheme 3.3

For reactions with S, specificity is found to decrease in the series cyanoisopropyl~methyl~t-butoxy>phenyl>benzoyloxy. Cyanoisopropyl (Scheme 3.3),⁷ t-butoxy and methyl radicals give exclusively tail addition.⁸ Phenyl radicals afford tail addition and *ca* 1% aromatic substitution.⁸ Benzoyloxy radicals give tail addition, head addition, and aromatic substitution (Scheme 3.4).^{8,9}



β-scission 1%

Scheme 3.4

With MMA, these radicals show a quite different order of specificity; regiospecificity decreases in the series cyanoisopropyl~methyl>phenyl >benzoyloxy>t-butoxy. Cyanoisopropyl and methyl radicals give exclusively tail addition. Benzoyloxy and phenyl radicals also react almost exclusively with the double bond (though benzoyloxy radicals give a mixture of head and tail addition¹⁰) and abstraction, while detectable, is a very minor (<1%) pathway.^{10,11} On the other hand, only 63% of *t*-butoxy radicals react with MMA by tail addition to give 1 (Scheme 3.5).¹² The remainder abstract hydrogen, either from the α -methyl (predominantly) to give 2 or the ester methyl to give 3.^{12,13} The radicals 1-3 and methyl (formed by β -scission) may then initiate polymerization.

These examples clearly show that the initiation pathways depend on the structures of the radical and the monomer. The high degree of specificity shown by a radical (*e.g. t*-butoxy) in its reactions with one monomer (*e.g.* S) must not be taken as a sign that a similarly high degree of specificity will be shown in reactions with all monomers (*e.g.* MMA).

Radicals can be classified according to their tendency to give aromatic substitution, abstraction, double bond addition, or β -scission and further classified in terms of the specificity of these reactions (see 3.4). With this knowledge, it should be possible to choose an initiator according to its suitability for use with a given monomer or monomer system so as to avoid the formation of undesirable end groups or, alternatively, to achieve a desired functionality.



β-scission 4%

Scheme 3.5

The importance of these considerations can be demonstrated by examining some of the possible consequences for radical-monomer systems. For the case of MMA polymerization initiated by a *t*-butoxy radical source, chains may be initiated by the radicals **1**, **2** or **3** (Scheme 3.5). A significant proportion of chains will therefore have an olefinic end group rather than an initiator-derived end group. These chain ends may be reactive, either during polymerization, leading to chain branching, or afterwards, possibly leading to an impairment in polymer properties (Section 8.2.2). Polystyrene (PS) formed with BPO as initiator will have a proportion of relatively unstable benzoate end groups formed by benzoyloxy radical reacting by head addition and aromatic substitution (Scheme 3.4).^{8,9} There is evidence that PS prepared with BPO as initiator is less thermally stable^{14,15} and less resistant to weathering and yellowing^{16,17} than that prepared using other initiators (Section 8.2.1).

3.2.2 Fragmentation

Many radicals undergo fragmentation or rearrangement in competition with reaction with monomer. For example, *t*-butoxy radicals undergo β -scission to form methyl radicals and acetone (Scheme 3.6).



| <" _ | L | |
|-------|------|-------|
| · · · | neme | ' i n |
| | | |

Benzoyloxy radicals decompose to phenyl radicals and carbon dioxide (Scheme 3.7).



Scheme 3.7

The reactivity of the monomer and the reaction conditions determine the relative importance of β -scission. Fragmentation reactions are generally favored by low monomer concentrations, high temperatures and low pressures. Their significance is greater at high conversion. They may also be influenced by the nature of the reaction medium.

Other radicals undergo rearrangement in competition with bimolecular processes. An example is the 5-hexenyl radical (5). The 6-heptenoyloxy radical (4) undergoes sequential fragmentation and cyclization (Scheme 3.8).¹⁸



The radicals formed by unimolecular rearrangement or fragmentation of the primary radicals are often termed secondary radicals. Often the absolute rate constants for secondary radical formation are known or can be accurately determined. These reactions may then be used as "radical clocks",^{19,20} to calibrate the absolute rate constants for the bimolecular reactions of the primary radicals (*e.g.* addition to monomers - see 3.4). However, care must be taken since the rate constants of some clock reactions (*e.g.* t-butoxy β -scission²¹) are medium dependent (see 3.4.2.1.1).

54

3.2.3 Reaction with Solvents, Additives, or Impurities

A typical polymerization system comprises many components besides the initiators and the monomers. There will be solvents, additives (*e.g.* transfer agents, inhibitors) as well as a variety of adventitious impurities that may also be reactive towards the initiator-derived radicals.

For the case of MMA polymerization with a source of *t*-butoxy radicals (DBPOX) as initiator and toluene as solvent, most initiation may be by way of solvent-derived radicals^{21,22} (Scheme 3.9). Thus, a high proportion of chains (>70% for 10% w/v monomers at 60 °C²²) will be initiated by benzyl rather than *t*-butoxy radicals. Other entities with abstractable hydrogens may also be incorporated as polymer end groups. The significance of these processes increases with the degree of conversion and with the (solvent or impurity):monomer ratio.



Scheme 3.9

There is potential for this behavior to be utilized in devising methods for the control of the types of initiating radicals formed and hence the polymer end groups.

3.2.4 Effects of Temperature and Reaction Medium on Radical Reactivity

The reaction medium may also modify the reactivity of the primary, or other radicals without directly reacting with them. For example, when *t*-butoxy reacts

with MMA (Scheme 3.5), the ratio of addition: $abstraction:\beta$ -scission varies according to the nature of the solvent²¹ and the reaction temperature^{23,24} (see 2.3.6 and 3.4.2.1.1).

For *t*-alkoxy radicals, polar and aromatic solvents favor abstraction over addition, and β -scission over either addition or abstraction (3.4.2.1.1). Addition, abstraction and β -scission have quite different Arrhenius parameters. As a further example the temperature dependence of the rate constants for addition of cumyloxy radicals to styrene, abstraction from isopropylbenzene, and β -scission to give methyl radicals is shown in Figure 3.1. Low temperatures favor abstraction over addition and both of these reactions over β -scission.



Figure 3.1 Temperature dependence of rate constants for reactions of cumyloxy radicals (a) β-scission to methyl radicals (— — —) (b) abstraction from isopropylbenzene (- - - -) and (c) addition to styrene (— —). Data are an extrapolation based on literature Arrhenius parameters.^{25,26} Adapted from Moad.²⁷

3.2.5 Reaction with Oxygen

Radicals, in particular carbon-centered radicals, react with oxygen at near diffusion-controlled rates.²⁸ Thus, for polymerizations carried out either in air or in incompletely degassed media, oxygen is likely to become involved in, and further complicate, the initiation process.

The reaction of oxygen with carbon-centered radicals (*e.g.* cyanoisopropyl, Scheme 3.10) affords an alkylperoxy radical $6^{29,30}$ This species may initiate polymerization so forming a relatively unstable peroxidic end group 7. With respect to most carbon-centered radicals, the alkylperoxy radicals 6 show an enhanced tendency to abstract hydrogen. The alkylperoxy radicals may abstract hydrogen from polymer, monomer, or other components in the system³¹ forming a potentially reactive hydroperoxide **8** and a new radical species (R•) which may initiate polymerization. The process is further complicated if 7 or **8** undergo

homolysis under the polymerization conditions. The peroxides derived from 7 and 8 may also be active as chain transfer agents.





3.2.6 Initiator Efficiency in Thermal Initiation

The proportion of radicals which escape the solvent cage to form initiating radicals is termed the initiator efficiency (f) which is formally defined as follows (eq. 1).^{*}

$$f = \frac{\text{(Rate of initiation of propagating chains)}}{n \text{ (Rate of initiator disappearance)}}$$
(1)

where n is the number of moles of radicals generated per mole of initiator.

The effective rate of initiation (R_i) in the case of thermal decomposition of an initiator (I_2) decomposing by Scheme 3.11 is given by eq. 2

| I_2 | $\xrightarrow{k_{d}}$ | $[2 I^{\bullet}] \longrightarrow I^{\bullet}$ | primary radical formation |
|------------------|-----------------------|---|-----------------------------|
| I∙ + M | $\xrightarrow{k_i}$ | $\mathbf{P}_1 \bullet$ | initiation |
| I∙ | | I'• | secondary radical formation |
| $I' \bullet + M$ | $\xrightarrow{k_i}$ | P₁• | initiation |

Scheme 3.11

$$R_{i} = k_{i}[I\bullet][M] + k_{i}'[I\bullet][M]$$

(2)

^{*} In some texts the initiator efficiency (*f*) is defined simply in terms of the yield of initiatorderived radicals (the fraction of radicals I• that undergo cage escape - Section 3.2.8). This number will always be larger than that obtained by application of eq. 1.

eq. 1 can then be written as follows (eq. 3)

$$f = \frac{(k_i [\mathbf{l} \bullet] [\mathbf{M}] + k_i' [\mathbf{l}' \bullet] [\mathbf{M}])}{2k_a [\mathbf{I}_2]}$$
(3)

If, as is usual, the k_i are not rate determining the rate of initiation is given by eq. 4.

$$R_{\rm i} = 2k_{\rm d} f[\mathbf{1}_2] \tag{4}$$

According to eq. 1, the term f should take into account all side reactions that lead to loss of initiator or initiator-derived radicals. These include cage reaction of the initiator-derived radicals (3.2.8), primary radical termination (3.2.9) and transfer to initiator (3.2.10). The relative importance of these processes depends on monomer concentration, medium viscosity and many other factors. Thus f is not a constant and typically decreases with conversion (see 3.3.1.1.3 and 3.3.2.1.3).

3.2.7 Photoinitiation

It is worthwhile to consider some of the special features of photoinitiation. The Jablonski diagram provides a convenient description of the events that follow absorption of light (Figure 3.2). A molecule in its ground state (S_0) absorbs a photon of light to be excited to the singlet state (S_1). As well as being electronically excited, the molecule will be vibrationally and rotationally excited. Certain reactions may take place from the excited singlet state. These will compete with fluorescence, and other deactivation processes that return the molecule to the ground state, and intersystem crossing to the triplet state (T_1). The triplet state is typically of lower energy than the excited singlet state. Chemical reaction then competes with phosphorescence and other deactivation processes.

Azo-compounds and peroxides undergo photodecomposition to radicals when irradiated with light of suitable wavelength. The mechanism appears similar to that of thermal decomposition to the extent that it involves cleavage of the same bonds. The photodecomposition of azo-compounds is discussed in Section 3.3.1.1.2 and peroxides in Sections 3.3.2.1.2 (diacyl peroxides) and 3.3.2.3.2 (peroxyesters). Specific photoinitiators are discussed in Section 3.3.4. It is also worth noting that certain monomers may undergo photochemistry and direct photoinitiation on irradiation of monomer is possible.

Clearly, unless monomer is the intended photoinitiator, it is important to choose an initiator that absorbs in a region of the UV-visible spectrum clear from the absorptions of monomer and other components of the polymerization medium. Ideally, one should choose a monochromatic light source that is specific for the chromophore of the photoinitiator or photosensitizer. It is also important in many experiments that the total amount of light absorbed by the sample is small. Otherwise the rate of initiation will vary with the depth of light penetration into the sample.



Figure 3.2 Jablonski diagram describing photoexcitation process.

In order to define the rate and efficiency of photoinitiation, consider the simplified reaction Scheme 3.12.

$$I_2 \xrightarrow{hv} 2I^{\bullet}$$
$$I^{\bullet} + M \xrightarrow{k_1} P_1^{\bullet}$$

Scheme 3.12

The quantum yield (Φ) is the yield of initiating radicals produced per photon of light absorbed (eq. 5)

$$\Phi = \frac{\text{(yield of initiating radicals)}}{n \text{ (photons absorbed)}}$$
(5)

which can also be expressed in terms of the rate of initiation (eq. 6).

$$\Phi = \frac{\text{(rate of initiation of propagating chains)}}{n \text{ (intensity of incident irradiation absorbed)}} = \frac{R_i}{nI_{abs}}$$
(6)

where *n* is the number of moles of radicals generated per mole of initiator and I_{abs} is the intensity of incident light absorbed.

The Beer-Lambert law (also often called Beer's law) relates I_{abs} to the total incident light intensity (I_0) (eq. 7).

$$\frac{I_{abs}}{I_0} = 1 - 10^{scd} = 1 - e^{\alpha cd}$$
(7)

and if αcd is small (<0.1 for <5% error) then this simplifies to eq. 8.

$$\frac{I_{\rm abs}}{I_0} \approx \alpha cd \tag{8}$$

where ε (= $\alpha/2.303$) is the molar extinction coefficient at the given wavelength, *c* is the concentration of the absorbing substance, and *d* is the pathlength. It can be seen that the term Φ embraces the same factors as $k_{d}f$ in thermal initiation. Care must be taken to establish how the molar extinction coefficient (ε or α) was determined since both decadic and natural forms are in common usage.

If the reaction with monomer is not the rate determining step, the rate of radical generation in photoinitiated polymerization is given by eq. 9

$$R_{i} = 2\Phi I_{abs}$$

= $2\Phi I_{0}(1 - e^{\alpha d[1,]}) = 2\Phi I_{0}(1 - 10^{\epsilon d[1,]})$ (9)

which for small $\alpha d[I_2]$ simplifies to eq. 10.

$$R_{\rm i} \approx 2\Phi I_0 \alpha d[I_2] \tag{10}$$

3.2.8 Cage Reaction and Initiator-Derived Byproducts

The decomposition of an initiator seldom produces a quantitative yield of initiating radicals. Most thermal and photochemical initiators generate radicals in pairs. The self-reaction of these radicals is often the major pathway for the direct conversion of primary radicals to non-radical products in solution, bulk or suspension polymerization. This cage reaction is substantial even in bulk polymerization at low conversion when the medium is essentially monomer. The importance of the process depends on the rate of diffusion of these species away from one another.

Thus, the size and the reactivity of the initiator-derived radicals and the medium viscosity (or microviscosity) are important factors in determining the initiator efficiency. Thus, the extent of the cage reaction is likely to increase with decreasing reaction temperature and with increasing conversion.^{32,33} The cage reaction, as well as lowering the initiation efficiency, can produce a range of byproducts. These materials may be reactive under the polymerization conditions or they may themselves have a deleterious influence on polymer properties. For example, the cage reaction of cyanoisopropyl radicals formed from the decomposition of AIBN produces, amongst other products (Scheme 3.13), MAN, which readily undergoes copolymerization to be incorporated into the final polymer,^{7,34} and tetramethylsuccinonitrile (9), which is claimed to be toxic and should not be present in polymers used for food contact applications.^{35,36}

In other cases, the cage reaction may simply lead to reformation of the initiator. This process is known as cage return and is important during the decomposition of BPO (Section 3.3.2.1.1) and DTBP (Section 3.3.2.4). Cage return lowers the rate of radical generation but does not directly yield byproducts. It is one factor contributing to the solvent and viscosity dependence of k_d and can lead to a reduced k_d at high conversion.

60

A variety of methods may be envisioned to decrease the importance of the cage reaction. One method, given the viscosity dependence of the cage reaction, is to conduct polymerizations in solution rather than in bulk. Another involves carrying out the polymerization in a magnetic field.³⁷ This is thought to reduce the rate of triplet-singlet intersystem crossing for the geminate pair.³⁸



Scheme 3.13

3.2.9 Primary Radical Termination

The primary radicals may also interact with other radicals present in the system after they escape the solvent cage. When this involves a propagating radical, the process is known as primary radical termination. The term also embraces the reactions of other initiator or transfer agent-derived radicals with propagating radicals. Most monomers are efficient scavengers of the initiator-derived radicals and the steady state concentration of propagating radicals is very low (typically $\leq 10^{-7}$ M). The concentrations of the primary and other initiator-derived radicals are very much lower (typically $\leq 10^{-9}$ M). Thus, with most initiators, primary radical termination has a very low likelihood during the early stages of polymerization.

Primary radical termination may involve combination or disproportionation with the propagating radical. It is often assumed that small radicals give mainly combination even though direct evidence for this is lacking. Both pathways are observed for reaction of cyanoisopropyl radicals with PS• (Scheme 3.14) (Section 7.4.3.2). The end group formed by combination is similar to that formed by head addition to monomer differing only in the orientation of the penultimate monomer unit.

If the rate of addition to monomer is low, primary radical termination may achieve greater importance. For example, in photoinitiation by the benzoin ether 12 both a fast initiating species (13, high k_i) and a slow initiating species (14, low

 k'_{i}) are generated (Scheme 3.15). The polymerization kinetics are complicated and the initiator efficiency is lowered by primary radical termination involving the dimethoxybenzyl radical (14, see 3.3.4.1.1).^{39,40}



Primary radical termination is also of demonstrable significance when very high rates of initiation or very low monomer concentrations are employed. It should be noted that these conditions pertain in all polymerizations at high conversion and in starved feed processes. Some syntheses of telechelics are based on this process (Section 7.5.1). Reversible primary radical termination by combination with a persistent radical is the desired pathway in many forms of living radical polymerization (Section 9.3).

3.2.10 Transfer to Initiator

Many of the initiators used in radical polymerization are susceptible to induced decomposition by various radical species. When the reaction involves the

propagating species, the process is termed transfer to initiator. The importance of this reaction depends on both the initiator and the propagating radical.

Diacyl peroxides are particularly prone to induced decomposition (Scheme 3.16). Transfer to initiator is of greatest importance for polymerizations taken to high conversion or when the ratio of initiator to monomer is high. It has been shown that, during the polymerization of S initiated by BPO, transfer to initiator can be the major pathway for the termination of chains.^{7,41}



Scheme 3.16

Transfer to initiator introduces a new end group into the polymer, lowers the molecular weight of the polymer, reduces the initiator efficiency, and increases the rate of initiator disappearance. Methods of evaluating transfer constants are discussed in Section 6.2.1.

3.2.11 Initiation in Heterogeneous Polymerization

Many polymerizations are carried out in heterogeneous media, usually watermonomer mixtures, where suspending agents or surfactants ensure proper dispersion of the monomer and control the particle size of the product.

Suspension polymerizations are often regarded as "mini-bulk" polymerizations since ideally all reaction occurs within individual monomer droplets. Initiators with high monomer and low water solubility are generally used in this application. The general chemistry, initiator efficiencies, and importance of side reactions are similar to that seen in homogeneous media.

Emulsion polymerizations most often involve the use of water-soluble initiators (*e.g.* persulfate see 3.3.2.6.1) and polymer chains are initiated in the aqueous phase. A number of mechanisms for particle formation and entry have been described, however, a full discussion of these is beyond the scope of this book. Readers are referred to recent texts on emulsion polymerization by Gilbert⁴² and Lovell and El-Aasser⁴³ for a more comprehensive treatment.

Radicals typically are generated in the aqueous phase and it is now generally believed that formation of an oligomer of average chain length z (z-mer, P_z^{\bullet}) occurs in the aqueous phase prior to particle entry.⁴⁴ The steps involved in forming a radical in the particle phase from an aqueous phase initiator are summarized in Scheme 3.17. The length of the z-mer depends on the particular monomer and is shorter for more hydrophobic monomers.

| initiator(aq) $\longrightarrow I^{\bullet}(aq)$ | (initiator-derived radical in aqueous phase) |
|---|--|
| $\xrightarrow{M} P_1 \bullet (aq)$ | (initiating radical in aqueous phase) |
| $\xrightarrow{M} P_z \bullet (aq)$ | (z-1 monomer additions to give z-mer) |
| $ \xrightarrow{\rho} P_{z} \bullet(p) $ | (transfer to particle phase, entry) |

Scheme 3.17

The concentration of monomers in the aqueous phase is usually very low. This means that there is a greater chance that the initiator-derived radicals (1•) will undergo side reactions. Processes such as radical-radical reaction involving the initiator-derived and oligomeric species, primary radical termination, and transfer to initiator can be much more significant than in bulk, solution, or suspension polymerization and initiator efficiencies in emulsion polymerization are often very low. Initiation kinetics in emulsion polymerization are defined in terms of the entry coefficient (ρ) - a pseudo-first order rate coefficient for particle entry.

Microemulsion and miniemulsion polymerization differ from emulsion polymerization in that the particle sizes are smaller (10-30 and 30-100 nm respectively vs 50-300 nm)⁴² and there is no monomer droplet phase. All monomer is in solution or in the particle phase. Initiation takes place by the same process as conventional emulsion polymerization.

3.3 The Initiators

Certain polymerizations (e.g. S, see 3.3.6.1) can be initiated simply by applying heat; the initiating radicals are derived from reactions involving only the monomer. More commonly, the initiators are azo-compounds or peroxides that are decomposed to radicals through the application of heat, light, or a redox process.

When initiators are decomposed thermally, the rates of initiator disappearance (k_d) show marked temperature dependence. Since most conventional polymerization processes require that k_d should lie in the range 10^{-6} - 10^{-5} s⁻¹ (half-life *ca* 10 h), individual initiators typically have acceptable k_d only within a relatively narrow temperature range (*ca* 20-30 °C). For this reason initiators are often categorized purely according to their half-life at a given temperature or *vice versa*.⁴⁵ For initiators which undergo unimolecular decomposition, the half-life is related to the decomposition rate constant by eq. 11.

$$t_{1} = \frac{\ln 2}{k_{d}} \tag{11}$$

The Arrhenius relationship can be rearranged as follows (eq. 12) to enable calculation of the temperature required to give a desired decomposition rate or half-life.

$$T(^{\circ}\mathrm{C}) = -273.15 - \frac{E_{\mathrm{a}}}{R \ln \left(\frac{k_{\mathrm{d}}}{A}\right)} = -273.15 - \frac{E_{\mathrm{a}}}{R \ln \left(\frac{\ln 2}{At_{\frac{1}{2}}}\right)}$$
(12)

The temperature at which the half-life is 10h is then given by the following expression (eq. 13).

$$T(^{\circ}C) = -273.15 - \frac{0.120277 E_{a}}{-10.8578 + \ln\left(\frac{1}{A}\right)}$$
(13)

The initiator in radical polymerization is often regarded simply as a source of radicals. Little attention is paid to the various pathways available for radical generation or to the side reactions that may accompany initiation. The preceding discussion (see 3.2) demonstrated that in selecting initiators (whether thermal, photochemical, redox, *etc.*) for polymerization, they must be considered in terms of the types of radicals formed, their suitability for use with the particular monomers, solvent, and the other agents present in the polymerization medium, and for the properties they convey to the polymer produced.

Many reviews detailing aspects of the chemistry of initiators and initiation have appeared.^{2,45,46} A non-critical summary of thermal decomposition rates is provided in the *Polymer Handbook*.^{47,48} The subject also receives coverage in most general texts and reviews dealing with radical polymerization. References to reviews that detail the reactions of specific classes of initiator are given under the appropriate sub-heading below.

Some characteristics of initiators used for thermal initiation are summarized in Table 3.1. These provide some general guidelines for initiator selection. In general, initiators which afford carbon-centered radicals (*e.g.* dialkyldiazenes, aliphatic diacyl peroxides) have lower efficiencies for initiation of polymerization than those that produce oxygen-centered radicals. Exact values of efficiency depend on the particular initiators, monomers, and reaction conditions. Further details of initiator chemistry are summarized in Sections 3.3.1 (azo-compounds) and 3.3.2 (peroxides) as indicated in Table 3.1. In these sections, we detail the factors which influence the rate of decomposition (*i.e.* initiator structure, solvent, complexing agents), the nature of the radicals formed, the susceptibility of the initiator to induced decomposition, and the importance of transfer to initiator and other side reactions of the initiator or initiation system. The reactions of radicals produced from the initiator are given detailed treatment in Section 3.4.

 Table 3.1 Guide to Properties of Polymerization Initiators

| Initiator Class | Example | Section |
|-----------------------------|---|---------|
| dialkyldiazenes | $\begin{array}{ccc} CN & CN \\ H_3C-\overset{C}{C}-N=N-\overset{C}{C}-CH_3 \\ \overset{C}{CH_3} & \overset{C}{CH_3} & \mathbf{AIBN} \end{array}$ | 3.3.1.1 |
| hyponitrites | $\begin{array}{ccc} CH_3 & CH_3 \\ H_3C-C-O-N=N-O-C-CH_3 \\ CH_3 & CH_3 \end{array}$ | 3.3.1.2 |
| diacyl peroxides | ООСН ₃ (CH ₂) ₉ CH ₂ -С-О-О-С-СН ₂ (CH ₂) ₉ CH ₃ LPO | 3.3.2.1 |
| diaroyl peroxides | O O H Ph·C-O-O-C·Ph BPO | 3.3.2.1 |
| peroxydicarbonates | H ₃ C O O CH ₃ CH-O-C-O-O-C-O-CH H ₃ C CH ₃ | 3.3.2.2 |
| peroxyesters | $\begin{array}{c} CH_3 & O CH_3 \\ H_3C-\overset{C}{C}-O-O-\overset{C}{C}-\overset{C}{C}-CH_3 \\ \overset{C}{CH}_3 & \overset{C}{CH}_3 \end{array}$ | 3.3.2.3 |
| peroxyoxalates | СН ₃ О СН ₃ Н ₃ С-С-О-О-С-С-О-О-С-СН ₃ СН ₃ СН ₃ DBPOX | 3.3.2.3 |
| dialkyl peroxides | $ \begin{array}{ccc} CH_3 & CH_3 \\ H_3C-C-O-O-C-CH_3 \\ CH_3 & CH_3 & \mathbf{D}\mathbf{T}\mathbf{B}\mathbf{P} \end{array} $ | 3.3.2.4 |
| dialkyl ketone peroxides | O-O-C(CH ₃) ₃ O-O-C(CH ₃) ₃ | 3.3.2.5 |
| hydroperoxides | СН ₃ Н ₃ С-С-О-О-Н СН ₃ | 3.3.2.5 |
| persulfate | O -O-S-O-O-S-O- O O O | 3.3.2.6 |
| disulfides | $ \begin{array}{cccc} C_{2}H_{5} & S & C_{2}H_{5} \\ C_{2}H_{5} & N-C-S-S-C-N & C_{2}H_{5} \end{array} $ | 3.3.5 |

a 1° = primary radical from initiator decomposition, 2° = secondary radical-derived by fragmentation of 1° radical. Species shown in parentheses may be formed under some conditions but are seldom observed in polymerizations of common monomers.

Table 3.1 (continued)

| Rac | licals generated ^a | Efficiency ^b | Transfer ^e |
|-----------|-------------------------------|-------------------------|-----------------------|
| 1° | alkyl | low | low |
| 1° 2° | alkoxy alkyl | high | low |
| (1° 2° | acyloxy) alkyl | low | high |
| 1° 2° | aroyloxy aryl | high | high |
| ⊺° (2° | alkoxycarbonyloxy alkoxy) | high | high |
| 1° 2° | alkoxy, acyloxy alkyl | med. | med. |
| 1° 2° | alkoxy alkyl | high | med. |
| I° 2° | alkoxy alkyl | high | low |
| l° 2° | alkoxy alkyl | med. | low |
| 1° 2° | hydroxy, alkoxy alkyl | high | high |
| Ι° | sulfate radical anion | low | low |
| Ι° | thiyl | high | high |

b Efficiency decreases as the importance of cage reactions increases. c Susceptibility to radical-induced decomposition.

3.3.1 Azo-Compounds

Two general classes of azo-compound will be considered in this section, the dialkyldiazenes (15) (3.3.1.1) and the dialkyl hyponitrites (16) (3.3.1.2).

Polymeric azo-compounds and multifunctional initiators with azo-linkages are discussed elsewhere (see 3.3.3 and 7.6.1) as are azo compounds, which find use as iniferters (see 9.3.4).

3.3.1.1 Dialkyldiazenes

The kinetics and mechanism of the thermal and photochemical decomposition of dialkyldiazenes (15) have been comprehensively reviewed by Engel.⁴⁹ The use of these compounds as initiators of radical polymerization has been covered by Moad and Solomon² and Sheppard.⁵⁰ The general chemistry of azo-compounds has also been reviewed by Koga *et al.*,⁵¹ Koenig,⁵² and Smith.⁵³

Dialkyldiazenes (15, R-alkyl) are sources of alkyl radicals. While there is clear evidence for the transient existence of diazenyl radicals (17; Scheme 3.18) during the decomposition of certain unsymmetrical diazenes^{49,51} and of *cis*-diazenes,⁵⁴ all isolable products formed in thermolysis or photolysis of dialkyldiazenes (15) are attributable to the reactions of alkyl radicals.

$$R-N=N-R' \longrightarrow \left[R-N=N\cdot\right] + \cdot R' \longrightarrow R\cdot + N\equiv N + \cdot R'$$

$$15 \qquad 17$$

Scheme 3.18

In the decomposition of symmetrical azo compounds the intermediacy of diazenyl radicals remains a subject of controversy. However, it is clear that diazenyl radicals, if they are intermediates, do not have sufficient lifetime to be trapped or to initiate polymerization. Ayscough *et al.*⁵⁵ photolyzed AIBN in a matrix at -196 °C and observed EPR signals which were attributed to the diazenyl radical, $(CH_3)_2(CN)C$ N=N• [this assignment has been questioned⁵¹]. However for AIBN decomposition in solution, at temperatures normally encountered in polymerizations, the finding, that the rate of decomposition is independent of solvent viscosity (*i.e.* no cage return) is evidence for concerted 2-bond cleavage.³¹ Commercially available dialkyldiazenes initiators (15) tend to be symmetrical and the R groups are generally tertiary with functionality to stabilize the incipient radical [*e.g.* cyano AIBN, (18-29), ester (AIBMe), amidinium salt (22, 23), amide (24, 25) or phenyl (21)]. Those most commonly encountered are the azonitriles, these include 2,2'-azobis(2-methylpropanenitrile) [better known as azobis-

(isobutyronitrile) or AIBN], 2,2'-azobis(2-methylbutanenitrile) (19), 1,1'-azobis(1cyclohexanenitrile) (20)). The initiator 18 exists as a mixture of diastereoisomers that have differing k_d (Table 3.2). Azoisooctane 26 and azo-*t*-butane 27 are high temperature initiators.



| Initiator | R (, R') | Solvent | Temp. range °C |
|-------------|--|-----------------|-------------------|
| diazenes (1 | 5) | | |
| 21 | (CH ₃) ₂ C(Ph) | toluene | 40-70(17) |
| 18 | (CH ₃) ₂ CHCH ₂ C(CH ₃)(CN) [€] | toluene | 60-70(2) |
| 18 | (CH ₃) ₂ CHCH ₂ C(CH ₃)(CN) [€] | toluene | 70-80(2) |
| AIBN | (CH ₃) ₂ C(CN) | benzene/toluene | 37-105(13) |
| AIBMe | (CH ₃) ₂ C(CO ₂ CH ₃) | benzene | 50-70(4) |
| 19 | $(CH_3)(C_2H_5)C(CN)$ | ethylbenzene | 80-100(3) |
| 20 | (<i>c</i> -C ₆ H ₁₀)C(CN) | toluene | 80-100(3) |
| 26 | (CH ₃) ₃ CHCH ₂ C(CH ₃) ₂ | diphenyl ether | 130-160(7) |
| 27 | (CH ₃) ₃ C | diphenyl ether | 165-200(6) |
| 30 | (Ph) ₃ C, Ph | benzene/toluene | 25-75(9) |
| hyponitrite | s (16) | | |
| 34 | (CH ₃) ₃ C | isooctane | 45-75(4) |
| 35 | (CH ₃) ₂ (Ph)C | cyclohexane | 40-70(12) |

| Table 3.2 | Selected | Kinetic | Data | for | Decomposi | tion of | Azo-Co | ompounds" |
|-----------|----------|---------|------|-----|-----------|---------|--------|-----------|
| | | | | | | | | |

a Arrhenius parameters recalculated from original data taken from the indicated references. Values of E_a and A rounded to 4 and 3 significant figures respectively. b Number of data points given in parentheses. c Calculated from the Arrhenius parameters shown and rounded to 2 significant figures. d Temperature for ten hour half-life calculated with eq. 13. e Diastereoisomers.

| E_a | $A \times 10^{-15}$ | $k_{\rm d} \times 10^6 (60 {}^{\circ}{\rm C})^{\rm c}$ | $10 \text{ h} t_{1/2}^{d}$ | Ref. | Initiator |
|----------------------|---------------------|--|----------------------------|----------|-----------------|
| kJ mol ⁻¹ | s ⁻¹ | s ⁻¹ | °C | | |
| 126.7 | 12.2 | 170 | 45 | 56 | 21 |
| 118.9 | 0.376 | 86 | 49 | 57 | 18 ^e |
| 123.7 | 1.39 | 56 | 52 | 57 | 18 ^e |
| 131.7 | 4.31 | 9.6 | 65 | 32,58-61 | AIBN |
| 124.0 | 0.248 | 8.9 | 66 | 62.63 | AIBMe |
| 137.8 | 20.3 | 5.0 | 69 | 64 | 19 |
| 149.1 | 71.0 | 0.30 | 88 | 65 | 20 |
| 137.0 | 0.10 | 0.033 | 109 | 66 | 26 |
| 180.4 | 91.7 | ~5×10 ⁻⁶ | 161 | 67 | 27 |
| 114.6 | 0.486 | 522 | 35 | 68-70 | 30 |
| 119.5 | 1 17 | 214 | 42 | 71 | 34 |
| 113.9 | 0.99 | 1370 | 29 | 72 | 35 |

Table 3.2 (continued)

Water-soluble azo compounds include 4,4'-azobis(4-cyanovaleric acid) (29) and the amidinium hydrochlorides (22 and 23).

$$\begin{array}{ccc} CN & CN \\ HOH_2C-CH_2-CH_2-CH_2-C-N=N-C-CH_2-CH_2-CH_2OH \\ CH_3 & CH_3 \\ & \mathbf{28} \\ HO_2C-CH_2-CH_2-C-N=N-C-CH_2-CH_2-CO_2H \\ CH_3 & CH_3 \\ & \mathbf{29} \end{array}$$

Unsymmetrical azo-compounds find application as initiators of polymerization in special circumstances, for example, as initiators of living radical polymerization [e.g. triphenylmethylazobenzene (**30**) (see 9.3.4)], as hydroxy radical sources [e.g. α -hydroperoxydiazene (**31**) (see 3.3.3.1)], for enhanced solubility in organic solvents [e.g. t-butylazocyclohexanecarbonitrile (**32**)], or as high temperature initiators [e.g. t-butylazoformamide (**33**)]. They have also been used as radical precursors in model studies of cross-termination in copolymerization (Section 7.4.3).



3.3.1.1.1 Thermal decomposition

While some details of the kinetics of radical production from dialkyldiazenes remain to be unraveled, their decomposition mechanism and behavior as polymerization initiators are largely understood. Kinetic parameters for some common azo-initiators are presented in Table 3.2.

Thermolysis rates (k_d) of dialkyldiazenes (15) show a marked dependence on the nature of R (and R'). The values of k_d increase in the series where R (=R') is aryl<primary<secondary<tertiary<allyl. In general, k_d is dramatically accelerated by α -substituents capable of delocalizing the free spin of the incipient radical.⁴⁹ For example, Timberlake⁷³ has found that for the case of dialkyldiazenes,

72

X-C(CH₃)₂-N=N-C(CH₃)₂-X that k_d increases in the series where X is CH₃<-OCH₃ <-SCH₃<-CO₂R~-CN<-Ph<-CH=CH₂ (see also Table 3.2). These results can be rationalized in terms of the relative stability of the radicals generated (R•, R'•).

However, steric factors are also important.⁷⁴ Rüchardt and coworkers showed, for a series of acyclic alkyl derivatives, that a good correlation exists between k_d and ground state strain.^{75,76} Additional factors are important for bicyclic and other conformationally constrained azo-compounds.^{49,51,77} Woll⁷⁸ has described a scheme for calculating k_d based on radical stability (HOMO π -delocalization energies) and ground state strain (steric parameters).

There have been numerous studies on the kinetics of decomposition of AIBN, AIBMe and other dialkyldiazenes.⁴⁶ Solvent effects on k_d are small by conventional standards but, nonetheless, significant. Data for AIBMe is presented in Table 3.3. The data come from a variety of sources and can be seen to increase in the series where the solvent is: aliphatic < ester (including MMA) < aromatic (including styrene) < alcohol. There is a factor of two difference between k_d in methanol and k_d in ethyl acetate. The value of k_d for AIBN is also reported to be higher in aromatic than in hydrocarbon solvents and to increase with the dielectric constant of the medium.^{31,79,80} The k_d of AIBMe and AIBN show no direct correlation with solvent viscosity (see also 3.3.1.1.3), which is consistent with the reaction being irreversible (*i.e.* no cage return).

Thermolysis rates are enhanced substantially by the presence of certain Lewis acids (*e.g.* boron and aluminum halides), and transition metal salts (*e.g.* Cu^{2^-} , Ag').⁴⁶ There is also evidence that complexes formed between azo-compounds and Lewis acids (*e.g.* ethyl aluminum sesquichloride) undergo thermolysis or photolysis to give complexed radicals which have different specificity to uncomplexed radicals.⁸¹⁻⁸³

| $k_d \times 10^5$ | Solvent | Temperature | Reference |
|-------------------|--------------------|-------------|-----------|
| s ⁻¹ | | °C | |
| 0.58 | cyclohexane | 60.0 | 62 |
| 0.72 | ethyl acetate | 60.0 | 63 |
| 0.74 | methyl isobutyrate | 60.0 | 63 |
| 0.83 | 1:1 MMA/S | 60.0 | 84 |
| 1.18^{a} | aliphatic esters | 60.0 | 85 |
| 0.88 | benzene | 60.0 | 62 |
| 0.91 | benzene | 60.0 | 63 |
| 1.01 | acetonitrile | 60.0 | 62 |
| 1.13 | S | 60.0 | 86 |
| 1.20 | methanol | 60.0 | 63 |
| 1.44 | methanol | 60.0 | 62 |

 Table 3.3 Solvent Dependence of Rate Constants for AIBMe Decomposition^a

a Calculated from the expression given: $\ln(k_d)=33.1-(14800/T)$; said to be valid for a range of aliphatic ester solvents including MMA.

3.3.1.1.2 Photochemical decomposition

74

The *trans*-dialkyldiazenes have λ_{max} 350-370 nm and ε 2-50 M⁻¹ cm⁻¹ and are photolabile. They are, therefore, potential photoinitiators.^{49,87} The efficiency and rate of radical generation depends markedly on structure.⁴⁹ Dialkyldiazenes are often depicted without indicating the stereochemistry about the nitrogen-nitrogen double bond. However, except when constrained in a ring system, the dialkyldiazenes can be presumed to have the *trans*-configuration.

Alicyclic *cis*-dialkyldiazenes are very thermolabile when compared to the corresponding *trans*-isomers, often having only transient existence under typical reaction conditions. It has been proposed⁴⁹ that the main light-induced reaction of the dialkyldiazenes is *trans-cis* isomerization. Dissociation to radicals and nitrogen is then a thermal reaction of the *cis*-isomer (Scheme 3.19).



Scheme 3.19

Therefore, the quantum yield for photoisomerization approximates that for nitrogen formation and both are typically $ca \ 0.5$. Where the *cis* isomer is thermally stable, quantum yields for initiator disappearance are low ($\phi < 0.1$).⁴⁹

An important ramification of the photolability of azo-compounds is that, when using dialkyldiazenes as thermal initiators, care must be taken to ensure that the polymerization mixture is not exposed to excessive light during its preparation.

3.3.1.1.3 Initiator efficiency

The proportion of 'useful' radicals generated from common dialkyldiazenes is never quantitative; typically it is the range 50-70% in media of low viscosity (*i.e.* in low conversion polymerizations).^{32,88,89} The main cause of this inefficiency is loss of radicals through self-reaction within the solvent cage.

For dialkyldiazenes where the α -positions are not fully substituted, tautomerization to the corresponding hydrazone may also reduce the initiator efficiency⁹⁰ (Scheme 3.20). This rearrangement is catalyzed by light and by acid.



There is also evidence for a radical-induced mechanism involving initial hydrogen abstraction (Scheme 3.21).





Conflicting statements have appeared on the sensitivity of f to the nature of the monomer involved. Braun and Czerwinski⁹¹ reported that for low conversion polymerizations, f is essentially the same in MMA, S, and NVP. Fukuda *et al.*⁹² reported that f varies between MMA and S. The solvent dependence of k_d may account for this apparent conflict (Table 3.3).

While the rate of azo-compound decomposition shows only a small dependence on solvent viscosity, the amount of cage reaction (and hence f) varies dramatically with the viscosity of the reaction medium and hence with factors that determine the viscosity (conversion, temperature, solvent, *etc.*).³¹

Most values of f have been measured at zero or low conversions. During polymerization the viscosity of the medium increases and the concentration of monomer decreases dramatically as conversion increases (*i.e.* as the volume fraction of polymer increases). The value of f is anticipated to drop accordingly.^{32,33,93-96} For example, with S polymerization in 50% (v/v) toluene at 70 °C initiated by 0.1 M AIBN the 'instantaneous' f was determined to vary from 76% at low conversion to <20% at 90-95% conversion (Figure 3.3).³² The assumption that the rate of initiation ($k_d f$) is invariant with conversion (common to most pre 1990s and many recent kinetic studies of radical polymerization) cannot be supported.

The viscosity dependence of f may lead to the initiator efficiency being dependent on the molecular weight of the polymer being produced. This, in turn, is a function of the initiator and monomer concentration. For example, initiator efficiencies are expected to be higher during oligomer synthesis than in preparation of high molecular weight polymer. Initiator efficiency has also been shown to depend on the size of the initiator-derived radicals.³³ There is an inverse relationship between the rate of escape from the solvent cage and radical size.

Initiator efficiency increases with reaction temperature (Table 3.4). It is also worth noting that apparent zero-conversion initiator efficiencies depend on the method of measurement. Better scavengers trap more radicals. The data in Table 3.4 suggest that monomers (MMA, S) are not as effective at scavenging radicals as the inhibitors used to measure initiator efficiencies. The finding suggests that in polymerization the initiator-derived radicals have a finite probability of undergoing self-reaction after they escape the solvent cage and numbers obtained by the inhibitor method should be considered as upper limits.



Figure 3.3 Cumulative (□) and instantaneous (◆) initiator efficiency (*f*) of AIBN as initiator in S polymerization (50% v/v toluene, 70 °C) as a function of monomer conversion (lines are a polynomial fit to the datapoints).^{1,32}

| f | Scavenger | Temperature | Solvent | Reference |
|---------------------|-------------------|-------------|----------------|-----------|
| 0 | Ũ | °С | | |
| 0.81 ^a | none | 98 | S | 86 |
| 0.72 ^a | none | 90 | S | 86 |
| 0.77 | galvinoxyl | 90 | chlorobenzene | 97 |
| 0.76 | nitroxide | 80 | chlorobenzene | 98 |
| 0.70^{b} | triphenylverdazyl | 80 | MMA | 85 |
| 0.63 ^a | none | 80 | S | 86 |
| $0.68-0.60^{\circ}$ | none | 60 | MMA/S/ benzene | 84 |
| 0.56b | triphenvlverdazvl | 60 | MMA | 85 |
| 0.48 | DPPH | 60 | not specified | 99 |
| 0.45 | none | 60 | benzene | 63 |
| 0.40^{a} | none | 60 | S | 86 |

 Table 3.4 Zero-Conversion Initiator Efficiency (f) for AIBMe under Various Reaction Conditions

a Estimated by analysis of polymerization kinetics. b Calculated using the expression $\ln f = 0.58$ -(330/T).⁸⁵ c [Polymer end groups]/[total products] with AIBMe- α -¹³C as initiator. Overall efficiency reduces from 0.68 at <16% conversion to 0.60 at 95% conversion (Figure 3.3).

The byproducts of decomposition of certain dialkyldiazenes can be a concern. Consider the case of AIBN decomposition (Scheme 3.13). The major byproduct is the ketenimine (10). $^{61,100-102}$ This compound is itself thermally labile and reverts to cyanoisopropyl radicals at a rate constant similar to that for AIBN thermolysis. 59,60,102 This complicates any analysis of the kinetics of initiation. 32,60

Another concern, is the potential reactivity of **10** as a transfer agent under polymerization conditions (see 3.3.1.1.4).¹⁰³ Tetramethylsuccinonitrile (**9**) appears to be essentially inert under polymerization conditions.^{*} However, the compound is reported to be toxic and may be a problem in polymers used in food contact applications.^{35,36} Methacrylonitrile (MAN) formed by disproportionation readily copolymerizes.^{7,34} The copolymerized MAN may affect the thermal stability of polymers. A suggestion¹⁰⁵ that copolymerized MAN may be a "weak link" in PS initiated with AIBN has been disputed.¹⁴

Some of the complications associated with the use of AIBN may be avoided by use of alternative azo-initiators. Azobis(methyl isobutyrate) (AIBMe) has a decomposition rate only slightly less than AIBN and has been promoted for use in laboratory studies of polymerization⁸⁵ because the kinetics and mechanism of its decomposition kinetics are not complicated by ketenimine formation.

The azonitrile **19** also shows similar decomposition kinetics to AIBN (Table 3.2). The initiators **19** and AIBMe also have greater solubility in organic solvents than AIBN.

3.3.1.1.4 Transfer to initiator

Dialkyldiazenes are often preferred over other (peroxide) initiators because of their lower susceptibility to induced decomposition. The importance of transfer to initiator during polymerizations initiated by AIBN has been the subject of some controversy. While the early work of Baysal and Tobolsky,¹⁰⁶ Bevington and Lewis¹⁰⁷ and others suggested that transfer to initiator was insignificant during polymerizations of MMA or S, a number of subsequent studies on polymerization kinetics report a significant transfer constant ($C_1 \ ca \ 0.1$).^{104,108-112} Studies of S polymerization initiated by ¹³C-labeled AIBN demonstrate that transfer to initiator has little importance in that system.⁷ Thus, other explanations for those irregularities in polymerization kinetics previously attributed to transfer to initiator efficiency with conversion (see 3.3.1.1.3). There is some evidence that transfer to initiator with use of importance during AIBN-initiated vinyl acetate polymerization.¹¹³

Even though AIBN has a low transfer constant, the ketenimine formed by combination of cyanoisopropyl radicals (Scheme 3.13) is anticipated to be more susceptible to induced decomposition (Scheme 3.22).¹⁰³



* Pryor and Fiske¹⁰⁴ have determined $C_1=3.7\times10^{-5}$ for 9 at 60 °C in S polymerization.

3.3.1.2 Hyponitrites

The hyponitrites (16), esters of hyponitrous acid (HO N=N OH), are low temperature sources of alkoxy or acyloxy radicals. A detailed study of the effect of substituents on k_d for the hyponitrite esters has been reported by Quinga and Mendenhall.¹¹⁴



While di-*t*-butyl (**34**) and dicumyl hyponitrites (**35**) have proved convenient sources of *t*-butoxy and cumyloxy radicals respectively in the laboratory,^{71,72,115-117} the utilization of hyponitrites as initiators of polymerization has been limited by difficulties in synthesis and commercial availability. Dialkyl hyponitrites (**16**) show only weak absorption at λ >290 nm and their photochemistry is largely a neglected area. The triplet sensitized decomposition of these materials has been investigated by Mendenhall *et al.*¹¹⁸

The hyponitrites generally appear somewhat more efficient with respect to radical generation than the dialkyldiazenes (see 3.3.1.1). However, a proportion of radicals is lost through cage reaction with formation of the corresponding dialkyl peroxides or ketone plus alcohol (Scheme 3.23).^{119,120} The disproportionation pathway is open only to hyponitrites with α -hydrogens. Kiefer and Traylor¹²¹ showed that the extent of cage reaction was strongly dependent on the medium viscosity.



Scheme 3.23

Approximately 5% of radicals undergo cage recombination when dicumyl hyponitrite (35) is decomposed in bulk MMA or S at 60 °C.⁷² Dicumyl peroxide, the product of cage recombination is likely to be stable under the conditions where hyponitrites are usually employed. Nonetheless, its formation is a concern since

7**8**

contamination of a product polymer with peroxide may impair its longer term durability.

Tertiary hyponitrites are not particularly susceptible to induced decomposition. However, the same is not true of primary and secondary hyponitrites.¹²² Isopropyl hyponitrite is reported¹²³ to undergo induced decomposition by a mechanism involving initial abstraction of a α -hydrogen (Scheme 3.24).

$$\begin{array}{c} H^{\bullet} \\ H^{\bullet} \\ H^{\bullet} \\ CH_{3} - \overset{H}{C} \overset{H}{} O^{-} \overset{H}{O} - \overset{H}{O} - \overset{H}{C} - CH_{3} \\ CH_{3} \\ CH_{3} \\ CH_{3} \\ CH_{3} \end{array} \xrightarrow{} RH + CH_{3} - \overset{C}{C} = O + N_{2} + \overset{H}{} O^{-} \overset{H}{C} - CH_{3} \\ CH$$

Scheme 3.24

3.3.2 Peroxides

The general chemistry of the peroxides has been covered in many books and reviews.^{2,46,52,124-131} Readers are referred in particular to Swern's Trilogy¹²⁷⁻¹²⁹ for an excellent background and a comprehensive coverage of the literature through 1970. The chemistry associated with their use as initiators of polymerization was reviewed by Moad and Solomon.²

Many types of peroxides (R-O-O-R) are known. Those in common use as initiators include: diacyl peroxides (36), peroxydicarbonates (37), peroxyesters (38), dialkyl peroxides (39), hydroperoxides (40), and inorganic peroxides [*e.g.* persulfate (41)]. Multifunctional and polymeric initiators with peroxide linkages are discussed in Sections 3.3.3 and 6.3.2.1.

| 0 0 R-C-O-O-C-R' | 0 0 " " R-O-C-O-O-C-O-R' | 0 R-C-O-O-R' |
|---------------------------|--------------------------------|--|
| 36 | 37 | 38 |
| R-0-0-R' | R-O-O-H | 0 0- <u>5</u> -0-0- <u>5</u> -0 0 0 |
| 39 | 40 | 41 |

Peroxides are used most commonly either as thermal initiators or as a component in a redox system. While peroxides are photochemically labile, they seldom find use as photoinitiators other than in laboratory studies because of their poor light absorption characteristics. They generally have low extinction coefficients and absorb in the same region as monomer. Kinetic parameters for decomposition of some important peroxides are given in Table 3.5.

| class | initiator | R | R' | solvent | temp. range ^b |
|---------------------------|-----------|---|---|----------------------|--------------------------|
| | | | | | °C |
| diacyl peroxides (36) | | | | | |
| | BPO | Ph- | Ph- | benzene ^e | 38-80(17) |
| | LPO | <i>п-</i> С ₁₁ Н ₂₃ - | <i>п-</i> С ₁₁ Н ₂₃ - | benzene | 35-70(8) |
| peroxydicarbonates (37) | | | | | |
| | 47 | (CH ₃) ₂ CH- | (CH ₃) ₂ CH- | benzene ^e | 35-60(10) |
| peroxyesters (38) | | | | | |
| | BPB | Ph- | (CH ₃) ₃ C- | benzene | 110-130(3) |
| | DBPOX | | | benzene | 35-55(3) |
| dialkyl peroxides (39) | | | | | |
| | DTBP | (CH ₃) ₃ C- | (CH ₃) ₃ C- | benzene | 100-135(4) |
| alkyl hydroperoxides (40) | | | | | |
| | 59 | (CH ₃) ₃ C- | - | benzene | 155-175(4) |
| inorganic peroxides | | | | | |
| | 41 | $K_2S_2O_8$ | - | NaOH ^r | 50-90(5) |

Table 3.5 Selected Kinetic Data for Decomposition of Peroxides^a

a Kinetic parameters recalculated from original data taken from the references indicated. Values rounded to 3 significant figures. b Number of data points given in parentheses. c Calculated from Arrhenius parameters shown and rounded to two significant figures. d Temperature for ten hour half life – see footnote d to Table 3.2. e In the presence of inhibitor added to prevent induced decomposition. f 0.1 M aqueous NaOH.

| Ea | $\Lambda \times 10^{-15}$ | $k_{\rm d} \times 10^6$ | $10 h t_{1/2}^{a.d}$ | Rcf. | Initiator |
|----------------------|---------------------------|--------------------------------------|----------------------|---------|-----------|
| kJ mol ⁻¹ | s ⁻¹ | s ⁻¹ (60 °C) ^c | °C | | |
| | | | | | |
| 139.0 | 9.34 | 1.5 | 78 | 132 | BPO |
| 125.3 | 0.393 | 8.9 | 66 | 133 | LPO |
| | | | | | |
| 126.7 | 9.75 | 130 | 46 | 134 | 47 |
| | | | | | |
| 144.0 | 1.53 | 0.04 | 105 | 135 | BPB |
| 110.0 | 0.310 | 1800 | 26 | 136 | DBPOX |
| | | | | | |
| 152.7 | 2.16 | 0.0025 | 125 | 137,138 | DTBP |
| | | | | | |
| 174,2 | 7.97 | - | 168 | 1.39 | 59 |
| | | | | 1.10 | |
| 148.0 | 709 | 4.4 | 69 | 140 | 41 |

Table 3.5 (continued)
3.3.2.1 Diacyl or diaroyl peroxides

Diacyl or diaroyl peroxides (36, R⁻ alkyl or aryl respectively) are given specific coverage in reviews by Fujimori,¹⁴¹ Bouillion *et al.*,¹⁴² and Hiatt.¹⁴³ They are sources of acyloxy radicals which in turn are sources of aryl or alkyl radicals. Commercially available peroxides of this type include dibenzoyl peroxide (BPO), didodecanoyl or dilauroyl peroxide (LPO), didecanoyl peroxide (42) and succinic acid peroxide (43).



3.3.2.1.1 Thermal decomposition

The rates of thermal decomposition of diacyl peroxides (36) are dependent on the substituents R. The rates of decomposition increase in the series where R is: aryl~primary alkyl<secondary alkyl<tertiary alkyl. This order has been variously proposed to reflect the stability of the radical (R•) formed on β -scission of the acyloxy radical, the nucleophilicity of R, or the steric bulk of R. For peroxides with non-concerted decomposition mechanisms, it seems unlikely that the stability of R• should by itself be an important factor.

For diaroyl peroxides (36, R=aryl), *m*- and *p*-electron withdrawing substituents retard the rate of decomposition while *m*- and *p*-electron donating and all *o*-substituents enhance decomposition rates. The *o*-substituent effect has been attributed to the sensitivity of homolysis to steric factors.

Only a few diacyl peroxides see widespread use as initiators of polymerization. The reactions of the diaroyl peroxides (**36**, R=aryl) will be discussed in terms of the chemistry of BPO (Scheme 3.25). The rate of β -scission of thermally generated benzoyloxy radicals is slow relative to cage escape, consequently, both benzoyloxy and phenyl radicals are important as initiating species. In solution, the only significant cage process is reformation of BPO (*ca* 4% at 80 °C in isooctane);^{141,145} only minute amounts of phenyl benzoate or biphenyl are formed within the cage. Therefore, in the presence of a reactive substrate (*e.g.* monomer), the production of radicals can be almost quantitative (see 3.3.2.1.3).

One of the most commonly encountered aliphatic diacyl peroxides (36, R=alkyl) is LPO. Lower diacyl peroxides (e.g. diacetyl peroxide) cannot be

conveniently handled in a pure state due to their susceptibility to induced decomposition. They are shock sensitive and may decompose explosively.



Scheme 3.25

In general, aliphatic diacyl peroxide initiators should be considered as sources of alkyl, rather than of acyloxy radicals. With few exceptions, aliphatic acyloxy radicals have a transient existence at best. For certain diacyl peroxides (**36**) where R is a secondary or tertiary alkyl group there is controversy as to whether loss of carbon dioxide occurs in concert with O-O bond cleavage. Thus, ester end groups observed in polymers prepared with aliphatic diacyl peroxides are unlikely to arise directly from initiation, but rather from transfer to initiator (see 3.3.2.1.4).

The high rate of decarboxylation of aliphatic acyloxy radicals is also the prime reason behind low initiator efficiencies (see 3.3.2.1.3). Decarboxylation occurs within the solvent cage and recombination gives alkane or ester byproducts. Cage return for LPO is 18-35% at 80 °C in *n*-octane as compared to only 4% for BPO under similar conditions.¹⁴⁴

Observed rates of disappearance for diacyl peroxides show marked dependence on solvent and concentration.¹⁴⁶ In part, this is a reflection of their susceptibility to induced decomposition (see 3.3.2.1.4 and 3.3.2.1.5). However, the rate of disappearance is also a function of the viscosity of the reaction medium. This is evidence for cage return (see 3.3.2.1.3).¹⁴⁵ The observation¹⁴⁴ of slow scrambling of the label in benzoyl-*carbonyl*-¹⁸O peroxide between the carbonyl and the peroxidic linkage provides more direct evidence for this process.

3.3.2.1.2 Photochemical decomposition

Diacyl peroxides have continuous weak absorptions in the UV to *ca* 280 nm (ϵ *ca* 50 M⁻¹ cm⁻¹ at 234 nm).¹⁴⁷ Although the overall chemistry in thermolysis and photolysis may appear similar, substantially higher yields of phenyl radical products are obtained when BPO is decomposed photochemically. It has been suggested that, during the photodecomposition of BPO, β -scission may occur in

concert with O–O bond rupture and give rise to formation of one benzoyloxy radical, one phenyl radical, and a molecule of carbon dioxide (Scheme 3.26).¹⁴⁸ Time resolved EPR experiments¹⁴⁹ have shown that photochemical decomposition of BPO does produce benzoyloxy radicals with discrete existence. It is, nonetheless, clear that the photochemically generated benzoyloxy radicals have substantially shorter life times in solution than those generated thermally.^{150,151} In these circumstances cage products also assume greater importance¹⁵¹ and initiator efficiencies are anticipated to be lower.



Scheme 3.26

It has also been suggested that photoexcited benzoyl peroxide is somewhat more susceptible to induced decomposition processes involving electron transfer than the ground state molecule. Rosenthal *et al.*¹⁵² reported on redox reactions with certain salts (including benzoate ion) and neutral molecules (*e.g.* alcohols).

3.3.2.1.3 Initiator efficiency

Ideally all reactions should result from unimolecular homolysis of the relatively weak O–O bond. However, unimolecular rearrangement and various forms of induced and non-radical decomposition complicate the kinetics of radical generation and reduce the initiator efficiency.⁴⁶ Peroxide decomposition induced by radicals and redox chemistry is covered in Sections 3.3.2.1.4 and 3.3.2.1.5 respectively.

Cage recombination is also a major factor limiting the efficiency of radical production from aliphatic diacyl peroxides. Initiator efficiency depends on the rate of β -scission of the acyloxy radical formed. If β -scission is slow, the only significant cage reaction involves regeneration of the diacyl peroxide (*e.g.* thermolysis of diaroyl peroxides). Cage return leads to a lowering of the rate of decomposition without reducing the initiator efficiency (see 3.3.2.1.1). However, if β -scission is rapid and decarboxylation occurs within the solvent cage, then combination of the alkyl or aryl radical with another radical to form an ester or alkane will reduce the initiator efficiency (*e.g.* thermolysis or photolysis of aliphatic diacyl peroxides and photolysis of all diacyl peroxides).

The importance of the cage reaction increases according to the viscosity of the reaction medium. This contributes to a decrease in initiator efficiency with conversion.^{33,153-155} Stickler and Dumont¹⁵⁶ determined the initiator efficiency during bulk MMA polymerization at high conversions (*ca* 80%) to be in the range 0.1-0.2 depending on the polymerization temperature. The main initiator-derived byproduct was phenyl benzoate.

Diacyl peroxides may also undergo non-radical decomposition *via* the carboxy inversion process to form an acylcarbonate (Scheme 3.27).⁴⁶ The reaction is of greatest importance for diaroyl peroxides with electron withdrawing substituents and for aliphatic diacyl peroxides (**36**) where R is secondary, tertiary or benzyl.¹⁵⁷ The reaction is thought to involve ionic intermediates and is favored in polar solvents¹⁵⁷ and by Lewis acids.¹⁵⁸ Other heterolytic pathways for peroxide decomposition have been described.¹⁵⁹



3.3.2.1.4 Transfer to initiator and induced decomposition

Transfer to initiator can be a major complication in polymerizations initiated by diacyl peroxides. The importance of the process typically increases with monomer conversion and the consequent increase in the [initiator]:[monomer] ratio.^{9,106,160-162} In BPO initiated S polymerization, transfer to initiator may be the major chain termination mechanism. For bulk S polymerization with 0.1 M BPO at 60 °C up to 75% of chains are terminated by transfer to initiator or primary radical termination (<75% conversion).⁷ A further consequence of the high incidence of chain transfer is that high conversion PS formed with BPO initiator tends to have a much narrower molecular weight distribution than that prepared with other initiators (*e.g.* AIBN) under similar conditions.

The mechanism of transfer to BPO involves homolytic attack on one of the oxygen atoms of the peroxidic linkage (Scheme 3.16) with formation of an ester end group and expulsion of a benzoyloxy radical. The end group formed (a secondary ester) is distinct from that formed in initiation. Such end groups may contribute to the reduced thermal stability of high conversion PS prepared with benzoyl peroxide (Section 8.2.1).^{14,163} In the case of VAc or VC polymerizations the chain end will be a hydrolytically unstable ketal or α -chloroester group respectively (Section 8.2.3).

Other radicals present in the reaction medium may also induce the decomposition of BPO and other diacyl peroxides. These include initiatorderived¹⁴⁶ and stable radicals (*e.g.* galvinoxyl,¹³² triphenylmethyl^{164,165} and nitroxides¹⁶⁶).

3.3.2.1.5 Redox reactions

The decomposition of diacyl peroxides (**36**) is catalyzed by various transition metal salts,^{46,167} for example, Cu^+ (Scheme 3.28).^{168,169} A side reaction is oxidation of alkyl radicals by the oxidized form of the metal salt (*e.g.* Cu^{2+}).



Nitro- and nitroso-compounds,^{170,171} amines, and thiols induce the decomposition of diacyl peroxides in what may be written as an overall redox reaction. Certain monomers have been reported to cause induced decomposition of BPO. These include AN,¹⁷² *N*-vinylcarbazole,¹⁷³⁻¹⁷⁷ *N*-vinylimidazole¹⁷⁸ and NVP.¹⁷⁷

The mechanism proposed for the production of radicals from the *N*, *N*-dimethylaniline/BPO couple^{179,180} involves reaction of the aniline with BPO by a $S_N 2$ mechanism to produce an intermediate (44). This thermally decomposes to benzoyloxy radicals and an amine radical cation (46) both of which might, in principle, initiate polymerization (Scheme 3.29). Pryor and Hendrikson¹⁸¹ were able to distinguish this mechanism from a process involving single electron transfer through a study of the kinetic isotope effect.



Scheme 3.29

It has been suggested that the amine radical cation (46) is not directly involved in initiating chains and that most polymerization is initiated by benzoyloxy radicals.¹⁷⁹ However, Sato *et al.*¹⁸² employed spin trapping (3.5.2.1) to demonstrate that the anilinomethyl radical (45) was formed from the radical cation (46) by loss of a proton and proposed that the radical 45 also initiates polymerization. Overall efficiencies for initiation by amine-peroxide redox

couples are very low; Imoto and Choe¹⁸⁰ report f ca 25%; Walling¹⁷⁹ reports f=2-5%.

3.3.2.2 Dialkyl peroxydicarbonates

The chemistry of peroxydicarbonates (37) and their use as initiators of polymerization has been reviewed by Yamada *et al.*,¹³⁴ Hiatt¹⁴³ and Strong.¹⁸³

Dialkyl peroxydicarbonates have been reported as low temperature sources of alkoxy radicals (Scheme 3.30)^{184,185} and these radicals may be formed in relatively inert media. However, it is established, for primary and secondary peroxydicarbonates, that the rate of loss of carbon dioxide is slow compared to the rate of addition to most monomers or reaction with other substrates.^{186,187} Thus, in polymerizations carried out with diisopropyl peroxydicarbonate (47), chains will be initiated by isopropoxycarbonyloxy (48) rather than isopropoxy radicals (49) (see 3.4.2.2).¹⁸⁸

A slow rate of β -scission also means that the main cage recombination process will be cage return to reform the peroxydicarbonate. Dialkyl peroxides are typically not found amongst the products of peroxydicarbonate decomposition. In these circumstances, cage recombination is unlikely to be a factor in reducing initiator efficiency.

Laboratory studies have generally focused on the diisopropyl, dicyclohexyl and di-*t*-butyl derivatives. These and the *s*-butyl and 2-ethylhexyl derivatives are commercially available.¹⁸⁹ The rates of decomposition of the peroxydicarbonates show significant dependence on the reaction medium and their concentration. This dependence is, however, less marked than for the diacyl peroxides (**36**) (see 3.3.1.1.4). Induced decomposition may involve a mechanism analogous to that described for diacyl peroxides. However, a more important mechanism for primary and secondary peroxydicarbonates involves abstraction of an α -hydrogen (Scheme 3.31).¹⁹⁰

$$\mathbf{R} \underbrace{\stackrel{\mathsf{CH}_3}{\leftarrow} H \stackrel{\mathsf{O}}{\leftarrow} C \stackrel{\mathsf{CH}_3}{\leftarrow} O \stackrel{\mathsf{O}}{\leftarrow} C \stackrel{\mathsf{CH}_3}{\leftarrow} O \stackrel{\mathsf{CH}_3}{\leftarrow} C \stackrel{\mathsf{CH}_3}{\leftarrow} \mathbf{R} \stackrel{\mathsf{CH}_3}{\leftarrow} H + \underbrace{\stackrel{\mathsf{CH}_3}{\leftarrow} C \stackrel{\mathsf{O}}{\leftarrow} O \stackrel{\mathsf{CH}_3}{\leftarrow} O \stackrel{\mathsf{CH}$$

Crano¹⁹¹ has investigated the reaction between diisopropyl peroxydicarbonate and tertiary amines. These experiments indicate the formation of radicals by loss of a hydrogen from the α -CH₂ of the amine. It seems likely that the mechanism of

radical formation is analogous to that observed for diacyl peroxide-amine systems (see 3.3.2.1.5).

3.3.2.3 Peroxyesters

The chemistry of peroxyesters (38) also commonly called peresters has been reviewed by Sawaki,¹⁹² Bouillion *et al.*¹⁹³ and Singer.¹⁹⁴ The peroxyesters are sources of alkoxy and acyloxy radicals (Scheme 3.32). Most commonly encountered peroxyesters are derivatives of *t*-alkyl hydroperoxides (*e.g. t*-butyl peroxybenzoate, BPB).



Aryl peroxyesters are generally unsuitable as initiators of polymerization owing to the generation of phenoxy radicals that can inhibit or retard polymerization

3.3.2.3.1 Thermal decomposition

The rates of decomposition of peroxyesters (38) are very dependent on the nature of the substituents R and R'. The variation in the decomposition rate with R follows the same trends as have been discussed for the corresponding diacyl peroxides (see 3.3.2.1.1).

Peroxyesters derived from secondary (*e.g.* peroxyisobutyrate esters) and tertiary acids (*e.g.* peroxypivalate esters) are believed to undergo concerted 2-bond cleavage leading to direct production of an alkoxy and an alkyl radical and a molecule of carbon dioxide.¹⁹⁵⁻¹⁹⁸ On the other hand, primary (*e.g.* peroxyacetate and peroxypropionate esters) and aromatic peroxyesters (*e.g.* BPB, Scheme 3.32) are thought to undergo I-bond scission to generate an acyloxy and an alkoxy radical.^{145,196} Evidence for the transient existence of acyloxy radicals includes the observation of substantial cage return.

For *t*-butyl peresters there is also a variation in efficiency in the series where R is primary>>secondary>tertiary. The efficiency of *t*-butyl peroxypentanoate in initiating high pressure ethylene polymerization is >90%, that of *t*-butyl peroxy-2-ethylhexanoate *ca* 60% and that of *t*-butyl peroxypivalate *ca* 40%.¹⁹⁶ Inefficiency is due to cage reaction and the main cage process in the case where R is secondary or tertiary is disproportionation with *t*-butoxy radicals to form *t*-butanol and an olefin.¹⁹⁶

Di-*t*-butyl peroxyoxalate (DBPOX) is a clean, low temperature, source of *t*-butoxy radicals (Scheme 3.33).¹³⁶ The decomposition is proposed to take place by concerted 3-bond cleavage to form two alkoxy radicals and two molecules of carbon dioxide.

$$\begin{array}{cccccccccc} & & & & & & & CH_3 & & CH_3 & & & CH_3 & & CH_3 & & & CH_3 & CH_3 & & CH_3 & & CH_3 & &$$

Scheme 3.33

The low conversion initiator efficiency of di-*t*-butyl peroxyoxalate (0.93-0.97)¹²¹ is substantially higher than for other peroxyesters [*t*-butyl peroxypivalate, 0.63; *t*-butyl peroxyacetate, 0.53 (60 °C, isopropylbenzene)¹⁹⁵]. The dependence of cage recombination on the nature of the reaction medium has been the subject of a number of studies.^{121,199,200} The yield of DTBP (the main cage product) depends not only on viscosity but also on the precise nature of the solvent. The effect of solvent is to reduce the yield in the order: aliphatic>aromatic>protic. It has been proposed¹⁹⁹ that this is a consequence of the solvent dependence of β -scission of the *t*-butoxy radical which increases in the same series (Section 3.4,2.1.1).

Transfer to initiator is generally of lesser importance than with the corresponding diacyl peroxides. They are, nonetheless, susceptible to the same range of reactions (see 3.3.2.1.4). Radical-induced decomposition usually occurs specifically to give an alkoxy radical and an ester (Scheme 3.34).

$$\bigcirc \overset{O}{\xrightarrow{-C-O-O-C-CH_3}} \longrightarrow \bigcirc \overset{O}{\xrightarrow{-C-O-R}} + \overset{CH_3}{\xrightarrow{-C-O-R}}$$

BPB

Scheme 3.34

Peroxyesters may undergo non-radical decomposition *via* the Criegee rearrangement (Scheme 3.35). This process is analogous to the carboxy inversion process described for diacyl peroxides (see 3.3.2.1.3) and probably involves ionic intermediates.

Scheme 3.35

The reaction is facilitated when R is electron withdrawing, when R has a high migratory aptitude (ability to stabilize a carbonium ion), and by polar reaction media.

3.3.2.3.2 Photochemical decomposition

Peroxyesters seldom find use as photoinitiators since photodecomposition requires light of 250-300 nm, a region where many monomers also absorb. This situation may be improved by the introduction of a suitable chromophore into the molecule or through the use of sensitizers.^{201,202} The peroxyester (**50**) is reported to have λ_{max} 366 nm and ϕ near unity.²⁰¹



50

3.3.2.4 Dialkyl peroxides

90

The chemistry of the dialkyl peroxides (**39**) has been reviewed by Matsugo and Saito,²⁰³ Sheldon²⁰⁴ and Hiatt.²⁰⁵ Dialkyl peroxides are high temperature sources of alkoxy radicals. Dialkyl peroxides commonly used as initiators have tertiary alkyl substituents. Those available commercially include di-*t*-butyl (DTBP) and dicumyl (**51**) peroxides, sources of *t*-butoxy and cumyloxy radicals respectively, **52** and a variety of dialkyl peroxyketals (*e.g.* **53-55**).^{206,207} These latter initiators, including 1,1-di-*t*-butylperoxycyclohexane (**53**), have decomposition rate constants k_d that are an order of magnitude greater than simple di-*t*-alkyl peroxides (*e.g.* DTBP, **51**, **52**)²⁰⁸ and can be shock sensitive. The peroxides **56-58** find application when volatility is an issue. For example, they are used in graft copolymerization by reactive extrusion (Section 7.6.4).²⁷





The decomposition of the peroxyketals (53) follows a stepwise, rather than a concerted mechanism. Initial homolysis of one of the O-O bonds gives an alkoxy radical and an α -peroxyalkoxy radical (Scheme 3.36).^{206,208-210} This latter species decomposes by β -scission with loss of either a peroxy radical to form a ketone as byproduct or an alkyl radical to form a peroxyester intermediate. The peroxyester formed may also decompose to radicals under the reaction conditions. Thus, four radicals may be derived from the one initiator molecule.



The relative importance of the various pathways depends on the alkyl groups (R). The rate constants for scission of groups (R•) from *t*-alkoxy radicals ($R^1R^2R^3C$ -O•) increase in the order isopropyl<ethyl<*t*-butylperoxy<methyl.²¹⁰ Thus, the pathway affording peroxyester and an alkyl radical is less important when R is methyl than when R is a higher alkyl group. If the pathway to alkylperoxy radicals is dominant, the resultant polymer is likely to have a proportion of peroxy end groups.^{206,211}

Solvent dependence of k_d for di-*t*-alkyl peroxides is small when compared to most other peroxide initiators.^{138,212} For di-*t*-butyl peroxide,¹³⁸ k_d is slightly greater (up to two-fold at 125 °C) in protic (*t*-butanol, acetic acid) or dipolar aprotic solvents than in other media (cyclohexane, triethylamine, tetrahydrofuran).

The chemistry of the di-*t*-butyl and cumyl peroxides is relatively uncomplicated by induced or ionic decomposition mechanisms. However, induced decomposition of di-*t*-butyl peroxide has been observed in primary or secondary alcohols^{213,214} (Scheme 3.37) and primary or secondary amines.²¹⁵ The reaction

involves oxidation of an α -hydroxyalkyl or α -aminoalkyl radical, to the corresponding carbonyl- or imino-compound and apparently requires coordination of the hydroxyl or aminyl hydrogen to the peroxidic oxygen.



The radical yield from simple di-*t*-alkyl peroxides (*i.e.* dicumyl, di-*t*-butyl) is reported to be almost 100%. The only significant cage reaction is reformation of the peroxide. The efficiencies of dialkyl peroxyketals and primary and secondary peroxides are lower.²⁰⁷ Lower efficiencies arise when the initially formed radicals undergo β -scission before cage escape or, in the case where primary or secondary alkoxy radicals are formed, by disproportionation within the solvent cage. Primary and secondary peroxides are also susceptible to a variety of induced and nonradical decomposition mechanisms. The initiator efficiency of di-*t*-butyl peroxide in styrene polymerization is reported to remain constant at close to 100% until *ca* 80% when it undergoes a dramatic reduction by more than an order of magnitude.²¹⁶ An explanation was not provided. It is possible, that at this conversion the rate of cage escape is slowed such that β -scission to give methyl radicals occurs within the solvent cage.

3.3.2.5 Alkyl hydroperoxides

The chemistry of alkyl hydroperoxides (40) has been reviewed by Porter,²¹⁷ Sheldon²⁰⁴ and Hiatt.²¹⁸ Alkyl hydroperoxides are high temperature sources of alkoxy and hydroxy radicals.²¹⁹ They are often encountered as components of redox systems.

The common initiators of this class are *t*-alkyl derivatives, for example, *t*-butyl hydroperoxide (59), *t*-amyl hydroperoxide (60), cumene hydroperoxide (61), and a range of peroxyketals (62). Hydroperoxides formed by hydrocarbon autoxidation have also been used as initiators of polymerization.



The ROO-H bond of hydroperoxides is weak compared to most other X-H bonds.^{*} Thus, abstraction of the hydroperoxidic hydrogen by radicals is usually an

^{*} D_{ROO-H} ~ 375 kJ mol⁻¹.²²⁰

exothermic process. The hydroperoxides can therefore be efficient transfer agents and radical-induced decomposition may be a major complication in their use as initiators.²²²

Primary and secondary hydroperoxides are also susceptible to induced decomposition through loss of an α -hydrogen. The radical formed is usually not stable and undergoes β -scission to give a carbonyl compound and hydroxy radical.²²³ It is reported that these hydroperoxides may also undergo non-radical decomposition with evolution of hydrogen.¹³⁷

Hydroperoxides react with transition metals in lower oxidation states ($Ti^{3^{-}}$, $Fe^{2^{-}}$, Cu^{+} , *etc.*) and a variety of other oxidants to give an alkoxy radical and hydroxide anion (Scheme 3.38).^{46,224,225}

$$\begin{array}{c} CH_3 \\ CH_3 - C - O - O - H \\ CH_3 \\ CH_3 \end{array} + Fe^{2+} \longrightarrow CH_3 - C - O + TOH + Fe^{3+} \\ CH_3 \\ CH_3 \\ CH_3 \end{array}$$

Scheme 3.38

With some systems, the hydroperoxide is reduced to hydroperoxy radical by the metal ion in its higher oxidation state (Scheme 3.39). Thus, it is possible to set up a catalytic cycle for hydroperoxide decomposition.

$$CH_{3} - C - O - O - H + Cu^{2+} \longrightarrow CH_{3} - C - O - O + H^{+} + Cu^{+}$$

$$CH_{3} - C - O - O + H^{+} + Cu^{+}$$

$$CH_{3} - C - O - O + H^{+} + Cu^{+}$$

$$CH_{3} - C - O - O + H^{+} + Cu^{+}$$

Scheme 3.39

With Ti^{4+} and Fe^{3+} this latter pathway is thought not to occur. The formation of ROO+, observed at high hydroperoxide concentrations, is attributed to the occurrence of induced decomposition.²²⁶

3.3.2.6 Inorganic peroxides



Inorganic peroxides [hydrogen peroxide (63), persulfate (41), peroxymonosulfate and peroxydiphosphate (64)] generally have limited usefulness as initiators in bulk or solution polymerization due to their poor solubility in

 $D_{\text{R3C-H}} \sim 385, D_{\text{R2CH-H}} \sim 396, D_{\text{RCH2-H}} \sim 410, D_{\text{RO-H}} \sim 435 \text{ kJ mol}^{-1,221}$

organic media. This means that the main use of these initiators is in aqueous²²⁷ or in part-aqueous heterogeneous media (*e.g.* in emulsion polymerization). They are often encountered as one component in a redox initiation system. The history of these systems has been reviewed by Bacon²²⁸ and Sosnovsky and Rawlinson.²²⁹ Their use is also described by Sarac.²³⁰

The following discussion concentrates on the chemistry of the two most common inorganic peroxides, persulfate and hydrogen peroxide.

3.3.2.6.1 Persulfate

Photolysis or thermolysis of persulfate ion (41) (also called peroxydisulfate) results in homolysis of the O-O bond and formation of two sulfate radical anions. The thermal reaction in aqueous media has been widely studied.^{231,232} The rate of decomposition is a complex function of pH, ionic strength, and concentration. Initiator efficiencies for persulfate in emulsion polymerization are low (0.1-0.3) and depend upon reaction conditions (*i.e.* temperature, initiator concentration).²³³

A number of mechanisms for thermal decomposition of persulfate in neutral aqueous solution have been proposed.²³² They include unimolecular decomposition (Scheme 3.40) and various bimolecular pathways for the disappearance of persulfate involving a water molecule and concomitant formation of hydroxy radicals (Scheme 3.41). The formation of polymers with negligible hydroxy end groups is evidence that the unimolecular process dominates in neutral solution. Heterolytic pathways for persulfate decomposition can be important in acidic media.



Normally, persulfate (41) can only be used to initiate polymerization in aqueous or part aqueous (emulsion) media because it has poor solubility in most organic solvents and monomers. However, it has been reported that polymerizations in organic solvent may be initiated by crown ether complexes of potassium persulfate.²³⁴⁻²³⁷ Quaternary ammonium persulfates can also serve as useful initiators in organic media.^{236,238} The rates of decomposition of both the crown ether complexes and the quaternary ammonium salts appear dramatically

greater than those of conventional persulfate salts (K⁺, Na⁻, NII₄⁺) in aqueous solution. The crown ether complex can be used to initiate polymerization at ambient temperature.²³⁴

In part, the accelerated decomposition might be attributed to the occurrence of induced decomposition and primary radical transfer.²³⁹ Persulfate (**41**) is also known to be a strong oxidant and, in this context, has been widely applied in synthetic organic chemistry.²⁴⁰ It is established that the rate of disappearance of persulfate in aqueous media is accelerated by the presence of organic compounds²³¹ and induced decomposition is an integral step in the oxidation of organic substrates (including ethers) by persulfate.²⁴¹

Persulfate (41) absorbs only weakly in the UV ($\varepsilon ca 25 \text{ M}^{-1} \text{ cm}^{-1} \text{ at } 250 \text{ nm}$).²⁴² Nonetheless, direct photolysis of persulfate ion has been used as a means of generating sulfate radical anion in laboratory studies.^{242,243}

Persulfate (41) reacts with transition metal ions (e.g. Ag^{1} , Fe^{21} , Ti^{31}) according to Scheme 3.42. Various other reductants have been described. These include halide ions, thiols (e.g. 2-mercaptoethanol, thioglycolic acid, cysteine, thiourea), bisulfite, thiosulfate, amines (triethanolamine, tetramethylethylenediamine, hydrazine hydrate), ascorbic acid, and solvated electrons (e.g. in radiolysis). The mechanisms and the initiating species produced have not been fully elucidated for many systems.²⁴⁴



Scheme 3.42

Various multicomponent systems have also been described. Three component systems in which a second reducing agent (*e.g.* sulfite) acts to recycle the transition metal salt, have the advantage that less metal is used (Scheme 3.43).

 $SO_3^{=} + Cu^{2+} \longrightarrow SO_3^{-} + Cu^{+}$

Scheme 3.43

Redox initiation is commonly employed in aqueous emulsion polymerization. Initiator efficiencies obtained with redox initiation systems in aqueous media are generally low. One of the reasons for this is the susceptibility of the initially formed radicals to undergo further redox chemistry. For example, potential propagating radicals may be oxidized to carbonium ions (Scheme 3.44). The problem is aggravated by the low solubility of the monomers (*e.g.* MMA, S) in the aqueous phase.

Scheme 3.44

3.3.2.6.2 Hydrogen peroxide

Homolytic scission of the O-O bond of hydrogen peroxide may be effected by heat or UV irradiation.²⁴⁵ The thermal reaction requires relatively high temperatures (>90 °C). Photolytic initiation generally employs 254 nm light. Reactions in organic media require a polar cosolvent (*e.g.* an alcohol).

Hydrogen peroxide also reacts with reducing agents (transition metals, metal complexes, solvated electrons, and some organic reagents) to produce hydroxyl radicals. It reacts with oxidizing agents to give hydroperoxy radicals. The reaction between hydrogen peroxide and transition metal ions in their lower oxidation state is usually represented as the simple process first described by Haber and Weiss (Scheme 3.45).²⁴⁶ However the mechanism is significantly more complex.

 H_2O_2 + Fe²⁺ \longrightarrow HO · + ^-OH + Fe³⁺ Scheme 3.45

Scheme 5.45

It has been suggested that the reactive species are metal complexed hydroxy radicals rather than "free" hydroxyl radicals.²⁴⁷⁻²⁵⁰ The reactions observed show dependence on the nature of the metal ion and quite different product distributions can be obtained from reaction of organic substrates with $Fc^{2-}H_2O_2$ (Fenton's Reagent) and $Ti^{3+}H_2O_2$. However, it is not clear whether these findings reflect the involvement of a different active species or simply the different rates and/or pathways for destruction of the initially formed intermediates.²⁵¹ Metal ions in their higher oxidation states (*e.g.* Fe^{3+}) can bring about the destruction of hydrogen peroxide according to Scheme 3.46.

$$H_2O_2$$
 + Fe³⁺ ----- HOO• + H⁺ + Fe²⁺

Scheme 3.46

The $Ti^{3^{+}}-H_2O_2$ system is preferred over Fenton's reagent because $Ti^{4^{+}}$ is a less powerful oxidizing agent than Fe³⁺ and the above mentioned pathway and other side reactions are therefore of less consequence.²⁵² Much of the discussion on redox initiation in Section 3.3.2.6.1 is also relevant to hydrogen peroxide.

3.3.3 Multifunctional Initiators

Multifunctional initiators contain two or more radical generating functions within the one molecule. They can be considered in two distinct classes according

to whether they undergo concerted (see 3.3.3.1) or non-concerted decomposition (see 3.3.3.2).

3.3.3.1 Concerted decomposition

Multifunctional initiators where the radical generating functions are in appropriate proximity may decompose in a concerted manner or in a way such that the intermediate species can neither be observed nor isolated. Examples of such behavior are peroxyoxalate esters (see 3.3.2.3.1) and α -hydroperoxy diazenes (*e.g.* **31**), derived peroxyesters (**65**)^{253,254} and bis- and multi-diazenes such as **66**.

The initiators (31) and (65) are low temperature sources of alkyl and hydroxy or acyloxy radicals respectively (Scheme 3.47).^{253,257,258} The α -hydroperoxy diazenes (*e.g.* 31) are one of the few convenient sources of hydroxy radicals in organic solution.^{253,254}



It has been reported that the α -hydroperoxy diazenes may undergo induced decomposition either by OH or H transfer.²⁵⁹

3.3.3.2 Non-concerted decomposition

Initiators where the radical generating functions are sufficiently remote from each other break-down in a non-concerted fashion. Examples include the azoperoxide $(68)^{260}$ and the bis-diazene $(67)^{.261}$ Their chemistry is often understandable in terms of the chemistry of analogous monofunctional initiators.²⁶⁰ This class also includes the dialkyl peroxyketals (see 3.3.2.4) and hydroperoxyketals (see 3.3.2.5).



The use of initiators such as **68** has been promoted for achieving higher molecular weights or higher conversions in conventional polymerization and for the production of block and graft copolymers. The use and applications of multifunctional initiators in the synthesis of block and graft copolymers is briefly described in Section 7.6.1.

3.3.4 Photochemical Initiators

Photoinitiation is most commonly used in curing or crosslinking processes and in initiating graft copolymerization. Major applications include inks and adhesives and the technologies such as laser direct imaging, holography and stereolithography. Photoinitiation also finds utility in small scale kinetic and mechanistic studies (*e.g.* pulsed laser polymerization, Section 4.5.2). Some approaches to living radical polymerization also make use of photoinitiation (Sections 3.3.4.2, 9.3.2 and 9.3.3).

General concepts have been discussed in Section 3.1.8. General reviews on photoinitiation include those by Pappas,²⁶²⁻²⁶⁴ Bassi,²⁶⁵ Mishra²⁶⁶ and Oster and Yang²⁶⁷ and Gruber.²⁶⁸ The applications of azo-compounds and peroxides as photoinitiators are considered in the sections on those initiators (see 3.3.1.1.2, 3.3.2.1.2, & 3.3.2.3.2). References to reviews on specific photoinitiators are given in the appropriate section below.

3.3.4.1 Aromatic carbonyl compounds

Many reviews have been written on the photochemistry of aromatic carbonyl compounds²⁶⁹ and on the use of these compounds as photoinitiators.²⁷⁰⁻²⁷⁵ Primary radicals are generated by one of the following processes:

(a) A unimolecular fragmentation involving, most commonly, either α -scission (Scheme 3.48; *e.g.* benzoin ethers, acylphosphine oxides)

$$\overset{O}{\xrightarrow{}} \overset{hv}{\xrightarrow{}} \overset{O}{\xrightarrow{}} \overset{$$

Scheme 3.48

or β -scission (Scheme 3.49; *e.g.* α -haloketones).



Examples of scission of bonds separated from the carbonyl group by a double bond or an aromatic ring are also known. Thus, the benzil monooxime (69) undergoes γ -scission (Scheme 3.50) (possibly by consecutive α - then β -scissions).



(b) A bimolecular process involving direct abstraction of hydrogen from a suitable donor (Scheme 3.51; *e.g.* with hydrocarbons, ethers, alcohols),



Scheme 3.51

or sequential electron and proton transfer (Scheme 3.52; *e.g.* with amines, thiols). The reaction pathway followed depends on whether H-donors or electron acceptors are present and the relative strengths of the bonds to the α - and β -substituents.



3.3.4.1.1 Benzoin and related compounds

Benzoin and a wide variety of related compounds (*e.g.* **12, 70-74**) have been extensively studied both as initiators of polymerization and in terms of their general photochemistry.^{271,273} The acetophenone chromophore absorbs in the near UV (300-400 nm). In the absence of hydrogen atom donors the mechanism of

99

radical generation is usually depicted as excitation to the $S_1(n,\pi^*)$ state followed by intersystem crossing to the $T_1(n,\pi^*)$ state and fragmentation; typically by α -scission (Scheme 3.53).



The benzoin ethers (75, R⁻alkyl; R'-H) and the α -alkyl benzoin derivatives (75, R=H, alkyl; R'=alkyl) undergo α -scission with sufficient facility that it is not quenched by oxygen or conventional triplet quenchers.²⁷⁶ This means that the initiators might be used for UV-curing in air. Unfortunately, it does not mitigate the usual effects of air as an inhibitor (Section 5.3.2). The products of α -scission (Scheme 3.53) are a benzoyl radical (13) and an α -substituted benzyl radical (76) both of which may, in principle, initiate polymerization.^{276,277}



Scheme 3.53

It should be pointed out that not all benzoin derivatives (75) are suitable for use as photoinitiators. Benzoin esters (75, R=acyl) undergo a side reaction leading to furan derivatives. Aryl ethers (75, R=aryl) undergo β -scission to give a phenoxy radical (an inhibitor) in competition with α -scission (Scheme 3.54). Benzoin derivatives with α -hydrogens (75 R'=11) are readily autoxidized and consequently can have poor shelf lives.

There are contradictory reports that phenyl glycolate esters (*e.g.* 72) undergo photochemistry analogous to the benzoin derivatives. However, a recent study²⁷⁸ suggests that the α -scission pathway is not significant. Photoinitiation with 72 generally involves hydrogen abstraction from solvent, monomer or other molecules of the initiator to form an initiating species and a relatively unreactive ketyl radical that decays by dimerization.²⁷⁸



Depending on the nature of the substituent R', the radical **76** (Scheme 3.53) may be slow to add to double bonds and primary radical termination can be a severe complication (see 3.2.9).^{39,40,279} The problems associated with formation of a relatively stable radical are mitigated with certain α -alkoxy (77) and α -alkanesulfonyl derivatives (79).²⁸⁰ In both cases the substituted benzyl radicals formed by α -scission (78 and 80 respectively) can themselves undergo a facile fragmentation to form a more reactive radical which is less likely to be involved in primary radical termination (Scheme 3.55, Scheme 3.56).



The acyl phosphonates, acyl phosphine oxides and related compounds (*e.g.* **81**, **82**) absorb strongly in the near UV (350-400 nm) and generally decompose by α -scission in a manner analogous to the benzoin derivatives.²⁸¹⁻²⁸⁵ Quantum yields vary from 0.3 to 1.0 depending on structure. The phosphinyl radicals are highly reactive towards unsaturated substrates and appear to have a high specificity for addition *vs* abstraction (see 3.4.3.2).

101



Klos *et al.*²⁸⁶ described a range of polymerizable benzoin derivatives as photoinitiators (*e.g.* **83**, **84**). These and other polymeric photoinitiators have advantages as initiators over low molecular weight analogs in circumstances where migratory stability is a problem.²⁸⁷⁻²⁸⁹

3.3.4.1.2 Carbonyl compound-tertiary amine systems

Photoredox systems involving carbonyl compounds and amines are used in many applications. Carbonyl compounds employed include benzophenone and derivatives, α -diketones [*e.g.* benzil, camphoroquinone (**85**),^{290,291} 9,10-phenanthrene quinone], and xanthone and coumarin derivatives. The amines are tertiary and must have α -hydrogens [*e.g.* N,N-dimethylaniline, Michler's ketone (**86**)]. The radicals formed are an α -aminoalkyl radical and a ketyl radical.



The reaction between the photoexcited carbonyl compound and an amine occurs with substantially greater facility than that with most other hydrogen donors. The rate constants for triplet quenching by amines show little dependence on the amine α -C-H bond strength. However, the ability of the amine to release an electron is important.²⁹² This is in keeping with a mechanism of radical generation which involves initial electron (or charge) transfer from the amine to the photoexcited carbonyl compound. Loss of a proton from the resultant complex (exciplex) results in an α -aminoalkyl radical which initiates polymerization. The

102

concurrently formed ketyl radicals are generally slow to initiate polymerization and consequently primary radical termination is a common complication with these initiator systems.

The electron transfer step is typically fast and efficient. Griller *et al.*²⁹² measured absolute rate constants for decay of benzophenone triplet in the presence of aliphatic tertiary amines in benzene as solvent. Values lie in the range $3-4\times10^9$ M⁻¹ s⁻¹ and quantum yields are close to unity.



Visible light systems comprising a photoreducible dye molecule $(e.g. 87)^{293}$ or an α -diketone $(e.g. 85)^{290}$ and an amine have also been described. The mechanism of radical production is probably similar to that described for the ketone amine systems described above (*i.e.* electron transfer from the amine to the photoexcited dye molecule and subsequent proton transfer). Ideally, the dye molecule is reduced to a colorless byproduct.

More efficient systems can be constructed by having the two components of the photoredox system in the one molecule.²⁹⁴

3.3.4.2 Sulfur compounds

The S-S linkage of disulfides and the C-S linkage of certain sulfides can undergo photoinduced homolysis. The low reactivity of the sulfur-centered radicals in addition or abstraction processes means that primary radical termination can be a complication. The disulfides may also be extremely susceptible to transfer to initiator (C₁ for **88** is *ca* 0.5, Sections 6.2.2.2 and 9.3.2). However, these features are used to advantage when the disulfides are used as initiators in the synthesis of telechelics²⁹⁵ or in living radical polymerizations.²⁹⁶ The most common initiators in this context are the dithiuram disulfides (**88**) which are both thermal and photochemical initiators. The corresponding monosulfides [*e.g.* (**89**)] are thermally stable but can be used as photoinitiators. The chemistry of these initiators is discussed in more detail in Section 9.3.2.



3.3.5 Redox Initiators

The early history of redox initiation has been described by Bacon.²²⁸ The subject has also been reviewed by Misra and Bajpai,²⁹⁷ Bamford²⁹⁸ and Sarac.²³⁰ The mechanism of redox initiation is usually bimolecular and involves a single electron transfer as the essential feature of the mechanism that distinguishes it from other initiation processes. Redox initiation systems are in common use when initiation is required at or below ambient temperature and they are frequently used for initiation of emulsion polymerization.

Common components of many redox systems are a peroxide and a transition metal ion or complex. The redox reactions of peroxides are covered in the sections on those compounds. Discussion on specific redox systems can be found in sections on diacyl peroxides (3.3.2.1.5), hydroperoxides (3.3.2.5), persulfate (3.3.2.6.1) and hydrogen peroxide (3.3.2.6.2).

Numerous redox systems have been described which do not involve peroxides including many metal ion free systems such as the photoredox reaction involving carbonyl compounds and tertiary amines (3.3.4.1.2). The following two sections describe redox systems based on the use of metal complexes and simple organic molecules. Various transition metal salts or complexes oxidize or reduce organic substrates by single electron transfer and radicals formed from the organic compound may initiate polymerization.²⁹⁸ We focus on metal complex-organic halide (3.3.5.1), and ceric ion-organic substrate systems (3.3.5.2).

3.3.5.1 Metal complex-organic halide redox systems

Metal complex-organic halide redox initiation is the basis of ATRP. Further discussion of systems in this context will be found in Section 9.4. The kinetics and mechanism of redox and photoredox systems involving transition metal complexes in conventional radical polymerization have been reviewed by Bamford.²⁹⁸

One photoredox system which has seen significant use comprises a transition metal in a low, typically zero, oxidation state (*e.g.* Mo(CO)₆, Re(CO)₆) and an organic halide. Radical production involves single electron transfer from the metal to the halogen substituent of the alkyl halide which then fragments to form a halide ion and an alkyl radical.²⁹⁹ Accordingly, the organic fragment of the alkyl halide should be a good electron acceptor, for example, CCl₄, CHCl₃, α -haloketones, α -haloesters. The use of polymeric halo compounds allows this chemistry to be used in the preparation of block and graft copolymers (Section 7.6.2).^{300,301}

The metal complexes most commonly used in these photoredox systems are manganese and rhenium carbonyls. The proposed mechanism of the photoredox

reaction involving $Mn_2(CO)_{10}$ is represented schematically as follows (Scheme 3.57). Quantum yields for photoinitiation are high.²⁹⁸ Redox couples involving similar metal complexes and an electron deficient monomer (typically a fluoro-olefin) have also been described.²⁹⁸



3.3.5.2 Ceric ion systems

Ceric ions oxidize various organic substrates and the mechanisms typically involve radical intermediates.³⁰² When conducted in the presence of a monomer these radicals may initiate polymerization.

The reaction of ceric ion with alcohols,³⁰³ amides and urethanes³⁰⁴ is thought to involve single electron transfer to the ceric ion and loss of a proton to give the corresponding oxygen- or nitrogen-centered radical (Scheme 3.58). The reaction may involve ligation of cerium. Mechanisms for ceric ion oxidation of alcohols which yield α -hydroxyalkyl radicals as initiating species have also been proposed.

Ceric ions react rapidly with 1,2-diols. There is evidence for chelation of cerium and these complexes are likely intermediates in radical generation.^{305,306} The overall chemistry may be understood in terms of an intermediate alkoxy radical which undergoes β -scission to give a carbonyl compound and a hydroxyalkyl radical (Scheme 3.59). However, it is also possible that there is concerted electron transfer and bond-cleavage. There is little direct data on the chemical nature of the radical intermediates.

The specificity for reaction with 1,2-diols over mono-ols and 1,3-diols accounts for the finding that oxidation of PVA gives specific cleavage of the 1,2-diol groups present as a consequence of head addition to monomer (see 4.4.3.2). The 1,3-glycol units in PVA also complex ceric ion and, while these complexes decompose only slowly under normal conditions, they undergo a facile photoinduced decomposition to generate initiating species.³⁰⁷



The reaction of ceric ions with polymer-bound functionalities gives polymerbound radicals. Thus, one of the major applications of ceric ion initiation chemistry has been in grafting onto starch, cellulose,^{305,306,308} polyurethanes and other polymers.³⁰⁴ The advantage of this over conventional initiating systems is that, ideally, no low molecular weight radicals which might give homopolymer contaminant are formed.

The ceric ion also is also known to trap carbon-centered radicals (initiatorderived species, propagating chains) by single electron transfer (Scheme 3.60).

$$m \cdot + Ce^{4+} \longrightarrow mn_{+} + Ce^{3+}$$

Scheme 3.60

3.3.6 Thermal Initiation

This section describes polymerizations of monomer(s) where the initiating radicals are formed from the monomer(s) by a purely thermal reaction (*i.e.* no other reagents are involved). The adjectives, thermal, self-initiated and spontaneous, are used interchangeably to describe these polymerizations which have been reported for many monomers and monomer combinations. While homopolymerizations of this class typically require above ambient temperatures, copolymerizations involving certain electron-acceptor-electron-donor monomer pairs can occur at or below ambient temperature.

Aspects of thermal initiation have been reviewed by Moad *et al.*,³⁰⁹ Pryor and Laswell,³¹⁰ Kurbatov,³¹¹ and Hall.³¹² It is often difficult to establish whether initiation is actually a process involving only the monomer. Trace impurities in the monomers or the reaction vessel may prove to be the actual initiators. Purely thermal homopolymerizations to high molecular weight polymers have only been demonstrated unequivocally for S and its derivatives and MMA. For these and other systems, the identity of the initiating radicals and the mechanisms by which they are formed remain subjects of controversy.

3.3.6.1 Styrene homopolymerization

The thermal polymerization of S has a long history.³¹⁰ The process was first reported in 1839, though the involvement of radicals was only proved in the 1930s. Carefully purified S undergoes spontaneous polymerization at a rate of *ca* 0.1% per hour at 60 °C and 2% per hour at 100 °C. At 180 °C, 80% conversion of monomer to polymer occurs in approximately 40 minutes. Polymer production is accompanied by the formation of S dimers and trimers which comprise *ca* 2% by weight of total products. The dimer fraction consists largely of *cis*- and *trans*-1,2-diphenylcyclobutanes (**90** and **91**) while the stereoisomeric tetrahydronaphthalenes (**92** and **93**) are the main constituents of the trimer fraction.³¹³



The two most widely accepted mechanisms for the spontaneous generation of radicals from S are the biradical mechanism (top half of Scheme 3.61) first proposed by Flory³¹⁴ and the Mayo³¹⁵ or MAH (molecule assisted homolysis) mechanism (lower part of Scheme 3.61).



The Mayo mechanism involves a thermal Diels-Alder reaction between two molecules of S to generate the adduct **95** which donates a hydrogen atom to another molecule of S to give the initiating radicals **96** and **97**. The driving force for the molecule assisted homolysis is provided by formation of an aromatic ring. The Diels-Alder intermediate **95** has never been isolated. However, related compounds have been synthesized and shown to initiate S polymerization.³¹⁰



The identification of both phenylethyl and 1-phenyl-1,2,3,4tetrahydronaphthalenyl end groups in polymerizations of styrene retarded by $FeCl_3/DMF$ provides the most compelling evidence for the Mayo mechanism.³¹⁶ The 1-phenyl-1,2,3,4-tetrahydronaphthalenyl end group is also seen amongst other products in the TEMPO mediated polymerization of styrene.^{317,318} However, the mechanism of formation of radicals **96** in this case involves reaction of the nitroxide with the Diels-Alder dimer (Scheme 3.63). The mechanism of nitroxide mediated polymerization is discussed further in Section 9.3.6.



Scheme 3.63

The Diels-Alder intermediate (95) is also rapidly trapped by aromatization in the presence of acids (Scheme 3.64). Thus, the observation by Buzanowski *et al.*,³¹⁹ of dramatically lower rates for S polymerizations carried out in the presence of various acid catalysts, is circumstantial evidence for the Mayo mechanism.



Scheme 3.64

Despite the body of evidence in favor of the Mayo mechanism, the formation of diphenylcyclobutanes (90, 91) must still be accounted for. It is possible that they arise via the 1,4-diradical 94 and it is also conceivable that this diradical is an intermediate in the formation of the Diels-Alder adduct 95 (Scheme 3.64) and could provide a second (minor) source of initiation. Direct initiation by diradicals is suggested in the thermal polymerization of 2,3,4,5,6-pentafluorostyrene where transfer of a fluorine atom from Diels-Alder dimer to monomer seems highly unlikely (high C-F bond strength) and for derivatives which cannot form a Diels-Alder adduct.

Thermal initiation of styrene has been shown to be third order in monomer. The average rate constants for third order initiation determined by Hui and Hamielec is $k_i = 10^{5.34} e^{(13810/T)} (M^2 s^{-1})$.³²⁰ The rate constant for formation of the Mayo dimer determined in trapping experiments with nitroxides (Scheme 3.63) or acid (Scheme 3.64) as $k_D = 10^{4.4} e^{(93500/RT)} (M^{-1} s^{-1})^{321}$ is substantially higher than is required to account for the rate of initiation. It has been postulated that radical production proceeds mainly through the isomer of **95** in which the phenyl group is axial.^{313.322} Both isomers of **95** can give rise to the trimers **92**, possibly by an ene reaction between **95** and S. However, the trimers **92** could also be formed by cage combination of radicals **96** and **97**.

3.3.6.2 Acrylate homopolymerization

Various acrylates, methacrylates and related compounds have been reported to undergo spontaneous polymerization.³¹⁰ A complication in studying thermal polymerization of MMA is the difficulty in eliminating impurity initiated polymerization. The monomer is extremely difficult to purify or retain in a "pure" state. These problems have led some to question whether there is any true spontaneous initiation.³²³ It is, in any event, clear that the rate of thermal polymerization of MMA is substantially less than that of S at the same temperature (at least 70-fold less at 90 °C).^{310,324}



Scheme 3.65

Dimer and trimer byproducts have been isolated from MMA polymerizations and these are suggestive of 1,4-diradical intermediates.³²⁵⁻³²⁸ Lingnau and Meyerhoff³²⁵ found that rates of spontaneous polymerization of MMA were substantially higher in the presence of transfer agents (RH). They were able to isolate the compound (**98**) that might come from trapping of the biradical intermediate (Scheme 3.65).

3.3.6.3 Copolymerization

Monomers that are strong electron donors may undergo spontaneous copolymerization with strong electron acceptor monomers by a radical mechanism. In certain cases homopolymers formed by an ionic mechanism accompany copolymer formation.^{312,329}

Examples where radical initiation is believed to be dominant include:

- (a) S with MAII,^{330,331} AA,³³² AN,^{333,334} vinylidene cyanide,³³⁵ or dimethyl 1,1dicyanoethane-2,2-dicarboxylate.³¹²
- (b) *p*-Methoxystyrene with trimethyl ethylenetricarboxylate³¹² or dimethyl cyanofumarate.³³⁶
- (c) 1,2-Dimethoxyethylene with MAH.³³⁷
- (d) Vinyl sulfides with a range of electrophilic monomers.³³⁸

Various mechanisms have been proposed to explain the initiation processes. The self-initiated copolymerizations of the monomer pairs S-MMA and S-AN proceed at substantially faster rates than pure S polymerization. For S-AN³³⁴ and S-MAH³³¹ the mechanism of initiation was proposed to be analogous to that of S homopolymerization (Scheme 3.62) but with acrylonitrile acting as the dienophile in the formation of the Diels-Alder adduct (Scheme 3.66).

Various oligomers formed by Diels-Alder/ene reactions are observed.^{333,334} For S-MAH polymerization Sato *et al.*³³¹ used spin trapping to identify the initiating species. On the other hand, in the case of S-AN copolymerization, the

finding that acid catalysts do not affect the rate of polymerization argues against the involvement of this species in the initiation mechanism.³³³ Acid catalysts, which effectively trap the Diels-Alder intermediate (**95**) by aromatization (see 3.3.6.1), have been found to lower the rate of thermal S homopolymerization dramatically.³¹⁹



Scheme 3.66

Other postulated mechanisms for spontaneous initiation include electron transfer followed by proton transfer to give two monoradicals,³³⁸ hydrogen atom transfer between a charge-transfer complex and solvent,³³⁰ and formation of a diradical from a charge-transfer complex.³³⁹

Hall^{312,329} has proposed a unifying concept based on tetramethylenes (resonance hybrids of 1,4-diradical and zwitterionic limiting structures - Scheme 3.67) to rationalize all donor-acceptor polymerizations. The predominant character of the tetramethylenes (zwitterionic or diradical) depends on the nature of the substituents.^{312,340} However, more evidence is required to prove the more global application of the mechanism.



Scheme 3.67

3.4 The Radicals

In this section, the reactions undergone by radicals generated in the initiation or chain transfer processes are detailed. Emphasis is placed on the specificity of radical-monomer reactions and other processes likely to take place in polymerization media under typical polymerization conditions. The various factors important in determining the rate and selectivity of radicals in addition and substitution processes have already been discussed in general terms in Sections 2.3 and 2.4 respectively.

3.4.1 Carbon-Centered Radicals

Carbon-centered radicals are produced as primary radicals in the decomposition of azo-compounds (*e.g.* Scheme 3.68),





as secondary radicals from peroxides by β -scission of the initially formed acyloxy or alkoxy radicals (*e.g.* Scheme 3.69),

$$\begin{array}{c} \bigcap_{\substack{H \\ H \\ H}} O \\ CH_3(CH_2)_{10} \cdot \overline{C} \cdot O \cdot O - \overline{C} \cdot (CH_2)_{10}CH_3 \longrightarrow \left[\begin{array}{c} O \\ CH_3(CH_2)_{10} \cdot \overline{C} \cdot O \cdot \right] \xrightarrow{-CO_2} CH_3(CH_2)_9CH_2 \cdot \\ LPO \\ Scheme 3.69 \end{array} \right]$$

and by transfer reactions (e.g. Scheme 3.70).

 $R \cdot + PhCH_3 \rightarrow PhCH_2 \cdot$

Scheme 3.70

In this section we consider the properties and reactions of three classes of carbon-centered radicals: alkyl radicals (3.4.1.1), aryl radicals (3.4.1.2) and acyl radicals (3.4.1.3).

3.4.1.1 Alkyl radicals

Primary radical termination involving alkyl radicals is described in Sections 2.5 and 7.4.3. Their reactions with monomers are also discussed in Sections 2.3 (fundamental aspects) and 4.5.4 (model propagation radicals). Their chemistry has been reviewed by Fischer and Radom,³⁴¹ Giese,^{342,343} Tedder,³⁴⁴ Beckwith,³⁴⁵ Rüchardt,⁷⁶ and Tedder and Walton.

Alkyl radicals, when considered in relation to heteroatom-centered radicals (*e.g. t*-butoxy, benzoyloxy), show a high degree of chemo- and regiospecificity in their reactions. A discussion of the factors influencing the rate and regiospecificity of addition appears in Section 2.3. Significant amounts of head addition are observed only when addition to the tail-position is sterically inhibited as it is in α , β -disubstituted monomers. For example, with β -alkylacrylates, cyclohexyl

radicals give head addition and the proportion can be correlated with the steric size of the β -substituent.³⁴⁸

Rate constants for reactions of carbon-centered radicals for the period through 1982 have been compiled by Lorand³⁴⁹ and Asmus and Bonifacic³⁵⁰ and for 1982-1992 by Roduner and Crocket.³⁵¹ The recent review of Fischer and Radom should also be consulted.³⁴¹ Absolute rate constants for reaction with most monomers lie in the range 10^5 - 10^6 M⁻¹ s⁻¹. Rate data for reaction of representative primary, secondary, and tertiary alkyl radicals with various monomers are summarized in Table 3.6.

In the absence of heteroatom containing substituents (e.g. halo-, cyano-), at or conjugated with the radical center, carbon-centered radicals have nucleophilic character. Thus, simple alkyl radicals generally show higher reactivity toward electron-deficient monomers (e.g. acrylic monomers) than towards electron-rich monomers (e.g. VAc, S) – Table 3.6.

Simple alkyl radicals thus seem ideal as initiating species:

- (a) They show a high degree of regiospecificity for tail vs head addition.
- (b) They show a high specificity for addition *vs* abstraction. Rate constants for hydrogen abstraction from monomers and solvents (*e.g.* toluene) are generally much smaller (*ca* 100-fold less) than those for addition to double bonds.
- (c) They react rapidly. Side reactions such as primary radical termination are thus minimal.

Thus alkyl radicals do not give unwanted end-group functionality and the kinetics of initiation are comparatively uncomplicated. However, this situation can be perturbed by substitution at or near the radical center.

3.4.1.1.1 α-Cyanoalkyl radicals

Thermal or photochemical decomposition of azonitriles (*e.g.* AIBN) affords α -cyanoalkyl radicals (Scheme 3.71).²⁹

$$\begin{array}{cccc} & & & CN & & & CN \\ H_3C - & C - N = N - & C - CH_3 & \longrightarrow & H_3C - & C + \\ CH_3 & CH_3 & & CH_3 & & CH_3 \end{array}$$

AIBN

Scheme 3.71

The reactions of cyanoisopropyl radicals with monomers have been widely studied. Methods used include time resolved EPR spectroscopy,³⁵² radical trapping³⁵³⁻³⁵⁵ and oligomer^{60,356} and polymer end group determination.^{60,357-364} Absolute³⁴¹ and relative reactivity data obtained using the various methods (Table 3.6) are in broad general agreement.

| Radical | Temp | $k_{\rm s} \times 10^{-5}$ ${ m M}^{-1} { m s}^{-1}$ | k/k _s | | | | |
|---|-------------------|---|-------------------|--------|------|-----|-----|
| | °C | | AMS | MA | MMA | AA | MAA |
| ĊН₂ОН | 25 | 0.23 | 1.2 | 31 | 26 | - | - |
| ĊH₂Ph | 25 | 0.011 | 0.77 | 0.39 | 1.9 | - | - |
| i J | 69 ^b | 3.2 ^h | 0.6 | 3.5 | - | 4.7 | 4.4 |
| CH3• | 65 | - | 1.16 | 1.3 | 1.8 | - | - |
| CH ₃ • | 25 | 2.6 | 1.2 | 1.3 | 1.9 | - | - |
| $\dot{C}H_2C(=O)C(CH_3)_3$ | 25 ^b | 19 | 2.1 | 0.26 | 0.68 | - | - |
| ĊH₂CN | 25 | 3.8 | 1.7 | 0.29 | 0.63 | - | - |
| CH₃ĊHPh | $100^{\rm f}$ | - | 1.1 | 1.5 | 1.9 | - | - |
| (СН ₃)2СН | 60 | 4.7 | - | - | 0.3 | - | - |
| Ò. | 20^{i} | - | 0.93 | 6.7 | 5.0 | - | - |
| (СН ₃) ₂ С́ОН | 25 | 7.3 | 0.27 | 47 | 22 | - | - |
| (CH ₃) ₃ C • | 25 | 1.3 | 0.45 | 8.5 | 5.1 | - | - |
| (CH ₃) ₂ CCO ₂ CH ₃ | $60^{\rm f}$ | - | | - | 0.7 | - | - |
| (CH ₃)2 ^C CO ₂ C(CH ₃)3 | 25 | 0.055 | 1.1 | 0.21 | 0.67 | - | - |
| PhCH ₂ C(CO ₂ Et) ₂ | 60 ^a | - | • | 0.0071 | - | - | - |
| (CH ₃)2ČCN | 30 ^d | - | 1.06 ^e | - | 0.56 | - | - |
| (CH ₃)2ČCN | 60 ^{d,f} | 0.03 | 0.95 | 0.3 | 0.56 | - | - |
| (CH ₃)2CCN | $100^{\rm f}$ | - | 0.87 | - | 0.56 | - | - |
| (CH ₃) ₂ ĊCN | $25^{\rm f}$ | 0.024 | 0.96 | 0.15 | 0.66 | - | - |
| \bigcirc · | 25 ^j | 1100 | - | - | 1.6 | - | - |
| \bigtriangledown · | 60 ^k | - | 1.31 | 0.73 | 1.16 | - | - |
| cı 🔊• | 25^{1} | - | - | 0.66 | 1.03 | - | - |

 Table 3.6 Kinetic Data for Reactions of Carbon-Centered Radicals

a ln acetic acid. b In acetonitrile. c 40 °C in toluene. d In benzene. Value based on the reported rate constant for addition to MAN⁶⁰ and the value of k_{MAN}/k_S shown. e 45 °C. f In toluene. g 30 °C, in ethyl acetate. h Reported values corrected using a more recent rate constant for the 5-hexenyl clock.³⁶⁵ i In methylene chloride. j In Freon 113. k In carbon tetrachloride. I In aqueous acetone.

Table 3.6 (continued)

| | k/ks | | | | Refs. | Radical | |
|--------|------|--------|--------|-------|-------------------|-------------|---|
| AN | MAN | VAc | PAc | VC | PhCH ₃ | | |
| 47 | 29 | 0.025 | 0.029 | 1.2 | - | 341 | Ċн₂он |
| 2.0 | 6.0 | 0.013 | 0.042 | - | - | 341,366 | $\dot{C}H_2Ph$ |
| 7.5 | - | - | - | - | - | 18 | i j |
| 2.2 | 2.7 | 0.038 | - | - | 0.000015 | 367,368 | CH ₃ • |
| 2.4 | 3.0 | 0.053 | 0.046 | 0.077 | - | 341 | CH3• |
| 0.28 | 0.49 | 0.034 | 0.046 | 0.037 | - | 341.369 | ĊH ₂ C(=O)C(CH ₃) ₃ |
| 0.29 | 0.45 | 0.034 | 0.031 | 0.031 | - | 341 | ĊH₂CN |
| 5.0 | - | - | - | - | - | 370,371 | CH₃ĊHPh |
| - | - | - | - | - | - | 372 | (CH ₃)₂ĊH |
| 24 | 13 | 0.12 | - | 0.016 | - | 342 | Ò. |
| 205 | 62 | 0.010 | 0.0066 | - | - | 341 | (СН ₃)2 ^Ċ OH |
| 40 | 13 | 0.032 | 0.013 | 0.12 | - | 341.373,374 | (CH ₃) ₃ C • |
| - | - | 0.038 | - | - | - | 357,375 | (CH ₃)₂ĊCO₂CH ₃ |
| 0.45 | 0.81 | 0.0032 | 0.011 | - | - | 341,357,375 | (CH ₃)2CCO2C(CH3)3 |
| 0.0088 | - | 0.0074 | - | - | - | 376 | PhCH ₂ C(CO ₂ Et) ₂ |
| - | - | 0.02 | - | - | - | 357,358 | (CH ₃)2 ^Ċ CN |
| 0.44 | 0.34 | 0.03 | - | 0.04° | - | 60,357-363 | (CH ₃)2 ^C CN |
| - | 0.49 | 0.05 | - | - | - | 358,362,364 | (CH ₃)₂ČCN |
| 0.84 | 0.44 | 0.017 | 0.033 | 0.25 | - | 341.352 | (CH ₃)2 ^C CN |
| - | - | - | - | - | 0.015 | 377 | \bigtriangledown . |
| 1.14 | 1.30 | 0.14 | 0.14 | 0.18 | - | 378 | \bigcirc . |
| 0.68 | - | - | - | - | - | 379 | cı∕∕∕ |

Absolute rate constants for addition reactions of cyanoalkyl radicals are significantly lower than for unsubstituted alkyl radicals falling in the range 10^3 - 10^4 M⁻¹s⁻¹.³⁴¹ The relative reactivity data demonstrate that they possess some electrophilic character. The more electron-rich VAc is very much less reactive than the electron-deficient AN or MA. The relative reactivity of styrene and acrylonitrile towards cyanoisopropyl radicals would seem to show a remarkable temperature dependence that must, from the data shown (Table 3.6), be attributed to a variation in the reactivity of acrylonitrile with temperature and/or other conditions.

Cyanoisopropyl radicals generally show a high degree of specificity in reactions with unsaturated substrates. They react with most monomers (*e.g.* S, MMA) exclusively by tail addition (Scheme 3.4). However, Bevington *et al.*^{113,362} indicated that cyanoisopropyl radicals give *ca* 10% head addition with VAc at 60 °C and that the proportion of head addition increases with increasing temperature.

 α -Cyanoalkyl radicals show relatively little tendency to abstract hydrogen from monomer, solvent, or polymer even in relation to other alkyl radicals.³⁸⁰ However, these radicals, like other carbon-centered radicals,²⁸ react with oxygen at diffusion controlled rates (Section 3.2.5). For polymerizations carried out in poorly degassed media, it has been proposed^{29,30} that abstraction products, peroxide linkages, and other defect structures may arise through the intermediacy of a alkylperoxy radical (Scheme 3.10).

The α -cyanoalkyl radicals can, in principle, react with substrates either at carbon or at nitrogen (Scheme 3.72). However, reaction at nitrogen to give a ketenimine is usually only observed in cases of reactions with other radicals (Section 5.2.2.1.3) or organometallic reagents.³⁸¹ There is a report of a ketenimine structure being formed in a radical substitution reaction (Section 4.4.2). There is as yet no evidence for ketenimine being produced in reactions with monomers or spin traps^{7,382} despite several studies aimed specifically at detecting such processes. It is anticipated that reaction through nitrogen would be favored by steric hindrance at the site of attack `and by electron donating substituents on the substrate. It is also likely that addition via the nitrogen will be required to observe this pathway).

A number of reports^{104,108,383} indicate that primary radical termination can be important during polymerizations initiated by azonitriles. However, for the case of S polymerization initiated by AIBN, NMR end group determination⁷ shows that primary radical termination is of little importance except when very high rates of initiation are employed (*e.g.* with high initiator concentrations at high temperatures). Cyanoalkyl radicals give a mixture of combination and disproportionation in their reactions with other radicals (see also Sections 2.5, 7.4.3.2, 7.4.3.3 and 7.4.3.5). This finding is significant for those who use azonitriles as initiators in producing telechelics (Section 7.5.1).



Scheme 3.72

3.4.1.2 Aryl radicals

Aryl radicals are produced in the decomposition of alkylazobenzenes and diazonium salts, and by β -scission of aroyloxy radicals (Scheme 3.73). Aryl radicals have been reported to react by aromatic subsitution (*e.g.* of S⁸) or abstract hydrogen (*e.g.* from MMA¹⁰) in competition with adding to a monomer double bond. However, these processes typically account for $\leq 1\%$ of the total. The degree of specificity for tail *vs* head addition is also very high. Significant head addition has been observed only where tail addition is retarded by steric factors (*e.g.* methyl crotonate¹⁰ and β -substituted methyl vinyl ketones^{379,384}).

Absolute rate constants for the attack of aryl radicals on a variety of substrates have been reported by Scaiano and Stewart $(Ph^{\cdot})^{377}$ and Citterio *et al.* $(p\text{-ClPh}^{\cdot})^{.379,384}$ The reactions are extremely facile in comparison with additions of other carbon-centered radicals $[e.g. k(S) = 1.1 \times 10^8 \text{ M}^{-1} \text{ s}^{-1} \text{ at } 25 \text{ °C}]^{.377}$ Relative reactivities are available for a wider range of monomers and other substrates (Table 3.6).^{377,378,385-387} Phenyl radicals do not show clear cut electrophilic or nucleophilic behavior.

3.4.1.3 Acyl radicals

Phenacyl radicals are produced by photodecomposition of initiators containing the phenone moiety (Scheme 3.74). These initiators include benzoin derivatives and acylphosphine oxides (see 3.3.4.1.1). Acyl radicals can be formed by
hydrogen abstraction from aldehydes. Various other sources have been described.³⁸⁸



Scheme 3.74

The general chemistry of acyl radicals has been recently reviewed.³⁸⁸ Acyl radicals have nucleophilic character. Absolute rate constants for substituted phenacyl radical addition to BA have been reported to be in the range $1.3-5.5 \times 10^5$ M⁻¹ s⁻¹ at 25 °C.²⁸⁵

Acyl radicals undergo decarbonylation. For aliphatic acyl radicals the rate constant for decarbonylation appears to be correlated with the stability of the alkyl radical formed. Values of the decarbonylation rate constant range from 4 s⁻¹ (for CH₃C(•)O) to 1.5×10^8 s⁻¹ [for (CH₃)₂C(Ph)C(•)O] at 298 °C.³⁸⁸ The loss of carbon monoxide from phenacyl radicals is endothermic and the rate constant is extremely low (*ca* 10^{-8} s⁻¹ at 298 °C).³⁸⁸ Consequently, the reaction is not observed during polymerization experiments.

3.4.2 Oxygen-Centered Radicals

Oxygen-centered radicals are arguably the most common of initiator-derived species generated during initiation of polymerization and many studies have dealt with these species. The class includes alkoxy, hydroxy and acyloxy radicals and the sulfate radical anion (formed as primary radicals by homolysis of peroxides or hyponitrites) and alkylperoxy radicals (produced by the interaction of carbon-centered radicals with molecular oxygen or by the induced decomposition of hydroperoxides).

There is an excellent, if non critical, compilation of absolute and relative rate data for reactions of oxygen-centered radicals covering the literature through 1982³⁸⁹ and for 1982-1992.³⁹⁰ Selected data from these and other sources are summarized in Table 3.7 and Table 3.8. The reactions of oxygen-centered radicals and their use in organic synthesis has been recently reviewed by Hartung *et al.*³⁹¹

The pathways whereby oxygen-centered radicals interact with monomers show marked dependence on the structure of the radical (Table 3.8). For example, with MMA the proportion of tail addition varies from 66% for *t*-butoxy to 99% for isopropoxycarbonyloxy radical. The reactions of oxygen-centered radicals are discussed in detail in the following sections.

3.4.2.1 Alkoxy radical

Alkoxy radicals are frequently encountered as initiating species in polymerizations and have been the subject of numerous laboratory studies. Most

work has concentrated on the chemistry of *t*-butoxy radical and relatively little attention has been paid to the chemistry of other alkoxy radicals. The chemistry of alkoxy radicals has been the subject of several reviews.³⁹²⁻³⁹⁵

Table 3.7 Selected Rate Data for Reactions of Oxygen-Centered Radicals^a

| Radical | Temp | $k_{\rm S} \times 10^{-5}$ | | | | $k/k_{\rm S}$ | | | | |
|---|------|----------------------------|-----|------|------|---------------|------|------|-------|------------------|
| | °C | $M^{-1}s^{-1}$ | AMS | МА | MMA | AN | MAN | VAc | PhCH₃ | refs. |
| (CH ₃) ₃ CO• | 60 | $\sim 9^{b}$ | 1.3 | 0.06 | 0.28 | 0.05 | 0.12 | 0.06 | 0.19 | 8,12,22,396 |
| (CH ₃) ₂ (Ph)CO• | 60 | ~30° | - | - | 0.1 | - | - | - | - | 72 |
| HO• | 60 | - | 1.2 | 0.34 | 0.63 | - | - | - | - | 397 |
| HO• | 25 | 200000 | - | - | 1.0 | 0.27 | 0.96 | - | - | |
| PhCO ₂ • | 24 | 5100 | - | - | - | - | - | - | - | 398 |
| PhCO ₂ • | 60 | - | - | 0.05 | 0.12 | < 0.05 | - | 0.36 | - | 399 |
| PhCO ₂ • | 60 | - | - | 0.02 | 0.11 | 0.02 | - | 0.26 | - | 10.11.22.400.401 |

a Overall reactivity. Reaction pathways are shown in Table 3.8. b Based on rate constant for β -scission as clock reaction¹⁰ and the yield of methyl radical-derived products observed in bulk S polymerization.⁸ c Based on the analysis of Rizzardo *et al.*⁷² but assuming a rate constant for β -scission for cumyloxy radical of 1.5×10^6 at 60 °C.

3.4.2.1.1 t-Butoxy radicals

The reactions of *t*-butoxy radicals are amongst the most studied of all radical processes. These radicals are generated by thermal or photochemical decomposition of peroxides or hyponitrites (Scheme 3.75).



Scheme 3.75

| Monomer | Radical | | Pathy | vay ^a | |
|---------------------------------------|---|------------------|-------|------------------|----|
| | | Α | в | C | D |
| L | (CH ₃) ₃ CO• ^{b,c,8} | 100 | - | - | - |
| | (CH ₃) ₂ CH ₂ O+ ^{b,c,402} | 100 | - | - | - |
| CH ₂ =C \checkmark B | CH ₃ CH ₂ O ^{-b.c.402} | 100 | - | - | - |
| | (CH ₂) ₂ (Ph)CO ^{•b.c.72} | 100 | - | - | - |
| | HO- ^{6.397} | 87 | 6 | - | 7 |
| · · · · · · · · · · · · · · · · · · · | PhCO ₂ • ^{c.d.8} | 80 | 6 | - | 14 |
| | (CH ₃) ₂ CHOCO ₂ • ^{b.c.403} | 95 | - | - | 5 |
| | (CH ₃) ₃ CO ^{•b.c.f.396} | 85 | - | 15 | - |
| K ° | HO• ^{6.397} | 83 | 3 | 5 | 9 |
| CH | | | | | |
| | | | | | |
| | | | | | |
| | | | | | |
| Á // 🔪 | | | | | |
| $ \sum$ | | | | | |
| | | | | | |
| н | (CH ₃) ₃ CO• ^{b,c,10} | 83 | 2 | - | 15 |
| | HO• ^{6.397} | 80 | 17 | - | 3 |
| CH ₂ =C \checkmark B | PhCO ₂ • ^{c,d,10} | 84 | 16 | - | - |
| 1 с_о_сн | - | | | | |
| | | | | | |
| ··· 0 | | | | | |
| C | (CH ₃) ₃ CCH ₂ C | ~66 ^g | - | ~33 ^g | h |
| ſ | (CH ₃) ₂ CO• | 66 | - | 30 | 4 |
| | (CH ₂) ₂ CO ^{•b,c,f,12} | 70 | - | 26 | 3 |
| | (CH ₂) ₂ (Ph)CO ^{•b.c.72} | 88 | - | 12 | - |
| СН₂=С′ ← в 🗸 р | (CH ₂) ₂ CHO ^{•b.e,123} | 92 | 6 | 8 | - |
| | CH ₂ CH ₂ O ^{•b.c.402} | 87 | 7 | 5 | 2 |
| | HO+ ^{b.397} | 93 | - | <l<sup>e</l<sup> | - |
| ۸ Ö | PhCO _{2*} e,d.10 | >99 | | - | - |
| | (CH ₃) ₂ CHOCO ₂ • ^{b,188} | | | | |

| Table 3.8 | Specificity Observed in the Reactions of Oxygen-Centered Radicals |
|-----------|---|
| | with Various Monomers at 60 °C |

| Table 3.8 (continued) | | | | | |
|--|--|-----------|----------|------------------|--------|
| Monomer | Radical | | Path | way ^a | |
| | | A | В | C | D |
| CH ₂ =C | (CH ₃) ₃ CO• ^{b,c,404} PhCO ₂ • ^{c,d,401} | 100 98 | 2 | - | - |
| CH ₂ =C B | (CH ₃) ₃ CO• ^{b.c.405} | 74 | - | 26 | - |
| CN CH ₂ =C | (CH ₃) ₃ CO• ^{b.c.404} _D PhCO₂• ^{c.d.401} | 79 76 | 15 24 | - | 6 - |
| ✓ O-C-CH ₃ ^A 0 | (out) on belige | 49 | | 49 | 4 |
| $CH_2 = C + B$ $CH_2 = C + B$ $O - C - CH_3$ | (CH ₃) ₃ CO- | 48 | - | 48 | 4 |

a Relative yields of products formed by pathway indicated. All data rounded to nearest 1%. A dash indicates that the product was not detected. b In bulk monomer, c Yields have been normalized to exclude β -scission products. d In 50% v/v acetone/monomer. e Total abstraction by benzoyloxy and phenyl radicals. f Addition:abstraction ratio shows solvent dependence.^{21,396} g Values approximate. Radical gives mainly β -scission and 1,5 H atom transfer h Product detected

In a polymerization reaction they may:

- (a) Initiate a chain by adding to the double bond of a monomer.
- (b) Abstract a hydrogen atom from the monomer, solvent, or another component of the reaction mixture to afford a new radical species and *t*-butanol (primary radical transfer).
- (c) Undergo β -scission to give methyl radicals and acetone (*e.g.* Scheme 3.6).

The relative importance of these processes depends strongly on the particular monomer(s) and the reaction conditions.

In contrast to most other oxygen-centered radicals [*e.g.* benzoyloxy (3.4.2.2.1), hydroxy (3.4.2.3)], *t*-butoxy radicals and other *t*-alkoxy radicals (3.4.2.1.2) show relatively high regiospecificity in reactions with carbon-carbon double bonds (Table 3.8). Nonetheless, significant amounts of head addition are observed with the halo-olefins,^{24,406} simple alkenes,⁴⁰⁷ vinyl acetate and methyl acrylate.⁴⁰⁴ Head addition is generally not observed with 1,1-disubstituted monomers. The exception is vinylidene fluoride^{24,406} where head addition predominates (Section 2.4). With allyl methacrylate (**99**)⁴⁰⁸ and allyl acrylate (**100**),⁴⁰⁹ *t*-butoxy radicals give substantially more addition to the acrylate double bond than to the allyl double bond (see Figure 3.4).

Studies of the relative reactivity of *t*-butoxy radicals with substituted styrenes,⁴¹⁰ toluenes^{411,412} and other substrates (see 2.3.3) indicate that they are slightly electrophilic in character. However, Sato and Otsu ¹³ found that the order of reactivity of *t*-butoxy radicals towards a series of monomers was different from that of the more electrophilic benzoyloxy radicals. They concluded that product radical stability was important in determining reactivity. Cuthbertson *et al.*⁴⁰⁶ examined the reactions of *t*-butoxy radicals toward fluoro-olefins and found a pattern of reactivities more characteristic of a nucleophilic species. The strength of the bond being formed plays an important role in determining regiospecificity. The factors influencing the specificity and rate of addition are discussed in greater detail in Section 2.3.2.



Figure 3.4 Relative reactivity of indicated site towards *t*-butoxy radicals for allyl methacrylate (99) and allyl acrylate (100)

Many investigations^{12,13,21,396,413,414} have shown that the reaction of *t*-butoxy radicals with monomers bearing sp^3 hydrogens invariably produces a mixture of initiating radicals arising from hydrogen abstraction and addition (Table 3.8). Simple alkenes (*e.g.* butenes),⁴⁰⁷ vinyl ethers⁴¹⁵ and higher acrylates (*e.g.* BMA - Figure 3.5)^{12,416} may give predominantly abstraction. The specificity seen in attack on the ester group has been attributed to polar factors.⁴¹⁶ The positions α - and β -to the ester oxygen are strongly deactivated towards attack by *t*-butoxy radicals.



Figure 3.5 Relative reactivity of indicated site towards *t*-butoxy radicals.

t-Butoxy radicals also undergo unimolecular fragmentation to produce acctone and methyl radicals (Scheme 3.6). Significant amounts of the β -scission products are obtained in the presence of even the most reactive monomers (*e.g.* S⁸). The reactions of methyl radicals have been discussed above (see 3.4.1.1).

The relative amounts of double bond addition, hydrogen abstraction and β -scission observed are dependent on the reactivity and concentration of the particular monomer(s) employed and the reaction conditions. Higher reaction temperatures are reported to favor abstraction over addition in the reaction of *t*-butoxy radicals with AMS⁴¹³ and cyclopentadiene.⁴¹⁷ However, the opposite trend is seen with isobutylene.^{23,24}

Pioncering work by Walling³⁹⁴ established that the specificity shown by *t*-butoxy radical is solvent dependent. Work^{21,22,396} on the reactions of *t*-butoxy radicals with a series of α -methylvinyl monomers has shown that polar and aromatic solvents favor abstraction over addition, and β -scission over either addition or abstraction. Recently, Weber and Fischer⁴¹⁸ and Tsentalovich *et al.*⁴¹⁹ reported absolute rate constants for β -scission of *t*-butoxy radicals in various solvents. These studies indicate that β -scission is strongly solvent dependent while abstraction is relatively insensitive to solvent.

| | | β-Scission | n | Abstra | action from a | cyclohexane |
|-------------------------|-------------------|-------------|------------------|------------------------------|--------------------|------------------------|
| solvent | k_{β} | E_{a} | $\log(A/s^{-1})$ | $k_{\rm abs} \times 10^{-5}$ | $E_{\mathfrak{a}}$ | $\log(A/M^{-1}s^{-1})$ |
| | s ^{-1 b} | kJ mol⁻¹ | | $M^{-1}s^{-1}b$ | kJ mol⁻¹ | |
| Fiigen 113 ^a | 8050 | 52.7 | 13.2 | 8.3 | 11.9 | 8.0 |
| DTBP | 12000 | 50.5 | 12.9 | - | - | - |
| C_6H_6 | 20300 | 48.7 | 12.8 | 9.6 | 12.1 | 8.5 |
| C_6H_5F | 21400 | 47.5 | 12.7 | 9.8 | 14.6 | 8.2 |
| a 1,1,2-trichlore | -1,2,2-trifl | uoroethane. | b Temperatu | re 298 K. | | |

 Table 3.9. Kinetic Data for Reactions of t-Butoxy Radicals in Various

 Solvents.

3.4.2.1.2 Other t-alkoxy radicals

Various *t*-alkoxy radicals may be formed by processes analogous to those described for *t*-butoxy radicals. The data available suggest that their propensities for addition *vs* abstraction are similar.⁷² However, rate constants for β -scission of *t*-alkoxy radicals show marked dependence on the nature of substituents α to oxygen (Figure 3.6).^{210,420,421} Polar, steric and thermodynamic factors are all thought to play a part in favoring this trend.³⁹³



Figure 3.6 Relative rate constants for β -scission of *t*-alkoxy radicals at 60 °C.⁴²¹

Thus, even if *t*-amyloxy radicals (101) show similar specificity for addition *vs* abstraction to *t*-butoxy radicals, abstraction will be of lesser importance.^{422,423} The reason is that most *t*-amyloxy radicals do not react directly with monomer. They undergo β -scission and initiation is mainly by ethyl radicals. Ethyl radicals are much more selective and give addition rather than abstraction. This behavior has led to *t*-amyl peroxides and peroxyesters being promoted as superior to the corresponding *t*-butyl derivatives as polymerization initiators.⁴²³

$$\begin{array}{c} CH_{3} & CH_{3} \\ C_{2}H_{5} - C - O \cdot \\ CH_{3} \end{array} \qquad \begin{array}{c} CH_{3} \\ Ph - C - O \cdot \\ CH_{3} \end{array} \qquad \begin{array}{c} CH_{3} \\ Ph - C - O \cdot \\ CH_{3} \end{array} \qquad \begin{array}{c} CH_{3} \\ CH_{3} \end{array} \qquad \begin{array}{c} CH$$

1,5-H atom transfer is another important unimolecular pathway for *t*-alkoxy radicals that have a suitably disposed hydrogen atom (Scheme 3.76).^{421,424,425}





Scheme 3.76

The rate constant of β -scission of cumyloxy radicals (102) is also significantly greater than that for *t*-butoxy radicals.^{26,420} β -Scission gives exclusively acetophenone and methyl radicals. For the case of S or MMA polymerization initiated by cumyloxy radicals at 60 °C, the proportion of methyl radical initiation is six-fold greater than is seen with *t*-butoxy radicals.⁷² The absolute rate constant for β -scission of 102 has been shown to be solvent dependent. The absolute rate constant (2.6×10⁵ s⁻¹ at 30 °C in CCl₄) increases *ca* seven-fold over the series CCl₄, C₆H₆, C₆H₅Cl, (CH₃)₃COH, CH₃CN, CH₃COOH.⁴²⁶ The rate constant for abstraction from cyclohexane remains at 1.2 ± 0.1×10⁶ M⁻¹ s⁻¹ in all solvents. For cumyloxy, and other *t*-alkoxy radicals, β -scission is much more sensitive to temperature than either addition or abstraction (Figure 3.1) such that at high temperatures it is likely to be the major process even in the presence of very reactive substrates.

3.4.2.1.3 Primary and secondary alkoxy radicals

Relatively few studies have dealt with the reactions of primary and secondary alkoxy radicals (isopropoxy, methoxy, *etc.*) with monomers. These radicals are conveniently generated from the corresponding hyponitrites (Scheme 3.77).^{123,402}

$$\begin{array}{ccc} CH_3 & CH_3 & CH_3 \\ H-C-O-N=N-O-C-H & & & & H-C-O \\ CH_3 & CH_3 & & & CH_3 \end{array}$$

Scheme 3.77

Primary and secondary alkoxy radicals generally show a reduced tendency to abstract hydrogen or to undergo β -scission when compared to the corresponding *t*-alkoxy radical.^{123,402} This has been correlated with the lesser nucleophilicity of these radicals.⁴²⁷

It has been suggested^{123,402} that primary and secondary alkoxy radicals may react with S by donation of a hydrogen atom to the monomer and production of an aldehyde.

3.4.2.2 Acyloxy and alkoxycarbonyloxy radicals

Aroyloxy radicals are formed by thermal or photochemical decomposition of diaroyl peroxides (see 3.3.2.1) and aromatic peroxyesters (3.3.2.3) (Scheme 3.78); alkoxycarbonyloxy radicals are similarly produced from peroxydicarbonates (3.3.2.2).

126 The Chemistry of Radical Polymerization

Aliphatic acyloxy radicals undergo facile fragmentation with loss of carbon dioxide (Scheme 3.69) and, with few exceptions,⁴²⁸ do not have sufficient lifetime to enable direct reaction with monomers or other substrates. The rate constants for decarboxylation of aliphatic acyloxy radicals are in the range $1-10 \times 10^9$ M⁻¹ s⁻¹ at 20 °C.⁴²⁹ Ester end groups in polymers produced with aliphatic diacyl peroxides as initiators most likely arise by transfer to initiator (see 3.3.2.1.4). The chemistry of the carbon-centered radicals formed by β -scission of acyloxy radicals is discussed above (see 3.4.1).



Scheme 3.78

3.4.2.2.1 Benzoyloxy radicals

Benzoyloxy radicals are electrophilic and show higher reactivity towards electron-rich (*e.g.* S, VAc) than electron-deficient (*e.g.* MMA, AN) monomers (Table 3.7).^{401,430} Product studies on the reactions of benzoyloxy radicals with simple olefins and monomers^{8,10,168,399,401,430-433} show that they have remarkably poor regiospecificity when adding to carbon-carbon double bonds. Their reactions invariably give a mixture of products from head addition and tail addition (Scheme 3.4 and Table 3.8).^{8,10,401,433} They also display a marked propensity for aromatic substitution.^{8,41,398} On the other hand, compared with alkoxy radicals, they show little tendency to abstract hydrogen.¹⁰



Scheme 3.79

Additions of benzoyloxy radicals to double bonds^{434,435} and aromatic rings (Scheme 3.79)¹⁴⁸ are potentially reversible. For double bond addition, the rate constant for the reverse fragmentation step is slow (k=10²-10³ s⁻¹ at 25 °C) with respect to the rate of propagation during polymerizations. Thus, double bond addition is effectively irreversible. However, for aromatic substrates, the rate of the reverse process is extremely fast. While the aromatic substitution products may be trapped with efficient scavenging agents (*e.g.* a nitroxide^{8,41} or a transition

metal¹⁶⁹), they are generally not observed under polymerization conditions.⁹ A different situation may pertain when redox initiation is used, as the oxidants employed may be effective radical traps. A small proportion of aromatic benzoate residues can be detected in high conversion PS prepared with benzoyl peroxide. However, it is likely that these arise through attack on PS rather than S.^{9,154}

The rate of β -scission of benzoyloxy radicals is such that in most polymerizations initiated by these radicals both phenyl and benzoyloxy end groups will be formed (Scheme 3.4). A reliable value for the rate constant for β -scission would enable the absolute rates of initiation by benzoyloxy radical to be estimated. Various values for the rate constant for β -scission have appeared. Many of the early estimates are low. The activation parameters (in CCl₄ solvent) determined by Chateauneuf *et al.*³⁹⁸ are $\log_{10} A = 12.6$ and $E_a = -35.97$ kJ mol⁻¹ which corresponds to a rate constant of 9×10^6 s⁻¹ at 60 °C.

The rate constant for β -scission is dependent on ring substituents. Rate constants for radicals X-C₆H₄CO₂• are reported to increase in the series where X is p-F $\leq p$ -CH₃O $\leq p$ -CH₃ $\sim p$ -Cl \leq H $\leq m$ -Cl.³⁹⁸ There is qualitative evidence that the relative rates for β -scission and addition are insensitive to solvent changes. For benzoyloxy radicals, similar relative reactivities are obtained from direct competition experiments¹⁰ as from studies on individual monomers when β -scission is used as a clock reaction.^{399,401}

The rate constants for benzoyloxy and phenyl radicals adding to monomer are high (> $10^7 \text{ M}^{-1} \text{ s}^{1}$ for S at 60 °C - Table 3.7). In these circumstances primary radical termination should have little importance under normal polymerization conditions. Some kinetic studies indicating substantial primary radical termination during S polymerization may need to be re-evaluated in this light.¹⁶¹ Secondary benzoate end groups in PS with BPO initiator may arise by head addition or transfer to initiator (Section 8.2.1).

3.4.2.2.2 Alkoxycarbonyloxy radicals

The chemistry of alkoxycarbonyloxy radicals in many ways parallels that of the aroyloxy radicals (*e.g.* benzoyloxy, see 3.4.2.2.1). Products attributable to the reactions of alkoxy radicals generally are not observed. This indicates that the rate of β -scission is slow relative to the rate of addition to monomers or other substrates.^{188,431}

The alkoxycarbonyloxy radicals show little tendency to abstract hydrogen.^{188,431} For example, in the reaction of isopropoxycarbonyloxy radicals with MMA, hydrogen abstraction, while observed, is a minor pathway ($\leq 1\%$). When isopropoxycarbonyloxy radicals abstract hydrogen, isopropanol is the expected byproduct since the intermediate acid undergoes facile decarboxylation. Formation of isopropanol is not evidence for the involvement of isopropoxy radicals (Scheme 3.80).



Scheme 3.80

Isopropoxycarbonyloxy radicals undergo facile reaction with aromatic substrates (e.g. toluene) by reversible aromatic substitution.^{169,436} Isopropoxycarbonyloxy radicals react with S to give ring substitution (ca 1%) as well as the expected double bond addition.⁴⁰³

3.4.2.3 Hydroxy radicals

Hydroxy radicals are produced by redox reactions involving hydrogen peroxide (see 3.3.2.6.2). They can also be generated in organic solution by thermal decomposition of α -hydroperoxydiazenes (see 3.3.3.1).

The transient radicals produced in reactions of hydroxy radicals with vinyl monomers in aqueous solution have been detected directly by EPR⁴³⁷⁻⁴³⁹ or UV spectroscopy.^{440,441} These studies indicate that hydroxy radicals react with monomers and other species at or near the diffusion-controlled limit (Table 3.7). However, high reactivity does not mean a complete lack of specificity. Hydroxy radicals are electrophilic and trends in the relative reactivity of the hydroxy radicals toward monomers can be explained on this basis.³⁹⁷

Grant *et al.*³⁹⁷ examined the reactions of hydroxy radicals with a range of vinyl and α -methylvinyl monomers in organic media. Hydroxy radicals on reaction with AMS give significant yields of products from head addition, abstraction and aromatic substitution (Table 3.8) even though resonance and steric factors combine to favor "normal" tail addition. However, it is notable that the extents of abstraction (with AMS and MMA) are less than obtained with *t*-butoxy radicals and the amounts of head addition (with MMA and S) are no greater than those seen with benzoyloxy radicals under similar conditions. It is clear that there is no direct correlation between reaction rate and low specificity.

Yields of aromatic substitution on S and AMS obtained by Grant *et al.*³⁹⁷ should be regarded as minimum yields until the efficiency of trapping of the cyclohexadienyl radicals under their reaction conditions is known. This may help reconcile the finding that, in aqueous media, aromatic substitution is reported to be the main reaction pathway.⁴⁴¹ Grant *et al.*³⁹⁷ also found that aromatic substitution on S proceeded by preferential *para* attack. This preference agrees with the calculated relative reactivity of the ring carbons based on frontier electron densities, but is otherwise unprecedented.⁴⁴²

3.4.2.4 Sulfate radical anion

The sulfate radical anion is formed by thermal, photochemical or redox decomposition of persulfate salts (41, see 3.3.2.6.1). Consequently, it is usually used in aqueous solution. However, crown ether complexes or alkylammonium salts may be used to generate the sulfate radical anion in organic solution (see 3.3.2.6.1).

Two pathways for the reaction of sulfate radical anion with monomers have been described (Scheme 3.81).²⁵² These are: (A) direct addition to the double bond or (B) electron transfer to generate a radical cation. The radical cation may also be formed by an addition-elimination sequence. It has been postulated that the radical cation can propagate by either cationic or a radical mechanism (both mechanisms may occur simultaneously). However, in aqueous media the cation is likely to hydrate rapidly to give a hydroxyethyl chain end.



Scheme 3.81

The preferred initiation pathway is dependent on the particular monomer involved and the reaction conditions. Generally radical cation formation (by either mechanism) is facilitated by low pH. The failure to detect an intermediate sulfate adduct led workers to propose that reactions of the sulfate radical anion with electron-rich alkenes and S derivatives proceeded by pathway (B) over a wide range of pH and reaction conditions.⁴⁴³⁻⁴⁴⁵ However, other workers rationalized similar data by allowing the initial formation of a sulfate adduct (pathway A).⁴⁴⁶ Detection of an intermediate in the reaction of sulfate radical anion with S⁴⁴⁷ or with cyclohexene²⁴² clearly points to addition being a major pathway in those cases. Moreover, PS formed with persulfate initiation is known to possess a high proportion of sulfate end groups.⁴⁴⁸⁻⁴⁵¹ Thus, the bulk of available evidence suggests that in initiation of S polymerization there is initial formation of a sulfate adduct (pathway A) and that, radical cations, if formed, are produced by subsequent elimination (Scheme 3.81).

In the case of electron-deficient monomers (*e.g.* acrylics) it is accepted that reaction occurs by initial addition of the sulfate radical anion to the monomer. Reactions of sulfate radical anion with acrylic acid derivatives have been shown to give rise to the sulfate adduct under neutral or basic conditions but under acidic conditions give the radical cation probably by an addition-elimination process.

The Chemistry of Radical Polymerization

Hydroxy radical and sulfate radical anion, though they may sometimes give rise to similar products, show quite different selectivity in their reactions with unsaturated substrates. In particular, the sulfate radical anion has a somewhat lower propensity for hydrogen abstraction than the hydroxyl radical. For example, the sulfate radical anion shows little tendency to abstract hydrogen from methacrylic acid.²⁵²

Sulfate radical anion may be converted to the hydroxyl radical in aqueous solution. Evidence for this pathway under polymerization conditions is the formation of a proportion of hydroxy end groups in some polymerizations. However, the hydrolysis of sulfate radical anion at neutral pII is slow ($k=10^7 \text{ M}^{-1} \text{ s}^{-1}$) compared with the rate of reaction with most monomers ($k=10^8-10^9 \text{ M}^{-1} \text{ s}^{-1}$, Table 3.7)⁴⁴⁰ under typical reaction conditions. Thus, hydrolysis should only be competitive with addition when the monomer concentration is very low. The formation of hydroxy end groups in polymerizations initiated by sulfate radical anion can also be accounted for by the hydration of an intermediate radical cation or by the hydrolysis of an initially formed sulfate adduct either during the polymerization or subsequently.

3.4.2.5 Alkylperoxy radicals

Alkylperoxy radicals are generated by the reactions of carbon-centered radicals with oxygen and in the induced decomposition of hydroperoxides (Scheme 3.82). Their reactions have been reviewed by Howard⁴⁵² and rate constants for their self reaction and for their reaction with a variety of substrates including various inhibitors have been tabulated.⁴⁵³

$$\begin{array}{cccc} & & & & & & & & \\ H_3C-C& & + & O_2 & & & & \\ CH_3 & & & & & \\ H_3C-C-O-O & & & & \\ H_3C-C-O-O-H & & & & \\ H_3C-C-O-O & & & \\ CH_3 & & & \\ H_3C-C-O-O & & \\ CH_3 & & \\ \end{array}$$

Scheme 3.82

Because of the importance of hydroperoxy radicals in autoxidation processes, their reactions with hydrocarbons are well known. However, reactions with monomers have not been widely studied. Absolute rate constants for addition to common monomers are in the range 0.09-3 M^{-1} s⁻¹ at 40 °C. These are substantially lower than k_i for other oxygen-centered radicals (Table 3.7).⁴⁵⁴

Epoxide formation may be a side reaction occurring during initiation by *t*-butylperoxy radicals. The mechanism proposed for this process is as follows (Scheme 3.83).²¹¹



Scheme 3.83

3.4.3 Other Heteroatom-Centered Radicals

Various other heteroatom-centered radicals have been generated as initiating species. These include silicon-, sulfur-, selenium- (see 3.4.3.1), nitrogen- and phosphorus-centered species (see 3.4.3.2). Kinetic data for reactions of these radicals with monomers is summarized in Table 3.10.

3.4.3.1 Silicon-centered radicals

Silicon centered radicals can be generated by transfer to silanes and by photolysis of polysilanes. Rate constants for addition to monomer are several orders of magnitude higher than similar carbon centered radicals.455,456 The radicals have nucleophilic character.

| Radical | Temp. | ks | | | ł | $k/k_{\rm S}^{\rm a}$ | | | |
|--|-------|---------------------|------|--------|-----------|-----------------------|-------------------|-------------------|-------|
| | °C | $M^{-1} s^{-1}$ | AMS | MA | MMA | AN | MAN | VAc | Refs. |
| (c-C ₆ H ₁₃)₂Si∙ | c | 2.2×10^{8} | - | - | 2.1 | - | - | - | 294 |
| (C ₂ H ₅) ₂ Si• | c | 1.6×10^{8} | - | - | 4.8 | 0.63 | 4.7 | - | 294 |
| C₂H₅S∙ | 60 | - | - | 0.036 | - | - | - | - | 457 |
| t-C₄H₃ S∙ | 60 | - | - | - | 1.0^{b} | - | 0.63 ^b | 0.13 ^b | 458 |
| CH ₃ (CH ₂) ₂ CH ₂ S• | 60 | - | - | - | 0.17 | - | - | - | 459 |
| PhS• | 23 | 2.0×10^{7} | - | - | 0.16 | - | - | - | 460 |
| PhS. | 60 | - | - | - | 0.2 | - | 0.1 | 0.002 | 461 |
| ρ-CIPhS∙ | 23 | 5.1×10^{7} | - | - | 0.10 | 0.0090 | 0.045 | 0.0009 | 462 |
| PhC(O)S+ | 22 | - | - | 0.03 | 0.12 | 0.0091 | - | 0.0025 | 463 |
| PhSe• | 23 | 2.2×10^{6} | 0.76 | 0.0078 | 0.019 | 0.0064 | 0.012 | 0.0005 | 464 |
| Ph ₂ P(O)• | с | 6.0×10 ⁷ | - | 0.58 | 1.33 | 0.33 | 0.83 | 0.027 | 465 |
| Ph ₂ P(O)• | 20 | 1.1×10^{7} | 1.27 | - | 1,45 | - | - | 0.25 | 466 |
| $(CH_3O)_2P(O)$ • | c.d | 2.2×10 ⁸ | - | 0.077 | 0.26 | 0.26 | 0.42 | 0.013 | 465 |

Table 3.10 Selected Rate Data for Reactions of Heteroatom-Centered Radicals

a Data rounded to two significant figures. b k/k_{MMA} . c Room temperature. relative reactivities for VAc and AN have been reported at 60 °C.⁴⁶⁷ d Similar

3.4.3.2 Sulfur- and selenium-centered radicals

Thiyl radicals are formed by transfer to thiols or by thermal or photochemical decomposition of disulfides (Scheme 3.84).



Scheme 3.84

Most studies have concerned the kinetics of arenethiyl radicals with monomers including S and its derivatives⁴⁶⁸⁻⁴⁷² and MMA.^{469,473} The radicals have electrophilic character and add more rapidly to electron-rich systems (Table 3.10). Relative reactivities of the monomers towards the benzoylthiyl radical have also been examined.⁴⁶³

It is established that the initial reaction involves predominantly tail addition to monomer.⁴⁷³ There is no evidence that abstraction competes with addition. It should be noted that the addition of arenethiyl radicals to double bonds is readily reversible.

A study on the kinetics of the reactions of phenylseleno radicals with vinyl monomers has also been reported.⁴⁶⁴

3.4.3.3 Phosphorus-centered radicals

Phosphinyl radicals (*e.g.* **103-107**) are generated by photodecomposition of acyl phosphinates or acyl phosphine oxides (see 3.3.4.1.1)^{282,466,474,475} or by hydrogen abstraction from the appropriate phosphine oxide.⁴⁶⁷



The reactivities of the various phosphinyl radicals with monomers have been examined (Table 3.10).^{283,465,467,475} Absolute rate constants are high, lying in the range 10^{6} - 10^{8} M⁻¹ s⁻¹ and show some solvent dependence. The rate constants are higher in aqueous acctonitrile solvent than in methanol. The high magnitude of the rate constants has been linked to the pyramidal structure of the phosphinyl radicals.⁴⁶⁵

The phosphinyl radicals (103-107) all show nucleophilic character (e.g. VAc is substantially less reactive than the acrylic monomers). However, the

132

nucleophilicity varies according to the number of oxygen substituents on phosphorous.^{465,467}

3.5 Techniques

The low concentration of initiator residues in polymers formed by radical polymerization means that they can usually only be observed directly in exceptional circumstances or in very low molecular weight polymers (Section 3.5.3). Thus, the study of the reactions of initiator-derived radicals with monomers has seen the development of some novel techniques. Three basic approaches have been employed. These involve:

- (a) Kinetic studies involving the observation of the disappearance of reactants and/or appearance of products using some time resolved spectroscopic technique (most often EPR spectroscopy or UV-visible spectophotometry Section 3.5.1).
- (b) Isolation of the initiator-monomer reaction by employing a reagent designed to trap the first-formed adduct. This usually involves conducting the polymerization in the presence of an appropriate inhibitor (Section 3.5.2).
- (c) Labeling the initiator such that the initiator-derived residues in the polymer can be more readily detected and quantified by chemical or spectroscopic analysis (Section 3.5.4).

3.5.1 Kinetic Studies

Time resolved EPR spectroscopy and UV-visible spectophotometry have proved invaluable in determining the absolute rate constants for radical-monomer reactions. The results of many of these studies are summarized in the Tables included in the previous section (3.4). Absolute rate constants for the reactions of carbon-centered radicals are reported in Table 3.6. These include *t*-butyl³⁷⁴ and cyanoisopropyl³⁵² radicals.

3.5.2 Radical Trapping

Radical traps used for the study of radical monomer reactions should meet a number of criteria:

- (a) The trap should ideally show a degree of specificity for reaction with the propagating species as opposed to the initiator-derived radicals.
- (b) All products from the reaction with monomer should be trapped with equal efficiency.
- (c) The trapped products should be stable under the reaction conditions.

Various reagents have been employed as radical traps. Those most commonly encountered are summarized in Table 3.11. The advantages, limitations and applications of each are considered in the following sections.

The Chemistry of Radical Polymerization

3.5.2.1 Spin traps

In spin trapping, radicals are trapped by reaction with a diamagnetic molecule to give a radical product.⁴⁷⁶ This feature (*i.e.* that the free spin is retained in the trapped product) distinguishes it from the other trapping methods. The technique involves EPR detection of the relatively stable radicals which result from the trapping of the more transient radicals. No product isolation or separation is required. The use of the technique in studies of polymerization is covered in reviews by Kamachi⁴⁷⁷ and Yamada *et al.*⁴⁷⁸

| Table 3.11 Radica | l Trapping Agents | for Studying | Initiatio | m |
|-------------------|-------------------|--------------|-----------|---|
|-------------------|-------------------|--------------|-----------|---|

| Trap | Initiating radicals trapped | Section |
|------------------------|--|---------|
| spin traps: | | |
| nitroso-compounds | most radicals | 3.5.2.1 |
| nitrones | most radicals | 3.5.2.1 |
| transition metal ions: | | |
| cupric ions | nucleophilic carbon-centered radicals | 3.5.2.2 |
| ferric ions | nucleophilic carbon-centered radicals | 3.5.2.2 |
| titanous ions | electrophilic carbon-centered radicals | 3.5.2.2 |
| metal hydrides: | - | |
| mercuric hydride | electrophilic carbon-centered radicals | 3.5.2.3 |
| Group VI hydrides | carbon-centered radicals | 3.5.2.3 |
| nitroxides | carbon-centered radicals | 3.5.2.4 |
| AMS dimer | most radicals | 3.5.2.5 |

The two most commonly employed spin traps are 2-methyl-2-nitrosopropane (108) (more commonly known as nitroso-*t*-butane) and phenyl *t*-butyl nitrone (109); both trap radicals to yield nitroxides (Scheme 3.85, Scheme 3.86).

 $\begin{array}{ccc} & & & & & & \\ H_3C-\overset{}{\overset{}}_{} - & N=O & & & \\ H_3C-\overset{}{\overset{}}_{} - & N=O & & \\ CH_3 & & & CH_3 & \\ \end{array}$

108

Scheme 3.85



Chalfont *et al.*⁴⁷⁹ were the first to apply the spin trapping technique in the study of radical polymerization. They studied radicals produced during S polymerization initiated by *t*-butoxy radicals with **108** as the radical trap. Since

134

that time many other systems have been studied using this trap (108).^{477.478} The use of 2,4,6-tri-*t*-butylnitrosobenzene (111) in the study of polymerization, has been advocated by Savedoff and Ranby⁴⁸⁰ and by Lane and Tabner.⁴³³ This nitroso-compound is reported to be more thermally and photochemically stable than 108. However 111 reacts with propagating radicals to give a mixture of anilino radicals (110) and nitroxides (112) as shown in Scheme 3.87.^{481,482} The ratio of 110 to 112 depends on the structure of the propagating radical. Formation of 110 is favored when the radical trapped is more hindered and/or more electron rich.



Scheme 3.87

Nitrones are generally more stable than nitroso-compounds and are therefore easier to handle. However, the nitroxides formed by reaction with nitrones [*e.g.* phenyl *t*-butyl nitrone (109)]^{483,484} have the radical center one carbon removed from the trapped radical (Scheme 3.86). The EPR spectra are therefore less sensitive to the nature of that radical and there is greater difficulty in resolving and assigning signals. Nitrones are generally less efficient traps than nitroso-compounds.⁴⁷⁶

There are several limitations on the use of the spin trapping technique when quantitative results are required. These are:

- (a) Not all radicals are trapped at equal rates or with equal efficiency.⁴⁸⁵
- (b) The product nitroxides may not be stable under the reaction conditions. Nitroxide stability is strongly dependent on the nature of the trapped species. Nitroxides react with radicals at or near diffusion controlled rates and they can also undergo β -scission either to regenerate the trapped radical or to form a new radical.
- (c) Side reactions involving the trap and the monomer may give rise to products which complicate the interpretation of the EPR spectra. Various side reactions have been described in the literature:⁴⁷⁶ the nitroso-compound (**108**) reacts with α -methylvinyl monomers by an ene reaction (Scheme 3.88);¹⁸⁸ *t*-butyl radicals produced by thermal or photochemical decomposition of (**108**) are trapped as di-*t*-butylnitroxide.

Many of the above-mentioned complications can be avoided or allowed for by carrying out appropriate control experiments. A further difficulty lies with the

The Chemistry of Radical Polymerization

sensitivity of the method. Minor initiation pathways ($\leq 5\%$) are extremely difficult to determine.



3.5.2.2 Transition metal salts

Certain transition metal salts can be used as radical traps (Scheme 3.89, Scheme 3.90).⁴⁸⁶ These include various cupric (*e.g.* Cu(OAc)₂, CuCl₂, Cu(SCN)₂),^{18,168,393,432,487} ferric (*e.g.* FeCl₃),^{316,488} and titanous salts (*e.g.* TiCl₃).³⁷⁹ These traps react with radicals by ligand- or electron-transfer to give products which can be determined by conventional analytical techniques.

Scheme 3.89

The rate of oxidation/reduction of radicals is strongly dependent on radical structure. Transition metal reductants (*e.g.* Ti^{III}) show selectivity for electrophilic radicals (*e.g.* those derived by tail addition to acrylic monomers or alkyl vinyl ketones - Scheme 3.89)³⁷⁹ while oxidants (Cu^{II}, Fe^{III}) show selectivity for nucleophilic radicals (*e.g.* those derived from addition to S - Scheme 3.90).¹⁸ A consequence of this specificity is that the various products from the reaction of an initiating radical with monomers will not all be trapped with equal efficiency and complex mixtures can arise.



The facile and reversible reaction of propagating species with transition metal halide complexes to form a polymeric halo-compound is one of the key steps in atom transfer radical polymerization (ATRP, see Section 9.4).

136

3.5.2.3 Metal hydrides

Metal hydride trapping agents have been used extensively in studying the reaction of alkyl radicals with monomers.^{489,490}

Alkyl mercuric hydrides are generated *in situ* by reduction of an alkyl mercuric salt with sodium borohydride (Scheme 3.91). Their use as radical traps was first reported by Hill and Whitesides⁴⁹¹ and developed for the study of radicalolefin reactions by Giese,^{489,490} Tirrell⁴⁹² and coworkers. Careful choice of reagents and conditions provides excellent yields of adducts of nucleophilic radicals (*e.g. n*-hexyl, cyclohexyl, *t*-butyl, alkoxyalkyl) to electron-deficient monomers (*e.g.* acrylics).

A consequence of the selectivity for electrophilic radicals is that not all products are trapped with equal efficiency. With electron-rich monomers (*e.g.* S) oligomerization may complicate analysis. Other possible complications in the utilization of this method have been discussed by Russell.⁴⁹³

Group IV hydrides (R₃SnH, R₃GeH) have also been used as trapping reagents.^{494,495} The reduction of alkyl halides by stannyl or germyl radicals affords alkyl radicals. These react with the group IV hydrides to set up a radical chain (Scheme 3.92).⁴⁹⁵ The alkyl radicals may react with a substrate (*e.g.* monomer) in competition with being trapped by the hydride. Absolute rate constants for the reactions of group IV hydrides with radicals are known. Thus the H-atom transfer step may be used as a radical clock to calibrate radical-monomer reactions.²⁰ This technique has seen widespread use in the study of intramolecular radical reactions.³⁴⁵ One limitation of the use of the group IV hydrides as radical traps in the study of polymerization is that the stannyl and germyl radicals may themselves add monomer, *albeit* reversibly.



3.5.2.4 Nitroxides

A well-known feature of the chemistry of nitroxides (*e.g.* **113-115**) is that they combine with carbon-centered radicals at near diffusion-controlled rates to give alkoxyamines. This feature led to the use of nitroxides as the reagents of choice in the inhibitor method for the determination of initiator efficiency.⁹² Rizzardo and Solomon⁴⁹⁶ applied this chemistry to develop one of the most versatile techniques for examining the initiation step of polymerization. The method is reliant on the initiator-derived radicals either not reacting or reacting only slowly with the nitroxide while the propagating radicals are efficiently scavenged to yield stable alkoxyamines (Scheme 3.93). The technique has been successfully used by several groups to study the reactions of heteroatom-centered (ethoxy,⁴⁰² isopropoxy,^{123,402} *t*-butoxy,^{8,10,12,21,22,177,188,396,404,406-410,496,497 cumyloxy,⁷² other *t*-alkoxy,⁴²¹ benzoyloxy,^{8,10,22,41,166,188,401} isopropoxycarbonyloxy,¹⁸⁸ hydroxy,^{253,397} thiyl,^{458,461,498} phosphinyl^{467,474,499}) and more reactive carbon-centered radicals (methyl, undecyl, *t*-butyl, phenyl)^{8,10,22,41,177,424,425,500-504} with monomers. The reaction has also been employed to detect radical intermediates in organic reactions and to identify primary radicals produced from photoinitiators.⁴⁷⁴}



Busfield and coworkers extended the technique to the study of less reactive carbon-centered radicals $(e.g. \text{ cyanoisopropyl})^{353,354}$ and short propagating radicals⁵⁰⁵⁻⁵⁰⁷. The very low concentration of nitroxide required to allow limited propagation was maintained by feeding with a syringe pump.

The reaction between nitroxides and carbon-centered radicals occurs at near (but not at) diffusion controlled rates. Rate constants and Arrhenius parameters for coupling of nitroxides and various carbon-centered radicals have been determined.⁵⁰⁸⁻⁵¹¹ The rate constants (20 °C) for the reaction of TEMPO with primary, secondary and tertiary alkyl and benzyl radicals are 1.2, 1.0, 0.8 and 0.5×10^9 M⁻¹s⁻¹ respectively. The corresponding rate constants for reaction of **115** are slightly higher. If due allowance is made for the afore-mentioned sensitivity to radical structure⁵¹⁰ and some dependence on reaction conditions,⁵¹¹ the reaction can be applied as a clock reaction to estimate rate constants for reactions between carbon-centered radicals and monomers^{504,506,507,512} or other substrates.²⁰

Major advantages of this method over other trapping techniques are that typical conditions for solution/bulk polymerization can be employed and that a very wide range of initiating systems can be examined. The application of the

138

technique is greatly facilitated by the use of a nitroxide possessing a UV chromophore (e.g. 113-115) which simplifies product analysis by liquid chromatography with UV detection.

Nitroxides have the property of quenching fluorescence. Thus radical trapping with nitroxides containing fluorophores (*e.g.* 114) can be monitored by observing the appearance of fluorescence.⁵¹³⁻⁵¹⁵ The method is highly sensitive and has been applied to quantitatively determine radical yields in PLP experiments (Section 4.5.2).



Scheme 3.93 (T=115)

Some limitations of the method arise due to side reactions involving the nitroxide. However, such problems can usually be avoided by the correct choice of nitroxide and reaction conditions. Nitroxides, while stable in the presence of most monomers, may act as oxidants or reductants under suitable reaction conditions.⁵¹⁶ The induced decomposition of certain initiators (*e.g.* diacyl peroxides) can be a problem (Scheme 3.94).^{166,177} There is some evidence that nitroxides may disproportionate with alkoxy radicals bearing α -hydrogens.¹²³ Side reactions with thiols have also been identified.⁴⁹⁸



Scheme 3.94

The Chemistry of Radical Polymerization

Various light-induced reactions including hydrogen atom abstraction, electron transfer and β -scission occur under the influence of UV light.⁵¹⁷⁻⁵²¹ Certain radicals, for example cyclohexadienyl radicals (Scheme 3.95), are trapped by disproportionation rather than coupling.⁸ Nitroxides are also reported to react by hydrogen abstraction with molecules that are extremely good hydrogen donors [*e.g.* S dimer (**95**)³¹⁸ and the ketenimine (**10**).¹⁰³]



Scheme 3.95

The reaction of radicals with nitroxides is reversible.³⁰⁹ This means that the highest temperature that the technique can reasonably be employed at is $ca \ 80 \ ^{\circ}C$ for tertiary propagating species and $ca \ 120 \ ^{\circ}C$ for secondary propagating species.²² These maximum temperatures are only guidelines. The stability of alkoxyamines is also dependent on solvent (polar solvents favor decomposition) and the structure of the trapped species. This chemistry has led to certain alkoxyamines being useful as initiators of living polymerization (Section 9.3.6). At elevated temperatures nitroxides are observed to add to monomer albeit slowly.^{318,522,523}

3.5.2.5 *a-Methystyrene dimer*

Watanabe *et al.*^{25,524-528} applied AMS dimer (116) as a radical trap to examine the reactions of oxygen-centered radicals (*e.g. t*-butoxy, cumyloxy, benzoyloxy). AMS dimer (116) is an addition fragmentation chain transfer agent (see 6.2.3.4) and reacts as shown in Scheme 3.96. The reaction products are macromonomers and may potentially react further. The reactivity of oxygen centered radicals towards 116 appears to be similar to that of S.²⁵ Cumyl radicals are formed as a byproduct of trapping and are said to decay mainly by combination and disproportionation.





3.5.3 Direct Detection of End Groups

In favorable circumstances initiator-derived end groups may be detected by spectroscopic methods or by chemical or chromatographic analysis. Most of the methods are sensitive only to a given type of end group in a given class of polymer. However, they have the advantage that no special chemistry or isolation steps are required. The main disadvantages associated with these methods are that they require foreknowledge of what the end groups are likely to be and, in general, they can only be applied to low molecular weight polymers.

3.5.3.1 Infra-red and UV-visible spectroscopy

 $UV^{153,529}$ and IR spectroscopy^{94,530,531} have been used for polymer end group determination and to study the kinetics and efficiency of initiation of polymerization. These techniques are not universally applicable. Ideally, it is required (a) that the chromophores are in a clear region of the spectrum and (b) that the positions of the absorptions are sensitive to the chemical environment of the chromophore such that end groups can be distinguished from residual initiator and initiator-derived byproducts.

Garcia-Rubio *et al.*^{153,529} examined S and MMA polymerizations initiated by BPO and have shown that UV can be used to distinguish and quantitatively determine aliphatic and aromatic benzoate groups in MMA and S polymerizations.

Buback *et al.*^{94,531,532} applied FTIR to follow the course of the initiation of S polymerization by AIBN and to determine initiator efficiency. Contributions to the IR signal due to cyanoisopropyl end groups, AIBN, and the ketenimine can be separated using curve resolution techniques.

3.5.3.2 Nuclear magnetic resonance spectroscopy

The sensitivity of modern NMR allows initiator residues to be determined directly in polymers of moderate molecular weight where the desired signals are discrete from those of the backbone carbons.⁵³⁰ Many examples can be found in the literature.^{34,533-538} The molecular weight limit is imposed both by sensitivity and the dynamic range of the spectrometer. Both resolution and sensitivity improve with field strength of the NMR spectrometer. Thus, one strategy for improving the ease of end group detection is to use the highest practicable field strength.^{537,538}

In some cases, it is possible to suppress NMR signals due to backbone carbons or hydrogens thus allowing obscured end group resonances to be observed. Several basic methods have been described in the literature. These are:

- (a) Subtraction of the spectrum of an exactly similar polymer but without the defect structure being sought.^{370,539} The procedure has the disadvantages that noise is added to the spectrum and that it requires preparation of a reference polymer. The method does not alleviate the dynamic range problems discussed above.
- (b) Use of a Hahn spin echo experiment to suppress signals from backbone atoms. It has been demonstrated^{7,540} that end group signals usually persist longer than backbone signals because of longer T₂ relaxation times. Moad *et al.*⁷ have applied the method to detect obscured cyanoisopropyl end groups in PMMA.
- (c) Use of pulse sequences that select for the number of attached hydrogens. For example, for PS prepared with AIBN a 'quaternary only' pulse sequence can be used to better visualize signals due to the quaternary carbons of the AIBNderived residues.⁷
- (d) Analysis of polymers prepared from NMR-inactive monomers. Hatada *et al.* used ¹H NMR to determine end groups in perdeuterated PS⁵⁴¹ and PMMA.^{542,543} Similarly, the use of NMR-inactive ¹²C-enriched monomers has been envisaged as an aid in detecting end groups in ¹³C NMR experiments.⁵⁴⁴
- (e) Use of two (2D) or three dimensional (3D) NMR methods. For example, Bevington and Huckerby⁵⁴⁵ applied ¹³C-¹H correlation spectroscopy to advantage to evaluate end groups when ¹³C signals are discrete yet ¹H signals are overlapping. Rinaldi and coworkers⁵⁴⁶⁻⁵⁴⁸ examined the end group structures of and define the initiation mechanism for polystyrene prepared with an acyl phosphine oxide initiator using 3D NMR.

These five techniques rely on suppressing signals due to the backbone carbons. The end group signals are not enhanced. Therefore, the sensitivity problems associated with detecting end groups in high molecular weight polymers are not entirely solved. However, the methods (b-d) do allow acquisition at higher spectrometer gain settings and assist in overcoming spectrometer dynamic range

problems. A drawback of the pulse sequence methods is that quantification may not be a straightforward exercise.

Selective labeling of the initiator with ¹³C allows substantial enhancement of the signals of the initiator residues relative to signals due to the backbone in ¹³C NMR spectra. Initiators labeled with or containing NMR active nuclei such as ¹⁹F or ³¹P can also be applied. These methods are described in Section 3.5.4.2.

3.5.3.3 Electron paramagnetic resonance spectroscopy

The application of EPR in the detection and quantification of species formed by spin-trapping the products of radical-monomer reactions is described in Section 3.5.2.1. The application of time-resolved EPR spectroscopy to study intermolecular radical-alkene reactions in solution is mentioned in Section 3.5.1.

3.5.3.4 Mass spectrometry

Some discussion on the use of mass spectrometry for end group determination can be found in recent texts.^{530,549} Traditionally mass spectrometric techniques have required polymers of relatively low molecular weight. Meisters *et al.*⁵⁵⁰ reported that fast atom bombardment mass spectrometry (FAB-MS) can be applied in the analysis of MMA oligomers to at least hexadecamer. For polymers that degrade by unzipping, pyrolysis GCMS has provided extremely useful data on initiation processes. Thus, Farina *et al.*^{459,551} and Ohtani *et al.*^{552,553} described the application of pyrolysis GCMS to determine end groups in PMMA, PS and copolymers.

Two relatively new techniques, matrix assisted laser desorption ionization-time of flight mass spectrometry (MALDI-TOF) and electrospray ionization (ESI), offer new possibilities for analysis of polymers with molecular weights in the tens of thousands. PS molecular weights as high as 1.5 million have been determined by MALDI-TOF. Recent reviews on the application of these techniques to synthetic polymers include those by Hanton⁵⁵⁴ and Nielen.⁵⁵⁵ The methods have been much used to provide evidence for initiation and termination mechanisms in various forms of living and controlled radical polymerization.⁵⁵⁶ Some examples of the application of MALDI-TOF and ESI in end group determination are provided in Table 3.12. The table is not intended to be a comprehensive survey.

MALDI-TOF can be applied to estimate molecular weights of very high molecular weight polymers. However, with the mass resolution of current instruments, molecular weights of less than 5000 are desirable for end groups to be reliably distinguished and determined. There are also issues with sensitivity dependence on molecular weight and composition. Sensitivity depends on volatility and the ease of cationization.⁵⁵⁴ For homopolymer samples MALDI-TOF is able to duplicate GPC distributions with reasonable precision when polydispersities are less than about 1.2. For broader molecular weight

144 The Chemistry of Radical Polymerization

distributions MALDI-TOF tends to underestimate the molecular weight and the polydispersity. Discrimination according to ease of cationization for low molecular weight polymers may be mitigated by end group derivatization. This, however, requires foreknowledge of the end groups.

| Polymerization Method | Technique | Polymer |
|-------------------------------------|-----------|---|
| Conventional - AIBN | MALDI-TOF | PMMA, ⁵⁵⁷ PS ⁵⁵⁷ |
| Conventional - with catalytic chain | MALDI-TOF | PMMA, copolymers ^{556,558,559} |
| transfer | | |
| Conventional, AIBN - with transfer | MALDI-TOF | PNVP ⁵⁶⁰ |
| to solvent | | |
| Conventional - photoinitiation | ESI | PMA ⁵⁶¹ |
| RAFT | MALDI-TOF | PNIPAM, ^{562,563} PS ⁵⁶⁴ other ⁵⁶⁵ |
| RAFT | ESI | PMA, ⁵⁶⁶ |
| ATRP | MALDI-TOF | РЕЛ, ⁵⁶⁷ РММЛ, ⁵⁶⁸ |
| NMP | MALDI-TOF | PBA, ⁵⁶⁹ PS ^{570,571} |
| NMP | ESI | PAN, ⁵⁰⁶ PS ⁵⁰⁷ |

 Table 3.12 Application of MALDI-TOF or ESI Mass Spectrometry to Polymers

 Prepared by Radical Polymerization

3.5.3.5 Chemical methods

Chemical analysis often allows end groups to be determined with high precision though the process is painstaking. A number of techniques have been developed for the chemical derivatization of polymer end groups so they can be more readily measured by spectrophotometric methods. One of the most used is the dyc-partition method introduced by Palit.^{451,572-574} Variants of this method have been applied to detect hydroxy,^{574,575} quaternary ammonium and sulfate end groups.^{450,451} A two step dealkylation-derivatization procedure⁵⁷⁶ was successfully used for determining *t*-butoxy end groups in PS. In that case the *t*-butoxy ends were first cleaved with trifluoroacetic acid to give hydroxy chain ends. This method was not applicable to PMMA. It was found the *t*-butoxy ends of PMMA could be determined by measuring the release of *t*-butyl chloride formed on treating the polymer with boron trichloride.⁵⁷⁶

Where the polymer end groups possess reactive functionality, for example hydroxy, amino, thiol or carboxy groups, post-polymerization derivatization may be used to facilitate detection and identification with NMR spectroscopy. Thus, trichloroacetyl isocyanate undergoes a facile reaction with protic end groups when added in slight excess to a solution of the polymer in an NMR tube (Scheme 3.97).^{577,578} The imidic hydrogens of the derivatives have a distinctive chemical shift in the region 8-11 ppm depending on the particular functional group. There is also a shift of the hydrogens α -to the chain end.



Scheme 3.97

3.5.4 Labeling Techniques

Various methods have been described whereby polymers are formed with an initiator that contains chromophores or other functionality to permit ready detection of initiator-derived end groups by chemical or spectroscopic methods.^{579,580} A potential disadvantage of this procedure is that the initiator is chemically modified and the specificity shown by the initiator-derived radicals may be different from that of the corresponding unlabeled species.

The best labeling system in this regard is isotopic labeling since it involves the minimum change from the standard initiator. Methods based on radiolabeling and stable isotopes detectable by NMR are described in Sections 3.5.4.1 and 3.5.4.2 respectively.

3.5.4.1 Radiolabeling

Polymer formed using radiolabeled initiators may be isolated and analyzed to determine the concentration of initiator-derived residues and calculate the initiator efficiency. Radiolabeled initiators have also been used extensively to establish the relative reactivity of monomers towards radicals.^{88,107,580-582}

Radiolabeling offers greater sensitivity than most other labeling methods. However, the technique has the disadvantage that end groups formed by initiation cannot be directly distinguished from initiator residues produced by other processes (*e.g.* primary radical termination or copolymerization of initiator byproducts) or from residual initiator. In general, the method gives the total initiator residues in the polymer. Analysis of the kinetics of polymerization can help to resolve these problems. A further disadvantage is that polymer isolation and purification is required.

For the case of initiators that produce both primary and secondary radicals (e.g. BPO) use of a doubly labeled initiator allows the different types of end groups to be distinguished [e.g. 117 and 118 - Scheme 3.98] and the reactivities of

monomers towards the primary radicals to be readily established by using the fragmentation step as a clock reaction.^{399,583}



3.5.4.2 Stable isotopes and nuclear magnetic resonance

NMR methods can be applied to give quantitative determination of initiatorderived and other end groups and provide a wealth of information on the polymerization process. They provide a chemical probe of the detailed initiation mechanism and a greater understanding of polymer properties. The main advantage of NMR methods over alternative techniques for initiator residue detection is that NMR signals (in particular ¹³C NMR) are extremely sensitive to the structural environment of the initiator residue. This means that functionality formed by tail addition, head addition, transfer to initiator or primary radical termination, and various initiator-derived byproducts can be distinguished.

Selective labeling of the initiator allows substantial enhancement of the signals of the initiator residues relative to the signals due to the backbone. Various stable isotopes have been employed in this context (including D, F, ¹⁵N and ³¹P), however, most work has involved the use of ¹³C-labeling (Table 3.13). The method has been reviewed. ^{536,584,585} The power of the technique is illustrated by the fact that one experiment allows the determination of:

- (a) The total fate of the initiator as a function of conversion (initiator efficiency, nature and amount of byproducts).
- (b) The chain ends (reactivity of primary radicals towards monomers, head vs tail addition, etc.).
- (c) The rate of polymerization.
- (d) The number average molecular weight ([end groups]/[monomer used]).

The use of ¹³C-labeled initiators in assessing the kinetics and efficiency of initiation^{2,14,32,60,84} requires that the polymer end groups, residual initiator, and various initiator-derived byproducts should each give rise to discrete signals in the NMR spectrum. So far this method has been demonstrated for homo- and copolymerizations of S and MMA prepared with AIBN- α -¹³C, AIBMe- α -¹³C or BPO-*carbonyl*-¹³C/BPO-*ring*-¹³C (1:1) as initiator.

 Table 3.13 Radical Polymerizations Performed with Initiators Labeled with

 Stable Isotopes

| Initiator | Polymer |
|---|--|
| $ \underbrace{ \left(\begin{array}{c} O \\ -13 \\ C \end{array} \right)^{-13} - O - O - O - 13 \\ C - O - O - O - 13 \\ C - O - O - O - O - O - O - O - O - O -$ | S, ⁹ MMA, ¹¹ MMA- <i>co</i> -S, ¹¹ other ^{400,586-594} |
| BPO- carbonyl- ¹³ C | |
| | S ¹⁴ |
| BPO -α- ¹¹ C | |
| F | S, ^{595,596} MMA, ^{595,596} MMA- <i>co</i> -S, ⁵⁹⁷ other ⁵⁹⁸ |
| | |
| СН ₃ - ¹³ Ċ-N=N- ¹³ Ċ-CH ₃ Н Н АРЕ-а- ¹³ С | MA, ⁵³⁹ MMA, ⁵³⁹ MPK, ⁵³⁹ VAe, ⁵³⁹ AN-co-S, ⁵⁹⁹ B-co- MMA, ⁶⁰⁰ MAII-co-S, ⁶⁰¹ MMA-co-S, ⁶⁰² other ^{371,603} |
| APE- α - ¹³ C / EASC ^a | B-co-MMA, ⁶⁰⁰ MMA-co-S ⁶⁰² |
| $\begin{array}{ccc} CN & CN \\ CH_3^{-13} \overset{CN}{C} & N=N & {}^{13} \overset{C}{C} - CH_3 \\ H & H \\ APN-\alpha^{-13} C \end{array}$ | AN-co-S ⁶⁰⁴ |
| $\begin{array}{ccc} CN & CN \\ CH_{3}^{-13}C - N^{-}N^{-13}C - CH_{3} \\ CH_{3} & CH_{3} \\ AlBN-\alpha^{-13}C \end{array}$ | MMA, ³⁵⁷ S, ^{7.32,357} VAc, ³⁵⁷ MMA- <i>co</i> -S, ^{2,357} MMA- <i>co</i> -VAc ³⁵⁷ |
| $\begin{array}{rccc} {}^{13}\text{CN} & {}^{13}\text{CN} \\ \text{CH}_3\text{-}\overset{\text{C}}{\text{C}}\text{-}\text{N}\text{=}\text{N}\text{-}\overset{\text{C}}{\text{C}}\text{-}\text{CH}_3 \\ & \overset{\text{C}}{\text{CH}_3} & \overset{\text{C}}{\text{CH}_3} \\ \end{array}$ | VAc, ¹¹³ VF ¹¹³ |
| С ¹⁵ N С ¹⁵ N СН ₃ -С-N=N-С-СН ₃ СН ₃ СН ₃ АIBN- ¹⁵ N | MMA- co -S, ³⁸² other ⁶⁰⁵ |
| CN CN ¹³ CH ₃ -C-N=N-C- ¹³ CH ₃ ¹³ CH ₃ ¹³ CH ₃ AIBN-86 - ¹³ C | AMS, ³⁵⁸ EA, ³⁶⁰ MMA, ^{359,606} MAN, ³⁵⁹ S, ^{359,606} VC, ³⁶¹ VP, ⁵⁸⁹ AN- <i>co</i> -MMA, ³⁶³ B- <i>co</i> -MMA, ⁶⁰⁰ EA- <i>co</i> -S ³⁶⁰ MAN- <i>co</i> -S, ³⁵⁹ MMA-MPK, ⁶⁰⁷ MMA-MVK, ⁶⁰⁷ MMA- <i>co</i> -S, ³⁵⁹ MMA- <i>m</i> VC, ³⁶¹ attac ^{589,600,603,608,609} |
| AIBN-88- ¹³ C/FASC ^a | $B_{-co-MMA} = \frac{600}{MMA_{-co-}} S^{81}$ |
| server bla commender | 13 DO 1911913 191197/ 1-DO-D |

Table 3.13 (continued)

| Initiator | Polymer |
|--|---|
| $\begin{array}{ccc} CN & CN \\ CD_3\text{-}C\text{-}N\text{-}N\text{-}C\text{-}CD_3 \\ CD_3 & CD_3 \\ \end{array}$ | EA, ³⁶⁰ EA- <i>co</i> -S ³⁶⁰ ,MMA- <i>co</i> -S ³⁸² |
| $ \begin{array}{cccc} ^{13}CN & O \\ CH_3 - \dot{C} & N=N & \overset{\vee}{\subset} & NH_2 \\ & CH_3 \\ AZOF-nitrile-^{13}C \end{array} $ | VAc ¹¹³ |
| CN Ο ¹³ CH ₃ -C-N ⁻ N-C-NH ₂ ¹³ CH ₃ AZOF-ββ- ¹³ C | MMA- <i>co</i> -S, ³⁶⁴ MAN- <i>co</i> -S, ³⁶² S- <i>co</i> -VAc ³⁶² |
| $\begin{array}{c} {}^{CO_2CH_3}_{CH_3} CO_2CH_3\\ {}^{CH_3^{-13}} \stackrel{C}{C} \! - \! N \! = \! N \! - \! ^{13} \stackrel{C}{C} \! - \! CH_3\\ {}^{CH_3} \stackrel{C}{C} \! H_3\\ \hline {}^{AIBMe \! - \! \alpha \! - \! ^{13}C} \end{array}$ | MMA, ³⁵⁷ S, ³⁵⁷ VAc, ³⁵⁷ MMA- <i>co</i> -S, ^{83,84,357} MMA- <i>co</i> -VAc, ³⁵⁷ |
| AIBMe-α- ¹³ C/EASC ^a | MMA-co-S ⁸³ |
| $\begin{array}{c} CO_2CH_3 CO_2CH_3 \\ {}^{13}CH_3\text{-} \overset{\circ}{C} N = N \overset{\circ}{C} {}^{-13}CH_3 \\ {}^{13}\overset{\circ}{CH}_3 {}^{13}\overset{\circ}{CH}_3 \end{array}$ | AN, 375 MA 375 , MMA, 375 S, 375 B-co-MMA, 600 MMAco-S ⁸¹ , other 600 |
| AIBMe-ββ- ¹³ C | |
| AIBMe-ββ- ¹³ C/EASC ^a | $B-co-MMA,^{600} MMA-co-S^{81}$ |

^a EASC = ethyl aluminum sesquichloride

Labeled initiators have been used in evaluating the relative reactivity of a wide range of monomers towards initiating radicals.³⁵⁹ The method involves determination of the relative concentrations of the end groups formed by addition to two monomers (*e.g.* **119** and **120**) in a binary copolymer formed with use of a labeled initiator. For example, when AIBMe- α -¹³C is used to initiate copolymerization of MMA and VAc (Scheme 3.99),³⁵⁷ the simple relationship (eq. 14) gives the relative rate constants for addition to the two monomers. Copolymerizations studied in this way are summarized in Table 3.13.

$$\frac{k_{\rm MMA}}{k_{\rm VAc}} = \frac{[\rm VAc] . [119]}{[\rm MMA] . [120]}$$
(14)



3.6 References

- 1. Solomon, D.H.; Moad, G. Makromol. Chem., Macromol. Symp. 1987, 10/11, 109.
- Moad, G.; Solomon, D.H. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 97.
- 3. Solomon, D.H.; Cacioli, P.; Moad, G. Pure Appl. Chem. 1985, 57, 985.
- 4. Hwang, E.F.J.; Pearce, E.M. Polym. Eng. Rev. 1983, 2, 319.
- 5. Mita, I. In *Aspects of Degradation and Stabilization of Polymers*; Jellincek, H.H.G., Ed.; Elsevier: Amsterdam, 1978; p 247.
- 6. Solomon, D.H. J. Macromol. Sci., Chem. 1982, A17, 337.
- 7. Moad, G.; Solomon, D.H.; Johns, S.R.; Willing, R.I. *Macromolecules* 1984, 17, 1094.
- 8. Moad, G.; Rizzardo, E.; Solomon, D.H. Macromolecules 1982, 15, 909.
- 9. Moad, G.; Solomon, D.H.; Johns, S.R.; Willing, R.I. *Macromolecules* **1982**, *15*, 1188.
- 10. Moad, G.; Rizzardo, E.; Solomon, D.H. Aust. J. Chem. 1983, 36, 1573.
- 11. Moad, G.; Rizzardo, E.; Solomon, D.H. Polym. Bull. 1984, 12, 471.
- Griffiths, P.G.; Rizzardo, E.; Solomon, D.H. J. Macromol. Sci., Chem. 1982, A17, 45.
- 13. Sato, T.; Otsu, T. Makromol. Chem. 1977, 178, 1941.
- 14. Krstina, J.; Moad, G.; Solomon, D.H. Eur. Polym. J. 1989, 25, 767.
- 15. Singh, M.; Nandi, U.S. J. Polym. Sci., Polym. Lett. Ed. 1979, 17, 121.
- Schildknecht, C.E. In *Polymerization Processes*; Schildknecht, C.E.; Skeist, I., Eds.; Wiley: New York, 1977; p 88.
- 17. Boundy, R.H.; Boyer, R.F., Eds. Styrene, Its Polymers, Copolymers and Derivatives; Reinhold: New York, 1952.
- 18. Citterio, A.; Arnoldi, A.; Minisci, F. J. Org. Chem. 1979, 44, 2674.
- 19. Griller, D.; Ingold, K.U. Acc. Chem. Res. 1980, 13, 317.
- 20. Newcomb, M. Tetrahedron 1993, 49, 1151.
- Grant, R.D.; Griffiths, P.G.; Moad, G.; Rizzardo, E.; Solomon, D.H. Aust. J. Chem. 1983, 36, 2447.
- 22. Bednarek, D.; Moad, G.; Rizzardo, E.; Solomon, D.H. *Macromolecules* **1988**, *21*, 1522.
- 23. Walling, C.; Thaler, W. J. Am. Chem. Soc. 1961, 83, 3877.
- 24. Elson, I.H.; Mao, S.W.; Kochi, J.K. J. Am. Chem. Soc. 1975, 97, 335.
- 25. Watanabe, Y.; Ishigaki, H.; Okada, H.; Suyama, S. Polym. J. 1997, 29, 693.

- 26. Baignee, A.; Howard, J.A.; Scaiano, J.C.; Stewart, L.C. J. Am. Chem. Soc. 1983, 105, 6120.
- 27. Moad, G. Prog. Polym. Sci. 1999, 24, 81.
- 28. Maillard, B.; Ingold, K.U.; Scaiano, J.C. J. Am. Chem. Soc. 1983, 105, 5095.
- 29. Hartzler, H.D. In *The Chemistry of the Cyano Group*; Rappoport, Z., Ed.; Wiley: London, 1970; p 671.
- 30. Bevington, J.C.; Troth, H.G. Trans. Faraday Soc. 1962, 58, 186.
- 31. Niki, E.; Kamiya, Y.; Ohta, N. Bull. Chem. Soc. Japan 1969, 42, 3220.
- 32. Moad, G.; Rizzardo, E.; Solomon, D.H.; Johns, S.R.; Willing, R.I. Makromol. Chem., Rapid Commun. 1984, 5, 793.
- 33. Achilias, D.S.; Kiparissides, C. Macromolecules 1992, 25, 3739.
- 34. Starnes, W.H., Jr.; Plitz, I.M.; Schilling, F.C.; Villacorta, G.M.; Park, G.S.; Saremi, A.H. *Macromolecules* **1984**, *17*, 2507.
- 35. Ishiwata, II.; Inoue, T.; Yoshihira, K. J. Chromatogr. 1986, 370, 275.
- Fordham, P.J.; Gramshaw, J.W.; Castle, L. Food Additives Contaminants 2001, 18, 461.
- 37. Simionescu, C.I.; Chiriac, A.P.; Chiriac, M.V. Polymer 1993, 34, 3917.
- 38. Turro, N.J.; Kraeutler, B. Acc. Chem. Res. 1980, 13, 369.
- Fischer, H.; Baer, R.; Hany, R.; Verhoolen, I.; Walbiner, M. J. Chem. Soc., Perkin Trans. 2 1990, 787.
- Buback, M.; Kowollik, C.; Kurz, C.; Wahl, A. Macromol. Chem. Phys. 2000, 201, 464.
- 41. Moad, G.; Rizzardo, E.; Solomon, D.H. J. Macromol. Sci., Chem. 1982, A17, 51.
- 42. Gilbert, R.G. *Emulsion Polymerization: A Mechanistic Approach*; Academic Press: London, 1995.
- 43. Lovell, P.A.; El-Aasser, M.S., Eds. *Emulsion Polymerization and Emulsion Polymers*; John Wiley & Sons: London, 1997.
- 44. Morrison, B.R.; Maxwell, I.A.; Gilbert, R.G.; Napper, D.H. ACS Symp. Ser. 1992, 492, 28.
- 45. Sheppard, C.S.; Kamath, V.R. Polym. Eng. Sci. 1979, 19, 597.
- 46. Barton, J.; Borsig, E. *Complexes in Free Radical Polymerization*; Elsevier: Amsterdam, 1988.
- Masson, J.C. In *Polymer Handbook*, 3rd ed.; Brandup, J.; Immergut, E.H., Eds.; Wiley: New York, 1989; p II/1.
- 48. Dixon, K.W. In *Polymer Handbook*, 4th ed.; Brandup, J.; Immergut, E.H.; Grulke, E.A., Eds.; John Wiley and Sons: New York, 1999; p II/1.
- 49. Engel, P.S. Chem. Rev. 1980, 80, 99.
- Sheppard, C.S. In *Encyclopedia of Polymer Science and Engineering*, 2nd ed.; Mark, H.F.; Bikales, N.M.; Overberger, C.G.; Menges, G., Eds.; Wiley: New York, 1985; Vol. 2, p 143.
- 51. Koga, G.; Koga, N.; Anselme, J.-P. In *The Chemistry of the Hydrazo, Azo and Azoxy Groups*; Patai, S., Ed.; Wiley: London, 1975; Vol. 16, part 2, p 861.
- 52. Koenig, T. In *Free Radicals*; Kochi, J.K., Ed.; Wiley-Interscience: New York, 1973; Vol. 1, p 113.
- 53. Smith, P.A.S. In *The Chemistry of Open Chain Organic Nitrogen Compounds*; Benjamin: New York, 1966; Vol. 2, p 269.

150

- 54. Neuman, R.C., Jr.; Grow, R.H.; Binegar, G.A.; Gunderson, H.J. J. Org. Chem. 1990, 55, 2682.
- 55. Ayscough, P.B.; Brooks, B.R.; Evans, H.E. J. Phys. Chem. 1964, 68, 3889.
- 56. Nelsen, S.F.; Bartlett, P.D. J. Am. Chem. Soc. 1966, 88, 137.
- 57. Overberger, C.G.; Berenbaum, M.B. J. Am. Chem. Soc. 1951, 73, 2618.
- 58. Van-Hook, J.P.; Tobolsky, S. J. Am. Chem. Soc. 1958, 80, 779.
- 59. Barbe, W.; Rüchardt, C. Makromol. Chem. 1983, 184, 1235.
- Krstina, J.; Moad, G.; Willing, R.I.; Danek, S.K.; Kelly, D.P.; Jones, S.L.; Solomon, D.II. Eur. Polym. J. 1993, 29, 379.
- 61. Talat-Erben, M.; Bywater, S. J. Am. Chem. Soc. 1955, 77, 3712.
- 62. Otsu, T.; Yamada, B. J. Macromol. Sci. Chem 1969, 3, 187.
- 63. Krstina, J.; Moad, G.; Solomon, D.H., unpublished results.
- 64. Duisman, W.; Rüchardt, C. Chem. Ber. 1978, 111, 596.
- 65. Overberger, C.G.; Berenbaum, M.B. J. Am. Chem. Soc. 1953, 75, 2078.
- Bandlish, B.K.; Garner, A.W.; Hodges, M.L.; Timberlake, J.W. J. Am. Chem. Soc. 1975, 97, 5856.
- 67. Martin, J.C.; Timberlake, J.W. J. Am. Chem. Soc. 1970, 92, 978.
- 68. Alder, M.G.; Leffler, J.E. J. Am. Chem. Soc. 1954, 76, 1425.
- 69. Cohen, S.G.; Cohen, F.; Wang, C.H. J. Org. Chem. 1963, 28, 1479.
- 70. Solomon, S.; Wang, C.H.; Cohen, S.G. J. Am. Chem. Soc. 1957, 79, 4104.
- 71. Kiefer, H.; Traylor, T.G. Tetrahedron Lett. 1966, 7, 6163.
- 72. Rizzardo, E.; Serelis, A.K.; Solomon, D.H. Aust. J. Chem. 1982, 35, 2013.
- Timberlake, J.W. In Substituent Effects in Radical Chemistry (NATO ASI Ser., Ser. C); Viche, H.G.; Janousek, Z.; Merenyi, R., Eds.; Reidel: Dordecht, 1986; Vol. 189, p 271.
- Overberger, C.G.; Hale, W.F.; Berenbaum, M.B.; Finestone, A.B. J. Am. Chem. Soc. 1954, 76, 6185.
- 75. Duisman, W.; Rüchardt, C. Tetrahedron Lett. 1974, 15, 4517.
- 76. Rüchardt, C. Top. Curr. Chem. 1980, 88, 1.
- 77. Firestone, R.A. J. Org. Chem. 1980, 45, 3604.
- 78. Wolf, R.A. ACS Symp. Ser. 1989, 404, 416.
- 79. Henrici-Olivé, G.; Olivé, S. Makromol. Chem. 1962, 58, 188.
- 80. Tanaka, H.; Fukuoka, K.; Ota, T. Makromol. Chem., Rapid, Commun. 1985, 6, 563.
- 81. Lyons, R.A.; Moad, G.; Senogles, E. Eur. Polym. J. 1993, 29, 389.
- Lyons, R.A.; Moad, G.; Scnogles, E. In *Pacific Polymer Conference Preprints*; Polymer Division, Royal Australian Chemical Insitute: Brisbane, 1993; Vol. 3, p 249.
- 83. Krstina, J.; Moad, G.; Solomon, D.H. Polym. Bull. 1992, 27, 425.
- Spurling, T.H.; Deady, M.; Krstina, J.; Moad, G. Makromol. Chem., Macromol. Symp. 1991, 51, 127.
- 85. Stickler, M. Makromol. Chem. 1986, 187, 1765.
- 86. O'Driscoll, K.F.; Huang, J. Eur. Polym. J. 1989, 7/8, 629.
- Drewer, R.J. In *The Chemistry of the Hydrazo, Azo and Azoxy Groups*; Patai, S., Ed.; Wiley: London, 1975; Vol. 16, part 2, p 935.
- 88. Ayrcy, G. Chem. Rev. 1963, 63, 645.
- 89. Fink, J.K. J. Polym. Sci., Polym. Chem. Ed. 1983, 21, 1445.

- 90. Cox, R.A.; Buncel, E. In *The Chemistry of the Hydrazo, Azo and Azoxy Groups*; Patai, S., Ed.; Wiley: London, 1975; Vol. 16, part 2, p 775.
- 91. Braun, D.; Czerwinski, W.K. Makromol. Chem. 1987, 188, 2371.
- 92. Fukuda, T.; Ma, Y.-D.; Inagaki, H. Macromolecules 1985, 18, 17.
- 93. Russell, G.T.; Napper, D.H.; Gilbert, R.G. Macromolecules 1988, 21, 2141.
- 94. Buback, M.; Huckestein, B.; Kuchta, F.-D.; Russell, G.T.; Schmid, E. Macromol. Chem. Phys. 1994, 195, 2117.
- 95. Sack, R.; Schulz, G.V.; Meyerhoff, G. Macromolecules 1988, 21, 3345.
- 96. Faldi, A.; Tirrell, M.; Lodge, T.P.; von Meerwall, E. *Macromolecules* **1994**, *27*, 4184.
- Bizilj, S.; Kelly, D.P.; Serelis, A.K.; Solomon, D.H.; White, K.E. Aust. J. Chem. 1985, 38, 1657.
- 98. Trecker, D.J.; Foote, R.S. J. Org. Chem. 1968, 33, 3527.
- 99. Kodaira, K.; Ito, K.; Iyoda, S. Polym. Commun. 1987, 28, 86.
- 100. Talat-Erben, M.; Bywater, S. J. Am. Chem. Soc. 1954, 77, 3710.
- 101. Jaffe, A.B.; Skinner, K.J.; McBride, J.M. J. Am. Chem. Soc. 1972, 94, 8510.
- 102. Hammond, G.S.; Trapp, O.D.; Keys, R.T.; Neff, D.L. J. Am. Chem. Soc. 1959, 81, 4878.
- Chung, R.P.-T.; Danck, S.K.; Quach, C.; Solomon, D.H. J. Macromol. Sci., Chem. 1994, A31, 329.
- 104. Pryor, W.A.; Fiske, T.R. Macromolecules 1969, 2, 62.
- 105. Cascaval, C.N.; Straus, S.; Brown, D.W.; Florin, R.E. J. Polym. Sci., Polym. Symp. 1976, 57, 81.
- 106. Baysal, B.; Tobolsky, A.V. J. Polym. Sci. 1952, 8, 529.
- 107. Bevington, J.C.; Lewis, T.D. Polymer 1960, 1, 1.
- 108. Pryor, W.A.; Coco, J.H. Macromolecules 1970, 3, 500.
- 109. May, J.A., Jr.; Smith, W.B. J. Phys. Chem. 1968, 72, 2993.
- 110. Ayrey, G.; Haynes, A.C. Makromol. Chem. 1974, 175, 1463.
- 111. Athey, R.D., Jr. J. Polym. Sci., Polym. Chem. Ed. 1977, 15, 1517.
- 112. Braks, J.G.; Huang, R.Y.M. J. Appl. Polym. Sci. 1978, 22, 3111.
- 113. Bevington, J.C.; Breuer, S.W.; Heseltine, E.N.J.; Huckerby, T.N.; Varma, S.C. J. Polym. Sci., Part A: Polym. Chem. 1987, 25, 1085.
- 114. Quinga, E.M.Y.; Mendenhall, G.D. J. Org. Chem. 1985, 50, 2836.
- 115. Dulog, L.; Klein, P. Chem. Ber. 1971, 104, 902.
- 116. Dulog, L.; Klein, P. Chem. Ber. 1971, 104, 895.
- 117. Protasiewicz, J.; Mendenhall, G.D. J. Org. Chem. 1985, 50, 3220.
- 118. Mendenhall, G.D.; Stewart, L.C.; Scaiano, J.C. J. Am. Chem. Soc. 1982, 104, 5109.
- 119. Mendenhall, G.D.; Quinga, E.M.Y. Int. J. Chem. Kinet. 1985, 17, 1187.
- 120. Druliner, J.D.; Krusic, P.D.; Lehr, G.F.; Tolman, C.A. J. Org. Chem. 1985, 50, 5838.
- 121. Kiefer, H.; Traylor, T.G. J. Am. Chem. Soc. 1967, 89, 6667.
- 122. Mendenhall, G.D.; Cary, L.W. J. Org. Chem. 1975, 40, 1646.
- 123. Busfield, W.K.; Jenkins, I.D.; Rizzardo, E.; Solomon, D.H.; Thang, S.H. J. Chem. Soc., Perkin Trans. 1 1991, 1351.
- 124. Ando, W., Ed. Organic Peroxides; Wiley: Chichester, 1992.

152

- 125. Sheppard, C.S. In *Encyclopedia of Polymer Science and Engineering*, 2nd ed.; Mark, H.F.; Bikales, N.M.; Overberger, C.G.; Menges, G., Eds.; Wiley: New York, 1987; Vol. 11, p 1.
- Patai, S., Ed. The Chemistry of Functional Groups, The Chemistry of Peroxides; Wiley: Chichester, UK, 1983.
- 127. Swern, D., Ed. Organic Peroxides; Wiley-Interscience: New York, 1970; Vol. 1.
- 128. Swem, D., Ed. Organic Peroxides; Wiley-Interscience: New York, 1971; Vol. 2.
- 129. Swern, D., Ed. Organic Peroxides; Wiley-Interscience: New York, 1971; Vol. 3.
- 130. Davies, A.G. Organic Peroxides; Butterworths: London, 1961.
- Hawkins, E.G.R. Organic Peroxides Their Formation and Reactions; Van Nostrand: Princeton, 1961.
- 132. Janzen, E.G.; Evans, C.A.; Nishi, Y. J. Am. Chem. Soc. 1972, 94, 8236.
- 133. Bawn, C.E.H.; Halford, R.G. Trans. Faraday Soc. 1955, 51, 780.
- 134. Yamada, M.; Kitagawa, K.; Komai, T. Plast. Ind. News 1971, 17, 131.
- 135. Blomquist, A.T.; Ferris, A. J. Am. Chem. Soc. 1951, 73, 3412.
- 136. Bartlett, P.D.; Benzing, E.P.; Pincock, R.E. J. Am. Chem. Soc. 1960, 82, 1762.
- 137. Hiatt, R.; Mill, T.; Irwin, K.C.; Castleman, J.K. J. Org. Chem. 1968, 33, 1421.
- 138. Huyser, E.S.; VanScoy, R. J. Org. Chem. 1968, 33, 3524.
- 139. Hiatt, R.; Strachan, W.M.J. J. Org. Chem. 1963, 28, 1893.
- 140. Kolthoff, I.M.; Miller, I.K. J. Am. Chem. Soc. 1951, 73, 3055.
- 141. Fujimori, K. In Organic Peroxides; Ando, W., Ed.; Wiley: Chichester, 1992; p 319.
- 142. Bouillion, G.; Lick, C.; Schank, K. In *The Chemistry of the Peroxides*; Patai, S., Ed.; Wilcy: London, 1983; p 279.
- Hiatt, R. In Organic Peroxides; Swern, D., Ed.; Wiley-Interscience: New York, 1971; Vol. 2, p 799.
- 144. Martin, J.C.; Hargis, J.H. J. Am. Chem. Soc. 1969, 91, 5399.
- 145. Pryor, W.A.; Morkved, E.H.; Bickley, H.T. J. Org. Chem. 1972, 37, 1999.
- 146. Nozaki, K.; Bartlett, P.D. J. Am. Chem. Soc. 1946, 68, 1686.
- 147. Sheldon, R.A.; Kochi, J.K. J. Am. Chem. Soc. 1970, 92, 4395.
- 148. Saltiel, J.; Curtis, H.C. J. Am. Chem. Soc. 1971, 93, 2056.
- 149. Yamauchi, S.; Hirota, N.; Takahara, S.; Sakuragi, H.; Tokumaru, K. J. Am. Chem. Soc. 1985, 107, 5021.
- 150. Grossi, L.; Lusztyk, J.; Ingold, K.U. J. Org. Chem. 1985, 50, 5882.
- 151. Nedelec, J.Y.; Lefort, D. Tetrahedron 1980, 36, 3199.
- 152. Rosenthal, I.; Mossoba, M.M.; Riesz, P. J. Magn. Reson. 1982, 47, 200.
- 153. Garcia-Rubio, L.H.; Mehta, J. ACS Symp. Ser. 1986, 202.
- 154. Garcia-Rubio, L.H.; Ro, N.; Patel, R.D. Macromolecules 1984, 17, 1998.
- 155. Navolokina, R.A.; Zilberman, E.N.; Krasavina, N.B.; Kharitonova, O.A. Izv. Vyssh. Uchebn. Zaved. Khim. Khim. Tekhnol. 1986, 29, 83.
- 156. Stickler, M.; Dumont, E. Makromol. Chem. 1986, 187, 2663.
- Walling, C.; Waits, H.P.; Milovanovic, J.; Pappiaonnou, C.G. J. Am. Chem. Soc. 1970, 92, 4927.
- 158. Sivaram, S.; Singhal, R.K.; Bhardwaj, I.S. Polym. Bull. 1980, 3, 27.
- 159. Curci, R.; Edwards, J.O. In *Organic Peroxides*; Swern, D., Ed.; Wiley: New York, 1971; Vol. 1, p 200.
- 160. Mayo, F.R.; Gregg, R.A.; Matheson, M.S. J. Am. Chem. Soc. 1951, 73, 1691.
- 161. Berger, K.C.; Deb, P.C.; Meyerhoff, G. Macromolecules 1977, 10, 1075.
- Anisimov, Y.N.; Ivanchev, S.S.; Yurzhenko, A.I. Polym. Sci. USSR (Engl. Transl.) 1967, 9, 692.
- 163. Moad, G.; Solomon, D.H.; Willing, R.I. Macromolecules 1988, 21, 855.
- Suehiro, T.; Kanoya, A.; Yamauchi, T.; Komori, T.; Igeta, S.-I. *Tetrahedron* 1968, 24, 1551.
- 165. Suehiro, T.; Kanoya, A.; Hara, H.; Nakahama, T.; Komori, T. *Bull. Chem. Soc. Japan* **1967**, *40*, 668.
- 166. Moad, G.; Rizzardo, E.; Solomon, D.H. Tetrahedron Lett. 1981, 22, 1165.
- Sosnovsky, G.; Rawlinson, D.J., Eds. Organic Peroxides. Metal Ion-Catalyzed Reactions of Symmetric Peroxides; Wiley-Interscience: New York, 1970; Vol. 1.
- 168. Kochi, J.K. J. Am. Chem. Soc. 1962, 84, 1572.
- 169. Kurz, M.E.; Kovacic, P. J. Org. Chem. 1968, 33, 1950.
- 170. Chalfont, G.R.; Hey, D.H.; Liang, K.S.Y.; Perkins, M.J. Chem. Commun. (London) 1967, 367.
- 171. Perkins, M.J.; Chalfont, G.R.; Hey, D.H.; Liang, K.S.Y. J. Chem. Soc. (B) 1971, 233.
- 172. Rusakova, A.; Margaritova, M.F. Vysokomol. Soedin. Ser. B 1967, 9, 515.
- 173. Jones, R.G.; Catterall, E.; Bilson, R.T.; Booth, R.G. J. Chem. Soc., Chem. Commun. 1972, 22.
- 174. Bevington, J.C.; Dyball, C.J.; Leech, J. Makromol. Chem. 1977, 178, 2741.
- 175. Bevington, J.C.; Dyball, C.J.; Leech, J. Makromol. Chem. 1979, 180, 657.
- 176. Sato, T.; Abe, M.; Otsu, T. Makromol. Chem. 1977, 178, 1259.
- 177. Bottle, S.; Busfield, W.K.; Jenkins, I.D.; Thang, S.; Rizzardo, E.; Solomon, D.H. *Eur. Polym. J.* **1989**, *25*, 671.
- 178. Dambatta, B.B.; Ebdon, J.R. Eur. Polym. J. 1986, 22, 783.
- 179. Walling, C. Free Radicals in Solution; Wiley: New York, 1957.
- 180. Imoto, M.; Choe, S. J. Polym. Sci. 1955, 15, 485.
- 181. Pryor, W.A.; Hendrickson, W.H., Jr. Tetrahedron Lett. 1983, 24, 1459.
- 182. Sato, T.; Kita, S.; Otsu, T. Makromol. Chem. 1975, 176, 561.
- 183. Strong, W.A. Ind. Eng. Chem. 1964, 56(12), 33.
- 184. McBay, H.C.; Tucker, O. J. Org. Chem. 1954, 19, 869.
- 185. Razuvaev, G.A.; Terman, L.M.; Petukhow, G.G. Dokl. Akad. Nauk. USSR (Engl. Transl.) 1961, 136, 111.
- 186. Cohen, S.G.; Sparrow, D.B. J. Am. Chem. Soc. 1950, 72, 611.
- 187. Van Sickle, D.E. J. Org. Chem. 1969, 34, 3446.
- Cuthbertson, M.J.; Moad, G.; Rizzardo, E.; Solomon, D.H. *Polym. Bull.* 1982, 6, 647.
- Pastorino, R.L.; Lewis, R.N. In *Modern Plastics Encyclopedia*; McGraw-Hill: New York, 1988; p 165.
- 190. Duynstee, E.F.J.; Esser, M.L.; Schellekens, R. Eur. Polym. J. 1980, 16, 1127.
- 191. Crano, J. J. Org. Chem. 1966, 31, 3615.
- 192. Sawaki, Y. In Organic Peroxides; Ando, W., Ed.; Wiley: Chichester, 1992; p 426.
- Bouillion, G.; Lick, C.; Schank, K. In *The Chemistry of the Peroxides*; Patai, S., Ed.; Wiley: London, 1983; p 287.
- Singer, L.A. In Organic Peroxides; Swern, D., Ed.; Wiley-Interscience: New York, 1970; Vol. 1, p 265.

Initiation

- 195. Nakamura, T.; Busfield, W.K.; Jenkins, I.D.; Rizzardo, E.; Thang, S.H.; Suyama, S. *J. Org. Chem.* **2000**, *65*, 16.
- 196. Buback, M. Macromol. Symp. 2002, 182, 103.
- Buback, M.; Klingbeil, S.; Sandmann, J.; Sderra, M.B.; Vogele, II.P.; Wackerbarth, H.; Wittkowski, L. Z. Phys. Chem. 1999, 210, 199.
- 198. Buback, M.; Sandmann, J. Z. Phys. Chem. 2000, 214, 583.
- 199. Niki, E.; Kamiya, Y. J. Am. Chem. Soc. 1974, 96, 2129.
- 200. Hiatt, R.; Traylor, T.G. J. Am. Chem. Soc. 1965, 87, 3766.
- Gupta, S.N.; Gupta, I.; Neckers, D.C. J. Polym. Sci., Polym. Chem. Ed. 1981, 19, 103.
- 202. Allen, N.S.; Hardy, S.J.; Jacobine, A.; Glaser, D.M.; Catalina, F.; Navaratnam, S.; Parsons, B.J. In *Radiation Curing of Polymers II*; Randell, D.R., Ed.; Royal Society of Chemistry: Cambridge, 1991; p 182.
- Matsugo, S.; Saito, I. In Organic Peroxides; Ando, W., Ed.; Wiley: Chichester, 1992; p 157.
- 204. Sheldon, R.A. In *The Chemistry of the Peroxides*; Patai, S., Ed.; Wiley: London, 1983; p 161.
- 205. Hiatt, R. In Organic Peroxides; Swern, D., Ed.; Wiley-Interscience: New York, 1971; Vol. 3, p 1.
- 206. Suyama, S.; Sugihara, Y.; Watanabe, Y.; Nakamura, T. Polym. J. 1992, 24, 971.
- 207. Drumright, R.E.; Kastl, P.E.; Priddy, D.B. Macromolecules 1993, 26, 2246.
- 208. Matsuyama, K.; Kimura, H. J. Org. Chem. 1993, 58, 1766.
- 209. Bischoff, C.; Platz, K.-H. J. Prakt. Chem 1973, 315, 175.
- 210. Suyama, S.; Wanatabe, Y.; Sawaki, Y. Bull. Chem. Soc. Japan 1990, 63, 716.
- 211. Wanatabe, Y.; Ishigaki, H.; Okada, H.; Suyama, S. Bull. Chem. Soc. Japan 1991, 64, 1231.
- Yamamoto, T.; Nakashio, Y.; Onishi; Hirota, M. Nippon Kagaku Kaishi 1985, 2296.
- 213. Huyser, E.S.; Feng, R.H.C. J. Org. Chem. 1969, 34, 1727.
- 214. Huyser, E.S.; Bredeweg, C.J. J. Am. Chem. Soc. 1964, 86, 2401.
- 215. Huyser, E.S.; Bredeweg, C.J.; Vanscoy, R.M. J. Am. Chem. Soc. 1964, 86, 4148.
- 216. Zetterlund, P.B.; Yamauchi, S.; Yamada, B. Macromol. Chem. Phys. 2004, 205, 778.
- 217. Porter, N.A. In Organic Peroxides; Ando, W., Ed.; Wiley: Chichester, 1992; p 101.
- 218. Hiatt, R. In *Organic Peroxides*; Swern, D., Ed.; Wiley-Interscience: New York, 1971; Vol. 2, p 1.
- 219. Hiatt, R.; Irwin, K.C. J. Org. Chem. 1968, 33, 1436.
- 220. Nangia, P.S.; Benson, S.W. J. Phys. Chem. 1979, 83, 1138.
- 221. Benson, S.W. Thermochemical Kinetics; Wiley: New York, 1976.
- 222. Hiatt, R.; Mill, T.; Mayo, F.R. J. Org. Chem. 1968, 33, 1416.
- 223. Hiatt, R.; Mill, T.; Irwin, K.C.; Castleman, J.K. J. Org. Chem. 1968, 33, 1428.
- 224. Hiatt, R.; Irwin, K.C.; Gould, C.W. J. Org. Chem. 1968, 33, 1430.
- Sosnovsky, G.; Rawlinson, D.J. In Organic Peroxides; Swern, D., Ed.; Wiley-Interscience: New York, 1971; Vol. 2, p 153.
- 226. Mulcahy, M.F.R.; Steven, J.R.; Ward, J.C. Aust. J. Chem. 1965, 18, 1177.
- Hamilton, C.J.; Tighe, B.J. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 261.

- 228. Bacon, R.G.R. Chem. Soc., Quart. Rev. 1955, 9, 287.
- Sosnovsky, G.; Rawlinson, D.J. In Organic Peroxides; Swern, D., Ed.; Wiley-Interscience: New York, 1971; Vol. 2, p 269.
- 230. Sarac, A.S. Prog. Polym. Sci. 1999, 24.
- 231. House, D.A. Chem. Rev. 1962, 62, 185.
- 232. Behrman, E.J.; Edwards, J.O. Rev. Inorg. Chem. 1980, 2, 179.
- Rudin, A.; Samanta, M.C.; Van Der Hoff, B.M.E. J. Polym. Sci., Polym. Chem. Ed. 1979, 17, 493.
- 234. Rasmussen, J.K.; Smith, H.K. J. Am. Chem. Soc. 1981, 103, 730.
- 235. Choi, K.Y.; Lee, C.Y. Ind. Eng. Chem. Res. 1987, 26, 2079.
- 236. Rasmussen, J.K.; Smith, H.K. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1982, 23(1), 152.
- Takeishi, M.; Ohkawa, H.; Hayama, S. Makromol. Chem., Rapid Commun. 1981, 2, 457.
- Rasmussen, J.K.; Heilmann, S.M.; Krepski, L.R.; Smith, H.K. ACS Symp. Ser. 1987, 326, 116.
- Rasmussen, J.K.; Heilmann, S.M.; Toren, P.E.; Pocius, A.V.; Kotnour, T.A. J. Am. Chem. Soc. 1983, 105, 6845.
- 240. Kim, Y.H. In Organic Peroxides; Ando, W., Ed.; Wiley: Chichester, 1992; p 387.
- Curci, R.; Delano, G.; Edwards, J.O.; DiFuria, F.; Gallopo, A.R. J. Org. Chem. 1974, 39, 3020.
- 242. Chawla, O.P.; Fessenden, R.W. J. Phys. Chem. 1975, 79, 2693.
- 243. Tang, Y.; Thorn, R.P.; Mauldin, R.L.; Wine, P.H. J. Photochem. Photobiol., A 1988, 44, 243.
- 244. Ebdon, J.R.; Huckerby, T.N.; Hunter, T.C. Polvmer 1994, 35, 250.
- 245. Brosse, J.-C.; Derouet, D.; Epaillard, F.; Soutif, J.-C.; Legeay, G.; Dusek, K. Adv. Polym. Sci. **1986**, *81*, 167.
- 246. Haber, F.; Weiss, J.J. Proc. R. Soc., London 1934, A147, 332.
- 247. Shiga, T.; Boukhors, A.; Douzou, P. J. Phys. Chem. 1967, 71, 4264.
- 248. Shiga, T.; Boukhors, A.; Douzou, P. J. Phys. Chem. 1967, 71, 3559.
- 249. Chiang, Y.S.; Craddock, J.; Mickewich, D.; Turkevich, J. J. Phys. Chem. 1966, 70, 3509.
- 250. Dixon, W.T.; Norman, R.O.C. J. Chem. Soc. 1963, 3119.
- 251. Walling, C. Acc. Chem. Res. 1975, 8, 125.
- Norman, R.O.C. In Chem. Soc. Spec. Publ. Essays on Free Radical Chemistry; Chem. Soc.: London, 1970; Vol. 24, p 117.
- Grant, R.D.; Rizzardo, E.; Solomon, D.H. J. Chem. Soc., Chem. Commun. 1984, 867.
- 254. Tczuka, T.; Narita, N. J. Am. Chem. Soc. 1979, 101, 7413.
- 255. Engel, P.S.; Y., C.; C., W. J. Am. Chem. Soc. 1991, 113, 4355.
- Engel, P.S.; Pan, L.; Whitmire, K.H.; Guzman-Jiminez, I.; Willcott, M.R.; Smith, W.B. J. Org. Chem. 2000, 65, 1016.
- 257. Nazran, A.S.; Warkentin, J. J. Am. Chem. Soc. 1982, 104, 6405.
- 258. Dixon, D.W. Adv. Oxygenated Processes 1988, 1, 179.
- Osei-Twum, E.Y.; McCallion, D.; Nazran, A.S.; Pannicucci, R.; Risbood, P.A.; Warkentin, J. J. Org. Chem. 1984, 49, 336.

Initiation

- Simionescu, C.; Comanita, E.; Pastravanu, M.; Dumitriu, S. Prog. Polym. Sci. 1986, 12, 1.
- 261. Engel, P.S.; Pan, L.; Ying, Y.; Alemany, L.B. J. Am. Chem. Soc. 2001, 123, 3706.
- 262. Pappas, S.P. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 4, p 337.
- 263. Pappas, S.P. In *Encyclopedia of Polymer Science and Engineering*, 2nd ed.; Mark, H.F.; Bikales, N.M.; Overberger, C.G.; Menges, G., Eds.; Wiley: New York, 1987; Vol. 11, p 186.
- 264. Pappas, S.P. J. Radiat. Curing 1987, 14, 6.
- 265. Bassi, G.L. J. Radiat. Curing 1987, 14, 18.
- 266. Mishra, M.K. J. Macromol. Sci., Rev. Macromol. Chem. 1983, C22, 409.
- 267. Oster, G.; Yang, N. Chem. Rev. 1968, 68, 125.
- 268. Gruber, II.F. Prog. Polym. Sci. 1992, 17, 953.
- 269. Wagner, P.J. Top. Curr. Chem. 1976, 66, 1.
- 270. Hageman, H.J. Prog. Org. Coat. 1985, 13, 123.
- 271. McGinniss, V.D. Dev. Polym. Photochem. 1982, 3, 1.
- 272. Berner, G.; Kirchmayr, R.; Rist, G. J. Oil Colour Chem. Assoc. 1978, 61, 105.
- 273. Ledwith, A. J. Oil Colour Chem. Assoc. 1976, 59, 157.
- 274. Pappas, S.P. Prog. Org. Coat. 1973, 2, 333.
- Heine, II.-G.; Rosenkranz, H.-J.; Rudolf, II. Angew. Chem., Int. Ed. Engl. 1972, 11, 974.
- 276. Pappas, S.P.; Chattopadhyay, A.K.; Carlblom, L.H. ACS Symp. Ser. 1976, 25, 12.
- 277. Hageman, H.J.; Overeem, T. Makromol. Chem., Rapid Commun. 1981, 2, 719.
- 278. Hu, S.K.; Wu, X.S.; Neckers, D.C. Macromolecules 2000, 33, 4030.
- 279. Lipscomb, N.T.; Tarshiani, Y. J. Polym. Sci., Part A: Polym. Chem. 1988, 26, 529.
- 280. Hageman, H.J.; Jansen, L.G.J. Makromol. Chem. 1988, 189, 2781.
- 281. Schnabel, W.; Sumiyoshi, T. In *New Trends in the Photochemistry of Polymers*; Allen, N.S.; Rabek, J.F., Eds.; Elsevier Applied Science: London, 1985; p 133.
- Baxter, J.E.; Davidson, R.S.; Hageman, H.J.; Overeem, T. Makromol. Chem. 1988, 189, 2769.
- 283. Majima, T.; Schnabel, W.; Weber, W. Makromol. Chem. 1991, 192, 2307.
- 284. Rutsch, W.; Dietliker, K.; Leppard, D.; Kohler, M.; Misev, L.; Kolczak, U.; Rist, G. Prog. Org. Coat. 1996, 27, 227.
- 285. Colley, C.S.; Grills, D.C.; Besley, N.A.; Jockusch, S.; Matousek, P.; Parker, A.W.; Towrie, M.; Turro, N.J.; Gill, P.M.W.; George, M.W. J. Am. Chem. Soc. 2002, 124, 14952.
- 286. Klos, R.; Gruber, H.; Greber, G. J. Macromol. Sci., Chem. 1991, A28, 925.
- 287. Carlini, C.; Angiolini, L. Adv. Polym. Sci. 1995, 123, 127.
- 288. Castelvetro, V.; Molesti, M.; Rolla, P. Macromol. Chem. Phys. 2002, 203, 1486.
- 289. Sarker, A.M.; Lungu, A.; Neckers, D.C. Macromolecules 1996, 29, 8047.
- 290. Cook, W.D. Polymer 1992, 33, 600.
- 291. Andrzejewska, E.; Linden, L.-A.; Rabek, J.F. Macromol. Chem. Phys. 1998, 199, 441.
- 292. Griller, D.; Howard, J.A.; Marriott, P.R.; Scaiano, J.C. J. Am. Chem. Soc. 1981, 103, 619.
- 293. Alexander, I.J.; Scott, R.J. Br. Polym. J. 1983, 15, 30.

- Corrales, T.; Catalina, F.; Peinado, C.; Allen, N.S. J. Photochem. Photobio. A 2003, 159, 103.
- 295. Nair, C.P.R.; Clouet, G. J. Macromol. Sci., Rev. Macromol. Chem. Phys. 1991, C31, 311.
- 296. Otsu, T.; Matsumoto, A. Adv. Polym. Sci. 1998, 136, 75.
- 297. Misra, N.; Bajpai, U.D.N. Prog. Polym. Sci. 1982, 8, 61.
- Bamford, C.H. In Comprehensive Polymer Science; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 123.
- 299. Bamford, C.H. In *Reactivity, Mechanism, and Structure in Polymer Chemistry*; Jenkins, A.D.; Ledwith, A., Eds.; Wiley-Interscience: London, 1974; p 52.
- 300. Eastmond, G.C.; Grigor, J. Makromol. Chem., Rapid Commun. 1986, 7, 375.
- 301. Bamford, C.H.; Eastmond, G.C.; Woo, J.; Richards, D.H. Polymer 1982, 23, 643.
- 302. Ho, T.-L. Synthesis 1973, 347.
- 303. Mino, G.; Kaizerman, S.; Rasmussen, E. J. Am. Chem. Soc. 1959, 81, 1494.
- Bamford, C.H.; Middleton, I.P.; Sataka, Y.; Al-Lamee, K.G. In Advances in Polymer Synthesis; Culbertson, B.M.; McGrath, J.E., Eds.; Plenum: New York, 1986; p 291.
- 305. Hebeish, A.; Guthrie, J.T. *The Chemistry and Technology of Cellulosic Copolymers*; Springer: Berlin, 1981.
- 306. McDowell, D.J.; Gupta, B.S.; Stannett, V.T. Prog. Polym. Sci. 1984, 10, 1.
- Hill, D.J.T.; McMillan, A.M.; O'Donnell, J.H.; Pomery, P.J. Makromol. Chem., Macromol. Symp. 1990, 33, 201.
- 308. Casinos, I. Polymer 1992, 33, 1304.
- Moad, G.; Rizzardo, E.; Solomon, D.H. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 141.
- 310. Pryor, W.A.; Lasswell, L.D. Adv. Free Radical Chem. 1975, 5, 27.
- 311. Kurbatov, V.A. Russ. Chem. Rev. (Engl. Transl.) 1987, 56, 505.
- 312. Hall, H.K., Jr. Agnew. Chem. Int. Ed. Engl. 1983, 22, 440.
- 313. Kirchner, K.; Riederle, K. Angew. Makromol. Chem. 1983, 111, 1.
- 314. Flory, P.J. J. Am. Chem. Soc. 1937, 59, 241.
- 315. Mayo, F.R. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1961, 2, 55.
- 316. Chong, Y.K.; Rizzardo, E.; Solomon, D.H. J. Am. Chem. Soc. 1983, 105, 7761.
- 317. Komber, H.; Gruner, M.; Malz, H. Macromol. Rapid Commun. 1998, 19, 83.
- 318. Moad, G.; Rizzardo, E.; Solomon, D.H. Polym. Bull. 1982, 6, 589.
- 319. Buzanowski, W.C.; Graham, J.D.; Priddy, D.B.; Shero, E. Polymer 1992, 33, 3055.
- 320. IIui, A.; Hamielec, A.E. J. Appl. Polym. Sci. 1972, 16, 749.
- 321. Kothe, T.; Fischer, H. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 4009.
- 322. Olaj, O.F.; Kaufmann, II.F.; Breitenbach, J.W. Makromol. Chem. 1977, 178, 2707.
- 323. Clouet, G.; Chaumont, P.; Corpart, P. J. Polym. Sci., Part A: Polym. Chem. 1993, 31, 2815.
- 324. Walling, C.; Briggs, E.R. J. Am. Chem. Soc. 1946, 68, 1141.
- 325. Lingnau, J.; Meyerhoff, G. Polymer 1983, 24, 1473.
- 326. Lingnau, J.; Meyerhoff, G. Macromolecules 1984, 17, 941.
- 327. Lingnau, J.; Stickler, M.; Meyerhoff, G. Eur. Polym. J. 1980, 16, 785.
- 328. Stickler, M.; Meyerhoff, G. Makromol. Chem. 1978, 179, 2729.
- 329. Hall, H.K., Jr.; Padias, A.B. Acc. Chem. Res. 1990, 23, 3.

158

Initiation

- 330. Matsuda, M.; Abe, K. J. Polym. Sci., Part A-1 1968, 6, 1441.
- 331. Sato, T.; Abe, M.; Otsu, T. Makromol. Chem. 1977, 178, 1061.
- 332. Spychaj, T.; Hamielee, A.E. J. Appl. Polym. Sci. 1991, 42, 2111.
- 333. Hasha, D.L.; Priddy, D.B.; Rudolf, P.R.; Stark, E.J.; de Pooter, M.; Van Damme, F. Macromolecules 1992, 25, 3046.
- 334. Kirchner, K.; Schlapkohl, H. Makromol. Chem. 1976, 177, 2031.
- 335. Stille, J.K.; Chung, D.C. Macromolecules 1975, 8, 83.
- Hall, H.K., Jr.; Padias, A.B.; Pandya, A.; Tanaka, H. Macromolecules 1987, 20, 247.
- 337. Kokubo, T.; Iwatsuki, S.; Yamashita, Y. Makromol. Chem. 1969, 123, 256.
- 338. Sato, T.; Abe, M.; Otsu, T. J. Macromol. Sci., Chem. 1981, A15, 367.
- 339. Gaylord, N.G.; Takahashi, A. Adv. Chem. Ser. 1969, 91, 94.
- 340. Jug, K. J. Am. Chem. Soc. 1987, 109, 3534.
- 341. Fischer, H.; Radom, L. Angew. Chem., Int. Ed. Engl. 2001, 40, 1340.
- 342. Giese, B. Angew. Chem., Int. Ed. Engl. 1983, 22, 753.
- 343. Ghosez-Giese, A.; Giese, B. ACS Symp. Ser. 1998, 685, 50.
- 344. Tedder, J.M. Angew. Chem., Int. Ed. Engl. 1982, 21, 401.
- 345. Beckwith, A.L.J. Tetrahedron 1981, 37, 3073.
- 346. Tedder, J.M.; Walton, J.C. Acc. Chem. Res. 1976, 9, 183.
- 347. Tedder, J.M.; Walton, J.C. Tetrahedron 1980, 36, 701.
- 348. Giese, B.; Lachhein, S. Angew. Chem., Int. Ed. Engl. 1981, 20, 967.
- Lorand, J.P. In Landoldt-Bornstein, New Series, Radical Reaction Rates in Solution; Fischer, H., Ed.; Springer-Verlag: Berlin, 1984; Vol. II/13a, p 135.
- Asmus, K.-D.; Bonifacic, M. In Landoldt-Börnstein, New Series, Radical Reaction Rates in Solution; Fischer, H., Ed.; Springer-Verlag: Berlin, 1984; Vol. II/13b.
- Roduner, E.; Crockett, R. In Landoldt-Bornstein, New Series, Radical Reaction Rates in Solution; Fischer, H., Ed.; Springer-Verlag: Berlin, 1995; Vol. II/18a, p 79.
- 352. Heberger, K.; Fischer, H. Int. J. Chem. Kinet. 1993, 25, 249.
- 353. Busfield, W.K.; Jenkins, I.D.; Van Le, P. Polym. Bull. 1997, 38, 149.
- 354. Busfield, W.K.; Jenkins, I.D.; Van Le, P. Polym. Bull. 1996, 36, 435.
- Busfield, W.K.; Jenkins, I.D.; Van Le, P. J. Polym. Sci., Part A: Polym. Chem. 1998, 36, 2169.
- 356. Gridnev, A.A.; Ittel, S.D. Macromolecules 1996, 29, 5864.
- 357. Krstina, J.; Moad, G.; Solomon, D.H. Eur. Polym. J. 1992, 28, 275.
- 358. Behari, K.; Bevington, J.C.; Huckerby, T.N. Makromol. Chem. 1987, 188, 2441.
- Bevington, J.C.; Huckerby, T.N.; Hutton, N.W.E. J. Polym. Sci., Polym. Chem. Ed. 1982, 20, 2655.
- 360. Bevington, J.C.; Huckerby, T.N.; Hutton, N.W.E. Eur. Polym. J. 1984, 20, 525.
- 361. Ayrey, G.; Jumangat, K.; Bevington, J.C.; Huckerby, T.N. *Polym. Commun.* **1983**, 24, 275.
- 362. Bevington, J.C.; Huckerby, T.N.; Varma, S.C. Eur. Polym. J. 1986, 22, 427.
- 363. Barson, C.A.; Bevington, J.C.; Huckerby, T.N. Polym. Bull. 1986, 16, 209.
- 364. Bevington, J.C.; Breuer, S.W.; Huckerby, T.N. Polym. Commun. 1984, 25, 260.
- 365. Chatgilialoglu, C.; Ingold, K.U.; Scaiano, J.C. J. Am. Chem. Soc. 1981, 103, 7739.
- 366. Heberger, K.; Walbiner, M.; Fischer, II. Angew. Chem., Int. Ed. Engl. 1992, 31, 635.
- 367. Szwarc, M. J. Polym. Sci. 1955, 16, 367.

- 368. Herk, L.; Stefani, A.; Szwarc, M. J. Am. Chem. Soc. 1961, 83, 3003.
- Beranek, I.; Fischer, H. In Free Radicals in Synthesis and Biology, Minisci, F., Ed.; Kluwer: Dordrecht, 1989; p 303.
- Bevington, J.C.; Cywar, D.A.; Huckerby, T.N.; Senogles, E.; Tirrell, D.A. *Eur. Polym. J.* **1990**, *26*, 41.
- 371. Bevington, J.C.; Cywar, D.A.; Huckerby, T.N.; Senogles, E.; Tirrell, D.A. Eur. Polym. J. 1990, 26, 871.
- 372. Nakamura, T.; Suyama, S.; Busfield, W.K.; Jenkins, I.D.; Rizzardo, E.; Thang, S.H. *Polymer* **1999**, 40, 1395.
- 373. Russell, G.A.; Jiang, W.; Hu, S.S.; Khanna, R.K. J. Org. Chem. 1986, 51, 5498.
- 374. Münger, K.; Fischer, H. Int. J. Chem. Kinet. 1985, 17, 809.
- 375. Bevington, J.C.; Lyons, R.A.; Senogles, E. Eur. Polym. J. 1992, 28, 283.
- 376. Santi, R.; Bergamini, F.; Citterio, A.; Sebastiano, R.; Nicolini, M. J. Org. Chem. 1992, 57, 4250.
- 377. Scaiano, J.C.; Stewart, L.C. J. Am. Chem. Soc. 1983, 105, 3609.
- Levin, Y.A.; Abul'khanov, A.G.; Nefedov, A.G.; Skorobogatova, M.S.; Ivanov, B.E. Dokl. Phys. Chem. (Engl. Transl.) 1977, 235, 728.
- 379. Citterio, A.; Minisci, F.; Vismara, E. J. Org. Chem. 1982, 47, 81.
- 380. Kuwac, Y.; Kamachi, M. Bull. Chem. Soc. Japan 1989, 62, 2474.
- Dzhabiyeva, Z.M.; Matkovskii, P.Y.; Pechatnikov, Y.L.; Byrikhina, N.A. Polym. Sci. USSR (Engl. Transl.) 1985, 27, 2416.
- 382. Bevington, J.C.; Huckerby, T.N.; Hutton, N.W.E. Eur. Polym. J. 1982, 18, 963.
- 383. Mahabadi, H.K.; O'Driscoll, K.F. Makromol. Chem. 1977, 178, 2629.
- 384. Citterio, A.; Vismara, E.; Bernardi, R. J. Chem. Res., Miniprint 1983, 4, 876.
- 385. Pryor, W.A.; Fiske, T.R. Trans. Faraday Soc. 1969, 65, 1865.
- 386. Dickerman, S.C.; Megna, I.S.; Skoultchi, M.M. J. Am. Chem. Soc. 1959, 81, 2270.
- 387. Bevington, J.C.; Ito, T. Trans. Faraday Soc. 1968, 64, 1329.
- 388. Chatgilialoglu, C.; Crich, D.; Komatsu, M.; Ryu, I. Chem. Rev. 1999, 99, 1991.
- Howard, J.A.; Scaiano, J.C. In Landoldt-Bornstein, New Series, Radical Reaction Rates in Solution; Fischer, H., Ed.; Springer-Verlag: Berlin, 1984; Vol. II/13d, p 5.
- Lusztyk, J. In Landoldt-Bornstein, New Series, Radical Reaction Rates in Solution; Fischer, H., Ed.; Springer-Verlag: Berlin, 1995; Vol. II/18d1, p 1.
- 391. Hartung, J.; Gottwald, T.; Spehar, K. Synthesis 2002, 1469.
- 392. Heicklein, J.P. Adv. Photochem. 1988, 14, 177.
- 393. Kochi, J.K. In *Free Radicals*; Kochi, J.K., Ed.; Wiley: New York, 1973; Vol. 2, p 665.
- 394. Walling, C. Pure. Appl. Chem. 1967, 15, 69.
- 395. Ingold, K.U. Pure Appl. Chem. 1967, 15, 49.
- 396. Grant, R.D.; Rizzardo, E.; Solomon, D.H. Makromol. Chem. 1984, 185, 1809.
- 397. Grant, R.D.; Rizzardo, E.; Solomon, D.H. J. Chem. Soc., Perkin Trans. 2 1985, 379.
- 398. Chateauneuf, J.; Lusztyk, J.; Ingold, K.U. J. Am. Chem. Soc. 1988, 110, 2886.
- 399. Bevington, J.C.; Harris, D.O.; Johnson, M. Eur. Polym. J. 1965, 1, 235.
- 400. Bevington, J.C.; Breuer, S.W.; Huckerby, T.N. Macromolecules 1989, 22, 55.
- Moad, G.; Rizzardo, E.; Solomon, D.H. Makromol. Chem., Rapid Commun. 1982, 3, 533.
- 402. Busfield, W.K.; Jenkins, I.D.; Thang, S.H.; Rizzardo, E.; Solomon, D.H. Eur. Polym. J. 1993, 29, 397.

Initiation

- 403. Cuthbertson, M.C.; Rizzardo, E., personal communication.
- 404. Griffiths, P.G.; Rizzardo, E.; Solomon, D.H. Tetrahedron Lett. 1982, 23, 1309.
- 405. Rizzardo, E., unpublished results.
- 406. Cuthbertson, M.J.; Rizzardo, E.; Solomon, D.H. Aust. J. Chem. 1985, 38, 315.
- 407. Cuthbertson, M.J.; Rizzardo, E.; Solomon, D.H. Aust. J. Chem. 1983, 36, 1957.
- 408. Busfield, W.K.; Jenkins, I.D.; Thang, S.H.; Rizzardo, E.; Solomon, D.H. Aust. J. Chem. 1985, 38, 689.
- 409. Busfield, W.K.; Jenkins, I.D.; Thang, S.H.; Rizzardo, E.; Solomon, D.H. J. Chem. Soc., Perkin Trans. 1 1988, 485.
- 410. Jones, M.J.; Moad, G.; Rizzardo, E.; Solomon, D.H. J. Org. Chem. 1989, 54, 1607.
- 411. Sakurai, H.; Hosomi, A. J. Am. Chem. Soc. 1967, 89, 458.
- 412. Walling, C.; McGuinness, J.A. J. Am. Chem. Soc. 1969, 91, 2053.
- 413. Encina, M.V.; Rivera, M.; Lissi, E.A. J. Polym. Sci. Polym. Chem. Ed. 1978, 16, 1709.
- 414. Kunitake, T.; Murakami, S. J. Polym. Sci., Polym. Chem. Ed. 1974, 12, 67.
- 415. Korth, H.-G.; Sustmann, R. Tetrahedron Lett. 1985, 26, 2551.
- 416. Griffiths, P.G.; Rizzardo, E., unpublished data.
- 417. Wong, P.C.; Griller, D.; Scaiano, J.C. J. Am. Chem. Soc. 1982, 104, 5106.
- 418. Weber, M.; Fischer, H. J. Am. Chem. Soc. 1999, 121, 7381.
- 419. Tsentalovich, Y.P.; Kulik, L.V.; Gritsan, N.P.; Yurkovskaya, A.V. J. Phys. Chem. A 1998, 102, 7975.
- 420. Walling, C.; Padwa, A. J. Am. Chem. Soc. 1963, 85, 1593.
- 421. Nakamura, T.; Watanabe, Y.; Suyama, S.; Tezuka, H. J. Chem. Soc., Perkin Trans. 2 2002, 1364.
- 422. Huyser, E.S.; Jankauskas, K.J. J. Org. Chem. 1970, 35, 3196.
- 423. Kamath, V.R.; Sargent, J.D., Jr. J. Coat. Technol. 1987, 59, 51.
- 424. Nakamura, T.; Busfield, W.K.; Jenkins, I.D.; Rizzardo, E.; Thang, S.II.; Suyama, S. Macromolecules 1997, 30, 2843.
- 425. Nakamura, T.; Busfield, W.K.; Jenkins, I.D.; Rizzardo, E.; Thang, S.H.; Suyama, S. J. Am. Chem. Soc. **1997**, 119, 10987.
- 426. Avila, D.V.; Brown, C.E.; Ingold, K.U.; Lusztyk, J. J. Am. Chem. Soc 1993, 115, 466.
- 427. Bertrand, M.P.; Surzur, J.-M. Tetrahedron Lett. 1976, 17, 3451.
- 428. Bertrand, M.P.; Oumar-Mahamat, H.; Surzur, J.M. Bull. Soc. Chim. Fr. 1985, 115.
- 429. Hilborn, J.W.; Pincock, J.A. J. Am. Chem. Soc. 1991, 113, 2683.
- 430. Bevington, J.C. Angew. Makromol. Chem. 1991, 185/186, 1.
- 431. Edge, D.J.; Kochi, J.K. J. Am. Chem. Soc. 1973, 95, 2635.
- 432. Kochi, J.K. J. Am. Chem. Soc. 1962, 84, 774.
- 433. Lane, J.; Tabner, B.J. J. Chem. Soc., Perkin Trans. 2 1984, 1823.
- 434. Barclay, L.R.C.; Griller, D.; Ingold, K.U. J. Am. Chem. Soc. 1982, 104, 4399.
- 435. Beckwith, A.L.J.; Thomas, C.B. J. Chem. Soc., Perkin Trans. 2 1973, 861.
- 436. Nakata, T.; Tokumaru, K.; Simamura, O. Tetrahedron Lett. 1967, 8, 3303.
- 437. Fischer, H. Z. Naturforsch. 1964, 19a, 866.
- 438. Fischer, II.; Giacometti, G. J. Polym. Sci., Polym. Symp. 1967, 16, 2763.
- 439. Roth, II.K.; Wunsche, P. Acta Polym. 1981, 32, 491.
- 440. Maruthamuthu, P. Makromol. Chem., Rapid, Commun. 1980, 1, 23.
- 441. McAskill, N.A.; Sangster, D.F. Aust. J. Chem. 1984, 37, 2137.

The Chemistry of Radical Polymerization

- 442. Sloane, T.M.; Brudzynski, R.J. J. Am. Chem. Soc. 1979, 101, 1495.
- 443. Ledwith, A.; Russell, P.J. J. Polym. Sci., Polym. Lett. Ed. 1975, 13, 109.
- Citterio, A.; Arnoldi, C.; Giordano, C.; Castaldi, G. J. Chem. Soc., Perkin Trans. 1 1983, 891.
- 445. Arnoldi, C.; Citterio, A.; Minisci, F. J. Chem. Soc., Perkin Trans. 2 1983, 531.
- 446. Fristad, W.E.; Peterson, J.R. Tetrahedron 1984, 40, 1469.
- 447. McAskill, A.; Sangster, D.F. Aust. J. Chem. 1979, 32, 2611.
- 448. Ghosh, N.N.; Mandal, B.M. Macromolecules 1986, 19, 19.
- 449. Misra, N.; Mandal, B.M. Macromolecules 1984, 17, 495.
- 450. Misra, N.; Mandal, B.M. J. Polym. Sci., Polym. Lett. Ed. 1985, 23, 63.
- 451. Banthia, A.K.; Mandal, B.M.; Palit, S.R. J. Polym. Sci., Polym. Chem. Ed. 1977, 15, 945.
- 452. Howard, J.A. Rev. Chem. Intermed. 1984, 5, 1.
- 453. Neta, P.; Huie, R.E.; Ross, A.B. J. Chem. Phys. Ref., Data 1990, 19, 413.
- Howard, J.A. In *Free Radicals*; Kochi, J.K., Ed.; Wiley: New York, 1973; Vol. 2, p
 3.
- Alonso, A.; Peinado, C.; Lozano, A.E.; Catalina, F.; Zimmermann, C.; Schnabel, W. J. Macromol. Sci., Chem. 1999, A36, 605.
- Lozano, A.E.; Alonso, A.; Catalina, F.; Peinado, C. Macromol. Theory Simul. 1999, 8, 93.
- 457. Scott, G.P.; Soong, C.C.; Allen, J.L.; Reynolds, J.L. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1963, 4(1), 67.
- 458. Busfield, W.K.; Heiland, K.; Jenkins, I.D. Tetrahedron Lett. 1994, 35, 6541.
- 459. Farina, M.; Di Silvestro, G.; Sozzani, P. Makromol. Chem. 1989, 190, 213.
- 460. Ito, O.; Matsuda, M. J. Am. Chem. Soc. 1979, 101, 5732.
- 461. Busfield, W.K.; Jenkins, I.D.; Heiland, K. Tetrahedron Lett. 1995, 36, 1109.
- 462. Ito, O.; Matsuda, M. J. Am. Chem. Soc. 1979, 101, 1815.
- 463. Sato, T.; Abe, M.; Otsu, T. Makromol. Chem. 1977, 178, 1951.
- 464. Ito, O. J. Am. Chem. Soc 1983, 105, 850.
- 465. Sumiyoshi, T.; Schnabel, W. Makromol. Chem. 1985, 186, 1811.
- Kajiwara, A.; Konishi, Y.; Morishima, Y.; Schnabel, W.; Kuwata, K.; Kamachi, M. Macromolecules 1993, 26, 1656.
- Bottle, S.E.; Busfield, W.K.; Grice, I.D.; Heiland, K.; Meutermans, W.; Monteiro, M. Prog. Pac. Polym. Sci. 1994, 3, 85.
- 468. Geers, B.N.; Gleicher, G.J.; Church, D.F. Tetrahedron 1980, 36, 997.
- 469. Ito, O.; Matsuda, M. J. Am. Chem. Soc. 1979, 101, 5732.
- 470. Ito, O.; Matsuda, M. J. Phys. Chem. 1984, 88, 1002.
- 471. Ito, O.; Matsuda, M. J. Org. Chem. 1983, 48, 2748.
- 472. Ito, O.; Matsuda, M. J. Org. Chem. 1983, 48, 2410.
- 473. Bessiere, J.-M.; Boutevin, B.; Sarraf, L. Polym. Bull. 1987, 18, 253.
- Baxter, J.E.; Davidson, R.S.; Hageman, H.J.; Overeem, T. Makromol. Chem., Rapid Commun. 1987, 8, 311.
- 475. Weber, M.; Turro, N.J. J. Phys. Chem. A 2003, ASAP.
- 476. Perkins, M.J. Adv. Phys. Org. Chem 1981, 17, 1.
- 477. Kamachi, M. Adv. Polym. Sci. 1987, 82, 207.
- 478. Yamada, B.; Westmoreland, D.G.; Kobatake, S.; Konosu, O. Prog. Polym. Sci. 1999, 24, 565.

Initiation

- 479. Chalfont, G.R.; Perkins, M.J.; Horsfield, A. J. Am. Chem. Soc. 1968, 90, 7141.
- 480. Savedoff, L.G.; Ranby, B. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1978, 19(1), 629.
- 481. Yamada, B.; Fujity, M.; Sakamoto, K.; Otsu, T. Polym. Bull. 1994, 33, 309.
- 482. Yamada, B.; Yoshikawa, E.; Otsu, T. Polymer 1992, 33, 3245.
- 483. Bevington, J.C.; Tabner, B.J.; Fridd, P.F. Rev. Roum. Chim. 1980, 25, 947.
- 484. Bevington, J.C.; Fridd, P.F.; Tabner, B.J. J. Chem. Soc., Perkin Trans. 2 1982, 1389.
- 485. Pichot, C.; Spitz, R.; Guyot, A. J. Macromol. Sci.- Chem. 1977, A11, 251.
- 486. Minisci, F. Acc. Chem. Res. 1975, 8, 165.
- 487. Caronna, T.; Citterio, A.; Ghirardini, M.; Minisci, F. Tetrahedron 1977, 33, 793.
- 488. Bamford, C.H.; Jenkins, A.D.; Johnston, R. Proc. R. Soc., London 1957, A239, 214.
- 489. Giese, B. Radicals in Organic Synthesis: Formation of Carbon-Carbon Bonds; Pergamon Press: Oxford, 1986.
- 490. Giese, B. Rev. Chem. Intermed. 1986, 7, 3.
- 491. Hill, C.L.; Whitesides, G.M. J. Am. Chem. Soc. 1974, 96, 870.
- 492. Jones, S.A.; Prementine, G.S.; Tirrell, D.A. J. Am. Chem. Soc. 1985, 107, 5275.
- 493. Russell, G.A. Acc. Chem. Res. 1989, 22, 1.
- 494. Pike, P.; Hershberger, S.; Hershberger, J. Tetrahedron Lett. 1985, 26, 6289.
- 495. Giese, B.; Gonzalez-Gomez, J.A.; Witzel, T. Angew. Chem., Int. Ed. Engl. 1984, 23, 69.
- 496. Rizzardo, E.; Solomon, D.H. Polym. Bull. 1979, 1, 529.
- 497. Busfield, W.K.; Jenkins, I.D.; Thang, S.H.; Rizzardo, E.; Solomon, D.H. Tetrahedron Lett. 1985, 26, 5081.
- 498. Aldabbagh, F.; Busfield, W.K.; Jenkins, I.D. Aust. J. Chem. 2001, 54, 313.
- 499. Busfield, W.K.; Grice, I.D.; Jenkins, I.D. Aust. J. Chem. 1995, 48, 625.
- 500. Nakamura, T.; Busfield, W.K.; Jenkins, I.D.; Rizzardo, E.; Thang, S.H.; Suyama, S. J. Org. Chem. 1997, 62, 5578.
- 501. Nakamura, T.; Watanabe, Y.; Tczuka, H.; Busfield, W.K.; Jenkins, I.D.; Rizzardo, E.; Thang, S.H.; Suyama, S. *Chem. Lett.* **1997**, 1093.
- Nakamura, T.; Busfield, W.K.; Jenkins, I.D.; Rizzardo, E.; Thang, S.H.; Suyama, S. Macromolecules 1996, 29, 8975.
- 503. Nakamura, T.; Busfield, W.K.; Jenkins, I.D.; Rizzardo, E.; Thang, S.H.; Suyama, S. J. Am. Chem. Soc. **1996**, 118, 10824.
- 504. Beckwith, A.L.J.; Poole, J.S. J. Am. Chem. Soc 2002, 124, 9489.
- 505. Busfield, W.K.; Jenkins, I.D.; Monteiro, M.J. Polymer 1997, 38, 165.
- 506. Zetterlund, P.B.; Busfield, W.K.; Jenkins, I.D. Macromolecules 1999, 32, 8041.
- 507. Zetterlund, P.B.; Busfield, W.K.; Jenkins, I.D. Macromolecules 2002, 35, 7232.
- 508. Beckwith, A.L.J.; Bowry, V.W.; Moad, G. J. Org. Chem. 1988, 53, 1632.
- 509. Chatcauncuf, J.; Lusztyk, J.; Ingold, K.U. J. Org. Chem. 1988, 53, 1629.
- 510. Bowry, V.W.; Ingold, K.U. J. Am. Chem. Soc. 1992, 114, 4992.
- 511. Beckwith, A.L.J.; Bowry, V.W.; Ingold, K.U. J. Am. Chem. Soc. 1992, 114, 4983.
- 512. Moad, G.; Rizzardo, E.; Solomon, D.H.; Beckwith, A.L.J. Polym. Bull. 1992, 29, 647.
- 513. Gerlock, J.L.; Zacmanidis, P.J.; Bauer, D.R.; Simpson, D.J.; Blough, N.V.; Salmeen, I.T. *Free Radical Res. Commun.* **1990**, *10*, 119.

- 514. Moad, G.; Shipp, D.A.; Smith, T.A.; Solomon, D.II. *Macromolecules* **1997**, *30*, 7627.
- Moad, G.; Shipp, D.A.; Smith, T.A.; Solomon, D.H. J. Phys. Chem. A 1999, 103, 6580.
- 516. Golubev, V.A.; Kozlov, Y.N.; Petrov, A.N.; Purmal, A.P. Prog. React. Kinet. 1991, 16, 35.
- 517. Anderson, D.R.; Keute, J.; Chapel, H.L.; Koch, T.H. J. Am. Chem. Soc. 1979, 101, 1904.
- 518. Keana, J.F.W.; Dinerstein, R.J.; Baitis, F. J. Org. Chem. 1971, 36, 209.
- 519. Johnston, L.J.; Tencer, M.; Scaiano, J.C. J. Org. Chem. 1986, 51, 2806.
- 520. Coxon, J.M.; Pattsalides, E. Aust. J. Chem. 1982, 35, 509.
- 521. Bottle, S.E.; Chand, U.; Micallef, A.S. Chem. Lett. 1997, 857.
- 522. Aldabbagh, F.; Busfield, W.K.; Jenkins, I.D.; Thang, S.H. Tetrahedron Lett. 2000, 41, 3673.
- 523. Connolly, T.J.; Scaiano, J.C. Tetrahedron Lett. 1997, 38, 1133.
- 524. Watanabe, Y.; Ishigaki, H.; Okada, H.; Suyama, S. Polym. J. 1997, 29, 733.
- 525. Watanabe, Y.; Ishigaki, H.; Okada, H.; Suyama, S. Polym. J. 1997, 29, 603.
- 526. Watanabe, Y.; Ishigaki, H.; Okada, H.; Suyama, S. Polym. J. 1997, 29, 366.
- 527. Watanabe, Y.; Ishigaki, H.; Okada, H.; Suyama, S. Polym. J. 1997, 29, 940.
- 528. Watanabe, Y.; Ishigaki, H.; Okada, H.; Suyama, S. Polym. J. 1998, 30, 192.
- 529. Shetty, S.; Garcia-Rubio, L.H. Polym. Mater. Sci. Eng. 1991, 65, 103.
- 530. Koenig, J.L. Spectroscopy of Polymers; Elsevier: New York, 1999.
- 531. Buback, M.; Huckestein, B.; Ludwig, B. Makromol. Chem., Rapid Commun. 1992, 13, 1.
- 532. Buback, M.; Huckestein, B.; Leinhos, U. Makromol. Chem., Rapid Commun. 1987, 8, 473.
- Carduner, K.R.; Carter, R.O.; Zinbo, M.; Gerlock, J.L.; Bauer, D.R. Macromolecules 1988, 21, 1598.
- 534. Meijs, G.F.; Morton, T.C.; Rizzardo, E.; Thang, S.H. *Macromolecules* **1991**, *24*, 3689.
- 535. Meijs, G.F.; Rizzardo, E.; Thang, S.H. Macromolecules 1988, 21, 3122.
- 536. Bevington, J.C.; Ebdon, J.R.; Huckerby, T.N. In *NMR Spectroscopy of Polymers*; Ibbett, R.N., Ed.; Blackie: London, 1993; p 51.
- 537. Hatada, K.; Kitayama, T.; Ute, K.; Terawaki, Y.; Yanagida, T. *Macromolecules* **1997**, *30*, 6754.
- 538. Hatada, K. NMR Spectroscopy of Polymers; Springer-Verlag: Berlin, 2003.
- 539. Bevington, J.C.; Cywar, D.A.; Huckerby, T.N.; Senogles, E.; Tirrell, D.A. *Eur. Polym. J.* **1988**, *24*, 699.
- 540. Johns, S.R.; Rizzardo, E.; Solomon, D.H.; Willing, R.I. Makromol. Chem., Rapid Commun. 1983, 4, 29.
- 541. Hatada, K.; Kitayama, T.; Masuda, E. Polym. J. 1985, 17, 985.
- 542. Kashiwagi, T.; Inaba, A.; Brown, J.E.; Hatada, K.; Kitayama, T.; Masuda, E. Macromolecules 1986, 19, 2160.
- 543. Hatada, K.; Kitayama, T.; Masuda, E. Polym. J. 1986, 18, 395.
- 544. Bevington, J.C.; Ebdon, J.R.; Huckerby, T.N. Eur. Polym. J. 1985, 21, 685.
- 545. Bevington, J.C.; Huckerby, T.N. Polymer 1992, 33, 1323.
- 546. Saito, T.; Rinaldi, P.L. J. Magn. Reson. 1998, 130, 135.

164

Initiation

- 547. Saito, T.; Rinaldi, P.L. J. Magn. Reson. 1998, 132, 41.
- 548. Meng, H.H.; Saito, T.; Rinaldi, P.L.; Wyzgoski, F.; Helfer, C.A.; Mattice, W.L.; Harwood, II.J. *Macromolecules* **2001**, *34*, 801.
- 549. Montaudo, G.; Montaudo, M.S.; Montaudo, G.; Lattimer, R.P., Eds.; CRC Press: Boca Raton, 1999; p 41.
- 550. Meisters, A.; Moad, G.; Rizzardo, E.; Solomon, D.H. Polym. Bull. 1988, 20, 499.
- 551. Farina, M. Makromol. Chem., Macromol. Symp. 1987, 10/11, 255.
- 552. Ohtani, H.; Ishiguro, S.; Tanaka, M.; Tsuge, S. Polym. J. 1989, 21, 41.
- 553. Ohtani, H.; Suzuki, A.; Tsuge, S. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 1880.
- 554. Hanton, S.D. Chem. Rev. 2001, 101, 527.
- 555. Nielen, M.W.F. Mass Spectrom. Rev. 1999, 18, 309.
- Maloney, D.R.; Hunt, K.H.; Lloyd, P.M.; Muir, A.V.G.; Richards, S.N.; Derrick, P.J.; Haddleton, D.M. J. Chem. Soc., Chem. Commun. 1995, 561.
- 557. Zammit, M.D.; Davis, T.P.; Haddleton, D.M.; Suddaby, K.G. Macromolecules 1997, 30, 1915.
- 558. Suddaby, K.G.; Hunt, K.H.; Haddleton, D.M. Macromolecules 1996, 29, 8642.
- 559. Haddleton, D.M.; Maloney, D.R.; Suddaby, K. Polymer 1997, 38, 6207.
- 560. Liu, Z.F.; Rimmer, S. Macromolecules 2002, 35, 1200.
- 561. Vana, P.; Davis, T.P.; Barner-Kowollik, C. Aust. J. Chem. 2002, 55, 315.
- 562. Schilli, C.; Lanzendoerfer, M.G.; Mueller, A.H.E. Macromolecules 2002, 35, 6819.
- 563. Ganachaud, F.; Monteiro, M.J.; Gilbert, R.G.; Dourges, M.A.; Thang, S.H.; Rizzardo, E. *Macromolecules* **2000**, *33*, 6738.
- Charmot, D.; Corpart, P.; Adam, H.; Zard, S.Z.; Biadatti, T.; Bouhadir, G. Macromol. Symp. 2000, 150, 23.
- 565. D'Agosto, F.; Hughes, R.; Charreyre, M.T.; Pichot, C.; Gilbert, R.G. Macromolecules 2003, 36, 621.
- 566. Vana, P.; Albertin, L.; Barner, L.; Davis, T.P.; Barner-Kowollik, C. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 4032.
- Norman, J.; Moratti, S.C.; Slark, A.T.; Irvine, D.J.; Jackson, A.T. Macromolecules 2002, 35, 8954.
- 568. Borman, C.D.; Jackson, A.T.; Bunn, A.; Cutter, A.L.; Irvine, D.J. *Polymer* **2000**, *41*, 6015.
- 569. Farcet, C.; Belleney, J.; Charleux, B.; Pirri, R. Macromolecules 2002, 35, 4912.
- Dourges, M.A.; Charleux, B.; Vairon, J.P.; Blais, J.C.; Bolbach, G.; Tabet, J.C. Macromolecules 1999, 32, 2495.
- Bartsch, A.; Dempwolf, W.; Bothe, M.; Flakus, S.; Schmidt-Naake, G. Macromol. Rapid. Commun. 2004, 24, 614.
- 572. Palit, S.R. Makromol. Chem. 1959, 36, 89.
- 573. Palit, S.R. Makromol. Chem. 1960, 38, 96.
- 574. Rizzardo, E.; Solomon, D.H. J. Macromol. Sci., Chem. 1979, A13, 997.
- 575. Ghosh, N.N.; Sengputa, P.K.; Pramanik, A. J. Polym. Sci., Part A 1965, 3, 1725.
- 576. Rizzardo, E.; Solomon, D.H. J. Macromol. Sci., Chem. 1979, A13, 1005.
- 577. Postma, A.; Donovan, R.; Davis, T.P.; Moad, G.; O'Shea, M. *Polymer* **2005**, submitted for publication.
- 578. Donovan, A.R.; Moad, G. Polymer 2005, 5005.
- 579. Kern, W. Chem. Ztg. 1976, 100, 401.

- 580. Bevington, J.C. Radical Polymerization; Academic Press: London, 1961.
- 581. Bevington, J.C. Trans. Faraday Soc. 1955, 51, 1392.
- 582. Bevington, J.C.; Ebdon, J.R. Developments in Polymerisation 1979, 2, 1.
- 583. Bevington, J.C. Makromol. Chem., Macromol. Symp. 1987, 10/11, 89.
- 584. Moad, G. Chem. Aust. 1991, 58, 122.
- Moad, G. In Annual Reports in NMR Spectroscopy; Webb, G.A., Ed.; Academic Press: London, 1994; Vol. 29, p 287.
- 586. Barson, C.A.; Bevington, J.C.; Breuer, S.W. Eur. Polym. J. 1989, 25, 259.
- 587. Barson, C.A.; Behari, K.; Bevington, J.C.; Huckerby, T.N. J. Macromol. Sci., Chem. 1988, A25, 1137.
- 588. Barson, C.A.; Bevington, J.C.; Huckerby, T.N. Polym. Bull. 1991, 25, 83.
- 589. Bevington, J.C.; Huckerby, T.N.; Varma, S.C. Eur. Polym. J. 1987, 19, 319.
- 590. Barson, C.A.; Bevington, J.C.; Huckerby, T.N. Polym. Bull. 1989, 22, 131.
- 591. Bevington, J.C.; Huckerby, T.N. Macromolecules 1985, 18, 176.
- 592. Barson, C.A.; Bevington, J.C.; Huckerby, T.N. Polymer 1991, 32, 3415.
- 593. Barson, C.A.; Bevington, J.C.; Breuer, S.W.; Huckerby, T.N. *Eur. Polym. J.* **1989**, 25, 527.
- 594. Barson, C.A.; Bevington, J.C.; Breuer, S.W.; Huckerby, T.N. Makromol. Chem., Rapid Commun. 1992, 13, 97.
- 595. Bevington, J.C.; Breuer, S.W.; Huckerby, T.N.; Hunt, B.J.; Jones, R. Eur. Polym. J. 1998, 34, 539.
- 596. Bevington, J.C.; Breuer, S.W.; Huckerby, T.N.; Hunt, B.J.; Jones, R. Eur. Polym. J. 1997, 33, 1225.
- 597. Bevington, J.C.; Huckerby, T.N.; Vickerstaff, N. Makromol. Chem., Rapid Commun. 1983, 4, 349.
- 598. Bevington, J.C.; Huckerby, T.N.; Vickerstaff, N. Makromol. Chem., Rapid Commun. 1988, 9, 791.
- 599. Cywar, D.A.; Tirrell, D.A. Macromolecules 1986, 19, 2908.
- 600. Fellows, C.M.; Senogles, E. Eur. Polym. J. 2001, 37, 1091.
- 601. Fellows, C.M.; Senogles, E. Eur. Polym. J. 1998, 34, 1249.
- 602. Fellows, C.M.; Senogles, E. Eur. Polym. J. 1999, 35, 9.
- 603. Bevington, J.C.; Bowden, B.F.; Cywar, D.A.; Lyons, R.A.; Senogles, E.; Tirrell, D.A. Eur. Polym. J. 1991, 27, 1239.
- 604. Prementine, G.S.; Tirrell, D.A. Macromolecules 1987, 20, 3034.
- 605. Kitayama, T.; Kishiro, S.; Masuda, E.; Hatada, K. Polym. Bull. 1991, 25, 205.
- Bevington, J.C.; Ebdon, J.R.; Huckerby, T.N.; Hutton, N.W.E. Polymer 1982, 23, 163.
- 607. Behari, K.; Bevington, J.C.; Huckerby, T.N. Polymer 1988, 29, 1867.
- 608. Bevington, J.C.; Huckerby, T.N. J. Macromol. Sci., Chem. 1983, A20, 753.
- 609. Bevington, J.C.; Huckerby, T.N.; Hunt, B.J. Br. Polym. J. 1985, 17, 43.

4.1 Introduction

The propagation step of radical polymerization comprises a sequence of radical additions to carbon-carbon double bonds. The factors that govern the rate and specificity of radical addition have been dealt with in general terms in Section 2.3. In order to produce high molecular weight polymers, a propagating radical must show a high degree of specificity in its reactions with unsaturated systems. It must give addition to the exclusion of side reactions that bring about the cessation of growth of the polymer chain. Despite this limitation, there is considerable scope for structural variation in homopolymers.

The asymmetric substitution pattern of most monomers means that addition gives rise to a chiral center and their polymers will have tacticity (Section 4.2).



Addition to double bonds may not be completely regiospecific. The predominant head-to-tail structure may be interrupted by head-to-head and tail-to-tail linkages (Section 4.3).



Intramolecular rearrangement of the initially formed radical may occur occasionally (*e.g.* backbiting - Section 4.4.3) or even be the dominant pathway (*e.g.* cyclopolymerization – Section 4.4.1, ring-opening polymerization – Section 4.4.2). These pathways can give rise to branches, rings, or internal unsaturation in the polymer chain.



This chapter is primarily concerned with the chemical microstructure of the products of radical homopolymerization. Variations on the general structure $(CH_2-CXY)_n$ are described and the mechanisms for their formation and the associated rate parameters are examined. With this background established, aspects of the kinetics and thermodynamics of propagation are also considered (Section 4.5).

4.2 Stereosequence Isomerism - Tacticity

The classical representation of a homopolymer chain, in which the end groups are disregarded and only one monomer residue is considered, allows no possibility for structural variation. However, possibilities for stereosequence isomerism arise as soon as the monomer residue is considered in relation to its neighbors and the substituents X and Y are different. The chains have tacticity (Section 4.2.1). Experimental methods for tacticity determination are summarized in 4.2.2 and the tacticity of some common polymers is considered in 4.2.3.

The following discussion is limited to polymers of mono- or 1,1-disubstituted monomers. Other factors become important in describing the types of stereochemical isomerism possible for polymers formed from other monomers (e.g. 1,2-disubstituted monomers).¹

4.2.1 Terminology and Mechanisms

Detailed discussion of polymer tacticity can be found in texts by Randall,² Bovey,^{1,3} Koenig,^{4,5} Tonelli⁶ and Hatada.⁷ In order to understand stereoisomerism in polymer chains formed from mono- or 1,1-disubstituted monomers, consider four idealized chain structures:

(a) The isotactic chain where the relative configuration of all the substituted carbons in the chain is the same.



For the usual diagrammatic representation of a polymer chain, this corresponds to the situation where similar substituents lie on the same side of a plane perpendicular to the page and containing the polymer backbone.



(b) The syndiotactic chain where the relative configuration of centers alternates along the chain.



(c) The heterotactic chain where the dyad configuration alternates along the chain.



(d) The atactic chain where there is a random arrangement of centers along the chain.*

For polymers produced by radical polymerization, while one of these structures may predominate, the idealized structures do not occur. It is necessary to define parameters to more precisely characterize the tacticity of polymer chains.

It should be stressed that this treatment of polymer stereochemistry only deals with relative configurations; whether a substituent is "up or down" with respect to that on a neighboring unit. Therefore, the smallest structural unit which contains stereochemical information is the dyad. There are two types of dyad; meso (m), where the two chiral centers have like configuration, and racemic (r), where the centers have opposite configuration (Figure 4.1).



Figure 4.1 Representation of meso (*m*) and racemic (*r*) dyads with polymer chains.

Confusion can arise because of the seemingly contradictory nomenclature established for analogous model compounds with just two asymmetric centers.⁸ In such compounds, the diastereoisomers are named as in the following example (Figure 4.2).

^{*} In the literature the term atactic is sometimes used to refer to any polymer that is not entirely isotactic or not entirely syndiotactic.



The Chemistry of Radical Polymerization

Figure 4.2 Representation of meso (*m*) and racemic (*r*) diastereoisomers of low molecular weight compounds.

It is usual to discuss triads, tetrads, pentads, *etc.* in terms of the component dyads. For example, the *mrrrmr* heptad is represented as shown in Figure 4.3.



Figure 4.3 Representation of *mrrrmr* heptad identifying component *n*-ads.

It is informative to consider how tacticity arises in terms of the mechanism for propagation. The radical center on the propagating species will usually have a planar sp^2 configuration. As such it is achiral and it will only be locked into a specific configuration after the next monomer addition. This situation should be contrasted with that which pertains in anionic or coordination polymerizations where the active center is pyramidal and therefore has chirality. This explains why stereochemical control is more easily achieved in these polymerizations.

The configuration of a center in radical polymerization is established in the transition state for addition of the next monomer unit when it is converted to a tetrahedral sp^3 center. If the stereochemistry of this center is established at random (Scheme 4.1; $k_m = k_r$) then a pure atactic chain is formed and the probability of finding a *meso* dyad, P(m), is 0.5.

Polymers formed from monosubstituted monomers (X-H) under the usual reaction conditions (e.g. 60 °C, bulk) appear almost atactic with only a slight

170

preference for syndiotacticity and values of P(m) in the range 0.45-0.52 (Table 4.1, Section 4.2.3).



If the reaction center adopts a preferred configuration with respect to the configuration of the penultimate unit in the chain (Scheme 4.1; $k_m \neq k_r$) then Bernoullian statistics apply. The stereochemistry of the chain is characterized by the single parameter, P(m) or P(r) [= 1-P(m)]. The *n*-ad concentrations can be calculated simply by multiplying the concentrations of the component dyads. Thus the relative triad concentrations are given by the following expressions (eq. 1-3)

$$mm = P(m)^2 \tag{1}$$

$$mr = rm = 2 P(m) P(r) = 2 P(m) (1 - P(m))$$
(2)

$$rr = P(r)^{2} = (1 - P(m))^{2}$$
(3)

Higher *n*-ads are calculated similarly. Thus for the *mrrrmr* heptad:

$$mrrrmr = 2 P(m) P(r) P(r) P(r) P(m) P(r) = 2 P(m)^2 P(r)^4$$

The factor 2 is introduced in the case of asymmetric *n*-ads which can be formed in two ways (mrrrmr = rmrrrm).



Where the nature of the preceding dyad is important in determining the configuration of the new chiral center (Scheme 4.2), first order Markov statistics

172 The Chemistry of Radical Polymerization

apply. Propagation is subject to a penpenultimate unit effect (also called an antepenultimate unit effect). Two parameters are required to specify the stereochemistry, P(m|r) [=1-P(m|m)] and P(r|r) [=1-P(r|m)], where P(i|j) is the conditional probability that given a j dyad, the next unit in the chain will be an i dyad.* It can be shown that

$$P(m) = P(m|r) / (P(m|r) + P(r|m))$$
(4)

The relative triad concentrations are then given by the following expressions (eq. 5-7)

$$mm = P(m) P(m|m) \tag{5}$$

$$mr = rm = 2 P(m) P(r|m) = 2 P(m) (1 - P(m|m))$$
(6)

$$rr = P(r) P(r|r) \tag{7}$$

Again the higher *n*-ads are calculated similarly. Thus for the *mrrrmr* heptad: mrrrmr = 2 P(m) P(r|m) P(r|r) P(r|r) P(m|r) P(r|m)

We can also write expressions to calculate P(m|r) and P(r|m) from the triad concentrations (eq. 8, 9).

$$P(m|r) = mr/(2 mm + mr)$$
(8)

$$P(r|m) - rm/(2 rr + rm) \tag{9}$$

The Coleman-Fox two state model describes the situation where there is restricted rotation about the bond to the preceding unit (Scheme 4.3). If this is slow with respect to the rate of addition, then at least two conformations of the propagating radical need to be considered each of which may react independently with monomer. The rate constants associated with the conformational equilibrium and two values of P(m) are required to characterize the process.



More complex situations may also be envisaged and it should always be borne in mind that the fit of experimental data to a simple model provides support for but does not prove that model. The power of the experiment to discriminate between models has to be considered.

^{*} In texts by Bovey^{1.3} and Tonelli⁶ P(i|j) is written Pj/i.

4.2.2 Experimental Methods for Determining Tacticity

The application of NMR spectroscopy to tacticity determination of synthetic polymers was pioneered by Bovey and Tiers.⁹ NMR spectroscopy is the most used method and often the only technique available for directly assessing tacticity of polymer chains.^{1,2,7,8,10,11} The chemical shift of a given nucleus in or attached to the chain may be sensitive to the configuration of centers three or more monomer units removed. Other forms of spectroscopy (*e.g.* IR spectroscopy^{12,13}) are useful with some polymers and various physical properties (*e.g.* the Kerr effect¹⁴) may also be correlated with tacticity.

The ambiguity of the NMR peak assignments may cause problems in tacticity determination. The usual method of assigning peaks to configurational sequences involves matching expected and measured peak intensities. There are obvious problems inherent in this approach and these are being redressed by the application of 2D NMR methods which in many cases can provide unambiguous assignments.¹⁵ These methods have been applied to make absolute tacticity assignments for PAA,¹⁶ PMMA,¹⁷⁻²⁰ PMAN,²¹ PVA,^{22,23} PVC^{24,25} and PVF.²⁶ In some cases, an *a priori* assignment of chemical shifts using theoretical methods (making use of the rotational isomeric state model and the γ -gauche effect) may also be possible.⁶ Such methods have been shown useful for polypropylenc, PVC and PVF.

Attention must also be paid to sample preparation methods.²⁷ The number average molecular weight of the polymer must be sufficiently high that signals due to sequences near the chain ends make no significant contribution to the spectrum. For PMMA with heptad resolution, this requires that \overline{M}_n is in excess of 30,000. Similarly, one must be concerned about structural irregularities introduced through head addition, backbiting and other processes.

4.2.3 Tacticities of Polymers

Many radical polymerizations have been examined from the point of view of establishing the stereosequence distribution. For most systems it is claimed that the tacticity is predictable within experimental error* by Bernoullian statistics [*i.e.* by the single parameter P(m) – see 4.2.1].

Tacticity is most often determined by NMR analysis and usually by looking at the signals associated with the -CXY- group (refer Figure 4.3). The analysis then provides the triad concentrations (mm, mr and rr) and the value of m or P(m) is given by eq. 10.

$$P(m) = mm + 0.5 mr$$
 (10)

^{*} It should be noted that, in some studies, deviations of 5-10% in expected and measured NMR peak intensities have been ascribed to experimental error. Such error is sufficient to hide significant departures from Bernoullian statistics.^{28,29}

Most polymers formed by radical polymerization have an excess of syndiotactic over isotactic dyads [*i.e.* $P(m) \le 0.5$]. P(m) typically lies in the range 0.4-0.5 for vinyl monomers and 0.2-0.5 for 1,1-disubstituted monomers. It is also generally found that P(m) (the fraction of isotactic dyads) decreases with decreasing temperature.³⁰ Data on tacticity for some common polymers are presented in Table 4.1.

There are exceptions to this general rule. For example, polymerizations of methacrylates with very bulky ester substituents (1-4) show a marked preference for isotacticity³¹ whereas polymerizations of MMA show a significant preference for syndiotacticity (Table 4.1). Polymerization of the acrylamide derivative **5** which has a bulky substituent on nitrogen also provides a polymer that is highly isotactic.^{32,33} AM and simple derivatives (NIPAM, DMAM) give polymers that are slightly syndiotactic (Table 4.1). Tacticity can be influenced by solvent and Lewis acids (Section 8.3).³⁴



An explanation for the preference for syndiotacticity during MMA polymerization was proposed by Tsuruta *et al.*³⁵ They considered that the propagating radical should exist in one of two conformations and showed, with models, that attack on the less hindered side of the preferred conformation (where steric interactions between the substituent groups are minimized) would lead to formation of a syndiotactic dyad while similar attack on the less stable conformation would lead to an isotactic dyad.

MMA polymerization is one of the most studied systems and was thought to be explicable, within experimental error, in terms of Bernoullian statistics. Moad *et al.*³⁶ have made precise measurements of the configurational sequence distribution for PMMA prepared from ¹³C-labeled monomer. It is clear that

Bernoullian statistics do not provide a satisfactory description of the tacticity.³⁶ This finding is supported by other work.^{28,37,38} First order Markov statistics provide an adequate fit of the data. Possible explanations include: (a) penpenultimate unit effects are important; and/or (b) conformational equilibrium is slow (Section 4.2.1). At this stage, the experimental data do not allow these possibilities to be distinguished.

It seems likely that other polymerizations will be found to depart from Bernoullian statistics as the precision of tacticity measurements improves. One study¹² indicated that vinyl chloride polymerizations are also more appropriately described by first order Markov statistics. However, there has been some reassignment of signals since that time.^{24,25}

The triad fractions for $PVAc^{22,39}$ seem to obey Bernoullian statistics. However, the concentrations of higher order *n*-ads cannot be explained even by first (or second) order Markov statistics suggesting either that ambiguities still remain in the signal assignments at this level or that there are unresolved complexities in the polymerization mechanism. Tacticities have been shown to be solvent and temperature dependent with the degree of syndiotacticity being significantly enhanced in fluoroalcohol solvents and by lower temperatures.^{40,41} Tacticity of vinyl esters is also dependent on the ester group.⁴²

| Monomer | Temp. | P(m) ^a | P(m m) | <i>P</i> (r m) | Solvent | Conv. | Ref. |
|---------|-------|------------------------|--------|----------------|------------------|-------|----------|
| | °C | | | | | % | |
| AN | 35 | 0.52 | - | - | H ₂ O | - | 43 |
| ΜΛ | 60 | 0.49 | - | - | toluene | <50 | 44.45 |
| AM | 0 | са 0.46 ^ь | - | - | methanol | 60 | 34 |
| DMAM | 60 | $ca 0.46^{b}$ | - | - | methanol | 73 | 34 |
| NIPAM | 60 | $ca 0.45^{b}$ | - | - | methanol | 82 | 34 |
| S | 80 | 0.46 | - | - | benzene | 92 | 46-48 |
| VAc | - | $0.46 \pm .01^{\circ}$ | _ | - | d | - | 22.49.50 |
| VC | 90 | (0.454) | 0.437 | 0.465 | е | - | 12 |
| VC | 5 | (0.406) | 0.391 | 0.424 | е | - | 12 |
| VC | -30 | (0.377) | 0.337 | 0.391 | d | - | 12 |
| MAN | 60 | 0.406 | - | - | bulk | 15 | 21 |
| 6 | 60 | $ca 0.14^{b}$ | - | - | methanol | 97 | 51 |
| 6 | 60 | $ca 0.28^{\rm b}$ | - | - | toluene | 95 | 51 |
| 7 | 60 | $< 0.1^{\circ}$ | - | - | methanol | 50 | 51 |
| ММА | 60 | (0.202) | 0.159 | 0.212 | benzene | 5 | 36 |

 Table 4.1 Tacticities of Selected Homopolymers

a Best fit number for P(m). The polymerization is believed to follow first order Markov statistics. b Bernoullian statistics not established. Values of P(m) estimated from triad distributions given. c Sec text. d Commercial samples or conditions of preparation unstated. e Suspension polymerization.

176 The Chemistry of Radical Polymerization

Further discussion on the effects of the reaction media and Lewis acids on tacticity appears in Section 7.2. Attempts to control tacticity by template polymerization and by enzyme mediated polymerization are described in Section 7.3. Devising effective means for achieving stereochemical control over propagation in radical polymerization remains an important challenge in the field.

4.3 Regiosequence Isomerism - Head vs Tail Addition

Most monomers have an asymmetric substitution pattern and the two ends of the double bond are distinct. For mono- and 1,1-disubstituted monomers (Section 4.3.1) it is usual to call the less substituted end "the tail" and the more substituted end "the head". Thus the terminology evolved for two modes of addition: head and tail; and for the three types of linkages: head-to-tail, head-to-head and tail-totail. For 1,2-di-, tri- and tetrasubstituted monomers definitions of head and tail are necessarily more arbitrary. The term "head" has been used for that end with the most substituents, the largest substituents or the best radical stabilizing substituent (Scheme 4.4).

With 1,3-diene based polymers, greater scope for structural variation is introduced because there are two double bonds to attack and the propagating species is a delocalized radical with several modes of addition possible (see 4.3.2).



Scheme 4.4

4.3.1 Monoene Polymers

Various terminologies for describing regiosequence isomerism have been proposed.^{1,4} By analogy with that used to describe stereosequence isomerism (Section 4.2), it has been suggested that a polymer chain with the monomer units connected by "normal" head-to-tail linkages should be termed isoregic, that with alternating head-to-head and tail-to-tail linkages, syndioregic, and that with a random arrangement of connections, aregic.¹

For mono- and 1,1-disubstituted monomers, steric, polar, resonance, and bondstrength terms (see Section 2.3) usually combine to favor a preponderance of tail addition; *i.e.* an almost completely isoregic structure. However, the occurrence of

head addition has been unambiguously demonstrated during many polymerizations. During the intramolecular steps of cyclopolymerization, 100% head addition may be obtained (Section 4.4.1).

The tendency for radicals to give tail addition means that a head-to-head linkage will, most likely, be followed by a tail-to-tail linkage (Scheme 4.5). Thus, head-to-head linkages formed by an "abnormal" addition reaction are chemically distinct from those formed in termination by combination of propagating radicals (Scheme 4.6).







In view of the potential problems associated with discriminating between the various types of head-to-head linkages, it is perhaps curious that, while much effort has been put into finding head-to-head linkages, relatively little attention has been paid to applying spectroscopic methods to detect tail-to-tail linkages where no such difficulty arises.

Even allowing for the above-mentioned complication, the number of head-tohead linkages is unlikely to equate exactly with the number of tail-to-tail linkages. The radicals formed by tail addition (T^{\bullet}) and those formed by head addition (H^{\bullet}) are likely to have different reactivities.

Consideration of data on the reactions for small radicals (Section 2.3) suggest that the primary alkyl radical (H^{\bullet}) is more likely to give head addition than the normal propagating species (T^{\bullet}) for three reasons:

(a) The propensity for head addition, which usually corresponds with attack at the more substituted end of the double bond, should decrease as the steric bulk of the attacking radical increases. Note that H• (a primary alkyl radical in the

case of mono- and 1,1-disubstituted monomers) will usually be less sterically bulky than T•.

- (b) Most common monomers have some dipolar character. H• and T• will usually be polarized similarly to the head and tail ends of the monomer respectively. This should favor T• adding tail and H• adding head.
- (c) The primary alkyl radical (H•) will be more reactive than T• with no α -substituent to stabilize or delocalize the free spin.

However, head addition is usually a very minor pathway and is difficult to determine experimentally. Analysis of the events which follow head addition presents an even more formidable problem. Therefore, there is little experimental data on polymers with which to test the above-mentioned hypothesis. Data for fluoro-olefins indicate that H• gives less head addition than T• (Section 4.3.1.3). No explanation for the observation was proposed.

The primary alkyl radical, H•, is anticipated to be more reactive and may show different specificity to the secondary or tertiary radical, T•. In VAe and VC polymerizations the radical H• appears more prone to undertake intermolecular (Sections 4.3.1.1 and 4.3.1.2) or intramolecular (4.4.3.2) atom transfer reactions.

4.3.1.1 Poly(vinyl acetate)

It is generally agreed that ca 1-2% of propagation steps during VAc polymerization involve head addition. There is some evidence that, depending on reaction conditions, a high proportion of the head-to-head linkages may appear at chain ends (Scheme 4.7) and that the number of head-to-head linkages may not equate with tail-to-tail linkages. The extent of head addition in VAc polymerization increases with the polymerization temperature.

The classic method for establishing the proportion of head addition occurring in VAc polymerization involves a two step process.⁵² The PVAc is converted to PVA by exhaustive hydrolysis and the number of 1,2-glycol units is determined by periodate cleavage.

The reliability of the chemical method has been assessed by Adelman and Ferguson.⁵³ They showed that, for low molecular weight PVA, a significant proportion of the 1,2-glycol units appear at chain ends as 2,3-dihydroxybutyl groups (*ca* 20% for $\overline{M}_n = 5,000$, PVAc prepared in methanol at 75 °C). The inference is that the radical formed by head addition is particularly active in interor intramolecular transfer and/or termination reactions. The result suggests that measurements of the decrease in molecular weight caused by periodate cleavage could underestimate the amount of head addition.⁵²

Analysis of ¹³C NMR spectra of PVA provides a direct estimate of the extent of head addition occurring in VAc polymerizations.^{39,54,55} Another advantage of the NMR method over chemical methods is that both head-to-head and tail-to-tail linkages can be observed. The polymers examined in these studies^{39,54} were of relatively high molecular weight and prepared by emulsion polymerization. They

possessed an equal number of head-to-head and tail-to-tail linkages. We have found that NMR can also be used to determine the fraction of head-to-head linkages in PVAc directly.

Scheme 4.7

The reaction conditions (solvent, temperature) may also influence the amount of head addition and determine whether the radical formed undergoes propagation or chain transfer.

4.3.1.2 Poly(vinyl chloride)

Establishment of the detailed microstructure of PVC has attracted considerable interest. This has been spurred by the desire to rationalize the poor thermal stability of the polymer (Chapter 1). Many reviews have appeared on the chemical microstructure of PVC and the mechanisms of "defect group" formation.⁵⁶⁻⁶⁰

Although head addition occurs during PVC polymerization to the extent of ca 1%, it is now thought that PVC contains few, if any, head-to-head linkages (<0.05%).^{61.62} Propagation from the radical formed by head addition is not competitive with a unimolecular pathway for its disappearance, namely, 1,2-chlorine atom transfer (see Scheme 4.8).

Rigo *et al.*⁶³ were the first to propose that head addition does occur but is immediately followed by a 1,2-chlorine atom shift. The viability of 1,2-chlorine atom shifts is well established in model studies and theoretical calculations.⁶⁴ Experimental support for this occurring during VC polymerization has been provided by NMR studies on reduced PVC.^{65,66} Starnes *et al.*⁶¹ proposed that head addition is followed by one or two 1,2-chlorine atom shifts to give chloromethyl or dichloroethyl branch structures respectively (Scheme 4.8). There also is kinetic data to support this hypothesis.



Scheme 4.8

Starnes *et al.*⁶⁷ have also suggested that the head adduct may undergo β -scission to eliminate a chlorine atom which in turn adds VC to initiate a new polymer chain. Kinetic data suggest that the chlorine atom does not have discrete existence. This addition-elimination process is proposed to be the principal mechanism for transfer to monomer during VC polymerization and it accounts for the reaction being much more important than in other polymerizations. The reaction gives rise to terminal chloroallyl and 1,2-dichloroethyl groups as shown in Scheme 4.8.

The presence of 1,2-dichloroethyl end groups and branch structures is likely to confuse attempts to determine head-to-head linkages by chemical methods (*e.g.* iodometric titration⁶⁸).

4.3.1.3 Fluoro-olefin polymers

Propagation reactions involving the fluoro-olefins, vinyl fluoride (VF),⁶⁹⁻⁷² vinylidene fluoride (VF2),^{69,72-74} and trifluoroethylene (VF3),⁷⁵ show relatively poor regiospecificity. This poor specificity is also seen in additions of small

radicals to the fluoro-olefins (see 2.3). Since the fluorine atom is small, the major factors affecting the regiospecificity of addition are anticipated to be polarity and bond strength.

The fraction of head-to-head linkages in the poly(fluoro-olefins) increases in the series $PVF2 < PVF \sim PVF3$ (Table 4.2). This can be rationalized in terms of the propensity of electrophilic radicals to add preferentially to the more electron rich end of monomers (*i.e.* that with the lowest number of fluorines). This trend is also seen in the reactions of trifluoromethyl radicals with the fluoro-olefins (see 2.3).

The proportion of head-to-head linkages in fluoro-olefin polymers also depends on the polymerization temperature 69,70,72,73 (Table 4.2).

| temperature | | % head addition | |
|-------------|-------------------|-------------------|------------------|
| °C | VF3 ⁷⁵ | VF2 ⁷³ | VF ⁷⁰ |
| 100 | | | 13.0 |
| 80 | 13.8 | 5.7 | 12.5 |
| 70 | | | 12.5 |
| 60 | | | 12.5 |
| 0 | 11.8 | 3.45 | - |
| -80 | 10.0 | 3.0 | - |

 Table 4.2 Temperature Dependence of Head vs Tail Addition for Fluoro-olefin

 Monomers

¹⁹F NMR studies have allowed regiosequence information to be determined at the pentad (VF) or heptad (VF2) level. Early studies⁷⁶ found that polymers formed by radical polymerization could be adequately described by Bernoullian statistics. However, Cais and Sloane⁷⁴ found that it was more appropriate to use first order Markov statistics to interpret regiospecificity. Their analysis suggests that the - CH₂• radical (formed by head addition) is much less likely to add head than the - CFX• radical [by a factor of ~14-18 for VF2 (depending on the polymerization temperature) or ~4 for VF]. No explanation for this selectivity was offered. The findings for fluoro-olefin propagation appear at variance with the considerations discussed above (see 4.3) and observations made for simple models. For example, with VF2, methyl radical is known to give much more head addition than trifluoromethyl radical (see 2.3).

4.3.1.4 Allyl polymers

Matsumoto *et al.*⁷⁷⁻⁸¹ have reported that substantial amounts (5-20%) of head addition occur during polymerization of allyl esters and that the proportion increases with polymerization temperature. They report that the proportion of

The Chemistry of Radical Polymerization

head-to-head linkages in poly(allyl esters) is also dependent on the molecular weight of the polymer chain. For short chains, the fraction is reported to be *ca* 10% irrespective of the nature of the ester group. For longer chains the proportion of head-to-head linkages decreases and the molecular weight dependence of this fraction increases according to the size/polarity of the ester group.

The very high levels of head addition and the substituent effects reported in these studies are inconsistent with expectations based on knowledge of the reactions of small radicals (see 2.3) and are at odds with structures formed in the intermolecular step of cyclopolymerization of diallyl monomers (see 4.4.1.1) where overwhelming tail addition is seen.

4.3.1.5 Acrylic polymers

Before the advent of NMR spectroscopy, a number of reports appeared suggesting the possibility of substantial head addition during polymerization of acrylate ester derivatives. Marvel *et al.*^{82,83} reported chemical degradation experiments that suggested that α -haloacrylate polymers contain halogen substituents in a 1,2-relationship. On this basis they proposed that these monomers polymerize in a head-to-head, tail-to-tail fashion. McCurdy and Laidler⁸⁴ suggested that irregularities in the heats of polymerization of methyl and higher acrylates and methacrylates could be rationalized if a fraction of units were arranged in head-to-head, tail-to-tail arrangement.

Since that time, many studies by NMR and other techniques on the microstructure of acrylic and methacrylic polymers formed by radical polymerization have proved their predominant head-to-tail structure.

There is, however, some evidence that a small amount of head addition during propagation occurs in the polymerization of acrylic monomers. On the basis of chemical analysis, Sawant and Morawetz⁸⁵ suggested that 4.6% of amide groups in PAM may be present as head-to-head linkages. Minigawa⁸⁶ has indicated the presence of a small percentage of head-to-head linkages in PAN.

4.3.2 Conjugated Diene Polymers

There is greater scope for structural variation in the diene based polymers than for the monoene polymers already discussed. The polymers contain units from overall 1,2- and *cis*- and *trans*-1,4-addition. Two mechanisms for overall 1,2addition may be proposed. These are illustrated in Scheme 4.9 and Scheme 4.10:

- (a) The delocalized allyl radical produced by addition to the 1- (or 4-) position may react in two ways to give overall 1,2-addition or 1,4-addition (Scheme 4.9).
- (b) By analogy with the chemistry seen with monoene monomers the propagating species could, in principle, add to one of the internal (2- or 3-) positions of the diene (Scheme 4.10).

Analyses of polymer microstructures do not allow these possibilities to be unambiguously distinguished. However, EPR experiments demonstrate that radicals add exclusively to one of the terminal methylenes.⁸⁷



Scheme 4.10

When used in conjunction with unsymmetrical dienes with substituents in the 2-position, the term 'tail addition' has been used to refer to addition to the methylene remote from the substituent. 'Head addition' then refers to addition to the methylene bearing the substituent (*i.e.* head addition \equiv 4,1- or 4,3-addition, tail addition \equiv 1,4- or 1,2-addition) as illustrated below for chloroprene (Scheme 4.11). Note that 1,2- and 4,3-addition give different structures while 1,4- and 4,1-addition give equivalent structures and a chain of two or more monomer units must be considered to distinguish between head and tail addition.

Tacticity is only a consideration for units formed by 1,2-addition. However, units formed by 1,4-addition may have a *cis*- or a *trans*-configuration.

In anionic and coordination polymerizations, reaction conditions can be chosen to yield polymers of specific microstructure. However, in radical polymerization while some sensitivity to reaction conditions has been reported, the product is typically a mixture of microstructures in which 1,4-addition is favored. Substitution at the 2-position (*e.g.* isoprene or chloroprene - Section 4.3.2.2) favors 1,4-addition and is attributed to the influence of steric factors. The reaction temperature does not affect the ratio of 1,2:1,4-addition but does influence the configuration of the double bond formed in 1,4-addition. Lower reaction temperatures favor *trans*-1,4-addition (Sections 4.3.2.1 and 4.3.2.2).

Early work on the microstructure of the diene polymers has been reviewed.¹ While polymerizations of a large number of 2-substituted and 2,3-disubstituted dienes have been reported,⁸⁸ little is known about the microstructure of diene polymers other than PB,⁸⁹ polyisoprene,⁹⁰ and polychloroprene.⁹¹



Scheme 4.11

4.3.2.1 Polybutadiene

The mechanism of B polymerization is summarized in Scheme 4.9. 1,2-, and *cis*- and *trans*-1,4-butadiene units may be discriminated by IR, Raman, or ¹H or ¹³C NMR spectroscopy.^{1,92-94} PB comprises predominantly 1,4-*trans*-units. A typical composition formed by radical polymerization is 57.3:23.7:19.0 for *trans*-1,4-:*cis*-1,4-:1,2-. While the ratio of 1,2- to 1,4-units shows only a small temperature dependence, the effect on the *cis*-*trans* ratio appears substantial. Sato *et al.*⁹³ have determined dyad sequences by solution ¹³C NMR and found that the distribution of isomeric structures and tacticity is adequately described by Bernoullian statistics. Kawahara *et al.*⁹⁴ determined the microstructure (ratio *trans*-1,4-:*cis*-1,4-:1,2- and dyad ratios) by performing ¹³C NMR measurements directly on PB latexes and obtained similar data to that obtained by solution ¹³C NMR. They⁹⁴ also characterized crosslinked PB.

4.3.2.2 Polychloroprene, polyisoprene

The mechanism of chloroprene polymerization is summarized in Scheme 4.11. Coleman *et al.*^{95,96} have applied ¹³C NMR in a detailed investigation of the microstructure of poly(chloroprene) also known as neoprene. They report a substantial dependence of the microstructure on temperature and perhaps on reaction conditions (Table 4.3). The polymer prepared at -150 °C essentially has a homogeneous 1,4-*trans*-microstructure. The polymerization is less specific at higher temperatures. Note that different polymerization conditions were employed as well as different temperatures and the influence of these has not been considered separately.

| temperature | unit | | | | | | |
|------------------|-----------|-------------------|-------------------|-------|-------|--|--|
| °C | 4,1-trans | 1,4- <i>trans</i> | 1,4 <i>-cis</i> a | 1,2-b | 4,3- | | |
| 90° | 75.1 | 10.3 | 7.8 | 2.9 | 4.1 | | |
| 40 ^d | 81.6 | 9.2 | 5.2 | 2.5 | 1.4 | | |
| 0ď | 90.4 | 5.5 | 1.8 | 2.1 | 1.1 | | |
| -40 ^d | 93.2 | 4.2 | 0.7 | 1.4 | 0.5 | | |
| -150e | 98.0 | 2.0 | < 0.2 | < 0.2 | < 0.2 | | |

Table 4.3 Microstructure of Poly(chloroprene) vs Temperature

a 1,4- and 4,1-*cis* not distinguished. b 25-50% of 1,2- are isomerized. c Reaction conditions not stated. d Emulsion polymer. e Polymer prepared by irradiation of crystalline monomer.

Poly(isoprene) can also be prepared by radical polymerization.⁹⁷ Although the ratio of 1,4-:1,2-:4,3- units is stated to be *ca* 90:5:5 irrespective of the polymerization temperature (range -20–50 °C), the proportion of *cis*-1,4-addition increases from 0 at -20 °C to 17.6% at 50 °C. EPR studies indicate that radicals add preferentially to the 1-position.⁸⁷

4.4 Structural Isomerism - Rearrangement

During most radical polymerizations, the basic carbon skeleton of the monomer unit is maintained intact. However, in some cases the initially formed radical may undergo intramolecular rearrangement leading to the incorporation of new structural units into the polymer chain. The rearrangement may take the form of ring closure (see 4.4.1), ring-opening (see 4.4.2) or intramolecular atom transfer (see 4.4.3).

The unimolecular rearrangement must compete with normal propagation. As a consequence, for systems where there is <100% rearrangement, the concentration of rearranged units in the polymer chain will be dependent on reaction conditions. The use of low monomer concentrations will favor the unimolecular process and it follows that the rearrangement process will become increasingly favored over normal propagation as polymerization proceeds and monomer is depleted (*i.e.* at high conversion). Higher reaction temperatures generally also favor rearrangement.

4.4.1 Cyclopolymerization

Diene monomers with suitably disposed double bonds may undergo intramolecular ring-closure in competition with propagation (Scheme 4.12). The term cyclopolymerization was coined to cover such systems. Many systems which give cyclopolymerization to the exclusion of "normal" propagation and crosslinking are now known. The subject is reviewed in a series of works by Butler.⁹⁸⁻¹⁰²



Scheme 4.12

Intramolecular cyclization is subject to the same factors as intermolecular addition (see 2.3). However, stereoelectronic factors achieve greater significance because the relative positions of the radical and double bond are constrained by being part of the one molecule (see 2.3.4) and can lead to head addition being the preferred pathway for the intramolecular step.

Geometric considerations in cyclopolymerization are optimal for 1,6-dienes (see 4.4.1.1). Instances of cyclopolymerization involving formation of larger rings have also been reported (see 4.4.1.4), as have examples where sequential intramolecular additions lead to bicyclic structures within the chain (see 4.4.1.2). Various 1,4- and 1,5-dienes are proposed to undergo cyclopolymerization by a mechanism involving two sequential intramolecular additions (see 4.4.1.3).

4.4.1.1 1,6-Dienes

The polymerization of nonconjugated diene monomers might be expected to afford polymer chains with pendant unsaturation and ultimately, on further reaction of these groups, crosslinked insoluble polymer networks. Thus, the finding by Butler *et al.*, ¹⁰³⁻¹⁰⁵ that polymerizations of diallylammonium salts, of general structure **8** [*e.g.* diallyldimethylammonium chloride (**9**)] gave linear saturated polymers, was initially considered surprising.



The explanation proposed involved sequential inter- and intramolecular addition steps. The presence of cyclic structures within the polymer chain was soon confirmed by degradation experiments.¹⁰⁶ However, these experiments did

not unambiguously define the precise nature of the cyclic units. Their nature was inferred on the basis of the then prevailing theory, that radical additions proceed so as to give the more stable product (a six-membered ring and a secondary radical). As a consequence, the structure of these cyclopolymers was not firmly established until the 1970s when spectroscopic studies showed that five-membered ring formation is the preferred (kinetic) pathway during cyclopolymerization of simple diallyl compounds (10).¹⁰⁷⁻¹¹²

Cyclopolymerizations of other 1,6-dienes afford varying ratios of five- and six-membered ring products depending on the substitution pattern of the starting diene. Substitution of the olefinic methine hydrogen (e.g. 11, $R-CH_3$) causes a shift from five- to six-membered ring formation. More bulky R substituents can prevent efficient cyclization and cross-linked polymers may result.



A vast range of symmetrical and unsymmetrical 1,6-diene monomers has now been prepared and polymerized and the generality of the process is well established.^{98,109} A summary of symmetrical 1,6-diene structures, known to give cyclopolymerization, is presented in Table 4.4 In many cases, the structure of the repeat units has not been rigorously established. Often the only direct evidence for cyclopolymerization is the solubility of the polymer or the absence of residual unsaturation. In these cases the proposed repeat unit structures are speculative.

The understanding of the mechanism of cyclopolymerization has been one of the initial driving forces responsible for studies on the factors controlling the mode of ring closure of 5-hexenyl radicals and other simple model compounds.¹¹³



Scheme 4.13

The preferential 1,5-ring closure of unsubstituted 5-hexenyl radicals has been attributed to various factors; these are discussed in greater detail in Section 2.3.4. The mode and rate of cyclization is strongly influenced by substituents. The results may be summarized as follows (Scheme 4.13):

(a) Methyl substitution at C-1 slows the rate of both 1,5 and 1,6-ring closure. Substituents which delocalize the spin into a π -system (CN, CO₂Me) may result in a predominance of six-membered ring products by rendering intramolecular addition readily reversible.

- (b) Substitution at C-5 dramatically retards 1,5-ring closure to the extent that 1,6-ring closure may predominate.
- (c) Substitution at C-6 retards 1,6-ring closure. If both the 5 and 6 positions are substituted 1,5-ring closure predominates.
- (d) Substitution at C-2, C-3, or C-4 facilitates both 1,5- and 1,6- ring closure.
- (e) Increased reaction temperatures favor 1,6-ring closure at the expense of 1,5-ring closure.

The presence of heteroatoms and the inclusion of sp^2 centers are also known to affect the rate and mode of cyclization.

Thus, on the basis of model studies, it is possible to reconcile the observation that diallyl monomers that are unsubstituted on the double bond (10, X–Z–CH₂, Y=CR₂, NR, O, *etc.*) give predominantly five-membered rings for the intramolecular step. Dimethallyl monomers and other similarly substituted monomers (11, R≠H) generally give predominantly six-membered rings (*e.g.* 11, X=Z=CH₂, R=CH₃ or CO₂R - Table 4.4). Dimethacrylic anhydride (11, Y=O, X=Z=C=O, R=CH₃) gives six-membered rings.¹¹⁴ It is surprising that dimethacrylic imides (11, Y=N-alkyl, X=Z=C=O, R= CH₃) are reported to give five-membered rings.^{114,115}



The observation by Matsumoto *et al.* (see 4.3.1.4) that significant amounts of head addition occur in polymerization of simple allyl monomers brings into question the origin of the small amounts of six-membered ring products that are formed in cyclopolymerization of simple diallyl monomers (Scheme 4.14). If the intermolecular addition step were to involve head addition, then the intramolecular step should give predominantly a six-membered ring product (14) (by analogy with chemistry seen for 1,7 dienes – see 4.4.1.4). Note that the repeat units 14 and 16, like 12 and 17 are the same; however, they are oriented differently in the chain.

188

If there is significant intermolecular head addition, the formation of sevenmembered units (13) might occur.¹¹⁶

The stereospecificity of the cyclization step has been examined both for model systems^{113,117} and in a few cyclopolymerizations.^{111,118,119} In formation of either five- or six-membered rings, there is a preference for the polymeric residues to end up *cis*- to each other. Note that for cyclopolymers with six-membered ring units, the ring stereochemistry is established in the intermolecular addition step (Scheme 4.15). In the case of five-membered ring units, ring stereochemistry is established during the intramolecular step (Scheme 4.16).



Scheme 4.15



Scheme 4.16

Unsymmetrical 1,6-dienes known to undergo cyclopolymerization include allyl (meth)acrylate (**18** X=II, CH₃; Y=H),¹²⁰ (**18** X=CH₃; Y=Ph)¹²¹ and (meth)acrylamide derivatives (**19** X=H, CH₃)^{120,122-125} and *o*-allyl (**20** X=H)¹²⁶ and *o*-isopropenylstyrene (**20** X=CH₃).¹²⁷ With these cyclopolymerizations initial addition is to the double bond with the α -phenyl or carbonyl group and residual double bonds are isopropenyl or allyl groups.^{124,125} For these examples, the cyclization step is relatively slow and reaction conditions are extremely important in obtaining soluble (uncrosslinked) polymers.


| Table 4.4 | Ring Sizes Formed in Cyclopolymerization of Symmetrical 1,6-Diene |
|-----------|---|
| | Monomers |

| | monomer | substituents | ring size ^a | refs. |
|----------------|------------------------------------|---|------------------------|-------------------|
| (a) all carbon | X V | X=Ph; Y=Z=H | 6 | 128,129 |
| skeleton | | X=CO ₂ H, CO ₂ R, CN; Y=Z=H | 6 | 130.131 |
| | | X=CO ₂ Me; Y=Z=CN | 6 | 118 |
| | γ∕~z | $X = CO_2R$; $Y = Z = CO_2R$ | 6 | 132 |
| (b) 4-nitrogen | XXX | X=H | 5 | 98,99.107,110.133 |
| (b) 4-muogen | l l | X=CH ₃ | 6 | |
| | | , , , , , , , , , , , , , , , , , , , | | |
| | | | | |
| | × .X × .X | X=H | 5 | 134 |
| | Y Y | X=CH ₂ | 5 | 114.115 |
| | - 大、大 | U U | | |
| | | | | |
| | Ý | | | |
| (c) 4-oxygen | ⇒× ⇒× | X=H | 5 | 80.136.127 |
| | | X=CH ₃ | 6 | 80,132-137 |
| | 0~0~0 | | | |
| | X | X=CO₀R | 6 | 138-140 |
| | T T | E | | |
| | | | | |
| | | V_V_ | 5-6 | 141.142 |
| (d) 3,5-oxygen | $\gamma^{\prime}\gamma^{\prime}$ | X=H: X=C_H_ | 5.0 | 142 |
| | 0, _0 | X=11, 1=0 ₃ 11 ₇ | | |
| | \uparrow | | | |
| ()] | Ŷ | | | 143-145 |
| (e) other | $\searrow^{X} \searrow^{X}$ | X=H | | 145 |
| neteroatom | | X=CH ₃ | | |
| substituents | Si | | | |
| | R´ `R' | | | |
| | | X=O, NH, NCH ₃ | 5+6+7? | 116 |
| | HaC-Si Si-CHa | | | |
| | H ₃ C × CH ₃ | | | |
| | | | 5-6 | 146 |
| | <u> </u> | | | |
| | 0,0 | | | |
| | сна сна | | | |
| | 33 | | | |

monomer substituents ring size^a refs. X=H X a (e) other X=CH₃ heteroatom Z substituents 148,149 X=H 148.150 X=CH₃ 151 X=H

a Predominant ring size. If not specified, it has not been unambiguously determined.

Propagation in cyclopolymerization may be substantially faster than for analogous monoene monomers.¹⁵² The various theories put forward to account for this observation are summarized in Butler's review.⁹⁸ A recent theoretical study by Tüzün *et al.*¹³³ looks at the effects of substituents on the rate of the cyclization step.

One contributing factor, which seems to have been largely ignored, is that the ring closed radical (in many cases a primary alkyl radical) is likely to be much more reactive towards double bonds than the allyl radical propagating species. This species will also have a different propensity for degradative chain transfer (a particular problem with allylamines and related monomers - see 6.2.6.4) and other processes which complicate polymerizations of the monoenes.

4.4.1.2 Triene monomers

Table 4.4 (continued)

Triallyl monomers [e.g. (21) or salts thereof] can potentially undergo two successive intramolecular cyclizations.^{153,154} However, in practice these materials give insoluble products.



A model study has demonstrated the pathways shown in Scheme 4.17. The first cyclization step gave predominantly five-membered rings, the second a mixture of six- and seven-membered rings.¹⁵⁵ Relative rate constants for the individual steps were measured. The first cyclization step was found to be some five-fold faster than for the parent 5-hexenyl system. Although originally put forward as evidence for hyperconjugation in 1,6-dienes, further work showed the rate acceleration to be steric in origin. 113,133



Scheme 4.17

The first cyclization gives a mixture of cis- and trans-isomers and only the cisisomer goes on to give bicyclic products. The relatively slow rate of the second cyclization step, and the formation of trans-product which does not cyclize, provides an explanation for the observation that radical polymerizations of triallyl monomers often give a crosslinked product.

4.4.1.3 1,4- and 1,5-dienes

Geometric considerations would seem to dictate that 1,4- and 1,5-dienes should not undergo cyclopolymerization readily. However, in the case of 1,4dienes, a 5-hexenyl system is formed after one propagation step. Cyclization via 1,5-backbiting generates a second 5-hexenyl system. Homopolymerization of divinyl ether (22) is thought to involve such a bicyclization. The polymer contains a mixture of structures including that formed by the pathway shown in Scheme 4.18.

It has been suggested that certain 1,5-dienes including o-divinylbenzene (23),¹⁵⁶ vinyl acrylate (24, X=H) and vinyl methacrylate (24, X=CH₃)¹²⁰ may also undergo cyclopolymerization with a monomer addition occurring prior to evelization and formation of a large ring. However, the structures of these cyclopolymers have not been rigorously established.



Scheme 4.18

Bicyclo[2,2,1]heptadiene derivatives (**25**) are set up to undergo ring closure to form a three-membered ring and it is proposed that polymers formed from (**25**) contain predominantly nortricyclene units.^{157,158}



4.4.1.4 1,7- and higher 1,n-dienes

Several polymerizations of 1,7- and higher diene monomers have been reported. Cyclization to large rings (> six-membered) has been postulated.¹⁵⁹⁻¹⁶⁴ However, in many examples, cyclization is not quantitative and crosslinked polymers are formed. Evidence for ring formation comes from kinetic data and, in particular, from the delay in the gel point from that expected (based on the assumptions that no cyclization occurs and that all pendant double bonds are available for crosslinking reactions). One common monomer that is thought to show such behavior is methylene-bis-acrylamide (ring structure not proven).^{159,160}

1,7-dienes give six-membered rings in preference to seven-membered rings; examples include ethylene glycol divinyl ether (26) and bis-acryloylhydrazine (27).¹⁶¹ This preference is also seen with model 6-heptenyl radicals.¹⁶⁵ One of the first reported examples of a 'cyclopolymerization' was that of the 1,11-diene, diallyl phthalate (28). A significant fraction (30-40%) of repeat units in the low conversion polymer was postulated to have a cyclic structure.^{162,163} NMR studies on polymers formed by exhaustive hydrolysis suggest the cyclopolymer contains eleven-membered rings.¹⁶⁴

Various dimethacrylates have been polymerized in an effort to synthesize a poly(methacrylate) with head-to-head linkages.^{114,115} Various 1,6- (*e.g.* dimethacrylamides - see Table 4.4), 1,7- (*e.g.* dimethacrylhydrazines) and 1,8- dienes (*e.g.* dimethacryloylureas) are reported to give head-to-head addition (five-,

six- or seven-membered rings respectively) or a mixture of head-to-head and head-to-tail addition. The 1,9-diene, *o*-dimethacryloylbenzene $(29)^{166}$ and the 1,10-diene 2,4-pentanediol dimethacrylate (30) give 100% cyclopolymerization and only head-to-tail addition (nine- and ten-membered rings respectively). Methacrylate derivatives of oligo- and polyhydroxy compounds analogous to 30 have been shown to undergo cyclopolymerization to give ladder polymers. These polymerizations are considered further in the section on template polymerization (see 8.3.5.2).



4.4.1.5 Cyclo-copolymerization

In this section we consider systems where the radical formed by propagation can cyclize to yield a new propagating radical. Certain 1,4-dienes undergo cyclo-copolymerization with suitable olefins. For example, divinyl ether and MAH are proposed to undergo alternating copolymerization as illustrated in Scheme 4.19.¹⁶⁷ These cyclo-copolymerizations can be quantitative only for the case of a strictly alternating copolymer. This can be achieved with certain electron donor-electron acceptor pairs, for example divinyl ether-maleic anhydride.



Scheme 4.19

4.4.2 Ring-Opening Polymerization

Much of the interest in ring-opening polymerizations stems from the fact that the polymers formed may have lower densities than the monomers from which they are derived (*i.e.* volume expansion may accompany polymerization).¹⁶⁸⁻¹⁷¹ This is in marked contrast with conventional polymerizations which typically involve a nett volume contraction. Such polymerizations are therefore of particular interest in adhesive, mold filling, and other applications where volume

contraction is undesirable. Their use in dental composite and adhesive compositions has attracted recent attention.¹⁷¹

Ring-opening polymerizations and copolymerizations also offer novel routes to polyesters and polyketones (Section 4.4.2.2). These polymers are not otherwise available by radical polymerization. Finally, ring-opening copolymerization can be used to give end functional polymers. For example, copolymerization of ketene acetals with, for example, S, and basic hydrolysis of the ester linkages in the resultant copolymer offers a route to α, ω -difunctional polymers (Section 7.5.4).



Scheme 4.20

Reviews on radical ring-opening polymerization include those by Sanda and Endo,¹⁷² Klemm and Schultz,¹⁷³ Cho,¹⁷⁴ Moszner *et al.*,¹⁷⁵ Endo and Yokozawa¹⁷⁶ Stansbury¹⁷⁰ and Bailey.¹⁷⁷ A review by Colombani¹⁷⁸ on addition-fragmentation processes is also relevant. Monomers used in ring-opening are typically vinyl (*e.g.* vinylcyclopropane - Scheme 4.20; Section 4.4.2.1) or methylene substituted cyclic compounds (*e.g.* ketene acetals - Section 4.4.2.2) where addition to the double bond is followed by β -scission.

However, there are also examples of addition across a strained carbon-carbon single bond, as occurs with bicyclobutane¹⁷⁹ and derivatives (Scheme 4.21, Scheme 4.22).^{180,181} Interestingly, I-cyano-2,2,4,4-tetramethylbicylobutane (**31**) is reported to provide a polyketenimine (Scheme 4.22).¹⁸² This is the only known examples of a α -cyanoalkyl radical adding monomer *via* nitrogen.



Scheme 4.22

For ring-opening to compete effectively with propagation, the former must be extremely facile. For example with $k_p \sim 10^2 \cdot 10^3 \text{ M}^{-1} \text{ s}^{-1}$ the rate constant for ring-opening (k_{β}) must be at least $\sim 10^5 \cdot 10^6 \text{ s}^{-1}$ to give >99% ring-opening in bulk

polymerization. The reaction conditions can be chosen so as to favor ringopening. Ring-opening will be favored by dilute reaction media and, usually, by higher polymerization temperatures.

The ring-opening reaction usually results in the formation of a new unsaturated linkage. When this is a carbon-carbon double bond, the further reaction of this group during polymerization leads to a crosslinked (and insoluble) structure and can be a serious problem when networks are undesirable. In many of the applications mentioned above, crosslinking is desirable.

4.4.2.1 Vinyl substituted cyclic compounds

There must be considerable driving force for ring-opening if it is to compete with propagation. In the case of vinylcyclopropane and derivatives (Scheme 4.20) this is provided by the relief of strain inherent in the three-membered ring. Rates of ring-opening of cyclopropylmethyl radicals are reported to be in the range 10^{5} - 10^{8} s⁻¹ depending on the substitution pattern.¹⁸³⁻¹⁸⁷



Many polymerizations of vinylcyclopropane and substituted derivatives (**32**) have now been reported.^{174,175,188-205} All examples give 100% opening of the cyclopropane ring. However, conversions and polymerization rates are often low, even when the double bond is activated towards addition by a phenyl substituent (**33**).^{205,206} For this example, the explanation for low polymerization rates probably lies with the reversibility of ring-opening. The reversibility of cyclopropylmethyl radical ring-opening has been established even for the parent system. The α -phenyl substituent reduces the rate of ring-opening by some two to three orders of magnitude^{185,207} and the equilibrium lies in favor of the ring-closed radical.²⁰⁷ Even though the rate constant for ring-opening is slow in the case of **33**, the monomer is unlikely to undergo polymerization without ring-opening. Such a polymerization should have a low ceiling temperature since **33** is structurally analogous to AMS (Section 44.5.1).

In the case of asymmetrically ring-substituted vinylcyclopropane derivatives (32, Y and/or $Z\neq H$), two pathways for ring-opening are available (Scheme 4.23).²⁰⁸ There have been a number of studies on substituent effects on ring-opening of cyclopropylmethyl radicals.¹⁸³⁻¹⁸⁷ Steric, polar and stereoelectronic factors are all important in determining the kinetics and preferred mode of ring-opening. Since this is a reversible process, the kinetic and thermodynamic products may be different.¹⁸⁷



Scheme 4.23

It has also been proposed that the ring-opened radicals may undergo ringclosure to a cyclobutane (Scheme 4.23).^{202,208} At this stage the only evidence for this pathway is observation of signals in the NMR spectrum of the polymer that cannot be rationalized in terms of the other structures. There is no precedent for 1,4-ring-closure of a 3-butenyl radical in small molecule chemistry and the result is contrary to expectation based on stereoelectronic requirements for intramolecular addition (Section 2.3.4). However, an alternate explanation has yet to be proposed. The possibility of carbonium ion intermediates should not be discounted.

1,2,2,3,3-Pentafluorovinylcyclopropane (**34**, Scheme 4.24) undergoes facile ring-opening polymerization exclusively as shown (*trans* double bond).²⁰⁹



The vinyloxirane (35, Scheme 4.25) undergoes ring-opening polymerization to give a polyether structure²¹⁰⁻²¹² with specific cleavage of the C-C bond. Other oxiranylmethyl radicals (without the phenyl substituent) are reported to give specific cleavage of the C-O bond.²¹³



Scheme 4.25

Rate constants for ring-opening of cyclobutylmethyl radicals²¹⁴ are less than those for the corresponding cyclopropylmethyl radicals by a factor of *ca* 10⁴.¹⁸³ This is consistent with the smaller degree of ring strain inherent in the fourmembered ring. Model studies have shown that *cis*- β -substituents on the cyclobutane ring lead to a markedly enhanced rate constant for ring-opening and a high specificity for cleavage of the more substituted bond.²¹⁴ The substituted vinylcyclobutane (**36**, stereochemistry unspecified) is reported to give >90% ringopening on polymerization in bulk at 60 °C and a single ring-opened product as shown in Scheme 4.26.²¹⁵



2-Phenyl-1-vinylcyclobutane (37) is also reported to give partial ringopening¹⁷⁴ while the vinylpropiolactones **38** and **39** give 100% ring-opening with loss of carbon dioxide.¹⁷⁴



For vinylcyclopentane (cyclopentylmethyl radical) and vinylcyclohexane (cyclohexylmethyl radical) derivatives, ring-opening is generally not a favorable process (Section 4.4.1). However, a number of ring-opening polymerizations involving five- or larger-membered rings have been reported where appropriate substitution is present to provide the driving force for the β -scission step. Examples are the vinylsulfones (40, n=0,1,2),²¹⁶⁻²¹⁸ which undergo ring-opening polymerization by scission of a relatively weak C-S bond and loss of sulfur

dioxide, and the spiro derivatives 41^{219} , 42^{220} and $43-44^{221}$ where ring-opening is facilitated by the concomitant aromatization of a cyclohexadiene derivative.



Polymerization of **42** gives between 43% (85 °C, bulk) and 98% (130 °C, bulk) ring-opening depending on reaction temperature.²²⁰ Near quantitative ring-opening has been obtained in the case of polymerizations of **43** and **44** where further driving force for ring-opening is provided by formation of a benzylic radical.²²¹ These monomers, **43** and **44**, also undergo ring-opening in copolymerization with S.

4.4.2.2 Methylene substituted cyclic compounds

The ring-opening polymerization of ketene acetals (45, X=O) provides a novel route to polyesters and many examples have now been reported (Scheme 4.27).²²² 227 A disadvantage of these systems is the marked acid sensitivity of the monomers which makes them relatively difficult to handle and complicates characterization. This area is covered by a series of reviews by Bailey *et al.*^{177,228-231}

The main driving force for ring-opening in polymerizations of these compounds is formation of a strong carbon-oxygen double bond. The nitrogen (45, X–N-CH₃, n–0) and sulfur (45, X–S, n–0) analogs undergo ring-opening polymerization (Table 4.5) with selective cleavage of the C-O bond to give polyamides or polythioesters respectively (Scheme 4.27). The specificity is most likely a reflection of the greater bond strength of C–O vs the C–S or C–N double bonds. The corresponding dithianes do not give ring-opening even though this would involve cleavage of a weaker C-S bond.^{232,233}



Scheme 4.27

The competition between ring-opening and propagation is dependent on ring size and substitution pattern. For the five-membered ring ketene acetal (45, X=O, n=0) ring-opening is not complete except at very high temperatures. However, with the larger-ring system (45, X=O, n=2) ring-opening is quantitative. This observation (for the n=2 system) was originally attributed to greater ring strain.

200

However, it may also reflect the greater ease with which the larger ring systems can accommodate the stereoelectronic requirements for β -scission (Section 2.3.4).¹¹³ Substituents (*e.g.* CH₃, Ph) which lend stabilization to the new radical center, or increase strain in the breaking bond, also favor ring-opening (Table 4.5).

| monomer | % ring-opening | conditions | ref. ^a |
|---------------------|-------------------------------------|---|-------------------|
| Ì}-0 | 100 | 160 °C, bulk, <i>t</i> Bu ₂ O ₂ | |
| ó. > | 87 | 120 °C | |
| | 50 | 60 °C | 224 |
| | bond A 61 ^b bond B 27 | 110 °C, bulk, tBu_2O_2 | 234 |
| \$-0, | bond A 100 ^c | 120 °C, bulk, <i>t</i> Bu ₂ O ₂ | 224.235 |
| °, ∕ Ph | | 30 °C hv | 236 |
| O A | bond B 100 | 120 °C, bulk, tBu_2O_2 | 237 |
| CH3 | 100 | 120 °C, bulk, tBu ₂ O ₂ | |
| CH ₃ | 100 | 120 °C bulk dbu.O. | 225 |
| 0 Ph | 100 | 120° C, buik, $15u_2O_2$ | |
| | 100 | 65-125 °C, benzene | 226 |
| <u> </u> | | various initiators | |
| | <100 | 120 °C, bulk, tBu_2O_2 | |
| | 100 | 120 °C, bulk, IBu_2O_2 | |
| | <100 | 120 °C, bulk, <i>t</i> Bu ₂ O ₂ | 237 |
| $\langle 0 \rangle$ | 100 | 120 °C, bulk, <i>t</i> Bu ₂ O ₂ | 238.239 |
| $\sqrt{2}$ | 100 | 120 °C, bulk, tBu_2O_2 | 240 |
| 0 | | | |

Table 4.5 Extent of Ring-opening During Polymerizations of
2-Methylene-1,3-dioxolane and Related Species

Table 4.5 (continued)



a Where no reference is given, the examples are taken from Bailey's review.¹⁷⁷ b Data for R=n-decyl. Specificity dependent on R, temperature, and monomer concentration. c Racemization accompanies polymerization of optically active monomer.²²⁴





The diene shown in Scheme 4.28 is also reported to give 100% ringopening.²²⁷ However, polymerization had to be carried out in very dilute solution to give a soluble (not crosslinked) product.

Rate constants for ring-opening of dioxolan-2-yl radicals have been measured by Barclay *et al.*²⁴³ as 10^3 - 10^4 s⁻¹ at 75 °C (Scheme 4.29). There is also evidence that ring-opening is reversible.^{243,244} Thus, isomerization of the initially formed product to one more thermodynamically favored is possible if propagation is slow.



Scheme 4.29

Bailey *et al.*^{177,245} observed that ring-opening polymerization of the monomers (**39**) and (**40**), which can potentially give rise to the same ring-opened radical, give different polymers. That formed from (**39**) has pendant vinyl groups, while that from (**40**) has in-chain double bonds. They proposed that, in radical polymerization of ketene acetals, ring-opening might be concerted with addition of the next monomer unit and various experiments were suggested to test the hypothesis.¹⁷⁷ One of these was carried out by Acar *et al.*,²²⁴ who showed that ring-

opening polymerization of optically active 4-phenyl-1,3-dioxolane was accompanied by racemization. This is evidence against concerted ring-opening.

It was proposed¹⁷⁷ that radical addition to **46** or **48** should occur exclusively at the respective methylene group to generate radicals **47** (Scheme 4.30).





If, however, radicals add preferentially to the vinyl group of **48**, ring-opening polymerization would give the polymer with in-chain double bonds specifically *via* resonance structure **49** (Scheme 4.31). Thus, the two pathways are readily distinguishable. No other ring-opening polymerizations of vinyl dioxolane derivatives appear to have been reported to date.



Scheme 4.31

4-Methylenedioxolane derivatives also undergo ring-opening. However, the ring-opened radical may undergo a further β -scission (*e.g.* **50**, Scheme 4.32).^{223,246-252} The extent of the second β -scission step depends on the nature of substituents at the 2-position and the reaction conditions (Table 4.6).



Scheme 4.32

Of the 4-methylene-1,3-dioxolanes reported thus far (Table 4.6), only the 2,2diphenyl derivative (**50**) is reported to give the polyketone quantitatively (Scheme 4.32). This requires temperatures in excess of 120 °C in bulk polymerization.^{246,247} The 2-phenyl-2-alkyl derivatives give <100% ring-opening but still give 100% elimination of the ring-opened product at 120 °C.²²³ The 2-phenyl derivative is

reported to afford ring-opening without elimination of benzaldehyde at temperatures less than 30 °C (photochemical initiation).²⁴⁹ At higher temperatures terpolymers are formed that comprise units that are non-ring-opened, ring-opened, and ring-opened with β -scission.

| Table 4.04-Methylene | 6 Extent of Ring-Ope -1,3-dioxolane and 2- | ning During Poly Methylene-1,4-di | merizations of oxane Derivatives | 8 |
|----------------------|---|--------------------------------------|----------------------------------|-------|
| monomer | % ring-opening | % elimination | conditions. | ref.a |
| ` | | | | |

| | - i - U | | | |
|-------------------|-----------|---------|--------------------------------|----------------|
| <u>}</u> | 30 | 100 | 130 °C, bulk | 177 |
| Ph Ph | 73 100 | 36 0 | 120 °C, bulk <30 °C, hv | 177.250 249 |
| CH ₃ | 23 | 100 | 120 °C, bulk | b,223 |
| | 18 | 100 | 60 °C, bulk | 248 |
| O Ph | 100 | 100 | 120 °C, bulk | 246.247 |
| | 10 | 0 | 0-120 °C, bulk 140 °C, bulk | 177 253 |
| 0 O O Ph | 40 | 0 | 140 °C, bulk | |
| | 20 | 0 | 80 °C, benzene | |
| | 40 | 0 | 80 °C, benzene | |
| O Ph | 100 | 0 | 80 °C, benzene | |

a Where no reference is given, the examples are taken from Bailey's review.¹⁷⁷ b Other 2-phenyl-2-alkyl derivatives are also reported to give <100% ring-opening and 100% elimination at 120 °C.

The structurally analogous five-membered ring α -alkoxyacrylates (Scheme 4.33) are slow to ring-open and do not undergo β -scission to form an acyl radical propagating species.^{177,253-255} This latter observation is probably a reflection of a higher bond strength for the bond α - to the carbonyl group. More ring-opening is observed for six-membered ring systems (Table 4.6).

Scheme 4.33

 Table 4.7 Extent of Ring-Opening During Polymerizations of
 2-Methylenetetrahydrofuran and Related Compounds

| monomer | % ring-opening | conditions | ref.a |
|---------|----------------|---|-------|
| | 40 | 120 °C, bulk, tBu_2O_2 | 242 |
| Z | 5 | 120 °C, bulk, tBu_2O_2 | 256 |
| сн3 | 15-20 | 120 °C, bulk, tBu_2O_2 | |
| | 0 | 120 °C, bulk, <i>t</i> Bu ₂ O ₂ | |
| Ph | 50 | 120 °C, bulk, <i>1</i> Bu ₂ O ₂ | 256 |
| ₹°) | 4-8 | 120 °C, bulk, <i>t</i> Bu ₂ O ₂ | |

^a Where no reference is given, the examples are taken from Bailey's review.¹

Monomers with only a single ring oxygen-atom give less facile ring-opening. For example, the 2-methylenetetrahydrofuran derivatives give substantially less ring-opening than the corresponding 2- or 4-methylene-1,3-dioxolanes (Table 4.7).

Seven- and eight-membered ring cyclic allyl sulfide derivatives (51, 52, 54-56) are stable in storage and handling and do not show the acid sensitivity of the cyclic acetal monomers above. They undergo facile ring-opening polymerization even at relatively low temperatures²⁵⁷⁻²⁶⁰ with quantitative ring-opening (Scheme 4.34, Scheme 4.35). The monomers also undergo facile ring-opening copolymerization with MMA and S.²⁶¹ The corresponding six-membered ring compound (53) appears unreactive in homopolymerization.

Ring-opening provides a thiyl radical propagating species. Although the polymers have a double bond on the backbone there is little or no crosslinking (Scheme 4.34, Scheme 4.35). There is, however, evidence of reversible addition

and addition-fragmentation involving this double bond.²⁶² Monomers containing multiple double bonds have been designed to provide ring-opening polymerization with crosslinking.²⁵⁹



4.4.2.3 Double ring-opening polymerization

While many factors affect the degree of volume change which accompanies polymerization, any volume increase is directly related to the number of rings opened in the propagation step and is inversely related to the size of the rings being broken. Consideration of these factors leads to the conclusion that appreciable volume expansion on polymerization should only be expected when two or more rings are opened¹⁷⁰ and substantial effort has been put into designing systems where two or more rings are opened on polymerization.

It should also be noted that for many of the applications where volume expansion is required (adhesives, composites, *etc.*) a crosslinked product is desirable and some monomers have been designed with this in mind. This does, however, make the products difficult to characterize. Some monomers with potential for double ring-opening are reported in Table 4.8.



Scheme 4.36

Various methylene derivatives of spiroorthoearbonates and spiroorthoesters have been reported to give double ring-opening polymerization (*e.g.* Scheme 4.36). Like the parent monocyclic systems, these monomers can be sluggish to polymerize and reactivity ratios are such that they do not undergo ready copolymerization with acrylic and styrenic monomers. Copolymerizations with VAc have been reported.¹⁷⁰ These monomers, like other acetals, show marked acid sensitivity.

The vinylcyclopropane derivatives substituted with a five- or six-membered acetal ring give single ring-opening with differing regiospecificity (Scheme $4.37^{202,203}$ and Scheme $4.38^{202,203,263}$).



Scheme 4.37 (double bond stereochemistry not specified)



Scheme 4.38 (double bond stereochemistry not specified)

Systems with substituents on the acetal ring²⁶⁴ or with larger acetal rings may give double ring-opening (*e.g.* Scheme 4.39).^{202,203}



Scheme 4.39 (double bond stereochemistry not specified)

| Monomer | % ring-opening | conditions | ref. ^a |
|----------------------|-----------------|--|-------------------|
| spiroorthocarbonates | | | |
| | 100 | 130 °C, bulk, 30% conv. ^b | |
| | 5-100 | 130 °C, PhCl, <50% conv. ^b | 170,206.265 |
| TX T | 0 | 165 °C, PhCl | 266 |
| spiroorthoesters | | | |
| | 10° | 120 °C, bulk, /Bu ₂ O ₂ | 267.268 |
| | 10 ^d | 120 °C, bulk, <i>t</i> Bu ₂ O ₂ | 269 |
| | 100 | 120 °C, bulk, tBu ₂ O ₂ | |
| | | | |
| other systems | | | 070 |
| | high | 100 °C, AIBN | 270 |
| | 100 | 130 °C, bulk, /Bu ₂ O ₂ | 271 |
| vinylcyclopropanes | | | 202 202 |
| \sim | 0° | 60 °C, bulk, AIBN | 202,203 |
| | 0e | 60 °C, bulk, AIBN | 202,203.263 |
| \sim | 46 | 60 °C, bulk, AIBN | 202,203 |

 Table 4.8 Extent of Double Ring-Opening During Polymerization of Polycyclic Monomers

a Where no reference is given, the examples are taken from Bailey's review.¹⁷⁷ b Insoluble and presumably crosslinked polymer formed at higher conversions. c 50:50 mixture single and double ring-opened products. d \geq 50% double ring-opened product. e Single ring-opened product only.

Analogous systems with six-, seven-, or eight-membered spirodithioacetal rings are reported to give single ring-opened products with no olefinic residues. A mechanism involving consecutive cyclopropane ring-opening and cyclization was proposed to rationalize this result.²⁷²

The spiro monomers 57-59 are reported to give single ring-opening.²⁷³⁻²⁷⁵ Solution polymerization of 57^{273} and 58 (R⁻CH₃, R'-C₃H₇)²⁷⁴ provided soluble products.



4.4.3 Intramolecular Atom Transfer

It has been known for some time that intramolecular atom transfer, or backbiting, complicates polymerizations of E (Scheme 4.40 - Section 4.4.3.1), VAc and VC (see 4.4.3.2). Recent work has shown that backbiting is also prevalent in polymerization of acrylate esters (Section 4.4.3.3) and probably occurs to some extent during polymerizations of most monosubstituted monomers.^{276,277}

Viswanadhan and Mattice²⁷⁸ carried out calculations aimed at rationalizing the relative frequency of backbiting in these and other polymerizations in terms of the ease of adopting the required conformation for intramolecular abstraction (see 2.4.4). More recent theoretical studies generally support these conclusions and provide more quantitative estimates of the Arrhenius parameters for the process.^{279,280}

Cases of "addition-abstraction" polymerization have also been reported where propagation occurs by a mechanism involving sequential addition and intramolecular 1,5-hydrogen atom transfer steps (Section 4.4.3.4).



Scheme 4.40

4.4.3.1 Polyethylene and copolymers

The extent of short-chain branching in PE may be quantitatively determined by a variety of techniques including IR,^{281,282} pyrolysis-GC,²⁸³ and γ -radiolysis.²⁸⁴ The most definitive information comes from ¹³C NMR studies.²⁸⁵⁻²⁹⁰ The typical

concentration of branch points in PE formed by radical polymerization is 8-25 per 1000 CH₂.²⁸⁷ These are made up of: ethyl, 1.2-11.3; butyl, 3.9-8.5; pentyl (amyl), 0.6-2.2; hexyl and longer, 0.5-2.8. The range of values for extent and type of short-chain branches arises because the branching process is extremely dependent on the polymerization conditions.²⁸⁷ High reaction temperatures and low pressures (monomer concentrations) favor the backbiting process.

The backbiting reaction first proposed by Roedel²⁹¹ (Scheme 4.40) is generally accepted as the mechanism for short chain branch formation during polymerization of E (for discussion on alternative mechanisms see^{292.293}). The preferential formation of butyl [*vs* propyl, pentyl, or longer branches] branches can be rationalized in terms of the stereoelectronic requirements imposed on the transition state (Section 2.4.4). The preferred coplanar arrangement of atoms is most readily achieved in a six-membered chair-like transition state.²⁹⁴ 1-Undecyl radicals are a simple model of the PE propagating species and give 1,5- and 1,6-H transfer in the ratio 3:1. Other intramolecular H transfers were not detected.²⁹⁵ Theoretical studies provide a picture of the transition state and a reasonable estimate of the Arrhenius parameters for backbiting.²⁷⁹ Both enthalpic and entropic factors favor 1,5-H transfer.

Direct formation of an ethyl branch would require backbiting *via* a highly strained four-membered transition state and, therefore, should have a low probability.^{294,296} The relatively large numbers of ethyl branches in PE is accounted for by the occurrence of two successive 1,5-H transfers which leads to either a pair of ethyl branches (Scheme 4.41) or a 2-ethylhexyl branch depending on the site of abstraction.²⁹⁷ This mechanism for ethyl branch formation requires that the radical formed by backbiting (secondary alkyl) should be substantially more prone to undertake backbiting than the normal propagating species (primary alkyl). This suggests that the former has a reduced rate of propagation (more sterically hindered radical) and/or an increased rate of intramolecular abstraction (Thorpe-Ingold effect).

Backbiting also occurs in ethylene copolymerizations with AN,²⁹⁸ (meth)acrylate esters²⁹⁰ and VAc. ^{280,290,299,300} The structures identified in E-BA copolymerization include **60-63** (X=BA). Structure **60** is formed when the BA terminated chain backbites. Structure **61** is formed when backbiting occurs across a BA unit. Structure **62** and **63** are from backbiting to a BA unit (**63** is from double backbiting). The concentration of comonomer is such that there are few comonomer sequences.

The incidence of the various structures depends strongly on the comonomer. In copolymerization with acrylates structures **62** and **63** dominate. In copolymerization with VAc structure **61** dominates and **62** and **63** are not observed. Structure **60** may be present in VAc copolymers to a very small extent but is not observed in acrylate copolymerizations. Structures **62** and **63** are not observed and cannot be formed in methacrylate copolymerizations.²⁹⁰ The results were interpreted²⁹⁰ in terms of the PVAc• propagating radical having a lesser

propensity for backbiting. This seems inconsistent with the observation of the products of backbiting during VAc homopolymerization (Section 4.4.3.2).³⁰¹ The data might also be rationalized in terms of the influence of polar and enthalpic factors on the facility of the various abstraction reactions (Section 2.4).



Scheme 4.41

Table 4.9 Structures Formed by Backbiting in Ethylene Copolymerizations^a

| | H K 60 | H X 61 | Market Karley Ma | ~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~~ |
|-------|--------------|--------------|--|--|
| E/BA | 0 | + | +++ | +++ |
| E/AA | | | 0 | 0 |
| E/VAc | | • + | 0 | 0 |
| E/BMA | 0 | I. | ٥ | 0 |
| E/MAA | U | + | U | U |

a Legend: --+ prevalent - weak 0 absent.

4.4.3.2 Vinyl polymers

There is evidence for backbiting during the polymerizations of VC 67,302 and VAc. $^{55,301,303-307}$ The mechanism is believed to be analogous to that discussed for PE above and should lead to the formation of 2,4-dichlorobutyl or 2,4 diacetoxybutyl branches (Scheme 4.42) respectively.



Scheme 4.42

The process is favored by low monomer concentrations as occurs at high conversions and in starved feed polymerizations.³⁰⁷ Theoretical calculations suggest that the incidence of backbiting should be strongly dependent on the tacticity of the penultimate dyad.³⁰⁸ Double backbiting in VC or VAc polymerization will lead to 2-chloroethyl or 2-acetoxy ethyl branches respectively (as for E in Scheme 4.41).³⁰²

There are no proven examples of 1,2-hydrogen atom shifts; this can be understood in terms of the stereoelectronic requirements on the process. The same limitations are not imposed on heavier atoms (*e.g.* chlorine). The postulate³⁰⁹ that ethyl branches in reduced PVC are all derived from chloroethyl branches formed by sequential 1,5-intramolecular hydrogen atom transfers as described for PE (Section 4.4.3.1) has been questioned.^{56,65} It has been shown that many of these ethyl branches are derived from dichloroethyl groups. The latter are formed by sequential 1,2-chlorine atom shifts which follow a head addition (Section 4.3.1.2).

4.4.3.3 Acrylate esters and other monosubstituted monomers

Recent work has shown that backbiting is prevalent in polymerizations and copolymerizations of acrylate esters.^{276,277,305,306,310-319} It is also observed in styrene polymerization at high temperature²⁷⁶ and probably occurs to some extent during polymerizations of most monosubstituted monomers. At high temperatures, and at low temperatures in very dilute solution, backbiting may be followed by fragmentation (Scheme 4.43).^{276,277,310-312,318} At lower temperatures short chain branch formation dominates.³¹³⁻³¹⁶ The backbiting process complicates the measurement of propagation rate constants for acrylates.³²⁰



The high temperature polymerization of acrylates with the backbiting-fragmentation process has been used to synthesize macromonomers based on acrylate esters.^{276,277,312} Interestingly, fragmentation shows a strong preference for giving the polymeric macromonomer **64** and a small radical **65**.^{276,277} An explanation for this specificity has yet to be proposed.

4.4.3.4 Addition-abstraction polymerization

Several examples of addition-abstraction polymerization have been reported. In these polymerizations, the monomers are designed to give quantitative rearrangement of the initially formed adduct *via* 1,5-hydrogen atom transfer (Scheme 4.44). The monomers (66) are such that the double bond is electron rich (vinyl ether) and the site for 1,5-H transfer is electron deficient. This arrangement favors intramolecular abstraction over addition. Thus compound 66a undergoes^{321,322} quantitative rearrangement during homopolymerization. For 66b, where the site of intramolecular attack is less electron deficient, up to 80% of propagation steps involve intramolecular abstraction. As expected, higher reaction temperatures and lower monomer concentrations favor the intramolecular abstraction pathway.



66 a R^1 =CN, R^2 =CO₂Et **b** R^1 = R^2 =CO₂Et

Scheme 4.44

4.5 **Propagation Kinetics and Thermodynamics**

In this section, we consider the kinetics of propagation and the features of the propagating radical $(P_n \cdot)$ and the monomer (M) structure that render the monomer polymerizable by radical homopolymerization (Section 4.5.1). The reactivities of monomers towards initiator-derived species (Section 3.3) and in copolymerization (Chapter 6) are considered elsewhere.

| | | | | Scheme 4.4 | 45 |
|------------------|---|---|---------------|------------------|---|
| $P_n \bullet$ | ÷ | Μ | \rightarrow | P_{n+1} • | $k_{p}(n)[P_{n}\bullet][M]$ |
| P_3 • | ÷ | М | \rightarrow | P₄• | $k_{p}(3)[P_{n}\bullet][M]$ |
| P_2^{\bullet} | ÷ | Μ | \rightarrow | P ₃ • | $k_{p}(2)[P_{n}\bullet][M]$ |
| ₽ ₁ • | ÷ | Μ | \rightarrow | P₂• | $k_{p}(1)[\mathbf{P}_{n}\bullet][\mathbf{M}]$ |

In the literature on radical polymerization, the rate constant for propagation, k_p , is often taken to have a single value (*i.e.* $k_p(1) - k_p(2) - k_p(3) - k_p(n)$ - refer Scheme 4.45). However, there is now good evidence that the value of k_p is dependent on chain length, at least for the first few propagation steps (Section 4.5.1), and on the reaction conditions (Section 8.3).

4.5.1 Polymerization Thermodynamics

Polymerization thermodynamics has been reviewed by Allen and Patrick,³²³ Ivin,³²⁴ Ivin and Busfield,³²⁵ Sawada³²⁶ and Busfield.³²⁷ In most radical polymerizations, the propagation steps are facile (k_p typically > 10² M⁻¹ s⁻¹ - Section 4.5.2) and highly exothermic. Heats of polymerization (ΔH_p) for addition polymerizations may be measured by analyzing the equilibrium between monomer and polymer or from calorimetric data using standard thermochemical techniques. Data for polymerization of some common monomers are collected in Table 4.10. Entropy of polymerization (ΔS_p) data are more scarce. The scatter in experimental numbers for ΔH_p obtained by different methods appears quite large and direct comparisons are often complicated by effects of the physical state of the monomer and polymers (*i.e.* whether for solid, liquid or solution, degree of crystallinity of the polymer).

The addition of radicals and, in particular, propagating radicals, to unsaturated systems is potentially a reversible process (Scheme 4.46). Depropagation is entropically favored and the extent therefore increases with increasing temperature (Figure 4.4). The temperature at which the rate of propagation and depropagation become equal is known as the ceiling temperature (T_c). Above T_c there will be net depolymerization.

$$P_n \bullet - M \xrightarrow{K_{eq}} P_{n+1} \bullet$$

Scheme 4.46

With most common monomers, the rate of the reverse reaction (depropagation) is negligible at typical polymerization temperatures. However, monomers with alkyl groups in the α -position have lower ceiling temperatures than monosubstituted monomers (Table 4.10). For MMA at temperatures <100 °C, the value of K_{eq} is <0.01 (Figure 4.4). AMS has a ceiling temperature of <30 °C and is not readily polymerizable by radical methods. This monomer can, however, be copolymerized successfully (Section 7.3.1.4).

The value of T_c and the propagation/depropagation equilibrium constant (K_{eq}) can be measured directly by studying the equilibrium between monomer and polymer or they can be calculated at various temperatures given values of $\Delta H_{\rm p}$ and $\Delta S_{\rm p}$ using eq. 11 and 12 respectively.

$$K_{\rm eq} = \exp\left(\frac{\Delta H_{\rm p}}{RT} - \frac{\Delta S_{\rm p}}{R}\right) = \frac{1}{[M]_{\rm eq}}$$
(11)

where $[M]_{eq}$ is the equilibrium monomer concentration.

$$T_{\rm c} = \frac{\Delta H_{\rm p}}{\Delta S_{\rm p} + R \ln[{\rm M}]} \tag{12}$$



Figure 4.4 Dependence of K_{eq} on temperature for selected monomers. Based on values of ΔH_p and ΔS_p shown in Table 4.10.

| monomer | Х | R | Δι | H _p (kJ mol | 1) | ΔS_n^c | T_c d |
|--------------------|---|-------------------------------|------------------|------------------------|-----------------------|--|---------|
| | | | а | b | с | $\mathbf{J} \operatorname{mol}_{t}^{F} \mathbf{K}^{*}$ | °C |
| ΛΛ | CO ₂ H | Н | 67 | - | - | - | - |
| MAA | CO ₂ H | CH3 | 43 | 65 | - | - | - |
| MA | CO ₂ CH ₃ | Н | 78 | - | - | - | - |
| MMA | CO ₂ CH ₃ | CH ₃ | 56 (58) | 55 | 56 ^{328,329} | $118^{328.329}$ | 202 |
| EMA | CO ₂ C ₂ H ₅ | CH ₃ | 60 (58) | - | 60^{330} | 124^{330} | 211 |
| BMA | CO ₂ C₄H ₉ | CH ₃ | 58 (60) | - | - | - | - |
| MEA ³³¹ | CO ₂ CH ₃ | C ₂ H ₅ | 32e | - | - | - | 22 |
| ΛN | CN | H | 75 ^{1°} | - | - | 109327 | 415 |
| ΜΛΝ | CN | CH_3 | 57 | - | 64 ³³² | $142^{g_{1}^{332}}$ | 177 |
| S | Ph | н | 69 (73) | 70 | 73 ³³³ | 104^{333} | 428 |
| AMS | Ph | CH_3 | - | 35 | 45 ³³⁴ | 148^{334} | 31 |
| VAc | O ₂ CCH ₃ | н | 88 (90) | - | - | - | - |
| VC | cī | н | 96 | 112 | - | - | - |

 Table 4.10 Thermodynamic Parameters for Polymerization of Selected Monomers (CH2=CRX)

a From calorimetry - data are for liquid monomer to amorphous solid polymer or for liquid monomer to polymer in monomer (in parentheses) and are taken from the Polymer Handbook unless otherwise indicated.³²⁷ All data are rounded to the nearest whole number. b From heat of combustion monomer and polymer - data are for liquid monomer to amorphous solid polymer and are taken from the Polymer Handbook.³²⁷ All data are rounded to the nearest whole number. c From studies of monomer-polymer equilibria - data are for liquid monomer to amorphous solid polymer. All data are rounded to the nearest whole number. d Calculated from numbers of ΔH_p (column c except for AN) and ΔS_p shown and [M] = 1.0. e Based on a measured T_c of 82 °C in bulk monomer and an assumed value for ΔS_p of 105 J mol⁻¹ K^{-1,331} A more reasonable value of ΔS_p of 120 J mol⁻¹ K⁻¹ would suggest a ΔH_p of 40 kJ mol⁻¹. f Partially crystalline polymer. g In benzonitrile solution.

Note that the value of T_c is dependent on the monomer concentration. In the literature, values of T_c may be quoted for [M] = 1.0 M, for $[M] = [M]_{eq}$ or for bulk monomer. Thus care must be taken to note the monomer concentration when comparing values of T_c . One problem with using the above method to calculate K_{eq} or T_c , is the paucity of data on ΔS_p . A further complication is that literature values of ΔH_p show variation of +2 kJ mol⁻¹ which may in part reflect medium effects.³²⁷ This "error" in ΔH_p corresponds to a significant uncertainty in T_c .

Steric factors appear to be dominant in determining ΔH_p and ΔS_p . The resonance energy lost in converting monomer to polymer is of secondary importance for most common monomers. It is thought to account for ΔH_p for VAc and VC being lower than for acrylic and styrenic monomers.

Evidence for the importance of steric factors comes from a consideration of the effect of α -alkyl substituents. It is found that the presence of an α -methyl substituent raises ΔH_p by at least 20 kJ mol⁻¹ (Table 4.10, compare entries for AA and MAA, MA and MMA, AN and MAN, S and AMS). The higher ΔH_p probably

reflects the greater difficulty in forming bonds to tertiary centers. This view is supported by the observation that higher alkyl substituents further increase ΔH_p [*e.g.* ethyl in MEA,³³¹ Table 4.10). Increasing the chain length of the α -substituent from methyl to ethyl should not greatly increase the thermodynamic stability of the radical, but steric factors will make the new bond both more difficult to form and easier to break.

Limited data suggest that the entropic term may be as important as the enthalpic term in determining polymerizability. The value of ΔS_p is lowered >20 J mol⁻¹ K⁻¹ by the presence of an α -methyl substituent (Table 4.10, compare entries for AN and MAN, S and AMS). This is likely to be a consequence of the polymers from α -methyl vinyl monomers having a more rigid, more ordered structure than those from the corresponding vinyl monomers.

There have been many studies on the polymerizability of α -substituted acrylic monomers.^{331,335-338} It is established that the ceiling temperature for α -alkoxyacrylates decreases with the size of the alkoxy group.³³⁵ However, it is of interest that polymerizations of α -(alkoxymethyl)acrylates (67)³³⁵ and α -(acyloxymethyl)acrylates (68)³³⁷ and captodative substituted monomers (69, 70)³³⁹ appear to have much higher ceiling temperatures than the corresponding α -alkylacrylates (*e.g.* methyl ethacrylate, MEA). For example, methyl α -ethoxymethacrylate³³⁵ readily polymerizes at 110 °C whereas MEA³³¹ has a very low ceiling temperature (Table 4.10). However, values of the thermodynamic parameters for these polymerizations have not yet been reported.



4.5.2 Measurement of Propagation Rate Constants

Methods for measurement of k_p have been reviewed by Stickler,^{340,341} van Herk ³⁴² and more recently by Beuermann and Buback.³⁴³ A largely non critical summary of values of k_p and k_t obtained by various methods appears in the *Polymer Handbook*.³⁴⁴ Literature values of k_p for a given monomer may span two or more orders of magnitude. The data and methods of measurement have been critically assessed by IUPAC working parties³⁴⁵⁻³⁵¹ and reliable values for most common monomers are now available.³⁴³ The wide variation in values of k_p (and k_t) obtained from various studies does not reflect experimental error but differences in data interpretation and the dependence of kinetic parameters on chain length and polymerization conditions.

Traditionally, measurement of k_p has required determination of the rate of polymerization under steady state (to give k_p/k_t^2) and non-steady state conditions

(to give k_p/k_t). The classical techniques in this context are the rotating sector ³⁵²⁻³⁵⁵ and related methods such as spatially intermittent polymerization (SIP).³⁵⁶

EPR methods that allow a more direct determination of k_p have been developed. These enable absolute radical concentrations to be determined as a function of conversion. With especially sensitive instrumentation, this can be done by direct measurement.³⁵⁷⁻³⁶⁰ An alternative method, applicable at high conversions, involves trapping the propagating species in a frozen matrix^{361,362} by rapid cooling of the sample to liquid nitrogen temperatures.

The radical concentration, when coupled with information on the rate of polymerization, allows k_p (and k_l) to be calculated. The EPR methods have been applied to various polymerizations including those of B, DMA, MMA,³⁶¹⁻³⁶⁶ S^{367,368} and VAc.³⁶⁹ Values for k_p are not always in complete agreement with those obtained by other methods (*e.g.* PLP, SIP) and this may reflect a calibration problem. Problems may also arise because of the heterogeneity of the polymerization reaction mixture,³⁶⁵ and insufficient sensitivity for the radical concentrations in low conversion polymerizations³⁶² or very low molecular weights. Some data must be treated with caution. However, the difficulties are now generally recognized and are being resolved.³⁶⁰

Pulsed laser photolysis (PLP) has emerged as the most reliable method for extracting absolute rate constants for the propagation step of radical polymerizations.³⁴³ The method can be traced to the work of Aleksandrov *et al.*³⁷⁰ PLP in its present form owes its existence to the extensive work of Olaj and coworkers³⁷¹ and the efforts of an IUPAC working party.³⁴⁵⁻³⁵¹ The method has now been successfully applied to establish rate constants, k_p (overall), for many polymerizations and copolymerizations.

In PLP the sample is subjected to a series of short (<30 ns) laser pulses at intervals τ . Analysis of the molecular weight distribution gives the length of chain formed between successive pulses (v) and this yields a value for k_p (eq. 13).

$$\mathbf{v} = k_{\mathbf{p}} \left[\mathbf{M} \right] \mathbf{\tau} \tag{13}$$

A molecular weight distribution for a PS sample obtained from a PLP experiment with S is shown in Figure 4.5. Olaj *et al.*³⁷¹ found empirically that v was best estimated from the points of inflection in the molecular weight distribution. Kinetic modeling of PLP has been carried out using Monte Carlo methods^{372,373} or by numerical integration.^{374,375} These studies confirm that the point of inflection in the molecular weight distribution is usually a good measure of v. With choice of polymerization conditions the values of v are relatively insensitive to the termination rate and mechanism and the occurrence of side reactions such as transfer to monomer. Some difficulties are experienced with high k_p monomers (acrylates, VAc) but appear to have been resolved through the use of low reaction temperatures and dilute media.³⁷⁵ These difficulties may arise through interference from backbiting.³²⁰ Independent determination of the rate of polymerization allows k_p/k_1 and hence k_t to be evaluated (Section 5.2).³⁷⁶

There are some reports that values of k_p are conversion dependent and that the value decreases at high conversion due to k_p becoming limited by the rate of diffusion of monomer. While conversion dependence of k_p at extremely high conversions is known, some data that indicate this may need to be reinterpreted, as the conversion dependence of the initiator efficiency was not recognized (Sections 3.3.1.1.3, 3.3.2.1.3 and 5.2.1.4).





4.5.3 Dependence of Propagation Rate Constant on Monomer Structure

Recent data for k_p are summarized in Table 4.11. Monomers have been grouped into three series according to the α -substituent (hydrogen, methyl, other). Some trends can be seen.

- (a) The Arrhenius A factor decreases by almost an order of magnitude in going from monomers with an α-hydrogen (20-80×10⁶ M⁻¹s⁻¹) to those with an α-methyl (2-5×10⁶ M⁻¹s⁻¹) and decreases further for those with a larger α-substituent, dimethyl itaconate (71) and the MA dimer (72), (0.2-1×10⁶ M⁻¹s⁻¹) (Table 4.11). The same overall trend is seen for analogous reactions of small radicals (Table 4.12, see also Section 2.3) and is predicted by theory.
- (b) Within both the α-hydrogen and α-methyl series, the lowest k_p values (for MAN, S, B) are associated with the highest activation energies and the more stable propagating radicals.

| monomer | $k_{\rm p} (60^{\circ} {\rm C})$ | Α | $E_{ m a}$ | reference |
|---------------------------|-----------------------------------|--|------------|-----------|
| | $M^{-1}s^{-1}$ | M ⁻¹ s ⁻¹ ×10 ⁶ | kJ mol⁻l | |
| α-H | | | | |
| MA | 28000 | 16.6 | 17.7 | 377 |
| BA | 31000 | 15.8 | 17.3 | 378 |
| DA | 39000 | 17.9 | 17.0 | 377 |
| VAc | 8300 | 14.7 | 20.7 | 379 |
| S^{c} | 340 | 42.7 | 32.5 | 349 |
| В | 200 | 80.5 | 35.7 | 380 |
| α -methyl | | | | |
| MĂA | 1200 | - | - | 381 |
| MAA(MeOH) ^d | 1000 | 0.60 | 17.7 | 381 |
| $MAA(H_2O)^e$ | 6700 | 1.72 | 15.3 | |
| MMA ^c | 820 | 2.67 | 22.4 | 348 |
| EMA ^c | 870 | 4.06 | 23.4 | 347 |
| <i>n</i> BMA ^c | 970 | 3.78 | 22.9 | 347 |
| iBMA | 1000 | 2.64 | 21.8 | 382 |
| EHMA | 1200 | 1.87 | 20.4 | 382 |
| DMA ^e | 1300 | 2.50 | 21.0 | 347 |
| HEMA | 3300 | 8.88 | 21.9 | 383 |
| GMA | 1600 | 4.41 | 21.9 | 383 |
| MAN | 59 | 2.69 | 29.7 | 384 |
| α -other | | | | |
| 71 | 25 | 0.20 | 24.9 | 385 |
| 72 | 30 | 1.25 | 29.5 | 386 |

 Table 4.11 Kinetic Parameters for Propagation in Selected Radical Polymerizations in Bulk Monomer

a Values are calculated from the Arrhenius parameters shown and given to two significant figures. b Values given to three significant figures. c IUPAC benchmark value. d 33 vol% MAA in methanol. Values are dependent on solvent and on concentration. e 15 vol% in water.

- (c) Within the series of alkyl acrylates and methacrylates there is a clear tendency for increase in k_p with increase in the length of the alkyl chain. The effect is small and, on the basis of the data shown in Table 4.11, cannot be assigned to a variation in A or E_a . However, there are reasonable theoretical grounds to expect this effect could be assigned to changes in the frequency factor.
- (d) The methacrylic monomers with protic substituents (MAA, HEMA) are associated with higher k_p values that are solvent and concentration dependent. The effect is suggestive of monomer-polymer and/or monomer-monomer association through hydrogen-bonding.

(e) The lowering of k_p with the increase in size of the α -substituent (MA>MMA>71~72) is associated with an increase in A and a decrease in E_a .



4.5.4 Chain Length Dependence of Propagation Rate Constants

It is usually assumed that propagation rate constants in homopolymerization (k_p) are independent of chain length and, for longer chains (length >20), there is experimental evidence to support this assumption.^{356,367} However, there is now a body of indirect evidence to suggest that the rate constants for the first few propagation steps $k_p(1)$, $k_p(2)$, *etc.* can be substantially different from $k_p(\text{overall})$ (refer Scheme 4.45). The effect can be seen as a special case of a penultimate unit effect (Section 7.3.1.2). Evidence comes from a number of sources, for example:

- (a) Chain transfer constants (k_p/k_{tr}) often show a marked chain length dependence for very short chain lengths (Section 5.3) indicating that k_p , k_{tr} or both are chain length dependent.³⁸⁷
- (b) The absolute rate constants for the reaction of small model radicals with monomers are typically at least an order of magnitude greater than the corresponding values of k_p (Table 4.12).³⁸⁸
- (c) Aspects of the kinetics of emulsion polymerization³⁸⁹ can be explained by invoking chain length dependence of k_p .
- (d) The apparent chain length dependence of k_p (average) in PLP experiments (Section 4.5.2) can be interpreted in this light.³⁷⁴ However, Olaj *et al.*³⁹⁰ have interpreted the same and similar data as suggesting a smaller decrease in k_p over a much longer range of chain lengths. They proposed that chain length dependence was a consequence of a change in the degree of solvation of the polymer chain and thus in the effective monomer concentration in the vicinity of the chain end. The explanation is analogous to that proposed to explain the bootstrap effect in copolymerization. Beuermann³⁴³ has questioned these interpretations pointing out that the interpretation of PLP data can be problematical due to the dependence of the shape of the molecular weight distribution on experimental parameters.

There have been attempts at direct measurements of these important kinetic parameters in AN,³⁹¹ MA,³⁹² MAN,^{393,394} MMA³⁹⁴ and S³⁹⁵ polymerizations. When the reaction is compared to a reference reaction care must be taken to establish the influence of chain length on the reference reaction.

Frequency factors for addition of small radicals to monomers are higher by more than an order of magnitude than those for propagation (Table 4.12). Activation energies are typically lower. However, trends in the data are very similar suggesting that the same factors are important in determining the relative reactivities for both small radicals and propagating species. The same appears to be true with respect to reactivities in copolymerization (Section 7.3.1.2).³⁸⁸

| ⊷СН₂-С́Н | HCH | | H ₃ C−C, CH ₃ |
|----------|--|-------|-------------------------------------|
| СО₂СН₃ | CO ₂ C(CH ₃) ₃ | | CO ₂ CH ₃ |
| PMA• | 73 | PMMA• | 74 |

Table 4.12 Rate Constants (25 °C) and Arrhenius Parameters for Propagation of Monomers CH₂=CR¹R² Compared with Rate Constants for Addition of Small Radicals³⁸⁸

| monomer | $k_{\rm p}^{\rm a}$ | logA | Ea | model | k_{a}^{a} | $\log A^b$ | Ea |
|---------|---------------------------------|------|----------------------|--------------------------------------|----------------|------------|----------|
| | M ⁻¹ s ⁻¹ | - | kJ mol⁻ ^t | | $M^{-1}s^{-1}$ | | kJ mol⁻¹ |
| Е | 77 | 7.27 | 34.3 | CH ₃ • | 12000 | 8.5 | 28.2 |
| S | 340 | 7.63 | 32.5 | PhCH₂• | 4700 | 8.5 | 30.8 |
| MA | 28000 | 7.22 | 17.7 | 73 | 1100000 | 8.5 | 15.6 |
| AN | | | | CH ₂ CN• | 410000 | 8.5 | 18.4 |
| MMA | 820 | 6.43 | 22.4 | 74 | 9700 | 7.5 | 22.4 |
| ΜΛΝ | 59 | 6.42 | 29.7 | C(CH ₃) ₂ CN+ | 2300 | 7.5 | 26.4 |

a Values at 60 °C calculated from the Arrhenius parameters shown and quoted to two significant figures. b Log *A* values based on recommendations of Fischer and Radom³⁸⁸ (refer Section 2.3.7).

4.6 References

- 1. Bovey, F.A. *Chain Structure and Conformation of Macromolecules*; Wiley: New York, 1982.
- 2. Randall, J.C. Polymer Sequence Determination; Academic Press: New York, 1977.
- 3. Bovey, F.A. Polymer Conformation and Configuration; Academic Press: New York, 1969.
- 4. Koenig, J.L. Chemical Microstructure of Polymer Chains; Wiley: New York, 1980.
- 5. Koenig, J.L. Spectroscopy of Polymers; Elsevier: New York, 1999.
- 6. Tonelli, A.E. NMR Spectroscopy and Polymer Microstructure; VCH: New York, 1989.
- 7. Hatada, K. NMR Spectroscopy of Polymers; Springer-Verlag: Berlin, 2003.
- 8. Farina, M. Top. Stereochem. 1987, 17, 1.
- 9. Bovey, F.A.; Tiers, G.V.D. J. Polym. Sci. 1960, 44, 173.
- Hatada, K.; Kitayama, T.; Ute, K. In Annual Reports in NMR Spectroscopy; Webb, G.A., Ed.; Academic Press: London, 1993; Vol. 26, p 100.
- 11. Bovcy, F.A.; Mirau, P. NMR of Polymers; Academic Press: New York, 1996.
- 12. King, J.; Bower, D.I.; Maddams, W.F.; Pyszora, H. Makromol. Chem. 1983, 184, 879.
- 13. Fox, T.G.; Schnecko, H.W. Polymer 1962, 3, 575.

- 14. Khanarian, G.; Cais, R.E.; Kometani, J.M.; Tonelli, A.E. *Macromolecules* **1982**, *15*, 866.
- 15. Bovey, F.A.; Mirau, P.A. Acc. Chem. Res. 1988, 21, 37.
- 16. Beshah, K. Makromol. Chem. 1993, 194, 3311.
- 17. Moad, G.; Rizzardo, E.; Solomon, D.H.; Johns, S.R.; Willing, R.I. *Macromolecules* 1986, 19, 2494.
- 18. Berger, P.A.; Kotyk, J.J.; Remsen, E.E. Macromolecules 1992, 25, 7227.
- 19. Kotyk, J.J.; Berger, P.A.; Remsen, E.E. Macromolecules 1990, 23, 5167.
- Schilling, F.C.; Bovey, F.A.; Bruch, M.D.; Kozlowski, S.A. Macromolecules 1985, 18, 1418.
- Dong, L., Hill, D.J.T., O'Donnell, J.H., Whittaker, A.K. *Macromolecules* 1994, 27, 1830.
- 22. Hikichi, K.; Yasuda, M. Polym. J. 1987, 19, 1003.
- 23. Gippert, G.P.; Brown, L.R. Polym. Bull. 1984, 11, 585.
- 24. Mirau, P.A.; Bovey, F.A. Macromolecules 1986, 19, 210.
- 25. Crowther, M.W.; Szeverenyi, N.M.; Levy, G.C. Macromolecules 1986, 19, 1333.
- 26. Bruch, M.D.; Bovcy, F.A.; Cais, R.E. Macromolecules 1984, 17, 2547.
- 27. Hatada, K.; Kitayama, T.; Terawaki, Y.; Chujo, R. Polym. J. 1987, 19, 1127.
- Chujo, R.; Hatada, K.; Kitamaru, R.; Kitayama, T.; Sato, H.; Tanaka, Y. Polym. J. 1987, 19, 413.
- 29. Chujo, R.; Hatada, K.; Kitamaru, R.; Kitayama, T.; Sato, H.; Tanaka, Y.; Horii, F.; Terawaki, Y. *Polym. J.* **1988**, *20*, 627.
- Elias, H.G. In *Polymer Handbook*, 3rd ed.; Brandup, J.; Immergut, E.H., Eds.; Wiley: New York, 1989; p II/357.
- 31. Nakano, T.; Mori, M.; Okamoto, Y. Macromolecules 1993, 26, 867.
- 32. Porter, N.A.; Rosenstein, I.J.; Breyer, R.A.; Bruhnke, J.D.; Wu, W.-X.; McPhail, A.T. J. Am. Chem. Soc. 1992, 114, 7664.
- 33. Porter, N.A.; Allen, T.; Breyer, R.A. J. Am. Chem. Soc. 1992, 114, 7676.
- 34. Habaue, S.; Isobe, Y.; Okamoto, Y. Tetrahedron 2002, 58, 8205.
- 35. Tsuruta, T.; Makimoto, T.; Kanai, H. J. Macromol. Chem. 1966, 1, 31.
- 36. Moad, G.; Solomon, D.H.; Spurling, T.H.; Johns, S.R.; Willing, R.I. Aust. J. Chem. 1986, 39, 43.
- 37. Reinmöller, M.; Fox, T.G. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1966, I, 999.
- Ferguson, R.C.; Ovenall, D.W. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem.) 1985, 26(1), 182.
- 39. Ovenall, D.W. Macromolecules 1984, 17, 1458.
- 40. Nagara, Y.; Yamada, K.; Nakano, T.; Okamoto, Y. Polymer J. 2001, 33, 534.
- 41. Nagara, Y.; Nakano, T.; Okamoto, Y.; Gotoh, Y.; Nagura, M. Polymer 2001, 42, 9679.
- 42. Yamada, K.; Nakano, T.; Okamoto, Y. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 220.
- 43. Kamide, K.; Yamazaki, H.; Okajima, K.; Hikichi, K. Polym. J. 1985, 17, 1233.
- 44. Matsuzaki, K.; Kanai, T.; Kawamura, T.; Matsumoto, S.; Uryu, T. J. Polym. Sci., Polym. Chem. Ed. 1973, 11, 961.
- 45. Suzuki, T.; Santee, E.R., Jr; Harwood, H.J.; Vogl, O.; Tanaka, T. J. Polym. Sci., Polym. Lett. Ed. 1974, 12, 635.
- 46. Sato, H.; Tanaka, Y.; Hatada, K. J. Polym. Sci., Polym. Phys. Ed. 1983, 21, 1667.
- 47. Kawamura, T.; Uryu, T.; Matsuzaki, K. Makromol. Chem., Rapid Commun. 1982, 3, 661.

- 48. Kawamura, T.; Toshima, N.; Matsuzaki, K. Macromol. Rapid Commun. 1994, 15, 479.
- 49. Bugada, D.C.; Rudin, A. J. Appl. Polym. Sci. 1985, 30, 4137.
- 50. Wu, T.K.; Ovenall, D.W. Macromolecules 1974, 7, 776.
- 51. Suito, Y.; Isobe, Y.; Habaue, S.; Okamoto, Y. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 2496.
- 52. Flory, P.J.; Leutner, F.S. J. Polym. Sci. 1950, 5, 267.
- 53. Adelman, R.L.; Ferguson, R.C. J. Polym. Sci., Polym. Chem. Ed. 1975, 13, 891.
- 54. Vercauteren, F.F.; Donners, W.A.B. Polymer 1986, 27, 993.
- 55. Amiya, S.; Uetsuki, M. Macromolecules 1982, 15, 166.
- 56. Starnes, W.H., Jr.; Wojciechowski, B.J. Makromol. Chem., Macromol. Symp. 1993, 70/71, 1.
- 57. Starnes, W.H., Jr. In *Developments in Polymer Degradation*.; Grassie, N., Ed.; Applied Science: London, 1981; Vol. 3, p 135.
- 58. Caraculacu, A.A. Pure Appl. Chem. 1981, 53, 385.
- 59. Hjertberg, T.; Sorvik, E. In *Degradation and Stabilisation of PVC*; Owen, E.D., Ed.; Elsevier Applied Science: Barking, 1984; p 21.
- 60. Starnes, W.II. Prog. Polym. Sci. 2002, 27, 2133.
- Starnes, W.H., Jr.; Schilling, F.C.; Abbas, K.B.; Cais, R.E.; Bovey, F.A. Macromolecules 1979, 12, 556.
- 62. Darricades-Llauro, M.F.; Michel, A.; Guyot, A.; Waton, H.; Petiaud, R.; Pham, Q.T. J. Macromol. Sci., Chem. 1986, A23, 221.
- 63. Rigo, A.; Palma, G.; Talamini, G. Makromol. Chem. 1972, 153, 219.
- 64. Fossey, J.; Nedelec, J.-Y. Tetrahedron 1981, 37, 2967.
- 65. Starnes, W.H., Jr.; Wojciechowski, B.J.; Velazquez, A.; Benedikt, G.M. Macromolecules 1992, 25, 3638.
- 66. Park, G.S.; Saleem, M. Polym. Bull. 1979, 1, 409.
- Starnes, W.H., Jr.; Schilling, F.C.; Plitz, I.M.; Cais, R.E.; Freed, D.J.; Hartless, R.L.; Bovey, F.A. *Macromolecules* 1983, 16, 790.
- Mitani, K.; Ogata, T.; Awaya, H.; Tomari, Y. J. Polym. Sci., Polym. Chem. Ed. 1975, 13, 2813.
- 69. Wilson, C.W., III; Santee, E.R., Jr. J. Polym. Sci., Part C 1965, 8, 97.
- 70. Cais, R.E.; Kometani, J.M. ACS Symp. Ser. 1984, 247, 153.
- 71. Ovenall, D.W.; Uschold, R.E. Macromolecules 1991, 24, 3235.
- 72. Görlitz, V.M.; Minke, R.; Trautvetter, W.; Weisgerber, G. Angew. Macromol. Chem. 1973, 29/30, 137.
- 73. Cais, R.E.; Kometani, J.M. Macromolecules 1984, 17, 1887.
- 74. Cais, R.E.; Sloane, N.J.A. Polymer 1983, 24, 179.
- 75. Cais, R.E.; Kometani, J.M. Macromolecules 1984, 17, 1932.
- 76. Ferguson, R.C.; Brame, E.G., Jr. J. Phys. Chem. 1979, 83, 1397.
- 77. Matsumoto, A.; Iwanami, K.; Oiwa, M. J. Polym. Sci., Polym. Lett. Ed. 1980, 18, 211.
- Matsumoto, A.; Iwanami, K.; Kawaguchi, N.; Oiwa, M. Technol. Rep. Kansai Univ. 1983, 24, 183.
- Matsumoto, A.; Kikuta, M.; Oiwa, M. J. Polym. Sci., Part C: Polym. Lett. 1986, 24, 7.
- 80. Matsumoto, A.; Tcrada, T.; Oiwa, M. J. Polym. Sci., Part A: Polym. Chem. 1987, 25, 775.
- 81. Matsumoto, A.; Iwanami, K.; Oiwa, M. J. Polym. Sci., Polym. Lett. Ed. 1981, 19, 497.

- 82. Marvel, C.S.; Cowan, J.C. J. Am. Chem. Soc. 1939, 61, 3156.
- 83. Marvel, C.S.; Dec, J.; Cooke, H.G., Jr.; Cowan, J.C. J. Am. Chem. Soc. 1940, 62, 3495.
- 84. McCurdy, K.G.; Laidler, K.J. Can. J. Chem. 1964, 42, 818.
- 85. Sawant, S.; Morawetz, H. J. Polym. Sci., Polym. Lett. Ed. 1982, 20, 385.
- 86. Minagawa, M. J. Polym. Sci., Polym. Chem. Ed. 1980, 18, 2307.
- Kamachi, M.; Kajiwara, A.; Saegusa, K.; Morishima, Y. Macromolecules 1993, 26, 7369.
- Henderson, J.N. In *Encyclopedia of Polymer Science and Engineering*, 2nd ed.; Mark, H.F.; Bikales, N.M.; Overberger, C.G.; Menges, G., Eds.; Wiley: New York, 1985; Vol. 2, p 515.
- Tate, D.P.; Bethea, T.W. In Encyclopedia of Polymer Science and Engineering, 2nd ed.; Mark, H.F.; Bikales, N.M.; Overberger, C.G.; Menges, G., Eds.; Wiley: New York, 1985; Vol. 2, p 537.
- Senyck, M.L. In *Encyclopedia of Polymer Science and Engineering*, 2nd cd.; Mark, H.F.; Bikales, N.M.; Overberger, C.G.; Menges, G., Eds.; Wiley: New York, 1987; Vol. 8, p 487.
- Stewart, C.A.; Takeshita, T.; Coleman, M.L. In *Encyclopedia of Polymer Science* and Engineering, 2nd ed.; Mark, H.F.; Bikales, N.M.; Overberger, C.G.; Menges, G., Eds.; Wiley: New York, 1986; Vol. 3, p 441.
- 92. Khachaturov, A.S.; Ivanova, V.P.; Podkorytov, I.S.; Osetrova, L.V. Vysokomol. Soedin. 1998, 40, 964.
- 93. Sato, H.; Takebayashi, K.; Tanaka, Y. Macromolecules 1987, 20, 2418.
- 94. Kawahara, S.; Bushimata, S.; Sugiyama, T.; Hashimoto, C.; Tanaka, Y. Rubber Chem. Technol. 1999, 72, 844.
- 95. Coleman, M.M.; Tabb, D.L.; Brame, E.G. Rubber Chem. Technol. 1977, 50, 49.
- 96. Coleman, M.M.; Brame, E.G. Rubber Chem. Technol. 1978, 51, 668.
- 97. Sato, H.; Ono, A.; Tanaka, Y. Polymer 1977, 18, 580.
- 98. Butler, G.B. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 4, p 423.
- Butler, G.B. In *Encyclopedia of Polymer Science and Engineering*, 2nd cd.; Mark, H.F.; Bikales, N.M.; Overberger, C.G.; Menges, G., Eds.; Wiley: New York, 1986; Vol. 4, p 543.
- 100. Butler, G.B. Acc. Chem. Res. 1982, 15, 370.
- 101. Butler, G.B. In *Polymeric Amines and Ammonium Salts*; Goethals, E.J., Ed.; Pergamon: New York, 1981; p 125.
- 102. Butler, G.B. Cyclopolymerization and Cyclocopolymerization; Marcel Dekker: New York, 1992.
- 103. Butler, G.B.; Bunch, R.L. J. Am. Chem. Soc. 1949, 71, 3120.
- 104. Butler, G.B.; Ingley, F.L. J. Am. Chem. Soc. 1951, 72, 894.
- 105. Butler, G.B.; Angelo, R.J. J. Am. Chem. Soc. 1957, 79, 3128.
- 106. Butler, G.B.; Crawshaw, A.; Miller, W.L. J. Am. Chem. Soc. 1958, 80, 3615.
- 107. Solomon, D.H. J. Polym. Sci., Polym. Symp. 1975, 49, 175.
- 108. Beekwith, A.L.J.; Hawthorne, D.G.; Solomon, D.H. Aust. J. Chem. 1976, 29, 995.
- 109. Solomon, D.H. J. Macromol. Sci., Chem. 1975, A9, 97.
- Solomon, D.H.; Hawthorne, D.G. J. Macromol. Sci., Rev. Macromol. Chem. 1976, C15, 143.
- 111. Johns, S.R.; Willing, R.I. J. Macromol. Sci., Chem. 1976, 10, 875.
- 112. Lancaster, J.E.; Bacchel, L.; Panzer, H.P. J. Polym. Sci., Polym. Lett. Ed. 1976, 14, 549.

- 113. Beckwith, A.L.J. Tetrahedron 1981, 37, 3073.
- 114. Xi, F.; Vogl, O. J. Macromol. Sci., Chem. 1983, A20, 321.
- 115. Otsu, T.; Ohya, T. J. Macromol. Sci., Chem. 1984, A21, 1.
- 116. Seyferth, D.; Robison, J. Macromolecules 1993, 26, 407.
- 117. Beckwith, A.L.J. Chem. Soc. Rev. 1993, 143.
- 118. Tsuda, T.; Mathias, L.J. Macromolecules 1993, 26, 6359.
- 119. Masterman, T.C.; Dando, N.R.; Weaver, D.G.; Seyferth, D. J. Polym. Sci., Part A: Polym. Phys. 1994, 32, 2263.
- 120. Fukuda, W.; Nakao, M.; Okumura, K.; Kakiuchi, H. J. Polym. Sci., Part A-1 1972, 10, 237.
- 121. Ichihashi, T.; Kawai, W. Kobunshi Kagaku 1971, 28, 225.
- 122. Trossarelli, L.; Guaita, M.; Priola, A. J. Polym. Sci., Part B 1967, 5, 129.
- 123. Trossarelli, L.; Guaita, M.; Priola, A. Makromol. Chem. 1967, 100, 147.
- 124. Kodaira, T.; Okumura, M.; Urushisaki, M.; Isa, K. J. Polym. Sci., Part A: Polym. Chem. 1993, 31, 169.
- 125. Kodaira, T.; Mae, Y. Polymer 1992, 33, 3500.
- 126. Yokata, K.; Takada, Y. Kobunshi Kagaku 1969, 26, 317.
- 127. Kaye, H. Macromolecules 1971, 4, 147.
- 128. Field, N.D. J. Org. Chem. 1960, 25, 1006.
- 129. Marvel, C.S.; Gall, E.J. J. Org. Chem. 1960, 25, 1784.
- 130. Marvel, C.S.; Vest, R.D. J. Am. Chem. Soc. 1959, 81, 984.
- 131. Milford, G.N. J. Polym. Sci. 1959, 41, 295.
- Thang, S.H.; Rizzardo, E.; Moad, G. US 5830966, 1996 (Chem. Abstr. 1994, 123, 229253).
- 133. Tüzün, N.S.; Aviyente, V.; Houk, K.N. J. Org. Chem. 2002, 67, 5068.
- 134. Miyake, T. Kogyo Kagaku Zasshi 1961, 64, 359.
- 135. Ohya, T.; Otsu, T. J. Polym. Sci., Polym. Chem. Ed. 1983, 21, 3503.
- Matsumoto, A.; Kitamura, T.; Oiwa, M.; Butler, G.B. J. Polym. Sci., Polym. Chem. Ed. 1981, 19, 2531.
- 137. Matsumoto, A.; Kitamura, T.; Oiwa, M.; Butler, G.B. Makromol. Chem., Rapid Commun. 1981, 2, 683.
- 138. Stansbury, J.W. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1990, 31(1), 503.
- 139. Mathias, L.J.; Kusefoglu, S.H.; Ingram, J.E. Macromolecules 1988, 21, 545.
- 140. Mathias, L.J.; Colletti, R.F.; Bielecki, A. J. Am. Chem. Soc. 1991, 113, 1550.
- 141. Aso, C.; Kunitake, T.; Ando, S. J. Macromol. Sci., Chem. 1971, A5, 167.
- 142. Raymond, M.A.; Dietrich, H.J. J. Macromol. Sci., Chem. 1972, A6, 207.
- 143. Billingham, N.C.; Jenkins, A.D.; Kronfli, E.B.; Walton, D.R.M. J. Polym. Sci., Polym. Chem. Ed. 1977, 15, 675.
- 144. Butler, G.B.; Stackman, R.W. J. Org. Chem. 1960, 25, 1643.
- 145. Butler, G.B.; Stackman, R.W. J. Macromol. Sci., Chem. 1969, A3, 821.
- 146. Kida, S.; Nozakura, S.-I.; Murahashi, S. Polym. J. 1972, 3, 234.
- 147. Butler, G.B.; Skinner, D.L.; Bond, W.C., Jr.; Rogers, C.L. J. Macromol. Sci., Chem. 1970, A4, 1437.
- 148. Berlin, K.D.; Butler, G.B. J. Am. Chem. Soc 1960, 82, 2712.
- 149. Benyon, K.I. J. Polym. Sci., Part A 1963, 1, 3357.
- 150. Berlin, K.D.; Butler, G.B. J. Org. Chem. 1960, 25, 2006.
- 151. Corfield, G.C.; Monks, H.H. J. Macromol. Sci., Chem. 1975, A9, 1113.
- 152. Butler, G.B.; Kimura, S. J. Macromol. Sci., Chem. 1971, A5, 181.
- 153. Matsoyan, S.G.; Pogosyan, G.M.; Elliasyan, M.A. Vysokomol. Soedin 1963, 5, 777.
- 154. Hawthorne, D.G.; Solomon, D.H. J. Macromol. Sci., Chem. 1975, A9, 149.
- 155. Beckwith, A.L.J.; Moad, G. J. Chem. Soc., Perkin Trans. 2 1975, 1726.
- 156. Costa, L.; Chiantore, O.; Guaita, M. Polymer 1978, 19, 202.
- 157. Wiley, R.H.; Rivera, W.H.; Crawford, T.H.; Bray, N.F. J. Polym. Sci. 1962, 61, 538.
- 158. Graham, P.J.; Buhle, E.L.; Pappas, N. J. Org. Chem. 1961, 26, 4658.
- 159. Paulrajan, S.; Gopalan, A.; Subbaratnam, N.R.; Venkatarao, K. Polymer 1983, 24, 906.
- 160. Gopalan, A.; Paulrajan, S.; Subbaratnam, N.R.; Rao, K.V. J. Polym. Sci., Polym. Chem. Ed. 1985, 23, 1861.
- Nishikubo, T.; Iizawa, T.; Yoshinaga, A.; Nitta, M. Makromol. Chem. 1982, 183, 789.
- 162. Simpson, W.; Holt, T.; Zetie, R.J. J. Polvm. Sci. 1953, 10, 489.
- 163. Haward, R.N. J. Polym. Sci. 1953, 10, 535.
- Matsumoto, A.; Iwanami, K.; Oiwa, M. J. Polym. Sci., Polym. Lett. Ed. 1980, 18, 307.
- 165. Beckwith, A.L.J.; Moad, G. J. Chem. Soc., Chem. Commun. 1974, 472.
- 166. Ohya, T.; Otsu, T. J. Polym. Sci., Polym. Chem. Ed. 1983, 21, 3169.
- 167. Barton, J.M.; Butler, G.B.; Chapin, E.C. J. Polym. Sci., Part A 1965, 3, 501.
- Bailey, W.J.; Sun, R.L. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1972, 13, 281.
- 169. Brady, R.F., Jr. J. Macromol. Sci., Rev. Macromol. Chem. Phys. 1992, C32, 135.
- Stansbury, J.W. In *Expanding Monomers*; Sadhir, R.K.; Luck, R.M., Eds.; CRC Press: Boca Raton, Florida, 1992; p 153.
- 171. Moszner, N.; Salz, U. Prog. Polym. Sci. 2001, 26, 535.
- 172. Sanda, F.; Endo, T. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 265.
- 173. Klemm, E.; Schulze, T. Acta Polym. 1999, 50, 1.
- 174. Cho, I. Prog. Polym. Sci. 2000, 25, 1043.
- 175. Moszner, N.; Zeuner, F.; Volkel, T.; Rheinberger, V. Macromol. Chem. Phys. 1999, 200, 2173.
- 176. Endo, T.; Yokozawa, T. In New Methods for Polymer Synthesis; Mijs, W.J., Ed.; Plenum: New York, 1992; p 155.
- 177. Bailey, W.J. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 283.
- 178. Colombani, D. Prog. Polym. Sci. 1999, 24, 425.
- 179. Hall, H.K., Jr.; Ykman, P.J. J. Polym. Sci., Macromol. Rev. 1976, 11, 1.
- 180. Bothe, H.; Schluter, A.-D. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1988, 29, 412.
- 181. Hall, H.K., Jr.; Padias, A.B. J. Polym. Sci., Part A: Polym. Chem. 2003, 41, 625.
- 182. Hall, H.K., Jr.; Padias, A.B. J. Am. Chem. Soc. 1971, 4193, 110.
- 183. Beckwith, A.L.J.; Moad, G. J. Chem. Soc., Perkin Trans. 2 1980, 1473.
- 184. Newcomb, M. Tetrahedron 1993, 49, 1151.
- 185. Masnovi, J.; Samsel, E.G.; Bullock, R.M. J. Chem. Soc., Chem. Commun 1989, 1044.
- 186. Ingold, K.U.; Maillard, B.; Walton, J.C. J. Chem. Soc., Perkin Trans. 2 1981, 970.
- 187. Beckwith, A.L.J.; Bowry, V.W. J. Org. Chem. 1989, 54, 2681.
- 188. Takahashi, T.; Yamashita, I.; Miyakawa, T. Bull. Chem. Soc. Japan 1964, 37, 131.
- 189. Takahashi, T.; Yamashita, I. J. Polym. Sci., Part B 1965, 3, 251.
- 190. Cho, I.; Ahn, K.-D. J. Polym. Sci., Polym. Chem. Ed. 1979, 17, 3169.

Propagation

- Lishanskii, I.S.; Zak, A.G.; Fedorova, E.F.; Khachaturov, A.S. Vysokomolekul. Soedin 1965, 7, 966.
- 192. Endo, T.; Watanabe, M.; Suga, K.; Yokozawa, T. J. Polym. Sci., Part A: Polym. Chem. 1989, 27, 1435.
- 193. Endo, T.; Suga, K. J. Polym. Sci., Part A: Polym. Chem. 1989, 27, 1831.
- 194. Endo, T.; Watanabe, M.; Suga, K.; Yokozawa, T. Makromol. Chem. 1989, 190, 691.
- 195. Endo, T.; Watanabe, M.; Suga, K.; Yokozawa, T. J. Polym. Sci., Part A: Polym. Chem. 1987, 25, 3039.
- 196. Cho, I.; Ahn, K.-D. J. Polym. Sci., Polym. Lett. Ed. 1977, 15, 751.
- 197. Cho, I.; Lee, J.-Y. Makromol. Chem., Rapid Commun. 1984, 5, 263.
- 198. Cho, I.; Song, S.S. J. Polym. Sci., Part A: Polym. Chem. 1989, 27, 3151.
- 199. Takahashi, T. J. Polym. Sci., Part A-1 1968, 6, 403.
- 200. Kennedy, J.P.; Elliot, J.J.; Butler, P.E. J. Macromol. Sci., Chem. 1968, A2, 1415.
- 201. Cho, I.; Song, S.S. Makromol. Chem., Rapid Commun. 1989, 10, 85.
- 202. Sanda, F.; Takata, T.; Endo, T. Macromolecules 1994, 27, 1099.
- 203. Sanda, F.; Takata, T.; Endo, T. J. Polym. Sci., Part A: Polym. Chem. 1993, 31, 2659.
- 204. Sanda, F.; Takata, T.; Endo, T. Macromolecules 1993, 26, 5748.
- 205. Sanda, F.; Takata, T.; Endo, T. Macromolecules 1992, 25, 6719.
- 206. Sanda, F.; Takata, T.; Endo, T. Macromolecules 1993, 26, 729.
- 207. Bowry, V.W.; Lusztyk, J.; Ingold, K.U. J. Chem. Soc., Chem. Commun 1990, 923.
- 208. Sanda, F.; Takata, T.; Endo, T. Macromolecules 1993, 26, 1818.
- 209. Yang, Z.-Y. J. Am. Chem. Soc. 2003, 125, 870.
- 210. Cho, I.; Kim, J.-B. J. Polvm. Sci., Polym. Lett. Ed. 1983, 21, 433.
- 211. Endo, T.; Kanda, N. Polym. Prepr. Jpn. 1987, 36, 140.
- Koizumi, T.; Nojima, Y.; Endo, T. J. Polym. Sci., Part A: Polym. Chem. 1993, 31, 3489.
- 213. Laurie, D.; Nonhebel, D.C.; Suckling, C.J.; Walton, J.C. *Tetrahderon* **1993**, *49*, 5869.
- 214. Beckwith, A.L.J.; Moad, G. J. Chem. Soc., Perkin Trans. 2 1980, 1083.
- 215. Hiraguri, Y.; Endo, T. J. Polym. Sci., Part C: Polym. Lett. 1989, 27, 333.
- 216. Cho, I.; Choi, S.Y. Makromol. Chem., Rapid Commun. 1991, 12, 399.
- 217. Cho, I.; Kim, S.-K.; Lee, M.-H. J. Polym. Sci., Polym. Symp. 1986, 74, 219.
- 218. Cho, I.; Lee, M.-H. J. Polym. Sci., Part C: Polym. Lett. 1987, 25, 309.
- 219. Errede, L.A. J. Polym. Sci. 1961, 49, 253.
- 220. Bailey, W.J.; Chou, J.L. Polym. Mater. Sci. Eng. 1987, 56, 30.
- Bailey, W.J.; Amone, M.J.; Chou, J.L. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1988, 29(1), 178.
- 222. Klemm, E.; Schulze, T. Makromol. Chem. 1993, 194, 2087.
- 223. Hiraguri, Y.; Endo, T. J. Polym. Sci., Part A: Polym. Chem. 1989, 27, 4403.
- 224. Acar, M.H.; Nambu, Y.; Yamamoto, K.; Endo, T. J. Polym. Sci., Part A: Polym. Chem. 1989, 27, 4441.
- 225. Bailey, W.J.; Gu, J.M.; Zhou, L.L. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1990, 31(1), 24.
- 226. Cho, I.; Song, K.Y. Makromol. Chem., Rapid Commun. 1993, 14, 377.
- 227. Cho, I.; Kim, S.-K. J. Polym. Sci., Part A: Polym. Chem. 1990, 28, 417.
- 228. Bailey, W.J.; Chen, P.Y.; Chen, S.-C.; Chiao, W.-B.; Endo, T.; Gapud, B.; Kuruganti, Y.; Lin, Y.-N.; Ni, Z.; Pan, C.-Y.; Shaffer, S.E.; Sidney, L.; Wu, S.-R.;

Yamamoto, N.; Yamazaki, N.; Yonczawa, K.; Zhou, L.-L. Makromol. Chem., Macromol. Symp. 1986, 6, 81.

- 229. Bailey, W.J. Polym. J. 1985, 17, 85.
- 230. Bailey, W.J. Makromol. Chem., Suppl. 1985, 13, 171.
- 231. Bailey, W.J.; Chou, J.L.; Feng, P.-Z.; Issari, B.; Kuruganti, V.; Zhou, L.-L. J. Macromol. Sci., Chem. 1988, A25, 781.
- 232. Kobayashi, S.; Kadokawa, J.; Shoda, S.; Uyama, H. Macromol. Reports 1991, A28 (Suppl. 1), 1.
- 233. Kobayashi, S.; Kadokawa, J.; Matsumura, Y.; Yen, I.F.; Uyama, H. Macromol. Reports 1992, A29 (Suppl. 3), 243.
- 234. Bailey, W.J.; Wu, S.-R.; Ni, Z. J. Macromol. Sci., Chem. 1982, A18, 973.
- 235. Bailey, W.J.; Wu, S.-R.; Ni, Z. Makromol. Chem. 1982, 183, 1913.
- 236. Endo, T.; Yako, N.; Azuma, K.; Nate, K. Makromol. Chem. 1985, 186, 1543.
- 237. Yokozawa, T.; Ilayashi, R.; Endo, T. J. Polym. Sci., Part A: Polym. Chem. 1990, 28, 3739.
- 238. Bailey, W.J.; Ni, Z.; Wu, S.-R. J. Polym. Sci., Polym. Chem. Ed. 1982, 20, 3021.
- 239. Endo, T.; Okawara, M.; Bailey, W.J.; Azuma, K.; Nate, K.; Yokona, H. J. Polym. Sci., Polym. Lett. Ed. 1983, 21, 373.
- 240. Bailey, W.J.; Ni, Z.; Wu, S.-R. Macromolecules 1982, 15, 711.
- 241. Bailey, W.J.; Arfaei, P.Y.; Chen, P.Y.; Chen, S.-C.; Endo, T.; Pan, C.-Y.; Ni, Z.; Shaffer, S.E.; Sidney, L.; Wu, S.-R.; Yamazaki, N. In Proc. IUPAC 28th Macromol. Symp.: Amherst, MA, 1982; p 214.
- 242. Sidney, L.N.; Shaffer, S.E.; Bailey, W.J. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1981, 22(2), 373.
- 243. Barclay, L.R.C.; Griller, D.; Ingold, K.U. J. Am. Chem. Soc. 1982, 104, 4399.
- 244. Beckwith, A.L.J.; Thomas, C.B. J. Chem. Soc., Perkin Trans. 2 1973, 861.
- Bailey, W.J.; Zhou, L.L. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1989, 30(1), 195.
- 246. Hiraguri, Y.; Endo, T. J. Polym. Sci., Part A: Polym. Chem. 1992, 30, 689.
- 247. Hiraguri, Y.; Endo, T. J. Am. Chem. Soc. 1987, 109, 3779.
- 248. Hiraguri, Y.; Endo, T. J. Polym. Sci., Part A: Polym. Chem. 1989, 27, 2135.
- 249. Cho, I.; Kim, B.-G.; Park, Y.-C.; Kim, C.-B.; Gong, M.-S. Makromol. Chem., Rapid Commun. 1991, 12, 141.
- 250. Pan, C.-Y.; Wu, Z.; Bailey, W.J. J. Polym. Sci., Part C: Polym. Lett. 1987, 25, 243.
- 251. Sugiyama, J.-I.; Yokozawa, T.; Endo, T. J. Am. Chem. Soc. 1993, 115, 2041.
- 252. Morariu, S.; Buruiana, E.C.; Simionescu, B.C. Polym. Bull. 1993, 30, 7.
- Miyagawa, T.; Sanda, F.; Endo, T. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 1861.
- 254. Bailey, W.J.; Feng, P.-Z. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1987, 28(1), 154.
- 255. Cho, I.; Lee, T.-W. Macromol. Chem. Rapid Commun. 1989, 10, 453.
- 256. Tsang, R.; Dickson, J.K., Jr.; Pak, H.; Walton, R.; Frascr-Reid, B. J. Am. Chem. Soc. 1987, 109, 3484.
- 257. Evans, R.A.; Moad, G.; Rizzardo, E.; Thang, S.H. Macromolecules 1994, 27, 7935.
- 258. Evans, R.A.; Rizzardo, E. Macromolecules 1996, 29, 6983.
- 259. Evans, R.A.; Rizzardo, E. Macromolecules 2000, 33, 6722.
- 260. Evans, R.A.; Rizzardo, E. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 202.
- 261. Harrisson, S.; Davis, T.P.; Evans, R.A.; Rizzardo, E. Macromolecules 2001, 34, 3869.

Propagation

- 262. Harrisson, S.; Davis, T.P.; Evans, R.A.; Rizzardo, E. Macromolecules 2000, 33, 9553.
- 263. Okazaki, T.; Sanda, F.; Endo, T. J. Polym. Sci., Part A: Polym. Chem. 1996, 34, 2029.
- 264. Okazaki, T.; Komiya, T.; Sanda, F.; Miyazaki, K.; Endo, T. J. Polym. Sci., Part A: Polym. Chem. 1997, 35, 2501.
- 265. Endo, T.; Bailey, W.J. J. Polym. Sci., Polym. Lett. Ed. 1975, 13, 193.
- 266. Sugiyama, J.-I.; Yokozawa, T.; Endo, T. J. Polym. Sci., Part A: Polym. Chem. 1990, 28, 3529.
- 267. Bailey, W.J.; Zheng, Z.-F. J. Polym. Sci., Part A: Polym. Chem. 1991, 29, 437.
- 268. Endo, T.; Bailey, W.J. J. Polym. Sci., Polym. Lett. Ed. 1980, 18, 25.
- 269. Tagoshi, H.; Endo, T. J. Polym. Sci., Part C: Polym. Lett. 1988, 26, 77.
- 270. Schulze, T.; Klemm, E. Polym. Bull. 1993, 31, 409.
- 271. Issari, B.; Bailey, W.J. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1988, 29(1), 217.
- 272. Okazaki, T.; Sanda, F.; Endo, T. Polymer J. 1998, 30, 365.
- 273. Sanda, F.; Takata, T.; Endo, T. Macromolecules 1994, 27, 3986.
- 274. Moszner, N.; Zeuner, F.; Fischer, U.K.; Rheinberger, V. Polym. Bull. 1998, 40, 447.
- 275. Okazaki, T.; Sanda, F.; Endo, T. J. Polym. Sci., Part A: Polym. Chem. 1997, 35, 2487.
- Chiefari, J.; Jeffery, J.; Moad, G.; Mayadunne, R.T.A.; Rizzardo, E.; Thang, S.H. Macromolecules 1999, 32, 5559.
- 277. Chiefari, J.; Jeffery, J.; Mayadunne, R.T.A.; Moad, G.; Rizzardo, E.; Thang, S.H. ACS Symp. Ser. 2000, 768, 297.
- 278. Viswanadhan, V.N.; Mattice, W.L. Makromol. Chem. 1985, 186, 633.
- 279. Toh, J.S.S.; Huang, D.M.; Lovell, P.A.; Gilbert, R.G. Polymer 2001, 42, 1915.
- 280. Filley, J.; McKinnon, J.T.; Wu, D.T.; Ko, G.H. Macromolecules 2002, 35, 3731.
- 281. Usami, T.; Takayama, S. Polym. J. 1984, 16, 731.
- 282. Blitz, J.P.; McFaddin, D.C. J. Appl. Polvm. Sci. 1994, 51, 13.
- 283. Ohtani, II.; Tsuge, S.; Usami, T. Macromolecules 1984, 17, 2557.
- 284. Bowmer, T.N.; O'Donnell, J.H. Polymer 1977, 18, 1032.
- 285. Cutler, D.J.; Hendra, P.J.; Cudby, M.E.A.; Willis, H.A. Polymer 1977, 18, 1005.
- 286. Usami, T.; Takayama, S. Macromolecules 1984, 17, 1756.
- 287. Axelson, D.E.; Levy, G.C.; Mandelkern, L. Macromolecules 1979, 12, 41.
- 288. Bovey, F.A.; Schilling, F.C.; McCrackin, F.L.; Wagner, H.L. Macromolecules 1976, 9, 76.
- 289. Bugada, D.C.; Rudin, A. Eur. Polym. J. 1987, 23, 809.
- 290. McCord, E.F.; Shaw, W.H.; Hutchinson, R.A. Macromolecules 1997, 30, 246.
- 291. Roedel, M.J. J. Am. Chem. Soc. 1953, 75, 6110.
- 292. Stoiljkovich, D.; Jovanovich, S. Makromol. Chem. 1981, 182, 2811.
- 293. Stoiljkovich, D.; Jovanovich, S. Br. Polym. J. 1984, 16, 291.
- 294. Huang, X.L.; Dannenberg, J.J. J. Org. Chem. 1991, 56, 5421.
- 295. Nedelec, J.Y.; LeFort, D. Tetrahedron 1975, 31, 411.
- 296. Beckwith, A.L.J.; Ingold, K.U. In *Rearrangements in Ground and Excited States*; de Mayo, P., Ed.; Academic Press: New York, 1980; Vol. 1, p 162.
- 297. Wilbourn, A.H. J. Polym. Sci. 1959, 34, 569.
- Randall, J.C.; Buff, C.J.; Keichtermans, M.; Gregory, B.H. Macromolecules 1992, 25, 2624.
- 299. Ketels, H.; Beulen, J.; Vandervelden, G. Macromolecules 1988, 21, 2032.
- 300. Ketels, H.; Dehaan, J.; Aerdts, A.; Vandervelden, G. Polymer 1990, 31, 1419.

- 301. Britton, D.; Heatley, F.; Lovell, P.A. Macromolecules 1998, 31, 2828.
- 302. Starnes, W.H.; Zaikov, V.G.; Chung, H.T.; Wojciechowski, B.J.; Tran, H.V.; Saylor, K.; Benedikt, G.M. *Macromolecules* 1998, 31, 1508.
- 303. Morishima, Y.; Nozakura, S. J. Polym. Sci., Polym. Chem. Ed. 1976, 14, 1277.
- 304. Melville, H.W.; Sewell, P.R. Makromol. Chem. 1959, 32, 139.
- Ahmad, N.M.; Britton, D.; Heatley, F.; Lovell, P.A. Macromol. Symp. 1999, 143, 231.
- 306. Britton, D.; Heatley, F.; Lovell, P.A. Macromolecules 2001, 34, 817.
- 307. Britton, D.; Heatley, F.; Lovell, P.A. Macromolecules 2000, 33, 5048.
- 308. Mattice, W.L.; Viswanadhan, V.N. Macromolecules 1986, 19, 568.
- 309. Hjertberg, T.; Sorvik, E. ACS Symp. Ser. 1985, 280, 259.
- 310. Yamada, B.; Azukizawa, M.; Yamazoe, H.; Hill, D.J.T.; Pomery, P.J. Polymer 2000, 41, 5611.
- 311. Azukizawa, M.; Yamada, B.; Hill, D.J.T.; Pomery, P.J. Macromol. Chem. Phys. 2000, 201, 774.
- 312. Hirano, T.; Yamada, B. Polymer 2003, 44, 347.
- Plessis, C.; Arzamendi, G.; Alberdi, J.M.; Agnely, M.; Leiza, J.R.; Asua, J.M. Macromolecules 2001, 34, 6138.
- 314. Plessis, C.; Arzamendi, G.; Leiza, J.R.; Schoonbrood, H.A.S.; Charmot, D.; Asua, J.M. Ind. Eng. Chem. Res. 2001, 40, 3883.
- Plessis, C.; Arzamendi, G.; Leiza, J.R.; Schoonbrood, H.A.S.; Charmot, D.; Asua, J.M. *Macromolecules* 2000, 33, 5041.
- Plessis, C.; Arzamendi, G.; Leiza, J.R.; Schoonbrood, H.A.S.; Charmot, D.; Asua, J.M. Macromolecules 2000, 33, 4.
- 317. Ahmad, N.M.; Heatley, F.; Lovell, P.A. Macromolecules 1998, 31, 2822.
- 318. Grady, M.C.; Simonsick, W.J.; Hutchinson, R.A. 2002, 182, 149.
- 319. Heatley, F.; Lovell, P.A.; Yamashita, T. Macromolecules 2001, 34, 7636.
- 320. Tanaka, K.; Yamada, B.; Willemse, R.; van Herk, A.M. Polym. J. 2002, 34, 692.
- 321. Sato, T.; Takahashi, H.; Tanaka, H.; Ota, T. J. Polym. Sci., Part A: Polym. Chem. 1988, 26, 2839.
- 322. Sato, T.; Ito, D.; Kuki, M.; Tanaka, H.; Ota, T. Macromolecules 1991, 24, 2963.
- 323. Allen, P.E.M.; Patrick, C.R. *Kinetics and Mechanisms of Polymerization Reactions*; Ellis Horwood: Chichester, 1974.
- Ivin, K.J. In Reactivity, Mechanism and Structure in Polymer Chemistry; Jenkins, A.D.; Ledwith, A., Eds.; Wiley: London, 1974; p 514.
- 325. Ivin, K.J.; Busfield, W.K. In *Encyclopedia of Polymer Science and Engineering*, 2nd ed.; Mark, H.F.; Bikales, N.M.; Overberger, C.G.; Menges, G., Eds.; Wiley: New York, 1987; Vol. 12, p 555.
- 326. Sawada, H. J. Macromol. Sci., Rev. Macromol. Chem. 1969, C3, 313.
- 327. Busfield, W.K. In *Polymer Handbook*, 3rd ed.; Brandup, J.; Immergut, E.H., Eds.; Wiley: New York, 1989; p II/295.
- 328. Ivin, K.J. Trans. Faraday Soc. 1955, 51, 1273.
- 329. Bywater, S. Trans. Faraday Soc. 1955, 51, 1267.
- 330. Cook, R.E.; Ivin, K.J. Trans. Faraday Soc. 1957, 53, 1273.
- 331. Penelle, J.; Collot, J.; Rufflard, G. J. Polym. Sci., Part A: Polym. Chem. 1993, 31, 2407.
- 332. Bywater, S. Can. J. Chem. 1957, 34, 552.
- 333. Bywater, S.; Worsfold, D.J. J. Polym. Sci. 1962, 58, 571.
- 334. Ivin, K.J.; leonard, J. Eur. Polym. J. 1970, 6, 331.
- 335. Yamada, B.; Satake, M.; Otsu, T. Makromol. Chem. 1991, 192, 2713.

Propagation

- 336. Madruga, E.M. In *Macromolecules 1992*; Kahovec, J., Ed.; VSP: Utrecht, 1992; p 109.
- 337. Avci, D.; Kusefoglu, S.H.; Thompson, R.D.; Mathias, L.J. *Macromolecules* 1994, 27, 1981.
- 338. Cheng, J.; Yamada, B.; Otsu, T. J. Polym. Sci., Part A: Polym. Chem. 1991, 29, 1837.
- 339. Tanaka, H. Prog. Polym. Sci. 2003, 28, 1171.
- Stickler, M. In Comprehensive Polymer Science; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: London, 1989; Vol. 3, p 59.
- Stickler, M. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 85.
- 342. Van Herk, A.M. J. Macromol. Sci., Rev. Macromol. Chem. Phys. 1997, 37, 633.
- 343. Beuermann, S.; Buback, M. Prog. Polym. Sci. 2002, 27, 191.
- 344. Kamachi, M.; Yamada, B. In *Polymer Handbook*, 4th ed.; Brandup, J.; Immergut, E.H.; Grulke, E.A., Eds.; John Wiley and Sons: New York, 1999; p II/77.
- 345. Buback, M.; Gilbert, R.G.; Russell, G.T.; Hill, D.J.T.; Moad, G.; O'Driscoll, K.F.; Shen, J.; Winnik, M.A. J. Polym. Sci., Part A: Polym. Chem. 1992, 30, 851.
- 346. Buback, M.; Garcia-Rubio, L.II.; Gilbert, R.G.; Napper, D.II.; Guillot, J.; Hamielec, A.E.; Hill, D.; O'Driscoll, K.F.; Olaj, O.F.; Shen, J.; Solomon, D.II.; Moad, G.; Stickler, M.; Tirrell, M.; Winnik, M.A. J. Polym. Sci., Part C: Polym. Lett. 1988, 26, 293.
- 347. Beuermann, S.; Buback, M.; Davis, T.P.; Gilbert, R.G.; Hutchinson, R.A.; Kajiwara, A.; Klumperman, B.; Russell, G.T. *Macromol. Chem. Phys.* 2000, 201, 1355.
- 348. Beuermann, S.; Buback, M.; Davis, T.P.; Gilbert, R.G.; Hutchinson, R.A.; Olaj, O.F.; Russell, G.T.; Schweer, J.; van Herk, A.M. Macromol. Chem. Phys. 1997, 198, 1545.
- Buback, M.; Gilbert, R.G.; Hutchinson, R.A.; Klumperman, B.; Kuchta, F.-D.; Manders, B.G.; O'Driscoll, K.F.; Russell, G.T.; Schweer, J. *Macromol. Chem. Phys.* 1995, 196, 3267.
- 350. Gilbert, R.G. Pure Appl. Chem. 1996, 68, 1491.
- 351. Gilbert, R.G. Pure Appl. Chem. 1992, 64, 1563.
- 352. Burnett, G.M.; Melville, H.W. Proc. R. Soc., London 1947, A189, 486.
- 353. Fukuda, T.; Ma, Y.-D.; Inagaki, II. Macromolecules 1985, 18, 17.
- Olaj, O.F.; Kremminger, P.; Schnöll-Bitai, I. Makromol. Chem., Rapid Commun. 1988, 9, 771.
- 355. Olaj, O.F.; Schnöll-Bitai, I.; Kremminger, P. Eur. Polym. J. 1989, 25, 535.
- 356. O'Driscoll, K.F.; Mahabadi, H.K. J. Polym. Sci., Polym. Chem. Ed. 1976, 14, 869.
- Bresler, S.E.; Kazbekov, E.N.; Fomichev, V.N.; Shadrin, V.N. Makromol. Chem. 1972, 157, 167.
- 358. Bresler, S.E.; Kazbekov, E.N.; Shadrin, V.N. Makromol. Chem. 1974, 175, 2875.
- 359. Kamachi, M. J. Polym. Sci., Part A: Polym. Chem. 2001, 40, 269.
- Yamada, B.; Westmoreland, D.G.; Kobatake, S.; Konosu, O. Prog. Polym. Sci. 1999, 24, 565.
- Carswell, T.G.; Hill, D.J.T.; Londero, D.I.; O'Donnell, J.H.; Pomery, P.J.; Winzor, C.L. *Polymer* **1992**, *33*, 137.
- 362. Carswell, T.G.; Hill, D.J.T.; Hunter, D.S.; Pomery, P.J.; O'Donnell, J.H.; Winzor, C.L. Eur. Polym. J. 1990, 26, 541.
- 363. Kamachi, M.; Kohno, M.; Kuwae, Y.; Nozakura, S.-I. Polym. J. 1982, 14, 749.
- 364. Shen, J.; Tian, Y.; Wang, G.; Yang, M. Makromol. Chem. 1991, 192, 2669.

- 365. Zhu, S.; Tian, Y.; Hamielec, A.E.; Eaton, D.R. Macromolecules 1990, 23, 1144.
- 366. Tonge, M.P.; Pace, R.J.; Gilbert, R.G. Macromol. Chem. Phys. 1994, 195, 3159.
- 367. Yamada, B.; Kageoka, M.; Otsu, T. Polym. Bull. 1992, 28, 75.
- 368. Yamada, B.; Kageoka, M.; Otsu, T. Macromolecules 1991, 24, 5234.
- 369. Kamachi, M.; Kuwae, Y.; Kohno, M.; Nozakura, S.-I. Polym. J. 1985, 17, 541.
- 370. Aleksandrov, H.P.; Genkin, V.N.; Kitai, M.S.; Smirnova, I.M.; Solokov, V.V. Sov. J. Quantum Electron. 1977, 7, 547.
- 371. Olaj, O.F.; Bitai, I.; Hinkelmann, F. Makromol. Chem. 1987, 188, 1689.
- 372. O'Driscoll, K.F.; Kuindersma, M.E. Macromol. Theory Simul. 1994, 3, 469.
- 373. Lu, J.; Zhang, H.; Yang, Y. Makromol. Chem., Theory Simul. 1993, 2, 747.
- 374. Deady, M.; Mau, A.W.H.; Moad, G.; Spurling, T.H. Makromol. Chem. 1993, 194, 1691.
- 375. Hutchinson, R.A.; Richards, J.R.; Aronson, M.T. Macromolecules 1994, 27, 4530.
- 376. Buback, M.; Huckestein, B.; Kuchta, F.-D.; Russell, G.T.; Schmid, E. Macromol. Chem. Phys. 1994, 195, 2117.
- 377. Buback, M.; Kurz, C.H.; Schmaltz, C. Macromol. Chem. Phys. 1998, 199, 1721.
- 378. Lyons, R.A.; Hutovic, J.; Piton, M.C.; Christie, D.I.; Clay, P.A.; Manders, B.G.; Kable, S.H.; Gilbert, R.G. Macromolecules 1996, 29, 1918.
- Hutchinson, R.A.; Paquet, D.A., Jr.; McMinn, J.H. DECHEMA Monogr. 1995, 131, 467.
- Deibert, S.; Bandermann, F.; Schweer, J.; Sarnecki, J. Makromol. Chem., Rapid Commun. 1992, 13, 351.
- 381. Beuermann, S.; Paquet, D.A., Jr.; McMinn, J.H.; Hutchinson, R.A. *Macromolecules* 1997, 30, 194.
- Hutchinson, R.A.; Beuermann, S.; Paquet, D.A., Jr; McMinn, J.H. Macromolecules 1997, 30, 3490.
- 383. Buback, M.; Kurz, C.H. Macromol. Chem. Phys. 1998, 199, 2301.
- 384. Shipp, D.A.; Smith, T.A.; Solomon, D.H.; Moad, G. Macromol. Rapid Commun. 1995, 16, 837.
- 385. Yee, L.H.; Coote, M.L.; Chaplin, R.P.; Davis, T.P. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 2192.
- 386. Tanaka, K.; Yamada, B.; Fellows, C.M.; Gilbert, R.G.; Davis, T.P.; Yee, L.H.; Smith, G.B.; Rees, M.T.L.; Russell, G.T. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 3902.
- 387. Starks, C.M. Free Radical Telomerization; Academic Press: New York, 1974.
- 388. Fischer, H.; Radom, L. Angew. Chem., Int. Ed. Engl. 2001, 40, 1340.
- 389. Morrison, B.R.; Maxwell, I.A.; Gilbert, R.G.; Napper, D.H. ACS Symp. Ser. 1992, 492, 28.
- 390. Olaj, O.F.; Vana, P.; Zoder, M. Macromolecules 2002, 35, 1208.
- 391. Zetterlund, P.B.; Busfield, W.K.; Jenkins, I.D. Macromolecules 1999, 32, 8041.
- Moad, G.; Rizzardo, E.; Solomon, D.H.; Beckwith, A.L.J. Polym. Bull. 1992, 29, 647.
- 393. Krstina, J.; Moad, G.; Willing, R.I.; Danek, S.K.; Kelly, D.P.; Jones, S.L.; Solomon, D.H. Eur. Polym. J. 1993, 29, 379.
- 394. Gridnev, A.A.; Ittel, S.D. Macromolecules 1996, 29, 5864.
- 395. Zetterlund, P.B.; Busfield, W.K.; Jenkins, I.D. Macromolecules 2002, 35, 7232.

5.1 Introduction

In this chapter we consider reactions that lead to the cessation of growth of one or more polymer chains. Three processes will be distinguished:

(a) The self-reaction of propagating radicals by combination and/or disproportionation (e.g. Scheme 5.1) (Section 5.2).





(b) Primary radical termination (Sections 3.2.9, 3.4, 5.2.2.1 and 7.4.3); the reaction of a propagating radical with an initiator-derived (I•, Scheme 5.2) or transfer agent-derived radical. The significance of this process is highly dependent on the structure of the radical (I•).



(c) Inhibition (Section 5.3); the reaction of a propagating radical with another species (Z•, Scheme 5.3) to give a dead polymer chain. Z• is usually of low molecular weight. Examples of inhibitors are "stable" radicals (e.g. nitroxides,

oxygen), non-radical species that react to give "stable" radicals (*e.g.* phenols, quinones, nitroso-compounds) and transition metal salts.

Scheme 5.3

Chain transfer, the reaction of a propagating radical with a non-radical substrate to produce a dead polymer chain and a new radical capable of initiating a new polymer chain, is dealt with in Chapter 6. There are also situations intermediate between chain transfer and inhibition where the radical produced is less reactive than the propagating radical but still capable of reinitiating polymerization. In this case, polymerization is slowed and the process is termed retardation or degradative chain transfer. The process is mentioned in Section 5.3 and, when relevant, in Chapter 6.

5.2 Radical-Radical Termination

The most important mechanism for the decay of propagating species in radical polymerization is radical-radical reaction by combination or disproportionation as shown in Scheme 5.1. This process is sometimes simply referred to as bimolecular termination. However, this term is misleading since most chain termination processes are bimolecular reactions.

Before any chemistry can take place the radical centers of the propagating species must come into appropriate proximity and it is now generally accepted that the self-reaction of propagating radicals is a diffusion-controlled process. For this reason there is no single rate constant for termination in radical polymerization. The average rate constant usually quoted is a composite term that depends on the nature of the medium and the chain lengths of the two propagating species. Diffusion mechanisms and other factors that affect the absolute rate constants for termination are discussed in Section 5.2.1.4.

Even though the absolute rate constant for reactions between propagating species may be determined largely by diffusion, this does not mean that there is no specificity in the termination process or that the activation energies for combination and disproportionation are zero or the same. It simply means that this chemistry is not involved in the rate-determining step of the termination process.

The relative importance of combination and disproportionation in relevant model systems and in polymerizations of some common monomers is considered in Sections 5.2.2.1 and 5.2.2.2 respectively. The significance of the termination mechanism on the course of polymerization and on the properties of polymers is discussed briefly in Section 5.2.2 and is further discussed in Section 8.2.

5.2.1 Termination Kinetics

A detailed treatment of termination kinetics is beyond the scope of this book. However, some knowledge is important in understanding the chemistry described in subsequent sections. There are a number of reviews of the kinetics of radicalradical termination of propagating species. Those by North¹ and O'Driscoll² provide a useful background. Significant advances in our knowledge of termination kinetics came with the development of pulsed laser methods. Recent reviews include those by Buback *et al.*,³ Russell⁴⁻⁷ and de Kock *et al.*^{8.9} Many of the issues surrounding termination have been summarized by one IUPAC working party.¹⁰⁻¹² Values of, and methods of determining, termination rate constants are currently being critically assessed by another working party.³

In Section 5.2.1.1 we provide an overview of the classical treatment of polymerization kinetics. Some aspects of termination kinetics are not well understood and no wholly satisfactory unified description is in place, Nonetheless, it remains a fact that many features of the kinetics of radical polymerization can be predicted using a very simple model in which radical-radical termination is characterized by a single rate constant. The termination process determines the molecular weight and molecular weight distribution of the polymer. In section 5.2.1.2, we define the terminology used in describing molecular weights and molecular weight distributions. In Section 5.2.1.3, we provide a simple statistical treatment based on classical kinetics and discuss the dependence of the molecular weight distribution on the termination process. Some of the complexities of termination associated with diffusion control and the dependence on chain length and on conversion are described in Section 5.2.1.4.

Termination in heterogeneous polymerization is discussed in Section 5.2.1.5 and the more controversial subject of termination during living radical polymerization is described in Section 5.2.1.6. Termination in copolymerization is addressed in Section 7.3.

5.2.1.1 Classical kinetics

The overall rate constant for radical-radical termination can be defined in terms of the rate of consumption of propagating radicals. Consider the simplified mechanism for radical polymerization shown in Scheme 5.4.

Ideally, as long as the rate constants for reinitiation (k_{iT}, k_{iM}) are high with respect to that for propagation (k_p) , the transfer reactions should not directly affect the rate of polymerization and they need not be considered further in this section. The overall rate constant for radical-radical termination (k_i) can be defined in terms of the rate of consumption of propagating radicals as shown in eq. 1:

$$R_{t} = -2k_{t}[\mathbf{P}\bullet]^{2} \tag{1}$$

where [P•] is the total concentration of propagating radicals and $k_i = k_{tc} + k_{td}$.

In many works on radical polymerization, the factor 2 is by convention incorporated into the rate constant.^{13,14} In this case $R_t = -k_t [\mathbf{P} \cdot]^2$. The termination rate constant is then sometimes expressed as $k_t = k_t c/2 + k_{td}$ to reflect the fact that only one polymer chain is formed when two propagating radicals combine whilst two are formed in disproportionation. In reading the literature and when comparing values of k_t , care must be taken to establish which definitions have been used.² In accord with the current IUPAC recommendation,¹⁵ in the following discussion, eq. 1 and $k_t = k_{tc} + k_{td}$ are used.

| initiation | | | | |
|-----------------------------|---------------|---|---|---|
| I_2 | \rightarrow | 2 I• | $R_{\rm i} - 2k_{\rm d} / [{ m I}_2]$ | |
| $I \bullet + M$ | \rightarrow | $P_1 \bullet$ | $k_{\rm i} \ge k_{\rm p}$ | |
| propagation | n | | | |
| $P_n \bullet + M$ | \rightarrow | P _{n+1} ● | $R_{p} = k_{p}[M][P\bullet]$ | |
| termination | by d | isproportionation | | |
| $P_n \bullet + P_m$ | • → | $\mathbf{P}_{n}^{H} + \mathbf{P}_{m}$ | $R_{td}=2k_{td}[P\bullet]^2$ | $R_{\rm t} = R_{\rm tc} + R_{\rm td}$ |
| termination | by co | ombination | | |
| $P_n \bullet + P_m \bullet$ | • → | P_{n+m} | $R_{\rm tc}=2k_{\rm tc}[{\rm P}\bullet]^2$ | |
| termination | by ci | hain transfer | | |
| $P_n \bullet + I_2$ | \rightarrow | $P_n + I \bullet$ | $R_{trl} = k_{trl} [I_2] [P \bullet]$ | $R_{\rm tr} = R_{\rm tr,I} + R_{\rm tr,M} + R_{\rm tr,T}$ |
| $P_n \bullet + M$ | \rightarrow | $\mathbf{P}_{n} + \mathbf{P}_{1} \bullet$ | $R_{trM} = k_{trM}[M][P\bullet]$ | |
| P _n • ∣ T | ~ | $P_n + T \bullet$ | $R_{\text{trT}} = k_{\text{trT}}[T][P^{\bullet}]$ | |
| M∙ + M | \rightarrow | P₂• | $k_{\rm iM} \ge k_{\rm p}$ | |
| $T \bullet + M$ | \rightarrow | $P_1 \bullet$ | $k_{\rm iT} > k_{\rm p}$ | |

Scheme 5.4

Application of a steady state approximation (that $R_t = R_i$, eq. 2) and a long chain approximation (negligible monomer consumption in the initiation or reinitiation steps) provides a number of useful relationships.

$$\frac{-\mathbf{d}[\mathbf{P}\bullet]}{\mathbf{d}t} = R_{i} - R_{i} = 2k_{d}f[\mathbf{I}_{2}] - 2k_{t}[\mathbf{P}\bullet]^{2} = 0$$
⁽²⁾

(a) The total concentration of propagating radicals ([P•]) (eq. 3):

$$[\mathbf{P}\bullet] = \left(\frac{k_{\rm d}f}{k_{\rm l}}\right)^{0.5} [\mathbf{I}_2]^{0.5}$$
(3)

(b) The mean lifetime of a propagating radical (τ) (eq. 4):

$$\boldsymbol{\tau} = \left(2k_{\rm d}f[\mathbf{I}_2]k_{\rm t}\right)^{-0.5} \tag{4}$$

(c) The average kinetic chain length (\bar{v}) (eq. 5):

$$\overline{v} = \frac{R_{p}}{R_{i}} = \frac{R_{p}}{R_{i}} = \frac{k_{p}[M]}{\left(2k_{d}f[I_{2}]k_{i}\right)^{0.5}}$$
(5)

(d) The number average degree of polymerization in the absence of chain transfer (eq. 6):

$$\overline{X}_{n} = \frac{k_{p}[M]}{\left(1 + \frac{k_{td}}{k_{t}}\right) \left(2k_{d} f[I_{2}]k_{t}\right)^{0.5}}$$
(6)

(e) The initiator efficiency (eq. 7):

$$f = \left(1 + \frac{k_{\rm td}}{k_{\rm t}}\right) \frac{R_{\rm p}}{\overline{X}_{\rm n} k_{\rm d} [\rm I_2]} \tag{7}$$

It also enables elimination of the radical concentration in the expression for rate of polymerization (eq. 8):

$$R_{\rm p} = \frac{-\mathbf{d}[\mathbf{M}]}{\mathbf{d}t} - k_{\rm p}[\mathbf{P}\bullet][\mathbf{M}]$$
$$= k_{\rm p} \left(\frac{k_{\rm d}f}{k_{\rm t}}\right)^{0.5} [\mathbf{I}_2]^{0.5} [\mathbf{M}]$$
(8)

In eq. 8, the rate of polymerization is shown as being half order in initiator (I_2) . This is only true for initiators that decompose to two radicals both of which begin chains. The form of this term depends on the particular initiator and the initiation mechanism. The equation takes a slightly different form in the case of thermal initiation (S), redox initiation, diradical initiation, etc. Side reactions also cause a departure from ideal behavior.

Eq. 8 can be recast in terms of the fractional conversion of monomer to polymer as in eq. 9:

$$-\frac{\mathrm{dln}([M]/[M]_{o})}{\mathrm{d}t} = \left(\frac{k_{d} f k_{p}^{2}}{k_{u}}\right)^{0.5} [I_{2}]^{0.5}$$
(9)

From this we can see that knowledge of $k_d f$ and R_p in a conventional polymerization process readily yields a value of the ratio k_p^{-2}/k_t . In order to obtain a value for k_t we require further information on k_p . Analysis of R_p data obtained under non-steady state conditions (when there is no continuous source of initiator radicals) yields the ratio k_p/k_t . Various non-steady state methods have been developed including the rotating sector method, spatially intermittent polymerization and pulsed laser polymerization (PLP). The classical approach for deriving the individual values of k_p and k_t by combining values for k_p^{-2}/k_t . with k_p/k_t obtained in separate experiments can, however, be problematical because the values of k_t are strongly dependent on the polymerization conditions (Section 5.2.1.4). These issues are thought to account for much of the scatter apparent in literature values of k_t .^{3.16} PLP and related methods yield absolute values of k_p directly (the methods used for extracting k_p are discussed in Section 4.5.2). These values may be combined with either k_p^{-2}/k_t or k_p/k_t to give k_t . The SP-PLP^{8.17.18} and PS-PLP^{17.19} techniques involve following the monomer

The SP-PLP^{8,17,18} and PS-PLP^{17,19} techniques involve following the monomer conversion induced by a single laser pulse or a sequence of laser pulses. These experiments are usually conducted at high pressure because rates of termination are lower and sensitivities are somewhat higher.¹⁷

EPR methods can be used to determine the radical concentration [P•] either directly^{20,21} or *via* trapping methods.²² Fluorescence experiments have also been designed to give [P•] for a particular conversion.²³⁻²⁵ Given [P•] and the rate of polymerization, k_p can be evaluated using eq. 8. Given the rate of initiation and [P•], k_t can be calculated using eq. 3.^{20,21,26} It is also possible to estimate k_t from the molecular weight distributions given k_p and [P•] using kinetic simulation.^{24,25}

For low conversions, values of the rate constants k_t for monosubstituted monomers (S and acrylates) are ~10⁸ M⁻¹s⁻¹ and those for methacrylates are ~10⁷ M⁻¹s⁻¹ and activation energies are small and in the range 3-8 kJ mol^{-1,17} These activation energies relate to the rate-determining diffusion process (Section 5.2.1.4) rather than to radical-radical coupling.

Values of termination constants for sterically hindered monomers may be several orders of magnitude lower than those for S and (methacrylates). Such monomers include various α -substituted methacrylates, itaconates, fumarates, and N-substituted itaconimides and maleimides. Values of k_t for these monomers have been reported to lie in the range 10-10⁵ M⁻¹s⁻¹ depending on the particular structure.²⁰

5.2.1.2 Molecular weights and molecular weight averages

The degree of polymerization of a polymer (X_i) is equal to the chain length *i* (the number of monomer units in the chain). If we neglect end groups,* the number molecular weight (M_n) is given by eq. 10:

^{*} By definition, the molar mass of the end groups should be included in the molecular weight of a polymer but the corresponding quantity is not included in the degree of polymerization.

239

$$M_{\rm i} = X_{\rm i} M_0 \tag{10}$$

where M_0 is the molecular weight or molar mass of the monomer or repeat unit.[†]

The number average molecular weight (M_n) is the average molecular weight of all of the polymer chains that make up a sample and is given by eq. 11:

$$\overline{M}_{n} = \frac{\sum n_{i} X_{i}}{\sum n_{i}} M_{0}$$
(11)

where n_i is the concentration of chains of length *i* (monomer units)

The weight average molecular weight (\overline{M}_{w}) is given by eq. 12:

$$\overline{M}_{w} = \frac{\sum w_{i} X_{i}}{\sum w_{i}} M_{0} = \frac{\sum n_{i} X_{i}^{2}}{\sum n_{i} X_{i}} M_{0}$$
(12)

where w_i is the weight of chains of length *i*.

i

The Z average molecular weight (\overline{M}_{1}) is provided by eq. 13:

$$M_{z} = \frac{\sum n_{i} X_{i}^{3}}{\sum n_{i} X_{i}^{2}} M_{0}$$
(13)

This term gives some information about the asymmetry of the molecular weight distribution and is important in analyzing sedimentation behavior in ultracentrifugation.

It is also useful to define the moments of the chain length distribution. The *j*th moment is defined in eq. 14:

$$\lambda^{j} = \sum n_{i} X_{i}^{j} \tag{14}$$

The zeroth moment $\lambda^0 = \sum n_i$ can be recognized as the total concentration of polymer chains and the first moment $\lambda^1 = \sum n_i X_i = \sum w_i$ is the total concentration of repeat or monomer units in those chains. The moments can be related to the molecular weight averages as follows:

$$\overline{M}_{n} = \frac{\lambda^{1}}{\lambda^{0}} M_{0}, \quad \overline{M}_{n} = \frac{\lambda^{2}}{\lambda^{1}} M_{0}, \quad \overline{M}_{z} = \frac{\lambda^{3}}{\lambda^{2}} M_{0}$$

The breadth of the molecular weight distribution is often discussed in terms of the dispersity $(D)^*$ and is expressed in terms of the moments as shown in eq. 15:

[†] In this book, in accord with common usage, we use the term molecular weight rather than molar mass when referring to polymers.

^{*} The dispersity is also commonly called the polydispersity index or the polydispersity.

$$D = \frac{\overline{X}_{u}}{\overline{X}_{n}} = \frac{\overline{M}_{u}}{\overline{M}_{n}} = \frac{\lambda^{0} \lambda^{1}}{\left(\lambda^{1}\right)^{2}}$$
(15)

In calculations the moments can be treated as concentrations. Kinetic simulation of radical polymerization to evaluate dispersities typically involves evaluation of the moments rather than the complete distribution. This method of moments is accurate as long as the kinetics are independent of chain length.

5.2.1.3 Molecular weight distributions

The simple statistical treatment of radical polymerization can be traced back to Schultz.²⁷ Texts by $Flory^{28}$ and Bamford *et al.*²⁹ are useful references.

The probability of a propagation event (ϕ) can be defined as shown in eq. 16:

$$\phi = \frac{R_{\rm p}}{R_{\rm p} + R_{\rm t} + R_{\rm tr}}$$

$$= \frac{k_{\rm p}[M]}{k_{\rm p}[M] + 2k_{\rm t}[P^{\bullet}] + k_{\rm tr}[I_2] + k_{\rm tr}[M] + k_{\rm tr}[T]}$$
(16)

A given chain will undergo *i*-1 propagation steps (each with probability ϕ) before terminating (with probability 1- ϕ). Thus, if termination is wholly by chain transfer or disproportionation, the chain length distribution is given by eq. 17 (Figure 5.1):

$$n_i = \phi^{i-1}(1-\phi) \tag{17}$$

This distribution is known as the Schultz-Flory or most probable distribution.²⁸ The moments of the molecular weight distribution are:

$$\lambda^0 = 1, \ \lambda^1 = (1 - \phi)^{-1}, \ \lambda^2 = (1 + \phi)(1 - \phi)^{-1}$$

and the average degrees of polymerization and dispersity are:

$$\overline{X}_{w} = \frac{1}{1-\phi}, \ \overline{X}_{w} = \frac{1+\phi}{1-\phi} \text{ and } D = \frac{X_{w}}{\overline{X}_{w}} = 1+\phi$$

and for long chains as $\phi \rightarrow 1, D \rightarrow 2$.

If termination is wholly by combination it can be shown²⁹ that the number distribution is given by eq. 18 (Figure 5.1):

$$n_i = (i-1)(1-\phi)^2 \phi^{i-2}$$
(18)

The moments of the molecular weight distribution are:

$$\lambda^0 = 1, \ \lambda^1 = 2(1-\phi)^{-1}, \ \lambda^2 = (4+2\phi)(1-\phi)^{-2}$$

240

and the average degrees of polymerization and dispersity are:

$$\overline{X}_n = \frac{2}{1-\phi}, \ \overline{X}_w = \frac{2+\phi}{1-\phi} \text{ and } D = \frac{\overline{X}_w}{\overline{X}_n} = \frac{2+\phi}{2}$$

The molecular weight distribution in this case is significantly narrower. For long chains as $\phi \rightarrow 1$ so $D \rightarrow 1.5$.



Figure 5.1 (a) Number and (b) GPC distributions for two polymers both with $\overline{X}_n = 100$. The number distribution of chains formed by disproportionation or chain transfer (-----, $\sum n_i = 1.0$, $\overline{X}_w / \overline{X}_n = 2.0$) is calculated using eq. 17. The number distribution of chains formed by combination (----, $\sum n_i = 1.0$, $\overline{X}_w / \overline{X}_n = 1.5$) is calculated using eq. 18.

For the more general case, the molecular weight distribution will be described by a weighted average of eqs. 17 and 18 (eq. 19):

$$n_{i} = \frac{R_{tc}}{R_{t} + R_{tr}} (i-1)(1-\phi)^{2} \phi^{i-2} + \frac{R_{td} + R_{tr}}{R_{t} + R_{tr}} \phi^{i-1}(1-\phi)$$
(19)

These equations predict that for oligomers with degree of polymerization less than 10, polydispersities significantly less than 1.5 will be obtained - Figure 5.2.

The above treatment only applies to polymerizations where there is negligible conversion of monomer, initiator, and transfer agents. Analytical treatments have been devised to take into account effects of conversion and more complex mechanisms. Discussion of these is beyond the scope of this book.

A common error is to confuse the GPC distribution with the weight distribution. The response of a refractive index detector is proportional to the mass of polymer. The GPC elution volume (V) typically scales according to the logarithm of the degree of polymerization (or the logarithm of the molecular

weight). Thus, $V \sim a+b \log i$ (where a and b are constants) and a volume increment (dV) will be proportional to di/i. It follows that the y-axis of the GPC distribution (e.g. Figure 5.1b) is proportional to iw_i or i^2n_i .



Figure 5.2 Dispersity (*D*) as a function of \overline{X}_n for polymers formed by (a) disproportionation or chain transfer (——) and (b) combination (-----).

5.2.1.4 Diffusion controlled termination

Termination by self-reaction of propagating radicals is a diffusion-controlled process even at very low conversion.³ The evidence for this includes the following:

- (a) Analogy with the known chemistry of small radicals. The rate constants for self-reaction of small radicals approach the diffusion-controlled limit and rate constants can be predicted using the Smoluchowski equation.
- (b) The value of k_t shows an inverse dependence on medium viscosity as anticipated for a diffusion controlled reaction.
- (c) The value of k_t decreases with increasing pressure (positive activation volume). For a reaction involving the combination of two species, the activation volume is expected to be negative.

However, while it is generally accepted that the rate of radical-radical reaction is dependent on how fast the radical centers of the propagating chains (P_i^{\bullet} and P_j^{\bullet}) come together, there remains some controversy as to the diffusion mechanism(s) and/or what constitutes the rate-determining step in the diffusion process. The steps in the process as postulated by North and coworkers³⁰⁻³² are shown conceptually in Scheme 5.5.



Scheme 5.5

Center of mass or translational diffusion is believed to be the rate-determining step for small radicals³³ and may also be important for larger species. However, other diffusion mechanisms are operative and are required to bring the chain ends together and these will often be the major term in the termination rate coefficient for the case of macromolecular species. These include:

- (a) Segmental motion. The internal reorganization of the chain required to bring the reactive ends together.
- (b) Reptation. The snaking of the chain through a viscous medium.
- (c) Reaction diffusion (also called residual termination). Chain end motion by addition of monomer to the chain end.

The relative importance of these mechanisms, and the value of the overall k_t , depends on the molecular weight and dispersity of the propagating species, the medium and the degree of conversion. The value of k_t is not a constant!



Figure 5.3 Conversion-time profile for bulk MMA polymerization at 50 °C with AIBN initiator illustrating the three conversion regimes. Data are taken from Balke and Hamielec.³⁴

In dealing with radical-radical termination in bulk polymerization it is common practice to divide the polymerization timeline into three or more conversion regimes.^{2,35} The reason for this is evident from Figure 5.3. Within each regime, expressions for the termination rate coefficient are defined according to the dominant mechanism for chain end diffusion. The usual division is as follows:

- (a) Low conversion prior to the onset of the autoacceleration phenomenon known as the gel or Norrish-Trommsdorff effect³⁶⁻³⁸ and characterized by highly mobile propagating species. Center of mass and/or segmental diffusion are the rate-determining mechanisms for chain end movement. Initiator efficiencies are high and approximately constant.
- (b) Medium to high conversion immediately after the onset of the gel effect. The diffusion mechanism is complex. Large chains become effectively immobile (on the timescale of the lifetime of a propagating radical) even though the chain ends may move by segmental diffusion, reptation or reaction diffusion. Monomeric species and short chains may still diffuse rapidly. Short-long termination dominates. Initiator efficiencies may reduce with conversion.
- (c) Very high conversion the polymerization medium is a glassy matrix. Most chains are immobile and reaction diffusion is the rate-determining diffusion mechanism. New chains are rapidly terminated or immobilized. Initiator efficiencies are very low.

The precise conversion ranges are determined by a variety of factors including the particular monomer, the molecular weight of the polymeric species and the solvent (if any). For bulk polymerization of S and MMA (a) is typically <20%, (b) is 20-85% and (c) is >85%. In solution polymerization, or for polymerizations carried out in the presence of chain transfer agents, the duration of the low conversion regime is extended and the very high conversion regime may not occur. Cage escape is also a diffusion controlled process, thus the initiator efficiency (*f*) and the rate of initiation ($k_{d}f$) generally decrease with conversion and depend on the conversion regime as indicated above (Sections 3.2.8, 3.3.1.1.3, 3.3.2.1.3. 3.3.2.4).

5.2.1.4.1 Termination at low conversion

Most in depth studies of termination deal only with the low conversion regime. Logic dictates that simple center of mass diffusion and overall chain movement by reptation or many other mechanisms will be chain length dependent. At any instant, the overall rate coefficient for termination can be expressed as a weighted average of individual chain length dependent rate coefficients (eq. 20):³⁹

$$k_{t} = \frac{\sum_{i=1}^{\infty} \sum_{j=1}^{\infty} k_{t}^{i,j} [\mathbf{P}_{i} \bullet] [\mathbf{P}_{j} \bullet]}{[\mathbf{P} \bullet]^{2}}$$
(20)

where $k_t^{i,j}$ is the rate coefficient for reaction between species of chain lengths i and j, and [P•] is the total radical concentration.

Mahabadi and O'Driscoll³⁹ considered that segmental motion and center of mass diffusion should be the dominant mechanisms at low conversion. They analyzed data for various polymerizations and proposed that $k_t^{i,j}$ should be dependent on chain length such that the overall rate constant obeys the expression:

$$k_1 \propto \overline{X_n}^{-\alpha} \tag{21}$$

where \overline{X}_n is the number average degree of polymerization and $\alpha - 0.5$ for short \overline{X}_n reducing to 0.1 for large \overline{X}_n .

Various expressions have been proposed for estimating how the overall rate coefficient k_t and the individual rate coefficients $k_t^{i,j}$ vary with the chain lengths of the reacting species, $^{2,39-46}_{2,42,46,47}$ simple relationships of the following forms are the most often applied: $^{32,42,46,47}_{32,42,46,47}$

(a) The harmonic mean is said to be of the functional form expected if chain end encounter or coil overlap is rate-determining:

$$k_{\rm t}^{\rm i,j} = k_{\rm to} \left(\frac{2i.j}{i+j}\right)^{-\alpha} \tag{22}$$

(b) The Smoluchowski mean is of the functional form expected if translational diffusion is rate-determining; it is known to provide a reasonable description of the termination kinetics of small radicals:

$$k_{\rm L}^{\rm i,j} = 0.5 k_{\rm to} (i^{-\alpha} + j^{-\alpha}) \tag{23}$$

or:

$$k_{t}^{i,j} = 2\pi \sigma p_{\text{spin}}(D^{i} + D^{j}) \tag{24}$$

where σ is a capture radius, p_{spin} is a spin multiplicity term, and D' and D' are chain length dependent diffusion constants. When $\alpha=1$, the Smoluchowski mean and the harmonic mean approximations are the same

(c) The geometric mean has no physical basis but has been suggested to best approximate the functional form of the segmental diffusion process:

$$k_{\rm r}^{\rm i,j} = k_{\rm to}(\rm i,j)^{-\alpha/2} \tag{25}$$

where α and k_{to} are constants.

245

While many data are suggestive of chain length dependence, the data are not usually suitable for or have not been tested with respect to model discrimination. Values of $k_t^{1,1}$ have been determined for a variety of small "monomeric" radicals to be *ca* 10⁹ M⁻¹ s^{-1.48} Taking k_{to} as $k_t^{1,1}$ and α as 1.0 in the geometric expression yields values of $k_t^{1,j}$ as shown in Figure 5.4a.⁴⁹ Use of the Smoluchowski mean or the harmonic mean approximation predicts a shallower dependence of $k_t^{1,j}$ on the chain length (Figure 5.4b). All expressions yield the same dependence for j=i.



Figure 5.4 Chain length dependence of $k_t^{i,j}$ predicted by (a) the geometric mean (eq, 25) or (b) the harmonic mean approximation (eq. 22) or the Smoluchowski mean (eq. 23) with α =1.0 and k_{to} =10⁹; i and j are the lengths of the reacting chains.

However, it has been pointed out that the value of k_{to} in the expressions eqs. 25-23 should not be confused with the small radical $k_t^{1,1}$, rather, the value of k_{to} represents the termination rate constant of a single unit chain if the implied diffusion mechanism was the rate-determining process.

Recent work has allowed values of $k_t^{i,i}$ and α for bulk polymerization in dilute solution to be estimated. This work suggests values of $k_{to} = k_t^{i,i} \sim 1 \times 10^8 \text{ M}^{-1}$ and $\alpha \sim 0.15$ -0.25 for both MMA and S.^{17,50} Some values of $k_t^{i,i}$ and α for S and methacrylates estimated from SP-PLP at high pressure experiments are shown in Table 5.1.

The value of the exponent α obtained in the above-mentioned experiments is in remarkable accord with predictions based on a consideration of excluded kinetic volume effects. Khokhlov⁵¹ proposed, that for a slow, chemically controlled, reaction between the ends of long chains α should be 0.16. The value of α was suggested to increase to 0.28 for chain end-mid chain reaction and to 0.43 for midchain-mid chain reaction. The latter provides one possible explanation for the greater exponent for higher acrylates (Table 5.1).⁵²

246

| Monomer | T (°C) | P (bar) | $k_{\rm p} ({\rm M}^{-1}{\rm s}^{-1})$ | $k_{10} (M^{-1} s^{-1})$ | α | ref |
|---------|--------|---------|--|--------------------------|------|-----|
| S | 40 | 1000 | 1600 | 7×10^{7} | 0.16 | 18 |
| MMA | 40 | 2000 | 1700 | 4×10^{7} | 0.14 | 52 |
| DMA | 40 | 1000 | 1400 | 3×106 | 0.15 | 52 |
| MA | 40 | 1000 | 28600 | 2×10^{8} | 0.15 | 52 |
| BA | 40 | 1000 | 35600 | 6×10^{7} | 0.14 | 52 |
| DA | 40 | 1000 | 39800 | 8×10 ⁷ | 0.43 | 52 |

 Table 5.1 Parameters Characterizing Chain Length Dependence of Termination

 Rate Coefficients in Radical Polymerization of Common Monomers^a

a Determined by the SP-PLP technique. Values apply to bulk polymerization at low conversion (up to 15% conversion).

For the situation where the chain length of one or both of the species is "small" (not entangled with itself or other chains) and conversion of monomer to polymer is low, the termination kinetics should be dominated by the rate of diffusion of the shorter chain. While the chain remains short, the time required for the chain reorganization to bring the reacting centers together will be insignificant and center of mass diffusion can be the rate-determining step. As the chain becomes longer, segmental diffusion will become more important. Thus, it is expected that $k_i^{(4)}$ should lie between an upper limit predicted by the Smoluchowski mean (eq. 23) and a lower limit predicted by the geometric mean (eq. 25) with the value being closer to the geometric mean value for higher chain lengths as shown in Figure 5.5.

Smith *et al.*⁵⁰ have recently suggested a composite model based on similar considerations to predict $k_1^{1,j}$ over the entire chain length range. Experimental data for $k_1^{1,j}$ for dodecyl methacrylate polymerization consistent with such a model have been provided by Buback *et al.*⁵³

Since shorter, more mobile, chains diffuse more rapidly (by center of mass diffusion or other mechanisms), they are more likely to be involved in termination. For this reason, most termination involves reaction of a long species with a short species. The lower mobility of long chains ensures that they are unlikely to react with each other. Cardenas and O'Driscoll⁵⁴ proposed that propagating species be considered as two populations; those with chain length below the entanglement limit and those above. This basic concept has also been adopted by other authors.^{24,55-58} Russell⁵⁵ has provided a detailed critique of these concepts. Direct experimental evidence for the importance of the dispersity of the propagating radicals on termination kinetics has been reported by Faldi *et al.*⁵⁶ O'Neil and Torkelson questioned the chain entanglement concept pointing out that for low conversions chain entanglements are unlikely even for chain lengths >100.



Figure 5.5 Chain length dependence of $k_t^{i,j}$ predicted by the Smoluchowski mean (eq. 23) with α =0.5 and k_{to} =10⁹ (upper series) and the geometric mean (eq. 25) with α =0.2 and k_{to} =10⁸ (lower series); i and j are the lengths of the reacting chains. For low conversions, $k_t^{i,j}$ is expected to lie between the values predicted by eqs. 23 and 25 (see text).

For larger species, even though the chains themselves may be in contact, chain end diffusion by segmental motion, reptation, or reactive diffusion will be required to bring the radical centers together. These terms are likely to be more important than center of mass diffusion. North¹ argued that diffusion of the reactive chain end of longer chains by segmental diffusion should be independent of chain length and has presented some experimental evidence for this hypothesis.

Bamford^{45,59-63} has proposed a general treatment for solving polymerization kinetics with chain length dependent k_t and considered in some detail the ramifications with respect to molecular weight distributions and the kinetics of chain transfer, retardation, *etc*.

5.2.1.4.2 Termination at medium to high conversions

Changes in the population of propagating species and the increase in the polymer concentration mean that the rate coefficient for radical-radical termination will decrease with conversion. The moderate conversion regime is characterized by the autoacceleration phenomenon known as the gel or Norrish-Trommsdorf effect.³⁶⁻³⁸ Various empirical relationships defining k_t or the rate of diffusion of long chains in terms of either the viscosity^{1,64} or the free volume^{34,35,44,65-69} have been proposed which enable the onset of the gel effect (Figure 5.3) to be predicted for a number of polymer systems.

Ito,⁷⁰ Tulig and Tirrell,⁷¹ and de Gennes⁷² have proposed expressions for k_t based on a reptation mechanism. More recently, the manner in which the termination rate coefficient scales with chain length for entangled systems has been considered in some detail in studies by O'Shaughnessy and coworkers.^{57,58,73,74} For the situation where both chains are long (entangled), the way in which the termination coefficient (or diffusion rates) should scale with chain length means that a long chain is unlikely to terminate by reaction with another long chain. Short-long termination is dominant. Measurements of the diffusion rate constants of oligomers and polymers provide some support for this theory.

The concept of reaction diffusion (also called residual termination) has been incorporated into a number of treatments.^{75,76} Reaction diffusion will occur in all conversion regimes. However at low and intermediate conversions the process is not of great significance as a diffusion mechanism. At high conversion long chains are essentially immobile and reaction diffusion becomes the dominant diffusion mechanism (when i and j are both "large" >100). The termination rate constant is determined by the value of k_p and the monomer concentration. In these circumstances, the rate constant for termination $k_t^{i,j}$ should be independent of the chain lengths i and j and should obey an expression of the form:⁷⁵

$$k_{t}^{i,j} - k_{tl}k_{p}[\mathbf{M}] \tag{26}$$

where k_{t1} is a constant.

5.2.1.5 Termination in heterogeneous polymerization

The kinetics of termination in suspension polymerization is generally considered to be the same as for solution or bulk polymerization under similar conditions and will not be discussed further. A detailed discussion on the kinetics of termination in emulsion polymerization appears in recent texts by Gilbert⁷⁷ and Lovell and El-Aasser⁷⁸ and readers should consult these for a more comprehensive treatment.

The steps involved in entry of a radical into the particle phase from an aqueous phase initiator have been summarized in Section 3.1.11. Aqueous phase termination prior to particle entry should be described by conventional dilute solution kinetics (Section 5.2.1.4.1). Note that chain lengths of the aqueous soluble species are short (typically <10 units).

Even though the chemical reactions are the same (*i.e.* combination, disproportionation), the effects of compartmentalization are such that, in emulsion polymerization, particle phase termination rates can be substantially different to those observed in corresponding solution or bulk polymerizations. A critical parameter is \overline{n} , the average number of propagating species per particle. The value of \overline{n} depends on the particle size and the rates of entry and exit.

Many emulsion polymerizations can be described by so-called zero-one kinetics. These systems are characterized by particle sizes that are sufficiently small that entry of a radical into a particle already containing a propagating radical always causes instantaneous termination. Thus, a particle may contain either zero or one propagating radical. The value of \overline{n} will usually be less than 0.4. In these systems, radical-radical termination is by definition not rate determining. Rates of polymerization are determined by the rates of particle entry and exit rather than by rates of initiation and termination. The main mechanism for exit is thought to be chain transfer to monomer. It follows that radical-radical termination, when it occurs in the particle phase, will usually be between a short species (one that has just entered) and a long species.

Treatments (Smith-Ewart,⁷⁹ pseudo-bulk⁷⁷) have been devised which allow for the possibility of greater than one radical per particle and for the effects of chain length dependent termination. Further discussion on these is provided in the references mentioned above.^{77,78}

Microemulsion and miniemulsion polymerization processes differ from emulsion polymerization in that the particle sizes are smaller (10-30 and 30-100 nm respectively vs 50-300 nm)⁷⁷ and there is no discrete monomer droplet phase. All monomer is in solution or in the particle phase. Initiation usually takes place by the same process as conventional emulsion polymerization. As particle sizes reduce, the probability of particle entry is lowered and so is the probability of radical-radical termination. This knowledge has been used to advantage in designing living polymerizations based on reversible chain transfer (*e.g.* RAFT, Section 9.5.2).⁸⁰⁻⁸²

5.2.1.6 Termination during living radical polymerization

It remains a common misconception that radical-radical termination is suppressed in processes such as NMP or ATRP. Another issue, in many people's minds, is whether processes that involve an irreversible termination step, even as a minor side reaction, should be called living. Living radical polymerization appears to be an oxymoron and the heading to this section a contradiction in terms (Section 9.1.1). In any processes that involve propagating radicals, there will be a finite rate of termination commensurate with the concentration of propagating radicals and the reaction conditions. The processes that fall under the heading of living or controlled radical polymerization (e.g. NMP, ATRP, RAFT) provide no exceptions.

In conventional radical polymerization, the chain length distribution of propagating species is broad and new short chains are formed continually by initiation. As has been stated above, the population balance means that, termination, most frequently, involves the reaction of a shorter, more mobile, chain with a longer, less mobile, chain. In living radical polymerizations, the chain lengths of most propagating species are similar (*i.e.* $i \sim j$) and increase with conversion. Ideally, in ATRP and NMP no new chains are formed. In practice,

250

some new chains may be formed, as, for example, from thermal initiation in S polymerization. In processes such as RAFT new small radicals are continuously formed by initiation as in the conventional process but form a much smaller part of the population as they undergo rapidly equilibration with longer dormant chains.

Diffusion mechanisms depend on chain length as follows:

- (a) Very short chains ($X_n < 10$ units). Translational diffusion is the most important diffusion mechanism.
- (b) Chains of moderate length ($X_n \sim 10-100$ units). Segmental motion of the chain ends is the rate-determining diffusion mechanism.
- (c) Long chains. Chains immobile, reaction diffusion is rate-determining.

On this basis it might be expected that at low conversions the extent of termination would be higher than in a conventional polymerization since all chains are short. Similarly, for higher conversions the extent of termination should be lower than in a conventional polymerization because most chains are long.⁸⁰ It has also been proposed that the molecular weight distribution in living radical polymerization might be analyzed to provide values of $k_1^{(4)}$ as a function of molecular weight. Recently, Vana *et al.*⁸³ have analyzed RAFT polymerization in this context. Their data suggests a chain length dependence in general agreement with that suggested by other methods. It can also be noted that the SP-PLP experiment is, in some respects, a good model of a living radical polymerization and also provides values of $k_1^{(4)}$.

It can also be noted that reversible chain transfer, in RAFT and similar polymerizations, and reversible activation-deactivation, in NMP and ATRP, provide other mechanisms for reaction diffusion.

5.2.2 Disproportionation vs Combination

Even though the rate of radical-radical reaction is determined by diffusion, this does not mean there is no selectivity in the termination step. As with small radicals (Section 2.5), self-reaction may occur by combination or disproportionation. In some cases, there are multiple pathways for combination and disproportionation. Combination involves the coupling of two radicals (Scheme 5.1). The resulting polymer chain has a molecular weight equal to the sum of the molecular weights of the reactant species. If all chains are formed from initiator-derived radicals, then the combination product will have two initiator-derived ends. Disproportionation involves the transfer of a β -hydrogen from one propagating radical to the other. This results in the formation of two polymer molecules. Both chains have one initiator-derived end. One chain has an unsaturated end, the other has a saturated end (Scheme 5.1).

Since the mode of termination clearly plays an important part in determining the polymer end groups and the molecular weight distribution, a knowledge of the disproportionation:combination ratio (k_{td}/k_{tc}) is vital to the understanding of structure-property relationships. Unsaturated linkages at the ends of polymer

chains, as may be formed by disproportionation, have long been thought to contribute to polymer instability and it has been demonstrated that both head-to-head linkages and unsaturated ends are weak links during the thermal degradation of PMMA (Section 8.2.2).⁸⁴⁻⁸⁷ Polymer chains with unsaturated ends may also be reactive during polymerization. Copolymerization of macromonomers formed by disproportionation is a possible mechanism for the formation of long chain branches.⁸⁸⁻⁹⁰ Such macromonomers may also function as transfer agents (Section 6.2.3.4 and 9.5.2).⁹⁰

Knowledge of k_{td}/k_{tc} is also important in designing polymer syntheses. For example, in the preparation of block copolymers using polymeric or multifunctional initiators (Section 7.6.1), ABA or AB blocks may be formed depending on whether termination involves combination or disproportionation respectively. The relative importance of combination and disproportionation is also important in the analysis of polymerization kinetics and, in particular, in the derivation of rate parameters.

5.2.2.1 Model studies

The determination of k_{td}/k_{tc} by direct analysis of a polymerization or the resultant polymer often requires data on aspects of the polymerization mechanism that are not readily available. For this reason, it is appropriate to consider the self-reactions of low molecular weight radicals which are structurally analogous to the propagating species. These model studies provide valuable insights by demonstrating the types of reaction that are likely to occur during polymerization and the factors influencing k_{td}/k_{tc} . These have been discussed in general terms in Section 2.4.

In these model studies, evaluation of k_{td}/k_{tc} is simplified because reactions that compete with disproportionation or combination are more readily detected and allowed for. However, by their very nature, model studies cannot exactly simulate all aspects of the polymerization process. Consequently, a number of factors must be borne in mind when using model studies to investigate the termination process. These stem from differences inherent in polymerization vs simple organic reactions and include:

- (a) There may be additional pathways open to the poly- or oligomeric radicals which are not available to the simple model species.⁹¹
- (b) In polymerization particular propagating species have only transient existence since they are scavenged by the addition of monomer or other reactions. Model studies are usually designed such that the self-reaction is the only process. This can lead to a very different and sometimes misleading product distribution. A knowledge of the reaction kinetics is extremely important in analyzing the results.
- (c) Reaction conditions (solvent, viscosity, *etc.*) chosen for the model experiment and the polymerization experiment are often very different.

Model carbon-centered radicals are conveniently generated from azocompounds. These have the advantage that radicals are generated in pairs and that transfer to initiator is generally not a serious problem. All of the major products from thermal or photochemical decomposition in an inert solvent are the products from radical-radical reaction. One frequently observed complication is polymerization of the unsaturated byproducts of disproportionation. This problem may be circumvented by conducting experiments in the presence of an inhibitor, the concentration of which can be chosen such that all radicals which escape the solvent cage are trapped and reactions of the initiator-derived radicals with other species are eliminated.⁸⁹ The value of k_{td}/k_{tc} is determined by analyzing the products of cage reaction. Most data indicate no difference in specificity between the cage and encounter (*i.e.* non-cage) processes.⁸⁹

5.2.2.1.1 Polystyrene and derivatives

The self reaction of substituted phenylethyl radicals (1) has been widely investigated.⁹²⁻⁹⁶ The findings of these studies are summarized in Table 5.2. Unless R^2 is very bulky (*e.g. t*-butyl, see below), combination is by far the dominant process with the value k_{td}/k_{tc} typically in the range 0.05-0.16. Thus, a small amount of disproportionation is always observed.



The value of k_{td}/k_{tc} shows no significant dependence on chain length for oligostyryl radicals (4a, b).^{95,96} On the basis of these findings, k_{td}/k_{tc} for PS• should also be small and non-zero.



For radicals 1, k_{td}/k_{tc} shows a marked dependence on the bulk of the substituent (R²). While phenylethyl radicals (2) and cumyl radicals (5) afford predominantly combination, there are indications of a substantial penultimate unit effect. The radicals 6, with an α -neopentyl substituent, give predominantly disproportionation. Termination in AMS polymerization might therefore also give substantial

disproportionation (However, AMS does not polymerize readily due to a very low ceiling temperature - Section 4.5.1).

| System | Structure | Temp. | $k_{\rm td}/k_{\rm tc}$ | System | Structure | Temp. | $k_{\rm td}/k_{\rm tc}$ |
|-----------------|------------|-------|-------------------------|-------------------|------------|-------|-------------------------|
| | | (°C) | | | | (°C) | |
| S ⁹⁵ | 2 | 20 | 0.073 | S ⁹⁶ | 4a | 141 | 0.090 |
| S^{95} | 2 | 80 | 0.081 | S^{96} | 4 a | 161 | 0.078 |
| S^{93} | 2 | 118 | 0.097 | S^{96} | 4 b | 80 | 0.159 |
| S ⁹⁵ | 3 | 20 | 0.141 | S^{96} | 4 b | 90 | 0.150 |
| S^{95} | 3 | 80 | 0.146 | S^{96} | 4 b | 100 | 0.134 |
| S ⁹³ | 3 | 118 | 0.107 | S ⁹⁶ | 4 b | 120 | 0.119 |
| S^{95} | 4 a | 80 | 0.156 | S ⁹⁶ | 4 b | 141 | 0.097 |
| S^{96} | 4 a | 90 | 0.146 | S^{96} | 4 b | 161 | 0.082 |
| S^{96} | 4 a | 90 | 0.141 | AMS ⁹⁷ | 5 | 20-60 | 0.05 |
| S^{96} | 4 a | 100 | 0.130 | AMS ⁹⁸ | 5 | 55 | 0.1 |
| S ⁹⁶ | 4 a | 120 | 0.109 | AMS ⁹⁹ | 6 | 55 | œ |

Table 5.2 Values of k_{td}/k_{tc} for Polystyryl Radical Model Systems

The value of k_{td}/k_{tc} for oligostyryl radicals (4) is reported to decrease with increasing temperature. With 1,3,5-triphenylpentyl radicals (4b) k_{td}/k_{tc} halves on increasing the temperature from 80 °C to 160 °C (Table 5.2).⁹⁶

The result indicates that the activation energy for combination is higher than that for disproportionation by ca 10 kJ mol⁻¹. A similar inverse temperature dependence is seen for other small radicals (Section 2.5). However, markedly different behavior is reported for polymeric radicals (Section 5.2.2.2.1).

Benzyl radicals and α - and β - substituted derivatives also undergo unsymmetrical coupling through the aromatic ring (Section 2.5). The formation of the α -o and α -p coupling products is reversible. Consequently, these materials are often only observed as transient intermediates.



Scheme 5.6

Direct aromatization of the quinonoid intermediates is a photochemically allowed but thermally forbidden rearrangement (Scheme 5.6). When phenylethyl radicals are generated photochemically at 20 °C there is evidence⁹⁵ of α -o coupling by way of the aromatized product 7. The products derived from these pathways can be trapped in thermal reactions by radical⁹⁸ or acid¹⁰⁰ catalyzed

aromatization. With benzyl radicals the ratio of $\alpha - o:\alpha - p$ and $[\alpha - o + \alpha - p]:\alpha - \alpha$ has been shown to increase with increasing temperature.¹⁰⁰ A transient species, presumed to be a quinonoid intermediate, has also been observed when oligomeric radicals 4 are generated thermally.⁹⁶

The formation of the quinonoid species is favored by substitution at the radical center (Section 2.4). Cumyl radicals $(5)^{97,98,101}$ are reported to give $\alpha - \alpha$, $\alpha - o$ and $\alpha - p$ coupling products in the ratio 77:8:15. Several studies have examined the reactions of *p*-substituted phenylethyl radicals. Electron withdrawing substituents favor disproportionation over combination. However, the effect is small.

A report by Businelli *et al.* suggests a remarkable solvent dependence for the combination: disproportionation ratio.¹⁰² These authors found that 1-phenylpentyl radicals (concentration, temperature unspecified) gave only combination in benzene solvent but combination: disproportionation products in a 1:1 ratio in acctonitrile solvent.

5.2.2.1.2 Poly(alkyl methacrylates)

The self-reactions of 2-carboalkoxy-2-propyl radicals (8-10) have been examined.^{89,103,104} The results of these studies are reported in Table 5.3. Combination is slightly favored over disproportionation. The value of k_{td}/k_{tc} for **8** was found to be essentially independent of temperature.

| СН₃ Н₃С−С́∙ | CH₃ H₃C−Ċ∙ | СН ₃ Н ₃ С−С́• | CH₃ CH₃ H₃C-Ċ-CH₂-Ċ• |
|---------------------------------|---|---|-------------------------|
| ĊO ₂ CH ₃ | ĊO ₂ C ₂ H ₅ | CO ₂ C ₄ H ₉ | CO2CH3 CO2CH3 |
| 8 | 9 | 10 | 11 |

| Structure | Temperature (°C) | $k_{ m td}/k_{ m tc}$ | ref. |
|-----------|--|---|--|
| 8 | 70-90 | 0.78 | 89 |
| 8 | 90 | 0.62 | 103 |
| 8 | 115 | 0.61 | 89.103 |
| 8 | 140 | 0.60 | 89.103 |
| 8 | 165 | 0.59 | 103 |
| 11 | 80 | ≤1.85 | 89 |
| 9 | 80 | 0.72 | 89 |
| 10 | 80 | 1.17 | 89 |
| 8, 10 | 80 | 1.22 | 105 |
| | Structure 8 8 8 8 8 11 9 10 8, 10 | Structure Temperature (°C) 8 70-90 8 90 8 115 8 140 8 165 11 80 9 80 10 80 8, 10 80 | StructureTemperature (°C) k_{td}/k_{tc} 870-900.788900.6281150.6181400.6081650.591180 ≤ 1.85 9800.7210801.178, 10801.22 |

Table 5.3 Values of k_{td}/k_{tc} for Methacrylate Ester Model Systems

Disproportionation increases in the series where the ester is methyl<ethyl
butyl suggesting that this process is favored by increasing the bulk of the ester alkyl group. This trend is also seen for polymeric radicals (Section

256

5.2.2.2.2). Bizilj *et al.*⁸⁹ reported that disproportionation is more important for oligomeric radicals. While combination products were unequivocally identified, analytical difficulties prevented a precise determination of the disproportionation products. Accordingly, they were only able to state a maximum value of k_{td}/k_{tc} . Their data show that $k_{td}/k_{tc} \le 1.85$ for the self reaction of **11** and ≤ 1.50 for reaction between **8** and **11**.

An early report¹⁰⁶ indicated that the self reaction of 2-carbomethoxy-2-propyl radicals (8), like cyanoisopropyl radicals (15) (Section 5.2.2.1.3), affords an unstable coupling product (analogous to a ketenimine). Precedent for a reversible unsymmetrical C-O coupling mode for radicals with a α -carbonyl group has recently been established for the case where normal C-C coupling is sterically very hindered.¹⁰⁷ However, the more recent studies on reactions of 2-carbomethoxy-2-propyl radicals (8) and related species provide no evidence for this pathway.^{89,103} Bizilj *et al.*⁸⁹ also demonstrated that during disproportionation of oligomeric radicals 12, the abstraction of a methyl hydrogen (to generate a terminal methylene group - 13, Scheme 5.7) is preferred \geq 10-fold over abstraction of a methylene hydrogen (to afford an internal double bond 14). One explanation is that the methyl hydrogens are more sterically accessible than the methylene hydrogens.



Scheme 5.7

5.2.2.1.3 Poly(methacrylonitrile)

A simple model for the propagating species in MAN polymerization is the cyanoisopropyl radical (15). The reactions of these radicals (from AIBN; Scheme 5.8) have been extensively studied. In contrast with the analogous esters 8-10 (Section 5.2.2.1.2), combination is by far the dominant process (Table 5.4).

Serelis and Solomon¹⁰⁸ found that primary radical termination of oligo(MAN) radicals (16) with 15 also gives predominantly combination. The ratio k_{td}/k_{tc} was found to have little, if any, dependence on the oligomer chain length (n<4). As with PMMA•, disproportionation involves preferential abstraction of a methyl

hydrogen and chains terminated in this way will, therefore, possess a potentially reactive terminal methylene (17).



Table 5.4 Values of k_{td}/k_{tc} for Reactions involving Cyanoisopropyl Radicals

| System | Structure | Temperature (°C) | $k_{\rm td}/k_{\rm tc}$ | ref. |
|------------|------------|------------------|-------------------------|---------|
| ΜΛΝ | 15 | 80 | 0.05-0.1 | 108-110 |
| MAN | 16 | 80 | 0.1 | 108 |
| MAN-co-S | 4 a | 90 | 0.61 | 111 |
| MAN-co-S | PS• | 98 | а | 112 |
| MAN-co-BMA | PBMA• | 25 | b | 113 |
| MAN-co-E | PE• | 80 | b | 114 |

a Predominantly combination. b Predominantly disproportionation.

Cyanoisopropyl radicals (15) undergo unsymmetrical C-N coupling in preference to C-C coupling.¹¹⁵ The preferential formation of the ketenimine is a reflection of the importance of polar and steric influences.¹¹⁶ However, the ketenimine is itself thermally unstable and a source of 15, thus the predominant isolated product is often from C-C coupling.

Preferential C-N coupling is also observed for oligomeric radicals (Scheme 5.9).¹¹⁷ A ketenimine (21) is the major product from the reaction of the "dimeric" MAN radical 18 with cyanoisopropyl radicals (15). Only one of the two possible ketenimines was observed; a result which is attributed to the thermal lability of ketenimine 19. If this explanation is correct then, although C-N coupling may

occur during MAN polymerization, ketenimine structures are unlikely to be found in PMAN by self-reaction of propagating radicals.



5.2.2.1.4 Polyethylene

The self reaction of primary alkyl radicals gives mainly combination.¹¹⁸ For primary alkyl radicals $[CH_3(CH_2)_nCH_2\bullet]$, k_{td}/k_{te} is reported to lie in the range 0.12-0.14, apparently independent of chain length (*n*=0-3).^{118,119}

5.2.2.2 Polymerization

A substantial number of studies give information on k_{td}/k_{tc} for polymerizations of S (5.2.2.2.1) and MMA (5.2.2.2.2). There has been less work on other systems. One of the main problems in assessing k_{td}/k_{tc} lies with assessing the importance of other termination mechanisms (*i.e.* transfer to initiator, solvent, *etc.*, primary radical termination).

Techniques applied in assessing the relative importance of disproportionation and combination include:

(a) The Gelation technique. This method was developed by Bamford *et al.*¹²⁰ In graft copolymerization, termination by combination will give rise to a crosslink while disproportionation (and most other termination reactions) will lead to graft formation. The initiation system based on a polymeric halo-compound [poly(vinyl trichloroacetate)/Mn₂(CO)₁₀/hv] was used to initiate polymerization and the time for gelation was used to calculate k_{td}/k_{tc} . In the original work, the results were calibrated with reference to data for S polymerization for which a k_{td}/k_{tc} of 0.0 was assumed. Recent studies suggest that, in S polymerization, disproportionation may account for 10-20% of

258

chains (Section 5.2.2.2.1). Thus the data may require minor adjustment. Systems studied with this technique include AN, MAN, MA, MMA, and S.

- (b) Molecular weight measurement. The mode of termination can be calculated by comparing the kinetic chain length (the ratio of the rate of propagation to the rate of initiation or termination) with the measured number average molecular weight.¹²¹⁻¹²³
- (c) Molecular weight distribution evaluation. This method relies on a precise evaluation of the molecular weight distribution.¹²⁴⁻¹²⁷ The mode of termination has a significant influence on the shape of the molecular weight distribution with the instantaneous dispersity (*D* being ~2.0 if termination occurs exclusively by disproportionation of propagating radicals and ~1.5 if termination involves only combination (Section 5.2.1.2).¹²⁸ Values of *D* are conversion dependent so the method should only be applied to very low conversion samples. Truncation of the ends of the distribution as a result of baseline selection difficulties will lead to the dispersity being underestimated.¹²⁹ A more precise but related method is to fit the entire molecular weight distribution using kinetic modeling methods.
- (d) End group determination. Polymer chains terminated by combination possess two initiator-derived chain ends. Disproportionation affords chains with only one such end. The value of k_{td}/k_{tc} can therefore be determined by evaluating the initiator-derived polymer end groups/molecule by applying eq. 27

$$k_{\rm td}/k_{\rm tc} = (2-x)/2(x-1) \tag{27}$$

where x is the number of initiator fragments per molecule. The errors inherent in this technique can be large since the polymer end groups typically comprise only a very small fraction of a polymer sample. The initiator-derived ends may be labeled for ease of detection. These techniques are described in Section 3.6. It is necessary to allow for side reactions. If there is transfer to monomer, solvent, *etc.*, the value of k_{td}/k_{tc} will be overestimated. The occurrence of transfer to initiator, primary radical termination, or copolymerization of initiator byproducts will lead to k_{td}/k_{tc} being underestimated.

(c) Mass spectrometry. Matrix-assisted laser desorption ionization time-of-flight mass spectroscopy (MALDI-TOF) has been used to determine k_{td}/k_{tc} in S and MMA polymerization.¹³⁰ Chains formed by disproportionation and chains formed by combination form two distinct distributions. Mass spectrometric end group determination is described in Section 3.5.3.4.

Evaluation of molecular weights after ultrasonic scission of high molecular weight polymers (PMMA and PS) in the presence of a radical trap has been claimed to provide evidence of the termination mechanism.¹³¹ However, scission gives radicals as shown in Scheme 5.10.

Scheme 5.10

5.2.2.2.1 Polystyrene

Hensley *et al.*¹³² reported the only direct experimental observation of head-tohead linkages in PS by 2D INADEQUATE NMR on ¹³C-enriched PS. The method did not enable these groups to be quantified with sufficient precision for evaluation of k_{td}/k_{tc} . Zammit *et al.*¹³⁰ studied chain distribution of low molecular weight PS prepared with AIBN initiator by MALDI-TOF. Separate distributions of chains formed by combination and disproportionation were observed. They estimated k_{td}/k_{tc} at 90 °C to be 0.057.

A wide range of less direct methods has been applied to determine k_{td}/k_{tc} in S polymerization. Most indicate predominant combination.^{122,125,133-148} However, distinction between a k_{td}/k_{tc} of 0.0 and one which is non-zero but ≤ 0.2 is difficult even with the precision achievable with the most modern instrumentation. Therefore, it is not surprising that many have interpreted the experimental finding of predominantly combination as meaning exclusively combination.

Olaj *et al.*¹²⁴ proposed that termination of S polymerization involves substantial disproportionation. They analyzed the molecular weight distribution of PS samples prepared with either BPO or AIBN as initiator at temperatures in the range 20-90 °C and estimated k_{td}/k_{tc} to be *ca* 0.2. In a more recent study, Olaj *et al.*¹⁴⁹ determined the molecular weight distribution of PS samples prepared with photoinitiation at 60 and 85 °C and estimated values of k_{td}/k_{tc} of 0.5 and 0.67 respectively. Dawkins and Yeadon¹²⁵ discussed the problems associated with estimating k_{td}/k_{tc} on the basis of dispersity measurements and determined that k_{td}/k_{tc} should be "substantially smaller" than suggested by Olaj *et al.*¹⁴⁹

Berger and Meyerhoff¹⁵⁰ also reported that termination involves substantial disproportionation. They determined the initiator fragments per molecule in PS prepared with radiolabeled AIBN and conducted a detailed kinetic analysis of the system. They also found a marked temperature dependence for k_{td}/k_{tc} . Values of k_{td}/k_{tc} ranged from 0.168 at 30 °C to 0.663 at 80 °C.

Other determinations of k_{td}/k_{tc} based on end group determination are at variance with these findings. End group analyses by NMR,^{146,147} radiotracer techniques,¹⁴²⁻¹⁴⁴ or chemical analysis¹³⁵ on PS formed with appropriately labeled initiators all indicate predominantly combination. Moad *et al.*^{146,147} used ¹³C NMR to define and quantify the end groups in samples of PS prepared at 60 °C with either ¹³C-labeled BPO or AIBN as initiator. This method has the advantage that the end groups from primary radical termination, transfer to initiator, residual initiator and any copolymerized initiator byproducts can be distinguished from the end groups formed by initiation (Section 3.5.3.2). They showed that, under the conditions employed (60 °C, bulk), there are 1.7±0.2 initiator-derived end groups

corresponding to a k_{td}/k_{tc} of *ca* 0.2. Other NMR end group determinations have yielded similar data. Barson *et al.*¹⁵¹ analyzed PS prepared with ¹³C-labeled AIBN by ¹³C NMR. Bevington *et al.*¹⁵² analyzed PS prepared with fluorinated BPO by ¹⁹F NMR. In each case there were *ca* 1.6 initiator-derived end groups per molecule (k_{td}/k_{tc} *ca* 0.3). Yoshikawa *et al.*¹⁵³ formed PS• from narrow dispersity (\overline{M}_{n} =1500, M_{w}/M_{n} =1.09) low molecular weight ω -bromopolystyrene by atom transfer to Cu(I) at 110 °C. They used NMR to estimate the fraction of chains formed by disproportionation as 0.07 (k_{td}/k_{tc} *ca* 0.08) and by GPC peak resolution to be 0.09 (k_{td}/k_{tc} *ca* 0.1).

The influence of substituents (*p*-Cl, *p*-OMe) on k_{td}/k_{tc} was investigated by Ayrey *et al.*¹⁴⁸ They found disproportionation was favored by the p-OMe substituent and that the extent of disproportionation increased with increasing temperature. This result is contrary to the model studies (Section 5.2.2.1.1) that show k_{td}/k_{tc} has little dependence on substituents and, indeed, suggest the opposite trend.

5.2.2.2.2 Poly(alkyl methacrylates)

Table 5.5 Determinations of k_{td}/k_{tc} for MMA Polymerization

| Temperature | Method ^a | | | | | | | | | | |
|-------------|----------------------|------------------|------------------|-----------|------------------|------------------|------------------|-----------|---------------------|------------------|------------------|
| (°C) | E ^{133,147} | E ¹³⁴ | E ¹⁵⁵ | G^{156} | E ¹⁵⁷ | P ¹⁵⁸ | E ¹⁵⁹ | M^{121} | E ^{85,160} | S ¹³⁰ | E ¹⁵¹ |
| -25 | - | - | - | - | - | 0.14 | - | - | - | - | - |
| 0 | 1.50 | - | - | - | - | 0.50 | - | - | - | - | - |
| 15 | - | - | - | - | - | 0.76 | - | - | - | - | - |
| 25 | 2.13 | - | - | 2.0 | - | - | - | - | - | - | - |
| 30 | - | - | - | - | - | 1.18 | - | - | - | - | - |
| 40 | - | - | 0.45 | - | - | - | - | - | - | - | - |
| 45 | - | - | - | - | - | 00 | - | - | - | - | - |
| 60 | 5.67 | 1.35 | 0.75 | 2.7 | 2.62 | - | 2.57 | 0.44 | 1.28 | - | 4.5 |
| 80 | - | - | 1.32 | 4.0 | - | - | - | - | - | - | - |
| 90 | - | - | - | - | - | - | - | - | - | 4.37 | - |
| 100 | - | - | - | - | - | - | - | - | 1.5 | - | - |

a Methods used (Section 5.2.2.2): G-gelation technique, M-molecular weight measurement, Pdispersity evaluation, E-end group determination, S-MALDI-TOF mass spectrometry

The nature of the termination reaction in MMA polymerization has been investigated by a number of groups using a wide range of techniques (Table 5.5). There is general agreement that there is substantial disproportionation. However, there is considerable discrepancy in the precise values of k_{td}/k_{tc} . In some cases the difference has been attributed to variations in the way molecular weight data are interpreted or to the failure to allow for other modes of termination under the polymerization conditions (chain transfer, primary radical termination).¹⁵⁴ In other cases the reasons for the discrepancies are less clear. MALDI-TOF mass
spectrometry provides a direct measurement of k_{td}/k_{tc} for low molecular weight MMA and this indicates a value of 4.37 at 90 °C.¹³⁰

Four studies suggest that k_{td}/k_{tc} has a significant temperature dependence (Table 5.5). Although not agreeing on the precise value of k_{td}/k_{tc} , all four studies indicate that the proportion of disproportionation increases with increasing temperature. These results are at variance with model studies that suggest that k_{td}/k_{tc} is independent of temperature. It was also proposed that the preferred termination mechanism is solvent dependent and that disproportionation is favored in more polar media.¹⁶¹

Hatada *et al.*^{160,162} showed that the disproportionation-derived unsaturated ends in PMMA can be determined directly by ¹H NMR. For PMMA prepared with BPO in toluene at 100 °C they found the number of chain ends per molecule formed from initiation reactions (from BPO and toluene-derived radicals) to be *ca* 1.25^{160} suggesting a k_{td}/k_{tc} of *ca* 1.5. They also demonstrated the preference for transfer of a methyl *vs* a methylene hydrogen in disproportionation. This is in line with the studies on model radicals (Section 5.2.2.1.2).

Values of k_{td}/k_{tc} for polymerizations of EMA and BMA and higher methacrylate esters have been determined.^{113,120,157,159} The extent of disproportionation increases with the size of the ester alkyl group.

5.2.2.3. Poly(methacrylonitrile)

Bamford *et al.*¹²⁰ examined MAN polymerization (25 °C, DMSO) using the gelation technique (Section 5.2.2.2) and have estimated that termination occurs predominantly by disproportionation ($k_{td}/k_{tc} = 1.86$). This result is at variance with the model studies (Section 5.2.2.1.3).

5.2.2.2.4 Poly(alkyl acrylates)

The termination mechanism in MA polymerization has been variously determined to be predominantly disproportionation^{137,157} or predominantly combination.^{120,159,163}

Ayrey *et al.*¹⁶³ suggested that transfer reactions may have led to erroneous conclusions being drawn in some of the earlier studies. They concluded that termination is almost exclusively by combination (25 °C, benzene). Bamford *et al.*¹²⁰ came to a similar conclusion using the gelation technique (25 °C, bulk) and determined that the polymerizations of higher acrylate esters also terminate predominantly by combination.

5.2.2.2.5 Poly(acrylonitrile)

There appears to be general agreement that termination in AN polymerization under a variety of conditions (10-90 °C, DMSO, DMF, H₂O) involves mainly combination.^{120,123,164,165} It was suggested that this may involve either C-N (ketenimine formation) or C-C coupling.¹⁶⁶

5.2.2.2.6 Poly(vinyl acetate)

Early reports^{137,157,167} suggested that termination during VAc polymerization involved predominantly disproportionation. However, these investigations did not adequately allow for the occurrence of transfer to monomer and/or polymer, which are extremely important during VAc polymerization (Sections 6.2.6.2 and 6.2.7.4 respectively). These problems were addressed by Bamford *et al.*¹²⁰ who used the gelation technique (Section 5.2.2.2) to show that the predominant radical-radical termination mechanism is combination (25 °C).

5.2.2.2.7 Poly(vinyl chloride)

Studies on VC polymerization are also complicated by the fact that only a small proportion of termination events may involve radical-radical reactions. Most termination is by transfer to monomer (Sections 4.3.1.2 and 6.2.6.3). Early studies on the termination mechanism which do not allow for this probably overestimate the importance of disproportionation.^{168,169}

Park and Smith ¹⁷⁰ attempted to allow for chain transfer in their examination of the termination mechanism during VC polymerization at 30 and 40 °C in chlorobenzene. They determined the initiator-derived ends in PVC prepared with radiolabeled AIBN and concluded that $k_{td}/k_{tc} = 3.0$. However, questions have been raised regarding the reliability of these measurements.^{171,172} Atkinson *et al.*¹⁷² applied the gelation technique (Section 5.2.2.2) to VC polymerization and proposed that termination involves predominantly combination.

5.2.2.3 Summary

Unequivocal numbers for k_{td}/k_{tc} are not yet available for most polymerizations and there is only qualitative agreement between values obtained in model studies and real polymerizations.

It is tempting to attribute problems in reconciling data from model studies and actual polymerizations to difficulties associated with data interpretation. The polymerization experiments are often complicated by other termination pathways, in particular chain transfer, which must be allowed for when assessing the results. It is notable in this context that the discrepancies are most evident for reactions carried out at higher temperatures (Sections 5.2.2.1.1 and 5.2.2.1.2).

However, some of the differences may be explicable in terms of an effect of molecular size. For many of the model systems at least one of the reaction partners is monomeric (*i.e.* **2**, **5**, **8-10**, **15**). Since combination is known to be more sensitive to steric factors than disproportionation (Section 2.4.3.2), k_{td}/k_{tc} may be anticipated to be higher for the corresponding propagating species. The values of k_{td}/k_{tc} reported for **3** or **4** are significantly greater than those for **2**. Similarly, **6** gives much more disproportionation than **5**. Thus, values of k_{td}/k_{tc} seen for systems involving monomeric model radicals (**2**, **5**, **8-10**, or **15**) should be considered only as a lower limit for the polymeric system.

Despite these problems in assessing k_{td}/k_{tc} , it is possible to make some generalizations:

- (a) Termination of polymerizations involving vinyl monomers (CH₂=CHX) involves predominantly combination.
- (b) Termination of polymerizations involving α-methylvinyl monomers (CH₂=C(CH₃)X) always involves a measurable proportion of disproportionation.
- (c) During disproportionation of radicals bearing an α -methyl substituent (for example, those derived from MMA), there is a strong preference for transfer of a hydrogen from the α -methyl group rather than the methylene group.
- (d) Within a series of vinyl or α -methylvinyl monomers, k_{td}/k_{te} appears to decrease as the ability of the substituent to stabilize a radical center increases. Thus, k_{td}/k_{te} for radicals $\sim C(\cdot)(CH_3)X$ or $\sim C(\cdot)HX$ decreases in the series where X is $CO_2R >> CN > Ph$.

5.3 Inhibition and Retardation

Inhibitors and retarders are used to stabilize monomers during storage or during processing (e.g. synthesis, distillation). They are often used to quench polymerization when a desired conversion has been achieved. They may also be used to regulate or control the kinetics of a polymerization process.

Inhibitors have been defined as species which, when added to a polymerization, react to consume and deactivate the initiator-derived radicals.¹⁷³ Retarders have been similarly defined as species which deactivate the propagating radicals.¹⁷³ According to this definition, a nitroxide added to a *t*-butoxy radicalinitiated polymerization of S should be called a retarder since the *t*-butoxy radicals appear not to react with the nitroxide. However, the initiator-derived and propagating radicals often show similar selectivity in their reactions and the distinction between inhibitors and retarders becomes blurred. In a cyanoisopropyl radical-initiated polymerization of S, an added nitroxide would be called an inhibitor when used in high concentration and a retarder when used at very low concentration. Generally the term inhibitor is used without reference to which radicals are scavenged. With many experimental techniques it is not possible to discriminate between scavenging of initiator-derived and oligomeric propagating radicals. Thus an inhibitor has come to mean any species that is able to rapidly and efficiently scavenge propagating and/or initiator-derived radicals and thus prevent polymer chain formation. The term retarder is commonly used to define species that slows rather than prevents polymerization.

Inhibitors or retarders that give inert products are called 'ideal'.¹⁷³ The term 'ideal inhibitor' has also been used to describe a species that stops all polymerization until such time as it is completely consumed (*i.e.* the induction period) and then allows polymerization to proceed at the normal rate. However, in many cases the products formed during inhibition or retardation are not inert. Four

main pathways for further reaction following the initial reaction with inhibitor or retarder are distinguished:

- (a) Slow reinitiation with reference to propagation following chain transfer (see, for example, Section 5.3.4).
- (b) Slow propagation with reference to normal propagation following addition (see, for example, Section 5.3.3).
- (c) Further reaction of the initially formed species as an inhibitor or retarder (see, for example, Sections 5.3.4, 5.3.5, 5.3.7).
- (d) Reversal of the reaction associated with inhibition or retardation (see, for example, Section 5.3.1 and Chapter 9).

The kinetics and mechanism of retardation and inhibition has been reviewed by Bamford,¹⁷³ Tüdos and Földes-Berezsnich,¹⁷⁴ Eastmond,¹⁷⁵ Goldfinger *et al.*¹⁷⁶ and Bovey and Kolthoff.¹⁷⁷

Common inhibitors include stable radicals (Section 5.3.1), oxygen (5.3.2), certain monomers (5.3.3), phenols (5.3.4), quinones (5.3.5), phenothiazine (5.3.6), nitro and nitroso-compounds (5.3.7) and certain transition metal salts (5.3.8). Some inhibition constants (k_z/k_p) are provided in Table 5.6. Absolute rate constants (k_z) for the reactions of these species with simple carbon-centered radicals are summarized in Table 5.7.

| Inhibitor | | | k_z/k_p | | |
|------------------------|----------------------|----------------------------|-------------------|-------|---------------------|
| | MMA | МА | AN | S | VAc |
| CuCl ₂ | 1030 | - | 100 ^e | 10000 | - |
| FeCl ₃ | $5000 k_{p}^{c}$ | $6800 k_{p}^{c}$ | 3.33° | 536 | $2300000 k_{p}$ |
| <i>p</i> -benzoquinone | 4.5 | $< 0.15 k_{\rm p}^{\rm b}$ | 0.91 ^b | 520 | - |
| nitrobenzene | 0.00464 ^h | 0.00464 ^b | - | 0.326 | 11.2^{b} |
| DPPH | 2000 | - | - | - | - |
| oxygen | 33000 | - | - | 14600 | - |
| anthracene | - | 0.098^{b} | 2.67 ^b | 2^d | 27.8 |
| <i>p</i> -hydroquinone | - | - | - | - | 0.7 |
| phenol | - | 0.0002 ^b | - | - | 0.06 |
| styrene | - | - | - | - | $40.8^{b.174}$ |

Table 5.6 Inhibition constants $(k_z/k_p, 60 \text{ °C}, \text{ bulk})$ for Various Inhibitors with Some Common Monomers^a

a Data taken from Eastmond¹³ unless otherwise stated and are rounded to three significant figures. b 50 °C. c in DMF. d 44.4 °C.

Whether a given species functions as an inhibitor, a retarder, a transfer agent or a comonomer in polymerization is dependent on the monomer(s) and the reaction conditions. For example, oxygen acts as an inhibitor in many polymerizations yet it readily copolymerizes with S. Reactivity ratios for VAc-S

copolymerization are such that small amounts of S are an effective inhibitor of VAc polymerization (r_s =0.02, r_{VAc} =22.3). The propagating chain with a terminal VAc adds to S preferentially even when VAc is present in large excess over S. The resultant propagating radical with a terminal S adds to VAc only slowly. The reactions of many inhibitors with propagating radicals may become reversible under some reaction conditions. In these circumstances, the reagent may find use as a control agent in living radical polymerization (Chapter 9).

Inhibitor Radical Temp. (°C) $k_{\rm z} \,({\rm M}^{-1}{\rm s}^{-1})$ refs. $\sim 1 \times 10^{9}$ TEMPO (23) prim. alkyl 60 178-180 2.9×10^9 27 oxygen benzyl 181 2.0×10^7 prim. alkyl 69 182 p-benzoquinone (38) 1826.5 x 10⁵ CuCh prim. alkyl 25

Table 5.7 Absolute Rate Constants (k_z) for the Reaction of Carbon-CenteredRadicals with Some Common Inhibitors

The effectiveness of inhibitors is measured in terms of the rate constant ratio k_z/k_p and the stoichiometric coefficient. The stoichiometric coefficient is the moles of radicals consumed per mole of inhibitor. These parameters may be determined by various methods. A brief description of the classical kinetic treatment for evaluating k_z/k_p follows. Consider the reaction scheme shown which describes ideal inhibition and retardation (Scheme 5.11).

| initiation | | | | |
|---------------------|---------------|---------------------------------------|--|---------------------------------------|
| I_2 | \rightarrow | 2 I• | $R_i=2 k_d f[I_2]$ | |
| I∙+M | \rightarrow | P₁• | $k_{\rm i} \ge k_{\rm p}$ | |
| inhibition | | | | |
| I∙+Z | \rightarrow | IZ (dead) | $R_z - k_z[Z][I \bullet]$ | |
| propagatio | n | | | |
| $P_n \bullet + M$ | \rightarrow | P _{n+1} ● | $R_p - k_p [M] [P \bullet]$ | |
| disproporti | onati | on | | |
| $P_n \bullet + P_m$ | • → | $\mathbf{P_n}^{H} + \mathbf{P_m}^{=}$ | $R_{\rm tc}=2k_{\rm tc}[{\rm P}\bullet]^2$ | $R_{\rm t} = R_{\rm tc} + R_{\rm td}$ |
| combinatio | m | | | |
| $P_n \bullet + P_m$ | • → | P _{n+m} | $R_{\rm tc}=2k_{\rm tc}[{\rm P}\bullet]^2$ | |
| retardation | | | | |
| $P_n \bullet + Z$ | \rightarrow | P _n Z (dead polymer) | $R_z - k_z[Z][P\bullet]$ | |

Scheme 5.11

With the omission of the reinitiation reaction, this scheme is the same as that for polymerization with chain transfer and an expression (eq 28) for the degree of polymerization similar in form to the Mayo equation can be derived.

$$\frac{1}{\overline{X}_{n}} = \frac{\left(1 + \frac{k_{\rm td}}{k_{\rm t}}\right) \left(2k_{\rm d} f[I_{2}]k_{\rm t}\right)^{0.5}}{k_{\rm p}[{\rm M}]} + \frac{k_{\rm z}[Z]}{k_{\rm p}[{\rm M}]}$$
(28)

If the amount of termination by radical-radical reaction is neglected the degree of polymerization and the kinetic chain length are given by eq. 29:

$$v \approx \overline{X}_{n} \approx \frac{k_{p}[M]}{k_{z}[Z]}$$
⁽²⁹⁾

If chains are very short we must include an additional term in the numerator for monomer consumption in the initiation step (eq. 30):

$$\overline{X}_{n} = \frac{k_{p}[M]}{k_{z}[Z]} + 1$$
(30)

Data on the rate of consumption of the inhibitor as a function of conversion may also be used to obtain k_c/k_p (eq. 31):

$$\frac{k_z}{k_p} = \frac{[M]}{[Z]} \frac{d[Z]}{d[M]} = \frac{d\log[M]}{d\log[Z]}$$
(31)

It is clear that many procedures used to evaluate chain transfer constants can also be used to evaluate the kinetics of inhibition. The following sections will show that the mechanism for inhibition is often more complex than suggested by Scheme 5.11.

5.3.1 'Stable' Radicals

The kinetics and mechanism of inhibition by stable radicals has been reviewed by Rozantsev *et al.*¹⁸³ Ideally, for radicals to be useful inhibitors in radical polymerization they should have the following characteristics:

- (a) They should not add to, abstract from, or otherwise react with the monomer, solvent, *etc*.
- (b) They should not undergo self reaction or unimolecular decomposition.
- (c) They must react rapidly with the propagating and/or the initiator-derived radicals to terminate polymer chains.

Examples of radicals which are reported to meet these criteria are diphenylpicrylhydrazyl [DPPH, (22)], Koelsch radical (26), nitroxides [*e.g.* TEMPO (23), Fremy's Salt (24)], triphenylmethyl (25), galvinoxyl (27), and verdazyl radicals [*e.g.* triphenylverdazyl (28)]. These reagents have seen practical application in a number of contexts. They have been widely utilized in the determination of initiator efficiency (Section 3.3.1.1.3) and in mechanistic investigations (Section 3.5.2).

Stable radicals can show selectivity for particular radicals. For example, nitroxides do not trap oxygen-centered radicals yet react with carbon-centered radicals by coupling at or near diffusion controlled rates.^{179,184} This capability was utilized by Rizzardo and Solomon¹⁸⁵ to develop a technique for characterizing radical reactions and has been extensively used in the examination of initiation of radical polymerization (Section 3.5.2.4). In contrast DPPH, while an efficient inhibitor, shows little selectivity and its reaction with radicals is complex.¹⁸⁶



The efficiency of these inhibitors may depend on reaction conditions. For example: the reaction of radicals with stable radicals (*e.g.* nitroxides) may be reversible at elevated temperatures (Section 7.5.3); triphenylmethyl may initiate polymerizations (Section 7.5.2). A further complication is that the products may be capable of undergoing further radical chemistry. In the case of DPPH (**22**) this is attributed to the fact that the product is an aromatic nitro-compound (Section 5.3.7). Certain adducts may undergo induced decomposition to form a stable radical which can then scavenge further.

5.3.2 Oxygen

The role of oxygen in radical and other polymerizations has been reviewed by Bhanu and Kishore.¹⁸⁷ Rate constants for the reaction of carbon-centered radicals with oxygen are extremely fast, generally $\geq 10^9$ M⁻¹ s⁻¹.^{181,188} The initially formed

species are peroxy radicals **29**. These may abstract hydrogen or add monomer (Scheme 5.12).



Scheme 5.12

Thus, while polymerization may proceed in the presence of oxygen, it is an efficient scavenger of both initiating and propagating species in radical polymerization and usually steps must be taken to exclude oxygen or to minimize its effects. Typically, this involves conducting the experiment under vacuum or an inert atmosphere (*e.g.* nitrogen) or in a refluxing solvent. Oxygen may act as an inhibitor or retarder of polymerization, copolymerize (*e.g.* S polymerization), and/or facilitate chain transfer (*e.g.* VAc polymerization) or inhibition with other species (*e.g.* phenols – Section 5.3.4).

The effect observed is dependent on the reactivity of the monomer and other agents present in the polymerization medium towards hydroperoxy radicals **29**. If addition of **29** to monomer is slow, in relation to normal propagation, then retardation or inhibition will be observed. It should also be noted that, polymeric peroxides, one of the products of reaction with oxygen are potentially sources of additional radicals. These may complicate polymerization and can impair the properties of the final polymer (Section 8.2).

5.3.3 Monomers

Certain monomers may act as inhibitors in some circumstances. Reactivity ratios for VAc-S copolymerization ($r_S=0.02$, $r_{VAc}=22.3$) and rates of cross propagation are such that small amounts of S are an effective inhibitor of VAc polymerization. The propagating chain with a terminal VAc is very active towards S and adds even when S is present in small amounts. The propagating radical with S adds to VAc only slowly. Other vinyl aromatics also inhibit VAc polymerization.¹⁷⁴

| Ph | SR | ОН | |
|--------------------|--------------------|-------|--|
| H ₂ C=C | H ₂ C=Ċ | Н2С=С | |
| Ph | CO ₂ R' | СО2Р | |
| 30 | 31 | 32 | |

1,1-diphenylethylene (30) acts as a reversible inhibitor in polymerizations of S and MMA (Section 9.3.6).¹⁸⁹ Olefins with captodative substitution such as 31

rapidly scavenge radicals to give new radicals **33** which are unable or slow to reinitiate polymerization (Scheme 5.13).^{190,191} Termination is believed to occur exclusively by combination, thus telechelic polymers are available by appropriate choice of the initiator. The head to head coupling product **34** is stable at normal polymerization temperatures. However, at higher temperatures **34** undergoes reversible homolysis and radicals **33** may initiate polymerization (Section 9.3.5).^{191,192}



The chemistry is dependent on the particular substituents. Oxygen analogs of **31**, α -alkoxyacylates (**32**), do not inhibit polymerization but readily polymerize and copolymerize with reactivity ratios similar to methacrylate esters.¹⁹¹⁻¹⁹⁴

5.3.4 Phenols

Phenolic inhibitors such as hydroquinone (35), monomethylhydroquinone (*p*-methoxyphenol) (36) and 3,5-di-*t*-butylcatechol (37) are added to many commercial monomers to prevent polymerization during transport and storage.



Studies with simple radicals show that carbon-centered radicals react with phenols by abstracting a phenolic hydrogen (Scheme 5.14). The phenoxy radicals may then scavenge a further radical by C-C or C-O coupling or (in the case of hydroquinones) by loss of a hydrogen atom to give a quinone. The quinone may then react further (Section 5.4.4). Thus two or more propagating chains may be terminated for every mole of phenol.¹⁹⁵





However, by themselves, phenols are poor polymerization inhibitors¹⁹⁶⁻¹⁹⁸ (see also Table 5.6) and are reported to act as accelerants in the ATRP of MMA.¹⁹⁹ They (*e.g.* hydroquinone) are more effective inhibitors in the presence of oxygen.^{196-198,200} The mechanism for inhibition is shown in Scheme 5.15. The reaction of carbon centered radicals (including initiating and propagating radicals) with oxygen is very fast in relation to propagation. Phenols are excellent scavengers of hydroperoxy radicals.



Scheme 5.15

5.3.5 Quinones

Quinones may react with carbon-centered radicals by addition at oxygen or carbon, or by electron transfer (Scheme 5.16).^{174,182,195,201,202} The preferred reaction pathway depends both on the attacking radical and the particular quinone (halogenated quinones react preferentially by electron transfer). The radical formed may then scavenge another radical. There is also evidence that certain quinones [*e.g.* chloranil, benzoquinone (**38**)] may copolymerize under some conditions.²⁰³

The absolute rate constants for attack of carbon-centered radicals on *p*-benzoquinone (**38**) and other quinones have been determined to be in the range 10^7-10^8 M⁻¹ s⁻¹.^{182,204} This rate shows a strong dependence on the electrophilicity of the attacking radical and there is some correlation between the efficiency of various quinones as inhibitors of polymerization and the redox potential of the quinone. The complexity of the mechanism means that the stoichiometry of inhibition by these compounds is often not straightforward. Measurements of moles of inhibitor consumed for each chain terminated for common inhibitors of this class give values in the range 0.05-2.0.¹⁷⁶





5.3.6 Phenothiazine



In contrast to phenols (Section 5.3.4), phenothiazine (**39**) is reported to be an excellent scavenger of both carbon-centered and oxygen-centered radicals by hydrogen atom transfer and is also used to stabilize monomers in storage.¹⁹⁸

5.3.7 Nitrones, Nitro- and Nitroso-Compounds

Many nitrones and nitroso-compounds have been exploited as spin traps in elucidating radical reaction mechanisms by EPR spectroscopy (Section 3.5.2.1). The initial adducts are nitroxides which can trap further radicals (Scheme 5.17).

$$\begin{array}{ccccccc} \mathsf{C}\mathsf{H}_3 & \mathsf{R}^{\bullet} & \mathsf{O}^{\bullet}\mathsf{C}\mathsf{H}_3 & \mathsf{R}^{\bullet} & \mathsf{O}^{\circ}\mathsf{C}\mathsf{H}_3 \\ \mathsf{O}=\mathsf{N}-\mathsf{C}-\mathsf{C}\mathsf{H}_3 & & \mathsf{R}^{\bullet} & \mathsf{R}^{\bullet}\mathsf{O}^{\circ}\mathsf{C}\mathsf{H}_3 \\ \mathsf{C}\mathsf{H}_3 & & \mathsf{C}\mathsf{H}_3 & & \mathsf{R}^{\bullet} & \mathsf{R}^{\bullet}\mathsf{N}-\mathsf{C}^{\bullet}\mathsf{C}\mathsf{C}\mathsf{H}_3 \\ & & \mathsf{C}\mathsf{H}_3 & & \mathsf{C}\mathsf{H}_3 \end{array}$$

Scheme 5.17

Aromatic nitro-compounds have also seen use as inhibitors in polymerization and as additives in radical reactions. The reactions of these compounds with radicals are very complex and may involve nitroso-compounds and nitroxide intermediates.^{205,206} In this case, up to four moles of radicals may be consumed per mole of nitro-compound. The overall mechanism in the case of nitrobenzene has been written as shown in Scheme 5.18. The alkoxyamine **40** can be isolated in

good yield from the decomposition of AIBN in the presence of nitrobenzene (Scheme 5.18, R=cyanoisopropyl).¹⁰⁹



5.3.8 Transition Metal Salts

Transition metal salts trap carbon-centered radicals by electron transfer or by ligand transfer. These reagents often show high specificity for reaction with specific radicals and the rates of trapping may be correlated with the nucleophilicity of the radical (Table 5.6). For example, PS• radicals are much more reactive towards ferric chloride than acrylic propagating species.²⁰⁷

Various transition metal salts have been applied in quantitative determination of initiation reactions (Section 3.5.2.2). Under some circumstances, the ligand transfer may be reversible under the polymerization conditions. This chemistry forms the basis of ATRP (Section 9.4).

5.4 References

- North, A.M. In *Reactivity, Mechanism and Structure in Polymer Chemistry*; Jenkins, A.D.; Ledwith, A., Eds.; Wiley: London, 1974; p 142.
- O'Driscoll, K.F. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 161.
- Buback, M.; Egorov, M.; Gilbert, R.G.; Kaminsky, V.; Olaj, O.F.; Russell, G.T.; Vana, P.; Zifferer, G. Macromol. Chem. Phys. 2002, 201, 2570.
- 4. Russell, G.T. Macromol. Theory Simul. 1995, 4, 519.
- 5. Russell, G.T. Macromol. Theory Simul. 1995, 4, 549.
- 6. Russell, G.T. Macromol. Theory Simul. 1995, 4, 497.
- 7. Russell, G.T.; Napper, D.H.; Gilbert, R.G. Macromolecules 1988, 21, 2133.
- de Kock, J.B.L.; van Herk, A.M.; German, A.L. J. Macromol. Sci., Rev. Macromol. Chem. Phys. 2001, C41, 199.
- de Kock, J.B.L.; Klumperman, B.; van Herk, A.M.; German, A.L. Macromolecules 1997, 30, 6743.
- Buback, M.; Garcia-Rubio, L.H.; Gilbert, R.G.; Napper, D.H.; Guillot, J.; Hamielec, A.E.; Hill, D.; O'Driscoll, K.F.; Olaj, O.F.; Shen, J.; Solomon, D.H.; Moad, G.; Stickler, M.; Tirrell, M.; Winnik, M.A. J. Polym. Sci., Part C: Polym. Lett. 1988, 26, 293.
- Buback, M.; Gilbert, R.G.; Russell, G.T.; Hill, D.J.T.; Moad, G.; O'Driscoll, K.F.; Shen, J.; Winnik, M.A. J. Polym. Sci., Part A: Polym. Chem. 1992, 30, 851.
- 12. Gilbert, R.G. Pure Appl. Chem. 1992, 64, 1563.
- Eastmond, G.C. In *Comprehensive Chemical Kinetics*; Bamford, C.H.; Tipper, C.F.H., Eds.; Elsevier: Amsterdam, 1976; Vol. 14A, p 1.

- Bamford, C.H. In *Encyclopedia of Polymer Science and Engineering*, 2nd cd.; Mark, H.F.; Bikales, N.M.; Overberger, C.G.; Menges, G., Eds.; Wiley: New York, 1988; Vol. 13, p 708.
- Mills, I.; Cvitas, T.; Homann, K.; Kallay, N.; Kuchitsu, K. Quantities, Units and Symbols in Physical Chemistry, 1988 ed.; Blackwell Scientific Publications: Oxford, 1988.
- Kamachi, M.; Yamada, B. In *Polymer Handbook*, 4th ed.; Brandup, J.; Immergut, E.H.; Grulke, E.A., Eds.; John Wiley and Sons: New York, 1999; p II/77.
- 17. Beuermann, S.; Buback, M. Prog. Polym. Sci. 2002, 27, 191.
- 18. Buback, M.; Busch, M.; Kowollik, C. Macromol. Theory Simul. 2000, 9, 442.
- Beuermann, S.; Buback, M.; Russell, G.T. Macromol. Chem. Phys. 1995, 196, 2493.
- Yamada, B.; Westmoreland, D.G.; Kobatake, S.; Konosu, O. Prog. Polym. Sci. 1999, 24, 565.
- 21. Kamachi, M. J. Polym. Sci., Part A: Polym. Chem. 2001, 40, 269.
- Carswell, T.G.; Hill, D.J.T.; Londero, D.I.; O'Donnell, J.H.; Pomery, P.J.; Winzor, C.L. *Polymer* 1992, 33, 137.
- Moad, G.; Shipp, D.A.; Smith, T.A.; Solomon, D.H. J. Phys. Chem. A 1999, 103, 6580.
- 24. Shipp, D.A.; Solomon, D.H.; Smith, T.A.; Moad, G. *Macromolecules* 2003, *36*, 2032.
- Moad, G.; Shipp, D.A.; Smith, T.A.; Solomon, D.H. Macromolecules 1997, 30, 7627.
- Zetterlund, P.B.; Yamauchi, S.; Yamada, B. Macromol. Chem. Phys. 2004, 205, 778.
- 27. Schulz, G.V.; Harborth, G. Z. Phys. Chem. 1939, B43, 25.
- Flory, P.J. Principles of Polymer Chemistry; Cornell University Press: Ithaca, New York, 1953.
- 29. Bamford, C.II.; Barb, W.G.; Jenkins, A.D.; Onyon, P.F. *The Kinetics of Vinyl Polymerization by Radical Mechanisms*; Butterworths: London, 1958.
- 30. North, A.M.; Reed, G.A. Trans. Faraday Soc. 1961, 57, 859.
- 31. North, A.M.; Reed, G.A. J. Polym. Sci., Part A 1963, 1, 1311.
- 32. Benson, S.W.; North, A.M. J. Am. Chem. Soc. 1962, 84, 935.
- 33. Fischer, H.; Paul, H. Acc. Chem. Res. 1987, 20, 200.
- 34. Balke, S.T.; Hamielec, A.E. J. Appl. Polym. Sci. 1973, 17, 905.
- 35. Soh, S.K.; Sundberg, D.C. J. Polym. Sci., Polym. Chem. Ed. 1982, 20, 1345.
- 36. Norrish, R.G.W.B., E. F. Proc. Roy. Soc. London 1939, A171, 147.
- 37. Trommsdorf, E.; Kohle, H.; Lagally, P. Makromol. Chem. 1948, 1, 169.
- 38. Norrish, R.G.W.; Smith, R.R. Nature 1942, 150, 336.
- 39. Mahabadi, H.K.; O'Driscoll, K.F. J. Polym. Sci., Polym. Chem. Ed. 1977, 15, 283.
- 40. Olaj, O.F.; Zifferer, G. Macromolecules 1987, 20, 850.
- 41. Zhu, S.; Hamielee, A.E. Macromolecules 1989, 22, 3093.
- 42. Yasukawa, T.; Murakami, K. Macromolecules 1981, 14, 227.
- 43. Yasukawa, T.; Murakami, K. Polymer 1980, 21, 1423.
- 44. Marten, F.L.; Hamielec, A.E. J. Appl. Polym. Sci. 1982, 27, 489.
- 45. Bamford, C.H. Polymer 1990, 31, 1720.
- 46. Olaj, O.F.; Zifferer, G.; Gleixner, G. Makromol. Chem. 1986, 187, 977.
- 47. Olaj, O.F.; Zifferer, G. Makromol. Chem., Rapid Commun. 1982, 3, 549.
- 48. Griller, D. In Landoldt-Bornstein, New Series, Radical Reaction Rates in Solution; Fischer, H., Ed.; Springer-Verlag: Berlin, 1984; Vol. II/13a, p 5.

- 49. Deady, M.; Mau, A.W.H.; Moad, G.; Spurling, T.H. Makromol. Chem. 1993, 194, 1691.
- 50. Smith, G.B.; Russell, G.T.; Heuts, J.P.A. Macromol. Theory Simul. 2003, 12, 299.
- 51. khokhlov, A.R. Makromol. Chem., Rapid Commun. 1981, 2, 633.
- 52. Buback, M.; Egorov, M.; Feldermann, A. Macromolecules 2004, 37, 1768.
- 53. Buback, M.; Egorov, M.; Junkers, T.; Panchenko, E. *Macromol. Rapid Commun.* **2004**, *25*, 1004.
- 54. Cardenas, J.N.; O'Driscoll, K.F. J. Polym. Sci., Polym. Chem. Ed. 1976, 14, 883.
- 55. Russell, G.T. Macromol. Theory Simul. 1994, 3, 439.
- 56. Faldi, A.; Tirrell, M.; Lodge, T.P. Macromolecules 1994, 27, 4176.
- 57. O'Shaughnessy, B.; Yu, J. Macromolecules 1994, 27, 5079.
- 58. O'Shaughnessy, B.; Yu, J. Macromolecules 1994, 27, 5067.
- 59. Bamford, C.H. Eur. Polym. J. 1989, 25, 683.
- 60. Bamford, C.H. Eur. Polym. J. 1993, 29, 313.
- 61. Bamford, C.H. Eur. Polym. J. 1990, 26, 719.
- 62. Bamford, C.H. Eur. Polym. J. 1991, 27, 1289.
- 63. Bamford, C.H. Eur. Polym. J. 1990, 26, 1245.
- 64. Verravalli, M.S.; Rosen, S.L. J. Polym. Sci., Part B: Polym. Phys. 1990, 28, 775.
- 65. Soh, S.K.; Sundberg, D.C. J. Polym. Sci., Polym. Chem. Ed. 1982, 20, 1299.
- 66. Soh, S.K.; Sundberg, D.C. J. Polym. Sci., Polym. Chem. Ed. 1982, 20, 1315.
- 67. Soh, S.K.; Sundberg, D.C. J. Polym. Sci., Polym. Chem. Ed. 1982, 20, 1331.
- 68. O'Neil, G.A.; Wisnudel, M.B.; Torkelson, J.M. Macromolecules 1998, 31, 4537.
- 69. O'Neil, G.A.; Torkelson, J.M. Macromolecules 1999, 32, 411.
- 70. Ito, K. Polym. J. 1980, 12, 499.
- 71. Tulig, T.J.; Tirrell, M. Macromolecules 1981, 14, 1501.
- 72. de Gennes, P.G. J. Chem. Phys. 1982, 76, 3322.
- 73. O'Shaughnessy, B.; Yu, J. Macromolecules 1998, 31, 5240.
- 74. Kim, J.U.; O'Shaughnessy, B. Macromolecules 2004, 37, 1630.
- 75. Buback, M.; Huckestein, B.; Russell, G.T. Macromol. Chem. Phys. 1994, 195, 539.
- 76. Chiu, W.Y.; Carrat, G.M.; Soong, D.S. *Macromolecules* **1983**, *16*, 348.
- Gilbert, R.G. Emulsion Polymerization: A Mechanistic Approach; Academic Press: London, 1995.
- Lovell, P.A.; El-Aasser, M.S., Eds. Emulsion Polymerization and Emulsion Polymers; John Wiley & Sons: London, 1997.
- 79. Smith, W.V.; Ewart, R.H. J. Chem. Phys. 1948, 16, 592.
- Krstina, J.; Moad, C.L.; Moad, G.; Rizzardo, E.; Berge, C.T.; Fryd, M. Macromol. Symp. 1996, 111, 13.
- Krstina, J.; Moad, G.; Rizzardo, E.; Winzor, C.L.; Berge, C.T.; Fryd, M. Macromolecules 1995, 28, 5381.
- Moad, G.; Chiefari, J.; Krstina, J.; Postma, A.; Mayadunne, R.T.A.; Rizzardo, E.; Thang, S.H. Polym. Int. 2000, 49, 933.
- Vana, P.; Davis, T.R.; Barner-Kowollik, C. Macromolecular Rapid Communications 2002, 23, 952.
- Cacioli, P.; Moad, G.; Rizzardo, E.; Serelis, A.K.; Solomon, D.H. *Polym. Bull.* 1984, 11, 325.
- Kashiwagi, T.; Inaba, A.; Brown, J.E.; Hatada, K.; Kitayama, T.; Masuda, E. Macromolecules 1986, 19, 2160.
- 86. Meisters, A.; Moad, G.; Rizzardo, E.; Solomon, D.H. Polym. Bull. 1988, 20, 499.
- 87. Manring, L.E. Macromolecules 1989, 22, 2673.
- 88. Bamford, C.H.; White, E.F.T. Trans. Faraday Soc. 1958, 54, 268.

- Bizilj, S.; Kelly, D.P.; Serelis, A.K.; Solomon, D.H.; White, K.E. Aust. J. Chem. 1985, 38, 1657.
- Cacioli, P.; Hawthorne, D.G.; Laslett, R.L.; Rizzardo, E.; Solomon, D.H. J. Macromol. Sci., Chem. 1986, A23, 839.
- 91. Morawetz, H. J. Polym. Sci., Polym. Symp. 1978, 62, 271.
- 92. Overberger, C.G.; Finestone, A.B. J. Am. Chem. Soc. 1956, 78, 1638.
- 93. Gibian, M.J.; Corley, R.C. J. Am. Chem. Soc. 1972, 94, 4178.
- 94. Shelton, J.R.; Liang, C.K. J. Org. Chem. 1973, 38, 2301.
- 95. Gleixner, G.; Olaj, O.F.; Breitenbach, J.W. Makromol. Chem. 1979, 180, 2581.
- 96. Schreck, V.A.; Serelis, A.K.; Solomon, D.H. Aust. J. Chem. 1989, 42, 375.
- 97. Nelsen, S.F.; Bartlett, P.D. J. Am. Chem. Soc. 1966, 88, 137.
- 98. Ncuman, R.C., Jr.; Amrich, M.J., Jr. J. Org. Chem. 1980, 45, 4629.
- 99. Fraenkel, G.; Geckle, M.J. J. Chem. Soc., Chem. Commun. 1980, 55.
- 100. Langhals, H.; Fischer, H. Chem. Ber. 1978, 111, 543.
- 101. Skinner, K.J.; Hochster, H.S.; McBride, J.M. J. Am. Chem. Soc. 1974, 96, 4301.
- Businelli, L.; Gnanou, Y.; Maillard, B. Macromolecular Chemistry and Physics 2000, 201, 2805.
- 103. Trecker, D.J.; Foote, R.S. J. Org. Chem. 1968, 33, 3527.
- 104. Kodaira, K.; Ito, K.; Iyoda, S. Polym. Commun. 1987, 28, 86.
- 105. Kelly, D.P.; Serelis, A.K.; Solomon, D.H.; Thompson, P.E. Aust. J. Chem. 1987, 40, 1631.
- 106. Mackie, J.S.; Bywater, S. Can. J. Chem. 1957, 35, 570.
- 107. Neumann, W.P.; Stapel, R. Chem. Ber. 1986, 119, 3422.
- 108. Serelis, A.K.; Solomon, D.H. Polym. Bull. 1982, 7, 39.
- 109. Gingras, B.A.; Waters, W.A. J. Chem. Soc. 1954, 1920.
- 110. Barbe, W.; Rüchardt, C. Makromol. Chem. 1983, 184, 1235.
- 111. Serelis, A.K. Personal Communication.
- 112. Konter, W.; Bömer, B.; Köhler, K.H.; Heitz, W. Makromol. Chem. 1981, 182, 2619.
- Barton, J.; Capek, I.; Juranicova, V.; Riedel, S. Makromol. Chem., Rapid Commun. 1986, 7, 521.
- 114. Guth, W.; Heitz, W. Makromol. Chem. 1976, 177, 1835.
- 115. Jaffe, A.B.; Skinner, K.J.; McBride, J.M. J. Am. Chem. Soc. 1972, 94, 8510.
- 116. Minato, T.; Yamabe, S.; Fujimoto, H.; Fukui, K. *Bull. Chem. Soc. Japan* **1978**, *51*, 1.
- 117. Krstina, J.; Moad, G.; Willing, R.I.; Danek, S.K.; Kelly, D.P.; Jones, S.L.; Solomon, D.H. Eur. Polym. J. 1993, 29, 379.
- 118. Gibian, M.J.; Corley, R.C. Chem. Rev. 1973, 73, 441.
- 119. Heitz, W. In *Telechelic Polymers: Synthesis and Applications*; Goethals, E.J., Ed.; CRC Press: Boca Raton, Florida, 1989; p 61.
- 120. Bamford, C.H.; Dyson, R.W.; Eastmond, G.C. Polymer 1969, 10, 885.
- 121. Stickler, M. Makromol. Chem. 1979, 180, 2615.
- 122. Burnett, G.M.; North, A.M. Makromol. Chem. 1964, 73, 77.
- 123. Bamford, C.H.; Jenkins, A.D.; Johnston, R. Trans. Faraday Soc. 1959, 55, 179.
- 124. Olaj, O.F.; Breitenbach, J.W.; Wolf, B. Monatsh. Chem. 1964, 95, 1646.
- 125. Dawkins, J.V.; Yeadon, G. Polymer 1979, 20, 981.
- 126. Baker, C.A.; Williams, R.J.P. J. Chem. Soc. 1956, 2352.
- 127. Henrici-Olive, G.; Olive, S. Fortschr. Hochpolym. Forsch. 1961, 2, 496.
- 128. Rudin, A. In Comprehensive Polymer Science, Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 239.
- 129. Moad, G.; Moad, C.L. Macromolecules 1996, 29, 7727.

- Zammit, M.D.; Davis, T.P.; Haddleton, D.M.; Suddaby, K.G. Macromolecules 1997, 30, 1915.
- 131. Catalgil-Giz, H.; Giz, A.; Oncul-Koc, A. Polym. Bull. 1999, 43, 215.
- Hensley, D.R.; Goodrich, S.D.; Harwood, H.J.; Rinaldi, P.L. Macromolecules 1994, 27, 2351.
- 133. Bevington, J.C.; Melville, H.W.; Taylor, R.P. J. Polym. Sci. 1954, 14, 463.
- 134. Ayrey, G.; Moore, C.G. J. Polym. Sci. 1959, 36, 41.
- 135. Hakozaki, J.; Yamada, N. In J. Chem. Soc., Japan, 96263r (1968) ed., 1967; Vol. 70, p 1560.
- 136. O'Driscoll, K.F.; Bevington, J.C. Eur. Polym. J. 1985, 21, 1039.
- 137. Bamford, C.H.; Jenkins, A.D. Nature 1955, 176, 78.
- 138. Mayo, F.R.; Gregg, R.A.; Matheson, M.S. J. Am. Chem. Soc. 1951, 73, 1691.
- 139. Johnson, D.H.; Tobolsky, A.V. J. Am. Chem. Soc. 1952, 74, 938.
- 140. Braks, J.G.; Huang, R.Y.M. J. Appl. Polym. Sci. 1978, 22, 3111.
- 141. Henrici-Olive, G.; Olive, S. J. Polym. Sci. 1960, 48, 329.
- 142. Bevington, J.C.; Melville, H.W.; Taylor, R.P. J. Polym. Sci. 1954, 12, 449.
- 143. Kolthoff, I.M.; O'Connor, P.R.; Hansen, J.L. J. Polym. Sci. 1955, 15, 459.
- 144. Arnett, L.M.; Peterson, J.H. J. Am. Chem. Soc. 1952, 74, 2031.
- 145. Bessiere, J.-M.; Boutevin, B.; Loubet, O. Polym. Bull. 1993, 31, 673.
- 146. Moad, G.; Solomon, D.H.; Johns, S.R.; Willing, R.I. *Macromolecules* **1984**, *17*, 1094.
- 147. Moad, G.; Solomon, D.H.; Johns, S.R.; Willing, R.I. *Macromolecules* **1982**, *15*, 1188.
- 148. Ayrey, G.; Levitt, F.G.; Mazza, R.J. Polymer 1965, 6, 157.
- 149. Olaj, O.F.; Kaufmann, H.F.; Breitenbach, J.W.; Bieringer, H. J. Polym. Sci., Polym. Lett. Ed. 1977, 15, 229.
- 150. Berger, K.C.; Meyerhoff, G. Makromol. Chem. 1975, 176, 1983.
- 151. Barson, C.A.; Bevington, J.C.; Hunt, B.J. Polymer 1998, 39, 1345.
- Bevington, J.C.; Breuer, S.W.; Huckerby, T.N.; Hunt, B.J.; Jones, R. Eur. Polym. J. 1998, 34, 539.
- 153. Yoshikawa, C.; Goto, A.; Fukuda, T. e-Polymers 2002, 2002, 13.
- 154. Allen, P.W.; Ayrey, G.; Merrett, F.M.; Moore, C.G. J. Polym. Sci. 1956, 22, 549.
- 155. Schulz, G.V.; Henrici-Olive, G.; Olive, S. Makromol. Chem. 1959, 31, 88.
- 156. Bamford, C.H.; Eastmond, G.C.; Whittle, D. Polymer 1969, 10, 771.
- 157. Chaudhuri, A.K.; Palit, S.R. J. Polym. Sci., Part A-1 1968, 6, 2187.
- 158. Braks, J.G.; Mayer, G.; Huang, R.Y.M. J. Appl. Polym. Sci. 1980, 25, 449.
- 159. Ayrey, G.; Haynes, A.C. Eur. Polym. J. 1973, 9, 1029.
- 160. Hatada, K.; Kitayama, T.; Ute, K.; Terawaki, Y.; Yanagida, T. Macromolecules 1997, 30, 6754.
- 161. Boudevska, II.; Brutchkov, C.; Platchkova, S. Makromol. Chem. 1981, 182, 3257.
- 162. Hatada, K.; Kitayama, T.; Masuda, E. Polym. J. 1986, 18, 395.
- 163. Ayrey, G.; Humphrey, M.J.; Poller, R.C. Polymer 1977, 18, 840.
- 164. Bailey, B.E.; Jenkins, A.D. Trans. Faraday Soc. 1960, 56, 903.
- 165. Bevington, J.C.; Eaves, D.E. Trans. Faraday Soc. 1959, 55, 1777.
- 166. Patron, L.; Bastianelli, U. Appl. Polym. Symp. 1974, 25, 105.
- 167. Funt, B.L.; Paskia, W. Can. J. Chem. 1960, 38, 1865.
- 168. Danusso, F.; Pajaro, G.; Sianesi, D. Chim. Ind. (Milan) 1959, 41, 1170.
- 169. Talamini, G.V., G. Chim. Ind. (Milan) 1964, 46, 16.
- 170. Park, G.S.; Smith, D.G. Makromol. Chem. 1970, 131, 1.
- 171. Starnes, W.H., Jr.; Plitz, I.M.; Schilling, F.C.; Villacorta, G.M.; Park, G.S.; Saremi, A.H. Macromolecules 1984, 17, 2507.

- 172. Atkinson, W.H.; Bamford, C.H.; Eastmond, G.C. Trans. Faraday Soc. 1970, 66, 1446.
- 173. Bamford, C.H. In Comprehensive Polymer Science; Agarwal, S.L.; Russo, S., Eds.; Pergamon: Oxford, 1992; Vol. Suppl. 1, p 1.
- 174. Tüdos, F.; Földes-Berezsnich, T. Prog. Polym. Sci. 1989, 14, 717.
- 175. Eastmond, G.C. In *Comprehensive Chemical Kinetics*; Bamford, C.H.; Tipper, C.F.H., Eds.; Elsevier: Amsterdam, 1976; Vol. 14A, p 153.
- 176. Goldfinger, G.; Yee, W.; Gilbert, R.D. In Encyclopedia of Polymer Science and Technology; Mark, H.F.; Gaylord, N.M.; Bikales, N.M., Eds.; Wiley: New York, 1967; Vol. 7, p 644.
- 177. Bovey, F.A.; Kolthoff, I.M. Chem. Rev. 1948, 42, 491.
- 178. Bowry, V.W.; Ingold, K.U. J. Am. Chem. Soc. 1992, 114, 4992.
- 179. Beckwith, A.L.J.; Bowry, V.W.; Moad, G. J. Org. Chem. 1988, 53, 1632.
- 180. Beckwith, A.L.J.; Bowry, V.W.; Ingold, K.U. J. Am. Chem. Soc. 1992, 114, 4983.
- 181. Maillard, B.; Ingold, K.U.; Scaiano, J.C. J. Am. Chem. Soc. 1983, 105, 5095.
- 182. Citterio, A.; Arnoldi, A.; Minisci, F. J. Org. Chem. 1979, 44, 2674.
- 183. Rozantsev, E.G.; Gol'dfein, M.D.; Trubnikov, A.V. Russ. Chem. Rev. (Engl. Transl.) 1986, 55, 1070.
- 184. Chateauneuf, J.; Lusztyk, J.; Ingold, K.U. J. Org. Chem. 1988, 53, 1629.
- 185. Rizzardo, E.; Solomon, D.H. Polym. Bull. 1979, 1, 529.
- 186. Hawthorne, D.G.; Solomon, D.H. J. Macromol. Sci., Chem. 1972, A6, 661.
- 187. Bhanu, V.A.; Kishore, K. Chem. Rev. 1991, 91, 99.
- 188. Neta, P.; Huie, R.E.; Ross, A.B. J. Chem. Phys. Ref., Data 1990, 19, 413.
- 189. Wieland, P.C.; Raether, B.; Nuyken, O. Macromol. Rapid Commun. 2001, 22, 700.
- 190. Mignani, S.; Janousek, Z.; Merenyi, R.; Viehe, H.G.; Riga, J.; Verbist, J. *Tetrahedron Lett.* **1984**, *25*, 1571.
- 191. Tanaka, II. Prog. Polym. Sci. 2003, 28, 1171.
- 192. Tanaka, H.; Teraoka, Y.; Sato, T.; Ota, T. Makromol. Chem. 1993, 194, 2719.
- 193. Hageman, H.J.; Oosterhoff, P.; Overeem, T.; Polman, R.J.; van der Werf, S. Makromol. Chem. 1985, 186, 2483.
- 194. Tanaka, H.; Kameshima, T.; Sasai, K.; Sato, T.; Ota, T. Makromol. Chem. 1991, 192, 427.
- 195. Kharasch, M.S.; Kawahara, F.; Nudenberg, W. J. Org. Chem. 1954, 19, 1977.
- 196. Levy, L.B. J. Appl. Polym. Sci. 1996, 60, 2481.
- 197. Levy, L.B. J. Polym. Sci., Part A: Polym. Chem. 1992, 30, 569.
- 198. Schulze, S.; Vogel, II. Chem. Eng. Technol. 1998, 21, 829.
- 199. Haddleton, D.M.; Clark, A.J.; Crossman, M.C.; Duncalf, D.J.; Heming, A.M.; Morsley, S.R.; Shooter, A.J. Chem. Commun. 1997, 1173.
- 200. Chen, S.; Tsai, L. Makromol. Chem. 1986, 187, 653.
- 201. Bevington, J.C.; Ghanem, N.A.; Melville, H.W. Trans. Faraday Soc. 1955, 51, 946.
- 202. Price, C.C.; Read, D.H. J. Polym. Res. 1946, 1, 44.
- 203. Yassin, A.A.; Rizk, N.A. Eur. Polym. J. 1977, 13, 441.
- 204. Golubev, V.B.; Mun, G.A.; Zubov, V.P. Russ. J. Phys. Chem. (Engl. Transl.) 1986, 60, 347.
- Perkins, M.J.; Chalfont, G.R.; Hey, D.H.; Liang, K.S.Y. J. Chem. Soc. (B) 1971, 233.
- 206. Chalfont, G.R.; Hey, D.H.; Liang, K.S.Y.; Perkins, M.J. Chem. Commun. (London) 1967, 367.
- 207. Bamford, C.II.; Jenkins, A.D.; Johnston, R. Proc. R. Soc., London 1957, A239, 214.

6.1 Introduction

Chain transfer is the reaction of a propagating radical with a non-radical substrate (X-Y, Scheme 6.1) to produce a dead polymer chain and a new radical (Y•) capable of initiating a polymer chain. The transfer agent (X-Y) may be a deliberate additive (*e.g.* a thiol) or it may be the initiator, monomer, polymer, solvent or an adventitious impurity.



Transfer without reinitiation is called inhibition and is discussed in Section 5.3. There are also situations where the reaction produces a dead polymer chain and a radical that is less reactive than the propagating radical but still capable of reinitiating polymerization. The process is then termed retardation or degradative chain transfer.

6.2 Chain Transfer

The general mechanism of chain transfer as first proposed by Flory,^{1,2} may be written schematically as shown in Scheme 6.2. The overall process involves a propagating chain (P_n^{\bullet}) reacting with a transfer agent (T) to terminate one polymer chain and produce a radical (T•) that initiates a new chain (P_1^{\bullet}).

Transfer agents find widespread use in both industrial and laboratory polymer syntheses. They are used to control:

- (a) The molecular weight of polymers
- (b) The polymerization rate and exotherm (by mitigating the gel or Norrish-Trommsdorff effect)
- (c) The polymer end groups.

transfer to transfer agent or solvent

 $P_n \bullet + T \longrightarrow P_n + T \bullet \qquad R_{trT} = k_{trT} [P \bullet][T]$ reinitiation $R_{iT} = k_{iT} [T \bullet] [M]; k_{iT} > k_{p}$ $T \bullet + M \rightarrow P_1 \bullet$ transfer to initiator $P_{n}\bullet + I_{2} \quad \rightarrow \quad P_{n} + I\bullet$ $R_{
m trT} - k_{
m trT} \left[{
m P} ullet
ight] \left[{
m I}_2
ight]$ reinitiation $I \bullet + M$ $\rightarrow P_1 \bullet$ $R_{iT} = k_{iT} [1\bullet][M]; k_i \ge k_p$ transfer to monomer $P_n \bullet + M \rightarrow P_n + M \bullet \qquad R_{trT} = k_{trT} [P \bullet][M]$ reinitiation $M \bullet + M \rightarrow P_1 \bullet$ $R_{iT} = k_{iT} [M \bullet] [M]; k_{iM} \ge k_p$ Scheme 6.2

General aspects of chain transfer have been reviewed by Chiefari and Rizzardo,³ Barson,⁴ Farina,⁵ Eastmond⁶ and Palit *et al.*⁷ The use of chain transfer in producing telechelic and other functional polymers has been reviewed by Boutevin,⁸ Heitz,⁹ Corner¹⁰ and Starks¹¹ and is discussed in Section 7.5.2. There are two main mechanisms which should be considered in any discussion of chain transfer: (a) atom or group transfer by homolytic substitution (Section 6.2.2) and (b) addition-fragmentation (Section 6.2.3).

Even in the absence of added transfer agents, all polymerizations may be complicated by transfer to initiator (Sections 3.2.10 and 3.3), solvent (Section 6.2.2.5), monomer (Section 6.2.6) or polymer (Section 6.2.7). The significance of these transfer reactions is dependent upon the particular propagating radicals involved, the reaction medium and the polymerization conditions. Thiol-ene polymerization consists of sequential chain transfer and reinitiation steps and ideally no monomer consumption by propagation (Section 7.5.3).

For efficient chain transfer, the rate constant for reinitiation following transfer $(k_{iT}; \text{ refer Scheme 6.2})$ must be greater than or equal to that for propagation (k_p) . In these circumstances, the presence of the transfer agent reduces the molecular weight of the polymer without directly influencing the rate of polymerization. If, however, $k_{iT} < k_p$ then polymerization will be retarded and the likelihood that the transfer agent-derived radical (T•) will undergo side reactions such as primary radical termination is increased. Thus, retardation is much more likely in polymerizations of high k_p monomers (*e.g.* MA, VAc) than it is with lower k_p monomers (*e.g.* S, MMA). Retardation is discussed in greater detail in Section 5.3.

Even when $k_{iT} \ge k_p$, the rate of polymerization at higher conversions will often be lower that in the absence of a transfer agent due to a reduced gel or Norrish-

Trommsdorf effect. One cause of this autoacceleration phenomenon is a reduced rate of radical-radical termination brought about by the immobilization of long chains through entanglement at higher conversions (Section 5.2.1.4). In the presence of a transfer agent, the population of short chains is higher and, because the ultimate molecular weight is lower, there are fewer chain entanglements.

The number average degree of polymerization (X_n) of polymer formed at any given instant during the polymerization can be expressed simply as the rate of monomer usage in propagation divided by the rate of formation of polymer molecules (the overall rate of termination). Thus according to classical kinetics, if termination is only by radical-radical reaction or chain transfer the degree of polymerization is given by eq. 1:

$$\overline{X}_{n} = \frac{k_{p}[\mathbf{M}]}{\left(1 + \frac{k_{n}}{k_{i}}\right)k_{i}[\mathbf{P}^{\bullet}] + k_{n,i}[\mathbf{T}] + k_{n,M}[\mathbf{M}]}$$
(1)

This can be rewritten as eq. 2:

.

$$\frac{1}{\overline{X}_{n}} = \frac{\left(1 + \frac{k_{n}}{k_{t}}\right)k_{t}[\mathbf{P}^{\bullet}]}{k_{p}[\mathbf{M}]} + \frac{k_{n,1}[\mathbf{T}]}{k_{p}[\mathbf{M}]} + \frac{k_{n,1}[\mathbf{I}]}{k_{p}[\mathbf{M}]} + \frac{k_{n,M}}{k_{p}}$$
(2)

The ratio k_{tr}/k_{p} is called the transfer constant (C_{tr}) and C_{T} , C_{1} and C_{M} are the transfer constants for transfer to transfer agent, initiator and monomer respectively. Appropriate substitution gives eq. 3:

$$\frac{1}{\overline{X}_{n}} = \frac{\left(1 + \frac{k_{n}}{k_{r}}\right)k_{r}[\mathbf{P}^{\bullet}]}{k_{p}[\mathbf{M}]} + C_{T}\frac{|\mathbf{T}|}{[\mathbf{M}]} + C_{I}\frac{[\mathbf{I}]}{[\mathbf{M}]} + C_{M}$$
(3)

The degree of polymerization in the absence of a chain transfer agent is given by eq. 4:

$$\frac{1}{\overline{X}_{n0}} = \frac{\left(1 + \frac{k_{ul}}{k_{l}}\right)k_{l}[\mathbf{P}^{\bullet}]}{k_{p}[\mathbf{M}]} + C_{l}\frac{[\mathbf{I}]}{[\mathbf{M}]} + C_{M}$$
(4)

Thus,

$$\frac{1}{\overline{X}_n} = \frac{1}{\overline{X}_{n0}} + C_{\mathrm{T}} \frac{[\mathrm{T}]}{[\mathrm{M}]}$$
(5)

This equation (eq. 5) is commonly known as the Mayo equation.¹² The equation is applicable at low (zero) conversion and is invalidated if the rate constants are chain length dependent.

The magnitude of a transfer constant depends on structural features of both the attacking radical and the transfer agent. A C_{tr} of unity has been called ideal. In these circumstances, the transfer agent:monomer ratio ([T]:[M]) will remain constant throughout the polymerization.¹⁰ This means that \overline{X}_n remains constant with conversion and the dispersity of the molecular weight distribution is thus minimized ($\overline{X}_w/\overline{X}_n$ close to 2.0). If C_{tr} is high (>>1), the transfer agent will be consumed rapidly during the carly stages of polymerization and the polymerization will be unregulated at higher conversion. If, on the other hand, C_{tr} is low (<<1), [T]:[M] will increase as the polymerization progresses and there will be a corresponding decrease in \overline{X}_n with conversion. In both circumstances, a broad molecular weight distribution will result from a high conversion batch polymerization. It is often possible to overcome these problems by establishing an incremental protocol for monomer and/or transfer agent addition such that [T]:[M] is maintained at a constant value throughout the polymerization.

The rate constants for chain transfer and propagation may well have a different dependence on temperature (i.e. the two reactions may have different activation parameters) and, as a consequence, transfer constants are temperature dependent. The temperature dependence of $C_{\rm tr}$ has not been determined for most transfer agents. Care must therefore be taken when using literature values of $C_{\rm tr}$ if the reaction conditions are different from those employed for the measurement of $C_{\rm tr}$. For cases where the transfer constant is close to 1.0, it is sometimes possible to choose a reaction temperature such that the transfer constant is 1.0 and thus obtain ideal behavior.¹³

The value of C_{tr} in homopolymerization can show significant chain length dependence for chain lengths ≤ 5 . Some values of transfer constants for homolytic substitution chain transfer agents are shown in Table 6.1.¹¹ The variation in C_{tr} with chain length can reflect variations in k_p or k_{tr} or (most likely) both. The data provided in Section 4.5.3 show that k_p can be dependent on chain length for at least the first few propagation steps. The magnitude of the effect on C_{tr} for a given monomer varies according to the particular transfer agent. This indicates the sensitivity of k_p and k_{tr} to the penultimate unit is different. Chain transfer constants in addition-fragmentation (Section 6.2.3.4) and catalytic chain transfer have also been shown to be chain length dependent (Section 6.2.5).

Bamford¹⁴ has provided evidence that, in copolymerization, penultimate unit effects can be important in determining the reactivity of propagating radicals toward transfer agents. The magnitude of this effect also depends on the particular monomers and transfer agent involved. The finding that the most pronounced remote unit effects are observed for the most bulky transfer agents (Section 6.2.2.4), has been taken as evidence that the magnitude of the remote unit effect is determined at least in part by steric factors. However, this view has been questioned.¹⁵

| Transfer | Monomer | T (°C) ^a | <i>C</i> ₁ | <i>C</i> ₂ | <i>C</i> ₃ | C_4 | C5-4 | Refs. |
|---------------------|---------|---------------------|-----------------------|-----------------------|-----------------------|----------------|----------|-------|
| Agent | | | | | | | | |
| $C_2H_5SH^b$ | MA | 50 | $0.94 \pm .07$ | $1.65 \pm .12$ | 1.57±.09 | $1.52 \pm .06$ | 1.57±.18 | 16 |
| $i-C_3H_7SH^{b}$ | MA | 50 | $0.54 \pm .08$ | $0.67 \pm .07$ | $0.70 \pm .08$ | $0.66 \pm .08$ | - | 17 |
| $C_2H_5SH^b$ | S | 50 | 7.1±.3 | 30±10 | - | - | 17±1 | 18 |
| CCl ₃ Br | S | 30 | $0.52 \pm .14$ | 9.4±4.6 | 37±3 | 96±12 | 460±61 | 19 |
| CCl ₄ | S | 76 | 0.0006 | 0.0025 | 0.0069 | 0.0115 | - | 20 |
| CCl ₄ | VAc | 60 | - | 0.13 | 0.47 | 0.67 | 0.80 | 21 |
| CCL | VC | 60 | 0.00284 | 0.0184 | 0.0280 | - | - | 22 |
| CHC13 | VC | 60 | 0.006 | 0.0141 | 0.0292 | - | - | 23 |
| D 11 1 | | e | | | | 1 (() | | |

Table 6.1 Chain Length Dependence of Transfer Constants (C_n)

a Bulk polymerization, medium comprises monomer – transfer agent. b The variation between C_2 , C_3 , C_4 and C_{5-6} is within experimental error.

6.2.1 Measurement of Transfer Constants

Various methods for estimating transfer constants in radical polymerization have been devised. The methods are applicable irrespective of whether the mechanism involves homolytic substitution or addition-fragmentation.

The most used method is based on application of the Mayo equation (eq. 5). For low (zero) conversion polymerizations carried out in the presence of added transfer agent T, it follows from eq. 5 that a plot of $1/\overline{X}_n vs [T]_0/[M]_0$ should yield a straight line with slope C_{tr} .¹² Thus, a typical experimental procedure involves evaluation of the degree of polymerization for low conversion polymerizations carried out in the presence of several concentrations of added transfer agent. The usual way of obtaining X_n values is by GPC analysis of the entire molecular weight distribution.

GPC-derived weight average molecular weights are often less prone to error than number average molecular weights. When termination is wholly by disproportionation or chain transfer and chains are long (>10 units), classical kinetics predicts $\overline{X}_n = \overline{X}_w/2$ (Section 5.2.1.3). It follows that C_{tr} can be obtained from the slope of a plot of $2/\overline{X}_w$ vs $[T]_0/[M]_0$.^{24,25} The errors introduced even when the dominant process for radical-radical termination is combination (*e.g.* S polymerization) are small as long as \overline{X}_n is small in relation to \overline{X}_{n0} .

It has been shown that equivalent information can be obtained by analysis of log(number chain length distribution) plots (the log CLD method).²⁴⁻²⁷ For the case where termination is wholly by disproportionation or chain transfer, it is possible to show that (eq. 6) applies:

$$\frac{d\ln(n_i)}{di} = \frac{d\ln[\phi^{i-1}(1-\phi)]}{di} = \ln(\phi)$$
(6)

For long chains ($X_n > 50$ for < 1% error)

$$\ln(\phi) \approx 1 - \frac{1}{\phi} = \frac{1}{(\overline{X}_n - 1)} \tag{7}$$

it is possible to write eq. 8 which is equivalent to the Mayo equation:

$$-\frac{\mathrm{dln}(n_{\rm i})}{\mathrm{d}i} \approx \frac{\left(1 + \frac{k_{\rm id}}{k_{\rm t}}\right) k_{\rm t}[\mathbf{P}^{\bullet}]}{k_{\rm p}[\mathrm{M}]} + \frac{k_{\rm tr,\mathrm{T}}}{k_{\rm p}} \frac{[\mathrm{T}]}{[\mathrm{M}]} + \frac{k_{\rm tr,\mathrm{I}}}{k_{\rm p}} \frac{[\mathrm{I}]}{[\mathrm{M}]} + \frac{k_{\rm tr,\mathrm{M}}}{k_{\rm p}} = \frac{1}{\overline{X}_{x0}} + \frac{k_{\rm tr,\mathrm{T}}}{k_{\rm p}} \frac{[\mathrm{T}]}{[\mathrm{M}]}$$
(8)
(9)

It follows that a plot of the slopes of the log CLD plots $vs [T]_0/[M]_0$ should yield a straight line with slope $-C_{tr}$.

In the more general case, where some termination is by combination it can be shown that for sufficiently large chain length (i):

$$\lim_{i \to \infty} \frac{\mathrm{dln}(n_i)}{\mathrm{d}i} = \mathrm{ln}(\phi) \tag{10}$$

While it is, in principle, desirable to take the limiting slope of the log CLD plot, in practice the limiting slopes are very susceptible to experimental noise and baseline choice issues. Moad and Moad²⁴ have shown that very little error is introduced by *systematically* taking the slope over the top 10% or the top 20% of the chain length distribution. The values for the slopes will overestimate $\ln(\phi)$. However, because the discrepancy is systematic, the "Mayo" analysis still provides a good estimate for C_{tr} (~6% error for the example in Figure 6.1).

The log CLD method can sometimes provide better quality data than the conventional Mayo method. It is less sensitive to experimental noise and has application in measuring the transfer constant to polymeric species where the distributions of the transfer agent and the polymer product partially overlap.²⁴

Problems arise with any of the abovementioned methods in the measurement of transfer constants for very active transfer agents. Bamford²⁸ proposed the technique of moderated copolymerization. In these experiments, the monomer of interest is copolymerized with an excess of a moderating monomer that has a much lower (preferably negligible) transfer constant. The method has also been applied to evaluate penultimate unit effects on the transfer constant.²⁸⁻³⁰



Figure 6.1 "Mayo plots" in which the calculated limiting slopes (triangles, —, C_{tr} (app) =0.184), "last 10% slopes" (circles, ---, C_{tr} (app)=0.180) and "top 20% slopes" (squares, ----, C_{tr} (app)=0.169) are graphed as a function of [T]/[M]. Data are for system with \overline{X}_n =5155, $k_{tc}/(k_{tc} + k_{td})$ = 1.0 and C_{tr} =0.184.²⁴ C_{tr} (app) is the apparent C_{tr} from the slope of the "Mayo plot".

Another classical method for evaluating transfer constants involves evaluation of the usage of transfer agent (or better the incorporation of transfer agent fragments into the polymer) and the monomer conversion:³¹

$$\frac{d[\mathbf{T}]}{d[\mathbf{M}]} = \frac{k_{\mathrm{trT}}[\mathbf{P}^{\bullet}][\mathbf{T}] + k_{\mathrm{iT}}[\mathbf{T}^{\bullet}][\mathbf{M}]}{k_{\mathrm{p}}[\mathbf{P}^{\bullet}][\mathbf{M}]}$$
(11)

For long chains, consumption of the monomer in the reinitiation step can be neglected and eq. 11 simplifies to eq. 12:

$$\frac{d[\mathrm{T}]}{d[\mathrm{M}]} = \frac{k_{\mathrm{tr}}[\mathrm{T}]}{k_{\mathrm{p}}[\mathrm{M}]} = C_{\mathrm{tr}} \frac{[\mathrm{T}]}{[\mathrm{M}]}$$
(12)

from which eq. 13 follows:

$$\frac{\mathrm{d}\ln[\mathrm{T}]}{\mathrm{d}\ln[\mathrm{M}]} = C_{\mathrm{tr}} \tag{13}$$

Thus, the slope of a plot of $\ln[T] vs \ln[M]$ will yield the transfer constant. This method does not rely on molecular weight measurements.

For the situation where short chains cannot be ignored eq. 11 can be transformed to eq. 14:

$$\frac{d[M]}{d[T]} = \frac{[M]}{C_{\rm tr}[T]} + 1$$
(14)

A number of authors have provided integrated forms of the Mayo equation³²⁻³⁶ which have application when the conversion of monomer to polymer is non-zero. Integration of eq. 12 provides eq. 15:

$$\frac{[\mathrm{T}]}{[\mathrm{T}]_{\mathrm{o}}} = \left(\frac{[\mathrm{M}]}{[\mathrm{M}]_{\mathrm{o}}}\right)^{C_{w}}$$
(15)

This enables substitution for [T] in eq. 16 to give eq. 17:^{34,36}

$$\frac{1}{\overline{X}_{n}} = \frac{1}{\overline{X}_{n0}} + \frac{[T] - [T]_{0}}{[M] - [M]_{0}}$$
(16)

$$\frac{1}{\overline{X}_{n}} = \frac{1}{\overline{X}_{n0}} + \frac{\left[T\right]_{0} \left[1 - \left(\begin{bmatrix}\mathbf{M}\\ [\mathbf{M}]_{0}\right)^{c_{0}}\right]}{\left[\mathbf{M}\right]_{0} \left(1 - \left(\frac{[\mathbf{M}]}{[\mathbf{M}]_{0}}\right)\right)}$$
(17)

Rearrangement and substitution of 1-x for [M]/[M]₀ provides eq. 18:

$$\ln\left(1 - \frac{[M]_{o}x}{[T]_{o}}\left(\frac{1}{\bar{X}_{u}} - \frac{1}{\bar{X}_{u^{0}}}\right)\right) = C_{u}\ln(1-x)$$
(18)

where x is the fractional conversion of monomer into polymer. Thus, a plot of

$$\ln\left(1 - \frac{[M]_o x}{[T]_o} \left(\frac{1}{X_n} - \frac{1}{X_{n0}}\right)\right) vs \ln(1-x)$$

. .

should provide a straight line passing through the origin with slope $C_{\rm tr}$. Bamford and Basahel²⁸⁻³⁰ have reported the derivation of a similar equation for copolymerization. This method is highly dependent on the precision of the conversion measurements since errors in conversions are magnified in $C_{\rm tr}$.

Cardenas and O'Driscoll³² and Stickler³³ have shown that, provided that the consumption of transfer agent is negligible with respect to monomer, a plot of

$$\frac{1}{\overline{X}_n} vs - \frac{[T]_o}{[M]_o} \frac{\ln(1-x)}{x}$$

should also yield a straight line with slope C_{tr} .^{32,33} Nair *et al.*³⁷ have proposed a modified Mayo equation for use when retardation through primary radical termination with transfer agent-derived radicals is significant.

Chain transfer is kinetically equivalent to copolymerization. The Q-e and 'Patterns of Reactivity' schemes used to predict reactivity ratios in copolymerization (Section 7.3.4) can also be used to predict reactivities (chain transfer constants) in chain transfer and the same limitations apply. Tabulations of the appropriate parameters can be found in the *Polymer Handbook*.^{38,39}

6.2.1.1 Addition-fragmentation

Some transfer agents react by addition-fragmentation (Section 6.2.3) or abstraction-fragmentation mechanisms. Both of these processes involve the formation of a short-lived intermediate. The reaction scheme for addition-fragmentation can be summarized schematically as follows (Scheme 6.3).

transfer addition $P_n \bullet + T \rightarrow [P_n T \bullet]$ $R_{add} = k_{add} [P \bullet][T]$ fragmentation $[P_n T \bullet] \rightarrow P_n + T \bullet \qquad R_{\beta} = k_{\beta} [P T \bullet]$ reinitiation $T \bullet + M \rightarrow P_1 \bullet \qquad R_{iT} = k_{iT} [T \bullet][M]; k_{iT} \ge k_p$

Scheme 6.3

The reactivity of the transfer agent (T) towards the propagating species and the properties of the adduct (P_nT •) are both important in determining the effectiveness of the transfer agent: if the lifetime of the intermediate (P_nT •) is significant, it may react by other pathways than β -scission; if it (P_iT •) undergoes coupling or disproportionation with another radical species the rate of polymerization will be retarded; if it adds to monomer (T copolymerizes) it will be an inefficient transfer agent.

If both addition and fragmentation are irreversible the kinetics differ little from conventional chain transfer. In the more general case, the rate constant for chain transfer is defined in terms of the rate constant for addition (k_{add}) and a partition coefficient which defines how the adduct is partitioned between products and starting materials (eq. 19).

$$k_{\rm tr} = k_{\rm add} \, \frac{k_{\beta}}{k_{\rm -add} + k_{\beta}} \tag{19}$$

Methods used for evaluating transfer constants are the same as for conventional chain transfer.

6.2.1.2 Reversible chain transfer

In some cases the product of chain transfer (P_n^T) is itself a transfer agent and chain transfer is reversible. Examples include alkyl iodides (Scheme 6.4) and certain addition-fragmentation transfer agents (*e.g.* macromonomers and thiocarbonylthio compounds) (Scheme 6.5).

For very active transfer agents, the transfer agent-derived radical (T•) may partition between adding to monomer and reacting with the polymeric transfer agent (P_n^{T}) even at low conversions. The transfer constant measured according to the Mayo or related methods will appear to be dependent on the transfer agent concentration (and on the monomer conversion).⁴⁰⁻⁴² A reverse transfer constant can be defined as follows (eq. 20):

$$C_{-\rm tr} = \frac{k_{-\rm tr}}{k_{\rm iT}} \tag{20}$$

and the rate of transfer agent consumption is then given by eq. 21:

This equation can be solved numerically to give values of C_{tr} and C_{tr} .^{40,41} For reversible addition-fragmentation chain transfer (RAFT) (Scheme 6.5), the rate constant for the reverse reaction is defined as shown in eq. 22:

$$k_{-tr} = k_{-\beta} \frac{k_{-add}}{k_{-add} + k_{\beta}}$$
(22)

Systems that give reversible chain transfer can display the characteristics of living polymerization. Such systems are discussed in Section 9.5.

6.2.2 Homolytic Substitution Chain Transfer Agents

Chain transfer most commonly involves transfer of an atom or group from the transfer agent to the propagating radical by a homolytic substitution (SH^2) mechanism. The general factors influencing the rate and specificity of these reactions have been dealt with in Section 2.4. Rate constants are determined by a combination of bond strength, steric and polar factors. Transfer agents that react by addition-fragmentation are dealt with in Section 6.2.3. Organometallic species that give catalytic chain transfer are discussed in Section 6.2.5.

The moiety transferred will most often be a hydrogen atom, for example, when the transfer agent is a thiol (*e.g. n*-butanethiol - Scheme 6.6, Section 6.2.2.1), a hydroperoxide (Section 3.3.2.5), the solvent (6.2.2.5), *etc.*

$$P_n + H - S(CH_2)_3 CH_3 \longrightarrow P_n - H + S(CH_2)_3 CH_3$$

Scheme 6.6

It is also possible to transfer a heteroatom (e.g. a halogen atom from bromotrichloromethane - Scheme 6.7, Section 6.2.2.4),

$$P_n^{\bullet} + Br - CCl_3 \longrightarrow P_n - Br + \cdot CCl_3$$

Scheme 6.7

or a group of atoms (e.g. from diphenyl disulfide - Scheme 6.8, Section 6.2.2.2).

$$P_n^* + PhS - SPh \longrightarrow P_n - SPh + SPh$$

Scheme 6.8

Group transfer processes are of particular importance in the production of telechelic or di-end functional polymers.

The following sections detail the chemistry undergone by specific transfer agents that react by atom or group transfer by a homolytic substitution mechanism. Thiols, disulfides, and sulfides are covered in Sections 6.2.2.1, 6.2.2.2 and 6.2.2.3 respectively, halocarbons in Section 6.2.2.4, and solvents and other agents in Section 6.2.2.5. The transfer constant data provided have not been critically

assessed or evaluated but are included to show the order of magnitude of these values and to provide a guide to the relative reactivity of the various reagents.

6.2.2.1 Thiols

Traditionally thiols or mercaptans are perhaps the most commonly used transfer agents in radical polymerization. They undergo facile reaction with propagating (and other) radicals with transfer of a hydrogen atom and form a saturated chain end and a thiyl radical (Scheme 6.6). Some typical transfer constants are presented in Table 6.2. The values of the transfer constants depend markedly on the particular monomer and can depend on reaction conditions.^{43,44}

Table 6.2 Transfer Constants (60 °C, bulk) for Thiols (RSH) with VariousMonomers^a

| Transfer agent | | | C_{u} | | |
|--|----------------------|---------------------|-------------------|------------------|--------------------|
| R | ΜΜΛ | ΜΛ | ΛN | S | VΛc |
| Н | - | - | 0.30 ^b | 5° | - |
| <i>п</i> -С ₄ Н ₉ - | 0.67^{31} | $1.7^{d.31}$ | - | 22 ³¹ | 48 ^{e.31} |
| <i>n</i> -C ₁₂ H ₂₅ - | 0.7^{45} | 1.5 ^{g.45} | 0.73 ^h | 16 ⁴⁶ | - |
| HO-CH ₂ CH ₂ - (3) | 0.62 | - | - | - | - |
| HOC(=O)CH ₂ CH ₂ - | 0.38 ^f | - | - | 9.4 | - |
| CH ₅ OC(=O)CH ₂ - | $0.30^{ m h.47}$ | $0.64^{g,h,47}$ | - | $1.4^{h.47}$ | $0.07^{h.47}$ |
| H ₃ N ⁺ -CH ₂ CH ₂ - | 0.11 ^{i,43} | | - | $11^{-4.3}$ | - |
| Ph- | 2.7^{48} | - | - | 0.08 | - |

a Numbers are taken from the Polymer Handbook⁴⁹ unless otherwise stated and have been rounded to two significant figures. b 50 °C. c At 70 °C. d In ethyl acetate solvent. e Substantial retardation observed.³⁰ f Extrapolated to 60 °C from the data given. The activation energies quoted⁵¹ appear to be calculated incorrectly. g BA. h In benzene solvent. i The corresponding free amine is reported to have a very low transfer constant in MMA polymerization.⁴³ It may be consumed in a Michael reaction with monomer.

Thiols react more rapidly with nucleophilic radicals than with electrophilic radicals. They have very large $C_{\rm tr}$ with S and VAc, but near ideal transfer constants ($C_{\rm tr} \sim 1.0$) with acrylic monomers (Table 6.2). Aromatic thiols have higher $C_{\rm tr}$ than aliphatic thiols but also give more retardation. This is a consequence of the poor reinitiation efficiency shown by the phenylthiyl radical. The substitution pattern of the alkanethiol appears to have only a small (<2-fold) effect on the transfer constant. Studies on the reactions of small alkyl radicals with thiols indicate that the rate of the transfer reaction is accelerated in polar solvents and, in particular, water.⁵² Similar trends are observed for transfer to 1 in S polymerization with $C_{\rm tr} = 1.4$ in benzene 3.6 in CH₃CN and 6.1 in 5% aqueous CH₃CN.⁴⁴ In copolymerizations, the thiyl radicals react preferentially with electron-rich monomers (Section 3.4.3.2).

Bamford and Basahel⁵³ have investigated the importance of penultimate unit effects on the reactivity of *n*-butanethiol in a number of copolymerizations (S-MMA, S-MA) using the technique of "moderated copolymerization". Their data indicate that penultimate unit effects are unimportant in these systems. More recently, de la Fuente and Madruga⁴⁵ have come to similar conclusions for the reactivity of dodecanethiol in BA-MMA copolymerization. This contrasts with findings for transfer to carbon tetrabromide (Section 6.2.2.4). It has also been found, again in contrast with halocarbons, that C_{tr} for various primary and secondary thiols is essentially independent of chain length for chain lengths ≥ 2 (Table 6.1).

A range of functional thiols [*e.g.* thioglycolic acid (2) and mercaptoethanol (3)] has been used to produce monofunctional polymers^{10,54-56} (Section 7.5.2) and thence as precursors for diblock copolymers.⁴⁷

$$\begin{array}{cccc} & & & & & \\ CH_3O-\overset{\circ}{C}-CH_2-SH & & HO-\overset{\circ}{C}-CH_2-SH & & HO-CH_2-CH_2-SH \\ 1 & & & & & & \\ \end{array}$$

6.2.2.2 Disulfides

A wide range of dialkyl⁵⁷ and diaryl disulfides,^{58,59} diaroyl disulfides,⁶⁰ and xanthogens⁶¹ has been used as transfer agents (Scheme 6.8). Their use ideally leads to the incorporation of functionality at both ends of the polymer chain, thus they find application in the synthesis of telechelics (Section 7.5.2).

The C-S bond of the sulfide end groups can be relatively weak and susceptible to thermal and photo- or radical-induced homolysis. This means that certain disulfides [for example 7-9] may act as iniferters in living radical polymerization and they can be used as precursors to block copolymers (Sections 7.5.1 and 9.3.2).



Aliphatic disulfides 4 are not particularly reactive in chain transfer towards MMA and S (Table 6.3). However, they appear to be ideal transfer agents ($C_{tr} \sim 1.0$) for VAc polymerizations.

The reactivity of diphenyl (6, X=H) and dibenzoyl (7, X=H) disulfide derivatives is higher than aliphatic derivatives. The value of C_{tr} depends markedly on the substituents, X, and on the pattern of substitution. Electron withdrawing substituents (*e.g.* X = *p*-CN or *p*-NO₂) may increase C_{tr} by an order of magnitude.^{59,60} However, these compounds also give marked retardation.

Compounds with a thiocarbonyl α to the S-S bond such as the dithiuram (e.g. **8**)^{62,63} and xanthogen disulfides (e.g. **9**)⁶⁴ have transfer constants that are much higher than other disulfides. In part, this may be due to the availability of another mechanism for induced decomposition (Scheme 6.9) involving addition to the C⁻S double bond and subsequent fragmentation. Thiocarbonyl double bonds are very reactive towards addition and an addition-fragmentation mechanism has been demonstrated for related compounds (Section 6.2.3.5).

Table 6.3 Transfer Constants for Disulfides (R-S-S-R) With Various Monomers^a

| Transfer agent | | $C_{\rm tr}$ | | | |
|---|---------------------|----------------|---------------------|--------|--|
| R | MMA | MÄ | S | VAc | |
| C ₂ H ₅ - (4 , n=1) | 0.00013 | - | - | - | |
| $n-C_4H_9-(4, n=3)$ | - | - | 0.0024^{57} | 1.0 | |
| PHCH ₂ - (5) | 0.0063 | - | 0.01 | - | |
| EIOC(C=O)CH ₂ - | 0.00065 | - | 0.015 | 1.5 | |
| Ph-(6, X=H) | 0.0085^{59} | - | 0.15 | - | |
| PhC(=O)- ^b (7, X=H) | 0.0010^{60} | - | 0.0036^{60} | - | |
| $p-CNC_6H_4C(=O)^{-b}(7, X=CN)$ | 0.029^{60} | - | 0.32^{60} | - | |
| $(CH_3)_2 NC(=S)^{-c}$ (8) | 0.53^{62} | - | 0.57^{63} | - | |
| $CH_{\circ}OC(=S)^{-d}(9)$ | 1.1 ^{64.d} | $4.9^{64,d,e}$ | 3.1 ^{64,d} | 64.d.f | |

a 60 °C, bulk unless indicated otherwise. Numbers are taken from the Polymer Handbook⁴⁹ unless otherwise stated, and have been rounded to two significant figures. Where a choice of numbers is available the average value has usually been quoted. b These numbers are reported incorrectly in the Polymer Handbook and many other compilations. c 80 °C, d in benzene. e BA. f inhibition.



Scheme 6.9

6.2.2.3 Monosulfides

Most monosulfides generally have very low transfer constants. Exceptions to this rule are allyl sulfides (Section 6.2.3.2) and thiocarbonylthic compounds such as the trithiocarbonates and dithioesters (Section 9.5.3) that react by an addition-fragmentation mechanism.

t-Butanesulfide (10) has a substantially higher transfer constant than other saturated monosulfides ($C_{tr} = 0.025$ in S polymerization,⁵⁷ *n*-butane sulfide has $C_{tr} = 0.0022$). This result appears counterintuitive if the reaction involves homolytic substitution on sulfur. Pryor and Pickering⁵⁷ proposed that this compound may react by hydrogen atom transfer and fragmentation as shown in Scheme 6.10.

$$P_{n} \xrightarrow{CH_{3} CH_{3}}_{CH_{2}-C-S-C-CH_{3}} \xrightarrow{CH_{3} CH_{3}}_{CH_{3} CH_{3}} \xrightarrow{CH_{3} CH_{3}}_{CH_{3} CH_{3}}$$

Scheme 6.10

6.2.2.4 Halocarbons

Halocarbons including carbon tetrachloride, chloroform, bromotrichloromethane⁶⁵ (Scheme 6.7) and carbon tetrabromide have been widely used for the production of telomers and transfer to these compounds has been the subject of a large number of investigations.¹¹ Representative data are shown in Table 6.4. Telomerization involving halocarbons has also been developed as a means of studying the kinetics and mechanism of radical additions.⁶⁶

 Table 6.4 Transfer Constants (60 °C, bulk) for Halocarbons with Various Monomers^a

| | | | $C_{\rm m} \times 10^4$ | | |
|------------------|------|------|-------------------------|------|---------------------|
| | MMA | BA | AN | S | VAc |
| CBr ₄ | 2700 | - | 500 | 2200 | 7.4×10 ⁶ |
| CCI4 | 2.4 | 3.2 | 0.85 | 130 | 9600 ⁵⁰ |
| CHCl3 | 1.8 | 0.89 | 5.7 | 0.5 | 150^{-50} |

a Numbers are taken from the Polymer Handbook⁴⁹ unless otherwise stated, and have been rounded to two significant figures.

The perhalocarbons, CCl_4 and CBr_4 , react with carbon-centered radicals by halogen-atom transfer to form a perhaloalkyl radical. Halogen atom abstractability decreases in the series iodine>bromine>chlorine. Halohydrocarbons may in principle react by hydrogen-atom, halogen-atom transfer or both. The preferred pathway can often be predicted by considering the relative C-X bond strengths (Section 2.4). For CHCl₃, transfer of a hydrogen atom is favored.

The halocarbons react more rapidly with nucleophilic radicals than with electrophilic radicals. Thus, values of C_{tr} with S and VAc are substantially higher than those with acrylic monomers (Table 6.4) where the transfer constant is close to ideal (C_{tr} =1.0). The haloalkyl radicals formed have electrophilic character (Section 2.3.2).

Bamford¹⁴ demonstrated that C_{tr} for transfer to carbon tetrabromide in copolymerization is subject to penultimate unit effects. He found $C_{S,S}$ =368, $C_{MA,S}$ =302, $C_{MMA,S}$ =60 (compare behavior observed with thiols - Section 6.2.2.1). The finding ($C_{MA,S}$ ~ $C_{S,S}$ >> $C_{MMA,S}$) suggested that steric factors were more important than either polar or electronic factors in determining the magnitude of the remote unit effect on C_{tr} . Bamford.¹⁴ proposed that k_{tr} is more sensitive to remote unit effects than k_p . The S/MMA/CBr₄ system has recently been reexamined by Harrisson *et al.*¹⁵ They also found penultimate unit effects to be important in the S/MMA/CCl₄ system. Further evidence for remote unit effects is that C_{tr} in MA and S polymerizations is chain length dependent for chain lengths ≤ 3 units (Table 6.1). A variation in C_{tr} with chain length for ethylene polymerization has been attributed to polar effects. The electron donating ability of the alkyl chain increases in the series: ethyl
butyl<hexyl.¹¹

Ameduri and Boutevin⁶⁷ showed that certain transition metal salts and complexes effectively catalyze transfer to the halocarbons. In these cases, initiation/reinitiation involves a redox reaction between the metal and the halocarbon. A transition metal in its oxidized form then reacts with the propagating radical by group transfer to regenerate the metal in its original oxidation state. Transition metal species that are effective in this context, include copper salts and RuCl₂(PPh₃). Effective transfer constants are substantially higher than when the transfer agent is used alone. Narrow polydispersities were not obtained. Nonetheless, these experiments can be considered to mark the beginnings of ATRP (Section 9.4).

Certain alkyl iodides give reversible chain transfer with S and some fluoroolefins (Section 9.5.4). In these cases, the polymerization can show some living characteristics.

6.2.2.5 Solvents and other reagents

Many solvents and additives have measurable transfer constants (Table 6.5). The accuracy of much of the transfer constant data in the literature is questionable with values for a given system often spanning an order of magnitude. In some cases the discrepancies may be real and reflect differences in experimental conditions. In other cases they are less clear and may be due to difficulties in molecular weight measurements or other problems.

Nonetheless, it is clear that the reactivity of solvents in transfer reactions depends on the nature of the propagating species and some general conclusions can be drawn. The propagating species derived from MMA has relatively little tendency to undertake transfer. That derived from VAc appears extremely reactive towards solvents and other transfer agents (note, however that many reagents give marked retardation with VAc⁵⁰). The factors influencing reactivity in hydrogen atom abstraction reactions are discussed in general terms in Section 2.4.

| Solvent | | | $C_{\rm tr} \times 10^4$ | | |
|---------------|------|------------------|--------------------------|------|-----|
| | MMA | MA | AN | S | VAc |
| benzene | 0.04 | 0.3 ^b | 2.5 | 0.02 | 3.0 |
| toluene | 0.20 | 2.7 | 5.8 | 0.12 | 21 |
| acctone | 0.20 | 0.23 | 1.1 | 0.32 | 12 |
| butan-2-onc | 0.45 | 3.2 ^b | 6.4 | 5.0 | 74 |
| ethyl acetate | 0.15 | - | 2.5 | 5.7 | 3.0 |
| triethylamine | 8.3 | 400 | 790 | 7.1 | 370 |

 Table 6.5 Transfer Constants (60 °C, bulk) for Selected Solvents and Additives with Various Monomers^a

a Numbers have been selected from the Polymer Handbook⁴⁹ or references given therein and have been rounded to two significant figures. $b \ 80 \ ^{\circ}C$.

Mechanisms for chain transfer depend on the particular solvent or reagent. Many solvents have abstractable hydrogens (*e.g.* acetone, butanone, toluene) and may react by loss of those hydrogens (Scheme 6.11).



Scheme 6.11

Benzene may react by addition as shown in Scheme 6.12 (this pathway is also open to other aromatic solvents). The cyclohexadienyl radical is a poor initiating species and may terminate a second chain by hydrogen atom transfer. According to this process, benzene is a retarder rather than a transfer agent.



Scheme 6.12

In the case of S, it has been proposed that reinitiation may occur by hydrogenatom transfer to monomer (Scheme 6.13).^{12,68}



Scheme 6.13

6.2.3 Addition-Fragmentation Chain Transfer Agents

296

Addition-fragmentation chain transfer has been reviewed by Rizzardo *et al*,⁶⁹ Colombani and Chaumont,⁷⁰ Colombani,⁷¹ Yagci and Reetz,⁷² and Chiefari and Rizzardo.³ Certain unsaturated compounds may act as transfer agents by a twostep addition-fragmentation mechanism. All of the compounds discussed in this section have the general structure **11** or **12** where C=X is a reactive double bond (X is most often carbon or sulfur) Z is a group chosen to give the transfer agent an appropriate reactivity with respect to the monomer(s), A is typically CH₂, O or S, B is typically O and R is a radical leaving group. Chain transfer to monomer in VC polymerization (Section 6.2.6.3) and transfer to benzene (6.2.2.5) can also be considered as examples of addition-fragmentation chain transfer.



Radical addition-fragmentation processes have been exploited in synthetic organic chemistry since the early 1970's.⁷³⁻⁷⁵ Allyl transfer reactions with allyl stannanes and the Barton-McCombie deoxygenation process with xanthates are two examples of reactions known to involve a $S_{11}2'$ mechanism. However, the first reports of addition-fragmentation transfer agents in polymerization appeared in the late 1980's.⁷⁶⁻⁷⁸ Mechanisms for addition-fragmentation chain transfer are shown in Scheme 6.14 and Scheme 6.15. Since functionality can be introduced to the products 14 or 16 in either or both the transfer (from Z, X, A, or B) and reinitiation (from R) steps, these reagents offer a route to a variety of end-functional polymers including telechelics.

Chain transfer



Scheme 6.14



Scheme 6.15

Rates of addition to transfer agents 11, 12 are determined by the same factors that determine rates of addition to monomers (Section 2.3). Substituents on the remote terminus of a double bond typically have only a minor influence. Thus, in most cases, the double bonds of the transfer agents have a reactivity towards propagating radicals that is comparable with that of the common monomers they resemble. With efficient fragmentation, transfer constants can be close to unity. The radicals formed by addition typically have low reactivity towards further propagation and other intermolecular reactions because of steric crowding about the radical center.

Efficient transfer requires that radicals formed by addition undergo facile β -scission (for 13) or rearrangement (for 15) to form a new radical that can reinitiate polymerization. The driving force for fragmentation of the intermediate radical is provided by cleavage of a weak A-R bond and/or formation of a strong C-X bond (for 11). If fragmentation leads preferentially back to starting materials the transfer constant will be low. If the overall rate of β -scission is slow relative to propagation then retardation may result. The adducts (13 and 15) then have the potential to undergo side reactions by addition (*e.g.* copolymerization of the transfer agent) or radical-radical termination. Retardation is an issue particularly for high k_p monomers such as VAc and MA. In designing transfer agents and choosing an R group (see 11, 12), a balance must be achieved between the leaving group ability of R and reinitiation efficiency by R*.

When the product of the reaction is itself a potential transfer agent or macromonomer (11, X=A=CH₂, X=A=S) block, graft or hyperbranched copolymer formation may be an issue particularly at high conversions.^{76,79} The design of transfer agents that give reversible addition-fragmentation chain transfer (RAFT) has provided one of the more successful approaches to living radical polymerization (Sections 9.5.2 and 9.5.3). The pathway can be blocked by choice of A (see 11). For example, when A is oxygen (vinyl ethers, Section 6.2.3.1) or bcars an alkyl substituent (*e.g.* A=CII-CH₃), the product is unreactive to radical addition.

If R and Z, A or X are connected to form a ring structure the result is a potential ring opening monomer. For many of the transfer agents in this section there are analogous ring-opening monomers described in Section 4.4.2.
The Chemistry of Radical Polymerization

6.2.3.1 Vinyl ethers

The vinyl ethers (11 X=CH₂, A=O) can be very effective chain transfer agents.^{78,80-82} The mechanism for chain transfer is shown in Scheme 6.16 for the case of α -benzyloxystyrene (17). A large part of the driving force for fragmentation is provided by formation of a strong carbonyl double bond. It is also important that R is a good radical leaving group.^{81,83} The ketene acetal 19⁸³ gives both copolymerization and chain transfer in S polymerization whereas with 20,⁸³ and 17⁷⁸ and 21-23⁸⁰ chain transfer is the only reaction detected. Transfer constants for some vinyl ether transfer agents are provided in Table 6.6. Those with a benzyl radical leaving group are designed for use in S or (meth)acrylate ester polymerization and give retardation in VAc polymerization. The polymers formed have a ketone end group (*e.g.* 18, Scheme 6.16). Additional functionality can be introduced on Z or R (refer 11) to modify reactivity or to tailor the end groups as in the examples (24-26).⁸²



Scheme 6.16



| Transfer agent | | C_{μ} for monomer ^b | | | References |
|----------------|-------|------------------------------------|-------------------|------------------|------------|
| | S | MMA | MA | VAc | 70.00 |
| 17 | 0.26 | 0.76 | 5.7° | 9.7 ^c | 78,80 |
| 21 | 0.036 | 0.081 | 0.3° | 12 ^c | 80 |
| 22 | 0.046 | 0.16 | 0.54 ^e | 20 ^e | 80 |
| 23 | 0.2 | 0.5 | 1.1° | - | 80 |

Table 6.6 Transfer Constants for Vinyl Ethers at 60 °C^a

a Bulk, medium comprises only monomer and transfer agent. b Transfer constants rounded to two significant figures. c Significant retardation observed.

The vinyl ether transfer agents, like other vinyl ethers, can show marked acid sensitivity. They are not suited for use with acid monomers. Even traces of acidic impurities in the monomer or the polymerization medium can catalyze decomposition of the transfer agent.



6.2.3.2 Allyl sulfides, sulfonates, halides, phosphonates, silanes

With allyl transfer agents (e.g. 11 X=CH₂, A=CH₂) such as allyl halides,⁸⁴⁻⁹⁰ sulfides,^{77,91,92} sulfones,⁸⁴ sulfonates,^{84,93} silanes⁸⁴ phosphonates⁸⁴ and similar compounds,⁸⁴ the main driving force is the weak single bond (A-R) of 11. A similar situation pertains with the corresponding dienyl transfer agents *e.g.* 11 X–CH₂, A is CH₂–CH₂-CH₂-.^{94,95} The proposed mechanism of chain transfer is shown in Scheme 6.17 for the case of the allyl sulfide 27. The product will be predominantly a macromonomer (28) that may be reactive under the polymerization conditions particularly at high conversion (Section 6.2.3.4).

Some typical transfer constants for allyl sulfides are given in Table 6.7. The values of $C_{\rm tr}$ for these reagents are less dependent on the particular monomer than those for halocarbons (Table 6.2) or thiol transfer agents (Table 6.4). The low transfer constant of **32** demonstrates the importance of the activating group Z (*cf.* **11**).







| Transfer agent | | $C_{\rm tr}$ for monomer ^b | | | References | |
|---|-------|---------------------------------------|---------------|------|-------------------|----------|
| | S | ΜМΛ | ΜΛ | ΜΛΝ | VΛc | |
| 29, R=Ph | 0.80 | 1.2 | 4.0° | - | $\sim 20^{\circ}$ | 77 |
| 29, R–CN | 1.8 | 1.4 | 1.6° | - | ~60 ^c | 77 |
| 29, R=CO ₂ Et | 0.95 | 0.74 | 2.2° | 0.42 | ~2 7 ° | 77.96.97 |
| 30, R=CH ₂ CO ₂ H | 0.95 | 1.1 | - | - | - | 91 |
| 30, R=CH ₂ CH ₂ NH ₂ | 0.79 | 0.91 | - | - | - | 91 |
| 30, R=CH ₂ CH ₂ OH | 0.77 | 1.2 | - | - | - | 91 |
| 31 | 1.27 | 0.74 | - | - | - | 91 |
| 32, R=CH₂CH₂CO₂CH₃^d | 0.016 | - | - | - | - | 35 |
| 33 | 0.35 | 1.11 | - | - | - | 95 |
| 34 | 1.51c | 0.33 ^c | - | - | - | 94 |
| | | | | | | · · |

Table 6.7 Transfer Constants for Allyl Sulfides at 60 °C^a

a Bulk, medium comprises only monomer and transfer agent. b Transfer constants rounded to two significant figures. c Significant retardation observed. d Transfer constants similar for various R.

Allyl sulfonates (35, 36) show analogous behavior. Transfer constants are reported in Table 6.8. Other compounds with weak A-R bonds (*cf.* 11) that have the capacity to act as transfer agents are listed in Table 6.9. Allyl bromides 43a, 44, and 45a give predominantly chain transfer whereas, the chlorides (*e.g.* 45b)

give copolymerization as well as chain transfer. 84,98 The silane **48** is also able to react as a comonomer. 84 Compounds **11** with R=oxygen are not transfer agents but are comonomers.















| Transfer agent | | $C_{\rm tr}$ for n | References | | |
|----------------|------------------|--------------------|-------------------|------|----|
| | S | MMA | BA | VAc | |
| 35 | 4.2 ^d | 0.72^{d} | 1.1 ^d | _c | 93 |
| 36 | 5.8 | 1.1 | 2.3 ^d | _e | 84 |
| 37 | - | 1.0 | - | - | 84 |
| 38 | 0.02 | 0.065 | 0.20 ^c | 2.8 | 84 |
| 39 | - | - | - | 3.9 | 3 |
| 40 | - | - | - | 0.05 | 3 |
| 41 | - | 3.0 | - | - | 99 |
| 42 | - | 1.9 | - | - | 84 |

Table 6.8 Transfer Constants for Allyl Sulfonates and Sulfoxides at 60 °C^a

a Bulk, medium comprises only monomer and transfer agent. b Transfer constants rounded to two significant figures. c Significant retardation observed. d 3.46 M monomer in benzene solution⁹³. e MA.



| Transfer agent | | References | | | |
|---------------------|------|---------------------|--------------------|-----|-----|
| | S | ММА | ΜΛ | VAc | |
| 43a , Hal=Br | | 1.5 | 2.3 | - | 84 |
| 44 | 2.9 | 2.3 | 5.3 | - | 84 |
| 45a , Hal=Br | - | 2.2 | 3.0 | - | 84 |
| 45b, Hal=Cl | - | 0.0075 ^d | 0.046 ^d | - | 84 |
| 46 | - | 0.4 | - | - | 84 |
| 47 | - | 3.0 | - | - | 84 |
| 48 | - | 0.08 ^d | - | - | 84 |
| 49 | - | 3.4 | - | _ | 99 |
| 50 | 8.1 | 7.4 ^c | - | - | 99 |
| 51 | 0.25 | - | - | - | 100 |

 Table 6.9 Transfer Constants for Allyl Halides, Phosphonates, Silanes and Stannanes at 60 °C^a

a Bulk, medium comprises only monomer and transfer agent. b Transfer constants rounded to two significant figures. c Significant retardation observed. d Copolymerization observed.

6.2.3.3 Allyl peroxides

In the case of allyl peroxides (12 X= CH₂, A=CH₂, B=O),¹⁰¹⁻¹⁰⁵ intramolecular homolytic substitution on the O-O bond gives an epoxy end group as shown in Scheme 6.18 (1,3-S₁₁*i* mechanism). The peroxides **52-59** are thermally stable under the conditions used to determine their chain transfer activity (Table 6.10). The transfer constants are more than two orders of magnitude higher than those for dialkyl peroxides such as di-*t*-butyl peroxide (C_1 =0.00023-0.0013) or di-isopropyl peroxide (C_1 =0.0003) which are believed to give chain transfer by direct attack on the O-O bond.⁴⁹ This is circumstantial evidence in favor of the additionfragmentation mechanism.



Scheme 6.18



Table 6.10Transfer Constants for Allyl Peroxide and Related Transfer Agents at
 $60 \ ^{\circ}C^{\circ}$

| Transfer agent | C_{α} for monomer ^b | | | | References |
|----------------|---------------------------------------|------------|-------------|------------------|------------|
| | S | MMA | MA | VAc | |
| 52 | 0.9 | 0.8 | - | - | 82,101 |
| 53 | 1.6 | 0.6 | 1.0 | - | 82,101 |
| 54 | 2.0 | 0.9 | 0.7 | - | 82 |
| 55 | 0.8 | 0.8 | - | - | 82,101 |
| 57 | 0.92 | 0.49 | 1.9 | - | 106 |
| 58 | 0.9 | - | - | - | 106 |
| 59 | 0.22 | 0.012 | 0.08 | 3.7 | 100 |
| 61 | 0.82° | 0.31° | - | - | 107 |
| 60 | 0.35 ^d | 0.05^{d} | 0.46^{ad} | 1.3 ^d | 44,47 |
| 62 | 0.14 | 0.57 | 1.31° | - | 108 |

a Bulk, medium comprises only monomer and transfer agent. b Transfer constants rounded to two significant figures. c Compound is also an initiator under the polymerization conditions. Transfer constant obtained using a modified Mayo equation.¹⁰⁷ d In benzene. e BA.

Peroxyacetals 58^{106} and peresters such as 61^{107} are also effective transfer agents, however, at typical polymerization temperatures (~60 °C) they are thermally unstable and also act as initiators. Compounds such as 62 which may give addition and 1,5-intramolecular substitution with fragmentation have also been examined for their potential as chain transfer agents (1,5-S_H*i* mechanism).¹⁰⁸



6.2.3.4 Macromonomers

The chain transfer agents $(11 \text{ X}=\text{CH}_2, \text{ A}=\text{CH}_2)$ are misnamed 'macromonomers' since in this context they do not behave as macromonomers. Copolymerization when it occurs is a side reaction. The mechanism is shown in Scheme 6.19 for MAA 'trimer' (63). The final product (65) is a also a 'macromonomer' and formation of the adduct (64) and chain transfer is reversible (see also Section 6.2.7.2 and Section 9.5.2).^{36,76,79,109}



The most used transfer agents in this class are the methacrylate macromonomers (*e.g.* **66-68**) and AMS dimer (**76**). The applications of these compounds are summarized in a review.¹¹⁰

The rate constants (k_{add}) for addition of the MMA propagating radical³⁶ (and other radicals⁷⁹) to **66-68** are believed to be similar. The transfer constant of **66** is thought to be lower than **67** and **68** by more than an order of magnitude because of

306 The Chemistry of Radical Polymerization

an unfavorable partition coefficient. The fragmentation of 70 preferentially gives back 66 and the MMA propagating radical rather than 67 and the monomeric radical 69 (Scheme 6.20). The result has been attributed to steric factors.^{36,111}



Scheme 6.20

Tanaka *et al.*¹⁰⁹ observed that the adduct 71 from the monomeric MMA radicals adding to dimer was persistent and suggested that 71 may also act as a retarder or inhibitor of polymerization. However, the higher adducts 70 appear to be transient and no retardation beyond that expected from a reduced gel effect is observed.³⁶



71

Transfer constants of the methacrylate macromonomers in MMA polymerization do not depend on the ester group but are slightly higher for MAA trimer. Compounds 72 and 73 are derived from the MMA trimer (67) by selective hydrolysis or hydrolysis and reesterification respectively. They offer a route to telechelic polymers.



In the case of polymerization of monosubstituted monomers (*e.g.* S, BA) with **66-68**, copolymerization of the macromonomer to form a graft copolymer is a significant side reaction.⁷⁶

| | | | _ | h | |
|----------------------|------------------|-------------------|----------------------|-------------------|------------|
| Transfer agent | Temperature (°C) | $C_{ m cr}$ í | for monomer | r | References |
| | | S | MMA | EA | |
| 66 | 60 °C | - | 0.013 | 0.12 ^d | 36,112 |
| 67 | 60 °C | 0.55 ^d | 0.19 | 0.84^{d} | 36.112 |
| 68 , n=2 | 60 °C | - | 0.31 | - | 36 |
| 68 , av. n=14 | 60 °C | - | 0.21 | - | 36 |
| 72 | 60 °C | - | 0.18 | - | 111 |
| 63 | 60 °C | - | 0.26 | - | 111 |
| 73 | 60 °C | - | 0.18 | - | 111 |
| 74 | 60 °C | - | 0.27 | - | 111 |
| 75 | 60 °C | - | 0.015 | - | 113 |
| 76 | 110 °C | 0.20 | 0.13 | - | 114,115 |
| 80 | 60 °C | $0.552^{c.d}$ | 0.123 ^{e.d} | - | 116 |

Table 6.11 Transfer Constants for Macromonomers^a

 a Bulk, medium comprises only monomer and transfer agent.
 b Transfer constants rounded to two significant figures.
 c Significant retardation observed.
 d. Copolymerization observed as side reaction.

For polymerization of MMA in the presence of the macromonomers 77,¹¹⁷ 78^{118} and 79^{119} where the leaving group is a primary or secondary radical, the adduct radical partitions between fragmentation and propagation. In the case of **80**, where the leaving group is a more stable secondary radical,¹¹⁶ fragmentation is the favored pathway but copolymerization is still observed.

| $\begin{array}{c} CH_3\\ CH_2-{C}-CH_3\\ CH_2={C}CO_2CH_2Ph\\ CO_2CH_2Ph\end{array}$ | CH ₃ CH ₂ −C−CH ₃ CH ₂ =C Ph Ph | CH₂ [−] CH₂ CH₂=C CO₂Me CO₂Me |
|--|--|--|
| 75 | 76 | 77 |



6.2.3.5 Thionoester and related transfer agents

Other transfer agents which react with propagating species by an additionfragmentation mechanism include the thione derivatives $(81-83)^{120-122}$ and RAFT agents (Chapter 9). The thiohydroxamic esters 82 and 83 are sometimes known as Barton esters because of the work of Barton and coworkers who explored their use as radical generators in organic chemistry.¹²³⁻¹²⁵ Transfer constants for some thione derivatives are provided in Table 6.12. The initiating species formed from 82 and 83 are acyloxy radicals which may undergo decarboxylation before initiating a new chain (Scheme 6.21).



Scheme 6.21

Benzyl thionobenzoate (81) is believed to be ineffective as a transfer agent in MMA polymerization because of an unfavorable partition coefficient. PMMA• is

a much better radical leaving group than benzyl radical. Analogous benzyl thiocarbonylthio compounds are also ineffective as RAFT agents (Section 9.5.3).

Table 6.12 Transfer Constants for Thionoester and Related Transfer Agentsat 60 °Ca

| Transfer agent | | $C_{\rm tr}$ for r | | References | |
|-----------------------------|-----|--------------------|------------------|------------------|-----|
| | S | MMA | MA | VAc | |
| 81 | 1.0 | ~0 | 1.2 ^c | >20 ^d | 122 |
| 82 R= $C_{15}H_{31}$ | 3.8 | 4.0 | ~20 ^c | ~36 ^d | 121 |
| 82 R=PhCH ₂ | 3.9 | 4.3 | - | ~80 ^d | 121 |
| 82 R=Ph | - | 2.8 | - | - | 121 |
| 83 R= $C_{15}H_{31}$ | 0.3 | 0.6 | 3.1 | 9.7° | 121 |
| 83 R=PhCH ₂ | 1.0 | 1.0 | - | 18^{d} | 121 |

a Bulk, medium comprises only monomer and transfer agent. b Transfer constants rounded to two significant figures. c Significant retardation observed. d Strong retardation observed.

These thiohydroxamic esters have seen use in grafting of PAN onto PE,¹²⁶ of PS, PAM and PNIPAM onto cellulose^{127,128} and of PS, PMMA, PVP and PAM onto poly(arylene ether sulfone).¹²⁹ The process involves derivitization of a parent carboxy functional polymer to form the thiohydoxamic ester **82** (R=polymer) which then behaves as a polymeric transfer agent and/or radical generator.

6.2.4 Abstraction-Fragmentation Chain Transfer

Other multistep mechanisms for chain transfer are possible. An example is abstraction-fragmentation chain transfer shown by silylcyclohexadienes (84, Scheme 6.22).¹³⁰



The cyclohexadiene **84** is a good H donor but the cyclohexadienyl radical **85** is slow to react and fragments to provide the silyl radical **86** which initiates polymerization. The reported transfer constant for **84** in styrene polymerization at 80 °C is very low (0.00045).¹³⁰

310 The Chemistry of Radical Polymerization

6.2.5 Catalytic Chain Transfer

Enikolopyan *et al.*¹³¹ found that certain Co^{II} porphyrin complexes (*e.g.* **87**) function as catalytic chain transfer agents. Later work has established that various square planar cobalt complexes (*e.g.* the cobaloximes **88-92**) are effective transfer agents.^{132,133} The scope and utility of the process has been reviewed several times, ^{110,134-138} most recently by Heuts *et al*, ¹³⁷ Gridnev, ¹³⁸ and Gridnev and Ittel.¹¹⁰ The latter two references^{110,138} provide a historical perspective of the development of the technique.

The major applications of catalytic chain transfer are in molecular weight control and in synthesis of macromonomers based on methacrylate esters. However, they have also been shown effective in polymerizations and copolymerizations of MAA, MAM, MAN, AMS, S and some other monomers.

A major advantage of catalytic transfer agents over conventional agents is that they have very high transfer constants. The value of $C_{\rm tr}$ in MMA polymerization is in the range $10^3 - 10^5$ (Table 6.13), thus only very small amounts are required to bring about a large reduction in molecular weight. Exact values for $C_{\rm tr}$ are dependent on the reaction conditions (Section 6.2.5.3)^{131,132,139,140} and, for chain lengths ≤ 12 , on the molecular weight of the propagating species.^{139,140} Ideally, they are not used up during polymerization (Section 6.2.5.1).



6.2.5.1 Mechanism

The mechanism proposed for catalytic chain transfer¹³² is shown in Scheme 6.23 for MMA polymerization. The Co^{II} complex (93) rapidly and reversibly

combines with carbon-centered radicals. The product, the alkyl Co^{III} complex (96), may eliminate the cobalt hydride (95) to form a macromonomer (94). Alternatively, the Co^{II} complex (93) may undergo disproportionation with the carbon-centered radical to give the same products (94 and 95). It is also possible that both mechanisms operate simultaneously. The cobalt hydride (95) reinitiates polymerization by donating a hydrogen atom to monomer and in doing so regenerates the cobalt complex (93). The majority of chains formed in the presence of these reagents will have one unsaturated end group (94). With S,¹⁴¹⁻¹⁴³ acrylate esters¹⁴⁴ and other monosubstituted monomers, the

With S, ⁽⁴⁾⁽⁴⁾ acrylate esters¹⁴⁴ and other monosubstituted monomers, the adduct (98) has greater intrinsic stability. The overall mechanism proposed for catalytic chain transfer shown in Scheme 6.24 for the case of S polymerization is similar to that for MMA polymerization. However, hydrogen transfer to cobalt gives products (97) that have a 1,2-disubstituted double bond and appear inert under the polymerization conditions. The greater stability of 98 is the probable cause of retardation in homopolymerizations involving, in particular, acrylate esters and VAc. Stability is such that certain cobalt complexes have been exploited in living polymerization of acrylate esters (Section 9.3.9.1). Higher temperatures favor chain transfer over coupling and polymerizations of acrylate esters to achieve molecular weight control have been successfully carried out at >110 °C. Molecular weight control with less retardation can also be achieved by carrying out polymerizations in the presence of small amounts of an added α -methyl vinyl monomer (*e.g.* AMS). In this case, the dominant transfer process involves the α -methyl vinyl monomer.^{3,145}



Chain transfer:



Macromonomers such as **66**, **68** and **94** are themselves catalytic chain transfer agents (Section 6.2.3.4) and transfer to macromonomer is one mechanism for chain extension of the initially formed species. The adduct species in the case of monomeric radical adding dimer (**100**) may also react by chain transfer to give **101** which is inert under polymerization conditions (Scheme 6.25). Polymerizations to

give trimer may contain a significant amount of **101** as a byproduct.¹¹¹ In the case of higher species seission is fast relative to chain transfer and the corresponding byproducts are not observed. It is also thought that the reaction of **93** with a propagating radical to give cobalt hydride **95** and macromonomer is reversible.¹⁴⁶

6.2.5.2 Catalysts

Many catalysts have been screened for activity in catalytic chain transfer. A comprehensive survey is provided in Gridnev and Ittel's review.¹¹⁰ The best known, and to date the most effective, are the cobalt porphyrins (Section 6.2.5.2.1) and cobaloximes (Sections 6.2.5.2.2 and 6.2.5.2.3). There is considerable discrepancy in reported values of transfer constants. This in part reflects the sensitivity of the catalysts to air and reaction conditions (Section 6.2.5.3).

6.2.5.2.1 Cobalt porphyrin and related complexes

Many Co^{II} porphyrins (87)^{110,131} and phthalocyanine complexes (102)¹¹⁰ have been examined for their ability to function as catalytic chain transfer agents and much mechanistic work has focused on the use of these catalysts. The more widespread application of these complexes has been limited because they often have only sparing solubility and they are highly colored.

While in most complexes the cobalt is coordinated to four nitrogens, there are some exceptions such as 103.¹⁴⁷



6.2.5.2.2 Cobalt (II) cobaloximes

Much of the recent literature relates to BF_2 -bridged Co^{ll} cobaloximes based on dimethyl (89) or diphenyl glyoxime (104).¹¹⁰ The BF_2 -bridged cobaloximes (*e.g.* 89) show greater stability to hydrolysis than analogous H-bridged species (*e.g.* 88). The diphenylglyoxime complexes (104) show enhanced air and hydrolytic stability

with respect to the corresponding dimethylglyoxime complexes (89) but are less active (Table 6.13 on page 316).

The activity in MMA polymerization can be dramatically affected by the apical ligands. Apical aquo or alcohol ligands are labile and rapidly exchange with the polymerization medium. Lewis base ligands (*e.g.* pyridine, triphenyl phosphine) are comparatively stable. In MMA polymerization, it is found that activity increases with the basicity of the ligand. With alkyl Co^{III} complexes, a different order is found possibly because the type of apical ligand also controls the rate of initial generation of the active Co^{III} complex.



6.2.5.2.3 Cobalt (III) cobaloximes

Various Co^{II} cobaloximes (90-92) have also been used as catalytic chain transfer agents.^{133,148,149} To be effective, the complex must be rapidly transformed into the active Co^{II} cobaloximes under polymerization conditions. The mechanism of catalytic chain transfer is then identical to that described above (6.2.5.1).

When R is secondary or tertiary alkyl, the Co^{II} species may be generated by Co^{III} -C bond homolysis. Thus, **107** is thermally labile and can be used both as an initiator and a catalytic chain transfer agent at 60 °C.¹⁴⁸ When R is primary alkyl, halogen or pseudohalogen the Co^{II} species is generated by radical induced reduction.^{149,150} The cobaloxime **108** based on diethylglyoxime is thermally stable at temperatures up to 100 °C but is rapidly reduced in the presence of AIBN at 60 °C.¹⁴⁹ These two cobaloximes (**107** and **108**) appear equally effective as catalytic chain transfer agents.¹⁴⁹ The corresponding cobaloxime based on dimethylglyoxime (**109**) is not readily reduced and appears inactive under the same conditions.



Scheme 6.26



Scheme 6.27



6.2.5.2.4 Other catalysts

Other complexes also react with propagating radicals by catalytic chain transfer.¹¹⁰ These include certain chromium,^{151,152} molybdenum^{152,153} and iron¹⁵⁴ complexes. To date the complexes described appear substantially less active than the cobaloximes and are more prone to side reactions.

6.2.5.3 Reaction conditions

Catalytic chain transfer has now been applied under a wide range of reaction conditions (solution, bulk, emulsion, suspension) and solvents (methanol, butan-2-one, water). The selection of the particular complex, the initiator, the solvent and the reaction conditions can be critical. For example:

- (a) Initiators that generate oxygen centered radicals (*e.g.* BPO) or primary alkyl radicals (*e.g.* LPO) are generally to be avoided. The Co^{ll} cobaloximes can react with the initiator-derived radicals to create a species that is inactive or less active under the polymerization conditions. Preferred initiators are those that resemble propagating species and azo compounds that generate tertiary radicals such as AIBN.
- (b) The Co^{II} cobaloximes can be extremely air sensitive and rigorous exclusion of air is essential for reproducibility. Co^{III} complexes (92) have enhanced air stability with respect to the Co^{II} cobaloximes.¹⁴⁹ Solutions are stable at room temperature even in the presence of air. The active species is generated in *situ* under the polymerization conditions. However, rigorous exclusion of air from the polymerization is still essential.

316 The Chemistry of Radical Polymerization

- (c) There are reports of extreme sensitivity to solvent and monomer purity.¹⁵⁵
- (d) In emulsion polymerization, the partition coefficient of the complex between the droplet, aqueous and particle phases is important.¹⁴⁹ The complex should partition preferentially into the particle phase and yet in *ab initio* polymerizations have sufficient water solubility to be able to transfer from the monomer droplet to the particle phase. The very high activity of the cobalt complex, and the concentration typically used, mean that there may be only a few molecules of complex per particle.
- (e) In solution polymerization, the apical ligand of cobaloxime complexes may exchange with the medium changing the activity and solubility of the complex.
- (f) Intermediate Co^{III} complexes may be relatively stable at low temperatures reducing that concentration of the active Co^{II} complex and the propagating radicals. For S and acrylate esters transfer constants and rates of polymerization increase with increasing temperature.¹¹⁵
- (g) Co^{III} complexes (alkyl Co^{III} catalysts, Co^{III} intermediates) are light sensitive and will dissociate to the active Co^{II} complex and propagating radicals on irradiation with visible light. For S and acrylate esters higher transfer constants can be achieved by irradiation of the sample.¹⁵⁶

Catalytic inhibition has been reported for MAM and MMA polymerizations with DMF solvent. $^{\rm 157}$

| Transfer | | $C_{\rm p}$ for monomer ^b | |
|----------|------------------------|--------------------------------------|----------------------------|
| agent | MMA | Sc | MA ^c |
| 89 | 32000-37000155.158,159 | 1500-7000 115.141,142,156,158 | 50-1000 ^{156,160} |
| 104 | 18000^{161} | 400 ¹⁶¹ | - |
| | | | |

Table 6.13 Transfer Constants for Cobalt Complexes at 60 °C^a

a Bulk medium comprises only monomer and transfer agent. b Transfer constants rounded to two significant figures. c The apparent transfer constant with monosubstituted monomers is strongly dependent on reaction conditions (see text). The lower limit shown is the effective transfer constant in bulk polymerization. The upper limit is the likely actual transfer constant.

6.2.6 Transfer to Monomer

Non-zero transfer constants (C_M) can be found in the literature for most monomers. Values of C_M for some common monomers are given in Table 6.14. For S and the (meth)acrylates the value is small, in the range 10^{-5} - 10^{-4} . Transfer to monomer is usually described as a process involving hydrogen atom transfer. While this mechanism is reasonable for those monomers possessing aliphatic hydrogens (*e.g.* MMA, VAc, allyl monomers), it is less acceptable for monomers possessing only vinylic or aromatic hydrogens (*e.g.* VC, S). The details of the mechanisms by which transfer occurs are, in most cases, not proven. Mechanisms

for transfer to monomer that involve loss of vinylic hydrogens seem unlikely given the high strength of the bonds involved.

Irrespective of the mechanism by which transfer to monomer occurs, the process will usually produce an unsaturated radical as a byproduct. This species initiates polymerization to afford a macromonomer that may be reactive under typical polymerization conditions.

| Monomer | Temperature (°C) | $C_{\rm M} \times 10^4$ | Ref. |
|----------------|------------------|-------------------------|---------|
| S | 60 | 0.6 | 162 |
| ΜΜΛ | 60 | 0.1 | 163 |
| MA | 60 | 0.4 | 164 |
| AN | 60 | 0.3 | 165 |
| VΛc | 60 | 1.8 | 166 |
| VC | 100 | 50 | 167.168 |
| allyl acetate | 80 | 1600 | 169 |
| allyl chloride | 80 | 700 | 169 |

 Table 6.14
 Selected Values for Transfer Constants to Monomer^a

a Values rounded to one significant figure and are taken from the references shown. There is considerable scatter in literature values for many monomers.⁴⁹

6.2.6.1 Styrene

The value of $C_{\rm M}$ has been determined by a number of groups as 6×10^{-5} (Table 6.14).⁴⁹ However, the mechanism of transfer has not been firmly established. A mechanism involving direct hydrogen abstraction seems unlikely given the high strength of vinylic and aromatic C-H bonds. The observed value of $C_{\rm M}$ is only slightly lower than $C_{\rm tr}$ for ethylbenzene ($\sim 7 \times 10^{-5}$).⁴⁹



Scheme 6.28

It has been proposed that transfer to monomer may not involve the monomer directly but rather the intermediate (110) formed by Diels-Alder dimerization (Scheme 6.28).¹⁷⁰ Since 110 is formed during the course of polymerization, its involvement could be confirmed by analysis of the polymerization kinetics.

6.2.6.2 Vinyl acetate

There is a considerable body of evidence (kinetic studies, chemical and NMR analysis) indicating that transfer to VAc monomer involves largely, if not exclusively, the acetate methyl hydrogen to give radical **111** (Scheme 6.29).^{171,172} This radical **(111)** initiates polymerization to yield a reactive macromonomer **(112)**.



Scheme 6.29

Starnes *et al.*¹⁷³ have provided support for the above mechanism (Scheme 6.29) by determining the unsaturated chain ends (**112**) in low conversion PVAc by ¹³C NMR. They were able to distinguish (**112**) from chain ends that might have been formed if transfer involved abstraction of a vinylic hydrogen. The number of unsaturated chain ends (**112**) was found to equate with the number of -CH₂OAc ends suggesting that most chains are formed by transfer to monomer. Starnes *et al.*¹⁷³ also found an isotope effect $k_{\rm H}/k_{\rm D}$ of 2.0 for the abstraction reaction with CH₂=CHO₂CCD₃ as monomer. This result is consistent with the mechanism shown in Scheme 6.28 but is contrary to an earlier finding.¹⁷⁴

Stein¹⁶⁶ has indicated that the reactivity of the terminal double bond of the macromonomer (112) is 80% that of VAc monomer. The kinetics of incorporation of 112 have also been considered by Wolf and Burchard¹⁷⁵ who concluded that 112 played an important role in determining the time of gelation in VAc homopolymerization in bulk.

6.2.6.3 Vinyl chloride

It has been proposed that chain transfer to monomer determines the length of the polymer molecules formed during VC polymerization.¹⁷⁶ The mechanism for transfer, involving an addition-elimination sequence consequent on head addition to monomer (Section 4.3.1.2), was first proposed by Rigo *et al.*¹⁷⁷ Direct evidence for this pathway has been provided by Starnes *et al.*¹⁷⁸ and Park and Saleem.¹⁷⁹ This pathway (Scheme 6.30) accounts for $C_{\rm M}$ for VC being much greater than $C_{\rm M}$ for other commercially important monomers (Table 6.14) where the analogous pathway is not available. Starnes and Wojciechowski¹⁸⁰ have reported kinetic data which suggest that the chlorine atom does not have a discrete existence but is transferred directly from the β -chloroalkyl radical to VC.





Scheme 6.30

6.2.6.4 Allyl monomers

Transfer to monomer is of particular importance during the polymerization of allyl esters (113, X=O₂CR), ethers (113, X=OR), amines (113, X=NR₂) and related monomers.^{169,181,182} The allylic hydrogens of these monomers are activated towards abstraction by both the double bond and the heteroatom substituent (Scheme 6.31). These groups lend stability to the radical formed (114) and are responsible for this radical adding monomer only slowly. This, in turn, increases the likelihood of side reactions (*i.e.* degradative chain transfer) and causes the allyl monomers to retard polymerization.





For allyl acetate a significant deuterium isotope effect supports the hydrogen abstraction mechanism (Scheme 6.31).¹⁸³ Allyl compounds with weaker CH₂-X bonds (**113** X=SR, SO₂R, Br, *etc.*) may also give chain transfer by an addition-fragmentation mechanism (Section 6.2.3).

Diallyl monomers find significant use in cyclopolymerization (Section 4.4.1). Transfer to monomer is of greater importance in polymerizations of allyl than it is in diallyl monomers.¹⁸⁴ This might, in part, reflect differences in the nature of the propagating species [*e.g.* a secondary alkyl (115) *vs* a primary alkyl radical (116)]. Electronic factors may also play a role.¹⁸⁵



The polymerizability of allyl monomers is thought to be directly related to the abstractability of α -hydrogens.¹⁸⁶

6.2.7 Transfer to Polymer

Two forms of transfer to polymer should be distinguished:

- (a) Intramolecular reaction or backbiting, which gives rise to short chain branches (length usually ≤5 carbons).
- (b) Intermolecular reaction, which generally results in the formation of long chain branches.

The intramolecular process does not give rise to a new polymer chain and is considered in Section 4.4.3. It will not be considered further in this section.

Available evidence suggests that the main reaction accounting for transfer to vinyl polymers (*e.g.* PMA, PVAc, PVC, PVF) usually involves abstraction of a methine hydrogen (Scheme 6.32) (Sections 6.2.7.3, 6.2.7.4, 6.2.7.5 and 6.2.7.6 respectively). However, definitive evidence for the mechanism is currently only available for a few polymers (*e.g.* PVAc, PVF).



Scheme 6.32

Table 6.15 Transfer Constants to Polymer^a

| Monomer | Temperature (°C) | $C_{ m P} { m x10^4}$ |
|---------|------------------|-----------------------|
| S | 60 | 1.9-16 |
| MMA | 60 | 0.1-360 |
| MA | 60 | 0.5-1.0 |
| AN | 60 | 3.5 |
| VAc | 60 | 1.4-47 |
| VC | 50 | 5 |
| Е | 175 | 110 |

^a Numbers are taken from the *Polymer Handbook*⁴⁹ and have been rounded to two significant figures.

Transfer constants to polymer (C_P) are not as readily determined as other transfer constants because the process need not lead to an overall lowering of molecular weight. If transfer occurs by hydrogen-atom abstraction from the polymer backbone then, for every polymer chain terminated by transfer, another branched chain is formed. In these circumstances the overall molecular weight remains constant. The extent of chain transfer can then be estimated by measuring the number of long chain branches or by analyzing the molecular weight distribution. As NMR measurement of long chain branching relies on determining the branch points, a major analytical problem is distinguishing the long chain branches from the short chain branches formed by backbiting.

The values of $C_{\rm P}$ to added polymer are measurable in circumstances where the added material is readily distinguishable from that being formed *in situ*, for example, if it is of significantly different molecular weight or if it is uniquely labeled.¹⁸⁷ Studies with model compounds suggest that oligomers of chain length ≥ 3 can be used to provide a good estimate of the transfer constant.^{188,189}

For some polymers, the value of C_P depends on the polymer molecular weight (*e.g.* Section 6.2.7.2). This may help account for the wide range of values for C_P in the literature (Table 6.15).

6.2.7.1 Polyethylene

The presence of long chain branches in low density polyethylene (LDPE) accounts for the difference in properties (*e.g.* higher melt strength, greater toughness for the same average molecular weight) between LDPE and linear low density polyethylene (LLDPE, made by coordination polymerization).

Long chain branching (>8 carbons) in polyethylene can be detected by ¹³C NMR analysis.¹⁹⁰⁻¹⁹³ However, the length and distribution of the branches are more difficult to determine. Measurements of long chain branching have been made by GPC-light scattering¹⁹⁴⁻¹⁹⁶ or GPC-viscometry.¹⁹⁶⁻¹⁹⁸ The extent of long chain branching is known to be strongly dependent on the reactor design and the reaction conditions employed. These studies indicate that, for a given sample, the branch frequency appears to decrease with increasing molecular weight of PE.¹⁹⁶ An explanation was not given.

6.2.7.2 Poly(alkyl methacrylates)

ω-Unsaturated poly(alkyl methacrylates) (e.g. 117) are produced during radical polymerization of MMA through termination by disproportionation (Sections 5.2.2.1.3 & 5.2.2.2.3). Schulz *et al.*¹⁹⁹ were the first to suggest that reactions of these species (117) may complicate MMA homopolymerization. The ω-unsaturated poly(alkyl methacrylates) may act as a chain transfer agent in polymerization by the mechanism shown in Scheme 6.33 (Section 6.2.3.4).

In polymerization of methacrylates, the adducts formed by addition to the macromonomer radicals are relatively unreactive towards adding further monomer

The Chemistry of Radical Polymerization

and most undergo β -scission. There are two possible pathways for β -scission: one pathway leads back to starting materials; the other gives a new propagating radical and a macromonomer. Transfer is catalytic in macromonomer.



Values of C_P measured in the presence of added PMMA (for example) will depend on how the PMMA was prepared and its molecular weight (*i.e.* on the concentration of unsaturated ends). PMMA formed by radical polymerization in

concentration of unsaturated ends). PMMA formed by radical polymerization in the presence of a good H-donor transfer agent (or by anionic polymerization) would have only saturated chain ends. These PMMA chains should have a different transfer constant to those formed by normal radical polymerization where termination occurs by a mixture of combination and disproportionation. This could account for some of the variation in the values of $C_{\rm P}$ for this polymer.

6.2.7.3 Poly(alkyl acrylates)

Chain transfer to polymer is reported as a major complication and is thought to be unavoidable in the polymerization of alkyl acrylates.²⁰⁰⁻²⁰² The mechanism is believed to involve abstraction of a tertiary backbone hydrogen (Scheme 6.32). It has been proposed that this process and the consequent formation of branches may contribute to the early onset of the gel or Norrish-Trommsdorff effect in the polymerization of these monomers. At high temperatures the radicals formed may undergo fragmentation.

Copolymerization of macromonomers formed by backbiting and fragmentation is a second mechanism for long chain branch formation during acrylate polymerization (Section 4.4.3.3). The extents of long and short chain branching in acrylate polymers in emulsion polymerization as a function of conditions have been quantified.²⁰²

6.2.7.4 Poly(vinyl acetate)

The degree of branching in PVAc is strongly dependent on the polymerization conditions. Differences in the degree of branching are thought to be one of the main factors responsible for substantial differences in properties between various commercial samples of PVAc or PVA.²⁰³⁻²⁰⁵

PVAc is known to contain a significant number of long chain branches. Branches to the acetate methyl may arise by copolymerization of the VAc macromonomer produced as a consequence of transfer to monomer (Section 6.2.6.2). Transfer to polymer may involve either the acetate methyl hydrogens (Scheme 6.34) or the methine (Scheme 6.35) or methylene hydrogens of the polymer backbone.







Scheme 6.35 Non-hydrolyzable branch formation.

The presence of hydrolyzable long chain branches in PVAc was established by McDowell and Kenyon²⁰⁶ in 1940. They observed a reduction in molecular weight obtained on successively hydrolyzing and reacetylating samples of PVAc. Only branches to the acetate methyl will be lost on hydrolysis of the polymer; *i.e.* on conversion of PVAc to PVA.

The proposal that PVAc also has non-hydrolyzable long chain branches stems from the finding that PVA also possesses long chain branches. Nozakura *et al.*^{171,207} suggested, on the basis of kinetic measurements coupled with chemical analysis, that chain transfer to PVAc involves preferential abstraction of backbone (methine) hydrogens (*ca* 5:1 *vs* the acetate methyl hydrogens at 60 °C).

¹H and ¹³C NMR studies on PVAc or PVA also provide information on the nature of branches.^{203,204,208,209} Dunn and Naravane²⁰³ and Bugada and Rudin²⁰⁴ proposed that the difference in intensity of the methylene and methine regions of the ¹³C NMR spectrum could be used as a quantitative measure of the non-hydrolyzable branches (short chain + long chain) in PVA. However, this approach has been questioned by Vercauteren and Donners²⁰⁴ because of the relatively large errors inherent in the method.

In order to prove that non-hydrolyzable long chain branches are present in a pre-existing sample of PVA, it is required that long chain branches can be distinguished from short chain branches. This distinction cannot be made solely on the basis of the ¹³C NMR data. Extents of long chain branching can be obtained from GPC coupled with viscometry, ultracentrifugation or low angle laser light scattering on PVAc or reacetylated PVA.^{205,210}

The extent of branching, of whatever type, is dependent on the polymerization conditions and, in particular, on the solvent and temperature employed and the degree of conversion. Nozakura *et al.*¹⁷¹ found that, during bulk polymerization of VAc, the extent of transfer to polymer increased and the selectivity (for abstraction of a backbone *vs* an acetoxy hydrogen) decreases with increasing temperature.

Adelman and Ferguson²⁰⁸ have suggested, on the basis of ¹H NMR data (detection of CH₃CH(OH)CH(OH)CH₂– ends) and chemical analyses (formation of acetaldehyde on periodate cleavage of 1,2-glycol units) on PVA, that the radical formed by head addition to VAc may be responsible for a high proportion of transfer events. Their PVAc was prepared in methanol at 60-75 °C and much of the transfer involves the solvent. ¹³C NMR^{209,211} studies on several commercial PVA samples showed that those materials had equal numbers of head-to-head and tail-to-tail linkages (Section 4.3.1.1) and indicated the presence of -CH₂OH ends (*i.e.* most transfer involves the normal propagating species). These polymers are likely to have been prepared by emulsion polymerization, thus most transfer will involve monomer or polymer.

Hatada *et al.*²¹² have indicated that PVAc prepared in aromatic solvents (benzene, chlorobenzene) at 60 °C has fewer branch points than the polymer prepared in ethyl acetate under similar conditions. They attributed this observation to complexation of the propagating radical in the aromatic solvents and the

different reactivity of this complexed radical. They have also reported that VAc polymerization is substantially slowed in aromatic solvents and this was also attributed to complexation of the propagating radical²¹³ (Section 8.3.1.1).

6.2.7.5 Poly(vinyl chloride)

The microstructure of PVC has been the subject of numerous studies (Sections 4.3.1.2 and 6.2.6.3).²¹⁴ Starnes *et al.*¹⁶⁸ determined the long chain branch points by NMR studies on PE formed by Bu₃SnH reduction of PVC. They concluded that the probable mechanism for the formation of these branches involved transfer to polymer that occurred by hydrogen abstraction of a backbone methine by the propagating radical (Scheme 6.32).

6.2.7.6 Poly(vinyl fluoride)

Ovenall and Uschold²¹⁵ have recently measured the concentration of branch points (tertiary F, Scheme 6.32) in PVF by ¹⁹F NMR. These were found to account for between 0.5 to 1.5% of monomer units depending on reaction conditions. Branching was found to be favored by lower reactor pressures or higher reactor temperatures. More branching was observed for polymers produced in batch as opposed to continuous reactors. This effect was attributed to longer residence time of the polymer in the reactor.

6.2.8 Transfer to Initiator

The mechanism and incidence of transfer to various initiators is discussed in Chapter 3. See, in particular, Sections 3.2.10 (introduction), 3.3.2.1.4 (dialkyl diazenes including AIBN), 3.3.2.1.4 (diacyl peroxides including BPO), 3.3.2.3.1 (peroxyesters), 3.3.2.4 (dialkyl peroxides) and 3.3.2.5 (alkyl hydroperoxides).

6.3 References

- 1. Flory, P.J. J. Am. Chem. Soc. 1937, 59, 241.
- 2. Flory, P.J. *Principles of Polymer Chemistry*; Cornell University Press: Ithaca, New York, 1953.
- 3. Chiefari, J.; Rizzardo, E. In *Handbook of Radical Polymerization*; Davis, T.P.; Matyjaszewski, K., Eds.; John Wiley & Sons: Hoboken, NY, 2002; p 263.
- Barson, C.A. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 171.
- 5. Farina, M. Makromol. Chem., Macromol. Symp. 1987, 10/11, 255.
- 6. Eastmond, G.C. In *Comprehensive Chemical Kinetics*; Bamford, C.H.; Tipper, C.F.H., Eds.; Elsevier: Amsterdam, 1976; Vol. 14A, p 153.
- 7. Palit, S.R.; Chatterjee, S.R.; Mukherjee, A.R. In *Encyclopedia of Polymer Science* and *Technology*; Mark, H., F.; Gaylord, N.G.; Bikales, N.M., Eds.; Wiley: New York, 1966; Vol. 3, p 575.
- 8. Boutevin, B. Adv. Polym. Sci. 1990, 94, 69.

- 9. Heitz, W. In *Telechelic Polymers: Synthesis and Applications*; Goethals, E.J., Ed.; CRC Press: Boca Raton, Florida, 1989; p 61.
- 10. Corner, T. Adv. Polym. Sci. 1984, 62, 95.
- 11. Starks, C.M. Free Radical Telomerization; Academic Press: New York, 1974.
- 12. Mayo, F.R. J. Am. Chem. Soc. 1943, 65, 2324.
- 13. Clouet, G.; Knipper, M. Makromol. Chem. 1987, 188, 2597.
- 14. Bamford, C.H. Polym. Commun. 1989, 30, 36.
- 15. Harrisson, S.; Kapfenstein-Doak, H.; Davis, T.P. Macromolecules 2001, 34, 6214.
- 16. Scott, G.P.; Foster, F.J. Macromolecules 1969, 2, 428.
- 17. Scott, G.P.; Elghoul, A.M.R. J. Polym. Sci., Part A-1 1970, 8, 2255.
- 18. Scott, G.P.; Wang, J.C. J. Org. Chem. 1963, 28, 1314.
- 19. Barson, C.A.; Mather, R.R.; Robb, J.C. Trans. Faraday Soc. 1970, 66, 2585.
- 20. Mayo, F.R. J. Am. Chem. Soc. 1948, 70, 3689.
- 21. Asahara, T.; Makishima, T. Kogyo Kagaku Zasshi 1966, 69, 2173.
- 22. Englin, B.A.; Onishchenko, T.A. Izv. Akad. Nauk. SSSR, Ser. Khim. 1969, 1906.
- Englin, B.A.; Onishchenko, T.A.; Freidlina, R.K. Izv. Akad. Nauk SSSR, Ser. Khim. 1968, 11, 2489.
- 24. Moad, G.; Moad, C.L. Macromolecules 1996, 29, 7727.
- 25. Heuts, J.P.A.; Davis, T.P.; Russell, G.T. Macromolecules 1999, 32, 6019.
- 26. Whang, B.Y.C.; Ballard, M.J.; Napper, D.H.; Gilbert, R.G. Aust. J. Chem. 1991, 44, 1133.
- 27. Clay, P.A.; Gilbert, R.G. Macromolecules 1995, 28, 552.
- 28. Bamford, C.H. J. Chem. Soc., Faraday Trans. 1 1976, 72, 2805.
- 29. Bamford, C.H.; Basahel, S.N. J. Chem. Soc., Faraday Trans. 1 1978, 74, 1020.
- 30. Bamford, C.H.; Basahel, S.N. Polymer 1978, 19, 943.
- 31. Walling, C. J. Am. Chem. Soc. 1948, 70, 2561.
- 32. Cardenas, J.N.; O'Driscoll, K.F. J. Polym. Sci., Polym. Chem. Ed. 1977, 15, 2097.
- 33. Stickler, M. Makromol. Chem. 1979, 180, 2615.
- 34. Harwood, H.J.; Mcdsker, R.E.; Rapo, A. In *MakroAkron 94 Abstracts*; IUPAC, 1994; p 16.
- 35. Sunder, A.; Muelhaupt, R. Macromol. Chem. Phys. 1999, 200, 58.
- 36. Moad, G.; Moad, C.L.; Rizzardo, E.; Thang, S.H. Macromolecules 1996, 29, 7717.
- 37. Nair, C.P.R.; Chaumont, P.; Colombani, D. Macromolecules 1995, 28, 3192.
- 38. Jenkins, A.D.; Jenkins, J. In *Polymer Handbook*, 4th ed.; Brandup, J.; Immergut, E.H.; Grulke, E.A., Eds.; John Wiley and Sons: New York, 1999; p II/321.
- 39. Greenley, R.Z. In *Polymer Handbook*, 4th ed.; Brandup, J.; Immergut, E.H.; Grulke, E.A., Eds.; John Wiley and Sons: New York, 1999; p II/309.
- 40. Moad, G.; Chiefari, J.; Mayadunne, R.T.A.; Moad, C.L.; Postma, A.; Rizzardo, E.; Thang, S.H. In *Macromol. Symp.*, 2002; Vol. 182, p 65.
- Chiefari, J.; Mayadunne, R.T.A.; Moad, C.L.; Moad, G.; Rizzardo, E.; Postma, A.; Skidmore, M.A.; Thang, S.H. *Macromolecules* 2003, *36*, 2273.
- 42. Chong, Y.K.; Krstina, J.; Le, T.P.T.; Moad, G.; Postma, A.; Rizzardo, E.; Thang, S.H. *Macromolecules* **2003**, *36*, 2256.
- 43. Nair, C.P.R.; Richou, M.C.; Chaumont, P.; Clouet, G. Eur. Polym. J. 1990, 26, 811.
- 44. Businelli, L.; Gnanou, Y.; Maillard, B. Macromol. Chem. Phys. 2000, 201, 2805.
- 45. de la Fuente, J.L.; Madruga, E.L. Macromol. Chem. Phys. 2000, 201, 2152.
- 46. Hutchinson, R.A.; Paquet, D.A.; McMinn, J.H. Macromolecules 1995, 28, 5655.
- Businelli, L.; Deleuze, H.; Gnanou, Y.; Maillard, B. Macromol. Chem. Phys. 2000, 201, 1833.
- 48. O'Brien, J.L.; Gornick, F. J. Am. Chem. Soc. 1955, 77, 4757.

- 49. Ueda, A.; Nagai, S. In *Polymer Handbook*, 4th ed.; Brandup, J.; Immergut, E.H.; Grulke, E.A., Eds.; John Wiley and Sons: New York, 1999; p II/97.
- 50. Clarke, J.T.; Howard, R.O.; Stockmayer, W.H. Makromol. Chem. 1961, 44, 427.
- 51. Roy, K.K.; Pramanick, D.; Palit, S.R. Makromol. Chem. 1972, 153, 71.
- 52. Tronche, C.; Martinez, F.N.; Horner, J.H.; Newcomb, M.; Senn, M.; Giese, B. *Tetrahedron Lett.* **1996**, *37*, 5845.
- 53. Bamford, C.II.; Basahel, S.N. J. Chem. Soc., Faraday Trans. 1 1980, 76, 112.
- 54. Boutevin, B.; El Idrissi, A.; Parisi, J.P. Makromol. Chem. 1990, 191, 445.
- 55. Boutevin, B.; Lusinchi, J.-M.; Pietrasanta, Y.; Robin, J.-J. *Eur. Polym. J.* **1994**, *30*, 615.
- 56. Boutevin, B.: Pietrasanta, Y. Makromol. Chem. 1985, 186, 817.
- 57. Pryor, W.A.; Pickering, T.L. J. Am. Chem. Soc. 1962, 84, 2705.
- Costanza, A.J.; Coleman, R.J.; Pierson, R.M.; Marvel, C.S.; King, C. J. Polym. Sci. 1955, 17, 319.
- 59. Otsu, T.; Kinoshita, Y.; Imoto, M. Makromol. Chem. 1964, 73, 225.
- 60. Tsuda, K.; Otsu, T. Bull. Chem. Soc. Japan 1966, 39, 2206.
- 61. Otsu, T.; Nayatani, K. Makromol. Chem. 1958, 73, 225.
- 62. Staudner, E.; Kysela, G.; Beniska, J.; Mikolaj, D. Eur. Polym. J. 1978, 14, 1067.
- 63. Beniska, J.; Staudner, E. J. Polym. Sci., Part C 1967, 16, 1301.
- 64. Popielarz, R.; Clouet, G. Makromol. Chem. 1993, 194, 2897.
- 65. Kimura, T.; Kodaira, T.; Hamashima, M. Polym. J. 1983, 15, 293.
- 66. Tedder, J.M. Angew. Chem., Int. Ed. Engl. 1982, 21, 401.
- 67. Ameduri, B.; Boutevin, B. Macromolecules 1990, 23, 2433.
- 68. Mayo, F.R. J. Am. Chem. Soc. 1953, 75, 6133.
- Rizzardo, E.; Chong, Y.K.; Evans, R.A.; Moad, G.; Thang, S.H. Macromol. Symp. 1996, 111, 1.
- 70. Colombani, D.; Chaumont, P. Acta Polym. 1998, 49, 225.
- 71. Colombani, D. Prog. Polym. Sci. 1999, 24, 425.
- 72. Yagci, Y.; Reetz, I. React. Funct. Polym. 1999, 42, 255.
- 73. Lewis, S.N.; Miller, J.J.; Winstein, S. J. Org. Chem. 1972, 37, 1478.
- 74. Giese, B. Radicals in Organic Synthesis: Formation of Carbon-Carbon Bonds; Pergamon Press: Oxford, 1986.
- 75. Motherwell, W.B.; Crich, D. Free Radical Chain Reactions in Organic Synthesis; Academic Press: London, 1992.
- Cacioli, P.; Hawthorne, D.G.; Laslett, R.L.; Rizzardo, E.; Solomon, D.H. J. Macromol. Sci., Chem. 1986, A23, 839.
- 77. Meijs, G.F.; Rizzardo, E.; Thang, S.H. Macromolecules 1988, 21, 3122.
- 78. Meijs, G.F.; Rizzardo, E. Makromol. Chem., Rapid Commun. 1988, 9, 547.
- 79. Rizzardo, E.; Harrison, D.; Laslett, R.L.; Meijs, G.F.; Morton, T.C.; Thang, S.H. Prog. Pacific Polym. Sci. 1991, 2, 77.
- 80. Meijs, G.F.; Rizzardo, E. Makromol. Chem. 1990, 191, 1545.
- 81. Dais, V.A.; Priddy, D.B.; Bell, B.; Sikkema, K.D.; Smith, P. J. Polym. Sci., Part A: Polym. Chem. 1993, 31, 901.
- 82. Rizzardo, E.; Meijs, G.F.; Thang, S.H. Macromol. Symp. 1995, 98, 101.
- Bailey, W.J.; Endo, T.; Gapud, B.; Lin, Y.-N.; Ni, Z.; Pan, C.-Y.; Shaffer, S.E.; Wu, S.-R.; Yamazaki, N.; Yonezawa, K. J. Macromol. Sci., Chem. 1984, A21, 979.
- 84. Meijs, G.F.; Rizzardo, E.; Thang, S.H. Polym. Bull. 1990, 24, 501.
- 85. Yamada, B.; Kobatake, S.; Aoki, S. Macromol. Chem. Phys. 1994, 195, 581.
- 86. Yamada, B.; Otsu, T. Makromol. Chem. 1991, 192, 333.
- 87. Yamada, B.; Otsu, T. Makromol. Chem., Rapid Commun. 1990, 11, 513.

- 88. Yamada, B.; Satake, M.; Otsu, T. Polym. J. 1992, 24, 563.
- 89. Yamada, B.; Kato, E.; Kobatake, S.; Otsu, T. Polym. Bull. 1991, 25, 423.
- 90. Yamada, B.; Kobatake, S.; Aoki, S. Polym. Bull. 1993, 31, 263.
- 91. Meijs, G.F.; Morton, T.C.; Rizzardo, E.; Thang, S.H. Macromolecules 1991, 24, 3689.
- 92. Mathias, L.J.; Thompson, R.D.; Lightsey, A.K. Polym. Bull. 1992, 27, 395.
- Sato, T.; Seno, M.; Kobayashi, M.; Kohna, T.; Tanaka, H.; Ota, T. Eur. Polym. J. 1995, 31, 29.
- 94. Jiang, S.; Viehe, H.G.; Oger, N.; Charmot, D. Macromol. Chem. Phys. 1995, 196, 2349.
- Nair, C.P.R.; Chaumont, P.; Charmot, D. J. Polym. Sci., Part A: Polym. Chem. 1995, 33, 2773.
- 96. Busfield, W.K.; Zayas-Holdsworth, C.I.; Thang, S.H. Polymer 2000, 41, 4409.
- 97. Busfield, W.K.; Zayas-Holdsworth, C.I.; Thang, S.H. J. Polym. Sci., Part A: Polym. Chem. 2000, 39, 2911.
- 98. Yamada, B.; Hirano, T.; Kobatake, S. Polym. Bull. 2003, 49, 305.
- 99. Zink, M.-O.; Colombani, D.; Chaumont, P. Eur. Polym. J. 1997, 33, 1433.
- 100. Chaumont, P.; Colombani, D. Macromol. Chem. Phys. 1995, 196, 3643.
- 101. Meijs, G.F.; Rizzardo, E.; Thang, S.H. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem.) 1992, 33(1), 893.
- 102. Shanmugananda Murthy, K.; Kishore, K. J. Polym. Sci., Part A: Polym. Chem. 1996, 34, 1415.
- 103. Colombani, D.; Chaumont, P. J. Polym. Sci., Part A: Polym. Chem. 1994, 32, 2687.
- 104. Colombani, D.; Chaumont, P. Macromolecules 1994, 27, 5972.
- 105. Colombani, D.; Chaumont, P. Polymer 1995, 36, 129.
- 106. Chaumont, P.; Colombani, D. Macromolecules 1995, 29, 819.
- 107. Chaumont, P.; Colombani, D. Macromol. Chem. Phys. 1995, 196, 947.
- Colombani, D.; Lamps, J.-P.; Chaumont, P. Macromol. Chem. Phys. 1998, 199, 2517.
- 109. Tanaka, H.; Kawa, H.; Sato, T.; Ota, T. J. Polym. Sci., Part A: Polym. Chem. 1989, 27, 1741.
- 110. Gridnev, A.A.; Ittel, S.D. Chem. Rev. 2001, 101, 3611.
- 111. Hutson, L.; Krstina, J.; Moad, C.L.; Moad, G.; Morrow, G.R.; Postma, A.; Rizzardo, E.; Thang, S.H. *Macromolecules* **2004**, *37*, 4441.
- 112. Harrison, D.S. MSc Thesis; Swinburne University: Hawthorn, Victoria.
- 113. Haddleton, D.M.; Topping, C.; Kukulj, D.; Irvine, D. Polymer 1998, 39, 3119.
- 114. Yamada, B.; Tagashira, S.; Aoki, S. J. Polym. Sci., Part A: Polym. Chem. 1994, 32, 2745.
- Chiefari, J.; Jeffery, J.; Moad, C.L.; Moad, G.; Postma, A.; Rizzardo, E.; Thang, S.H. *Macromolecules* 2005, in press.
- 116. Nair, C.P.R.; Chaumont, P. J. Polym. Sci., Part A: Polym. Chem. 1999, 37, 2511.
- 117. Kobatake, S.; Yamada, B. J. Polym. Sci., Part A: Polym. Chem. 1996, 34, 95.
- 118. Ilirano, T.; Yamada, B. Polymer 2003, 44, 347.
- 119. Chiefari, J.; Jeffery, J.; Mayadunne, R.T.A.; Moad, G.; Rizzardo, E.; Thang, S.H. ACS Symp. Ser. 2000, 768, 297.
- 120. Meijs, G.F.; Morton, T.C.; Le, T.P.T. Polym. Int. 1991, 26, 239.
- 121. Meijs, G.F.; Rizzardo, E. Polym. Bull. 1991, 26, 291.
- 122. Meijs, G.F.; Rizzardo, E.; Le, T.P.T.; Chong, Y.K. Macromol. Chem. Phys. 1992, 193, 369.
- 123. Crich, D.; Quintero, L. Chem. Rev. 1989, 89, 1413.

- 124. Barton, D.H.R.; Bridon, D.; Fernandez-Picot, I.; Zard, S.Z. *Tetrahedron* **1987**, *43*, 2733.
- 125. Barton, D.H.R.; Samadi, M. Tetrahedron 1992, 48, 7083.
- 126. Bergbreiter, D.E.; Jing, Z. J. Polym. Sci., Part A: Polym. Chem. 1992, 30, 2049.
- 127. Daly, W.H.; Evenson, T.S. ACS Symp. Ser. 1998, 685, 377.
- 128. Daly, W.H.; Evenson, T.S.; Iacono, S.T.; Jones, R.W. Macromol. Symp. 2001, 174, 155.
- 129. Daly, W.H.; Evenson, T.S. Polymer 2000, 41, 5063.
- Studer, A.; Amrein, S.; Scieth, F.; Schulte, T.; Walton, J.C. J. Amer. Chem. Soc. 2003, 125, 5726.
- Enikolopyan, N.S.; Smirnov, B.R.; Ponomarev, G.V.; Belgovskii, I.M. J. Polym. Sci., Polym. Chem. Ed. 1981, 19, 879.
- Burczyk, A.F.; O'Driscoll, K.F.; Rempel, G.L. J. Polym. Sci., Polym. Chem. Ed. 1984, 22, 3255.
- 133. Gridnev, A.A. Polym. Sci. USSR (Engl. Transl.) 1989, 31, 2369.
- Karmilova, L.V.; Ponomarev, G.V.; Smirnov, B.R.; Bel'govskii, I.M. Russ. Chem. Rev. (Engl. Transl.) 1984, 53, 132.
- Davis, T.P.; Haddleton, D.M.; Richards, S.N. J. Macromol. Sci., Rev. Macromol. Chem. Phys. 1994, C34, 243.
- 136. Parshall, G.W.; Ittel, S.D. Homogeneous Catalysis; Wiley: New York, 1992.
- 137. Heuts, J.P.A.; Roberts, G.E.; Biasutti, J.D. Aust. J. Chem. 2002, 55, 381.
- 138. Gridnev, A. J. Polym. Sci. Pol. Chem. 2000, 38, 1753.
- 139. Sanayei, R.A.; O'Driscoll, K.F. J. Macromol. Sci., Chem. 1989, A26, 1137.
- Smirnov, B.R.; Marchenko, A.P.; Plotnikov, V.D.; Kuzayev, A.I.; Yenikolopyan, N.S. Polym. Sci. USSR (Engl. Transl.) 1981, 23, 1169.
- 141. Roberts, G.E.; Barner-Kowollik, C.; Davis, T.P.; Heuts, J.P.A. *Macromolecules* 2003, *36*, 1054.
- 142. Roberts, G.E.; Davis, T.P.; Heuts, J.P.A. J. Polym. Sci., Part A: Polym. Chem. 2003, 41, 752.
- Heuts, J.P.A.; Forster, D.J.; Davis, T.P.; Yamada, B.; Yamazoe, H.; Azukizawa, M. Macromolecules 1999, 32, 2511.
- 144. Roberts, G.E.; Heuts, J.P.A.; Davis, T.P. Macromolecules 2000, 33, 7765.
- 145. Chiefari, J.; Jeffery, J.; Moad, G.; Rizzardo, E.; Thang, S.H. Polym. Prepr. 1999, 40(2), 344.
- 146. Li, Y.; Wayland, B.B. Macromol. Rapid Commun. 2003, 24, 307.
- 147. Wang, W.X.; Stenson, P.A.; Marin-Becerra, A.; McMaster, J.; Schroder, M.; Irvine, D.J.; Freeman, D.; Howdle, S.M. *Macromolecules* 2004, 37, 6667.
- 148. Hawthorne, D.G. US 5324879, 1994 (Chem. Abstr. 1987, 107, 237504).
- 149. Krstina, J.; Moad, C.L.; Moad, G.; Rizzardo, E.; Berge, C.T.; Fryd, M. Macromol. Symp. 1996, 111, 13.
- 150. Gridnev, A.A.; Bel'govskii, I.M.; Enikolopyan, N.S. Dokl. Akad. Nauk SSSR (Engl. Transl.) 1986, 289, 748.
- 151. Tang, L.; Norton, J.R.; Edwards, J.C. Macromolecules 2003, 36, 9716.
- 152. Tang, L.; Norton, J.R. Macromolecules 2004, 37, 241.
- 153. Grognec, E.C.; Claverie, J.; Poli, R. J. Am. Chem. Soc. 2001, 123, 9513.
- 154. Gibson, V.C.; O'Reilly, R.K.; Wass, D.F.; White, A.J.P.; Williams, D.J. Macromolecules 2003, 36, 2591.
- 155. Pierik, S.C.J.; Vollmerhaus, R.; van Herk, A.M. Macromol. Chem. Phys. 2003, 204, 1090.

- 156. Picrik, S.C.J.; Vollmerhaus, R.; van Herk, A.M.; German, A.L. Macromol. Symp. 2002, 182, 43.
- 157. Suddaby, K.G.; O'Driscoll, K.F.; Rudin, A. J. Polym. Sci., Part A: Polym. Chem. 1992, 30, 643.
- 158. Suddaby, K.G.; Maloney, D.R.; Haddleton, D.M. Macromolecules 1997, 30, 702.
- 159. Heuts, J.P.A.; Forster, D.J.; Davis, T.P. Macromolecules 1999, 32, 3907.
- 160. Pierik, S.C.J.; van Herk, A.M. Macromol. Chem. Phys. 2003, 204, 1406.
- 161. Heuts, J.P.A.; Muratore, L.M.; Davis, T.P. Macromol. Chem. Phys. 2000, 201, 2780.
- 162. Mayo, F.R.; Gregg, R.A.; Matheson, M.S. J. Am. Chem. Soc. 1951, 73, 1691.
- 163. Baysal, B.; Tobolsky, A.V. J. Polym. Sci. 1952, 8, 529.
- 164. Mahadevan, V.; Santhappa, M. Makromol. Chem. 1955, 16.
- 165. Das, S.K.; Chatterjee, S.R.; Palit, S.R. Proc. R. Soc., London 1955, A227, 252.
- 166. Stein, D.J. Makromol. Chem. 1964, 76, 170.
- 167. Kuchanov, S.I.; Olenin, A.V. Polym. Sci. USSR (Engl. Transl.) 1973, 15, 2712.
- Starnes, W.H., Jr.; Schilling, F.C.; Plitz, I.M.; Cais, R.E.; Freed, D.J.; Hartless, R.L.; Bovey, F.A. *Macromolecules* 1983, 16, 790.
- 169. Bartlett, P.D.; Altschul, R. J. Am. Chem. Soc. 1945, 67, 816.
- 170. Pryor, W.A.; Coco, J.H. Macromolecules 1970, 3, 500.
- 171. Nozakura, S.-I.; Morishima, Y.; Murahashi, S. J. Polym. Sci., Part A-1 1972, 10, 2853.
- 172. Melville, H.W.; Sewell, P.R. Makromol. Chem. 1959, 32, 139.
- 173. Starnes, W.H., Jr.; Chung, H.; Benedikt, G.M. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem.) 1993, 34(1), 604.
- 174. Litt, M.; Chang, K.H.S. ACS Symp. Ser. 1981, 165, 455.
- 175. Wolf, C.; Burchard, W. Makromol. Chem. 1976, 177, 2519.
- 176. Vidotto, G.; Crosato-Arnaldi, A.; Talamini, G. Makromol. Chem. 1968, 114, 217.
- 177. Rigo, A.; Palma, G.; Talamini, G. Makromol. Chem. 1972, 153, 219.
- 178. Starnes, W.II., Jr.; Schilling, F.C.; Abbas, K.B.; Cais, R.E.; Bovey, F.A. Macromolecules 1979, 12, 556.
- 179. Park, G.S.; Saleem, M. Polym. Bull. 1979, 1, 409.
- 180. Starnes, W.H., Jr.; Wojciechowski, B.J. Makromol. Chem., Macromol. Symp. 1993, 70/71, 1.
- 181. Zubov, V.P.; Kumar, M.V.; Masterova, M.N.; Kabanov, V.A. J. Macromol. Sci., Chem. 1979, A13, 111.
- 182. Butler, G.B. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 4, p 423.
- 183. Bartlett, P.D.; Tate, F.A. J. Am. Chem. Soc. 1953, 75, 91.
- Butler, G.B. Cyclopolymerization and Cyclocopolymerization; Marcel Dekker: New York, 1992.
- 185. Tüzün, N.S.; Aviyente, V.; Houk, K.N. J. Org. Chem. 2002, 67, 5068.
- 186. Vaidya, R.A.; Mathias, L.J. J. Polym. Sci., Polym. Symp. 1986, 74, 243.
- 187. Bevington, J.C.; Melville, H.W.; Taylor, R.P. J. Polym. Sci. 1954, 14, 463.
- 188. Schulz, G.V.; Stein, D.J. Makromol. Chem. 1962, 52, 1.
- 189. Lim, D.; Wichterle, O. J. Polym. Sci. 1958, 29, 579.
- 190. Usami, T.; Takayama, S. Macromolecules 1984, 17, 1756.
- 191. Axelson, D.E.; Levy, G.C.; Mandelkern, L. Macromolecules 1979, 12, 41.
- 192. Bovey, F.A.; Schilling, F.C.; McCrackin, F.L.; Wagner, H.L. Macromolecules 1976, 9, 76.
- 193. Bugada, D.C.; Rudin, A. Eur. Polym. J.: 1987, 23, 809.

- 194. Rudin, A.; Grinshpun, V.; O'Driscoll, K. J. Liq. Chromatog. 1984, 7, 1809.
- 195. Grinshpun, V.; Rudin, A.; Russell, K.E.; Scammell, M.V. J. Polym. Sci., Part B: Polym. Phys. 1986, 24, 1171.
- 196. Pang, S.; Rudin, A. Polym. Mat. Sci. Eng. 1991, 65, 95.
- 197. Bugada, D.C.; Rudin, A. Eur. Polym. J. 1987, 23, 847.
- 198. Martin, J. J. Appl. Polym. Sci. 1990, 40, 1801.
- 199. Schulz, G.V.; Henrici, G.; Olive, S. J. Polym. Sci. 1955, 17, 45.
- 200. Fox, T.G.; Gratch, S. Ann. New York Acad. Sci. 1953, 57, 367.
- 201. Lovell, P.A.; Shah, T.H.; Heatley, F. Polym. Commun. 1991, 32, 98.
- 202. Ahmad, N.M.; Heatley, F.; Lovell, P.A. Macromolecules 1998, 31, 2822.
- 203. Dunn, A.S.; Natavane, S.R. Br. Polym. J. 1980, 75.
- 204. Bugada, D.C.; Rudin, A. Polymer 1984, 25, 1759.
- 205. Bugada, D.C.; Rudin, A. J. Appl. Polvm. Sci. 1985, 30, 4137.
- 206. McDowell, W.H.; Kenyon, W.O. J. Am. Chem. Soc. 1940, 62, 415.
- 207. Nozakura, S.-I.; Morishima, Y.; Murahashi, S. J. Polym. Sci., Part A-1 1972, 10, 2781.
- 208. Adelman, R.L.; Ferguson, R.C. J. Polym. Sci., Polym. Chem. Ed. 1975, 13, 891.
- 209. Ovenall, D.W. Macromolecules 1984, 17, 1458.
- 210. Agarwal, S.H.; Jenkins, R.F.; Porter, R.S. J. Appl. Polym. Sci. 1982, 27, 113.
- 211. Vercauteren, F.F.; Donners, W.A.B. Polymer 1986, 27, 993.
- 212. Hatada, K.; Terawaki, Y.; Kitayama, T.; Kamachi, M.; Tamaki, M. Polym. Bull. 1981, 4, 451.
- 213. Kamachi, M.; Liaw, D.J.; Nozakura, S.-I. Polym. J. 1979, 11, 921.
- 214. Starnes, W.H. Prog. Polym. Sci. 2002, 27, 2133.
- 215. Ovenall, D.W.; Uschold, R.E. Macromolecules 1991, 24, 3235.

This Page is Intentionally Left Blank

7 Copolymerization

7.1 Introduction

Copolymerizations are processes that lead to the formation of polymer chains containing two or more discrete types of monomer unit. Several classes of copolymer that differ in sequence distribution and/or architecture will be considered:

(a) Statistical copolymers are formed when a mixture of two or more monomers is polymerized in a single process and where the arrangement of the monomers within the chains is dictated purely by kinetic factors (Section 7.3).



copolymer, *e.g.* poly(methyl methacrylate-*co*-styrene) random copolymer, *e.g.* poly(methyl methacrylate-*ran*-styrene) statistical copolymer, *e.g.* poly(methyl methacrylate-*stat*-styrene)

(b) Under some conditions (Section 7.3.1.3) the monomer units alternate in the chain. These copolymers are called alternating copolymers.



alternating copolymer, e.g. poly(methyl methacrylate-alt-styrene)
(c) In living polymerizations, compositional drift as monomer is converted during the polymerization process and leads to the formation of gradient or tapered copolymers (Section 9.6).



gradient copolymer, e.g. poly(methyl methacrylate-grad-styrene)

(d) Block or segmented copolymers are usually prepared by multi-step processes (Section 7.5 and Section 9.7). The blocks may be a homopolymer or may themselves be copolymers.





diblock copolymer, e.g. poly(methyl methacrylate)-block-polystyrene



segmented or multiblock copolymer

(e) Graft copolymers and branched (co)polymers are also usually prepared by multi-step processes (Section 7.5). However, they are also formed by copolymerization of macromonomers (Section 7.6.5) and can form as a consequence of intramolecular rearrangement (Section 4.3). The backbone and the pendant chains may be of the same or different composition and may themselves be copolymers.



graft or branched copolymer

(f) Special classes of branched copolymers are star polymers, dendrimers, hyperbranched copolymers and microgels (Section 9,8).

In this chapter, we restrict discussion to approaches based on conventional radical polymerization. Living polymerization processes offer greater scope for controlling polymerization kinetics and the composition and architecture of the resultant polymer. These processes are discussed in Chapter 9.

7.2 Copolymer Depiction

IUPAC recommendations suggest that a copolymer structure, in this case poly(methyl methacrylate-co-styrene) or copoly(methyl methacrylate/styrene), should be represented as **1**. The most substituted carbon of the configurational repeat unit should appear first. This same rule would apply to the copolymer segments shown in Section 7.1. However, as was mentioned in Chapter 1, in this book, because of the focus on mechanism, we have adopted the more traditional depiction **2** which follows more readily from the polymerization mechanism.



7.3 Propagation in Statistical Copolymerization

Statistical copolymers are formed when mixtures of two or more monomers are polymerized by a radical process. Many reviews on the kinetics and mechanism of statistical copolymerization have appeared¹⁻⁹ and some detail can be found in most text books on polymerization. The term 'random copolymer', often used to describe these materials, is generally not appropriate since the incorporation of monomer units is seldom a purely random process. The

arrangement of monomer units in the chains is dictated by the inherent reactivities of the monomers and radicals involved which may, in turn, be influenced by the reaction conditions (solvent, temperature, *etc.*). These factors mean that it is only in special circumstances (Section 7.3.1.1), when monomer reactivities are equal, that there will be a direct correspondence between the copolymer composition and the ratio of monomers in the feed.

In most copolymerizations, the monomers are consumed at different rates dictated by the steric and electronic properties of the reactants. Consequently, both the monomer feed and copolymer composition will drift with conversion. Batch copolymers will generally not be homogeneous in composition at the molecular level. Unfortunately, the detail of the chemical composition of copolymers is not always readily measurable. Many of the traditional techniques only give the average composition (the average ratio of monomers). In living polymerization processes, where ideally all chains grow throughout the polymerization, composition drift is captured within the chain structure. All chains have similar composition and are called gradient or tapered copolymers (Section 9.6).

The detailed microstructure and compositional heterogeneity of copolymers can have a determining influence on copolymer properties. This has been recognized for many years,¹⁰ though the implications are often not fully appreciated. When copolymers with specific properties are required, it is generally not sufficient to control only the average number of functional groups/per polymer molecule.¹¹⁻¹³ It is important to have the functionality distributed in a particular manner along the individual chains (monomer sequence distribution) and amongst the chains (chemical heterogeneity). The microstructure and the degree of heterogeneity can be controlled by designing the monomer feed and/or by selecting the functional monomers according to their inherent reactivity and sometimes by choosing the initiator or transfer agent. The effects of specificity in the initiation and termination steps on the compositional heterogeneity are considered in Section 7.4.5.

Any understanding of the kinetics of copolymerization and the structure of copolymers requires a knowledge of the dependence of the initiation, propagation and termination reactions on the chain composition, the nature of the monomers and radicals, and the polymerization medium. This section is principally concerned with propagation and the effects of monomer reactivity on composition and monomer sequence distribution. The influence of solvent and complexing agents on copolymerization is dealt with in more detail in Section 8.3.1.

Propagation in copolymerization could, in principle, be discussed under the same headings as used for the discussion of propagation in Chapter 4. However, remarkably little information is currently available on the tacticity, extents of head vs tail addition, and propensity for rearrangement in copolymerization.

7.3.1 Propagation Mechanisms in Copolymerization

Studies on radical copolymerization and related model systems have demonstrated that many factors can influence the rate and course of propagation in copolymerization. These include:

- (a) The structure of the propagating species and the likelihood of significant remote unit effects.
- (b) The possibility of complex formation between monomers, between monomer and solvent, *etc*.
- (c) The kinetics and thermodynamics of copolymerization and the possibility that depropagation is competitive with propagation.
- (d) The nature of the medium and the manner in which it changes during the course of the copolymerization.

The various copolymerization models that appear in the literature (terminal, penultimate, complex dissociation, complex participation, *etc.*) should not be considered as alternative descriptions. They are approximations made through necessity to reduce complexity. They should, at best, be considered as a subset of some overall scheme for copolymerization. Any unified theory, if such is possible, would have to take into account all of the factors mentioned above. The models used to describe copolymerization reaction mechanisms are normally chosen to be the simplest possible model capable of explaining a given set of experimental data. They do not necessarily provide, nor are they meant to be, a complete description of the mechanism. Much of the impetus for model development and drive for understanding of the mechanism of copolymerization comes from the need to predict composition and rates. Developments in models have followed the development and application of analytical techniques that demonstrate the inadequacy of an earlier model.

7.3.1.1 Terminal model

The simplest model for describing binary copolymerization of two monomers, M_A and M_B , is the terminal model. The model has been applied to a vast number of systems and, in most cases, appears to give an adequate description of the overall copolymer composition; at least for low conversions. The limitations of the terminal model generally only become obvious when attempting to describe the monomer sequence distribution or the polymerization kinetics. Even though the terminal model does not always provide an accurate description of the copolymerization process, it remains useful for making qualitative predictions, as a starting point for parameter estimation and it is simple to apply.

The terminal model involves a number of approximations:¹

(a) It is assumed that the copolymer composition is dictated by the relative rates of only four propagation reactions (Scheme 7.1). It is implicit in the model that only the last added monomer unit determines reactivity of the propagating

radicals. Note that P_A^{\bullet} and P_B^{\bullet} are propagating species where the terminal (last added) monomer units are M_A and M_B respectively.

| | | | Scheme 7.1 | |
|---------------------|---------------|---------------|------------------|---|
| $P_B \bullet +$ | MB | \rightarrow | P _B • | $k_{\text{pBB}}[P_{\text{B}}\bullet][M_{\text{B}}]$ |
| $P_{B^{\bullet}}$ + | M_{Λ} | \rightarrow | P_A • | $k_{pBA}[P_B\bullet][M_A]$ |
| $P_{A^{\bullet}}$ + | M_B | \rightarrow | P _B ∙ | $k_{\text{pAB}}[\mathbf{P}_{A}\bullet][\mathbf{M}_{B}]$ |
| P _A • + | M_A | \rightarrow | P _A • | $k_{\text{pAA}}[P_{\text{A}}\bullet][M_{\text{A}}]$ |

(b) It is assumed that chains are long and therefore the influence of the initiation and termination steps on the rate of monomer consumption can be neglected. The rates of monomer disappearance can then be written as shown in eqs. 1 and 2.

$$R_{\Lambda} = -\frac{\mathbf{d}[\mathbf{M}_{\Lambda}]}{\mathbf{d}t} = k_{p\Lambda\Lambda}[\mathbf{P}_{\Lambda}\bullet][\mathbf{M}_{\Lambda}] + k_{pB\Lambda}[\mathbf{P}_{B}\bullet][\mathbf{M}_{\Lambda}]$$
(1)

$$R_{\rm B} - -\frac{d[M_{\rm B}]}{dt} = k_{\rm pAB}[P_{\rm A}\bullet][M_{\rm B}] + k_{\rm pBB}[P_{\rm B}\bullet][M_{\rm B}]$$
(2)

The ratio of these equations provides an expression for the instantaneous copolymer composition (eq. 3).

$$\frac{\mathrm{d}[\mathrm{M}_{\mathrm{A}}]}{\mathrm{d}[\mathrm{M}_{\mathrm{B}}]} = \frac{k_{\mathrm{pAA}}[\mathrm{P}_{\mathrm{A}}\bullet][\mathrm{M}_{\mathrm{A}}] + k_{\mathrm{pBA}}[\mathrm{P}_{\mathrm{B}}\bullet][\mathrm{M}_{\mathrm{A}}]}{k_{\mathrm{pAB}}[\mathrm{P}_{\mathrm{A}}\bullet][\mathrm{M}_{\mathrm{B}}] + k_{\mathrm{pBB}}[\mathrm{P}_{\mathrm{B}}\bullet][\mathrm{M}_{\mathrm{B}}]}$$
(3)

(c) A third assumption is that the concentrations of the two propagating species, P_{Λ} • and P_{B} •, achieve a steady state (eq. 4).

$$k_{\text{pAB}}[\mathbf{P}_{A} \bullet][\mathbf{M}_{B}] = k_{\text{pBA}}[\mathbf{P}_{B} \bullet][\mathbf{M}_{A}]$$
(4)

This allows elimination of the radical concentrations from the above equation and the copolymer composition equation (eq. 5), $^{14-16}$ also known as the Mayo-Lewis equation, can now be derived.

$$\frac{F_{\rm A}}{F_{\rm B}} = \frac{\mathrm{d}[\mathrm{M}_{\rm A}]}{\mathrm{d}[\mathrm{M}_{\rm B}]} = \frac{[\mathrm{M}_{\rm A}]}{[\mathrm{M}_{\rm B}]} \left(\frac{r_{\rm AB}[\mathrm{M}_{\rm A}] + [\mathrm{M}_{\rm B}]}{[\mathrm{M}_{\rm A}] + r_{\rm BA}[\mathrm{M}_{\rm B}]} \right) = \frac{f_{\rm A}}{f_{\rm B}} \left(\frac{r_{\rm AB}f_{\rm A} + f_{\rm B}}{f_{\rm A} - r_{\rm BA}f_{\rm B}} \right)$$
(5)

where F_A (= 1- F_B) and f_A (= 1- f_B) are the instantaneous mole fractions of monomer A in the polymer and in the monomer feed respectively, and r_{AB} and r_{BA} are the monomer reactivity ratios which are defined in eqs. 6 and 7. The reactivity ratios, r_{AB} and r_{BA} , are often abbreviated to r_A and r_B . The notation used (r_{AB} and r_{BA}) is preferred since it allows discussion of situations involving more than two monomers (*e.g.* terpolymerization, Section 7.3.2.4).

$$r_{\rm AB} = \frac{k_{\rm pAA}}{k_{\rm pAB}} \tag{6}$$

$$r_{\rm BA} = \frac{k_{\rm pBB}}{k_{\rm pBA}} \tag{7}$$

Other convenient forms of the copolymer composition equation are eq. 8:

$$\frac{F_{\rm A}}{F_{\rm B}} = \frac{1 + r_{\rm AB} x}{1 + r_{\rm BA} / x} \tag{8}$$

where $x = f_A/f_B$ and eq. 9:

$$F_{A} = \frac{r_{AB}f_{A}^{2} + f_{A}f_{B}}{r_{AB}f_{A}^{2} + 2f_{A}f_{B} + r_{BA}f_{B}^{2}} = \frac{(r_{AB} - 1)f_{A}^{2} + f_{A}}{(r_{AB} + r_{BA} - 2)f_{A}^{2} + (2 - r_{BA})f_{A} + r_{BA}}$$
$$= \frac{1 + r_{AB}x}{2 + r_{AB}x + r_{BA}/x}$$
(9)

(d) It is also implicit in this treatment that medium effects are negligible and that there is no participation by monomer-monomer or monomer-solvent complexes.

Table 7.1 Terminal Model Reactivity Ratios for Some Common Monomer Pairs^a

| Monomer B | Monomer A | | | | | | |
|-------------|-------------|------------|-------------|------------|-----------|-------------|-------------|
| | S | MMA | MA | AN | VC | MAII | VAc |
| S | Ν. | 0.51 | 0.77 | 0.40 | 17 | 0.002 | 22 |
| MMA | 0.49 | Ι. | - | 2.0 | - | 5.2 | 27 |
| MA | 0.12 | - | λ | 1.02 | - | 2.8 | 9.0 |
| ΛN | 0.05 | 0.25 | 0.80 | ١ | 3.3 | 6.0 | 5.0 |
| VC | 0.04 | - | - | 0.057 | \ | 0.30 | 1.4 |
| МАН | 0.021 | 0.018 | 0.011 | 0.00 | 0.008 | 1 | 0.003 |
| VAc | 0.02 | 0.03 | | 0.02 | 0.73 | 0.055 | ١ |
| a r. tabula | tod vorting | Ilu e. hou | el zontally | Values (al | on from 1 | aurior at a | 117 or from |

 r_{AB} tabulated vertically, r_{BA} horizontally. Values taken from Laurier *et al.*¹⁷ or from Greenley's compilation.¹⁸ All values rounded to two significant figures.

Thus, the terminal model allows the copolymer composition for a given monomer feed to be predicted from just two parameters; the reactivity ratios r_{AB} and r_{BA} . Some values of terminal model reactivity ratios for common monomer pairs are given in Table 7.1. Values for other monomers can be found in data

compilations.¹⁸ Literature values of reactivity ratios for most monomer pairs can span a considerable range. This can reflect experimental error, uncertain polymerization mechanism and/or inappropriate experimental design. No critical assessment has been made of the data in Table 7.1. Inclusion does not imply that the terminal model adequately describes the system or that the values shown are the best values.



Figure 7.1 Plot of the instantaneous copolymer composition (F_A) vs monomer feed composition (f_A) for the situation where (a) $r_{AB}=r_{BA}=1.0$ (_____), (b) $r_{AA}=r_{BA}=0.5$ (.....), (c) $r_{AB}=r_{BA}=0.01$ (----), (d) $r_{AB}=0.5$, $r_{BA}=2.0$ (....).

It is informative to consider some of the implications of the terminal model and, in particular, how the relative magnitudes of the reactivity ratios affect the copolymer composition (Figure 7.1):

- (a) For the special case where $r_{AB} = r_{BA} = 1.0$, the monomers are utilized according to their respective proportions in the monomer feed. The product is a random copolymer. The value of F_A always equals f_A irrespective of the starting f_A . Copolymerizations of structurally similar monomers come closest to achieving this ideal. Examples are, copolymerizations of isotopically labeled monomers or mixtures of (meth)acrylic esters (with non-bulky ester groups) *e.g.* MMA and BMA).
- (b) For many copolymerizations (e.g. S-MMA, S-AN) $r_{AB} < 1$ and $r_{BA} < 1$. In these cases, because cross-propagation is favored over homopropagation, there is a tendency towards alternation. In the extreme, where the values of both r_{AB} and r_{BA} approach zero (e.g. S-MAH), cross propagation occurs to the virtual exclusion of homopropagation and the product is an alternating copolymer.
- (c) Where $r_{AB}>1$ and $r_{BA}<1$ (or $r_{AB}<1$ and $r_{BA}>1$), the copolymer will always be richer in one monomer than it is in the other. These copolymerizations have no azeotropic composition. Copolymerizations of VAc, NVP and VC with

styrenic and (meth)acrylic monomers are in this class. The special case where the product $r_{AB}r_{BA}$ is unity ($r_{AB}=1/r_{BA}$) has been called ideal because the probabilities of a given monomer adding to the two propagating radicals are identical.¹⁹

(d) The converse situation, where both r_{AB} and r_{BA} are greater than one, is very rarely encountered. In this case, homopropagation is always favored over cross-propagation and, as a consequence, there will be a degree of blockiness in the copolymer.

In cases where $r_{AB}>1$ and $r_{BA}>1$ or $r_{AB}<1$ and $r_{BA}<1$, there will always be exactly one 'azeotropic composition' or 'critical point' where the copolymer composition will exactly reflect the monomer feed composition (Figure 7.1).

i.e.
$$\frac{d[\mathbf{M}_A]}{d[\mathbf{M}_B]} = \frac{[\mathbf{M}_A]}{[\mathbf{M}_B]} = x$$
 or $F_A = f_A$

Substitution into the copolymer composition equation (eq. 8) shows that this condition is satisfied when:

$$x = \frac{1 - r_{AB}}{1 - r_{BA}}$$

... ..

The existence of an azeotropic composition has some practical significance. By conducting a polymerization with the monomer feed ratio equal to the azeotropic composition, a high conversion batch copolymer can be prepared that has no compositional heterogeneity caused by drift in copolymer composition with conversion. Thus, the complex incremental addition protocols that are otherwise required to achieve this end, are unnecessary. Composition equations and conditions for azeotropic compositions in ternary and quaternary copolymerizations have also been defined.^{20,21}

The overall rate of propagation in copolymerization is given by eq. 10.

$$R_{p} = -\frac{d[M]}{dt} = \bar{k}_{p}[P\bullet][M]$$
$$= k_{pAA}[P_{A}\bullet][M_{A}] + k_{pBA}[P_{B}\bullet][M_{A}] + k_{pAB}[P_{A}\bullet][M_{B}] + k_{pBB}[P_{B}\bullet][M_{B}] \quad (10)$$

where [M] (=[M_A]+[M_B]) is the total monomer concentration and [P•] (-[P_A•]+[P_B•]) is the total concentration of propagating radicals.

An expression (eq. 11) for the overall rate constant for propagation in copolymerization (k_p) can now be formulated.

$$\bar{k}_{p} = \frac{r_{AB}f_{A}^{2} + f_{A}f_{B}^{2} + 2f_{A}f_{B}}{r_{AB}f_{A}/k_{pAA} + r_{BA}f_{B}/k_{pBB}}$$
(11)

Note that value of k_p is usually not constant with conversion since it depends on the monomer feed composition.

7.3.1.2 Penultimate model

The general features of the penultimate model in what have become known as the explicit and implicit forms are described in Section 7.3.1.2.1. Evidence for remote unit effects coming from small molecule radical chemistry and experiments other than copolymerization is discussed in Section 7.3.1.2.2. In Sections 7.3.1.2.3 and 7.3.1.2.4 specific copolymerizations are discussed. Finally, in Section 7.3.1.2.5, we consider the origin of the penultimate unit effects. A general recommendation is that when trying to decide on the mechanism of a copolymerization, first consider the explicit penultimate model.²

7.3.1.2.1 Model description

The influence of penultimate units on the kinetics of copolymerization and the composition of copolymers was first considered in a formal way by Merz *et al.*²² and Ham.⁸ They consider eight propagation reactions (Scheme 7.2).

| $P_{AA} +$ | M_A | \rightarrow | \mathbf{P}_{AA} • | $k_{\text{pAAA}}[P_{\text{AA}}\bullet][M_{\text{A}}]$ |
|--------------------|-------------|---------------|------------------------------|---|
| $P_{AA} \bullet +$ | M_{B} | \rightarrow | \mathbf{P}_{AB} • | $k_{\text{pAAB}}[P_{AA}\bullet][M_B]$ |
| $P_{AB} \bullet +$ | M_A | \rightarrow | P _{BA} ∙ | $k_{\text{pABA}}[P_{\text{AB}}\bullet][M_{\text{A}}]$ |
| $P_{AB} \bullet +$ | $M_{\rm B}$ | \rightarrow | $\mathbf{P}_{\mathbf{BB}}$ • | $k_{pABB}[P_{AB}^{\bullet}][M_B]$ |
| $P_{BA} +$ | M_A | \rightarrow | P_{AA} • | $k_{pBAA}[\mathbf{B}_{BA}\bullet][\mathbf{M}_{A}]$ |
| $P_{BA} \bullet +$ | M_{B} | \rightarrow | P _{AB} ∙ | $k_{pBAB}[P_{BA}\bullet][M_B]$ |
| $P_{BB}\bullet$ + | M_A | \rightarrow | P _{BA} ∙ | $k_{pBBA}[P_{BB}\bullet][M_A]$ |
| $P_{BB} \bullet +$ | M_{B} | \rightarrow | P _{BB} • | $k_{pBBB}[P_{BB}\bullet][M_B]$ |
| | | | Scheme 7.2 | |

From this scheme it can be seen that the copolymer composition is determined by the values of four monomer reactivity ratios.

$$r_{AAB} = \frac{k_{pAAA}}{k_{pAAB}} \qquad r_{BAB} = \frac{k_{pBAA}}{k_{pBAB}} \qquad r_{ABA} = \frac{k_{pABB}}{k_{pABA}} \qquad r_{BBA} = \frac{k_{pBBB}}{k_{pBBA}}$$

Fukuda *et al.*²³ were the first to recognize that a further two radical reactivity ratios were required to completely define the polymerization kinetics.

$$s_{\rm A} = \frac{k_{\rm pAAA}}{k_{\rm pBAA}}$$
 $s_{\rm B} = \frac{k_{\rm pBBB}}{k_{\rm pABB}}$

^{*} The reactivity ratios r_{AAB} , r_{BAB} , r_{BBA} and r_{ABA} are sometimes abbreviated to r_{AA} , r_{BA} , r_{BA} and r_{AB} or to r_A , r_A^{-1} , r_B , r_B^{-1} respectively. The notation used (r_{AAB} , r_{BAB} , r_{BBA} and r_{ABA}) is preferred since it allows discussion of situations involving more than two monomers.

In traditional treatments of copolymerization kinetics, the values of the ratios s_A and s_B are implicitly set equal to unity (Section 7.3.1.2.2). Since they contain no terms from cross propagation, these parameters have no direct influence on either the overall copolymer composition or the monomer sequence distribution; they only influence the rate of polymerization.

The instantaneous copolymer composition is described by the following equation (eq. 12):

$$\frac{F_{A}}{F_{B}} = \frac{1 + \frac{r_{BAB}x(1 + r_{AAB}x)}{1 + r_{BAB}x}}{1 + \frac{r_{ABA}(r_{BBA} + x)}{x(r_{ABA} + x)}}$$
(12)

By substituting $\bar{r}_{AB} = r_{BAB} \frac{1 + r_{AAB}x}{1 + r_{BAB}x}$ and $\bar{r}_{BA} = r_{ABA} \frac{x + r_{BBA}}{x + r_{ABA}x}$

eq. 12 may be written in a form similar to the terminal model copolymer composition equation (eq. 8) as eq. 13.

$$\frac{F_{\rm A}}{F_{\rm B}} = \frac{1 + F_{\rm AB} x}{1 + \bar{F}_{\rm BA} / x} \tag{13}$$

Cases have been reported where the application of the penultimate model provides a significantly better fit to experimental composition or monomer sequence distribution data. In these copolymerizations $r_{AAB} \neq r_{BAB}$ and/or $r_{ABA} \neq r_{BBA}$. These include many copolymerizations of AN,²⁴⁻²⁶ B,²⁷ MAH^{28,29} and VC.³⁰ In these cases, there is no doubt that the penultimate model (or some scheme other than the terminal model) is required. These systems are said to show an explicit penultimate effect. In binary copolymerizations where the explicit penultimate model applies there may be between zero and three azeotropic compositions depending on the values of the reactivity ratios.³¹

It is possible to define average propagation rate constants for copolymerization subject to a penultimate group effect as follows.

$$\overline{k}_{pAA} = k_{pAAA} \frac{1 + r_{AAB}x}{r_{AAB}x + 1/s_B}$$
 and $\overline{k}_{pBB} = k_{pBBB} \frac{x + r_{BBA}}{r_{BBA} + x/s_A}$

Note that the values of \bar{r}_{AB} , \bar{r}_{BA} , \bar{k}_{PAA} , and \bar{k}_{PBB} are dependent on the monomer feed composition and hence on conversion. These parameters may be substituted for r_{AB} , r_{BA} , k_{PAA} and k_{PBB} in eq. 11 to provide an expression for the overall rates of propagation (eq. 14) and of polymerization (eq.15).

$$\bar{k}_{p} = \frac{\bar{r}_{AB}f_{A}^{2} + f_{A}f_{B}^{2} + 2f_{A}f_{B}}{\bar{r}_{AB}f_{A}/\bar{k}_{pAA} + \bar{r}_{BA}f_{B}/\bar{k}_{pBB}}$$
(14)

$$R_{\rm p} = -\frac{d[\mathbf{M}]}{dt} = \bar{k}_{\rm p}[\mathbf{P}\bullet][\mathbf{M}] = \frac{\bar{r}_{\rm AB}f_{\rm A}^2 + f_{\rm A}f_{\rm B}^2 + 2f_{\rm A}f_{\rm B}}{\bar{r}_{\rm AB}f_{\rm A}/\bar{R}_{\rm pA} + \bar{r}_{\rm BA}f_{\rm B}/\bar{R}_{\rm pB}}$$
(15)

where $\overline{R}_{pA} = R_{pA} \frac{1 + r_{AAB}x}{r_{AAB}x + 1/s_B}$

For many systems, the copolymer composition appears to be adequately described by the terminal model yet the polymerization kinetics demand application of the penultimate model. These systems where $r_{AAB}=r_{BAB}$ and $r_{ABA}=r_{BBA}$ but $s_A\neq s_B$ are said to show an implicit penultimate effect. The most famous system of this class is MMA-S copolymerization (Section 7.3.1.2.3).

Penpenultimate and higher order remote unit effect models may also affect the outcome of copolymerizations. However, in most cases, experimental data, that are not sufficiently powerful to test the penultimate model, offer little hope of testing higher order models. The importance of remote unit effects on copolymerization will only be fully resolved when more powerful analytical techniques become available.

7.3.1.2.2 Remote substituent effects on radical addition



Figure 7.2 Chain end terminology.

In small molecule chemistry it is well established that β - and more remote substituents (Figure 7.2) can have a substantial influence on radical conformation, formation and reactivity. Thus, it should be anticipated that the nature of the penultimate unit of the propagating chain could significantly modify its reactivity towards monomers and other species. However, the magnitude of the effect will be dependent on the exact nature of the remote substituent and the reactants. It is

344

also important to remember that, in copolymerization, we consider the effect of the penultimate unit on a rate constant ratio, not on the rate constant for a particular reaction.

Experimental studies on models of the propagating radicals in S-AN copolymerization^{32,33} and a few other systems³⁴ provide support for an explicit penultimate unit effect. Of particular interest is the data of Tirrell and coworkers. They investigated the relative reactivity of S and AN towards various γ -substituted propyl radicals (Scheme 7.3 and Table 7.2). They found that:

- (a) There is only a small effect on radical reactivity when the γ -substituent is a styryl unit (at a PSAN chain end), a phenyl, or an alkyl group.
- (b) An electrophilic γ -cyano substituent has a marked effect on radical reactivity.
- (c) The relative reactivities of simple model radicals correlate well with the reactivities of propagating species estimated from copolymerization data assuming an explicit penultimate model.



Scheme 7.3

Table 7.2 Relative Rates for Addition of Substituted Propyl Radicals $(R^3R^2CHCH_2CHR^1 \cdot)$ to AN and S at 100 °C^a

| B ¹ | R ² | R3 | $\frac{k_{AN}}{k_{S}}$ |
|----------------|-------------------|-----------------|------------------------|
| Н | Н | CH ₃ | 24.5 ± 1.1^{32} |
| Н | Н | C_3H_7 | 26.3 ± 2.4^{32} |
| Н | Н | Ph | 22.6 ± 2.0^{32} |
| Ph | Н | Н | 5.0 ^c |
| Ph | Н | Ph | 4.8 ± 0.3^{33} |
| Ph | PSAN ⁶ | Ph | $4.2^{d.25}$ |
| Н | Н | CN | 6.8 ± 0.6^{32} |
| Ph | Н | CN | $1.9{\pm}0.1^{33}$ |
| Ph | PSAN" | CN | $1.7^{d,25}$ |

a refer Scheme 7.3. b Poly(AN-co-S) chain. c Value for 1-phenylethyl radical from Table 3.6 Section 3.4.1.1. Value from penultimate model reactivity ratios for AN-S copolymerization at 60 °C.²⁵

Other experimental data seem to provide support for an implicit penultimate model. Thus, simple (monomeric) model radicals for the propagating radical chain

ends in S-MMA copolymerization show similar (though not the same) chemospecificity to the corresponding propagating radicals in radical addition (Table 7.3). However, rate constants for addition appear more than an order of magnitude higher for the lower molecular weight species. There are many other examples of this type. Additional data on the rate constants for the reactions of small radicals with monomers can be found in Section 3.4.

| R ¹ | R ² | B^3 | $\frac{k_{\rm S}}{{ m M}^{-1}{ m s}^{-1}}$ | $\frac{k_{\text{MMA}}}{k_{\text{s}}}$ |
|---------------------------------|--------------------|--------|--|---------------------------------------|
| Ph | Н | Н | 1100 35 | 1.935 |
| Ph | PSMMA ^a | н | 93 ^b | 2.2° |
| CO ₂ CH ₃ | CH _a | CH_3 | 6030 35 | 0.6135 |
| CO_2CH_3 | PSMMA ^a | н | 180 ^b | 0.52 ^c |

Table 7.3 Relative Rates for Addition of Substituted Methyl Radicals (R³R²R¹C•)to MMA and S at ~25 °C

a Poly(MMA-*co*-S) chain. b based on k_p for homopropagation. c Value from terminal model reactivity ratios for MMA-S copolymerization at 25 °C.²³

Further examples of significant penultimate unit effects come from studies of rate constants for addition of the first propagating species to monomer (Scheme 7.4). There is a strong dependence on the particular initiating species. The data in Table 7.4 were provided in Fischer and Radom's review.³⁵

$$\begin{array}{c} R^{4} & R^{1} \\ R^{3} - C - CH_{2} - C \\ R^{2} & CO_{2}CH_{3} \end{array} \xrightarrow{R^{1}} CH_{2} = C \xrightarrow{k_{p1}} \\ CO_{2}CH_{3} \end{array}$$

Scheme 7.4

It is known that the penultimate unit influences the conformation of both model radicals and propagating radicals.³⁵⁻³⁸ Since addition requires a particular geometric arrangement of the reactants, there are enthalpic barriers to overcome for addition to take place and also potentially significant effects on the entropy of activation. Comparisons of the rate constants and activation parameters for homopropagation with those for addition of simple model radicals to the same monomers also provide evidence for significant penultimate unit effects (Section 4.5.4).

There is also clear evidence that penultimate group effects are important in determining the stereochemistry of addition in many homopolymerizations and copolymerizations. This is made evident from the fact that most homopolymers have tacticity (*i.e.* $P(m) \neq 0.5$, Section 4.2). Indeed, for some homopolymerizations there is evidence that the configuration of the penpenultimate unit may also influence the stereochemistry of addition.³⁹ If penpen- and penultimate units

influence the stereospecificity of addition, it is also reasonable to expect that they might affect the rate and chemospecificity of addition.

 \mathbb{R}^3 R^4 R^2 R k_{p1} $\underline{k_{(rel)}}$ $k_{(rel)}$ н PMACH₂^{*} CO₂CH₃ н 19000 3.1 Н CH_3 CHa CH_3 6120 1.0 Н CH_3 CH_3 ÔН 3290 0.54 Н н н Ph 22400 37. _ н н н OH 18110 3.0 CH_3 PMMACH₂^a CO₂CH₃ н 345 0.58 _ CH_3 CH_3 CH_3 CH_3 600 1.0 CH_3 CH_3 CH_3 OH 1205 2.0 CH_3 н Н Ph 4.4 2640 CH_3 н Н OH 3290 5.5

 Table 7.4 Rate Constants (298 K) for Addition of Substituted Propyl Radicals to (Meth)acrylate Esters (Scheme 7.4)³⁵

a Long chain propagating radical.

Penultimate unit effects are also important in both substitution^{40,41} and in addition-fragmentation chain transfer.^{42.44} Some examples are provided in Sections 6.2, 6.2.2.4, 6.2.3.4 and 9.5.

Based on the above data, it would seem unusual if reactivity of the propagating species in copolymerization were insensitive to the nature of the last added monomer units. However, while there are ample experimental data to suggest that copolymerizations should be subject to penultimate unit effects that affect the rate and/or copolymer composition, the origin and magnitude of the effect is not always easily predictable.

7.3.1.2.3 MMA-S copolymerization

MMA-S copolymerization has been investigated by many groups.^{23,45-49} Fukuda *et al.*²³ followed established procedure to confirm that the overall composition of MMA-S copolymers was satisfactorily predicted by the terminal model with $r_{AAB}-r_{BAB}-0.52$ and $r_{BBA}-r_{ABA}-0.46$. They applied the rotating sector method to determine absolute values of the overall propagation and termination rate constants. The data showed that the observed dependence of the rate of copolymerization on monomer feed composition, which had previously been attributed to an effect of the kinetics of termination, was in fact due to a composition dependence of the overall propagation rate constant. Fukuda *et al.*²³ proposed an explanation in terms of an implicit penultimate unit effect. Values of the radical reactivity ratios s_A (=0.52) and s_B (=0.30) were estimated which accounted for the data. Determinations of propagation rate constants using PLP, while suggesting slightly different values of s_A and s_B (Table 7.5), confirm the basic result.^{45,46,50,51}

| M_A | M_2 | r _{AB} | $r_{\rm BA}$ | ŕ _{AB} ŕ _{BA} | \$ _A | s _B | s _A s _B | Temp. °C | ref. |
|-------|--------------|-----------------|--------------|---------------------------------|-----------------|----------------|-------------------------------|-------------|------|
| MMA | S | 0.46 | 0.52 | 0.24 | 0.52 | 0.30 | 0.16 | 40 | 23 |
| MMA | S | (0.46) | (0.52) | 0.24 | 0.65 | 0.37 | 0.24^{a} | 40 | 52 |
| MMA | S | (0.46) | (0.52) | 0.24 | 0.80 | 0.30 | 0.24^{a} | 25 | 52 |
| MMA | \mathbf{S} | (0.46) | (0.52) | 0.24 | 0.47 | 0.18 | 0.08 | 25 | 50 |
| EMA | S | 0.35 | 0.62 | 0.21 | 0.21 | 0.62 | 0.13 | 25 | 53 |
| BMA | S | 0.45 | 0.72 | 0.32 | 0.63 | 0.56 | 0.35 | 25 | 53 |
| LMA | S | 0.45 | 0.57 | 0.26 | 0.33 | 0.59 | 0.19 | 25 | 53 |
| MA | S | 0.19 | 0.73 | 0.14 | 0.26 | 1.10 | 0.29 | 25 | 54 |
| MOS | S | 0.82 | 1.12 | 0.92 | 1.0 | 1.0 | 1.0 | 25 | 55 |
| MMA | MOS | 0.29 | 0.32 | 0.09 | 0.60 | 0.36 | 0.22 | 25 | 55 |

 Table 7.5. Implicit Penultimate Model Reactivity Ratios

a Assuming that $r_{AB}r_{BA} = s_A s_B$.

If the terminal model adequately explains the copolymer composition, as is often the case, the terminal model is usually assumed to apply. Even where statistical tests show that the penultimate model does not provide a significantly better fit to experimental data than the terminal model, this should not be construed as evidence that penultimate unit effects are unimportant.⁴⁹ It is necessary to test for model discrimination, rather than merely for fit to a given model. In this context, it is important to remember that composition data are of very low power when it comes to model discrimination. For MMA-S copolymerization, even though experimental precision is high, the penultimate model confidence intervals are quite large; $0.4 < r_{AAB}/r_{BAB} < 2.7$, $0.3 < r_{BBA}/r_{ABA} < 2.2$.⁴⁹ The terminal model ($r_{AAB}=r_{BAB}$, $r_{BBA}=r_{ABA}$) is only one of a number of possible solutions and the experimental composition data do not rule out the possibility of quite substantial penultimate unit effects. The same point was made more recently by Kaim.⁴⁷

Triad information is more powerful, but typically is subject to more experimental error and signal assignments are often ambiguous (Section 7.3.3.2). Triad data for the MMA-S system are consistent with the terminal model and support the view that any penultimate unit effects on specificity are small.⁵⁶⁻⁵⁸

Further evidence that penultimate unit effects are small in the MMA-S system comes from comparing the reactivities of small model radicals with the reactivity ratios (Section 7.3.1.2.2 and Table 7.4).

7.3.1.2.4 Other copolymerizations

The kinetics of many copolymerizations have now been examined with absolute (overall) propagation rate constants being determined by the rotating sector, PLP or ESR methods. A similar situation as pertains for the MMA-S

system applies in many cases. The terminal model appears to adequately describe copolymer composition but the kinetic data require a penultimate or more complex model. A summary of some recent data to which the implicit terminal model has been applied is provided in Table 7.5.

The values of s_A and s_B are not well defined by kinetic data.⁵⁹⁻⁶¹ The wide variation in s_A and s_B for MMA-S copolymerization shown in Table 7.5 reflects the large uncertainties associated with these values, rather than differences in the rate data for the various experiments. Partly in response to this, various simplifications to the implicit penultimate model have been used (*e.g.* $r_{AB}r_{BA}=s_As_B^{52}$ and $s_A=s_B$). These problems also prevent trends in the values with monomer structure from being established.

It has been pointed out that analysis of terpolymerization data or copolymerization with chain transfer could, in principle, provide a test of the model.^{2,3} However, to date experimental uncertainty has prevented this.

7.3.1.2.5 Origin of penultimate unit effects

Some theoretical justifications for the prevalence of systems which show an implicit penultimate effect have appeared. These are summarized in the recent reviews by Coote and Davis.^{2,3}

 $P_{ij} \bullet + M_k \rightarrow P_{jk} \bullet k_{pijk} [P_{ij} \bullet] [M_k]$

Scheme 7.5

Fukuda *et al.*^{9.62} have argued that, in most copolymerizations, penultimate substituents should mainly influence the enthalpy for addition to monomer. It was proposed that enthalpy change $(-\Delta H_{ijk})$ is given by the eq. 16 (refer Scheme 7.5) which contains a constant term (ΔH_o) and the 'stabilization energies' of the product propagating radical (U_{ijk}) , the reactant propagating radical (U_{ij}) and the monomer (U_k) .

$$-\Delta H_{ijk} = -\Delta H_0 + U_{jk} - (U_{ij} + U_k) \tag{16}$$

If the Evans-Polyani rule (Section 2.4.1)⁶³ applies, the activation energy E_{ijk} will be proportional to the reaction enthalpy $(-\Delta H_{ijk})$ and eq. 17 will hold.

$$E_{ijk} = \beta + \alpha (-\Delta H_{ijk}) = \beta + \alpha [-\Delta H_0 + U_{jk} - (U_{ij} + U_k)]$$
(17)

where β and α are constants.

If it is assumed that penultimate unit effects on the reaction entropy are insignificant, the terms in eqs. 18 and 19 corresponding to the stabilization energy of the reactant propagating radical will cancel and $r_{iij}=r_{jij}$. There should be no explicit penultimate unit effect on copolymer composition. On the other hand, the radical reactivity ratio s_i (eq. 20) compares two different propagating radicals so

there is no cancellation of the penultimate unit effect. On the basis of this argument, the penultimate unit effect is expected to be implicit.

$$r_{\rm iij} = \frac{k_{\rm piii}}{k_{\rm piij}} = \frac{A_{\rm piii}}{A_{\rm piij}} e^{E_{\rm poi} - E_{\rm piij}}$$
(18)

$$r_{\rm jij} = \frac{k_{\rm pjii}}{k_{\rm pjij}} = \frac{A_{\rm pjii}}{A_{\rm pjij}} e^{E_{\rm sjii} - E_{\rm pjij}}$$
(19)

$$s_{i} = \frac{k_{\text{pjii}}}{k_{\text{piii}}} = \frac{A_{\text{pjii}}}{A_{\text{piii}}} e^{E_{\text{pjii}} - E_{\text{pjii}}}$$
(20)

It also follows from this treatment that

$$r_{\rm AB}r_{\rm BA} = s_{\rm A}s_{\rm B} \tag{21}$$

However, plots which would demonstrate this relationship show considerable scatter.⁴⁴

The above argument is also at odds with the conventional wisdom that the well-known tendency for monomer alternation in copolymerization can primarily be attributed to polar factors. It was suggested⁹ that, in most cases, radical stabilization could provide an alternate explanation. A discussion on the relative importance of steric polar and radical stabilization effects on radical addition appears in Section 2.3.

It has been argued that for a majority of copolymerizations, composition data can be adequately predicted by the terminal model copolymer composition equation (eqs. 5-9). However, in that composition data are not particularly good for model discrimination, any conclusion regarding the widespread applicability of the implicit penultimate model on this basis is premature.

Heuts *et al.*,⁶⁴ while not disputing that penultimate units might influence the activation energies, proposed on the basis of theoretical calculations that penultimate unit effects of the magnitude seen in the S-AN and other systems (*i.e.* 2-5 fold) can also be explained by variations in the entropy of activation for the process. They also proposed that this effect would mainly influence rate rather than specificity.

7.3.1.3 Models involving monomer complexes

Mechanisms for copolymerization involving complexes between the monomers were first proposed to explain the high degree of alternation observed in some copolymerizations. They have also been put forward, usually as alternatives to the penultimate model, to explain anomalous (not consistent with the terminal model) composition data in certain copolymerizations.⁶⁵⁻⁷⁴

While there is clear evidence for complex formation between certain electron donor and electron acceptor monomers, the evidence for participation of such complexes in copolymerization is often less compelling. One of the most studied systems is S-MAH copolymerization.^{28,75} However, the models have been applied to many copolymerizations of donor-acceptor pairs. Acceptor monomers have substituents such as carboxy, anhydride, ester, amide, imide or nitrile on the double bond. Donor monomers have substituents such as alkyl, vinyl, aryl, ether, sulfide and silane. A partial list of donor and acceptor monomers is provided in Table 7.6.⁶⁵:

Common features of polymerizations involving such monomer pairs are:

- (a) A high degree of monomer alternation in the chain is observed.
- (b) The copolymer composition cannot be rationalized on the basis of the terminal model (Section 7.3.1.1).
- (c) The rate of copolymerization is usually very much faster than that of either homopolymerization.
- (d) Many of the monomers do not readily undergo homopolymerization or copolymerization with monomers of like polarity.
- (e) For most systems there is spectroscopic evidence for some form of donoracceptor interaction.

| Donors | Acceptors |
|--|---|
| dienes (e.g. B, isoprene) | MAA, itaconic acid |
| heterocyclic dienes (e.g. furan, indole, | (meth)acrylate esters (e.g. MA and MMA) |
| thiophene) | |
| vinylbenzene and derivatives (e.g. S, | cinnamate esters |
| AMS) | |
| vinyl heteroaromatics (e.g. vinyl | methacrylamides |
| pyridine, vinyl carbazole) | |
| vinyl esters (e.g. VAc) | cyanoethylenes (e.g. AN and |
| | 1,1-dicyanoethylene) |
| vinyl ethers (e.g. ethyl vinyl ether) | maleate, fumarate esters |
| vinyl sulfides | MAII, eitraconic anhydride |
| vinyl halides | maleimides (e.g. N-phenylmaleimide) |

Table 7.6 List of Donor and Acceptor Monomers

However, these observations are not proof of the role of a donor-acceptor complex in the copolymerization mechanism. Even with the availability of sequence information it is often not possible to discriminate between the complex model, the penultimate model (Section 7.3.1.2) and other, higher order, models.²⁸ A further problem in analyzing the kinetics of these copolymerizations is that many donor-acceptor systems also give spontaneous initiation (Section 3.3.6.3).

Equilibrium constants for complex formation (K) have been measured for many donor-acceptor pairs. Donor-acceptor interaction can lead to formation of highly colored charge-transfer complexes and the appearance of new absorption bands in the UV-visible spectrum may be observed. More often spectroscopic evidence for complex formation takes the form of small chemical shift differences in NMR spectra or shifts in the positions of the UV absorption maxima. In analyzing these systems it is important to take into account that some solvents might also interact with donor or acceptor monomers.

Since intermediates usually cannot be observed directly, the exact nature of the donor-acceptor complex and the mechanisms for their interaction with radicals are speculative. At least three ways may be envisaged whereby complex formation may affect the course of polymerization:

(a) The complex participation model.⁷⁵⁻⁷⁷ A binary complex is formed that is much more reactive than either of the non-complexed monomers. The monomers are incorporated into the chain in pairs (Scheme 7.6). If reaction with the complexed monomer competes with addition to uncomplexed monomer, the mechanism may be described in terms of six reactivity ratios and one equilibrium constant.

| P _A ∙ + | M _A | \rightarrow | P _A • | $k_{pAA}[P_A\bullet][M_A]$ |
|-----------------------------|----------------------------|---------------|-------------------|--|
| $P_{\Lambda^{\bullet}} \ +$ | M_{B} | \rightarrow | P _B • | $k_{\text{pAB}}[\mathbf{P}_{A}\bullet][\mathbf{M}_{B}]$ |
| $P_B\bullet \ +$ | M_A | \rightarrow | P _A • | $k_{pBA}[P_B\bullet][M_A]$ |
| $P_{B^{\bullet}}$ + | M_B | \rightarrow | P_{B}^{\bullet} | $k_{pBB}[P_B \bullet][M_B]$ |
| $P_{\Lambda^{\bullet}}$ = | $M_{\text{B}}M_{\text{A}}$ | \rightarrow | P_{Λ} • | $k_{\text{pABA}}[P_{\Lambda}\bullet][M_{\Lambda}M_{B}]$ |
| $P_{A} \bullet +$ | $M_A M_B$ | \rightarrow | P _B • | $k_{\text{pAAB}}[P_A\bullet][M_AM_B]$ |
| $P_{B}\bullet \pm$ | $M_{\rm B}M_{\rm A}$ | \rightarrow | $P_A \bullet$ | $k_{\text{pBBA}}[P_{B}\bullet][M_{A}M_{B}]$ |
| $P_{B} \bullet +$ | $M_A M_B$ | \rightarrow | $P_B \bullet$ | $k_{\text{pBAB}}[P_{\text{B}}\bullet][M_{\text{A}}M_{\text{B}}]$ |
| M_A + | M_{B} | - | $M_A M_B$ | |

Scheme 7.6

- (b) The complex dissociation model.⁷⁸⁻⁸⁰ A binary complex is formed that is much more reactive than either of the non-complexed monomers. The complex dissociates after addition and only a single monomer unit is incorporated on reaction with the complex (Scheme 7.7).
- (c) Formation of a less reactive complex. This could have the effect of reducing the overall monomer concentration and perhaps altering the ratio of reactive monomers in the feed. However, the fraction of monomer complexed is typically small.

| $P_A \bullet$ | + | M _A | \rightarrow | $P_A \bullet$ | | $k_{\text{pAA}}[P_{\text{A}}\bullet][M_{\text{A}}]$ |
|----------------------------------|---|---------------------------|---------------|----------------------|-------------------|--|
| $P_A\bullet$ | + | M_{B} | \rightarrow | P _B ∙ | | $k_{\text{pAB}}[\mathbf{P}_{A}\bullet][\mathbf{M}_{B}]$ |
| $P_B \bullet$ | ÷ | M _A | \rightarrow | $P_A\bullet$ | | $k_{pBA}[P_B^{\bullet}][M_A]$ |
| $P_{\mathrm{B}}\bullet$ | + | M_{B} | \rightarrow | P _B ∙ | | $k_{\text{pBB}}[P_{\text{B}}\bullet][M_{\text{B}}]$ |
| $P_{\Lambda}\bullet$ | + | $M_{\text{B}}M_{\Lambda}$ | -> | $P_A \bullet + N$ | И _в | $k_{\text{pABA}}[P_A\bullet][M_AM_B]$ |
| $P_A \bullet$ | ł | $M_A M_B$ | \rightarrow | P _B • ⊢ N | \mathcal{A}_{A} | $k_{\text{pAAB}}[P_{\text{A}}\bullet][M_{\text{A}}M_{\text{B}}]$ |
| $\mathrm{P}_{\mathrm{B}}\bullet$ | + | M_BM_A | \rightarrow | P _A • + N | И _В | $k_{\text{pBBA}}[P_{B}\bullet][M_{A}M_{B}]$ |
| P_B^\bullet | + | $M_A M_B$ | \rightarrow | $P_B \bullet + N$ | Л _л | $k_{\rm pBAB}[P_{\rm B}\bullet][M_{\rm A}M_{\rm B}]$ |
| $M_{\rm A}$ | + | M_B | ÷ | $M_{\rm A}M_{\rm B}$ | | |
| | | | | | | |

Scheme 7.7

Several studies on the reactivities of small radicals with donor-acceptor monomer pairs have been carried out to provide insight into the mechanism of copolymerizations of donor-acceptor pairs. Tirrell and coworkers⁸¹⁻⁸³ reported on the reaction of *n*-butyl radicals with mixtures of N-phenylmaleimide and various donor monomers (e.g. S, 2-chloroethyl vinyl ether). Jenkins and coworkers⁸⁴ have examined the reaction of t-butoxy radicals with mixtures of AN and VAc. Both groups have examined the S-AN system (see also Section 7.3.1.2). In each of these donor-acceptor systems only simple (one monomer) adducts are observed. Incorporation of monomers as pairs is not an important pathway (*i.e.* the complex participation model is not applicable). Furthermore, the product mixtures can be predicted on the basis of what is observed in single monomer experiments. The reactivity of the individual monomers (towards initiating radicals) is unaffected by the presence of the other monomer (*i.e.* the complex dissociation model is not applicable). Unless propagating species are shown to behave differently, these results suggest that neither the complex participation nor complex dissociation models apply in these systems.

7.3.1.4 Copolymerization with depropagation

Propagation reactions in radical polymerization and copolymerization are generally highly exothermic and can be assumed to be irreversible. Exceptions to this general rule are those involving monomers with low ceiling temperatures (Section 4.5.1). The thermodynamics of copolymerization has been reviewed by Sawada.⁸⁵

Some of the most important systems known to involve reversible propagation steps are:

- (a) Copolymerizations of AMS. Studies on copolymerizations of AMS with AN,^{86,87} BA,⁸⁸ MMA^{87,89-94} and S^{86,95} have been reported.
- (b) Copolymerizations with sulfur dioxide and earbon monoxide.85

Copolymerizations of other monomers may also be subject to similar effects given sufficiently high reaction temperatures (at or near their ceiling temperatures - Section 4.5.1). The depropagation of methacrylate esters becomes measurable at temperatures >100 °C (Section 4.5.1).⁹⁶ O'Driscoll and Gasparro⁸⁶ have reported on the copolymerization of MMA with S at 250 °C.

The analysis of these systems requires, in addition to reactivity ratios, equilibrium constants for any reversible propagation steps. The reaction scheme is shown in Scheme 7.8. Penultimate unit effects are not considered. In 1960, Lowry⁹⁷ developed theory to cover copolymerization involving depropagation of only one monomer. Howell *et al.*⁹⁸ have carried out a more general treatment, allowing for all propagation steps being reversible, and provided expressions for predicting sequence distribution for these systems. Other treatments of copolymerization with depropagation are those of Wittmer⁹⁴ and Kruger *et al.*⁹⁹

| | | | | Scheme 7.8 |
|-------------------|---|----------------|---|-------------------|
| $P_{B}\bullet$ | ÷ | MB | = | $P_{BB}\bullet$ |
| $P_B{\bullet}$ | ÷ | M_A | + | $P_{BA}\bullet$ |
| $P_A \bullet$ | ÷ | M_B | + | P _{AB} ∙ |
| $P_{A^{\bullet}}$ | ÷ | M _A | ÷ | P _{AA} • |

7.3.2 Chain Statistics

The arrangement of monomer units in copolymer chains is determined by the monomer reactivity ratios which can be influenced by the reaction medium and various additives. The average sequence distribution to the triad level can often be measured by NMR (Section 7.3.3.2) and in special cases by other techniques.^{100,101} Longer sequences are usually difficult to determine experimentally, however, by assuming a model (terminal, penultimate, *etc.*) they can be predicted.^{7,102} Where sequence distributions can be accurately determined they provide, in principle, a powerful method for determining monomer reactivity ratios.

7.3.2.1 Binary copolymerization according to the terminal model

If chains are long such that the initiation and termination reactions have a negligible effect on the average sequence distribution, then according to the terminal model, P_{AA} , the probability that a chain ending in monomer unit M_A adds another unit M_A , is given by eq. 22:⁸

$$P_{AA} = \frac{k_{PAA}[P_A\bullet][M_A]}{k_{PAA}[P_A\bullet][M_A] + k_{PAB}[P_A\bullet][M_B]} = \frac{r_{AB}x}{r_{AB}x + 1}$$
(22)

Similarly, $P_{AB} = \frac{1}{r_{AB}x + 1} = 1 - P_{AA}$, $P_{BB} = \frac{r_{BA}/x}{r_{BA}/x + 1}$, $P_{BA} = \frac{1}{r_{BA}/x + 1} = 1 - P_{BB}$

354

The probability of a given sequence is the product of the probabilities of the individual steps that give rise to that sequence. Thus, the fraction of isolated sequences of monomer M_A which are of length *n* is:

$$N_{n}^{A} = P_{AA}^{n-1} P_{AB} = \frac{\left(r_{AB}x\right)^{n-1}}{\left(r_{AB}x+1\right)^{n}}$$
(23)

while the number average sequence length for monomer units $M_{\Lambda}\xspace$ is:

$$\overline{N}^{A} = \frac{1}{P_{AB}} = r_{AB}x + 1 \tag{24}$$

Expressions for the dyad, triad and higher order n-ad fractions can also be derived in terms of these probabilities. Thus the dyad fractions are given by eqs. 25-27.

$$F_{AA} = P_A P_{AA} = \frac{P_{BA} \left(1 - P_{AB} \right)}{P_{AB} + P_{BA}}$$
(25)

$$F_{\rm AB} - 2P_{\rm A}P_{\rm AB} - 2P_{\rm B}P_{\rm BA} = \frac{2P_{\rm BA}P_{\rm AB}}{P_{\rm AB} + P_{\rm BA}}$$
(26)

The mirror image sequences, such as the AB and BA dyads, cannot be distinguished.

$$F_{\rm BB} = P_{\rm B} P_{\rm BB} = \frac{P_{\rm AB} (1 - P_{\rm BA})}{P_{\rm AB} + P_{\rm BA}} = 1 - F_{\rm AA} - F_{\rm AB}$$
(27)

The six triad fractions are:

$$F_{AAA} = \frac{P_{BA} (1 - P_{AB})^2}{P_{AB} + P_{BA}} \qquad F_{AAB} = \frac{2P_{BA} P_{AB} (1 - P_{AB})}{P_{AB} + P_{BA}} \qquad F_{BAB} = \frac{P_{BA} P_{AB}^2}{P_{AB} + P_{BA}}$$
$$F_{BBB} = \frac{P_{AB} (1 - P_{BA})^2}{P_{AB} + P_{BA}} \qquad F_{BBA} = \frac{2P_{BA} P_{AB} (1 - P_{BA})}{P_{AB} + P_{BA}} \qquad F_{ABA} = \frac{P_{BA}^2 P_{AB}^2}{P_{AB} + P_{BA}}$$

Because $F_{AAA}+F_{AAB}+F_{BAB}+F_{BBB}+F_{BBA}+F_{BBB}=1$ and $2F_{ABA}+F_{BBA}=2F_{BAB}+F_{AAB}$, there are only four independent triad fractions.

7.3.2.2 Binary copolymerization according to the penultimate model

With the penultimate model, the probability that a chain with a terminal M_{BA} dyad will add a M_A unit is given by eq. 28:

$$P_{BAA} = \frac{k_{pBAA}[P_{BA}][M_A]}{k_{pBAA}[P_{BA}][M_A] + k_{pBAB}[P_{BA}][M_B]} = \frac{[M_A]}{[M_A] + [M]_B / r_{BAB}}$$
(28)
$$- \frac{r_{BAB}x}{r_{BAB}x + 1} = 1 - P_{BAB}$$

The probability that a chain with a terminal M_{AA} dyad will add a M_A unit is eq. 29:

$$P_{AAA} = \frac{k_{pAAA}[P_{AA}][M_A]}{k_{pAAA}[P_{AA}][M_A] + k_{pAAB}[P_{AA}][M_B]} = \frac{[M_A]}{[M_A] + [M]_B / r_{AAB}}$$
(29)
$$= \frac{r_{AAB}x}{r_{AAB}x + 1} - 1 - P_{AAB}$$

Eqs. 30 and 31 are derived similarly:

$$P_{\Lambda BB} = \frac{r_{\Lambda B\Lambda}}{r_{\Lambda B\Lambda} x + 1} = 1 - P_{\Lambda B\Lambda}$$
(30)

$$P_{\rm BBA} = \frac{x}{r_{\rm BBA}x + x} = 1 - P_{\rm BBB} \tag{31}$$

The probability that a chain with a terminal M_A will add a M_B can be expressed in terms of these probabilities as shown in eq. 32:

$$P_{\rm AB} = \frac{P_{\rm AAB}}{P_{\rm AAB} + P_{\rm BAA}} \tag{32}$$

7.3.2.3 Binary copolymerization according to other models

Expressions for predicting monomer sequence distribution with higher order models⁸ and for monomer complex and other models have also been proposed.

There are at least two additional complications that need to be considered when attempting to predict sequence distribution or measure reactivity on the basis of sequence data:

(a) The effects of chain tacticity. Chain ends of differing tacticity may have different reactivity towards monomers.¹⁰¹ When tacticity is imposed on top of monomer sequence distribution there are then six different dyads and twenty different triads to consider; analytical problems are thus severe. The tacticity of copolymers is usually described in terms of the coisotacticity parameters σ_{AB} and σ_{BA} ;¹⁰³ σ_{AB} is the probability of generating a meso dyad when a chain ending in A adds monomer B. Coisotacticity parameters have to date been reported for only a few copolymers including MMA-S,¹⁰⁴ MMA-MA,¹⁰⁵ and MMA-MAA.^{106,107} These data are likely to change due to the complexities associated with data analysis and NMR signal assignment (see also 7.3.3.2).

356

Copolymers involving only monosubstituted monomers are usually assumed to have random tacticity (*i.e.* $\sigma_{AB} = \sigma_{BA} = 0.5$).

(b) The effects of the reaction medium. Harwood^{108,109} observed that copolymers of the same composition have the same monomer sequence distribution irrespective of the solvent used for the copolymerization. He termed this the 'bootstrap effect'. This applies even though estimates of monomer reactivity ratios made on the basis of composition data may be significantly different. Much argument for and against the 'bootstrap effect' has appeared.^{1.3,110} Solvent effects on copolymerization and the 'bootstrap effect' are considered in more detail in Section 8.3.1.2.

The full picture of the factors affecting copolymer sequence distribution and their relative importance still needs to be filled in.

7.3.2.4 Terpolymerization

Terpolymerizations or ternary copolymerizations, as the names suggest, are polymerizations involving three monomers. Most industrial copolymerizations involve three or more monomers. The statistics of terpolymerization were worked out by Alfrey and Goldfinger in 1944.¹¹¹ If we assume terminal model kinetics, ternary copolymerization involves nine distinct propagation reactions (Scheme 7.9).

| P _A • + | M_A | \rightarrow | $\mathbf{P}_{\mathbf{A}}$ • | $k_{p \wedge \Lambda} [\mathbf{P}_{\Lambda} \bullet] [\mathbf{M}_{\Lambda}]$ |
|---------------------------|-------------|---------------|-----------------------------|--|
| $P_{\Lambda^{\bullet}}$ + | M_B | \rightarrow | P _B • | $k_{pAB}[P_{A}\bullet][M_{B}]$ |
| P _A ∙ + | $M_{\rm C}$ | \rightarrow | P _C ∙ | $k_{_{\mathrm{pAC}}}[\mathbf{P}_{\mathrm{A}}\bullet][\mathbf{M}_{\mathrm{C}}]$ |
| P _B ∙ + | M_A | \rightarrow | P_{A} • | $k_{_{\mathrm{pBA}}}[\mathrm{P}_{\mathrm{B}}\bullet][\mathrm{M}_{_{\mathrm{A}}}]$ |
| P _B ∙ + | M_{B} | \rightarrow | $P_{B}\bullet$ | $k_{_{\mathrm{P}}\mathrm{BB}}[\mathrm{P}_{\mathrm{B}}\bullet][\mathrm{M}_{_{\mathrm{B}}}]$ |
| P _B • + | $M_{\rm C}$ | \rightarrow | P_{C} • | $k_{_{\mathrm{PBC}}}[\mathrm{P}_{\mathrm{B}}\bullet][\mathrm{M}_{\mathrm{C}}]$ |
| P _C • + | M_A | \rightarrow | P _A ∙ | $k_{pCA}[P_{C}\bullet][M_{A}]$ |
| P _C • + | $M_{\rm B}$ | \rightarrow | P _B • | $k_{pCB}[P_{C}\bullet][M_{B}]$ |
| P _C • + | M_{C} | -> | P_C • | $k_{\rm pCC}[{ m P}_{ m C}\bullet][{ m M}_{ m C}]$ |

Scheme 7.9

Six reactivity ratios are then required to describe the system.

$$r_{\rm AB} = \frac{k_{\rm pAA}}{k_{\rm pAB}} \quad r_{\rm BA} = \frac{k_{\rm pBB}}{k_{\rm pBA}} \quad r_{\rm AC} = \frac{k_{\rm pAA}}{k_{\rm pAC}} \quad r_{\rm BC} = \frac{k_{\rm pBB}}{k_{\rm pBC}} \quad r_{\rm CA} = \frac{k_{\rm pCC}}{k_{\rm pCA}} \quad r_{\rm CB} = \frac{k_{\rm pCC}}{k_{\rm pCB}}$$

Application of a steady state assumption (eqs. 33-35) enables derivation of the composition relationship (eq. 36).

$$k_{pAB}[P_A\bullet][M_B] - k_{pAC}[P_A\bullet][M_C] = k_{pBA}[P_B\bullet][M_A] + k_{pCA}[P_C\bullet][M_A]$$
(33)

$$k_{pAB}[P_{B}\bullet][M_{A}] + k_{pBC}[P_{B}\bullet][M_{C}] = k_{pAB}[P_{A}\bullet][M_{B}] + k_{pCB}[P_{C}\bullet][M_{B}]$$
(34)

$$k_{pCA}[P_{C}\bullet][M_{A}] + k_{pCB}[P_{C}\bullet][M_{B}] = k_{pAC}[P_{A}\bullet][M_{C}] + k_{pBC}[P_{B}\bullet][M_{C}]$$
(35)

 $d\mathbf{M}_{A}: d\mathbf{M}_{B}: d\mathbf{M}_{C} = P_{A}: P_{B}: P_{C}$

$$= \mathbf{M}_{A} \left[\frac{\mathbf{M}_{A}}{r_{CA} r_{BA}} + \frac{\mathbf{M}_{B}}{r_{BA} r_{CB}} + \frac{\mathbf{M}_{C}}{r_{CA} r_{BC}} \right] \left[\mathbf{M}_{A} + \frac{\mathbf{M}_{B}}{r_{AB}} + \frac{\mathbf{M}_{C}}{r_{AC}} \right]$$
$$: \mathbf{M}_{B} \left[\frac{\mathbf{M}_{A}}{r_{AB} r_{CA}} + \frac{\mathbf{M}_{B}}{r_{AB} r_{CB}} + \frac{\mathbf{M}_{C}}{r_{CB} r_{AC}} \right] \left[\mathbf{M}_{B} + \frac{\mathbf{M}_{A}}{r_{BA}} + \frac{\mathbf{M}_{C}}{r_{BC}} \right]$$
$$: \mathbf{M}_{C} \left[\frac{\mathbf{M}_{A}}{r_{AC} r_{BA}} + \frac{\mathbf{M}_{B}}{r_{BC} r_{AB}} + \frac{\mathbf{M}_{C}}{r_{AC} r_{BC}} \right] \left[\mathbf{M}_{C} + \frac{\mathbf{M}_{A}}{r_{CA}} + \frac{\mathbf{M}_{B}}{r_{CB}} \right]$$
(36)

The terpolymer composition can be predicted on the basis of binary copolymerization experiments. If, however, one (or more) monomer is slow to propagate one of the reactivity ratios will approach zero and eq. 36 will become indeterminate. This situation arises in terpolymerizations involving, for example, MAH or AMS. Alfrey and Goldfinger¹¹² derived eq. 37 for the case where one monomer (C) is slow to propagate (*i.e.* $k_{pCC} \rightarrow 0$ and hence r_{CA} and $r_{CB} \rightarrow 0$). Expressions for other cases, for example, where two monomers (B and C) are slow to propagate, were also derived.¹¹² An equation related to eq. 37 has application in the analysis of binary copolymerizations in the presence of a transfer agent (Section 7.5.6).¹¹³

$$d\mathbf{M}_{A}: d\mathbf{M}_{B}: d\mathbf{M}_{C} = P_{A}: P_{B}: P_{C}$$

$$= \mathbf{M}_{A} \left[\frac{R\mathbf{M}_{A}}{r_{BA}} + \frac{\mathbf{M}_{B}}{r_{BA}} + \frac{R\mathbf{M}_{C}}{r_{BC}} \right] \left[\mathbf{M}_{A} + \frac{\mathbf{M}_{B}}{r_{AB}} + \frac{\mathbf{M}_{C}}{r_{AC}} \right]$$

$$: \mathbf{M}_{B} \left[\frac{R\mathbf{M}_{A}}{r_{AB}} + \frac{\mathbf{M}_{B}}{r_{AB}} + \frac{\mathbf{M}_{C}}{r_{AC}} \right] \left[\mathbf{M}_{B} + \frac{\mathbf{M}_{A}}{r_{BA}} + \frac{\mathbf{M}_{C}}{r_{BC}} \right]$$

$$: \mathbf{M}_{C} \left[\frac{\mathbf{M}_{A}}{r_{AC}r_{BA}} + \frac{\mathbf{M}_{B}}{r_{BC}r_{AB}} + \frac{\mathbf{M}_{C}}{r_{AC}r_{BC}} \right] \left[R\mathbf{M}_{A} + \mathbf{M}_{B} \right]$$
(37)

where $R = \frac{k_{\text{pCA}}}{k_{\text{pCB}}}$.

The value of R can only be evaluated by conducting a terpolymerization.

The complexity of the terpolymer composition equation (eq. 36) can be reduced to eq. 41 through the use of a modified steady state assumption (eqs. 38-40). However, while these equations apply to component binary copolymerizations it is not clear that they should apply to terpolymerization even though they appear to work well. It can be noted that when applying the Q-e scheme a terpolymer equation of this form is implied.

$$k_{pAB}[\mathbf{P}_{A}\bullet][\mathbf{M}_{B}] = k_{pBA}[\mathbf{P}_{B}\bullet][\mathbf{M}_{A}]$$
(38)

$$k_{pAC}[\mathbf{P}_{A}\bullet][\mathbf{M}_{C}] = k_{pCA}[\mathbf{P}_{C}\bullet][\mathbf{M}_{A}]$$
(39)

$$k_{pBC}[P_{B}\bullet][M_{C}] = k_{pCB}[P_{C}\bullet][M_{B}]$$
(40)

 $d\mathbf{M}_{A}: d\mathbf{M}_{B}: d\mathbf{M}_{C} = P_{A}: P_{B}: P_{C}$

$$= \mathbf{M}_{A} \left[\mathbf{M}_{A} + \frac{\mathbf{M}_{B}}{r_{AB}} + \frac{\mathbf{M}_{C}}{r_{AC}} \right]$$

$$: \mathbf{M}_{B} \frac{r_{BA}}{r_{AB}} \left[\mathbf{M}_{B} + \frac{\mathbf{M}_{A}}{r_{BA}} + \frac{\mathbf{M}_{C}}{r_{BC}} \right]$$

$$: \mathbf{M}_{C} \frac{r_{CA}}{r_{AC}} \left[\mathbf{M}_{C} + \frac{\mathbf{M}_{A}}{r_{CA}} + \frac{\mathbf{M}_{B}}{r_{CB}} \right]$$
(41)

Azeotropic compositions are rare for terpolymerization and Ham¹¹⁴ has shown that it follows from the simplified eqs. 38-40 that ternary azeotropes should not exist. Nonetheless, a few systems for which a ternary azeotrope exists have now been described (this is perhaps a proof of the limitations of the simplified equations) and equations for predicting whether an azeotropic composition will exist for copolymerizations of three or more monomers have been formulated.^{20,115} This work also shows that a ternary azeotropic composition for any of the three possible binary copolymerizations of the monomers involved.

7.3.3 Estimation of Reactivity Ratios

Methods for evaluation of reactivity ratios comprise a significant proportion of the literature on copolymerization. There are two basic types of information that can be analyzed to yield reactivity ratios. These are (a) copolymer composition/conversion data (Section 7.3.3.1) and (b) the monomer sequence distribution (Section 7.3.3.2). The methods used to analyze these data are summarized in the following sections.

7.3.3.1 Composition data

The traditional method for determining reactivity ratios involves determinations of the overall copolymer composition for a range of monomer feeds at 'zero conversion'. Various methods have been applied to analyze this data. The Fineman-Ross equation (eq. 42) is based on a rearrangement of the copolymer composition equation (eq. 9). A plot of the quantity on the left hand side of eq. 9 vs the coefficient of r_{AB} will yield r_{AB} as the slope and r_{BA} as the intercept.

$$\frac{f_{\rm A}(1-2F_{\rm A})}{F_{\rm A}(1-f_{\rm A})} = r_{\rm AB} \frac{f_{\rm A}^2(F_{\rm A}-1)}{F_{\rm A}(1-f_{\rm A})^2} + r_{\rm BA}$$
(42)

Early methods such as the Intersection,¹⁴ and Fineman-Ross¹¹⁶ methods do not give equal weighting to the experimental points such that there is a non-linear dependence of the error on the composition. Consequently, these methods can give erroneous results.

These problems were addressed by Tidwell and Mortimer^{117,118} who advocated numerical analysis by non-linear least squares and Kelen and Tüdos^{119,120} who proposed an improved graphical method for data analysis. The Kelen-Tüdos equation is as follows (eq. 43):

$$\eta = \xi \left(r_{\rm A} + \frac{r_{\rm BA}}{\alpha} \right) - \frac{r_{\rm BA}}{\alpha} \tag{43}$$

where $\eta = \frac{x(y-1)}{y(\alpha + x^2/y)}$, $\xi = \frac{x^2/y}{\alpha + x^2/y}$ and α is a constant.

A plot of η vs ξ should yield a straight line with intercepts of $-r_{BA}/\alpha$ and r_{AB} at $\xi=0$ and $\xi=1$ respectively. A value of α corresponding to the highest and lowest values of $(x^2/y)^{0.5}$ used in the experiments results in a symmetrical distribution of experimental data on the plot. Greenley^{18,121,122} has re-evaluated much data using the Kelen-Tüdos method and has provided a compilation of these and other results in the Polymer Handbook.¹⁸

It is also possible to derive reactivity ratios by analyzing the monomer (or polymer) feed composition vs conversion and solving the integrated form of the Mayo Lewis equation.^{10,123} The following expression (eq. 44) was derived by Meyer and Lowry:¹²³

$$conversion = 1 - \left(\frac{f_{\rm A}}{f_{\rm Ao}}\right)^{\alpha} \left(\frac{f_{\rm B}}{f_{\rm Bo}}\right)^{\beta} \left(\frac{f_{\rm Ao} - \delta}{f_{\rm A} - \delta}\right)^{\gamma}$$
(44)

where
$$\alpha = \frac{r_{BA}}{1 - r_{BA}}$$
 $\beta = \frac{r_{AB}}{1 - r_{AB}}$ $\delta = \frac{1 - r_{AB}r_{BA}}{(1 - r_{AB})(1 - r_{BA})}$ $\gamma = \frac{1 - r_{BA}}{2 - r_{AB} - r_{BA}}$

360

Numerical approaches for estimating reactivity ratios by solution of the integrated rate equation have been described.¹²⁴⁻¹²⁶ Potential difficulties associated with the application of these methods based on the integrated form of the Mayo-Lewis equation have been discussed.¹²⁴⁻¹²⁷ One is that the expressions become undefined under certain conditions, for example, when r_{AB} or r_{BA} is close to unity or when the composition is close to the azeotropic composition. A further complication is that reactivity ratios may vary with conversion due to changes in the reaction medium.

Clearly, great care must be taken in the estimation of reactivity ratios from composition/conversion data. Many papers have been written on the merits of various schemes and comparisons of the various methods for reactivity ratio calculation have appeared.¹²⁸⁻¹³² Given appropriate design of the experiment, graphical methods for the estimation of reactivity ratios can give reasonable values. They also have the virtue of simplicity and do not require the aid of a computer. However, as a general rule, the use of such methods is not recommended except as an initial guide. It is more appropriate to use some form of non-linear least squares regression analysis to derive the reactivity ratios. The use of "error in variable" methods^{6,133-135} which take into account the error structure of the experimental data is highly recommended.

It is also possible to process copolymer composition data to obtain reactivity ratios for higher order models (*e.g.* penultimate model or complex participation, *etc.*). However, composition data have low power in model discrimination (Sections 7.3.1.2 and 7.3.1.3). There has been much published on the subject of the design of experiments for reactivity ratio determination and model discrimination.^{49,118,136,137} Attention must be paid to the information that is required; the optimal design for obtaining terminal model reactivity ratios may not be ideal for model discrimination.⁴⁹

One final point should be made. The observation of significant solvent effects on k_p in homopolymerization and on reactivity ratios in copolymerization (Section 8.3.1) calls into question the methods for reactivity ratio measurement which rely on evaluation of the polymer composition for various monomer feed ratios (Section 7.3.2). If solvent effects are significant, it would seem to follow that reactivity ratios in bulk copolymerization should be a function of the feed composition.¹³⁸ Moreover, since the reaction medium alters with conversion, the reactivity ratios may also vary with conversion. Thus the two most common sources of data used in reactivity ratio determination (i.e. low conversion composition measurements and composition conversion measurements) are potentially flawed. A corollary of this statement also provides one explanation for any failure of reactivity ratios to predict copolymer composition at high conversion. The effect of solvents on radical copolymerization remains an area in need of further research.

7.3.3.2 Monomer sequence distribution

NMR spectroscopy has made possible the characterization of copolymers in terms of their monomer sequence distribution. The area has been reviewed by Randall,¹⁰⁰ Bovey,¹³⁹ Tonelli,¹⁰¹ Hatada¹⁴⁰ and others. Information on monomer sequence distribution is substantially more powerful than simple composition data with respect to model discrimination.^{25,49} Although many authors have used the distribution of triad fractions to confirm the adequacy or otherwise of various models, only a few^{25,58,141} have used dyad or triad fractions to calculate reactivity ratios directly.

Terminal model reactivity ratios may be estimated from the initial monomer feed composition and the dyad concentrations in low conversion polymers using the following relationships (eqs. 45, 46).

$$r_{\rm AB} = \frac{f_{\rm B}}{f_{\rm A}} \frac{2F_{\rm AA}}{F_{\rm AB}} \tag{45}$$

$$r_{\rm BA} = \frac{f_{\rm A}}{f_{\rm B}} \frac{2F_{\rm BB}}{F_{\rm AB}} \tag{46}$$

Note that the dyad concentrations can be easily calculated from the triad concentrations (eqs. 47-49).

$$F_{AA} = F_{AAA} + \frac{F_{AAB}}{2}$$
(47)

$$F_{AB} = F_{ABA} + F_{BAB} + \frac{F_{AAB}}{2} + \frac{F_{ABB}}{2}$$
(48)

$$F_{\rm BB} = F_{\rm BBB} + \frac{F_{\rm ABB}}{2} \tag{49}$$

Similarly, penultimate model reactivity ratios can be estimated from initial monomer feed composition and triad concentrations using eqs. 50-53.

$$r_{AAB} = \frac{f_B}{f_A} \frac{2F_{AAA}}{F_{AAB}}$$
(50)

$$r_{\rm BAB} = \frac{f_{\rm B}}{f_{\rm A}} \frac{F_{\rm AAB}}{2F_{\rm BAB}} \tag{51}$$

$$r_{ABA} = \frac{f_A}{f_B} \frac{F_{ABB}}{2F_{ABA}}$$
(52)

$$r_{\rm RBA} = \frac{f_{\rm B}}{f_{\rm A}} \frac{2F_{\rm BBB}}{F_{\rm ABB}}$$
(53)

While sequence distributions are usually subject to more experimental noise than composition data, this is often outweighed by the greater information content. In principle, reactivity ratios can be estimated from a single copolymer sample. The consistency in reactivity ratios estimated with eqs. 45 and 46 for copolymers prepared with different monomer feed compositions and/or obtaining the same result from eqs. 50 and 51 ($r_{AAB} \neg r_{BAB}$) and eqs. 52 and 53 ($r_{ABA} \neg r_{BBA}$) are evidence for the applicability of the terminal model.^{28,142} Consistent reactivity ratios from application of eqs. 50-53 to copolymers prepared using a range of monomer feed compositions is evidence for the penultimate unit model. A limitation in the use of these equations is the precision of triad distribution data.

Another serious problem in applying these methods is that unambiguous assignments of NMR signals to monomer sequences are, as yet, only available for a few systems. Moreover, assignments are complicated by the fact that the sensitivity of chemical shifts to tacticity may be equal or greater than their sensitivity to monomer sequence.^{140,143}

The usual experiment is to prepare a series of copolymers each containing a different ratio of the monomers. A correlation of expected and measured peak intensities may then enable peak assignment.^{24,25} However, this method is not foolproof and papers on signal reassignment are not uncommon.^{56,104,143} 2D NMR methods,¹⁴³ decoupling experiments,⁵⁶ special pulse sequences²⁸ and analyses of isotopically labeled^{144,145} or regioregular⁵⁶ polymers have greatly facilitated analysis of complex systems. In principle, these methods allow a "mechanism-free" signal assignment.

7.3.4 Prediction of Reactivity Ratios

Various methods for predicting reactivity ratios have been proposed.¹⁴⁶ These schemes are largely empirical although some have offered a theoretical basis for their function. They typically do not allow for the possibility of variation in reactivity ratios with solvent and reaction conditions. They also presuppose a terminal model. Despite their limitations they are extremely useful for providing an initial guess in circumstances where other data is unavailable.

The most popular methods are the Q-e (Section 7.3.4.1) and 'Patterns of Reactivity' schemes (Section 7.3.4.2). Both methods may also be used to predict transfer constants (Section 6.2.1). For further discussion on the application of these and other methods to predict rate constants in radical reactions, see Section 2.3.7.

7.3.4.1 Q-e scheme

The method for the prediction of reactivity ratios in most widespread usage is the Q-e scheme.^{17,147} This scheme was devised in 1947 by Alfrey and Price¹⁴⁸ who

proposed that the rate constant for reaction of radical (R^{\bullet}) with monomer (M) should be dependent on polarity and resonance terms according to the following expression (eq. 54):

$$k_{\rm RM} = P_{\rm R} Q_{\rm M} e^{-\mathcal{C}_{\rm R} \mathcal{C}_{\rm M}} \tag{54}$$

where $P_{\rm R}$ and $Q_{\rm M}$ are the 'general reactivity' of the radical and monomer respectively. It has been proposed that these take into account resonance factors. The *e* values are related to the polarity of the radical or monomer ($e_{\rm R}$ and $e_{\rm M}$ are assumed to be the same). The parameters $P_{\rm R}$ are eliminated in the expressions for the reactivity ratios. The reactivity ratios $r_{\rm AB}$ and $r_{\rm BA}$ depend on Q and e as shown in eqs. 55, 56,

$$r_{AB} = \frac{k_{AA}}{k_{AB}} = \frac{Q_{A}}{Q_{B}} e^{-e_{A}(e_{A} - e_{B})}$$
(55)

$$r_{\rm BA} = \frac{k_{\rm BB}}{k_{\rm BA}} = \frac{Q_{\rm B}}{Q_{\rm A}} e^{-c_{\rm B}(c_{\rm B}-c_{\rm A})}$$
(56)

S is taken as the reference monomer with Q=1.0 and e = -0.8. Values for other monomers are derived by regression analysis based on literature or measured reactivity ratios. The *Q*-*e* values for some common monomers as presented in the *Polymer Handbook*¹⁴⁹ are given in Table 7.7. The accuracy of *Q*-*e* parameters is limited by the quality of the reactivity ratio data and can also suffer from inappropriate statistical treatment employed in their derivation.^{17,18} A further problem is that the data analysis makes no allowance for the dependence of reactivity ratios on reaction conditions. Reactivity ratios can be dependent on solvent (Section 7.3.1.2), reaction temperature, pII, *etc.* It follows that values of *e* and perhaps *Q* for a given monomer should depend on the medium, the monomer ratio and the particular comonomer. This is especially true for monomers which contain ionizable groups (*e.g.* MAA, AA, vinyl pyridine) or are capable of forming hydrogen bonds (*e.g.* HEMA, HEA).

There have, however, been attempts to correlate Q-e values and hence reactivity ratios to, for example, ¹³C NMR chemical shifts¹⁵⁰ or the results of MO calculations¹⁵¹⁻¹⁵³ and to provide a better theoretical basis for the parameters. Most recently, Zhan and Dixon¹⁵³ applied density functional theory to demonstrate that Q values could be correlated to calculated values of the relative free energy for the radical monomer reaction ($P_A \cdot + M_B \rightarrow P_A \cdot$). The e values were correlated to values of the electronegativities of monomer and radical.

The NMR method of predicting Q-e values appears attractive since spectra can be measured under the particular reaction conditions (solvent, temperature, pH). Thus, it may be possible to predict the dependence of the Q-e values and reactivity ratios on the reaction medium.¹⁵⁰

| Monomer | Q | е | $\log r_{1S}$ | π | U | v |
|---------|-------|-------|---------------|--------|-------|-------|
| В | 1.70 | -0.50 | 0.1461 | -0.100 | -0.30 | 0.41 |
| S | 1.0 | -0.8 | 0 | 0 | 0 | 0 |
| MAA | 0.98 | 0.62 | -0.2807 | 0.002 | -0.95 | 0.62 |
| AMS | 0.97 | -0.81 | -0.2219 | -0.77 | -0.04 | -0.03 |
| MAN | 0.86 | 0.69 | -0.4815 | 0.432 | -2.08 | 0.44 |
| BMA | 0.82 | 0.28 | -0.2757 | 0.267 | -1.49 | 0.26 |
| AA | 0.83 | 0.88 | - | - | - | - |
| MMA | 0.78 | 0.40 | -0.3372 | 0.339 | -1.18 | 0.23 |
| AN | 0.48 | 1.23 | -1.3980 | 0.701 | -2.6 | 0.42 |
| MA | 0.45 | 0.64 | -0.7447 | 0.421 | -2.34 | 0.16 |
| BA | 0.38 | 0.85 | -0.7447 | 0.443 | -2.22 | 0.12 |
| VC | 0.056 | -0.16 | -1.26 | 0.128 | -0.90 | -1.16 |
| VAc | 0.026 | -0.88 | -1.699 | 0.315 | -0.44 | -1.56 |

Table 7.7 Q- e^{149} and Patterns¹⁵⁴ Parameters for Some Common Monomers

7.3.4.2 Patterns of reactivity scheme

Bamford, Jenkins and coworkers¹⁵⁵⁻¹⁵⁷ concluded that many of the limitations of the *Q-e* scheme stemmed from its empirical nature and proposed a new scheme containing a radical reactivity term, based on experimentally measured values of the rate constant for abstraction of benzylic hydrogen from toluene ($k_{3,T}$), a polar term (the Hammett σ value) and two constants α and β which are specific for a given monomer or substrate (eq. 57):¹⁴⁶

$$\log k = \log k_{3,\mathrm{T}} + \alpha \sigma + \beta \tag{57}$$

and reactivity ratios are then defined by eqs. 58 and 59:

$$\log r_{AB} = \sigma_A(\alpha_A - \alpha_B) + \beta_A - \beta_B \tag{58}$$

$$\log r_{\rm BA} = \sigma_{\rm B}(\alpha_{\rm B} - \alpha_{\rm A}) + \beta_{\rm B} - \beta_{\rm A} \tag{59}$$

In the revised Patterns scheme reactivity ratios involving S are used as reference reactions.^{154,158} Reactivity ratios are then given by eqs. 60 and 61:

$$\log r_{\rm AB} \quad \log r_{\rm AS} - u_{\rm B} \pi_{\rm A} - \nu_{\rm B} \tag{60}$$

$$\log r_{\rm BA} = \log r_{\rm BS} \cdot u_{\rm A} \pi_{\rm B} \cdot v_{\rm A} \tag{61}$$

where r_{AS} (= k_{AA}/k_{aS}) is the reactivity ratio of the monomer (A) with S (log r_{AS} is the counterpart of Q in the Q-e scheme), π is a polarity term and is strongly correlated with the Hammet σ parameter (it is the counterpart of e) and u and v are constants. Tabulations of the Patterns parameters can be found in the Polymer Handbook¹⁵⁴ and a subset of this data is reproduced in Table 7.7. The scheme can also be used to predict chain transfer constants.

The Patterns scheme has been tested for its capacity to predict ¹³C NMR chemical shifts of the CH_2 = carbon of monomers $(CH_2=CXY)^{159}$ and in evaluating the reactivities of small radicals towards monomers.¹⁶⁰

7.4 Termination in Statistical Copolymerization

This section begins with a brief discussion of copolymerization kinetics and various models that have been used to describe termination. These models were derived with the presumption that the terminal model describes propagation in copolymerization. The "chemical control model" (Section 7.4.1) and the various diffusion control models (Section 7.4.2) as originally conceived largely fell from use with the advent of methods that allowed absolute values for the overall propagation rate constant in copolymerization to be reliably determined (*e.g.* PLP). Application of these methods pointed to the failure of terminal model kinetics by demonstrating that the overall propagation rate constant was strongly dependent on the monomer feed composition. Thus, 'anomalies' in copolymerization kinetics previously attributed to variation in the termination rate constant monomer feed composition rate constant. The so-called implicit and explicit penultimate models described in Section 7.3.1.2.1 were derived.

More recent work has shown that the observed variation in propagation rate constants with composition is not sufficient to define the polymerization rates.^{52,161,162} There remains some dependence of the termination rate constant on the composition of the propagating chain. Thus, the "chemical control" (Section 7.4.1) and the various diffusion control models (Section 7.4.2) have seen new life and have been adapted by substituting the terminal model propagation rate constants (k_{pXY}) with implicit penultimate model propagation rate constants (\bar{k}_{pXY} - Section 7.3.1.2.2).

The chain length dependence of termination rate constants (Section 5.2.1.4) should not be ignored when considering copolymerization kinetics. It has been pointed out that average chain lengths in copolymerization will be a function of the monomer feed composition¹⁶¹ especially in copolymerizations with disparate propagation rate constants. Factors determining the rate of copolymerization are not fully resolved and copolymerization kinetics remains a topic of discussion and an area in need of further study.

7.4.1 Chemical Control Model

The rate of copolymerization often shows a strong dependence on the monomer feed composition. Many theories have been developed to predict the rate of copolymerization based on the terminal model for chain propagation (Section 7.3.1.1). This usually requires an overall rate constant for termination in copolymerization that is substantially different from that observed in homopolymerization of any of the component monomers.

366

In early work, it was assumed that the rate constant for termination was determined by the monomer unit at the reacting chain ends. The kinetics of copolymerization were then dictated by the rate of initiation, the rates of the four propagation reactions (Scheme 7.1) and rates of three termination reactions (Scheme 7.10).¹⁶³⁻¹⁶⁵

Scheme 7.10

The instantaneous rate of monomer consumption in binary copolymerization is then given by eq. 62:

$$R_{p} = -\frac{d[M_{A} + M_{B}]}{dt}$$
$$= k_{pAA}[P_{A}\bullet][M_{A}] + k_{pBA}[P_{B}\bullet][M_{A}] - k_{pAB}[P_{A}\bullet][M_{B}] + k_{pBB}[P_{B}\bullet][M_{B}]$$
(62)

Use of the steady state approximation

$$R_{t} = k_{tAA} [P_{A} \bullet]^{2} + 2k_{tAB} [P_{A} \bullet] [P_{B} \bullet] + k_{tBB} [P_{B} \bullet]^{2} = R_{t} = 2k_{d} f [I_{2}]$$

allows the concentrations of the active species to be eliminated. Thus eq. 63:

$$-\frac{d[M_{A} + M_{B}]}{dt} = \frac{\left(k_{pAA}k_{pBA}[M_{A}]^{2} + 2k_{pAB}k_{pBA}[M_{A}][M_{B}] + k_{pBB}k_{pAB}[M_{B}]^{2}\right)R_{i}^{0.5}}{k_{tAA}k_{pBA}^{2}[M_{A}]^{2} + 2k_{tAB}k_{pAB}k_{pBA}[M_{A}][M_{B}] + k_{tBB}k_{pAB}^{2}[M_{B}]^{2}}$$
(63)

which can be rewritten as eq. 64:¹⁶⁵

$$\frac{-d[M_{A} + M_{B}]}{dt} = \frac{(r_{AB}[M_{A}]^{2} + 2[M_{A}][M_{B}] + r_{BA}[M_{B}]^{2})R_{i}^{0.5}}{\delta_{A}^{2}r_{AB}^{2}[M_{A}]^{2} + 2\phi\delta_{A}\delta_{B}r_{AB}r_{BA}[M_{A}][M_{B}] + \delta_{B}^{2}r_{BA}^{2}[M_{B}]^{2}}$$
(64)

where:

$$\phi = \frac{k_{\text{tAB}}}{2(k_{\text{tAA}}.k_{\text{tBB}})^{0.5}} \qquad \delta_{\text{A}} = \frac{2k_{\text{tAA}}^{0.5}}{k_{\text{pAA}}} \qquad \delta_{\text{B}} = \frac{2k_{\text{tBB}}^{0.5}}{k_{\text{pBB}}} \qquad r_{\text{AB}} = \frac{k_{\text{pAA}}}{k_{\text{pAB}}} \quad r_{\text{BA}} = \frac{k_{\text{pBB}}}{k_{\text{pBA}}}$$

In evaluating the kinetics of copolymerization according to the chemical control model, it is assumed that the termination rate constants k_{tAA} and k_{tBB} are known from studies on homopolymerization. The only unknown in the above expression is the rate constant for cross termination (k_{tAB}). The rate constant for this reaction in relation to k_{tAA} and k_{tBB} is given by the parameter ϕ .

Values of ϕ required to fit the rate of copolymerization by the chemical control model were typically in the range 5-50 though values <1 are also known. In the case of S-MMA copolymerization, the model requires ϕ to be in the range 5-14 depending on the monomer feed ratio. This "chemical control" model generally fell from favor with the recognition that chain diffusion should be the rate determining step in termination.

However, recent work based on the assumption of the implicit penultimate model suggests a value of ϕ for S-MMA copolymerization to be in the range 2-3.^{52,161} This value is in remarkably good agreement with that suggested by experiments with simple model radicals. These experiments also indicate that cross termination is 2-3 times faster than either homotermination reaction (Section 7.4.3.1).

7.4.2 Diffusion Control Models

In the classical diffusion control model it is assumed that propagation occurs according to the terminal model (Scheme 7.1). The rate of the termination step is limited only by the rates of diffusion of the polymer chains. This rate may be dependent on the overall polymer chain composition. However, it does not depend solely on the chain end.^{166,167}

$$P_{(AB)}\bullet + P_{(AB)}\bullet \longrightarrow \text{products} \quad k_{((AB)}[P_{(AB)}\bullet][P_{(AB)}\bullet]$$

North and coworkers^{166,168} proposed that chains terminate with a rate constant which is determined by the rate of diffusion. Thus

$$-\frac{d[M_{A} + M_{B}]}{dt} = \frac{(r_{AB}[M_{A}]^{2} + 2[M_{A}][M_{B}] + r_{BA}[M_{B}]^{2})R_{i}^{0.5}}{\varepsilon_{A}r_{AB}[M_{A}] + \varepsilon_{B}r_{BA}[M_{B}]} =$$
(65)

where $\varepsilon_{\rm A} = k_{\rm t(AB)}^{0.5} / k_{\rm pAA}, \varepsilon_{\rm B} = k_{\rm t(AB)}^{0.5} / k_{\rm pBB}$.

and $k_{t(AB)}$ is the copolymer-composition dependent rate constant for termination. It is not a constant. In eq. 65, the value of $k_{t(AB)}$ is obtained by fitting the experimental data. Various methods have then been proposed to estimate a dependence of $k_{t(AB)}$ on the monomer feed composition and the rate constants for homotermination (eqs. 66-68).^{166,169}

$$k_{\rm I(AB)} = F_{\rm A} k_{\rm IAA} + F_{\rm B} k_{\rm IBB} \tag{66}$$

$$k_{t(AB)} = F_A^2 k_{tAA} + F_A F_B k_{tAB} + F_B^2 k_{tBB}$$
(67)

$$k_{i(AB)} = F_A^2 k_{iAA} + 2F_A F_B \phi (k_{iAA} k_{iBB})^{0.5} + F_B^2 k_{iBB}$$
(68)

In eq. 68, ϕ is defined as in the chemical control model but this expression is cast in terms of the monomer feed composition rather than the radical chain end population.

More complex models for diffusion-controlled termination in copolymerization have appeared.¹⁷⁰⁻¹⁷³ Russo and Munari¹⁷¹ still assumed a terminal model for propagation but introduced a penultimate model to describe termination. There are ten termination reactions to consider (Scheme 7.11). The model was based on the hypothesis that the type of penultimate unit defined the segmental motion of the chain ends and their rate of diffusion.

| $P_{AA} +$ | P _{AA} ∙ | \rightarrow | products | $k_{\text{tAAAAA}}[P_{AA}\bullet][P_{AA}\bullet]$ |
|--------------------------------|-------------------|---------------|----------|---|
| $P_{AA}\bullet \ +$ | $P_{AB}\bullet$ | \rightarrow | products | $-2k_{\rm iaaab}[\mathbf{P}_{\rm aa}\bullet][\mathbf{P}_{\rm ab}\bullet]$ |
| $P_{\Lambda\Lambda} \bullet +$ | $P_{BA}\bullet$ | \rightarrow | products | $2 k_{\text{maba}} [P_{\text{ma}} \bullet] [P_{\text{ma}} \bullet]$ |
| $P_{\Lambda\Lambda} \bullet +$ | P_{AB} • | \rightarrow | products | $2 k_{\text{tAAAB}} [\mathbf{P}_{AA} \bullet] [\mathbf{P}_{AB} \bullet]$ |
| $P_{AB} \bullet +$ | $P_{AB}\bullet$ | \rightarrow | products | $k_{\text{tABAB}}[\mathbf{P}_{AB}\bullet][\mathbf{P}_{AB}\bullet]$ |
| $P_{AB}\bullet$ + | P _{BA} • | -> | products | $2 k_{\text{tABBA}} [\mathbf{P}_{\text{AB}} \bullet] [\mathbf{P}_{\text{BA}} \bullet]$ |
| $P_{AB}\bullet \ +$ | $P_{BB}\bullet$ | \rightarrow | products | $2 k_{\text{tabbb}} [P_{\text{ab}} \bullet] [P_{\text{bb}} \bullet]$ |
| $P_{BA}\bullet \ \pm$ | $P_{BA}\bullet$ | \rightarrow | products | $k_{\scriptscriptstyle \mathrm{tBABA}}[\mathrm{P}_{\scriptscriptstyle \mathrm{BA}}\bullet][\mathrm{P}_{\scriptscriptstyle \mathrm{BA}}\bullet]$ |
| $P_{BA}\bullet \ +$ | $P_{BB}\bullet$ | \rightarrow | products | $2 k_{\text{BABB}} [P_{\text{BA}} \bullet] [P_{\text{BB}} \bullet]$ |
| $P_{BB} \bullet +$ | P _{BB} • | \rightarrow | products | $k_{\text{rBBBB}}[P_{\text{BB}}\bullet][P_{\text{BB}}\bullet]$ |

Scheme 7.11

The rate constants for the cross termination terms are approximated as the geometric mean of the corresponding homotermination terms. Thus:

$$k_{\text{tAAAB}} = 2(k_{\text{tAAAA}} k_{\text{tABAB}})^{0.5} k_{\text{tAABA}} = 2(k_{\text{tAAAA}} k_{\text{tBABA}})^{0.5} k_{\text{tAABB}} = 2(k_{\text{tAAAA}} k_{\text{tBBBB}})^{0.5} k_{\text{tABBB}} = 2(k_{\text{tABAA}} k_{\text{tBBBB}})^{0.5} k_{\text{tABBA}} = 2(k_{\text{tABAB}} k_{\text{tBBBB}})^{0.5} k_{\text{tBABA}} = 2(k_{\text{tBABA}} k_{\text{tBBBB}})^{0.5} k_{\text{tBBBB}} = 2(k_{\text{tBABA}} k_{\text{tBBBB}})^{0.5}$$

which allows the rate of polymerization to be defined in terms of four termination rate constants (eq. 69).

$$-\frac{d[M_{A} + M_{B}]}{dt} = \frac{(r_{AB}[M_{A}]^{2} + 2[M_{A}][M_{B}] + r_{BA}[M_{B}]^{2})R_{i}^{0.5}}{\delta_{AA}r_{AB}^{2}[M_{A}]^{2} + \delta_{BA}r_{AB}[M_{A}][M_{B}]} + \frac{\delta_{BB}r_{BA}^{2}[M_{B}]^{2} + \delta_{AB}r_{BA}[M_{A}][M_{B}]}{r_{BA}[M_{A}] + [M_{B}]}$$
(69)
where
$$\delta_{AA} = \frac{k_{tAAAA}^{0.5}}{k_{pAA}}$$
 $\delta_{AB} = \frac{k_{tABAB}^{0.5}}{k_{pAB}}$ $\delta_{BA} = \frac{k_{tBABA}^{0.5}}{k_{pBA}}$ $\delta_{BB} = \frac{k_{tBBBB}^{0.5}}{k_{pBB}}$

This model provides a better description of the rate of copolymerization for some systems but has been criticized as having too many adjustable parameters.¹⁷⁴

Fukuda and coworkers¹⁶² have recently derived a model equivalent to the Russo-Munari model but where the implicit penultimate model is used to describe the propagation kinetics.

7.4.3 Combination and Disproportionation during Copolymerization

It is important to realize that, even if the rate of termination is determined by the rates of chain diffusion, the chain end composition and the ratio of combination to disproportionation are not. Knowledge or prediction of the overall rate of termination offers little insight into the detailed chemistry of the termination processes not involved in the rate-determining step.

Even when only the terminal monomer unit is considered, radical-radical termination in binary copolymerization involves at least seven separate reactions (Scheme 7.12). There are two homotermination processes and one cross termination process to consider. In the case of cross termination, there are two pathways for disproportionation. There are then at least three pieces of information to be gained:

- (a) The value of k_{td}/k_{tc} for cross termination.
- (b) The specificity for hydrogen transfer in disproportionation (*i.e.* from monomer A to monomer B or vice versa).
- (c) The relative rates of homo- and cross-termination.

Homotermination of A-ended chains

Scheme 7.12

370

Perhaps because of this complexity, few studies on determining k_{td}/k_{tc} in cross termination in copolymerization have been reported and most of the available data come from model studies. It is also usually assumed, without specific justification, that penultimate unit effects are unimportant in determining which reactions occur and that values of k_{td}/k_{tc} for the homotermination reactions are similar to those in the corresponding homopolymerizations.

Three types of model study have been performed. The first approach has been to decompose a mixture of two initiators (*i.e.* one to generate radical A, the other to generate radical B). With this method experimental difficulties arise because the two types of radical may not be generated at the same rate and because homotermination products from cage recombination complicate analysis.

A second approach has been to use an unsymmetrical initiator which allows the two radicals of interest to be generated simultaneously in equimolar amounts.¹⁷⁵ In this case, analysis of the cage recombination products provides information on cross termination uncomplicated by homotermination. Analysis of products of the encounter reaction can also give information on the relative importance of cross and homotermination. However, copolymerization of unsaturated products can cause severe analytical problems.

A third technique is to examine the products of primary radical termination in polymerizations carried out with high concentrations of initiator.^{176,177} Values of k_{td}/k_{tc} ratios in primary radical termination have been reported for a number of polymerizations carried out with AIBN (model for PMAN•) or AIBMe (model for PMMA•) initiation.

7.4.3.1 Poly(methyl methacrylate-co-styrene)

In termination, the rate determining step is the rate at which the chain ends are brought together by diffusion. Since propagation is rapid with respect to termination, the relative radical concentrations are more important than the termination rate constants in determining the products of termination.¹⁷⁸ The relative radical concentrations are in turn determined by the values of the reactivity ratios and the propagation rate constants. These considerations ensure that, during MMA-S copolymerization, the instantaneous concentration of chains ending in S is significantly greater than that of those with a terminal MMA unit.¹⁷⁸ Therefore, homotermination of chains ending in S and cross termination are the most important processes. There is comparatively little homotermination between chains ending in MMA (Table 7.8).

The reaction between the PMMA and PS model radicals (4 and 5, generated from the unsymmetrical azo-compound 3) has been studied as a model for cross-termination in MMA-S copolymerization (Scheme 7.13).^{178,179} The value for $k_{td}/k_{tc}(90^{\circ}\text{C})$ for the cross reaction was 0.56. In disproportionation, transfer of hydrogen from the PS• model 5 to the PMMA• radical 4 was *ca* 5.1 times more prevalent than transfer in the reverse direction (from 4 to 5). The value of $k_{td}/k_{tc}(90^{\circ}\text{C})$ is between those of $k_{td}/k_{tc}(90^{\circ}\text{C})$ for the self-reaction of these radicals

under similar conditions (0.13 and 0.78 for 5 and 4 respectively). Analysis of the encounter products indicated a small preference for cross termination over either homotermination process.¹⁷⁸

| Table 7.8 | Identity of Chain End Units Involved in Radical-Radical Termination in |
|-----------|--|
| | MMA-S Copolymerization ^a |

| Reaction | <u>'Chemical Control'</u> | | | 'Diffusion Control" |
|--------------|---------------------------|------|------|---------------------|
| | φ=13 | ф=3 | φ=1 | |
| -S• + -S• | 0.18 | 0.47 | 0.72 | 0.57 |
| -S• + -MMA• | 0.81 | 0.51 | 0.26 | 0.37 |
| -MMA• + MMA• | 0.01 | 0.02 | 0.02 | 0.06 |
| | 170 | | | |

 a Calculated by kinetic simulation.¹⁷⁸ b Calculated using the classical chemical control model (7.4.1). c Calculated using the diffusion control model of Russo and Munari¹⁷¹ (7.4.2).

Both S polymerization initiated by AIBMe^{176,180} (*i.e.* $PS \cdot + 4$) and MMA polymerization initiated by 1,1'-azobis-1-phenylethane¹⁷⁶ (*i.e.* PMMA \cdot + 1-phenylethyl radical) are reported to give predominantly combination. Ito¹⁷⁶ has concluded that cross termination is not particularly favored over homotermination in S-MMA copolymerization.





Several experimental studies on S-MMA copolymerization have appeared: all suggest predominant combination.¹⁸¹⁻¹⁸³ Ohtani *et al.*¹⁸³ analyzed the end groups of PSMMA (60°C, AIBN, chloroform) by pyrolysis-gas chromatography to find values for the number of end groups per molecule of between 1.56-1.77 (increasing with polymer M_n) which corresponds to an overall k_{td}/k_{tc} of between 0.39 and 0.21. Estimation of k_{td}/k_{tc} for cross termination requires knowledge of the

extents of homo- and cross termination. Bevington *et al.*¹⁸¹ examined S-MMA copolymerization (60°C, benzene) using the radiotracer method and found that the cross termination reaction involves predominantly combination (k_{td}/k_{tc} for the homotermination processes were taken to be 0 and 5.67 for chains ending in S and MMA respectively). Chen *et al.*¹⁸² conducted an analysis of polymerization kinetics and came to a similar conclusion. Both groups assumed a "chemical control model" for termination (Section 7.4.1) and the results may need to be reinterpreted.

7.4.3.2 Poly(methacrylonitrile-co-styrene)

Analysis of the products from the thermal decomposition of the mixed azo compound 6 showed that in the cross-reaction of radicals 5 and 7 $k_{td}/k_{tc}(90^{\circ}\text{C})$ is 0.61.¹⁷⁹ This study also found that in disproportionation, hydrogen transfer from 5 to 7 is *ca* 2.2 times more frequent than transfer from 7 to 5. Both self-reactions involve predominantly combination (Scheme 7.14). The values of $k_{td}/k_{tc}(80^{\circ}\text{C})$ are 0.16 and 0.05 for radicals 5 (Section 5.2.2.1.1) and 7 (Section 5.2.2.1.3) respectively. It is clear that values of k_{td}/k_{tc} for homotermination cannot be used as a guide to the value for k_{td}/k_{tc} in cross-termination.



Scheme 7.14

The reaction of oligostyrene radicals with cyanoisopropyl radicals (7) has been studied by several groups and reported to give exclusively combination (98°C, toluene),^{180,184} or mainly combination (60°C, ethyl acetate;¹⁸⁵ 98°C, toluene¹⁸⁶). Moad *et al.*¹⁸⁶ examined S oligomerization in toluene at 98°C using high concentrations of AIBN as initiator. While the major products arose from combination, they also isolated and identified small amounts of disproportionation products thus demonstrating that disproportionation does occur.

7.4.3.3 Poly(butyl methacrylate-co-methacrylonitrile)

Barton *et al.*¹⁸⁷ have reported that primary radical termination between PBMA• and cyanoisopropyl radicals (7) involves largely disproportionation.

7.4.3.4 Poly(butyl methacrylate-co-methyl methacrylate)



The value of $k_{td}/k_{tc}(80^{\circ}\text{C})$ in the cross-reaction between radicals 4 and 8 has been examined.¹⁷⁵ This system is a model for cross-termination in MMA-BMA copolymerization. The value of k_{td}/k_{tc} (1.22) is similar to that found for the self-reaction of 8 (1.17) and much larger than that for the self-reaction of 4 (0.78). There is a small preference (*ca* 1.4 fold) for the transfer of hydrogen from the butyl ester (8) to the methyl ester (4).

7.4.3.5 Poly(ethylene-co-methacrylonitrile)

Guth and Heitz¹⁷⁷ have reported that primary radical termination between PEradicals and cyanoisopropyl radicals (7) involves substantial disproportionation. Both homotermination processes involve largely combination (Sections 5.2.2.1.3 and 5.2.2.1.4).

7.5 Functional and End-Functional Polymers

Functional and end-functional polymers are precursors to block and graft copolymers and, in some cases, polymer networks. Copolymers with in-chain functionality may be simply prepared in copolymerizations by using a functional monomer. However, obtaining a desired distribution requires consideration of the chain statistics and, for low molecular weight polymers, the specificity of the initiation and termination processes. These issues are discussed in Section 7.5.6

End-functional polymers, including telechelic^{*} and other di-end functional polymers, can be produced by conventional radical polymerization with the aid of functional initiators (Section 7.5.1), chain transfer agents (Section 7.5.2), monomers (Section 7.5.4) or inhibitors (Section 7.5.5). Recent advances in our understanding of radical polymerization offer greater control of these reactions and hence of the polymer functionality. Reviews on the synthesis of end-functional polymers include those by Colombani,¹⁸⁸ Tezuka,¹⁸⁹ Ebdon,¹⁹⁰ Boutevin,¹⁹¹ Heitz,¹⁸⁰ Nguyen and Maréchal,¹⁹² Brosse *et al.*,¹⁹³ and French.¹⁹⁴

^{*} A telechelic polymer is a di-end-functional polymer where both ends possess the same functionality.

Living polymerization processes lend themselves to the synthesis of end functional polymers; their use in this context is described in Chapter 9. In this section we limit discussion to processes based on conventional radical polymerization.

7.5.1 Functional Initiators

Predominantly di-end-functional polymers may be prepared by conducting polymerizations with high concentrations of a functional initiator. Some of the first commercial products of this class, carboxy and hydroxy-terminated polybutadienes, were produced by this route.¹⁹⁴

The synthesis of telechelics by what Tobolsky¹⁹⁵ termed dead-end polymerization is described in several reviews.^{191,193} In dead-end polymerization very high initiator concentrations and (usually) high reaction temperatures are used. Conversion ceases before complete utilization of the monomer because of depletion of the initiator. Target molecular weights are low (1000-5000) and termination may be mainly by primary radical termination. The first use of this methodology to prepare telechelic polystyrene was reported by Guth and Heitz.¹⁷⁷

When a polymer is prepared by radical polymerization, the initiator derived chain-end functionality will depend on the relative significance and specificity of the various chain end forming reactions. Thus, for the formation of telechelic polymers:

- (a) The reaction of the initiator-derived radicals with monomer must involve double bond addition (*i.e.* no primary radical transfer).
- (b) Secondary radical formation (e.g. by β -scission in the case of acyloxy or alkoxy radicals) should either be negligible or not involve loss of the desired functionality.
- (c) Chain end formation by chain transfer to monomer, polymer, solvent, *etc.* must be minimal. Chain transfer to initiator may be tolerated if the initiator functionality is transferred.
- (d) All radical-radical termination (reaction with primary or propagating radicals) should involve combination.

These conditions severely limit the range of initiators and monomers that can be used and require that attention to reaction conditions is of paramount importance. The relatively low incidence of side reactions associated with the use of azo-compounds (Section 3.3.1) has led to these initiators being favored for this application. Functional azo compounds used in telechelic syntheses include 9, ¹⁹⁶⁻¹⁹⁸ 10^{199,200} and 11^{201,202}. The acylazide end groups formed with initiator 11 may be thermally transformed to isocyanate ends.^{201,202}

$$\begin{array}{ccc} & & CN & & CN \\ & & & R(CH_2)_2 - \overset{-}{C} - N = N - \overset{-}{C} - (CH_2)_2 R \\ & & & CH_3 & & CH_3 \\ & & & CH_3 \\ \mathbf{9} & R = CO_2 H & \mathbf{10} & R = CH_2 OH & \mathbf{11} & R = CON_3 \end{array}$$

Simple azo-compounds (AIBN or AIBMe) have also been used to produce telechelic polymers.^{177,184,194} The nitrile and ester functions can be elaborated to reactive carboxy, hydroxy or amino groups and used in polyester or polyurethane formation (*e.g.* Scheme 7.15). Functionalities (number of end groups/molecule) of 1.7 for PE and 2.0 for PS were reported. The latter number seems high given that PS• is known to give some disproportionation both in reaction with cyanoisopropyl radicals ($k_{\rm td}/k_{\rm tc}(90^{\circ}\text{C}) = 0.61$, Section 7.4.3.2) and in self reaction (Section 5.2.3.1.3). A possible explanation is that the unsaturated by-product from cage-disproportionation (*e.g.* MAN from AIBN, Section 3.3.1.1.3)¹⁸⁶ may copolymerize. This may result in an apparent functionality of ≥ 2 .



Scheme 7.15

There have been many studies on the applications of peroxide initiators to the synthesis of α, ω -dihydroxy and α, ω -dicarboxy oligomers. Succinic (12, n=2) and glutaric acid peroxides (12, n=3) have been used to synthesize carboxy end-functional polybutadiene.¹⁹⁴ This use of peroxides is complicated by the tendency of acyloxy radicals and alkoxy radicals to undergo β -scission and by the various pathways that may compete with double bond addition (Section 3.4.2). However, alkoxycarbonyloxy radicals undergo β -scission only slowly (Section 3.4.2.2.2) and peroxydicarbonates have been used to form polymers with carbonate end groups²⁰³ Guth and Heitz¹⁷⁷ reported that ethylene polymerized with peroxydicarbonate initiator has a functionality of only *ca* 1.1. As explanation, they proposed that primary radical termination involving the alkoxycarbonyloxy radical involves disproportionation rather than coupling. The carbonate ends were hydrolyzed to hydroxy ends.²⁰³

$$\begin{bmatrix} 0 & 0 \\ HO & (CH_2)_0 & 0 \end{bmatrix}_2$$

The use of ring substituted diacyl peroxides has also been reported.²⁰⁴ Both the aryl and aroyloxy ends possess the desired functionality. Other initiators used in this context include peroxides (*e.g.* hydrogen peroxide),

Disulfide derivatives and hexasubstituted ethanes²⁰⁵ may also be used in this context to make end-functional polymers and block copolymers. The use of dithiuram disulfides as thermal initiators was explored by Clouet, Nair and coworkers.²⁰⁶ Chain ends are formed by primary radical termination and by transfer to the dithiuram disulfide. The chain ends formed are thermally stable under normal polymerization conditions. The use of similar compounds as photo-iniferters, when some living characteristics may be achieved, is described in Section 9.3.2.1.1.

7.5.2 Functional Transfer Agents

Suitably functionalized transfer agents offer a route to end-functional and block and graft polymers.^{180,191,207-209} Living polymerization processes involving degenerate or reversible chain transfer (*e.g.* RAFT) are discussed in Section 9.5. For radical polymerization in the presence of a transfer agent, it must be remembered that the initiation and termination steps will always be responsible for a fraction of the chain ends. Therefore, to achieve the highest degree of functionality, an initiator should be chosen which gives the same type of end group as the transfer agent.

Chains with undesired functionality from termination by combination or disproportionation cannot be totally avoided. In attempts to prepare a monofunctional polymer, any termination by combination will give rise to a difunctional impurity. Similarly, when a difunctional polymer is required, termination by disproportionation will yield a monofunctional impurity. The amount of termination by radical-radical reactions can be minimized by using the lowest practical rate of initiation (and of polymerization). Computer modeling has been used as a means of predicting the sources of chain ends during polymerization and examining their dependence on reaction conditions (Section 7.5.6).^{210,211} The main limitations on accuracy are the precision of rate constants which characterize the polymerization.

Depending on the choice of transfer agent, mono- or di-end-functional polymers may be produced. Addition-fragmentation transfer agents such as functional allyl sulfides (Scheme 7.16), benzyl ethers and macromonomers have application in this context (Section 6.2.3).²¹²⁻²¹⁶ The synthesis of PEO-block copolymers by making use of PEO functional allyl peroxides (and other transfer agents) has been described by Businelli *et al.*²¹⁷ Boutevin *et al.*^{218.219} have described the telomerization of unsaturated alcohols with mercaptoethanol or dithiols to produce telechelic diols in high yield.



7.5.3 Thiol-ene Polymerization

Thiol-ene polymerization was first reported in 1938.²²⁰ In this process, a polymer chain is built up by a sequence of thiyl radical addition and chain transfer steps (Scheme 7.17). The thiol-ene process is unique amongst radical polymerizations in that, while it is a radical chain process, the rate of molecular weight increase is more typical of a step-growth polymerization. Polymers ideally consist of alternating residues derived from the diene and the dithiol. However, when dienes with high k_p and relatively low k_t monomers (e.g. acrylates) are used, short sequences of units derived from the diene are sometimes formed.

Addition





Scheme 7.17

Dithiols and dienes may react spontaneously to afford dithiols or dienes depending on the monomer dithiol ratio.²²¹ However, the precise mechanism of radical formation is not known. More commonly, photoinitiation or conventional radical initiators are employed. The initiation process requires formation of a radical to abstract from thiol or add to the diene then propagation can occur according to the steps shown in Scheme 7.17 until termination occurs by radicalradical reaction. Termination is usually written as involving the monomer-derived radicals. The process is remarkably tolerant of oxygen and impurities. The kinetics of the thiol-ene photopolymerization have been studied by Bowman and coworkers.222,223

The process may be used to form linear polymers. Nuyken and Völkel^{224,225} described a method for telechelic production, based on the radical initiated reaction of difunctional transfer agents with dienes (e.g. divinyl benzene (13),dimethacrylate esters). However, currently the most common use of thiol-ene

polymerization is to form network polymers in a photoinitiated process.²²⁶. Dienes employed include divinyl benzene (13), diethylene glycol diacrylate (15) and a variety of nonconjugated dienes. The latter include many monomers not commonly used in conventional radical polymerization such as diallyl trimethylolmethane (16) and the bis-norbornene derivative (17). Diacetylenes (*e.g.* 14) have also been used. The thiols used include simple aliphatic and aromatic dithiols (*e.g.* octanedithiol). Network polymers typically incorporate a compound with multiple thiol groups, for example, the tetrathiol 18.



One may envisage polymerizations analogous to the thiol-ene process using other bis- or multi transfer agents (*e.g.* radical-induced hydrosilylation between bis-silanes and dienes). However, none has been described or achieved significance.

7.5.4 Functional Monomers

Ketene acetals and related monomers undergo ring-opening polymerization to produce polyesters (Section 4.4.2.2). Copolymerization of such monomers with, for example, S (Scheme 7.18), and basic hydrolysis of the ester linkages in the resultant copolymer offers a route to α,ω -difunctional polymers.²²⁷ A limitation on the use of these particular ring-opening monomers is that they are relatively unreactive towards propagating radicals (*e.g.* PS•) thus rates of copolymerization are slow.



Scheme 7.18

Other ring-opening copolymerizations (of, for example, the cyclic allyl sulfide **19**), also yield polymers with in-chain ester groups and copolymerize more readily (Section 4.4.2.2).



Ebdon and coworkers²²⁸⁻²³² have reported telechelic synthesis by a process that involves copolymerizing butadiene or acetylene derivatives to form polymers with internal unsaturation. Ozonolysis of these polymers yields di-end functional polymers. The α,ω -dicarboxylic acid telechelic was prepared from poly(S-*stat*-B) (Scheme 7.19). Precautions were necessary to stop degradation of the PS chains during ozonolysis.²²⁸ The presence of pendant carboxylic acid groups, formed by ozonolysis of 1,2-diene units, was not reported.



Scheme 7.19

End-functional polymers are also produced by copolymerizations of monosubstituted monomers with α -methylvinyl or other monomers with high transfer constants in the presence of catalytic chain transfer agents (Section 6.2.5).²³³⁻²³⁶ Thus, copolymerization of BA with as little as 2% AMS in the presence of cobaloxime provides PBA with AMS at the chain end.²³⁷

380

7.5.5 Functional Inhibitors

Inhibitors (Section 5.3), including transition metal complexes and nitroxides, may be used to prepare mono-end-functional polymers. If an appropriate initiator is employed, di-end-functional polymers are also possible.

Only one polymer molecule is produced per mole of inhibitor. The inhibitor must be at least equimolar with the number of chains formed. Concentrations must be chosen (usually very low) to give the desired molecular weight.

7.5.6 Compositional Heterogeneity in Functional Copolymers

The copolymer composition equation only provides the average composition. Not all chains have the same composition. There is a statistical distribution of monomers determined by the reactivity ratios. When chains are short, compositional heterogeneity can mean that not all chains will contain all monomers.

In early work, while compositional heterogeneity was recognized and could be predicted, it was difficult to measure. Now, methods such as GPC combined with NMR and/or MALDI,^{238,239} GPC coupled with FTIR²⁴⁰ and two dimensional HPLC or GPC²⁴¹⁻²⁴⁵ can provide a direct measure of the composition distribution.

Chain compositional heterogeneity is of particular relevance to functional copolymers which find widespread use in the coatings and adhesives industries.^{13,240,246} In these applications, the functional copolymer and a crosslinking agent are applied together and are cured to form a network polymer. The functional copolymers are based on functional monomers with reactive groups (*e.g.* OH). It is desirable that all copolymer molecules have a functionality of at least two. Nonfunctional polymer will not be incorporated and could plasticize the network or be exuded from the polymer. Monofunctional polymers are not involved in crosslink formation and will produce dangling ends.

Various factors are important in determining the composition and molecular weight distribution of multicomponent copolymers (*e.g.* monomer reactivity ratios, reaction conditions). Stockmayer²⁴⁷ was one of the first to report on the problem and presented formulae for calculating the instantaneous copolymer composition as a function of chain length. Others^{11,12,248-250} have examined the variation in copolymer composition with chain length by computer simulation. One method of ensuring a functionality of at least one is to use a functional initiator or transfer agent.

The influence of selectivity in the initiation, termination or chain transfer steps on the distribution of monomer units within the copolymer chain is usually neglected. Galbraith *et al.*¹¹ provided the first detailed analysis of these factors. They applied Monte Carlo simulation to examine the influence of the initiation and termination steps on the compositional heterogeneity and molecular weight distribution of binary and ternary copolymers. Spurling *et al.*²⁵⁰ extended this treatment to consider additionally the effects of conversion on compositional heterogeneity.

The ends of polymer chains are often not representative of the overall chain composition. This arises because the initiator and transfer agent-derived radicals can show a high degree of selectivity for reaction with a particular monomer type (Section 3.4). Similarly, there is specificity in chain termination. Transfer agents show a marked preference for particular propagating species (Section 6.2.2 and 6.2.3). The kinetics of copolymerization are such that the probability for termination of a given chain by radical-radical reaction also has a marked dependence on the nature of the last added units (Section 7.4.3).

The effect of the initiation and termination processes on compositional heterogeneity can be seen in data presented in Figure 7.3 and Figure 7.4. The data come from a computer simulation of the synthesis of a hydroxy functional oligomer prepared from S, BA, and HEA with a thiol chain transfer agent. The recipe is similar to those used in some coatings applications.



Figure 7.3 Distribution of monomers [HEA(— — —), BA(——), S (------)] within chains as a function of chain length for a HEA:BA:S copolymer prepared with butanethiol chain transfer agent.²⁵⁰

In this copolymerization, most termination is by chain transfer and most chains are initiated by transfer agent-derived radicals. The thiyl radicals generated from the transfer agent react faster with S than they do with acrylate esters (Scheme 7.20).



Scheme 7.20



Figure 7.4 Molecular weight distributions for HEA:BA:S copolymer prepared with butanethiol chain transfer agent: (a) all chains (———); (b) chains without HEA (— — —).²⁵⁰

The thiol shows a preference to react with propagating radicals with a terminal S unit (Scheme 7.21). This selectivity is due both to chemospecificity in the reaction with thiol and to the relative concentrations of the various propagating species (determined by the reactivity ratios).

A preponderance of chains that both begin and end in S results and this means that short chains are much richer in S than in the acrylic monomers (Figure 7.3). This also has an influence on the fraction of chains that contain the functional monomer (Figure 7.4). The fraction of HEA in very short chains is much less than that in the polymer as a whole and a significant fraction of these short chains contain no functional monomer.



Scheme 7.21

384

In this copolymerization, the reactivity ratios are such that there is a tendency for S and the acrylic monomers to alternate in the chain. This, in combination with the above-mentioned specificity in the initiation and termination steps, causes chains with an odd number of units to dominate over those with an even number of units.

It is possible to exercise control over this form of compositional heterogeneity (*i.e.* the functionality distribution) by careful selection of the functional monomer and/or the transfer agent taking into account the reactivities of the radical species, monomers, and transfer agents, and their functionality.^{11,250} Relative reactivities of initiator and transfer agent-derived radicals towards monomers are summarized in Section 3.4. Some values for transfer constants are provided in Chapter 6.

The overall composition at low conversion of binary copolymers formed in the presence of a chain transfer agent can be predicted analytically using an expression analogous to that used to describe terpolymerization where one monomer does not undergo propagation (Section 7.3.2.4).²³⁶ Making the appropriate substitutions, eq. 37 becomes eq. 70:

$$d\mathbf{M}_{A}: d\mathbf{M}_{B}: d\mathbf{T} = P_{A}: P_{B}: P_{T}$$

$$= \mathbf{M}_{A} \left[\frac{R\mathbf{M}_{A}}{r_{BA}} + \frac{\mathbf{M}_{B}}{r_{BA}} + C_{B}RT \right] \left[\mathbf{M}_{A} + \frac{\mathbf{M}_{B}}{r_{AB}} + C_{A}T \right]$$

$$: \mathbf{M}_{B} \left[\frac{R\mathbf{M}_{A}}{r_{AB}} + \frac{\mathbf{M}_{B}}{r_{AB}} + C_{A}T \right] \left[\mathbf{M}_{B} + \frac{\mathbf{M}_{A}}{r_{BA}} + C_{B}T \right]$$

$$: T \left[\frac{\mathbf{M}_{A}}{C_{A}r_{BA}} + \frac{\mathbf{M}_{B}}{r_{BC}r_{AB}} + C_{A}C_{B}T \right] \left[R\mathbf{M}_{A} + \mathbf{M}_{B} \right]$$
(70)

where T is the concentration of transfer agent, C_A and C_B are the transfer constants of the transfer agent in polymerizations of monomer A and B respectively and $R=k_{iA}/k_{iB}$ is the relative rate of initiation by the transfer agent-derived radical. The average molecular weight is given by eq. 71.

$$X_{n} \approx \frac{1}{P_{T}} \tag{71}$$

The T containing sequences can be evaluated using expressions analogous to those described in Section 7.3.2.1 to provide the chain end compositions and the chain length distribution.

7.6 Block & Graft Copolymerization

Many block and graft copolymer syntheses involve radical polymerization at some stage of the overall preparation. This section deals with direct syntheses of

block and graft copolymers by conventional radical processes. Formation of block and graft copolymers by living radical polymerization is discussed in Chapter 9.

In the standard nomenclature $[poly(M_A)-graft-poly(M_B)]$ the first named monomer(s) form the backbone while those named second are the grafts or arms. Thus, PMMA-graft-PS indicates a backbone of PMMA and grafts of PS.

Graft copolymerizations are categorized according to their method of formation into three main types.²⁵¹

- (a) Grafting onto, where reactive functionality on one polymer chain reacts with functionality on a second chain. Condensation of polymer bound functionality with end-functional polymers is a grafting onto process. Processes for the formation of functional polymers are discussed in Section 7.5.
- (b) Grafting from, where active sites are created on the polymer chain from which new polymerization is initiated.
- (c) Grafting through, where a propagating species reacts with pendant unsaturation on another polymer chain. The copolymerization of macromonomers is a grafting through process (Section 7.6.5).

Four types of 'grafting from' processes are distinguished by the mechanism of radical formation.

- (a) Formation of radicals on or at the end of a polymer chain by decomposition of bonded initiator functionality (often an azo or peroxide linkage) (Section 7.6.1).
- (b) Formation of radicals by transformation of a polymer bound functionality to radicals typically by some form of redox or multi-step process (Section 7.6.2).
- (c) Formation of radicals on non-functional polymer substrates by irradiation with, for example, γ -rays or an electron beam (Section 7.6.3)
- (d) Formation of radicals on non-functional polymer substrates by radicals abstracting hydrogen (Section 7.6.4). Transfer to polymer during polymerization also causes branching in a grafting from process (Section 6.2.7).

Specific forms of graft copolymers may go under different names.

- (a) Branched polymers where the backbone and the arms are of the same composition
- (b) Comb polymers where the arms are of uniform length (*e.g.* PMMA-*comb*-PS)
- (c) Hyperbranched polymers when there are branches on branches.

7.6.1 Polymeric and Multifunctional Initiators

Multifunctional initiators contain two or more radical generating functions within the one molecule. The chemistry of these initiators has been the subject of several reviews.²⁵²⁻²⁵⁵ As long as the radical generating functions are sufficiently remote their decompositions are independent events. If decomposition occurs

under sufficiently different reaction conditions, these initiators can be used to form polymers with end groups that contain initiator moieties. The polymeric initiators can be subsequently utilized to yield higher molecular weight polymers, to achieve higher degrees of conversion, and in the production of block and graft copolymers.



The multifunctional initiators may be di- and tri-, azo- or peroxy-compounds of defined structure (e.g. 20^{256}) or they may be polymeric azo- or peroxycompounds where the radical generating functions may be present as side chains²⁵⁷ or as part of the polymer backbone.²⁵⁸⁻²⁶¹ Thus, amphiphilic block copolymers were synthesized using the polymeric initiator **21** formed from the reaction between an α, ω -diol and AIBN (Scheme 7.22).²⁶² Some further examples of multifunctional initiators were mentioned in Section 3.3.3.2. It is also possible to produce less well-defined multifunctional initiators containing peroxide functionality from a polymer substrate by autoxidation or by ozonolysis.²⁶³

The success of the multifunctional initiators in the preparation of block and graft copolymers depends critically on the kinetics and mechanism of radical production. In particular, the initiator efficiency, the susceptibility to and mechanism of transfer to initiator, and the relative stability of the various radical generating functions. Each of these factors has a substantial influence on the nature and homogeneity of the polymer formed. Features of the kinetics of polymerizations initiated by multifunctional initiators have been modeled by O'Driscoll and Bevington²⁶⁴ and Choi and Lei.²⁶⁵

$$\begin{array}{cccc} CN & CN \\ CH_3 - C - N = N - C - CH_3 \\ CH_3 & CH_3 \end{array} \xrightarrow{HO \{CH_2 CH_2 \cdot O\}_{H}} HO \left\{ \begin{array}{cccc} (CH_2 CH_2 O)_{D} & CO & COO \\ (CH_2 CH_2 O)_{D} & CO & COO \\ CH_3 - C - N = N - C - CH_3 \\ CH_3 & CH_3 \end{array} \right\}_{m} \\ \end{array}$$

Scheme 7.22

A final class of multifunctional initiators is based on the use a (multi)functional polymer and a low molecular weight redox agent. Radicals on the polymer chain are generated from the polymer bound functionality by a redox reaction. Ideally, no free initiating species are formed. The best known of this class are the polyol-redox and related systems. Polymers containing hydroxy or glycol and related functionality are subject to one electron oxidation by species such as ceric ions or periodate (Scheme 7.23).^{266,267} Substrates such as cellulose,

386

chitin and poly(vinyl alcohol) provide graft copolymers. The chemistry is briefly discussed in Section 3.3.5.2. Hydroxy end-functional polymers such as poly(ethylene glycol) and poly(4-hydroxybutyrate) yield block copolymers. A further example of this approach, which makes use of a halogen functional polymer, can be found in Section 7.6.2.



Scheme 7.23

7.6.2 Transformation Reactions

Block and graft copolymer syntheses by what have come to be known as 'transformation reactions' involve the preparation of polymeric species by some mechanism which leaves a terminal functionality that allows polymerization to be continued by another mechanism as shown schematically in Scheme 7.24. Examples of transformation of anionic, cationic, Ziegler-Natta, and group transfer polymerization to radical polymerization have been reported. Examples of transformation of radical to ionic polymerization are also known. Additional examples that involving transformation to or from living radical polymerization (NMP, ATRP or RAFT) can be found in Chapter 9. The success of the transformation reactions depends on the efficiency of the transformation process and the avoidance of processes that might lead to concurrent homopolymerization. The general area of block polymer synthesis through 'transformation reactions' has been reviewed by Stewart²⁶⁸, Schue,²⁶⁹ Abadie and Ourahmoune.²⁷⁰ and Eastmond²⁷¹ The mechanism of termination also plays an important role in determining the type of block copolymers that may be formed. If standard polymerization conditions are employed, an ABA or AB block may be produced depending on whether termination occurs by combination or disproportionation.

Scheme 7.24 (* - active center; *e.g.* anion, cation, radical)

One of the earliest examples of this methodology involves the reaction of a polymeric anion (formed by living anionic polymerization) with molecular oxygen to form a polymeric hydroperoxide which can be decomposed either thermally or, preferably, in a redox reaction to initiate block polymer formation with a second monomer (Scheme 7.25). However, the usual complications associated with initiation by hydroperoxides apply (Section 3.3.2.5).

$$\begin{array}{cccc} & & & & & & \\ & & & & \\ & & & & \\ & & & & \\ & & & \\ & & & \\ & & & \\ & & & \\ & & & \\ & & & \\ & & & \\ & & & \\ & & & \\ & & & \\ & & & \\ & & & \\$$

The reactions of polymeric anions with appropriate azo-compounds or peroxides to form polymeric initiators provide other examples of anion-radical transformation (*e.g.* Scheme 7.26).^{270,272-274} However, the polymeric azo and peroxy compounds have limited utility in block copolymer synthesis because of the poor efficiency of radical generation from the polymeric initiators (7.5.1).



Tung *et al.*²⁷⁵ have reported on the use of a polymeric thiol transfer agent for use in block copolymer production. Various methods have been used for the anion \rightarrow thiol conversion. Near quantitative yields of thiol are reported to have been obtained by terminating anionic polymerization with ethylene sulfide and derivatives (Scheme 7.27). Transfer constants for the polymeric thiols are reported to be similar to those of analogous low molecular weight compounds.²⁷⁵

Scheme 7.27

The preparation of ABA triblock polymers requires use of a telechelic bisthiol prepared by termination of anionic polymerization initiated by a difunctional initiator. The relative yields of homopolymer, di- and triblock obtained in these experiments depend critically on conversion.²⁷⁵

Richards *et al.* carried out extensive studies on the use of mercury,^{276,277} lead^{278,279} and silver compounds to terminate anionic polymerization and form polymeric organometallic species which can be used to initiate polymerization.

Bamford, Eastmond and coworkers²⁸⁰⁻²⁸⁵ have employed metal complexpolymeric halide redox systems to initiate block and graft copolymerization. The polymeric halides can be synthesized by a variety of techniques, including radical polymerization,²⁸¹ anionic polymerization (Scheme 7.28),²⁸⁰

388

group transfer polymerization (Scheme 7.29),²⁸⁴



cationic polymerization (Scheme 7.30),²⁸³



Scheme 7.30

and functionalization of a polymer with carboxylic acid, hydroxy, amino, or etherurethane groups with a haloisocyanate (Scheme 7.31).²⁸⁶



Scheme 7.31

The efficiency of the halide \rightarrow radical transformation is reported to be near quantitative. The yield of block or graft is then limited by the efficiency of the halide synthesis. Whether AB or ABA blocks are formed depends on the termination mechanism. Similar halo-compounds have been used to initiate ATRP (Section 9.4).

7.6.3 Radiation-Induced Grafting Processes

Radiation-induced grafting and curing processes have been discussed in a number of reviews.^{263,287-291} The process is widely used for surface modification. Recent applications are the modification of fuel cell membranes and improving

surface biocompatibility. Common substrates for radiation-induced grafting are the poly(fluoro-olefins) and the polyolefins. The usual radiation sources in this context are γ -rays (*e.g.* a ⁶⁰Co source) and electron beams.

The detailed chemistry of radiation grafting has, in most cases, not been rigorously established. Process characterization is complicated by the fact that often only surface layers are involved and, in other cases, by the substrates being cross-linked or intractable.

Three main processes for radiation-induced grafting are described:

- (a) Pre-irradiation the substrate is irradiated (in an 'inert' environment) then brought in contact with monomer.
- (b) Peroxidation the substrate is irradiated in an atmosphere of oxygen or air to form peroxidic groups, which are then thermally decomposed in the presence of monomer.
- (c) Mutual irradiation the substrate and monomer are brought together then irradiated.

These processes compete with radiation-induced crosslinking, scission and, for case (c), polymerization.

The radiation sensitivity of polymers and monomers is characterized by a G value; the number of radicals formed per 100 e.v. (16 aJ) absorbed. Radiation sensitive groups include -COOH, C-halogen, -SO₂-, -NH₂ and -C=C-. Radiation resistant groups are aromatic rings. It appears that the presence of aromatic moieties also offers some degree of radiation protection to the polymer chain as a whole.

7.6.4 Radical-Induced Grafting Processes

Radical induced grafting may be carried out in solution, in the melt phase,^{292-²⁹⁵ or as a solid state process.²⁹⁶ This section will focus on melt phase grafting to polyolefin substrates but many of the considerations are generic. The direct grafting of monomers onto polymers, in particular polyolefins, in the melt phase by reactive extrusion has been widely studied. Most recently, the subject has been reviewed by $Moad^{293}$ and by Russell.²⁹² More details on reactive extrusion as a technique can be found in volumes edited by Xanthos,²⁹⁴ Al Malaika²⁹⁵ and Baker *et al.*²⁹⁷ The process most often involves combining a free-radical initiator (most commonly a peroxide) and a monomer or macromonomer with the polyolefin as they are conveyed through the extruder. Monomers commonly used in this context include: MAII (Section 7.6.4.1), maleimide derivatives and maleate esters (Section 7.6.4.2), (meth)acrylic acid and (meth)acrylate esters (Section 7.6.4.3), S, AMS and derivatives (Section 7.6.4.4), vinylsilanes (Section 7.6.4.5) and vinyl oxazolines (Section 7.6.4.6).}

A major issue is the control of the side reactions that accompany grafting. These reactions include radical-induced degradation of the substrate by crosslinking and/or chain scission and homopolymerization of the graftee monomer.

Polyethylenes (HDPE, LDPE, LLDPE, high E content - EP) are prone to branching or crosslinking caused by radical-radical combination. This process is characterized by the formation of gels or a partially insoluble product. Polypropylene (PP) and low-density ethylene/ α -olefin copolymers may also undergo crosslinking under some conditions. However, the most oftenencountered side reaction is degradation caused by the initially formed radical undergoing β -scission. This susceptibility to chain scission is well documented and is used to advantage in the synthesis of controlled rheology PP.

A major challenge is then to devise conditions so as to maximize grafting and minimize or control these side reactions. Some discussion of many of these parameters is provided in the reviews mentioned above. It is significant that many recent publications and patents in the area of reactive extrusion relate, not to the development of new reactions or processes, but to the selection of operating parameters.

The monomer acts to trap radicals that might otherwise undergo chain scission or crosslinking. More degradation is seen with less reactive monomers. Use of a higher monomer concentration may result in less degradation of the polyolefin substrate. However, it is often found that the dependence of grafting yield on monomer concentration passes through a maximum. If the monomer concentration becomes too high, phase separation can occur. This results in reduced grafting yields and an increased likelihood for homopolymerization. In these circumstances, higher graft levels can better be achieved by multipoint/multipass addition of monomer and initiator or by use of a comonomer or other coagent.

It is also necessary to select the initiator according to the particular monomer(s) and the substrate. Factors to consider in this context, aside from initiator half-lives and decomposition rates, are the partition coefficient of the initiator between the monomer and polyolefin phases and the reactivity of the monomer *vs* the polyolefin towards the initiator-derived radicals.



Grafting is most commonly carried out with peroxides that are sources of t-alkoxy radicals (e.g. 22-25). At the high temperatures usually used, the extent of

β-scission is likely to be significant thus radicals involved in abstraction are likely to be a mixture of *t*-alkoxy and alkyl radicals. Several authors^{298,299} have pointed out that R₃CO-H and CH₃-H bond strengths are similar (Section 2.2.2). Even accepting the validity of the Evans-Polyani approach in this context, it must also be noted that C-C bonds are significantly stronger than R₃CO-C bonds (Section 2.4.6). Thus, methyl radical is anticipated to have a greater propensity for addition over abstraction than a *t*-alkoxy radical. The tendency for addition *vs* abstraction is greater for higher alkyl radicals. Abstraction:addition ratios are also temperature dependent (Section 3.2.4). Lower temperatures favor abstraction over addition and, for *t*-alkoxy radicals, both of these reactions are favored over β-scission. The regiospecificity of hydrogen abstraction by *t*-alkoxy and methyl radical is also very different. The methyl radical shows a much greater specificity for methine>methylene>methyl.²⁹³

While it is important that the initiator-derived radicals react preferentially with the polyolefin substrate, the specificity shown by the initiator-derived radicals may be of only minor importance in determining the ultimate product distribution. The species that abstracts hydrogen is, in many cases, not an initiator-derived radical. This follows from the observation that up to 20 monomer units may be grafted per initiator-derived radical generated.^{300,301} Care must be taken in interpreting such data as it is not always clear whether a high number of monomer units grafted per radical generated means a long graft length or a large number of graft sites. Nonetheless, it is clear that in some instances, where graft lengths have been characterized, that most abstraction must occur by way of the propagating species formed by addition of monomer. In these cases, chain transfer is also a major factor in limiting the length of the grafted chain.

An alternative to the direct use of peroxides in monomer grafting is to first functionalize the polymer with initiator or transfer agent functionality.

7.6.4.1 Maleic anhydride graft polyolefins

With a history of more than 25 years, the free radical-induced grafting of MAH onto polyolefin substrates is one of the most studied polyolefin modification processes.^{293,298,302} The process has been carried out in the melt phase, in various forms of extruders and batch mixers, and there are numerous patents covering various aspects of the process. It has also been carried out successfully in solution and in the solid state. The materials have a range of applications including their use as precursors to graft copolymers, either directly, or during the preparation of blends.²⁹⁷

Many of the structures for MAH-modified polyolefins that appear in the literature are wholly speculative, and are based on a proposed mechanism for the grafting reaction rather than an analysis of the reaction or reaction products. In early work, product characterization took the form of determining overall grafting levels by titration or IR spectroscopy. In more recent work, with the availability of

392

additional characterization techniques, it has been shown that the structure depends strongly on the particular polyolefin substrate and the synthesis conditions.^{303,304}

In early work it was often assumed, without specific proof, that MAH was grafted to polyolefins as single units (Scheme 7.32). This followed from its known sluggishness in homopolymerization and from a consideration of ceiling temperature. Recent NMR studies indicate that MAH is attached to PP³⁰³ and model substrates²⁹² as single units. However, other studies suggest that a fraction of units may be grouped either as oligo-MAH grafts^{305,306} or as adjacent grafts formed by sequential intramolecular abstraction and grafting (Scheme 7.33).²⁹² Differing reaction conditions used in the various works confuses analysis of the situation.²⁹³



Scheme 7.32

Typical levels of MAH in grafted PP of 0.5-2 wt % correspond to only one or two units per chain. If the MAH units are grouped it follows that many chains may contain no MAH. It has also been suggested that for PP all MAH may appear at the chain ends. This is rationalized in terms of the reaction of mid chain radicals with MAH always being followed by intramolecular chain transfer and chain scission as shown in Scheme 7.34. This pathway would be favored by the slow rate of homopropagation of MAH.

394

Substantial work has also been carried out on grafting to HDPE,^{303,307} LLDPE^{303,308-310} and EP copolymers.^{303,311} In many early studies, MAH grafting onto PE and ethylene copolymers seemed always to be accompanied by some degree of crosslinking as indicated by a partially insoluble product. However, the recent literature demonstrates that extrusion conditions can be designed to avoid or minimize crosslinking and provide a completely soluble product and still obtain very high grafting yields (~ 80%).³⁰⁸ The different outcome in these latter studies is attributed to the differences in the effectiveness of mixing, initiator concentrations, and the method of reagent introduction.³¹²





In principle the MAH may be attached to LDPE or LLDPE at methine sites or at methylene sites. Heinen et al.³⁰³ have used ¹³C NMR to study the grafting of ¹³C-labelled MAH to each of these substrates. Their work suggests that in EP (and in LLDPE) there is a preference for attachment at methine sites such that a sequence of greater than three methylenes is required before grafting to a methylene site is observed.

Various monomers and reagents can be added to improve grafting yields and to decrease the significance of side reactions such as chain scission (PP) or crosslinking (PE). The effect of various comonomers on grafting yield of MAH onto LLDPE (e.g. S, MMA, maleate esters)³⁰⁹ and PP (e.g. S, AA, MAA, MMA, NVP)^{298,313,314} has been studied. Colai *et al.*³¹⁵ have recently reported on the use of furan derivatives (e.g. 26) in this context. The use of the coagents can substantially increase grafting yields and reduces the degradation in the case of PP. Grafting yields decrease in the series where the comonomer is S >> AMS > MMA > VAc >(no comonomer) > NVP. Several explanations for the comonomer effect have been proposed. Higher grafting yields have been attributed²⁹⁸ to formation of a charge transfer complex between the comonomer and MAH (7.3.1.3) and to the greater reactivity of this species. A second explanation is that the comonomer is a more effective trap than MAH for the polyolefin derived radical.³¹⁵ However, it is also possible that more efficient grafting may simply be due to attachment of a longer chain length graft rather than a greater number of graft sites. Hu et al.²⁹⁸ have provided NMR data for S-MAH grafts from PP suggesting that the graft is a copolymer chain and not a single S-MAH pair. It would also appear from the copolymer composition that S and MAH do not show the same tendency to alternate in the chain in graft copolymer formation as is seen in conventional free radical copolymerization in solution at lower temperatures (7.3.1.3).²⁹⁸ It was found the S-MAH ratio in the graft exceeds the initial S-MAH ratio irrespective of that ratio.²⁹⁸ These observations do not preclude the involvement of a charge transfer complex but do show that the monomers are not incorporated pairwise.

Various solvents, transfer agents and inhibitors have also been used to enhance grafting yields or limit side reactions during polymer modification. If inhibitors can have specificity for the monomer-derived propagating species, it may be possible to prevent homopolymerization while not interfering with abstraction from the polymer backbone by the initiator-derived radicals. Such inhibitors would reduce grafting yields by limiting the length of the grafted chain. Gaylord et al. have reported that various 'electron donor additives' are effective in limiting the amount of crosslinking (various PE, 307,310,316 EP³¹¹) or chain scission (PP^{317,318}) that occurs during melt phase maleation. The additives used included various amides (e.g. dimethyl acetamide, dimethyl formamide, caprolactam, stearamide), $^{307,311,316-318}$ sulfoxides (e.g. dimethyl sulfoxide), 307,310,316 and phosphites (e.g. hexamethylphosphoramide, triethyl phosphite). 307,310,316 A mechanism of action for these coagents based on the propensity of MAH to form charge transfer complexes was proposed.³⁰² Gaylord *et al.*³⁰² also showed that these agents act as inhibitors of MAH homopolymerization, but not of MMA polymerization, and this may explain why the additives cause a ca two-fold reduction in grafting yields with MAH but are not effective in suppressing homopolymerization during grafting of methacrylic monomers. The effectiveness of certain of these coagents has been disputed.^{298,319} Wu and Su³¹⁹ found that stearamide is only useful for low initiator levels and then does not completely suppress crosslinking during grafting

of MAH to EP. It was suggested that, under the process conditions, stearamide acts as a transfer agent. Another report²⁹⁸ suggests that the effect of these coagents in reducing crosslinking of PP might be duplicated simply by using lower initiator concentrations.

7.6.4.2 Maleate ester and maleimide graft polyolefins



The melt phase grafting of dialkyl maleates, usually the diethyl (27) or dibutyl esters (28), onto PP,^{320,321} LLDPE³²⁰⁻³²⁶ and EP³²⁷⁻³²⁹ has been studied. Their use has been advocated over MAII (Section 7.6.4.1) due to their lower volatility and lower toxicity. All may and have been used as precursors to nylon/polyester grafts. However, the maleate esters are significantly less reactive towards free radical addition than MAII and grafting yields are generally lower. Like MAII, the maleate esters show little tendency to homopolymerize. NMR studies suggest that 27 is grafted onto the PE as isolated units even with relatively high dimethyl maleate:polyolefin ratios (1:1 weight ratio).³²³ Even though maleate esters are less reactive than MAH, conditions can be found such that the side reactions associated with peroxide induced grafting (crosslinking, chain seission) appear to be negligible (as indicated by little change in the GPC molecular weight distribution and no insoluble product). This may reflect the greater solubility of the maleate esters in the polyolefin melt.

As with MAH, the extent of grafting varies dramatically with the polyolefin substrate. Some differences have been attributed to variations in the type and amount of stabilizers present in the polyolefins substrate.³²⁶ In the case of isotactic PP, the maximum graft levels attained with **27** were found to correspond to only one unit of DEM per PP molecule³²⁰ This would support a mechanism whereby grafts appear only at the chain ends. Higher graft levels were obtained with atactic PP. The higher reactivity of the atactic PP (and atactic sequences in

isotactic PP) has been attributed to the greater conformational mobility in atactic sequences and less steric hindrance to grafting.^{320,330} It has also been found that grafting yields for EP with a blocky structure are higher than for 'random' copolymers of similar overall composition and molecular weight. This is circumstantial evidence that maleate ester units are preferentially grafted to methylene sites in EP.^{328,329}

More elaborate maleate and maleimide derivatives have provided a route to grafting various functionalities onto PP. Examples, include antioxidants (29 and 30)³³¹ and the oxazoline derivative (31)^{332,333} for which very high grafting yields were reported.

7.6.4.3 (Meth)acrylate graft polyolefins

Various (meth)acrylic monomers have been successfully grafted onto polyolefins. Most studies deal with functional monomers. Grafting yields obtained with PP are usually low (<20%) and are dependent on the particular monomer. Liu *et al.*³³⁴ carried out a comparative study on the grafting of various functional methacrylates onto PP. The experiments were performed in a batch mixer at 180 °C with 7 wt% monomer and 0.05 wt% **22** as an initiator. Grafting levels (wt%) obtained under these conditions were as follows: HPMA (1), TBAEMA (1), GMA (0.8), HEMA (0.4), DMAEMA (0.3), **32** (0.2). Grafting yields to PE appear generally higher.

 $\Lambda\Lambda$,³³⁵ and less often MAA or itaconic acid³³⁶ have been successfully grafted onto polyolefins. In the case of AA, grafting is often accompanied by homopolymerization.³³⁵

Baker and coworkers examined the grafting of methacrylate esters containing secondary or tertiary amino groups such as TBAEMA^{299,337,338} or DEAEMA^{299,339,340} onto LLDPE. Peroxides undergo induced decomposition in the presence of amino-functional monomers. This problem was overcome by using phenylazotriphenylmethane - a source of phenyl radicals - as initiator.²⁹⁹ This gave good grafting yields and no discernible side reactions. The mechanism of grafting was explored using squalane and eisocosane as model substrates.^{341,342} In these experiments, only single unit grafts were observed and little homopolymerization was detected for temperatures above 130 °C. The findings were rationalized in terms of the occurrence of intra-molecular hydrogen abstraction and a low ceiling temperature for polymerization.^{341,342}.

Many studies on the melt phase grafting of GMA onto polyolefins (PP, $^{298,300,301,334,343-347}$ EP, 343 LDPE, 343,348 LLDPE, 349,350 HDPE 343,349,351,352) have been reported. The experiments have been conducted in batch mixers and reactive extruders. Grafting efficiencies onto PP obtained in melt phase grafting experiments with GMA alone are typically very low (<20%). However, it is reported that initiator selection is important in determining the grafting yield.^{300,301} Use of a short half-life initiator **37** ($t_{1/2}$ ~6.6 s) gave two to three-fold higher grafting yields than **22**($t_{1/2}$ ~212 s) under similar processing conditions.³⁰⁰ The use

of comonomers, in particular, styrene, both enhances the grafting efficiency of GMA and reduces PP degradation although some crosslinking occurs when high styrene levels are employed.^{298,300,345,347,352}

Grafting of GMA onto LDPE and EP is more efficient.³⁴³ No crosslinking was observed and high grafting yields were attributed in part to the high solubility of GMA in these polyolefins. Grafting efficiencies for GMA are significantly higher than those observed with other methacrylates. Little has been reported on the structure of the GMA graft copolymers. However, Galluci and Going³⁴³ provided circumstantial evidence that GMA is attached to LDPE as oligo(GMA) blocks rather than as single units.



Al Malaika *et al.*^{353,354} have reported on the grafting of antioxidant moieties onto PP as mono- (e.g. 33) or bis-(meth)acrylic derivatives (34). Moderate grafting yields (10–40%) and some homopolymerization was observed in the case of the monoacrylate. However, with the bis-acrylate (34) close to 100% grafting yield was reported.

In another study, the monoacrylate **35** was grafted onto PP in the presence of tris(acryloylmethyl)propane (**36**) as coagent.³⁵⁵ Again close to 100% grafting yield was obtained. This was so despite the fact that **35** was anticipated to be an inhibitor of free radical reactions (in fact, phenols are poor inhibitors of (meth)acrylate polymerization - Section 5.3.4). The tris-acrylate **36** and related species have previously been used for producing crosslinked/branched PP.^{356,357} The structure of the graft was not established. The remarkable finding was that the final products in the processes involving **36** were not crosslinked and, indeed, were completely soluble in xylene. It was proposed that crosslinking did in fact occur but that the initially formed product underwent *in-situ* degradation by chain scission on further processing to ultimately yield a soluble, gel-free material.³⁵⁸

One would not expect this strategy to be useful for grafting onto PE or other polymers less susceptible to shear induced chain scission.

7.6.4.4 Styrenic graft polyolefins

Although there are several reports^{345,359} of direct grafting of S onto polyolefins, S and AMS are more often encountered as coagents when grafting MAII and (meth)acrylic and other monomers. Recent reports describe the use of the functional styrene derivative **38** to attach oxazoline groups to ABS and **39**^{360,361} to introduce isocyanato groups into PP or PE. Grafting yields with **39** onto PP were improved with use of S as a coagent.³⁶⁰



7.6.4.5 Vinylsilane graft polyolefins

The attachment of trialkoxysilane functionality to polyolefins (HDPE, LDPE, PP) though grafting of vinylsilanes (e.g. 40, 41) or silane functional acrylates (e.g. 42) has been widely studied.³⁶² The principal application of these materials is the preparation of moisture curable crosslinked polyolefins that are widely used in the cable industry.³⁶² Silane treatment has also been used for surface modification of polyolefins³²⁴ and silane grafted polyolefins might also serve as precursors to graft copolymers.

The vinylsilanes (e.g. 40, 41) do not readily homopolymerize. Forsyth *et al.*³⁶³ explored the mechanism of grafting these monomers using dodecane as a model for PE. Their work suggests that multiple monomer units are attached through a sequence of addition and intramolecular hydrogen atom transfer steps by a mechanism analogous to that shown in Scheme 7.33 on page 394.



7.6.4.6 Vinyl oxazoline graft polyolefins

The oxazoline moiety has been used in place of anhydride (from MAH) or epoxy groups (from GMA) as a reactive functionality for use in polymer modification by reactive extrusion³⁶⁴. Polyolefins containing oxazoline functionality are also used as precursors to graft copolymers or as *in situ* compatibilizers or toughening agents. Several methods have been devised for attaching the oxazoline functionality to polyolefins by free radical-induced grafting. The free radical-induced grafting of 2-isopropenyl-2-oxazoline onto PP was reported by Liu and Baker.^{365,366} Vainio *et al.*^{332,333} employed the maleate ester (**31**) to produce an oxazoline functional PP.

7.6.5 Polymerization and Copolymerization of Macromonomers

In the present context, a macromonomer is defined as an oligomer or polymer chain terminated with a double bond or other group such that the material is able to act as a comonomer in a radical copolymerization. The copolymerization of macromonomers with conventional low molecular weight monomers will give a graft copolymer. Since the chain length of the macromonomer determines the chain length of the graft, an important use of these compounds is in the synthesis of graft copolymers with well-defined graft lengths which are also known as polymer brushes.



Various macromonomers have been described in the literature; many are based on polymers of S or (meth)acrylate esters [*e.g.* **43-46**]. The relative merits of macromonomers have been assessed in reviews by Hadjichristidis,³⁶⁷ Capek and coworkers,^{368,369} Ito and coworkers,^{370,371} Meijs and Rizzardo,³⁷² Gnanou and Lutz,³⁷³ and Rempp and Franta³⁷⁴

Most macromonomers do not readily undergo homopolymerization or do so only sluggishly. The intrinsic reactivity of double bonds of macromonomers is often similar to that of the lower molecular weight monomers they resemble. However, the propagating species generated have low reactivity towards further propagation due to adverse steric factors. Oligomethacrylates (46) and similar macromonomers do not undergo homopolymerization or copolymerization with methacrylate esters because of competing addition-fragmentation chain transfer (Scheme 7.35, see also Section 6.2.3.4). On the other hand, with acrylates or S, copolymerization dominates over fragmentation at lower polymerization

temperatures (<80 °C).³⁷⁵ Higher reaction temperatures favor fragmentation to the extent that it is possible to synthesize block copolymers by this form of RAFT polymerization (Section 9.5.2).





The reactivity of macromonomers in copolymerization is strongly dependent on the particular comonomer-macromonomer pair. Solvent effects and the viscosity of the polymerization medium can also be important. Propagation may become diffusion controlled such that the propagation rate constant and reactivity ratios depend on the molecular weight of the macromonomer and the viscosity or, more accurately, the free volume of the medium.

Primary radical transfer may complicate the initiation process. Due to the low concentration of reactive double bonds, it is even more important than usual to select initiators with a low propensity for hydrogen atom abstraction. The greater viscosity of reaction media containing high concentrations of macromonomer can also cause reduced initiator efficiencies as compared to those for conventional polymerizations. Low rates of diffusion of propagating species may reduce rates of termination. A good solvent for macromonomer and polymer will facilitate interpenetration of the polymer chains by the monomer. The balance between these factors can lead to overall rates of copolymerization that are higher or lower than those of conventional radical copolymerization not involving macromonomers (Section 8.3.1). These factors are also largely responsible for the reactivity ratios of macromonomer showing significant solvent dependence.

7.7 References

- 1. Madruga, E.L. Prog. Polym. Sci. 2002, 27, 1879.
- Coote, M.L.; Davis, T.P. In Handbook of Radical Polymerization; Davis, T.P.; Matyjaszewski, K., Eds.; John Wiley & Sons: Hoboken, 2002; p 263.
- 3. Coote, M.L.; Davis, T.P. Prog. Polym. Sci. 1999, 24, 1217.
- 4. Braun, D.; Czerwinski, W.K. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 207.

- 5. Tirrell, D.A. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 195.
- 6. Hamielec, A.E.; MacGregor, J.F.; Penlidis, A. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 17.
- Tirrell, D.A. In Encyclopedia of Polymer Science and Engineering, 2nd ed.; Mark, H.F.; Bikales, N.M.; Overberger, C.G.; Menges, G., Eds.; Wiley: New York, 1985; Vol. 4, p 192.
- 8. Ham, G.E. In *Copolymerization*; Ham, G.E., Ed.; John Wiley and Sons: New York, 1964; p 1.
- 9. Fukuda, T.; Kubo, K.; Ma, Y.-D. Prog. Polym. Sci. 1992, 17, 875.
- 10. Skeist, I. J. Am. Chem. Soc. 1946, 68, 1781.
- 11. Galbraith, M.N.; Moad, G.; Solomon, D.H.; Spurling, T.H. *Macromolecules* **1987**, 20, 675.
- 12. O'Driscoll, K.F. J. Coat. Technol. 1983, 55, 57.
- 13. Hill, L.W.; Wicks, Z.W. Prog. Org. Coat. 1982, 10, 55.
- 14. Mayo, F.R.; Lewis, F.M. J. Am. Chem. Soc. 1944, 66, 1594.
- 15. Alfrey, T.; Goldfinger, G. J. Chem. Phys. 1944, 12, 205.
- 16. Wall, F.T. J. Am. Chem. Soc. 1944, 66, 2050.
- 17. Laurier, G.C.; O'Driscoll, K.F.; Reilly, P.M. J. Polym. Sci., Polym. Symp. 1985, 72, 17.
- 18. Greenley, R.Z. In *Polymer Handbook*, 4th ed.; Brandup, J.; Immergut, E.H.; Grulke, E.A., Eds.; John Wiley and Sons: New York, 1999; p II/181.
- 19. Eastmond, G.C. In *Comprehensive Chemical Kinetics*; Bamford, C.H.; Tipper, C.F.H., Eds.; Elsevier: Amsterdam, 1976; Vol. 14A, p 302.
- 20. Moad, G.; Solomon, D.H.; Spurling, T.H.; Vearing, D.J. Aust. J. Chem. 1986, 39, 1877.
- 21. Ham, G.E. J. Macromol. Sci., Chem. 1991, A28, 733.
- 22. Merz, E.; Alfrey, T.; Goldfinger, G. J. Polym. Sci. 1946, 1, 75.
- 23. Fukuda, T.; Ma, Y.-D.; Inagaki, H. Macromolecules 1985, 18, 17.
- 24. Hill, D.J.T.; O'Donnell, J.H.; O'Sullivan, P.W. Macromolecules 1982, 15, 960.
- 25. Hill, D.J.T.; Lang, A.P.; O'Donnell, J.H.; O'Sullivan, P.W. Eur. Polym. J. 1989, 25, 911.
- 26. Lin, J.; Petit, A.; Neel, J. Makromol. Chem. 1987, 188, 1163.
- 27. Van Der Meer, R.; Alberti, J.M.; German, A.L.; Linssen, H.N. J. Polym. Sci., Polym. Chem. Ed. 1979, 17, 3349.
- 28. Hill, D.J.T.; O'Donnell, J.T.H.; O'Sullivan, P.W. Macromolecules 1985, 18, 9.
- 29. Brown, A.S.; Fujimora, K.; Craven, I. Makromol. Chem. 1988, 189, 1893.
- 30. Guillot, J.; Vialle, J.; Guyot, A. J. Macromol. Sci., Chem. 1971, A5, 735.
- 31. Moad, G.; Solomon, D.H.; Spurling, T.H.; Vearing, D.J. Aust. J. Chem. 1985, 38, 1287.
- 32. Jones, S.A.; Prementine, G.S.; Tirrell, D.A. J. Am. Chem. Soc. 1985, 107, 5275.
- 33. Cywar, D.A.; Tirrell, D.A. J. Am. Chem. Soc. 1989, 111, 7544.
- 34. Giese, B.; Engelbrecht, R. Polym. Bull. 1984, 12, 55.
- 35. Fischer, II.; Radom, L. Angew. Chem., Int. Ed. Engl. 2001, 40, 1340.
- 36. Tanaka, II.; Sasai, K.; Sato, T.; Ota, T. Macromolecules 1988, 21, 3534.
- 37. Tanaka, H.; Sakai, I.; Sasai, K.; Sato, T.; Ota, T. J. Polym. Sci., Part C: Polym. Lett. 1988, 26, 11.
- 38. Kajiwara, A.; Nanda, A.K.; Matyjaszewski, K. Macromolecules 2004, 37, 1378.

402

- 39. Moad, G.; Solomon, D.H.; Spurling, T.H.; Johns, S.R.; Willing, R.I. Aust. J. Chem. 1986, 39, 43.
- 40. Bamford, C.H. Polym. Commun. 1989, 30, 36.
- 41. Harrisson, S.; Kapfenstein-Doak, H.; Davis, T.P. Macromolecules 2001, 34, 6214.
- 42. Chong, Y.K.; Krstina, J.; Le, T.P.T.; Moad, G.; Postma, A.; Rizzardo, E.; Thang, S.II. *Macromolecules* **2003**, *36*, 2256.
- 43. Moad, G.; Moad, C.L.; Rizzardo, E.; Thang, S.H. Macromolecules 1996, 29, 7717.
- 44. Fukuda, T.; Yoshikawa, C.; Kwak, Y.; Goto, A.; Tsujii, Y. ACS Symp. Ser. 2003, 854, 24.
- 45. Coote, M.L.; Zammit, M.D.; Davis, T.P.; Willet, G.D. *Macromolecules* **1997**, *30*, 8182.
- 46. Coote, M.L.; Johnston, L.P.M.; Davis, T.P. Macromolecules 1997, 30, 8191.
- 47. Kaim, A.; Oracz, P. Macromol. Theory Simul. 1997, 6, 565.
- 48. Schweer, J. Makromol. Chem., Theory Simul. 1993, 2, 485.
- 49. Moad, G.; Solomon, D.H.; Spurling, T.H.; Stone, R.A. *Macromolecules* 1989, 22, 1145.
- 50. Davis, T.P.; O'Driscoll, K.F.; Piton, M.C.; Winnik, M.A. J. Polym. Sci., Part C: Polym. Lett. 1989, 27, 181.
- 51. Olaj, O.F.; Bitai, I.; Hinkelmann, F. Makromol. Chem. 1987, 188, 1689.
- 52. Olaj, O.F.; Schnöll-Bitai, I.; Kremminger, P. Eur. Polym. J. 1989, 25, 535.
- 53. Davis, T.P.; O'Driscoll, K.F.; Piton, M.C.; Winnik, M.A. *Macromolecules* 1990, 23, 2113.
- 54. Davis, T.P.; O'Driscoll, K.F.; Piton, M.C.; Winnik, M.A. Polym. Int. 1991, 24, 65.
- 55. Piton, M.C.; Winnik, M.A.; Davis, T.P.; O'Driscoll, K.F. J. Polym. Sci., Part A: Polym. Chem. 1990, 28, 2097.
- 56. Aerdts, A.M.; de Haan, J.W.; German, A.L. Macromolecules 1993, 26, 1965.
- 57. Maxwell, I.A.; Aerdts, A.M.; German, A.L. Macromolecules 1993, 26, 1956.
- 58. Uebel, J.J.; Dinan, F.J. J. Polym. Sci., Polym. Chem. Ed. 1983, 21, 917.
- 59. Beuermann, S.; Buback, M. Prog. Polym. Sci. 2002, 27, 191.
- 60. van Herk, A.M.; Droge, T. Macromol. Theory Simul. 1997, 6, 1263.
- 61. Heuts, J.P.A.; Coote, M.; Davis, T.P.; Johnston, L.P.M. ACS Symp. Ser. 1998, 685, 120.
- 62. Fukuda, T.; Ma, Y.D.; Kubo, K.; Inagaki, H. Macromolecules 1991, 24, 370.
- 63. Evans, M.G.; Polanyi, M. Trans. Faraday Soc. 1938, 34, 11.
- 64. Heuts, J.P.A.; Clay, P.A.; Christie, D.I.; Piton, M.C.; Hutovic, J.; Kable, S.H.; Gilbert, R.G. Prog. Pac. Polym. Sci 1994, 3, 203.
- 65. Cowie, J.M.G. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwali, P., Eds.; Pergamon: Oxford, 1989; Vol. 4, p 377.
- 66. Cowie, J.M.G. In *Alternating Copolymers*; Cowie, J.M.G., Ed.; Plenum: New York, 1985; p 19.
- 67. Cowie, J.M.G. In *Alternating Copolymers*; Cowie, J.M.G., Ed.; Plenum: New York, 1985; p 1.
- 68. Hill, D.J.T.; O'Donnell, J.H.; O'Sullivan, P.W. Prog. Polym. Sci. 1982, 8, 215.
- Furakawa, J. In *Encyclopedia of Polymer Science*, 2nd ed.; Mark, H.F.; Bikales, N.M.; Overberger, C.G.; Menges, G., Eds.; Wiley: New York, 1985; Vol. 4, p 233.
- Shirota, Y. In *Encyclopedia of Polymer Science and Engineering*, 2nd ed.; Mark, H.F.; Bikales, N.M.; Overberger, C.G.; Menges, G., Eds.; New York: Wiley, 1985; Vol. 3, p 327.
- 71. Shirota, Y.; Mikawa, H. J. Macromol. Sci., Rev. Macromol. Chem. 1977, C16, 129.

- 72. Ebdon, J.R.; Towns, C.R.; Dodgson, K. J. Macromol. Sci., Rev. Macromol. Chem. Phys. 1986, 26, 523.
- 73. Rätzsch, M.; Vogl, O. Prog. Polym. Sci. 1991, 16, 279.
- 74. Rzayev, Z.M.O. Prog. Polym. Sci. 2000, 25, 163.
- 75. Cais, R.E.; Farmer, R.G.; Hill, D.J.T.; O'Donnell, J.H. *Macromolecules* 1979, 12, 835.
- 76. Seiner, J.A.; Litt, M. Macromolecules 1971, 4, 308.
- 77. Pittman, C.U.; Rounsefell, T.D. Macromolecules 1975, 8, 46.
- 78. Hill, D.J.T.; O'Donnell, J.H.; O'Sullivan, P.W. Macromol. 1983, 16, 1295.
- 79. Tsuchida, E.; Tomono, T. Makromol. Chem. 1971, 141, 265.
- 80. Karad, P.; Schneider, C. J. Polym. Sci., Polym. Chem. Ed. 1978, 16, 1137.
- 81. Prementine, G.S.; Jones, S.A.; Tirrell, D.A. Macromolecules 1989, 22, 52.
- 82. Saito, J.; Tirrell, D.A. Eur. Polym. J. 1993, 29, 343.
- 83. Jones, S.A.; Tirrell, D.A. J. Polym. Sci., Part A: Polym. Chem. 1987, 25, 3177.
- Bottle, S.E.; Busfield, W.K.; Grice, I.D.; Heiland, K.; Meutermans, W.; Monteiro, M. Prog. Pac. Polym. Sci. 1994, 3, 85.
- 85. Sawada, H. J. Macromol. Sci., Rev. Macromol. Chem. 1974, C11, 257.
- 86. O'Driscoll, K.F.; Gasparro, F.P. J. Macromol. Sci., Chem. 1967, A1, 643.
- 87. Wittmer, P. Makromol. Chem. 1967, 103, 188.
- 88. McManus, N.T.; Dube, M.A.; Penlidis, A. Polym. React. Eng. 1999, 7, 131.
- Palmer, D.E.; McManus, N.T.; Penlidis, A. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 1753.
- Palmer, D.E.; McManus, N.T.; Penlidis, A. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 1981.
- 91. Martinet, F.; Guillot, J. J. Appl. Polym. Sci. 1999, 72, 1611.
- 92. Martinet, F.; Guillot, J. J. Appl. Polym. Sci. 1997, 65, 2297.
- 93. Izu, M.; O'Driscoll, K.F.; Hill, R.J.; Quinn, M.J.; Harwood, H.J. Macromolecules 1972, 5, 90.
- 94. Wittmer, P. Adv. Chem. Ser. 1971, 99, 140.
- 95. Fischer, J.P. Makromol. Chem. 1972, 155, 211.
- 96. Hutchinson, R.A.; Paquet, D.A.; Beuermann, S.; McMinn, J.H. Ind. Eng. Chem. Res. 1998, 37, 3567.
- 97. Lowry, G.G. J. Polym. Sci. 1960, 17, 463.
- 98. Howell, J.A.; Izu, M.; O'Driscoll, K.F. J. Polym. Sci., Part A-1 1970, 8, 699.
- 99. Kruger, IL; Bauer, J.; Rubner, J. Makromol. Chem. 1987, 188, 2163.
- 100. Randall, J.C. Polymer Sequence Determination; Academic Press: New York, 1977.
- 101. Tonelli, A.E. NMR Spectroscopy and Polymer Microstructure; VCII: New York, 1989.
- 102. Koenig, J.L. Chemical Microstructure of Polymer Chains; Wiley: New York, 1980.
- 103. Bovey, F.A. J. Polym. Sci. 1962, 62, 197.
- 104. Kale, L.T.; O'Driscoll, K.F.; Dinan, F.J.; Uebel, J.J. J. Polym . Sci. Part A 1986, 24, 3145.
- Lopez-Gonzalez, M.M.C.; Fernandez-Garcia, M.; Barreles-Rienda, J.M.; Madruga, E.M.; Arias, C. *Polymer* 1993, 34, 3123.
- 106. Klesper, E.; Johnsen, A.; Gronski, W.; Wehrli, F.W. Makromol. Chem. 1975, 176, 1071.
- 107. Johnsen, A.; Klesper, E.; Wirthlin, T. Makromol. Chem. 1976, 177, 2397.
- 108. Park, K.Y.; Santee, E.R.; Harwood, H.J. Eur. Polym. J. 1989, 25, 651.
- 109. Harwood, H.J. Makromol. Chem., Macromol. Symp. 1987, 10/11, 331.

404

- Davis, T.P.; Heuts, J.P.A.; Barner-Kowollik, C.; Harrisson, S.; Morrison, D.A.; Yee, L.H.; Kapfenstein-Doak, H.M.; Coote, M.L. Macromol. Symp. 2002, 182, 131.
- 111. Alfrey, T.; Goldfinger, G. J. Chem. Phys. 1944, 12, 322.
- 112. Alfrey, T.; Goldfinger, G. J. Chem. Phys. 1946, 14, 115.
- 113. Moad, G.; Chiefari, J.; Mayadunne, R.T.A.; Moad, C.L.; Postma, A.; Rizzardo, E.; Thang, S.H. In *Macromol. Symp.*, 2002; Vol. 182, p 65.
- 114. Ham, G.E. J. Macromol. Sci., Chem. 1967, A1, 93.
- 115. Tarasov, A.I.; Tskhai, V.A.; Spasski, S.S. Vysokomol. Soedin. 1960, 2, 1601.
- 116. Fineman, M.; Ross, S.D. J. Polym. Sci. 1950, 5, 259.
- 117. Tidwell, P.W.; Mortimer, G.A. J. Macromol. Sci., Rev. Macromol. Chem. 1970, C4, 261.
- 118. Tidwell, P.W.; Mortimer, G.A. J. Polym. Sci., Part A 1965, 3, 369.
- 119. Kelen, T.; Tüdos, F.; Turcsányi, B. Polym. Bull. 1981, 2, 71.
- 120. Kelen, T.; Tüdos, F. J. Macromol. Sci., Chem. 1975, A9, 1.
- 121. Greenley, R.Z. J. Macromol. Sci., Chem 1980, A14, 445.
- Greenley, R.Z. In *Polymer Handbook*, 3rd ed.; Brandup, J.; Immergut, E.H., Eds.; Wiley: New York, 1989; p II/153.
- 123. Meyer, V.E.; Lowry, G.G. J. Polym. Sci., Part A 1965, 3, 369.
- 124. Francis, A.P.; Solomon, D.II.; Spurling, T.H. J. Macromol. Sci., Chem. 1974, A8, 469.
- 125. Van den Brink, M.; Van Herk, A.M.; German, A.L. J. Polym. Sci., Part A: Polym. Chem. 1999, 37, 3793.
- 126. Giz, A. Macromol. Theory Simul. 1998, 7, 391.
- 127. Plaumann, H.P.; Branston, R.E. J. Polym. Sci., Part A: Polym. Chem. 1989, 27, 2819.
- 128. Hautus, F.L.M.; Linssen, H.N.; German, A.L. J. Polym. Sci., Polym. Chem. Ed. 1984, 22, 3661.
- 129. Hautus, F.L.M.; Linssen, H.N.; German, A.L. J. Polym. Sci., Polym. Chem. Ed. 1984, 22, 3487.
- 130. Leicht, R.; Fuhrman, J. J. Polym. Sci., Polym. Chem. Ed. 1983, 21, 2215.
- 131. McFarlane, R.C.; Reilly, P.M.; O'Driscoll, K.F. J. Polym. Sci., Polym. Chem. Ed. 1980, 18, 251.
- 132. Burke, A.L.; Duever, T.A.; Penlidis, A. Ind. Eng. Chem. Res. 1997, 36, 1016.
- O'Driscoll, K.F.; Kale, L.T.; Garcia Rubio, L.H.; Reilly, P.M. J. Polym. Sci., Polym. Chem. Ed. 1984, 22, 2777.
- 134. Patino-Leal, H.; Reilly, P.M.; O'Driscoll, K.F. J. Polym. Sci., Polym. Lett. Ed. 1980, 18, 219.
- 135. Van Der Meer, R.; Linssen, H.N.; German, A.L. J. Polym. Sci., Polym. Chem. Ed. 1978, 16, 2915.
- 136. Burke, A.L.; Duever, T.A.; Penlidis, A. J. Polym. Sci., Part A: Polym. Chem. 1993, 31, 3065.
- 137. Kelen, T.; Tüdos, F. Makromol. Chem. 1990, 191, 1863.
- 138. Hill, D.J.T.; Lang, A.P.; Munro, P.D.; O'Donnell, J.H. Eur. Polym. J. 1992, 28, 391.
- 139. Bovey, F.A. Chain Structure and Conformation of Macromolecules; Wiley: New York, 1982.
- 140. Hatada, K. NMR Spectroscopy of Polymers; Springer-Verlag: Berlin, 2003.
- 141. Rudin, A.; O'Driscoll, K.F.; Rumack, M.S. Polymer 1981, 22, 740.
- 142. Brown, P.G.; Fujimori, K. Makromol. Chem., Rapid Commun. 1993, 14, 677.
- 143. Moad, G.; Willing, R.I. Polym. J. 1991, 23, 1401.
- 144. Moad, G. In *Annual Reports in NMR Spectroscopy*; Webb, G.A., Ed.; Academic Press: London, 1994; Vol. 29, p 287.
- 145. Moad, G. Chem. Aust. 1991, 58, 122.
- 146. Jenkins, A.D. In *Reactivity, Mechanism and Structure in Polymer Chemistry*; Jenkins, A.D.; Ledwith, A., Eds.; Wiley: London, 1974; p 117.
- 147. Semchikov, Y.D. Polym. Sci. USSR (Engl. Transl.) 1990, 32, 177.
- 148. Alfrey, T.; Price, C.C. J. Polym. Sci. 1947, 2, 101.
- 149. Greenley, R.Z. In *Polymer Handbook*, 4th ed.; Brandup, J.; Immergut, E.H.; Grulke, E.A., Eds.; John Wiley and Sons: New York, 1999; p II/309.
- 150. Borchardt, J.K. J. Macromol. Sci., Chem. 1985, A22, 1711.
- 151. Davis, T.P.; Rogers, S.C. Eur. Polym. J. 1993, 29, 1311.
- 152. Rogers, S.C.; Mackrodt, W.C.; Davis, T.P. Polymer 1994, 35, 1258.
- 153. Zhan, C.G.; Dixon, D.A. J. Phys. Chem. A 2002, 106, 10311.
- 154. Jenkins, A.D.; Jenkins, J. In *Polymer Handbook*, 4th ed.; Brandup, J.; Immergut, E.H.; Grulke, E.A., Eds.; John Wiley and Sons: New York, 1999; p II/321.
- 155. Bamford, C.H.; Jenkins, A.D.; Johnston, R. Trans. Faraday Soc. 1959, 55, 418.
- 156. Bamford, C.H.; Jenkins, A.D. J. Polm. Sci. 1961, 59, 530.
- 157. Bamford, C.H.; Jenkins, A.D. Trans. Faraday Soc. 1963, 59, 530.
- 158. Jenkins, A.D. Eur. Polym. J. 1989, 25, 721.
- 159. Jenkins, A.D.; Hatada, K.; Kitayama, T.; Nishiura, T. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 4336.
- 160. Jenkins, A.D. Polymer 1999, 40, 7045.
- 161. Olaj, O.F.; Zoder, M.; Vana, P.; Zifferer, G. Macromolecules 2004, 37, 1544.
- 162. Fukuda, T.; Goto, A.; Kwak, Y.; Yoshikawa, C.; Ma, Y.D. Macromol. Symp. 2002, 182, 53.
- 163. Melville, H.W.; Valentine, L. Proc. R. Soc., London 1950, A200, 358.
- 164. Melville, H.W.; Valentine, L. Proc. R. Soc., London 1950, A200, 337.
- 165. Mayo, F.R.; Walling, C. Chem. Rev. 1950, 46, 191.
- 166. Atherton, J.N.; North, A.M. Trans. Faraday Soc. 1962, 58, 2049.
- 167. North, A.M. In *Reactivity, Mechanism and Structure in Polymer Chemistry*; Jenkins, A.D.; Ledwith, A., Eds.; Wiley: London, 1974; p 142.
- 168. North, A.M.; Postlethwaite, D. Polymer 1964, 5, 237.
- 169. Chiang, S.S.M.; Rudin, A. J. Macromol. Sci., Chem. 1975, A9, 237.
- 170. O'Driscoll, K.F.; Wertz, W.; Husar, A. J. Polym. Sci., Part A-1 1967, 5, 2159.
- 171. Russo, S.; Munari, S. J. Macromol. Sci., Chem. 1968, A2.
- 172. Bonta, G.; Gallo, B.M.; Russo, S. J. Chem. Soc., Faraday Trans. 1 1975, 69, 1727.
- 173. Bonta, G.; Gallo, B.M.; Russo, S. J. Chem. Soc., Faraday Trans. 1 1973, 69, 329.
- 174. O'Driscoll, K.F.; Huang, J. Eur. Polym. J. 1989, 7/8, 629.
- 175. Kelly, D.P.; Serelis, Λ.K.; Solomon, D.H.; Thompson, P.E. Aust. J. Chem. 1987, 40, 1631.
- 176. Ito, K. Polymer 1985, 26, 1253.
- 177. Guth, W.; Heitz, W. Makromol. Chem. 1976, 177, 1835.
- 178. Moad, G.; Serelis, A.K.; Solomon, D.H.; Spurling, T.H. Polym. Commun. 1984, 25, 240.
- 179. Serelis, A.K. Personal Communication.
- Heitz, W. In *Telechelic Polymers: Synthesis and Applications*; Goethals, E.J., Ed.; CRC Press: Boca Raton, Florida, 1989; p 61.
- 181. Bevington, J.C.; Melville, H.W.; Taylor, R.P. J. Polym. Sci. 1954, 14, 463.
- 182. Chen, C.Y.; Wu, Z.Z.; Kuo, J.F. Polym. Eng. Sci. 1987, 27, 553.

Copolymerization

- 183. Ohtani, H.; Suzuki, A.; Tsuge, S. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 1880.
- 184. Konter, W.; Bömer, B.; Köhler, K.H.; Heitz, W. Makromol. Chem. 1981, 182, 2619.
- 185. Kodaira, T.; Ito, K.; Iyoda, S. Polym. Commun. 1988, 29, 83.
- 186. Moad, G.; Solomon, D.H.; Johns, S.R.; Willing, R.I. Macromolecules 1984, 17, 1094.
- Barton, J.; Capek, I.; Juranicova, V.; Riedel, S. Makromol. Chem., Rapid Commun. 1986, 7, 521.
- 188. Colombani, D.; Chaumont, P. Acta Polym. 1998, 49, 225.
- 189. Tezuka, Y. Prog. Polym. Sci. 1992, 17, 471.
- 190. Ebdon, J.R. In New methods of Polymer Synthesis; Ebdon, J.R., Ed.; Blackie: Glasgow, 1991; p 162.
- 191. Boutevin, B. Adv. Polym. Sci. 1990, 94, 69.
- Nguyen, II.A.; Marechal, E. J. Macromol. Sci., Rev. Macromol. Chem. Phys. 1988, C28, 187.
- 193. Brosse, J.C.; Derouet, D.; Epaillard, F.; Soutif, J.-C.; Legeay, G.; Dusek, K. Adv. Polym. Sci. 1987, 81, 167.
- 194. French, D.A. Rubber. Chem. Technol. 1969, 42, 71.
- 195. Tobolsky, A.V. J. Am. Chem. Soc. 1958, 80, 5927.
- 196. David, G.; Robin, J.J.; Boutevin, B. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 2740.
- 197. David, G.; Boutevin, B.; Robin, J.J.; Loubat, C.; Zydowicz, N. Polym. Int. 2002, 51, 800.
- 198. Bamford, C.H.; Jenkins, A.D.; Wayne, R.P. Trans. Faraday Soc. 1960, 56, 932.
- 199. Reed, S.F. J. Polym. Sci., Polym. Chem. Ed. 1973, 11, 55.
- 200. Bamford, C.H.; Jenkins, A.D.; Johnston, R. Trans. Faraday Soc. 1959, 55, 179.
- 201. Idage, B.B.; Vernekar, S.P.; Ghatge, N.D. J. Polym. Sci., Polym. Chem. Ed. 1983, 21, 385.
- 202. Ghatge, N.D.; Vernekar, S.P.; Wadgaonkar, P.P. Makromol. Chem., Rapid Commun. 1983, 4, 307.
- 203. Friedlander, H.N. J. Polym. Sci. 1962, 58, 455.
- 204. Bresler, L.S.; Barantsevich, E.N.; Polyansky, V.I. Makromol. Chem. 1982, 183, 2479.
- 205. Edelmann, D.; Ritter, H. Makromol. Chem. 1993, 194, 2375.
- 206. Nair, C.P.R.; Clouct, G. J. Macromol. Sci., Rev. Macromol. Chem. Phys. 1991, C31, 311.
- 207. Corner, T. Adv. Polym. Sci. 1984, 62, 95.
- 208. Starks, C.M. Free Radical Telomerization; Academic Press: New York, 1974.
- 209. Boutevin, B.; Lusinchi, J.-M.; Pietrasanta, Y.; Robin, J.-J. Eur. Polym. J. 1994, 30, 615.
- 210. Pryor, W.A.; Coco, J.H. Macromolecules 1970, 3, 500.
- 211. Deady, M.; Mau, A.W.H.; Moad, G.; Spurling, T.H. Makromol. Chem. 1993, 194, 1691.
- 212. Meijs, G.F.; Morton, T.C.; Rizzardo, E.; Thang, S.H. *Macromolecules* **1991**, *24*, 3689.
- 213. Colombani, D. Prog. Polym. Sci. 1997, 22, 1649.
- 214. Chiefari, J.; Rizzardo, E. In *Handbook of Radical Polymerization*; Davis, T.P.; Matyjaszewski, K., Eds.; John Wiley & Sons: Hoboken, NY, 2002; p 263.

- 215. Rizzardo, E.; Chong, Y.K.; Evans, R.A.; Moad, G.; Thang, S.H. Macromol. Symp. 1996, 111, 1.
- 216. Hutson, L.; Krstina, J.; Moad, C.L.; Moad, G.; Morrow, G.R.; Postma, A.; Rizzardo, E.; Thang, S.H. *Macromolecules* 2004, 37, 4441.
- Businelli, L.; Deleuze, II.; Gnanou, Y.; Maillard, B. Macromol. Chem. Phys. 2000, 201, 1833.
- 218. Boutevin, B.; El Idrissi, A.; Parisi, J.P. Makromol. Chem. 1990, 191, 445.
- 219. Boutevin, B.; Pietrasanta, Y. Makromol. Chem. 1985, 186, 817.
- 220. Kharasch, M.S.; Read, J.; Mayo, F.R. Chem. Ind. (London) 1938, 57, 752.
- 221. Klemm, E.; Sensfuss, S. J. Macromol. Sci., Chem. 1991, A28, 875.
- 222. Cramer, N.B.; Bowman, C.N. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 3311.
- 223. Cramer, N.B.; Reddy, S.K.; O'Brien, A.K.; Bowman, C.N. *Macromolecules* 2003, 36, 7964.
- 224. Nuyken, O.; Völkel, T. Makromol. Chem. 1990, 191, 2465.
- 225. Nuyken, O.; Völkel, T. Makromol. Chem., Rapid Commun. 1990, 11, 365.
- 226. Morgan, C.R.; Magnotta, F.; Ketley, A.D. J. Polym. Sci., Part A: Polym. Chem. 1977, 15, 627.
- 227. Bailey, W.J.; Endo, T.; Gapud, B.; Lin, Y.-N.; Ni, Z.; Pan, C.-Y.; Shaffer, S.E.; Wu, S.-R.; Yamazaki, N.; Yonezawa, K. J. Macromol. Sci., Chem. **1984**, *A21*, 979.
- 228. Rimmer, S.; Ebdon, J.R. J. Polym. Sci., Part A: Polym. Chem. 1996, 34, 3573.
- 229. Ebdon, J.R.; Flint, N.J. Eur. Polym. J. 1996, 32, 289.
- 230. Ebdon, J.R.; Flint, N.J. J. Polym. Sci., Part A: Polym. Chem. 1995, 33, 593.
- 231. Ebdon, J.R.; Flint, N.J.; Hodge, P. Eur. Polym. J. 1989, 25, 759.
- 232. Ebdon, J.R. Macromol. Symp. 1994, 84, 45.
- 233. Gridnev, A.A.; Simonsick, W.J.; Ittel, S.D. J. Polym. Sci. Pol. Chem. 2000, 38, 1911.
- 234. Chiefari, J.; Jeffery, J.; Mayadunne, R.T.A.; Moad, G.; Rizzardo, E.; Thang, S.H. ACS Symp. Ser. 2000, 768, 297.
- 235. Kukulj, D.; Heuts, J.P.A.; Davis, T.P. Macromolecules 1998, 31, 6034.
- 236. Moad, G.; Chiefari, J.; Moad, C.L.; Postma, A.; Mayadunne, R.T.A.; Rizzardo, E.; Thang, S.H. *Macromol. Symp.* **2002**, *182*, 65.
- Chiefari, J.; Jeffery, J.; Moad, C.L.; Moad, G.; Postma, A.; Rizzardo, E.; Thang, S.II. *Macromolecules* 2005, in press.
- 238. Montaudo, M.S. Macromolecules 2001, 34, 2792.
- 239. Montaudo, M.S. Polymer 2002, 43, 1587.
- 240. Provder, T.; Whited, M.; Huddleston, D.; Kuo, C.Y. Prog. Org. Coat. 1997, 32, 155.
- 241. Dawkins, J.V. Adv. Chem. Ser. 1995, 247, 197.
- 242. Pasch, H. Adv. Polym. Sci. 1997, 128, 1.
- 243. Berek, D. Prog. Polym. Sci. 2000, 25, 873.
- 244. Macko, T.; Hunkeler, D. Adv. Polym. Sci. 2003, 163, 61.
- 245. Pasch, H.; Mequanint, K.; Adrian, J. e-Polymers 2002, 005.
- 246. Spinelli, H.J. Am. Chem. Soc., Org. Coat. Plas. Chem., Reprints. 1982, 529.
- 247. Stockmayer, W.H. J. Chem. Phys. 1945, 13, 199.
- 248. Fueno, T.; Fukawa, J. J. Polym. Sci., Part A 1964, 2, 3681.
- 249. Mirabella, F.M., Jr. Polymer 1977, 18, 705.
- Spurling, T.H.; Deady, M.; Krstina, J.; Moad, G. Makromol. Chem., Macromol. Symp. 1991, 51, 127.
- 251. Remmp, P.F.; Lutz, P.J. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 4, p 403.

Copolymerization

- 252. Simionescu, C.; Comanita, E.; Pastravanu, M.; Dumitriu, S. Prog. Polym. Sci. 1986, 12, 1.
- 253. Nuyken, O.; Weidner, R. Adv. Polym. Sci. 1986, 73, 145.
- 254. Kuchanov, S.I. In *Comprehensive Polymer Science*; Agarwal, S.L.; Russo, S., Eds.; Pergamon: Oxford, 1992; Vol. Suppl. 1, p 23.
- 255. Ameduri, B.; Boutevin, B.; Gramain, P. Adv. Polym. Sci. 1997, 127, 87.
- 256. Piirma, I.; Chou, L.P.H. J. Appl. Polym. Sci. 1979, 24, 2051.
- 257. Nukyen, O.; Weidner, R. Makromol. Chem. 1988, 189, 1331.
- 258. Qiu, X.-Y.; Ruland, W.; Heitz, W. Angew. Makromol. Chem. 1984, 125, 69.
- 259. Yagci, Y.; Tunca, U.; Biçak, N. J. Polym. Sci., Part C: Polym. Lett. 1986, 24, 49.
- 260. Yagci, Y.; Onen, A.; Schnabel, W. Macromolecules 1991, 24, 4620.
- 261. Hazar, B.; Baysal, B.M. Polymer 1986, 27, 961.
- 262. Walz, R.; Bomer, B.; Heiz, W. Makromol. Chem. 1977, 178, 2527.
- Hoffmann, A.S.; Backsai, R. In *Copolymerization*; Ham, G.E., Ed.; John Wiley and Sons: New York, 1964; p 335.
- 264. O'Driscoll, K.F.; Bevington, J.C. Eur. Polym. J. 1985, 21, 1039.
- 265. Choi, K.Y.; Lee, G.D. AIChE J. 1987, 33, 2067.
- 266. Jenkins, D.W.; Hudson, S.M. Chem. Rev. 2001, 101, 3245.
- 267. McDowell, D.J.; Gupta, B.S.; Stannett, V.T. Prog. Polym. Sci. 1984, 10, 1.
- 268. Stewart, M.J. In New Methods of Polymer Synthesis; Ebdon, J.R., Ed.; Blackie: Glasgow, 1991; p 107.
- Schue, F. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 4, p 359.
- 270. Abadie, M.J.M.; Ourahmoune, D. Br. Polym. J. 1987, 19, 247.
- 271. Eastmond, G.C. Pure & Appl. Chem. 1981, 53, 657.
- 272. Riess, G.; Reeb, R. ACS Symp. Ser. 1981, 166, 477.
- 273. Abadie, M.J.M.; Ourahmoune, D.; Mendjel, H. Eur. Polym. J. 1990, 26, 515.
- 274. Ren, Q.; Zhang, II.; Zhang, X.; Huang, B. J. Polym. Sci., Part A: Polym. Chem. 1993, 31, 847.
- 275. Tung, L.H.; Lo, G.Y.S.; Griggs, J.A. J. Polym. Sci., Polym. Chem. Ed. 1985, 23, 1551.
- 276. Lindsell, W.E.; Service, D.M.; Soutar, I.; Richards, D.H. Br. Polym. J. 1987, 19, 255.
- 277. Cunliffe, A.V.; Hayes, G.F.; Richards, D.H. J. Polym. Sci., Polym. Lett. Ed. 1976, 14, 483.
- 278. Abadie, M.J.M.; Schue, F.; Souel, T.; Richards, D.H. Polymer 1981, 22, 1076.
- 279. Abadie, M.; Burgess, F.J.; Cunliffe, A.V.; Richards, D.H. J. Polym. Sci., Polym. Lett. Ed. 1976, 14, 477.
- 280. Bamford, C.H.; Eastmond, G.C.; Woo, J.; Richards, D.H. Polymer 1982, 23, 643.
- 281. Bamford, C.H.; Dyson, R.W.; Eastmond, G.C. Polymer 1969, 10, 885.
- 282. Eastmond, G.C.; Parr, K.J.; Woo, J. Polymer 1988, 29, 950.
- 283. Eastmond, G.C.; Woo, J. Polymer 1990, 31, 358.
- 284. Eastmond, G.C.; Grigor, J. Makromol. Chem., Rapid Commun. 1986, 7, 375.
- Alimoglu, A.K.; Bamford, C.H.; Ledwith, A.; Mullik, S.U. Polym. Sci. USSR (Engl. Transl.) 1980, 21, 2651.
- 286. Bamford, C.H.; Middleton, I.P.; Al-Lamee, K.G.; Paprotny, J. Br. Polym. J. 1987, 19, 269.
- 287. Dargaville, T.R.; George, G.A.; Hill, D.J.T.; Whittaker, A.K. Prog. Polym. Sci. 2003, 28, 1355.
- 288. Kabanov, V.Y.; Kudryavtsev, V.N. High Energy Chem. 2003, 37, 1.

- 289. Kato, K.; Uchida, E.; Kang, E.T.; Uyama, Y.; Ikada, Y. Prog. Polym. Sci. 2003, 28, 209.
- 290. Bhattacharya, A. Prog. Polym. Sci. 2000, 25, 371.
- 291. Kabanov, V.Y.; Aliev, R.E.; Kudryavtsev, V.N. Radiat. Phys. Chem. 1991, 37, 175.
- 292. Russell, K.E. Prog. Polym. Sci. 2002, 27, 1007.
- 293. Moad, G. Prog. Polym. Sci. 1999, 24, 81.
- 294. Xanthos, M. Reactive Extrusion; Hanser: Munich, 1992.
- 295. Al-Malaika, S., Ed. Reactive Modifiers for Polymers; Chapman & Hall: London, 1996.
- 296. Ratzsch, M.; Arnold, M.; Borsig, E.; Bucka, H.; Reichelt, N. Prog. Polym. Sci. 2002, 27, 1195.
- 297. Baker, W.E.; Scott, C.E.; Hu, G.-H. Reactive Polymer Blending; Hanser: Munich, 2001.
- 298. Hu, G.II.; Flat, J.-J.; Lambla, M. In *Reactive Modifiers for Polymers*; Al-Malaika, S., Ed.; Chapman & Hall: London, 1996; p 1.
- 299. Xie, H.-Q.; Baker, W.E. In New Advances in Polyolefins; Chung, T.C., Ed.; Plenum: New York, N. Y., 1993; p 101.
- 300. Wong, B.; Baker, W.E. Annu. Tech. Conf. Soc. Plast. Eng. 1996, 54(1), 283.
- 301. Huang, H.; Liu, N.C. J. Appl. Polym. Sci. 1998, 67, 1957.
- 302. Gaylord, N.G. In *Reactive Extrusion*; Xanthos, M., Ed.; Hanser: Munich, 1992; p 55.
- Heinen, W.; Rosenmöller, C.H.; Wenzel, C.B.; de Groot, H.J.M.; Lugtenburg, J. Macromolecules 1996, 29, 1151.
- 304. Zhang, M.Z.; Dhuamel, J.; van Duin, M.; Meessen, P. Macromolecules 2004, 37, 1877.
- 305. De Roover, B.; Sclavons, M.; Carlier, V.; Devaux, J.; Legras, R.; Momatz, A. J. Polym. Sci., Part A: Polym. Chem. 1995, 33, 829.
- 306. De Roover, B.; Devaux, J.; Legras, R. J. Polym. Sci., Part A: Polym. Chem. 1996, 34, 1195.
- 307. Gaylord, N.G.; Mehta, R. J. Appl. Polym. Sci. 1989, 38, 359.
- 308. Bray, T.; Damiris, S.; Grace, A.; Moad, G.; O'Shea, M.; Rizzardo, E.; van Diepen, G. Macromol. Symp. 1997, 129, 109.
- 309. Samay, G.; Nagy, T.; White, J.L. J. Appl. Polvm. Sci. 1995, 56, 1423.
- 310. Gaylord, G.N.; Mehta, R.; Mohan, D.R.; Kumar, V. J. Appl. Polym. Sci. 1992, 44, 1941.
- 311. Gaylord, N.G.; Mehta, N.; Mehta, R. J. Appl. Polym. Sci. 1987, 33, 2549.
- 312. Kowalski, R.C. In *Reactive Extrusion*; Xanthos, M., Ed.; Hanser: Munich, 1992; p 7.
- Hu, G.H.; Flat, J.J.; Lambla, M. Makromol. Chem., Macromol. Symp. 1993, 75, 137.
- 314. Hu, G.-H.; Flat, J.-J.; Lambla, M. Annu. Tech. Conf. Soc. Plast. Eng. 1994, 52(3), 2775.
- 315. Coiai, S.; Passaglia, E.; Aglietto, M.; Ciardelli, F. Macromolecules 2004, 37, 8414.
- 316. Gaylord, N.G.; Mehta, R. J. Polym. Sci., Part A: Polym. Chem. 1988, 26, 1189.
- 317. Gaylord, N.G.; Deshpande, A.B. Polym. Mater. Sci. Eng. 1992, 67, 109.
- 318. Gaylord, G.N.; Mishra, M.K. J. Polym. Sci., Polym. Lett. Ed. 1983, 21, 23.
- 319. Wu, C.H.; Su, A.C. Polymer 1992, 33, 1987.
- 320. Ruggeri, G.; Aglietto, M.; Petragnani, A.; Ciardelli, F. Eur. Polym. J. 1983, 19, 863.
- 321. Benedetti, E.; Aldo, D.A.; Aglietto, M.; Ruggeri, G.; Vergamini, P.; Ciadelli, F. Polym. Eng. Sci. 1986, 26, 9.

Copolymerization

- 322. Aglietto, M.; Bertani, R.; Ruggeri, G.; Segre, A.L. Macromolecules 1990, 23, 1928.
- 323. Aglietto, M.; Bertani, R.; Ruggeri, G. Makromol. Chem. 1992, 193, 179.
- 324. Konar, J.; Scn, A.K.; Bhowmick, A.K. J. Appl. Polym. Sci. 1993, 48, 1579.
- 325. Rosales, C.; Marques, L.; Gonzalez, J.; Perera, R.; Rojas, B.; Vivas, M. Polym. Eng. Sci. 1996, 36, 2247.
- 326. Rosales, C.; Perera, R.; Ichazo, M.; Gonzalez, J.; Rojas, H.; Sanchez, A.; Barrios, A.D. J. Appl. Polym. Sci. 1998, 70, 161.
- 327. Greco, R.; Musto, P.; Scarinzi, G. Polym. Mater. Sci. Eng. 1987, 57, 770.
- 328. Greco, R.; Riva, F.; Musto, P.V.; Maglio, G. J. Appl. Polym. Sci. 1989, 37, 789.
- 329. Greco, R.; Maglio, G.; Musto, P.V.; Scarinzi, G. J. Appl. Polym. Sci. 1989, 37, 777.
- 330. Garcia-Martinez, J.M.; Laguna, O.; Collar, E.P. J. Appl. Polym. Sci. 1997, 65, 1333.
- 331. Al-Malaika, S. Polym.-Plast. Technol. Eng. 1990, 29, 73.
- 332. Vainio, T.; Hu, G.-H.; Lambla, M.; Scppala, J.V. J. Appl. Polym. Sci. 1997, 63, 883.
- 333. Vainio, T.; Hu, G.-H.; Lambla, M.; Seppala, J.V. J. Appl. Polym. Sci. 1996, 61, 843.
- 334. Liu, N.C.; Xie, H.Q.; Baker, W.E. Polymer 1993, 34, 4680.
- Oromehie, A.R.; Hashemi, S.A.; Meldrum, I.G.; Waters, D.N. Polym. Int. 1997, 42, 117.
- Pesetskii, S.S.; Jurkowski, B.; Krivoguz, Y.M.; Urbanowicz, R. J. Appl. Polym. Sci. 1997, 65, 1493.
- 337. Song, Z.; Baker, W.E. Polymer 1992, 33, 3266.
- 338. Song, Z.; Baker, W.E. J. Appl. Polym. SciJ. Polym. Sci., Part A: Polym. Chem. 1992, 44, 2167.
- 339. Simmons, A.; Baker, W.E. Polym. Eng. Sci. 1989, 29, 1117.
- 340. Oliphant, K.E.; Baker, W.E. Annu. Tech. Conf. Soc. Plast. Eng. 1994, 52(2), 1524.
- 341. Wong Shing, J.B.; Baker, W.E.; Russell, K.E.; Whitney, R.A. J. Polym. Sci., Part A: Polym. Chem. 1994, 32, 1691.
- 342. Wong Shing, J.B.; Baker, W.E.; Russell, K.E. J. Polym. Sci., Part A: Polym. Chem. 1995, 33, 633.
- 343. Galluci, R.R.; Going, R.C. J. Appl. Polvm. Sci. 1982, 27, 425.
- 344. Chen, L.-F.; Wong, B.; Baker, W.E. Polym. Eng. Sci. 1996, 36, 1594.
- 345. Sun, Y.-J.; Hu, G.-H.; Lambla, M. Angew. Makromol. Chem 1995, 229, 1.
- 346. Sun, Y.-J.; Hu, G.-H.; Lambla, M. J. Appl. Polym. Sci. 1995, 57, 1043.
- 347. Cartier, H.; Hu, G.H. J. Polym. Sci., Part A: Polym. Chem. 1998, 36, 1053.
- 348. Zhang, X.; Yin, Z.; Li, L.; Yin, J. J. Appl. Polvm. Sci. 1996, 61, 2253.
- 349. Liu, T.M.; Evans, R.; Baker, W.E. Annu. Tech. Conf. Soc. Plast. Eng. 1995, 53(2), 1564.
- 350. Pesneau, I.; Champagne, M.F.; Huneault, M.A. J. Appl. Polym. Sci. 2004, 91, 3180.
- 351. Torres, N.; Robin, J.J.; Boutevin, B. J. Appl. Polym. Sci. 2001, 81, 581.
- 352. Cartier, H.; Hu, G.H. J. Polym. Sci., Part A: Polym. Chem. 1998, 36, 2763.
- 353. Al-Malaika, S.; Ibrahim, A.Q.; Rao, J.; Scott, G. J. Appl. Polym. Sci. 1992, 44, 1287.
- 354. Al-Malaika, S.; Scott, G.; Wirjosentono, B. Polym. Degrad. Stab. 1993, 40, 233.
- 355. Al-Malaika, S.; Suharty, N. Polym. Degrad. Stab. 1995, 49, 77.
- 356. Wang, X.; Tzoganakis, C.; Rempel, G.L. J. Appl. Polym. Sci. 1996, 61, 1395.
- 357. Kim, B.K. Korea Polym. J. 1996, 4, 215.
- Al-Malaika, S. In *Reactive Modifiers for Polymers*; Al-Malaika, S., Ed.; Chapman & Hall: London, 1996; p 266.
- 359. Kim, B.S.; Kim, S.C. J. Appl. Polym. Sci. 1998, 69, 1307.
- 360. Hu, G.H.; Li, H.X.; Feng, L.F.; Pessan, L.A. J. Appl. Polym. Sci. 2003, 88, 1799.
- 361. Braun, D.; Schmitt, M.W. Polym. Bull. 1998, 40, 189.

- Munteanu, D. In *Reactive Modifiers for Polymers*; Al-Malaika, S., Ed.; Chapman & Hall: London, 1996; p 196.
- 363. Forsyth, J.C.; Baker, W.E.; Russell, K.E.; Whitney, R.A. J. Polym. Sci., Part A: Polym. Chem. 1997, 35, 3517.
- 364. Rösch, J.; Holger, W.; Müller, P.; Schäfer, R.; Wörner, C.; Friedrich, C.; Kressler, J.; Mülhaupt, R. *Macromol. Symp.* **1996**, *102*, 241.
- 365. Liu, N.C.; Baker, W.E. In *Reactive Modifiers for Polymers*; Al-Malaika, S., Ed.; Chapman & Hall: London, 1996; p 163.
- 366. Liu, N.C.; Baker, W.E. Polymer 1994, 35, 988.
- Hadjichristidis, N.; Pitsikalis, M.; latrou, II.; Pispas, S. Macromol. Rapid Commun. 2003, 24, 979.
- 368. Capek, I. Adv Polym. Sci. 1999, 145, 1.
- 369. Capek, I.; Akashi, M. J. Macromol. Sci., Rev. Macromol. Chem. Phys. 1993, C33, 369.
- 370. Ito, K.; Kawaguchi, S. Adv. Polym. Sci. 1999, 142, 129.
- 371. Ito, K. Prog. Polym. Sci. 1998, 23, 581.
- 372. Meijs, G.F.; Rizzardo, E. J. Macromol. Sci., Rev. Macromol. Chem. Phys. 1990, C30, 305.
- 373. Gnanou, Y.; Lutz, P. Makromol. Chem. 1989, 190, 577.
- 374. Rempp, P.F.; Franta, E. Adv. Polym. Sci. 1984, 58, 1.
- Cacioli, P.; Hawthorne, D.G.; Laslett, R.L.; Rizzardo, E.; Solomon, D.H. J. Macromol. Sci., Chem. 1986, A23, 839.

8 Controlling Polymerization

8.1 Introduction

Radical polymerization is often the preferred mechanism for forming polymers and most commercial polymer materials involve radical chemistry at some stage of their production cycle. From both economic and practical viewpoints, the advantages of radical over other forms of polymerization are many (Chapter 1). However, one of the often-cited "problems" with radical polymerization is a perceived lack of control over the process: the inability to precisely control molecular weight and distribution, limited capacity to make complex architectures and the range of undefined defect structures and other forms of "structure irregularity" that may be present in polymers prepared by this mechanism. Much research has been directed at providing answers for problems of this nature. In this, and in the subsequent chapter, we detail the current status of the efforts to redress these issues. In this chapter, we focus on how to achieve control by appropriate selection of the reaction conditions in "conventional" radical polymerization.

Minor (by amount) functionality is introduced into polymers as a consequence of the initiation, termination and chain transfer processes (Chapters 3, 5 and 6 respectively). These groups may either be at the chain ends (as a result of initiation, disproportionation, or chain transfer,) or they may be part of the backbone (as a consequence of termination by combination or the copolymerization of byproducts or impurities). In Section 8.2 we consider three polymers (PS, PMMA and PVC) and discuss the types of defect structure that may be present, their origin and influence on polymer properties, and the prospects for controlling these properties through appropriate selection of polymerization conditions.

Structural irregularity is also introduced in the propagation step either through a lack of regio- or stereochemical specificity in radical addition to monomer or by rearrangement of the propagating species (Section 4.4). In Section 8.3 the influence of the reaction media and added reagents on the stereochemistry and rate of radical polymerization is explored. With this knowledge we consider the prospects for controlling polymer structure and properties by appropriate choice of reaction conditions (solvent, temperature, pressure) or through the use of complexing agents and templates to direct the course of polymerization.¹

8.2 Controlling Structural Irregularities

The functional groups introduced into polymer chains as a consequence of the initiation or termination processes can be of vital importance in determining certain polymer properties. Some such functionality is generally unavoidable. However, the types of functionality can be controlled through selection of initiator, solvent and reaction conditions and should not be ignored.

Such functionality can also be of great practical importance since functional initiators, transfer agents, *etc.* are applied to prepare end-functional polymers (see Section 7.5) or block or graft copolymers (Section 7.6). In these cases the need to maximize the fraction of chains that contain the reactive or other desired functionality is obvious. However, there are also well-documented cases where "weak links" formed by initiation, termination, or abnormal propagation processes impair the thermal or photochemical stability of polymers.

Thus, it is important to know, understand and control the kinetics and mechanism of the entire polymerization process so that desirable aspects of the polymer structure can be maximized while those reactions that lead to an impairment of properties or a less than ideal functionality can be avoided or minimized. A corollary is that it is important to know how a particular polymer was prepared before using it in a critical application.

8.2.1 "Defect Structures" in Polystyrene

There is a substantial literature on the thermal and photochemical degradation of PS and it is well established that polymer properties are sensitive to the manner in which a particular sample of PS is prepared. For example, it has been reported that PS prepared by anionic polymerization shows enhanced stability with respect to that prepared by a radical mechanism.²⁻¹⁰ This has often been attributed to the presence of "weak links" in the latter polymers. However, the precise nature of the "weak links" remains the subject of some controversy. The situation is further confused by all PS prepared by radical mechanisms often being considered as a class without reference to the particular polymerization conditions employed in their preparation. In many cases the polymers are "commercial samples" with details of the method of preparation incomplete or unstated.

In some cases the "weak links" in radical PS may be peroxidic linkages.^{11,12} Such groups may become incorporated in polymers formed by radical polymerization through copolymerization of adventitious oxygen (Section 5.3.2). Peroxidic linkage may be avoided by paying careful attention to monomer purification and rigorous exclusion of oxygen from the polymerization. Head-tohead linkages, such as those formed by termination by combination, have been proposed as a source of thermal instability.⁸ However, there is also evidence that thermal behavior depends on the particular radical initiator or reaction conditions (solvent, temperature, conversion) employed in polymer preparation. It also appears that in some cases the thermal degradation of radical PS can be interpreted in terms of initiation by random chain scission uncomplicated by processes initiated at weak links.¹²

BPO is commonly used as an initiator for S polymerizations and copolymerizations and it has been reported that its use can lead to yellowing and impaired stability in PS.^{13,14} The initiation and termination pathways observed for S polymerization when BPO is used as initiator have been discussed in Sections 3.2 and 3.4.2.2.1. These give rise to benzoyloxy and phenyl end-groups as follows (Scheme 8.1).

Initiation:



Transfer to initiator/primary radical termination:



Scheme 8.1

NMR studies^{15,16} on polymers prepared with ¹³C-labeled BPO have shown that the primary benzoyloxy and phenyl end groups formed by tail addition to monomer are thermally stable under conditions where the polymer degrades. They persist to > 50% weight loss at 300°C under nitrogen. Thus, these groups are unlikely to be directly responsible for the poor thermal stability of PS prepared with BPO as initiator. On the other hand, the secondary benzoate end groups, formed by head addition or transfer to initiator, appear extremely labile under these conditions. Their half life at 300°C is <5 min.

Studies with model compounds show that secondary benzoate esters eliminate benzoic acid to form unsaturated chain ends as shown in Scheme 8.2.¹⁵

Unsaturation has long been thought to be a "weak link" in PS.^{4,17} It has been found that for BPO initiated S polymerization at high conversion most chain termination may be by way of transfer to initiator or primary radical termination.¹⁸ Therefore, if these groups are responsible for initiating the chain degradation process, it provides a plausible explanation for high conversion PS formed with BPO initiator being less thermally stable than either a similar low conversion polymer or a polymer prepared with a different initiator.



Scheme 8.2

These examples show how initiator selection can be critical in determining the properties of PS prepared by radical polymerization. If thermal stability were of importance, then, since some initiator-derived ends cannot be avoided, a preferred initiator would be one which gives rise to end groups that do not readily eliminate or dissociate. End groups formed with AIBN initiator appear stable with respect to the polymer backbone,¹⁹ Many other systems remain to be studied.



Scheme 8.3

The majority of polymers formed by living radical polymerization (NMP, ATRP, RAFT) will possess labile functionality at chain ends. Recent studies have examined the thermal stability of polystyrene produced by NMP with TEMPO (Scheme 8.3),^{20,21} ATRP and RAFT (Scheme 8.4).²² In each case, the end groups

Controlling Polymerization

are observed to degrade at relatively low temperatures (~ 200 °C) by cross disproportionation or thermal elimination to leave an unsaturated chain end. Thermal elimination has been proposed as a simple and convenient method of removing reactive chain ends when this is desirable. For each method of polymerization, various methods of replacing the chain functionality with hydrogen or a more desirable functionality have been devised (Chapter 9).



Scheme 8.4

8.2.2 "Defect Structures" in Poly(methyl methacrylate)

There have been many studies on the thermal and thermo-oxidative degradation of PMMA.^{23,24} It is well established that the polymer formed by radical polymerization can be substantially less stable than predicted by consideration of the idealized structure and that the kinetics of polymer degradation are dependent on the conditions used for its preparation. There is still some controversy surrounding the details of thermal degradation mechanisms and, in particular, the initiation of degradation.²³

The thermal degradation of 'ideal' PMMA chains, such as might be formed by anionic polymerization, is thought to be initiated by a random scission process involving cleavage of backbone or side chain bonds.²⁵⁻²⁷ The polymer formed by radical polymerization contains weak links. PMMA degrades by unzipping or depropagation (*i.e.* the reverse of radical polymerization). Any structures that are less stable than the backbone or side chain bonds and which give rise to propagating radicals constitute weak links.

Unstable structures are known to arise by chain termination. Mechanisms for radical-radical termination in MMA polymerization have been discussed in Sections 5.2.2.1.2 and 5.2.2.2.2 and these are summarized in Scheme 8.5. It is established that both disproportionation and combination occur to substantial extents. The head-to-head linkages 1 and the unsaturated chain ends 2 both constitute weak links in PMMA.^{26,28-33} The presence of these groups account for

PMMA formed by radical polymerization being significantly less stable than that formed by anionic polymerization.





Head-to-head linkages (1) are thermally unstable at temperatures above 180°C and may undergo spontaneous scission to form propagating radicals (Scheme 8.6).^{29,31-33}



Scheme 8.6

The bond β - to the double bond of the unsaturated disproportionation product **2** is also weaker than other backbone bonds.^{10,30,32,33} However, it is now believed that the instability of unsaturated linkages is due to a radical-induced decomposition mechanism (Scheme 8.7).³⁰ This mechanism for initiating degradation is analogous to the addition-fragmentation chain transfer observed in polymerizations carried out in the presence of **2** at lower temperatures (see 6.2.3.4, 7.6.5 and 9.5.2).

To avoid these stability problems, it is necessary to minimize the proportion of chains that terminate by radical-radical reaction. One way of achieving this is to conduct the polymerization in the presence of an appropriate chain transfer agent. For example, if polymerization is performed in the presence of a H-donor chain transfer agent, conditions can be chosen such that most chains terminate by hydrogen-atom transfer. Bagby *et al.*³⁴ examined the thermal stability of PMMA formed with dodecanethiol. These polymer chains will then possess, more

Controlling Polymerization

thermally stable, saturated end groups (3, see Scheme 8.5).³⁴ If terminated by a proton source, anionic PMMA also has saturated chain ends (3).



Scheme 8.7

It has also been suggested that, for polymers formed in the presence of air, peroxidic linkages may be weak links.²³ However, in this context, it is of interest that PMMA appears more thermally stable under air than it is under nitrogen (higher initial decomposition temperature).^{24,32,35,36} Various explanations have been suggested. Peterson *et al.*^{24,36} have attributed this to the propagating radicals PMMA• formed as a consequence of weak link scission being trapped by oxygen to form as hydroperoxy radicals (Scheme 8.8). Other radical traps (nitric oxide) also stabilize the polymer.^{24,36}

Scheme 8.8

There are other sources of unsaturated chain ends in PMMA formed by radical polymerization:

- (a) End groups similar to those formed by disproportionation (2) are formed in chain transfer to certain addition-fragmentation transfer agents (*e.g.* allyl sulfides, see 6.2.3.2) or cobalt chain transfer agents (see 6.2.5).
- (b) Unsaturated chain ends can arise by primary radical transfer or transfer to MMA. This involves abstraction of the α -methyl or the ester methyl hydrogens. If the monomer-derived radicals so-formed initiate polymerization, the polymer will contain end groups 4 and 5. The *t*-butoxy and other *t*-alkoxy radicals show a propensity for abstraction (see 3.4.2.1).³⁷⁻³⁹



Note, however, that chain ends 4 and 5 may give different chemistry to those formed in termination by disproportionation (2, see Scheme 8.5) or the processes under (a) above. Chain scission β to the double bond will not lead to a MMA propagating species. It is not established whether the presence of these ends will give impaired thermal stability.

However, the presence of unsaturated chain ends can have other consequences for polymer properties:

- (a) Propagating radicals initiated by abstraction products will not contain an initiator residue at one chain end.³⁹ Experiments which depend on determination of initiator-derived chain ends may be in error and some literature data may need to be reinterpreted in this light.⁴⁰ Syntheses of telechelic or end-functional polymers based on the use of functional initiators will also be detrimentally affected (see 7.5.1).
- (b) The unsaturated end groups (2, 4 and 5) may be reactive under polymerization conditions (*i.e.* the polymer chains can be considered as macromonomers) and may copolymerize leading to graft formation (see 7.6.5).⁴¹ The end groups (2) may also give chain transfer by an addition-fragmentation mechanism (see 6.2.3.4 and 9.5.2).

It is of interest that thermogravimetric analysis has been used as a means of determining end group purity of PMMA macromonomers formed by catalytic chain transfer.

As in the case of PS (Section 8.2.1) polymers formed by living radical polymerization (NMP, ATRP, RAFT) have thermally unstable labile chain ends. Although PMMA can be prepared by NMP, it is made difficult by the incidence of cross disproportionation.⁴² Thermal elimination, possibly by a homolysis-cross disproportionation mechanism, provides a route to narrow polydispersity macromonomers.⁴³ Chemistries for end group replacement have been devised in the case of polymers formed by NMP (Section 9.3.6), ATRP (Section 9.4) and RAFT (Section 9.5.3).

8.2.3 "Defect Structures" in Poly(vinyl chloride)

Mechanisms of thermal degradation of PVC, the structure of PVC and the stabilization of PVC have been the subject of many reviews. Those by Starnes,⁴⁴ Endo⁴⁵ and Ivan⁴⁶ are some of the more recent. Defect structures in PVC arise during the propagation and chain transfer steps. As with PMMA, PVC formed by

anionic polymerization is much more stable than that formed by radical polymerization. The relative stabilities of structures that may be formed by anomalous reactions have been established by comparing the stabilities of low molecular weight model compounds.⁴⁴

PVC formed with diacyl peroxide or peroxydicarbonate initiators will contain a proportion of potentially labile α -haloester chain ends (6, Scheme 8.9). However, it is believed that most chain ends in PVC are formed by transfer to monomer as is discussed in Sections 4.3.1.2 and 6.2.6.3.⁴⁷



Scheme 8.9

8.3 Controlling Propagation

Given the important role that steric and polar factors play in determining the rate and regiospecificity of radical additions (see 2.3), it might be anticipated that reagents which coordinate with the propagating radical and/or the monomer and thereby modify the effective size, polarity, or inherent stability of that species, could alter the outcome of propagation.

The aspects of polymer structure to be controlled have already been discussed in Chapter 4. For the case of a homopolymer, these are:

- (a) Stereosequence isomerism (Section 4.2); the tacticity of the polymer chain. Most polymers formed by radical polymerization have an excess of syndiotactic over isotactic dyads. P(m) typically lies in the range 0.4-0.5 for vinyl monomers and 0.2-0.5 for 1,1-disubstituted monomers. The physical properties of polymers depend on chain stereochemistry. If tacticity control can be achieved, a further challenge is to control the chirality of the chain.
- (b) Regiosequence isomerism (Section 4.3); the extent of head vs tail addition
- (c) Structural isomerism (Section 4.4); rearrangement during propagation. A particular challenge is to control the incidence of short chain branching in PE and in polyacrylates

For the case of copolymers, it is also possible to control the arrangement of monomer units in the chain.

The reagents used for controlling polymer structure may be low molecular weight (e.g. the solvent - Sections 8.3.1-8.3.3, Lewis acids - Section 8.3.4) or

polymeric (*e.g.* template polymers - Section 8.3.5, enzymes - Section 8.3.6). Control over polymer structure may also be achieved in a topological polymerization where the monomer is crystalline or organized such that the spatial arrangement on the monomer is appropriately constrained (Section 8.3.7).

For greatest effect propagation involving the complexed or constrained species should dominate over normal propagation. For this to occur one of the following should apply:

- (a) Either the monomer or propagating species is completely complexed. This requires that concentration of the reagent whether mono- or polymeric to be at least stoichiometric with the species to be complexed throughout the polymerization.
- (b) The reactivity of the complexed species is many-fold greater than that of any remaining uncomplexed species and that the equilibrium and rate constants associated with complex formation are high.

Bearing these requirements in mind, the more desirable way of controlling propagation would appear to be to complex the propagating radical (P•). Whereas the initial monomer concentrations are typically in the range ~2-10 M, the typical "steady state" concentration of P• is usually very low (~10⁻⁶-10⁻⁷ M) (Scheme 8.10). Therefore, only a small concentration of a catalytic reagent would be required to complex all radicals. However, for this strategy to be successful, the reagent should interact specifically with P• and not associate strongly with either the monomer or the polymer. In any competitive equilibrium, the difference in concentrations (up to 10⁸-fold) would clearly favor interaction with monomer or polymer over P•.

$$[P^{\bullet}] \sim 10^{-6} \cdot 10^{-7} \text{ M} \quad [M]_{0} \sim 2 \cdot 10 \text{ M}$$
Scheme 8.10

In seeking a suitable complexing agent for the propagating species, one approach is to consider the various species $(X \cdot)$ that are known to reversibly add carbon-centered radicals (Scheme 8.11). Many such reagents have been described in the organic literature. Such species find use as mediators in living radical polymerization. Notable examples are nitroxides (in NMP, Section 9.3.6), dithioesters (in RAFT, Section 9.5.3) and various organometallic complexes (in ATRP, Section 9.4). These species (Z \cdot) react with carbon-centered radicals (R \cdot) at near diffusion controlled rates yet the Z–R bonds of the adduct are relatively weak. The bond strength depends on the nature of R and the functionality on Z. Under the appropriate reaction conditions, the X-R bond may undergo reversible

homolysis allowing monomer insertion by a radical mechanism. Could the proximity of X influence the course of propagation?

Most of the studies on polymerization have been concerned with studying the utility of reagents conferring living characteristics on the polymerization (*e.g.* achieving narrow polydispersities, making block copolymers, *etc.*) and at controlling the rate of polymerization. Only a few have explicitly looked for effects on polymer structure. Several studies have explored NMP utilizing chiral and bulky nitroxides. ^{48,49} The use of chiral metal complexes in ATRP has also been explored. ^{50,51} No significant influence on polymer structure (on chirality or tacticity) was observed. All evidence suggests that propagating radicals in these processes (NMP, RAFT, ATRP) behave as free (uncomplexed) propagating radicals. To date there is little evidence that complexation of radicals by the reagents discussed above occurs or, if there is, that the complexation influences the regio- or stereospecificity of radical addition.



Scheme 8.11

An early report⁵² that the stereoregularity of MMA propagation is influenced by Co-porphyrin has not been confirmed by subsequent studies. Giese et al.⁵³ reported that cyclohexyl radicals generated from alkylcobaloximes and cyclohexyl radicals generated from other sources show different specificity in atom transfer reactions. However, they⁵³ and Clarke and Jones⁵⁴ have also provided evidence that the radicals generated from square planar cobalt complexes behave as "normal" radicals in simple radical additions. The utility of cobalt complexes as complexing agents in controlling propagation is limited by side reactions that give chain transfer (these may be used to advantage in macromonomer preparation -Section 6.2.5). The importance of these reactions can be controlled by limiting the application to monosubstituted monomers and by changing the ligands on cobalt (Section 9.3.9.1). Radical polymerization of bulky methacrylamide derivatives (e.g. 7),^{55.56} maleimides⁵⁷ or methacrylate esters (e.g. 8)⁵⁸ provides stereospecific polymerization (Section 4.2.3). More recent work⁵⁹ has shown that the polymerization of 8 in the presence of a chiral cobalt(II) salophen complex (9) leads to isotactic chains with a one-handed helical structure. It was proposed that 9 selectively retards chain growth of one helix leading to an excess of the onehanded helices. Chiral initiators and transfer agents have also been used to induce chirality.60



Puzin *et al.*⁶¹ reported that the tacticity of PMMA prepared in bulk is influenced (slight increase in syndiotacticity) by very small amounts of titanocene dichloride (10^{-3} M). Selective complexation of the propagating radical was postulated.

Cyclopolymerization of the bis-methacrylates $(10, 11)^{62,63}$ or bis-styrene derivatives $(12)^{64}$ has been used to produce heterotactic polymers and optically active atactic polymers. Cyclopolymerization of racemic 13 by ATRP with a catalyst based on a chiral ligand (Scheme 8.12) gave preferential conversion of the (S,S)-enantiomer.^{65,66}



Scheme 8.12

Control of the polymerization process by changing the reaction medium, through the use of Lewis acids or with templates, has been studied by various groups since the 1950s. Most studies have focused on control of polymerization kinetics or control of reactivity ratios and hence composition in copolymerization. A lesser number of studies have focused on controlling the stereochemistry of the polymer chains. A survey of these studies is provided in the sections that follow. This section is entitled controlling propagation, however, not surprisingly, many of the reagents/reaction conditions mentioned also have an influence on termination kinetics and with conventional methods these effects are not always easy to distinguish. Stereocontrol in radical polymerization was recently reviewed by Habauc and Okamoto⁶⁷ and Matsumoto.⁶⁸

8.3.1 Organic Solvents and Water

Solvent effects on radical polymerization have been reviewed by Coote and Davis,⁶⁹ Coote *et al.*,⁷⁰ Barton and Borsig,⁷¹ Gromov,⁷² and Kamachi⁷³ A summary of kinetic data is also included in Beuermann and Buback's review.⁷⁴ Most literature on solvent effects on the propagation step of radical polymerization deals with influences of the medium on rate of polymerization.

Solvent effects for polymerizations in supercritical CO_2 and in ionic liquids are considered separately in Sections 8.3.2 and 8.3.3 respectively. In this section, we concentrate on effects of organic solvents and water on the rate and stereospecificity of the propagation step of radical polymerization. We exclude from consideration effects where the solvent is itself a reactant providing byproducts by acting as a comonomer or chain transfer agent (chain transfer to solvent is considered in Section 6.2.2.5). We also exclude differences between bulk and solution polymerization that can be ascribed to a simple concentration effect. In solution polymerization, the rate of propagation should be slowed with reference to that seen in bulk monomer simply because of dilution of monomer. Other reactions of the propagating radicals that do not depend on the monomer concentration can proceed at the same or a similar rate notwithstanding any influence of chain length. These include, radical-radical termination, chain transfer to species other than monomer and intra-molecular rearrangement by cyclization, ring opening or backbiting.

An attractive feature of using the solvent as an agent to control propagation in solution polymerization is that solvents when used are usually present in very large excess in relation to any radical species. Of course, economic, solubility, toxicity, waste disposal, and other considerations limit the range of solvents that can be employed in an industrial polymerization process.

Solvent effects on the reactions of small radicals have been discussed in general terms in Chapter 2 (see 2.3.6.2 & 2.4.5). Small, yet easily discernible, solvent effects have been reported for many reactions involving neutral radicals. These effects on the rates of radical reactions often appear insignificant when

compared with the much larger effects observed for similar reactions involving ionic species which may range to orders of magnitude.⁷⁵

Where monomers or radicals are charged, readily ionizable or capable of forming hydrogen bonds, mechanisms whereby the solvent could affect radical reactivity by disruption or involvement of hydrogen bonding may seem obvious. For other systems mechanisms are often still a matter of controversy even in the case of small radicals (Section 2.3.6.2). There are at least three mechanisms whereby the solvent might modify the outcome of a radical process:

- (a) Formation of a monomer or radical complex with different reactivity and/or specificity than the uncomplexed species.
- (b) Solvation of a transition state or intermediate that may have polar character.
- (c) Preferential solvation of one or more reactants leading to local concentrations being different from those in the medium as a whole.

Furthermore, at least three forms of radical-solvent interaction should be considered:

- (a) Reversible addition to the solvent molecule. For example, formation of a cyclohexadienyl radical in the case of aromatic solvents.
- (b) Formation of a charge transfer complex.
- (c) Orbital interaction with a C-H σ -bond or a π -system but without development of charge separation or bond formation.⁷⁶

8.3.1.1 Homopolymerization

The values of the rate parameters for many homopolymerizations have been shown to be solvent dependent.⁷¹⁻⁷⁴ Large solvent effects are reported for monomers which are ionizable (*e.g.* MAA, AA), give precipitation polymerization (AN), or contain hydroxy or amide groups (*e.g.* HEA, HEMA, AM, NIPAM) which can form hydrogen bonds. Some of the biggest solvent effects are reported for water *vs* other solvents. Substantial dependence of the propagation rate constants on monomer concentration has also been reported with water as solvent. For example, in MAA polymerization at 25 °C the propagation rate constant increases from 600 to 3900 M⁻¹ s⁻¹ on lowering the monomer concentration from 9.34 to 1.71 M.⁷⁴ No pronounced concentration dependence is seen with non-polar solvents.

Very large solvent effects are also observed for systems where the monomers can aggregate either with themselves or another species. For example, the apparent k_p for polymerizable surfactants, such as certain vinyl pyridinium salts and alkyl salts of dimethylaminoalkyl methacrylates, in aqueous solution above the critical micelle concentration (cmc) are dramatically higher than they are below the cmc in water or in non-aqueous media.⁷⁷ This does not mean that the value for the k_p is higher. The heterogeneity of the medium needs to be considered. In the micellar system, the effective concentration of double bonds in the vicinity of the

propagating species can be up to 100-fold greater than the concentration of monomer in the medium considered as a whole. The number of surfactant molecules per micelle can also influence the molecular weight. However, the microstructure (tacticity) of the polymer chains is claimed to be the same as that obtained in bulk polymerization (see also Section 8.3.7).

For less polar monomers, the most extensively studied homopolymerizations are vinyl esters (*e.g.* VAc), acrylate and methacrylate esters and S. Most of these studies have focused wholly on the polymerization kinetics and only a few have examined the microstructures of the polymers formed. Most of the carly rate data in this area should be treated with caution because of the difficulties associated in separating effects of solvent on k_p , k_t and initiation rate and efficiency.

One of the most dramatic examples of a solvent effect on propagation taken from the early literature is for vinyl acetate polymerization.^{78,79} Kamachi *et al.*⁷⁸ reported a *ca.* 80-fold reduction in k_p (30°C) on shifting from ethyl acetate to benzonitrile solvent (Table 8.1). Effects on polymer structure were also reported. Hatada *et al.*⁸⁰ conducted a ¹H NMR study on the structure of the PVAc formed in various solvents. They found that PVAc (M _n~20000) produced in ethyl acetate solvent has ~0.7 branches/chain while that formed in aromatic solvents is essentially unbranched.

| Solvent | $k_{\rm p} \times 10^{-2} ({\rm M}^{-1} {\rm s}^{-1})$ | Solvent | $k_{\rm p} \times 10^{-2} ({\rm M}^{-1} {\rm s}^{-1})$ |
|----------------|--|------------------------|--|
| benzonitrile | 8 | fluorobenzene | 97 |
| phenyl acetate | 37 | benzene-d ₆ | 113 |
| anisole | 48 | benzene | 117 |
| chlorobenzene | 61 | ethyl acetate | 637 |
| ethyl benzoate | 37 | | |

Table 8.1 Solvent Effect on Homopropagation Rate Constants for VAc at 30°C⁷⁸

Solvent effects on k_p in polymerizations of MMA⁸¹⁻⁸⁷ and S^{81,84-86,88} have been widely studied but are generally small by comparison and there appears to be no clear correlation with solvent dielectric constant or other solvent properties. When solvent effects are observed, does the solvent modify the reactivity of the propagating radicals, the reactivity of the monomer, the homogeneity of the reaction medium or all of these? Experimental data from, for example, PLP experiments (Section 4.5.2) can be used to calculate the propagation rate constants as a function of the reaction medium. Equally one can assume that k_p remains constant and calculate the effective monomer concentration in the proximity of the chain end. The experimental data typically do not allow casy discrimination between whether either or both are varying nor should one necessarily expect a universal rule to apply. Explanations for apparent conversion and chain length dependence of k_p can also be formulated in terms of effects on local monomer concentrations. Termination rate constants may also be affected by solvent quality.^{70,74}

The heterogeneity of the reaction medium is also important in determining the molecular weight and k_p in solution polymerization of macromonomers.⁸⁹ The magnitude of the effect varies according to the solvent quality. PS macromonomer chains in good solvents (*e.g.* toluene) have an extended conformation whereas in poor solvents (*e.g.* methylcyclohexane) chains are tightly coiled.⁸⁹ As a consequence, the radical center may see an environment that is medium dependent (see also Sections 7.6.5 and 8.3.7).

The tacticity of polymers formed by radical polymerization can also be influenced by solvent and by temperature.⁹⁰ Fluoro-alcohol solvents have been shown to have a significant influence on the tacticity of PVAc and other vinyl esters.⁹¹ Different effects are seen for VAc (more syndiotactic, fraction of *rr* dyads enhanced), vinyl propionate and other vinyl alkanoates (more heterotactic, fraction of *mr* dyads enhanced) and vinyl benzoate (more isotactic, fraction of mm dyads enhanced).⁹² The effect is greater for lower polymerization temperatures and for more bulky fluoro-alcohols. The effect of fluoro-alcohol solvents on polymerization of methacrylate esters has also been investigated^{93,94} and data for -40 °C are shown in Table 8.2. Polymerization in fluoro-alcohol solvents enhances syndiotacticity of PMMA and PEMA.⁹⁴ For PtBMA, syndiotacticity is reduced.⁹⁴ Again, the effect is greatest at the lowest reaction temperature. These solvent effects were attributed to steric factors associated with hydrogen bonding to the ester C=O. The solvent is said to enhance the bulkiness of the ester group of both the propagating radical and the monomer.^{93,94}

| Solvent ^a | MMA | EMA | tBMA |
|---------------------------------------|---------------|-------------------------------|---------------|
| | nim:mr:rr | mm:mr:rr | mm:mr:rr |
| toluene | 1.0:23.0:76.0 | 4.7:18.2:77.1 | 2.7:22.8:74.4 |
| methanol | 1.6:23.4:74.9 | 11.8:16.2:72.0 | - |
| HFIP | 1.3:19.6:79.1 | 4.4:15.3:80.3 | - |
| PFTB | 0.5:16.6:82.9 | :82.9 0.9:14.7:84.4 1.4:33.2: | 1.4:33.2:65.4 |
| $= 11\Gamma 1D = 1 \dots d1 \dots d1$ | | ana thutanal | |

 Table 8.2 Effect of Solvent on Tacticity of Poly(alkyl methacrylate) at -40 °C⁹⁴

a HFIP = hexafluoro-isopropanol, PFTB = perfluoro-*t*-butanol

Tacticity of MAA is influenced by solvent,^{90,95} the presence of amines (Table 8.3)⁹⁰ and complexation. PMAA appears more isotactic when formed in a nonhydrogen-bonding solvent.^{90,95} Polymerization of MAA in CHCl₃ in the presence of 14 or 15 also yields a more isotactic polymer.⁹⁰ Polymerization of zinc complexes of MAA also yields more isotactic polymers.⁹⁶



Table 8.3 Effect of Amines on Tacticity of Poly(methacrylic acid) at 60 °C⁹⁰

| Solvent ^a | Amine | mm:mr:rr ^a |
|----------------------|-------------------|-----------------------|
| MeOH | none | 4.0:34.6:61.4 |
| MeOH | 14 ^b | 3.8:29.1:67.0 |
| CIICl ₃ | none | 8.1:41.0:50.9 |
| CHCl3 | 15 | 12.3:47.0:40.7 |
| CHCl3 | 14^{b} | 16.3:48.8:34.9 |

a Polymerization of MAA (0.1 M) in presence of amine (stoichiometric NH₂) with AIBN initiator (0.004 M). Tacticity determined for PMMA obtained by esterification of PMAA formed. b (R,R)-configuration.

8.3.1.2 Copolymerization

The effects of solvent on radical copolymerization are mentioned in a number of reviews.^{69-72,97,98} For copolymerizations involving monomers that are ionizable or form hydrogen bonds (AM, MAM, HEA, HEMA, MAA, *etc.*) solvent effects on reactivity ratios can be dramatic. Some data for MAA-MMA copolymerization are shown in Table 8.4.⁹⁹

For MMA-MAA copolymerizations carried out in the more hydrophobic solvents (toluene, dioxane), MAA is the more reactive towards both propagating species while in water MMA is the more reactive. In solvents of intermediate polarity (alcohols, dipolar aprotic solvents), there is a tendency towards alternation. For these systems, choice of solvent could offer a means of controlling copolymer structure.

For copolymerizations between non-protic monomers solvent effects are less marked. Indeed, early work concluded that the reactivity ratios in copolymerizations involving only non-protic monomers (*e.g.* S, MMA, AN, VAc, *etc.*) should show no solvent dependence.^{100,101} More recent studies on these and other systems (*e.g.* AN-S,¹⁰²⁻¹⁰⁵ E-VAc,¹⁰⁶ MAN-S,¹⁰⁷ MMA-S,¹⁰⁸⁻¹¹⁰ MMA-VAc¹¹¹) indicate small yet significant solvent effects (some recent data for AN-S copolymerization are shown in Table 8.5). However, the origin of the solvent effect in these cases is not clear. There have been various attempts to rationalize solvent effects on copolymerization by establishing correlations between radical reactivity and various solvent and monomer properties.^{71,72,97,99} None has been entirely successful.

| solvent | $r_{ m MMA}$ | r_{MAA} | |
|--------------|--------------|-----------|--|
| toluene | 0.10 | 1.06 | |
| dioxane | 0.12 | 1.33 | |
| acetonitrile | 0.27 | 0.03 | |
| acetone | 0.31 | 0.63 | |
| DMSO | 0.78 | 0.23 | |
| isopropanol | 0.78 | 0.33 | |
| ethanol | 0.80 | 0.60 | |
| acetic acid | 0.80 | 0.78 | |
| DMF | 0.98 | 0.68 | |
| water | 2.61 | 0.43 | |

 Table 8.4
 Solvent Dependence of Reactivity Ratios for MMA-MAA

 Copolymerization at 70°C^{a.99}

a Reactivity ratios estimated from composition data.

 Table 8.5
 Solvent Dependence of Penultimate Model Reactivity Ratios for S-AN

 Copolymerization at 60°C¹⁰³

| Solvent | $r_{\rm SS}$ | rAS | $r_{\rm SA}$ | $r_{\rm AA}$ | |
|--------------|--------------|-------|--------------|--------------|--|
| bulk | 0.232 | 0.566 | 0.087 | 0.036 | |
| toluene | 0.242 | 0.566 | 0.109 | 0.133 | |
| acetonitrile | 0.322 | 0.621 | 0.105 | 0.052 | |
| | | | | | |

The solvent in a bulk copolymerization comprises the monomers. The nature of the solvent will necessarily change with conversion from monomers to a mixture of monomers and polymers, and, in most cases, the ratio of monomers in the feed will also vary with conversion. For S-AN copolymerization, since the reactivity ratios are different in toluene and in acetonitrile, we should anticipate that the reactivity ratios are different in bulk copolymerizations when the monomer mix is either mostly AN or mostly S. This calls into question the usual method of measuring reactivity ratios by examining the copolymer composition for various monomer feed compositions at very low monomer conversion. We can note that reactivity ratios can be estimated for a single monomer feed composition by analyzing the monomer sequence distribution. Analysis of the dependence of reactivity ratios determined in this manner of monomer feed ratio should therefore provide evidence for solvent effects. These considerations should not be ignored in solution polymerization either.

Harwood¹⁵² proposed that the solvent need not directly affect monomer reactivity, rather it may influence the way the polymer chain is solvated. Evidence for the proposal was the finding for certain copolymerizations, while the terminal model reactivity ratios appear solvent dependent, copolymers of the same overall composition had the same monomer sequence distribution. This was explained in terms of preferential monomer sorption such that the polymer composition determined the relative monomer concentration in the vicinity of the reactive chain end. This phenomenon was called "the bootstrap effect".^{112,113} A partition coefficient K was defined as eq. 1:

$$K = \frac{[M_{\rm A}]/[M_{\rm B}]}{[M_{\rm Ao}]/[M_{\rm Bo}]} \tag{1}$$

where $[M_A]/[M_B]$ is the ratio of monomer concentrations in the vicinity of the reactive chain end and $[M_{Ao}]/[M_{Bo}]$ is the global ratio. The conditional probabilities which determine the triad fractions are dependent on $[M_A]/[M_B]$ rather than $[M_{Ao}]/[M_{Bo}]$. The value of $[M_A]/[M_B]$ is determined by the polymer composition.

The apparent terminal model reactivity ratios are then: $r_{AB}^{app} = r_{AB}K$ and $r_{BA}^{app} = r_{BA} / K$. It follows that $r_{AB}^{app} r_{BA}^{app} = r_{AB}r_{BA} = const$. The bootstrap effect does not require the terminal model and other models (penultimate, complex participation) in combination with the bootstrap effect have been explored.^{103,114,115} Variants on the theory have also appeared where the local monomer concentration is a function of the monomer feed composition.¹¹⁶

The effects of solvent on reactivity ratios and polymerization kinetics have been analyzed for many copolymerizations in terms of this theory.⁹⁸ These include copolymerizations of S with MAH,^{117,118} S with MAA,¹¹² S with MMA,^{116,117,119-121} S with HEMA,¹²² S with BA,^{123,124} S with AN,^{103,115,125} S with MAN,¹¹³ BA with MMA^{126,127} and tBA with HEMA.¹²⁸ It must, however, be pointed out that while the experimental data for many systems are consistent with a bootstrap effect, it is usually not always necessary to invoke the bootstrap effect and much effort has been put into finding evidence both for or against the theory.^{69,70,98,129,130} If a bootstrap effect applies, then reactivity ratios cannot be determined by analysis of composition or sequence data in the normal manner discussed in Section 7.3.3.

Studies on the reactions of small model radicals with monomers provide indirect support but do not prove the bootstrap effect.¹³¹ Krstina *et al.*¹³¹ showed that the reactivities of MMA and MAN model radicals towards MMA, S and VAc were independent of solvent. However, small but significant solvent effects on reactivity ratios are reported for MMA/VAc¹¹¹ and MMA/S^{117,119} copolymerizations. For the model systems, where there is no polymer coil to solvate, there should be no bootstrap effect and reactivities are determined by the global monomer ratio $[M_{Ao}]/[M_{Bo}]$.¹³¹

Other phenomena attributed to a bootstrap or similar effects include

(a) The dependence of copolymer composition on molecular weight in certain copolymerizations.¹³²⁻¹³⁴ There are other explanations for the molecular weight dependence of copolymer composition that relate to specificity shown in the

initiation process (Section 7.5.6). However, these effects only apply to relatively low molecular weights (<20 units).

(b) The observation of significant solvent effects in macromonomer copolymerization.¹³⁵ Tsukahara *et al.*¹³⁵ found that when copolymerizing macromonomers, the choice of solvent has a substantial influence on the reactivity ratios, the molecular weight of the polymer, and the particle size distribution of the final product. They interpreted their data in terms of the effects of solvent on the degree of interpretation between unlike polymer chains.

8.3.2 Supercritical Carbon Dioxide

Polymerization, including radical polymerization, in supercritical CO_2 has been reviewed.^{136,137} It should be noted supercritical CO_2 while a good solvent for many monomers is a very poor solvent for polymers such as the (meth)acrylates and S. As a consequence, with the exception of certain fluoropolymers and polymerizations taken to very low conversion, most polymerizations in supercritical CO_2 are of necessity precipitation, dispersion or emulsion polymerizations.

Several studies have been directed towards determining the kinetics of radical polymerization in supercritical CO₂ using PLP (Section 4.5.2). While some early results^{138,139} suggested that $k_p(CO_2)$ for MMA was not significantly different to $k_p(bulk)$, more recent work has shown that $k_p(CO_2)$ for MMA¹⁴⁰ and various acrylate esters (MA,⁷⁴ BA,^{140,141} DA⁷⁴) are significantly reduced from values for bulk polymerization. Values of $k_p(CO_2)$ for S¹⁴² and VAc¹⁴³ are not significantly different to $k_p(bulk)$.

8.3.3 Ionic liquids

Room temperature ionic liquids are currently receiving considerable attention as environmentally friendly alternatives to conventional organic solvents in a variety of contexts.¹⁴⁴ The ionic liquids have this reputation because of their high stability, inertness and, most importantly, extremely low vapor pressures. Because they are ionic and non-conducting they also possess other unique properties that can influence the yield and outcome of organic transformations. Polymerization in ionic liquids has been reviewed by Kubisa.¹⁴⁵ Commonly used ionic liquids are tetra-alkylammonium, tetra-alkylphosphonium, 3-alkyl-1-methylimidazolium (**16**) or alkyl pyridinium salts (**17**). Counter-ions are typically PF₆⁻ and BF₄⁻, though many others are known.



Harrison *et al.*^{146,147} have used PLP (Section 4.5.2) to examine the kinetics of MMA polymerization in the ionic liquid **18** (bmimPF₆). They report a large (*ca* 2-fold) enhancement in k_p and a reduction in k_t . This property makes them interesting solvents for use in living radical polymerization (Chapter 9). Ionic liquids have been shown to be compatible with $\Lambda TRP^{148-156}$ and $R\Lambda FT$.^{157,158} However, there are mixed reports on compatibility with NMP.^{159,160} Widespread use of ionic liquids in the context of polymerization is limited by the poor solubility of some polymers (including polystyrene) in ionic liquids.

There is also some evidence that the ionic liquid medium affects polymer structure. Biedron and Kubisa¹⁵⁰ reported that the tacticity of PMA prepared in the chiral ionic liquid **19** is different from that prepared in conventional solvent. It is also reported that reactivity ratios for MMA-S copolymerization in the ionic liquid **18**¹⁶¹ differ from those observed for bulk copolymerization.

8.3.4 Lewis Acids and Inorganics

Lewis acids are known to form complexes both with monomers and with propagating species. Their addition to a polymerization medium, even in catalytic amounts, can bring about dramatic changes in rate constants in homopolymerization (Section 8.3.4.1) and reactivity ratios in copolymerization (Section 8.3.4.2). Early work in this area has been reviewed by Bamford¹⁶² and Barton and Borsig.⁷¹ There is significant current interest in using Lewis Acids in establishing tacticity control in homopolymerization (see 8.3.4.1).

8.3.4.1 Homopolymerization

In 1957, Bamford *et al.*¹⁶³ reported that the addition of small amounts of lithium chloride brought about a significant (up to two-fold) enhancement in the rate of polymerization of AN in DMF and led to a higher molecular weight polymer. Subsequent studies have shown this to be a more general phenomenon for polymerizations involving, in particular, acrylic and vinylheteroaromatic monomers in the presence of a variety of Lewis acids.⁷¹

For the case of polymerization of AN in DMF, measurements of the absolute rate constants associated with the polymerizations indicated that the rate of initiation (by AIBN) was not significantly affected by added lithium salts. The enhancement in the rate of polymerization was therefore attributed to an increase in k_p . The value of k_t remains essentially unchanged except when very high concentrations of the Lewis acid are employed.

Zubov *et al.*¹⁶⁴ suggested that during MMA polymerization in the presence of Lewis acids (*e.g.* AlBr₃) complexation occurs preferentially with the propagating radical rather than with monomer. They suggested a mechanism in which the metal ion is transferred to the incoming monomer in the transition state for addition so as to remain with the active chain end. It is known that Lewis acids can bring about significant changes in the appearance of the EPR spectra of MMA propagating radicals and related species.¹⁶⁵

Although it is clear that added Lewis acids affect the rate of polymerization and the molecular weight of homopolymers formed in their presence,⁷¹ the effect on polymer structure is small. There are reports that Lewis acids affect the tacticity.^{67,68,71,90} Otsu and Yamada¹⁶⁶ found a slightly greater proportion of isotactic (*mm*) triads in PMMA formed by bulk polymerization of a 1:1 complex of MMA with zinc chloride than is observed for a similar polymerization of MMA alone. However, for polymerizations carried out in solution or in the presence of lesser amounts of zinc chloride, no effect was observed.¹⁶⁶ For MMA polymerization in solution at 60 °C, a small though significant effect on tacticity (increase in isotactic triads) is seen on addition of 0.2 M scandium triflate¹⁶⁷ and lesser effects with ytterbium triflate and hafnium chloride (Table 8.6).



Lewis acids have a much greater effect on tacticity in polymerization of α -alkoxymethacrylates such as **20**,^{168,169} acrylamides (including AM, NIPAM, DMAM)¹⁷⁰⁻¹⁷³ and methacrylamides (including MAM, MMAM) (Table 8.6).^{170,171,174} The solvent has a significant effect on the magnitude of the effect observed and little influence is observed for polymerizations carried out in aqueous media. The effect of Lewis acids on tacticity is significantly greater for lower polymerization temperatures. In the polymerizations of acrylamide and methacrylamides a very significant influence on tacticity was seen for 10 mole% Yb(OTI)₃ with respect to monomer and the effect was not significantly enhanced for greater concentrations of Lewis acid.

It is also possible that complexation of monomer or propagating species could influence the regiospecificity of addition. However, since the effect is likely to be an enhancement of the usual tendency for head-to-tail addition, perhaps it is not surprising that such effects have not been reported.

| Monomer | Solvent | Lewis Acid | Conc. (M) | mm:mr:rr | $P(m)^a$ |
|------------------------|-------------------|----------------------|-----------|----------|----------|
| NIPAM ^{b,172} | CHCl ₃ | none | 0 | - | 0.45 |
| 172 | CHCl₃ | Yb(OTf)₃ | 0.20 | - | 0.58 |
| 172 | CHCl ₃ | Y(OTf) ₃ | 0.20 | - | 0.62 |
| 172 | MeOH | Y(OTf) ₃ | 0.20 | - | 0.80 |
| c.172 | H_2O | $Y(OTf)_3$ | 0.20 | - | 0.57 |
| MMAM ^{d,174} | MeOH | none | 0 | 2:29:69 | 0.17 |
| 174 | MeOH | $Sc(OTf)_3$ | 0.20 | 28:55:17 | 0.56 |
| 174 | MeOH | Y(OTf) ₃ | 0.20 | 46:40:14 | 0.66 |
| 174 | MeOH | Yb(OTf) ₃ | 0.20 | 46:44:10 | 0.68 |
| 175 | THF | Yb(OTf) ₃ | 0.20 | 32:50:18 | 0.57 |
| 175 | H_2O | Yb(OTf) ₃ | 0.20 | 2:31:67 | 0.18 |
| MAM ¹⁷⁵ | MeOH | none | 0 | 7:39:54 | 0.27 |
| 175 | MeOH | Yb(OTf) ₃ | 0.20 | 36:50:14 | 0.61 |
| 175 | MeOH | Y(OTf)₃ | 0.20 | 33:49:18 | 0.58 |
| MMΛ ^{c.67} | toluene | none | 0 | 3:33:64 | 0.19 |
| 67 | tolucne | HfCl4 | 0.20 | 6:36:58 | 0.21 |
| 67 | CHCl ₃ | Yb(OTf) ₃ | 0.24 | 10:36:54 | 0.23 |
| 67 | toluene | Sc(OTf) ₃ | 0.20 | 14:46:40 | 0.30 |

 Table 8.6. Effect of Lewis Acids on Tacticity of Polymers Formed in High Conversion Radical Polymerizations at 60 °C

a P(m) =0.5 mr + mm. There is evidence of non-Bernoullian statistics for some examples. b NIPAM (2.4 M) with AIBN (0.02 M) polymerized for 24 h at 60°C. c NIPAM (2.4 M) with K₂S₂O₈ (0.02 M) polymerized for 24 h at 60°C. d MMAM (2.0 M) with AIBN (0.02 M) polymerized for 24 h at 60°C. e MAM (2.4 M) with AIBN (0.02 M) polymerized for 2 h at 60°C. f MMA (2.4 M) with AIBN (0.02 M) polymerized for 2 h at 60°C.

8.3.4.2 Copolymerization

The kinetics of copolymerization and the microstructure of copolymers can be markedly influenced by the addition of Lewis acids. In particular, Lewis acids are effective in enhancing the tendency towards alternation in copolymerization of donor-acceptor monomer pairs and can give dramatic enhancements in the rate of copolymerization and much higher molecular weights than are observed for similar conditions without the Lewis acid. Copolymerizations where the electron deficient monomer is an acrylic monomer (*e.g.* AN, MA, MMA) and the electron rich monomer is S or a diene have been the most widely studied.^{164,176-184} Strictly alternating copolymers of MMA and S can be prepared in the presence of, for example, diethylaluminum sesquichloride. In the absence of Lewis acids, there is only a small tendency for alternation in MAA-S copolymerization; terminal model reactivity ratios are *ca* 0.51 and 0.49 - Section 7.3.1.2.3. Lewis acids used include: EtAlCl₂, Et₂AlCl, Et₃Al₂Cl₃, ZnCl₂, TiCl₄, BCl₃, LiClO₄ and SnCl₄.

Various mechanisms (not mutually exclusive) for the influence of Lewis acid on copolymerization have been proposed:

- (a) A ternary complex is formed between acceptor, donor, and Lewis acid. An alternating polymer may be formed by homopolymerization of such a complex.^{176,177}
- (b) The Lewis acid forms a binary complex with the acceptor monomer. The electron deficiency of the double bond is enhanced by complexation with the Lewis acid and thus its reactivity towards nucleophilic radicals is greater.¹⁸²
- (c) Spontaneous copolymerization, possibly by a biradical mechanism.¹⁸⁵
- (d) Complexation of the propagating radical to create a species with selectivity different to that of the normal propagating radical.

Most recent work is in accord with mechanism (b). In an effort to distinguish these mechanisms studies on model propagating species have been carried out.¹⁸⁶⁻¹⁸⁹ For S-MMA polymerization initiated by AIBMe- α -¹³C (Scheme 8.13) it has been established by end group analysis that extremely small amounts of ethyl aluminum sesquichloride (<10⁻³ M with 1.75 M monomers) are sufficient to cause a substantial enhancement in specificity for adding S in the initiation step. This result suggests that complexation of the propagating radical may be sufficient to induce alternating copolymerization but does not rule out other hypotheses.



Scheme 8.13

The primary aim of most studies on Lewis acid controlled copolymerization has been the elucidation of mechanism and only low conversion polymerizations are reported. Sherrington *et al.*¹⁸⁴ studied the high conversion synthesis of alternating MMA-S copolymers in the presence of Lewis acids on a preparative scale. Many Lewis acids were found to give poor control (*i.e.* deviation from 50:50 composition) and were further complicated by side reactions including cross-linking. They found that the use of catalytic BCl₃ as the Lewis acid and photoinitiation gave best results.

Matyjaszewski and coworkers ^{190,191} have explored living radical copolymerization (ATRP and RAFT) in the presence of Lewis acids.

Controlling Polymerization

8.3.5 Template Polymerization

The possibility of using a template polymer to organize the monomer units prior to their being "zipped up" by the attack of a radical species has long attracted interest and the field of template polymerization has been the subject of a number of reviews.¹⁹²⁻¹⁹⁵ Template polymerization can also be found under such headings as molecular imprinting, supramolecular chemistry and topological or topochemical polymerization (Section 8.3.7) though some of these terms have additional meaning. Template polymerization, as used here and as its name suggests, involves the formation of a *daughter* polymer on a preformed *parent* polymer.

The interest in this area may be seen to stem from the biological area where the phenomenon is well known and accounts for the regularity in the structure of natural proteins and polynucleotides. Such polymers are efficiently synthesized by enzymes which are capable of organizing monomer units within regularly structured molecular-scale spaces and exploiting weak forces such as hydrogen bonds and Van der Waal forces to control the polymerization process.

The literature distinguishes two limiting forms of template polymerization.¹⁹²⁻

- (a) Where the monomer is associated with the template and, ideally, initiation, propagation, and termination all occur on the template.
- (b) Where only the propagating chain associates with the template. The rate of polymerization is limited by the rate at which monomer is attached from the bulk solution.

The interaction of the template with monomer and/or the propagating radical may involve solely Van der Waals forces or it may involve charge transfer complexation, hydrogen bonding, or ionic forces (Section 8.3.5.1). In other cases, the monomer is attached to the template through formal covalent bonds (Section 8.3.5.2).

8.3.5.1 Non-covalently bonded templates

In 1972, Buter *et al.*¹⁹⁶ reported that polymerization of MMA in the presence of isotactic PMMA leads to a greater than normal predominance of syndiotactic sequences during the early stages of polymerization. Other investigations of this system supporting^{197,198} and disputing¹⁹⁹ this finding appeared. The mechanism of the template polymerization is thought to involve initial stereocomplex formation between the oligomeric PMMA propagating radical (predominantly syndiotactic) and the isotactic template polymer with subsequent monomer additions being directed by the environment of the template. Isotactic and syndiotactic PMMA have been shown to form a 1:2 stereocomplex.¹⁹⁷ Recently, Serizawa *et al.*²⁰⁰ showed that comparatively pure isotactic PMMA could be prepared within the

confines of a matrix comprising a thin porous film of syndiotactic PMMA. The matrix influenced both the molecular weight and the tacticity.

The nature of the interaction between the monomer and the template is more obvious in cases where specific ionic or hydrogen bonding is possible. For example, *N*-vinylimidazole has been polymerized along a PMAA template^{201,202} and acrylic acid has been polymerized on a *N*-vinylpyrrolidone template.²⁰³ The daughter PAA had a similar degree of polymerization to the template and had a greater fraction of isotactic triads than PAA formed in the absence of the template.

It is well known that rates of polymerizations can increase markedly with the degree of conversion or with the polymer concentration. Some workers have attributed this solely or partly to a template effect. It has been proposed²⁰⁴ that adventitious template polymerization occurs during polymerizations of AA, MAA and AN, and that the gel or Norrish-Trommsdorff effect observed during polymerizations of these monomers is linked to this phenomenon. However, it is difficult to separate possible template effects from the more generic effects of increasing solution viscosity and chain entanglement at high polymer concentrations on rates of termination and initiator efficiency (Section 5.2.1.4).

There are also reports of template effects on reactivity ratios in copolymerization. For example, Polowinski²⁰⁵ has reported that both kinetics and reactivity ratios in MMA-MAA copolymerization in benzene are affected by the presence of a PVA template.



21

A template polymer may allow the use of monomers that do not otherwise undergo polymerization. An example is the dinitrobenzoate derivative **21**; nitrobenzene derivatives are usually thought of as radical inhibitors (see 5.3.7), thus radical polymerization of monomer with such functionality is unlikely to be successful. Polymerization of **21** on a poly(N-vinylearbazole) template succeeded in producing a high molecular weight polymer.²⁰⁶ It was envisaged that the monomer **21** forms a charge transfer complex with the electron donating carbazole group.

8.3.5.2 Covalently bonded templates

Template polymerizations where the monomer is covalently bound to the template clearly have limitations if polymers of high molecular weight or large quantities are required. However, their use offers much greater control over

daughter polymer structure. The product in such cases is a ladder polymer and this may be viewed as a special case of cyclopolymerization (Section 4.4.1).



Scheme 8.14

Kämmerer *et al.*²⁰⁷⁻²⁰⁹ have conducted extensive studies on the template polymerization of acrylate or methacrylate derivatives of polyphenolic oligomers **22** with $\overline{X}_n \leq 5$ (Scheme 8.14). Under conditions of low "monomer" and high initiator concentration they found that \overline{X}_n for the daughter polymer was the same as \overline{X}_n for the parent. The possibility of using such templates to control microstructure was considered but not reported.

Feldman *et al.*^{210,211} and Wulff *et al.*²¹² have examined other forms of template controlled oligomerization of acrylic monomers. The template (**23**) has initiator and transfer agent groups attached to a rigid template of precisely defined structure.^{210,211} Polymerization of MMA in the presence of **23** gave a 3 unit oligo(MMA) as *ca* 66% of the polymeric product. The stereochemistry of the oligomer was reported to be "different" from that of atactic PMMA.



Wulff *et al.*²¹² attached vinyl groups to a large chiral sugar based template molecule and then copolymerized this substrate with various monomers. With MMA and MAN they achieved some optical induction. This approach has been extended in studies of higher molecular weight systems.²¹³⁻²¹⁶ Thus, PVA was esterified with methacryloyl chloride to give a "multimethacrylate" (24) and

440

polymerized to give a ladder polymer. "Multimethacrylates" based on PHEMA were also described. The daughter polymer was hydrolyzed to PMMA but only characterized in terms of molecular weight. The value of \bar{X}_n for the daughter polymer was greater than \bar{X}_n for the parent template indicating some intertemplate reaction. These workers also examined the copolymerization of partially methacrylated PVA with MMA. It has not been established whether the tacticity of parent PVA or the presence of head-to-head and tail-to-tail linkages has an effect on the microstructure of the daughter polymer.²¹³

Saito *et al.*²¹⁷⁻²¹⁹ have examined the polymerization of multimethacrylates prepared from β -cyclodextrin. Polymerization using ATRP conditions gave a bimodal molecular weight distribution for the derived PMMA composed predominantly of oligomers of 7 or 14 units indicating that there was little intermolecular reaction

A new form of template polymerization based on ring-opening polymerization of 4-methylenedioxalane has been reported by Endo and coworkers (Scheme 8.15).^{220,221} For this system, the monomer is covalently bound and the daughter polymer is released from the template as a consequence of the polymerization process.



Scheme 8.15

8.3.6 Enzyme Mediated Polymerization

A number of recent papers have explored enzyme-mediated polymerization. Monomers polymerized include MMA, S, AM and derivatives. The area has been reviewed by Singh and Kaplan²²² and Gross *et al.*²²³

One of the most used systems involves use of horseradish peroxidase, a β -diketone (most commonly 2,4-pentandione), and hydrogen peroxide.²²² Since these enzymes contain iron(II), initiation may involve decomposition of hydrogen peroxide by a redox reaction with formation of hydroxy radicals. However, the proposed initiation mechanism²²² involves a catalytic cycle with enzyme activation by hydrogen peroxide and oxidation of the β -diketone to give a species which initiates polymerization. Some influence of the enzyme on tacticity and molecular

weight has been reported. However, further study is required to define the origin of the effects observed.^{222,223}

8.3.7 Topological Radical Polymerization

In this section we consider topological or topochemical polymerizations where monomers are constrained by being part of an organic crystal,^{68,224} a Langmuir-Blodgett film, a liquid crystal, a lipid bilayer, a micellar aggregate,²²⁵ or a supramolecular assembly.²²⁶ Unlike template polymerization there is no parent polymer to organize the monomer. Rather polymerization occurs in a crystalline or otherwise organized phase that may comprise only the monomer.



Certain monomers crystallize in a conformation such that they can be zipped together without changing the symmetry of the crystal lattice. In the crystalline state, the arrangement of monomers is strictly determined by crystal packing. Polymerization is usually initiated by irradiation with UV, X- or γ -rays and is assumed to proceed by a radical mechanism. For example, muconic acid esters (25, 27, 29) and ammonium salts (26, 28, 30) can be stereospecifically polymerized in the crystalline state to high conversion.^{224,227,228} This form of
polymerization requires engineering a crystal containing monomer units in appropriate juxtaposition.

Amphiphilic molecules and macromolecules form micelles in aqueous media where the more hydrophobic segments are aggregated to form a core while the more hydrophilic head groups are exposed to the aqueous medium. The molecules can contain monomer functionality within either the more hydrophilic or more hydrophobic segment. The polymerization of surface active monomers has been reviewed by Nagai.²²⁵ Micelles are not static but are dynamic structures and there is rapid exchange of the surfactant monomers between the micellar phase and the aqueous phase and between individual micelles. It has been stated that radical polymerization of linear polymerizable surfactants (surfiners) formed into micelles above the critical micelle concentration (cmc) is unlikely to be controlled by the topology of the micelle.²²⁹ The mobility of the surfactant species is generally high and the rate of exchange of surfactant molecules between the micelle and solution is rapid with respect to the rate of polymerization. As a consequence, neither molecular weight nor polymer stereochemistry is controlled. Nonetheless, rates of polymerization can be high with respect to rates of polymerization observed for a similar monomer concentration in non-organized media. The effective local monomer concentrations in micellar systems can approach, or by organization surpass, those seen in bulk polymerization.

There are a few exceptions to this general rule. One of the few examples of an effect on polymer stereochemistry was provided by Dais *et al.*²³⁰ who found that polymerization of **31** above the cmc initiated by γ -irradiation at 25 °C yields polymer composed entirely of syndiotactic dyads P(m) = 0. When the double bond was distant from the polar head group in **32**, the tacticity observed was similar to that observed in solution polymerization $P(m)\sim 0.18$. Polymerization of **31** at higher temperatures (50 °C) initiated by AIBN also showed no sign of tacticity control. The stereospecific polymerization of **31** was attributed to organization of the methacrylate moiety on the surface of the micelle.



Cetyltrimethylammonium 4-vinylbenzoate (33) forms rod-like micelles that can be stabilized by radical polymerization. The resulting structure, was observed by small-angle neutron scattering to retain its original rod-like architecture and showed enhanced thermal stability and did not dissociate upon dilution.

Some of the more remarkable examples of this form of topologically controlled radical polymerization were reported by Percec *et al.*²³¹⁻²³⁴ Dendron macromonomers were observed to self-assemble at a concentration above 0.20 mol/L in benzene to form spherical micellar aggregates where the polymerizable double bonds are concentrated inside. The polymerization of the aggregates initiated by AIBN showed some living characteristics. Dispersities were narrow and molecular weights were dictated by the size of the aggregate. The shape of the resultant macromolecules, as observed by atomic force microscopy (AFM), was found to depend on \overline{X}_n . With $\overline{X}_n < 20$, the polymer remained spherical. On the other hand, with $\overline{X}_n > 20$, the polymer became cylindrical.^{231,232}

Further examples of micellar stabilization when micelles are composed of block copolymers formed by living radical polymerization are mentioned in Section 9.9.2.

8.4 References

- 1. Moad, G.; Solomon, D.H. Aust. J. Chem. 1990, 43, 215.
- 2. Lehrle, R.S.; Peakman, R.E.; Robb, J.C. Eur. Polym. J. 1982, 18, 517.
- 3. Rudin, A.; Samanta, M.D.; Reilly, P.M. J. Appl. Polym. Sci. 1979, 24, 171.
- 4. Cameron, G.G.; Meyer, J.M.; McWalter, I.T. Macromolecules 1978, 11, 696.
- 5. Wall, L.A.; Straus, S.; Florin, R.E.; Fetters, L.J. J. Res. Nat. Bur. Stand. 1973, 77, 157.
- Cascaval, C.N.; Straus, S.; Brown, D.W.; Florin, R.E. J. Polym. Sci., Polym. Symp. 1976, 57, 81.
- 7. Singh, M.; Nandi, U.S. J. Polym. Sci., Polym. Lett. Ed. 1979, 17, 121.
- 8. Howell, B.A.; Cui, Y.M.; Priddy, D.B. Thermochim. Acta 2003, 396, 167.
- 9. Guaita, M. Br. Polym. J. 1986, 18, 226.
- 10. Grassie, N.; Melville, H.W. Proc. R. Soc., London 1949, A199, 1.
- 11. Grassic, N.; Kerr, W.W. Trans. Faraday Soc. 1959, 55, 1050.
- 12. Peterson, J.D.; Vyazovkin, S.; Wight, C.A. Macromol. Chem. Phys. 2001, 202, 775.
- Schildknecht, C.E. In *Polymerization Processes*; Schildknecht, C.E.; Skeist, I., Eds.; Wiley: New York, 1977; p 88.
- 14. Boundy, R.H.; Boyer, R.F., Eds. Styrene, Its Polymers, Copolymers and Derivatives; Reinhold: New York, 1952.
- 15. Krstina, J.; Moad, G.; Solomon, D.H. Eur. Polym. J. 1989, 25, 767.
- 16. Moad, G.; Solomon, D.H.; Willing, R.I. Macromolecules 1988, 21, 855.
- 17. Cameron, G.G.; MacCallum, J.R. J. Macromol. Sci., Rev. Macromol. Chem. 1967, C1, 327.
- Moad, G.; Solomon, D.H.; Johns, S.R.; Willing, R.I. *Macromolecules* 1984, 17, 1094.
- Krstina, J.; Moad, G.; Willing, R.I.; Danek, S.K.; Kelly, D.P.; Jones, S.L.; Solomon, D.H. *Eur. Polym. J.* **1993**, *29*, 379.
- Roland, A.I.; Stenzel, M.; Schmidt-Naake, G. Angew. Makromol. Chem 1998, 254, 69.
- 21. Roland, A.I.; Schmidt-Naake, G. J. Anal. Appl. Pyrol. 2001, 58, 143.
- 22. Postma, A.; Davis, T.P.; Moad, G.; O'Shca, M. Macromolecules 2005, 38, 5371.
- 23. Holland, B.J.; Hay, J.N. Polym. Degr. Stab. 2002, 77, 435.
- 24. Peterson, J.D.; Vyazovkin, S.; Wight, C.A. J. Phys. Chem. B 1999, 103, 8087.

- 25. Stoliarov, S.I.; Westmoreland, P.R.; Nyden, M.R.; Forney, G.P. Polymer 2003, 44, 883.
- 26. Manring, L.E. Macromolecules 1988, 21, 528.
- 27. Manring, L.E. Macromolecules 1991, 24, 3304.
- 28. Hodder, A.N.; Holland, K.A.; Rae, I.D. J. Polym. Sci., Polym. Lett. Ed. 1983, 21, 403.
- 29. Manring, L.E.; Sogah, D.Y.; Cohen, G.M. Macromolecules 1989, 22, 4652.
- 30. Manring, L.E. Macromolecules 1989, 22, 2673.
- Cacioli, P.; Moad, G.; Rizzardo, E.; Serelis, A.K.; Solomon, D.H. Polym. Bull. 1984, 11, 325.
- 32. Kashiwagi, T.; Kirata, T.; Brown, J.E. Macromolecules 1985, 18, 131.
- 33. Meisters, A.; Moad, G.; Rizzardo, E.; Solomon, D.H. Polym. Bull. 1988, 20, 499.
- 34. Bagby, G.; Lehrle, R.S.; Robb, J.C. Polymer 1969, 10, 683.
- 35. Kashiwagi, T.; Inaba, A.; Brown, J.E.; Hatada, K.; Kitayama, T.; Masuda, E. *Macromolecules* **1986**, *19*, 2160.
- 36. Peterson, J.D.; Vyazovkin, S.; Wight, C.A. Macromol. Rapid Commun. 1999, 20, 480.
- 37. Rizzardo, E.; Solomon, D.H. J. Macromol. Sci., Chem. 1979, A13, 1005.
- 38. Rizzardo, E.; Solomon, D.H. Polvm. Bull. 1979, 1, 529.
- 39. Bednarek, D.; Moad, G.; Rizzardo, E.; Solomon, D.H. Macromolecules 1988, 21, 1522.
- 40. Aliwi, S.M.; Bamford, C.H. J. Chem. Soc., Faraday Trans. 1 1977, 73, 776.
- 41. Bamford, C.H.; White, E.F.T. Trans. Faraday Soc. 1958, 54, 268.
- 42. Moad, G.; Ercole, F.; Johnson, C.H.; Krstina, J.; Moad, C.L.; Rizzardo, E.; Spurling, T.II.; Thang, S.II.; Anderson, A.G. ACS Symp. Ser. **1998**, 685, 332.
- 43. Solomon, D.H.; Rizzardo, E.; Cacioli, P. US 4581429, 1986 (Chem. Abstr. 1985, 102, 221335q).
- 44. Starnes, W.H. Prog. Polym. Sci. 2002, 27, 2133.
- 45. Endo, K. Prog. Polym. Sci. 2002, 27, 2021.
- 46. Ivan, B. Adv. Chem. Ser. 1996, 249, 19.
- 47. Vidotto, G.; Crosato-Arnaldi, A.; Talamini, G. Makromol. Chem. 1968, 114, 217.
- 48. Puts, R.D.; Sogah, D.Y. Macromolecules 1997, 30, 3323.
- 49. Ananchenko, G.; Matyjaszewski, K. Macromolecules 2002, 35, 8323.
- 50. Haddleton, D.M.; Duncalf, D.J.; Kukulj, D.; Heming, A.M.; Shooter, A.J.; Clark, A.J. J. Mater. Chem. 1998, 8, 1525.
- 51. Johnson, R.M.; Ng, C.; Samson, C.C.M.; Fraser, C.L. *Macromolecules* 2000, 33, 8618.
- 52. Ozerhwskii, B.V.; Reshchupkin, V.P. Dokl. Phys. Chem. (Engl. Transl.) 1981, 254, 731.
- Giese, B.; Ghosez, A.; Gobel, T.; Hartung, J.; Huter, O.; Koch, A.; Kroder, K.; Springer, R. In *Free Radicals in Chemistry and Biology*; Minisci, F., Ed.; Kluwer: Dordrecht, 1989; p 97.
- 54. Clark, A.J.; Jones, K. Tetrahedron Lett. 1989, 30, 5485.
- 55. Porter, N.A.; Rosenstein, I.J.; Breyer, R.A.; Bruhnke, J.D.; Wu, W.-X.; McPhail, A.T. J. Am. Chem. Soc. 1992, 114, 7664.
- 56. Porter, N.A.; Allen, T.; Breyer, R.A. J. Am. Chem. Soc. 1992, 114, 7676.
- 57. Nakano, T.; Tamada, D.; Miyazaki, J.; Kakiuchi, K.; Okamoto, Y. *Macromolecules* 2000, *33*, 1489.
- 58. Nakano, T.; Mori, M.; Okamoto, Y. Macromolecules 1993, 26, 867.

444

Controlling Polymerization

- 59. Nakano, T.; Okamoto, Y. *Macromolecules* **1999**, *32*, 2391.
- 60. Nakano, T.; Shikisai, Y.; Okamoto, Y. Polym. J. 1996, 28, 51.
- Puzin, Y.I.; Prokudina, E.M.; Yumagulova, R.K.; Muslukhov, R.R.; Kolesov, S.V. Dokl. Phys. Chem. (Engl. Transl.) 2002, 386, 211.
- 62. Nakano, T.; Okamoto, Y.; Sogah, D.Y.; Zheng, S. Macromolecules 1995, 28, 8705.
- 63. Nakano, T.; Sogah, D.Y. J. Am. Chem. Soc. 1997, 117, 534.
- 64. Wulff, G.; Dhal, P.K. Angew. Chem. Int. Ed. Engl. 1989, 28, 196.
- 65. Tsuji, M.; Sakai, R.; Satoh, T.; Kaga, II.; Kakuchi, T. *Macromolecules* **2002**, *35*, 8255.
- 66. Kakuchi, T.; Tsuji, M.; Satoh, T. ACS Symp. Ser. 2003, 854, 206.
- 67. Okamoto, Y.; Habaue, S.; Isobe, Y.; Nakano, T. Macromol. Symp. 2002, 183, 83.
- Matsumoto, A. In *Handbook of Radical Polymerization*; Davis, T.P.; Matyjaszewski, K., Eds.; John Wiley & Sons: Hoboken, 2002; p 691.
- 69. Coote, M.L.; Davis, T.P. In *Handbook of Solvents*; Wypych, G., Ed.; William Andrew Publishing: Norwich, New York, 2001; p 777.
- Coote, M.L.; Davis, T.P.; Klumperman, B.; Monteiro, M.J. J. Macromol. Sci. Rev. Macromol. Chem. Phys. 1998, C38, 567.
- 71. Barton, J.; Borsig, E. Complexes in Free Radical Polymerization; Elsevier: Amsterdam, 1988.
- 72. Gromov, V.F.; Khomiskovskii, P.M. Russ. Chem. Rev. (Engl. Transl.) 1979, 48, 1040.
- 73. Kamachi, M. Adv. Polym. Sci. 1981, 38, 55.
- 74. Beuermann, S.; Buback, M. Prog. Polym. Sci. 2002, 27, 191.
- 75. Reichardt, C. Solvent Effects in Organic Chemistry; Verlag Chemie: Weinheim, 1978.
- 76. Brumby, S. Polym. Commun. 1989, 30, 13.
- 77. Paleos, C.M.; Malliaris, A. J. Macromol. Sci., Rev. Macromol. Chem. Phys. 1988, C28, 403.
- 78. Kamachi, M.; Liaw, D.J.; Nozakura, S.-I. Polym. J. 1979, 11, 921.
- 79. Kamachi, M.; Satoh, J.; Nozakura, S. J. Polym. Sci., Polym. Chem. Ed. 1978, 16, 1789.
- Hatada, K.; Terawaki, Y.; Kitayama, T.; Kamachi, M.; Tamaki, M. Polym. Bull. 1981, 4, 451.
- 81. Morrison, B.R.; Piton, M.C.; Winnik, M.A.; Gilbert, R.G.; Napper, D.II. Macromolecules 1993, 26, 4368.
- 82. Kamachi, M.; Liaw, D.J.; Nozakura, S. Polym. J. 1981, 13, 41.
- 83. Bamford, C.II.; Brumby, S. Makromol. Chem. 1967, 105, 122.
- 84. O'Driscoll, K.F.; Monteiro, M.J.; Klumpermann, B. J. Polym. Sci., Part A: Polym. Chem. 1997, 35, 515.
- 85. Zammit, M.D.; Davis, T.P.; Willet, G.D.; O'Driscoll, K.F. J. Polym. Sci. A: Polym. Chem. 1997, 35, 2311.
- 86. Olaj, O.F.; Schnoll-Bitai, I. Monatsh. Chem. 1999, 130, 731.
- 87. Coote, M.L.; Davis, T.P. Eur. Polym. J. 2000, 36, 2423.
- 88. Burnett, G.M.; Cameron, G.G.; Joiner, S.N. Trans. Faraday Soc. 1973, 69, 322.
- 89. Tsutsumi, K.; Tsukahara, Y.; Okamota, Y. Polym. J. 1994, 26, 13.
- 90. Nakano, T.; Okamoto, Y. ACS Symp. Ser. 1997, 685, 451.
- 91. Nagara, Y.; Yamada, K.; Nakano, T.; Okamoto, Y. Polymer J. 2001, 33, 534.
- 92. Yamada, K.; Nakano, T.; Okamoto, Y. Macromolecules 1998, 31, 7598.

- Isobe, Y.; Yamada, K.; Nakano, T.; Okamoto, Y. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 4693.
- 94. Isobe, Y.; Yamada, K.; Nakano, T.; Okamoto, Y. Macromolecules 1999, 32, 5979.
- Krakovyak, M.G.; Anufrieva, E.V.; Sycheva, E.A.; Sheveleva, T.V. Macromolecules 1993, 26, 7375.
- 96. Ishigaki, Y.; Takahashi, K.; Fukuda, H. Macromol. Rapid Commun. 2000, 21, 1024.
- 97. Plochocka, K. J. Macromol. Sci., Rev. Macromol. Chem. 1981, C20, 67.
- 98. Madruga, E.L. Prog. Polym. Sci. 2002, 27, 1879.
- 99. Georgiev, G.S.; Dakova, I.G. Macromol. Chem. Phys. 1994, 195, 1695.
- 100. Lewis, F.M.; Walling, C.; Cummings, W.; Briggs, E.R.; Mayo, F.R. J. Am. Chem. Soc. 1948, 1519.
- 101. Price, C.C.; Walsh, J.G. J. Polym. Sci. 1951, 6, 239.
- 102. Pichot, C.; Zaganiaris, E.; Guyot, A. J. Polym. Sci., Polym. Symp. 1975, 52, 55.
- 103. Hill, D.J.T.; Lang, A.P.; Munro, P.D.; O'Donnell, J.H. Eur. Polym. J. 1992, 28, 391.
- 104. Hill, D.J.T.; Lang, A.P.; Munro, P.D.; O'Donnell, J.H. Eur. Polym. J. 1989, 28, 391.
- 105. Asakura, J.; Yoshihara, M.; Matsubara, Y.; Maeshima, T. J. Macromol. Sci., Chem. 1981, A15, 1473.
- 106. Van Der Meer, R.; Aarts, M.W.A.M.; German, A.L. J. Polym. Sci., Polym. Chem. Ed. 1980, 18, 1347.
- 107. Cameron, G.G.; Esslemont, G.F. Polymer 1972, 13, 435.
- 108. Ito, T.; Otsu, T. J. Macromol. Sci. Chem 1969, A3, 197.
- 109. San Roman, J.; Madruga, E.L. Angew. Makromol. Chem. 1980, 86, 1.
- 110. Bonta, G.; Gallo, B.M.; Russo, S. Polymer 1975, 16, 429.
- 111. Busfield, W.K.; Low, R.B. Eur. Polym. J. 1975, 11, 309.
- 112. Harwood, H.J. Makromol. Chem., Macromol. Symp. 1987, 10/11, 331.
- 113. Park, K.Y.; Santee, E.R.; Harwood, H.J. Eur. Polym. J. 1989, 25, 651.
- 114. Christov, L.K.; Georgiev, G.S. Macromol. Theory Simul. 2000, 9, 715.
- 115. Klumperman, B.; Kraeger, I.R. Macromolecules 1994, 27, 1529.
- 116. Maxwell, I.A.; Aerdts, A.M.; German, A.L. Macromolecules 1993, 26, 1956.
- 117. Klumperman, B.; O'Driscoll, K.F. Polymer 1993, 34, 1032.
- 118. Klumperman, B.; Vonk, G. Eur. Polym. J. 1994, 30, 955.
- 119. Davis, T.P. Polym. Commun. 1990, 31, 442.
- 120. Coote, M.L.; Johnston, L.P.M.; Davis, T.P. Macromolecules 1997, 30, 8191.
- 121. Kaim, A. Macromol. Theory Simul. 1997, 6, 907.
- 122. Fernandez-Monreal, C.; Sanchez-Chaves, M.; Martinez, G.; Madruga, E.L. Acta Polym. 1999, 50, 408.
- 123. Fernandez-Garcia, M.; Fernandez-Sanz, M.; Madruga, E.L.; Cuervo-Rodriguez, R.; Hernandez-Gordo, V.; Fernandez-Monreal, M.C. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 60.
- 124. Chambard, G.; Klumperman, B.; German, A.L. Polymer 1999, 40, 4459.
- 125. Kaim, A. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 846.
- 126. Madruga, E.L.; Fernandez-Garcia, M. Macromol. Chem. Phys. 1996, 197, 3743.
- 127. de la Fuente, J.L.; Madruga, E.L. Macromol. Chem. Phys. 1999, 200, 1639.
- 128. Fernandez-Monreal, C.; Martinez, G.; Sanchez-Chaves, M.; Madruga, E.L. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 2043.
- 129. Coote, M.L.; Davis, T.P. Prog. Polym. Sci. 1999, 24, 1217.
- Coote, M.L.; Davis, T.P. In Handbook of Radical Polymerization; Davis, T.P.; Matyjaszewski, K., Eds.; John Wiley & Sons: Hoboken, 2002; p 263.
- 131. Krstina, J.; Moad, G.; Solomon, D.H. Eur. Polym. J. 1992, 28, 275.

Controlling Polymerization

- 132. Semchikov, Y.D. Polym. Sci. USSR (Engl. Transl.) 1990, 32, 177.
- 133. Semchikov, Y.D. Macromol. Symp. 1996, 111, 317.
- 134. Kuchanov, S.I.; Russo, S. Macromolecules 1997, 30, 4511.
- 135. Tsukahara, Y.; Hayashi, N.; Jiang, X.L.; Yamashita, Y. Polym. J. 1989, 21, 377.
- 136. Cooper, A.I. J. Mater. Chem. 2000, 10, 207.
- 137. Kendall, J.L.; Canelas, D.A.; Young, J.L.; DeSimone, J.M. Chem. Rev. 1999, 99, 543.
- van Herk, A.M.; Manders, B.G.; Canelas, D.A.; Quadir, M.A.; DeSimone, J.M. Macromolecules 1997, 30, 4780.
- Quadir, M.A.; DeSimone, J.M.; van Herk, A.M.; German, A.L. Macromolecules 1998, 31, 6481.
- 140. Beuermann, S.; Buback, M.; Schmaltz, C.; Kuchta, F.D. Macromol. Chem. Phys. 1998, 199, 1209.
- 141. Beuermann, S.; Buback, M.; Schmaltz, C. Macromolecules 1998, 31, 8069.
- 142. Beuermann, S.; Buback, M.; Isemer, C.; Lacik, I.; Wahl, A. *Macromolecules* 2002, *35*, 3866.
- 143. Beuermann, S.; Buback, M.; Nelke, D. Macromolecules 2001, 34, 6637.
- 144. Welton, T. Chem. Rev. 1999, 99, 2071.
- 145. Kubisa, P. Prog. Polym. Sci. 2004, 29, 3.
- 146. Harrisson, S.; Mackenzie, S.R.; Haddleton, D.M. Chem. Commun. 2002, 2850.
- 147. Harrisson, S.; Mackenzie, S.R.; Haddleton, D.M. Macromolecules 2003, 36, 5072.
- 148. Biedron, T.; Kubisa, P. Macromol. Rapid Commun. 2001, 22, 1237.
- 149. Biedron, T.; Kubisa, P. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 2799.
- 150. Biedron, T.; Kubisa, P. Polym. Int. 2003, 52, 1584.
- 151. Sarbu, T.; Matyjaszewski, K. Macromol. Chem. Phys. 2001, 202, 3379.
- 152. Ma, H.Y.; Wan, X.H.; Chen, X.F.; Zhou, Q.F. J. Polym. Sci., Part A: Polym. Chem. 2003, 41, 143.
- 153. Zhao, Y.L.; Zhang, J.M.; Jiang, J.; Chen, C.F.; Xi, F. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 3360.
- 154. Sarbu, T.; Pintauer, T.; McKenzie, B.; Matyjaszewski, K. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 3153.
- Carmichael, A.J.; Haddleton, D.M.; Bon, S.A.F.; Seddon, K.R. Chem. Commun. 2000, 1237.
- 156. Ma, H.; Wan, X.; Chen, X.; Zhou, Q.-F. Polymer 2003, 44, 5311.
- 157. Perrier, S.; Davis, T.P.; Carmichael, A.J.; Haddleton, D.M. Eur. Polym. J. 2003, 39, 417.
- 158. Perrier, S.; Davis, T.P.; Carmichael, A.J.; Haddleton, D.M. Chem. Commun. 2002, 2226.
- 159. Ryan, J.; Aldabbagh, F.; Zetterlund, P.B.; Yamada, B. 2004, 25, 930.
- 160. Zhang, H.W.; Hong, K.; Mays, J.W. Polym. Bull. 2004, 52, 9.
- 161. Zhang, H.W.; Hong, K.L.; Jablonsky, M.; Mays, J.W. Chem. Commun. 2003, 1356.
- Bamford, C.H. In *Alternating Copolymers*; Cowie, J.M.G., Ed.; Plenum: New York, 1985; p 75.
- 163. Bamford, C.II.; Jenkins, A.D.; Johnston, R. Proc. R. Soc., London 1957, A241, 364.
- Zubov, V.P.; Valuev, L.I.; Kabanov, V.A.; Kargin, V.A. J. Polym. Sci., Part A-1 1971, 9, 833.
- 165. Tanaka, T.; Kato, H.; Sakai, I.; Sato, T.; Ota, T. Makromol. Chem., Rapid Commun. 1987, 8, 223.
- 166. Otsu, T.; Yamada, B. J. Macromol. Sci. Chem 1966, A1, 61.

- Isobe, Y.; Nakano, T.; Okamoto, Y. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 1463.
- 168. Habaue, S.; Yamada, H.; Uno, T.; Okamoto, Y. J. Polym. Sci., Part A: Polym. Chem. 1997, 35, 721.
- 169. Baraki, H.; Habauc, S.; Okamoto, Y. Macromolecules 2001, 34, 4724.
- 170. Okamoto, Y.; Habaue, S.; Isobe, Y. ACS Symp. Ser. 2003, 854, 59.
- 171. Habaue, S.; Isobe, Y.; Okamoto, Y. Tetrahedron 2002, 58, 8205.
- 172. Isobe, Y.; Fujioka, D.; Habaue, S.; Okamoto, Y. J. Am. Chem. Soc. 2001, 123, 7180.
- 173. Ray, B.; Isobe, Y.; Matsumoto, K.; Habaue, S.; Okamoto, Y.; Kamigaito, M.; Sawamoto, M. *Macromolecules* 2004, 37, 1702.
- 174. Suito, Y.; Isobe, Y.; Habaue, S.; Okamoto, Y. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 2496.
- 175. Isobe, Y.; Suito, Y.; Habaue, S.; Okamoto, Y. J. Polym. Sci., Part A: Polym. Chem. 2003, 41, 1027.
- 176. Hirooka, M.; Yabuuchi, H.; Morita, S.; Kawasumi, S.; Nakaguchi, K. J. Polym. Sci., Polym. Lett. 1967, 5, 47.
- 177. Hirooka, M.; Yabuuchi, H.; Morita, S.; Kawasumi, S.; Nakaguchi, K. J. Polym. Sci., Part A-1: Polym. Chem. 1968, 6, 1381.
- 178. Momtaz-Afchar, J.; Polton, A.; Tardi, M.; Sigwalt, P. Eur. Polym. J. 1985, 21, 1067.
- 179. Rogueda, C.; Polton, A.; Tardi, M.; Sigwalt, P. Eur. Polym. J. 1989, 25, 1259.
- 180. Rogueda, C.; Polton, A.; Tardi, M.; Sigwalt, P. Eur. Polym. J. 1989, 25, 1251.
- 181. Rogueda, C.; Tardi, M.; Polton, A.; Sigwalt, P. Eur. Polym. J. 1989, 25, 885.
- 182. Golubev, V.B.; Zubov, V.P.; Georgiev, G.S.; Stoyachenko, I.L.; Kabanov, V.A. J. Polym. Sci., Polym. Chem. Ed. 1973, 11, 2463.
- 183. Seno, M.; Matsumura, N.; Nakamura, H.; Sato, T. J. Appl. Polym. Sci. 1997, 63, 1361.
- 184. Sherrington, D.C.; Slark, A.T.; Taskinen, K.A. Macromol. Chem. Phys. 2002, 203, 1427.
- 185. Wang, H.; Chu, G.; Srisiri, W.; Padias, A.B.; Hall, H.K. Acta Polym. 1994, 45, 26.
- 186. Lyons, R.A.; Moad, G.; Senogles, E. Eur. Polym. J. 1993, 29, 389.
- 187. Krstina, J.; Moad, G.; Solomon, D.H. Polym. Bull. 1992, 27, 425.
- 188. Fellows, C.M.; Senogles, E. Eur. Polym. J. 2001, 37, 1091.
- 189. Fellows, C.M.; Senogles, E. Eur. Polym. J. 1999, 35, 9.
- 190. Kirci, B.; Lutz, J.F.; Matyjaszewski, K. Macromolecules 2002, 35, 2448.
- 191. Lutz, J.F.; Kirci, B.; Matyjaszewski, K. Macromolecules 2003, 36, 3136.
- 192. Bamford, C.H. Chem. Aust. 1982, 49, 341.
- 193. Tan, Y.Y. In Recent Advances in Mechanistic and Synthetic Aspects of Polymerization; Dordrecht: Reidel, 1987; p 281.
- 194. Tan, Y.Y. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 245.
- 195. Polowinski, S. Prog. Polym. Sci. 2002, 27, 537.
- 196. Buter, R.; Tan, Y.Y.; Challa, G. J. Polym. Sci., Part A-1 1972, 10, 1031.
- 197. Schomaker, E.; Challa, G. Macromolecules 1988, 21, 3506.
- 198. Nodono, M.; Makino, T.; Nishida, K. React. Funct. Polym. 2003, 57, 157.
- 199. Matsuzaki, K.; Kanai, T.; Ichijo, C.; Yuzawa, M. Makromol. Chem. 1984, 185, 2291.
- 200. Serizawa, T.; Hamada, K.; Akashi, M. Nature 2004, 429, 52.

Controlling Polymerization

- 201. van de Grampel, H.T.; Tan, Y.Y.; Challa, G. Macromolecules 1991, 24, 3767.
- 202. van de Grampel, H.T.; Tan, Y.Y.; Challa, G. Macromolecules 1991, 24, 3773.
- 203. Ferguson, J.; Al-Alawi, S.; Grannmayeth, R. Eur. Polym. J. 1985, 19, 475.
- 204. Chapiro, A. Pure Appl. Chem. 1981, 53, 643.
- 205. Polowinski, S. Eur. Polym. J. 1983, 19, 679.
- 206. Natansohn, A. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1984, 25(2), 65.
- 207. Kammerer, H. Angew. Chem., Int. Ed. Engl. 1965, 4, 952.
- 208. Kern, W.; Kammerer, H. Pure Appl. Chem. 1967, 15, 421.
- 209. Kammerer, H.; Onder, N. Makromol. Chem. 1968, 111, 67.
- 210. Feldman, K.S.; Bobo, J.S.; Ensel, S.M.; Lee, Y.B.; Weinreb, P.H. J. Org. Chem. 1990, 55, 474.
- 211. Feldman, K.S.; Lee, Y.B. J. Am. Chem. Soc. 1987, 109, 5850.
- 212. Wulff, G.; Kemmerer, R.; Vogt, B. J. Am. Chem. Soc. 1987, 109, 7449.
- 213. Jantas, R. J. Polym. Sci., Part A: Polym. Chem. 1990, 28, 1973.
- 214. Jantas, R.; Polowinski, S. J. Polym. Sci., Part A: Polym. Chem. 1986, 24, 1819.
- 215. Jantas, R. J. Polym. Sci., Part A: Polym. Chem. 1994, 32, 295.
- Bamford, C.H. In *Developments in Polymerization 2*; Howard, R.N., Ed.; Applied Science: London, 1979; Vol. 49, p 215.
- 217. Saito, R.; Kobayashi, H. Macromolecules 2002, 35, 7207.
- 218. Saito, R.; Okuno, Y.; Kobayashi, H. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 3539.
- 219. Saito, R.; Yamaguchi, K. Macromolecules 2003, 36, 9005.
- 220. Sugiyama, J.-I.; Yokozawa, T.; Endo, T. Macromolecules 1994, 27, 1987.
- 221. Sugiyama, J.-I.; Yokozawa, T.; Endo, T. J. Am. Chem. Soc. 1993, 115, 2041.
- 222. Singh, A.; Kaplan, D.L. J. Polym. Envir. 2002, 10, 85.
- 223. Gross, R.A.; Kumar, A.; Kalra, B. Chem. Rev. 2001, 101, 2097.
- 224. Matsumoto, A. ACS Symp. Ser. 2000, 768, 93.
- 225. Nagai, K. Trends Polym. Sci. 1996, 4, 122.
- 226. Tajima, K.; Aida, T. Chem. Commun. 2000, 2399.
- 227. Nagahama, S.; Matsumoto, A. J. Polym. Sci., Part A: Polym. Chem. 2004, 42, 3922.
- 228. Nagahama, S.; Tanaka, T.; Matsumoto, A. Angew. Chem. Int. Ed. Engl. 2004, 43, 3811.
- 229. Cochin, D.; Zana, R.; Candau, F. Macromolecules 1993, 26, 5765.
- 230. Dais, P.; Palcos, C.M.; Nika, G.; Malliaris, A. Macromol. Chem. Phys. 1993, 194, 445.
- 231. Percec, V.; Ahn, C.II.; Ungar, G.; Yeardley, D.J.P.; Moller, M.; Sheiko, S.S. *Nature* 1998, 391, 161.
- Percec, V.; Ahn, C.H.; Cho, W.D.; Jamieson, A.M.; Kim, J.; Leman, T.; Schmidt, M.; Gerle, M.; Moller, M.; Prokhorova, S.A.; Sheiko, S.S.; Cheng, S.Z.D.; Zhang, A.; Ungar, G.; Yeardley, D.J.P. *J. Am. Chem. Soc.* **1998**, *120*, 8619.
- 233. Percec, V.; Mitchell, C.M.; Cho, W.D.; Uchida, S.; Glodde, M.; Ungar, G.; Zeng, X.B.; Liu, Y.S.; Balagurusamy, V.S.K.; Heiney, P.A. J. Am. Chem. Soc 2004, 126, 6078.
- 234. Percee, V.; Bera, T.K.; Glodde, M.; Fu, Q.Y.; Balagurusamy, V.S.K.; Heiney, P.A. *Chem. Eur. J.* **2003**, *9*, 921.

This Page is Intentionally Left Blank

9 Living Radical Polymerization

9.1 Introduction

The first demonstration of living polymerization and the current definition of the process can be attributed to Sware.^{1,2} Living polymerization mechanisms offer polymers of controlled composition, architecture and molecular weight distribution. They provide routes to narrow dispersity end-functional polymers, to high purity block copolymers, and to stars and other more complex architectures. Traditional methods of living polymerization are based on ionic, coordination or group transfer mechanisms. Ideally, the mechanism of living polymerization involves only initiation and propagation steps. All chains are initiated at the commencement of polymerization and propagation continues until all monomer is consumed. The combination of a living mechanism with the scope and versatility of the radical process should allow a wider selection of monomers and monomer combinations and more freedom in choosing reaction conditions. This potential and the applications that follow have provided the impetus for the very significant research efforts that have been devoted to this area over the last decade. In this chapter, we discuss the various approaches that have been developed in moving towards a living radical polymerization paying particular attention to the mechanism and the scope of each method.

At the time of the first edition of this book (1995),³ this field was still very much in its infancy. NMP was described, though little had been published in the open literature, and methods such as ATRP and RAFT had not been reported. Since 1995, the area has expanded dramatically and by themselves living/ controlled/mediated processes now account for a very substantial fraction of all research on radical polymerization (Chapter 1). The development of this field over this period can be followed in the publications following successful ACS symposia held in 1997,⁴ 2000⁵ and 2002⁶ and SML meetings held in 1996⁷ and 2001.⁸ Publications continue to appear at a rapid rate. Matyjaszewski⁹ has provided an overview of the history and development of living radical polymerization through 2001 in the *Handbook of Radical Polymerization*.¹⁰

9.1.1 Living? Controlled? Mediated?

The terminology used in this chapter deserves some mention. Currently there is controversy over the use of the terms "living" and "controlled" in the context of

describing a radical polymerization.¹¹⁻¹⁵ The current IUPAC recommendation, that a living polymerization is "a chain polymerization from which irreversible chain transfer and irreversible chain termination (deactivation) are absent", would preclude use of the term "living" in the context of a radical process.¹⁶ The use of the adjective "controlled" by itself to designate these polymerizations is also contrary to IUPAC recommendations.¹⁶ The adjective "controlled" should only be used when the particular aspect of polymerization that is being controlled is specified. It is not recommended that "controlled" be used in an exclusive sense to mean a particular form of polymerization since the word has an established, much wider, usage. The construct "controlled living polymerization" would seem acceptable when used to refer to those living polymerizations whose outcomes are defined by controlling the reaction conditions or other features. The word "controlled" should not be used to indicate that systems have a lower degree of livingness. Other terms such as "pseudo-living" and "quasi-living" are also discouraged.¹⁶ It has been stated that the definition of living polymerization "tolerates no restrictive adjectives implying something close to but not strictly living".¹¹

For this book, we have decided to entitle this chapter "Living Radical Polymerization" and use the term throughout. It is a chapter describing various approaches to living radical polymerization. We do not intend to imply that termination is absent from all or, indeed, any of the polymerizations described, only that the polymerizations display at least some of the observable characteristics normally associated with living polymerization.

9.1.2 Tests for Living (Radical) Polymerization

Following on from the above, various methods have been described to test and/or rank the "livingness" of polymerization processes.^{11,12,17-20} All of these tests have limitations. The following list paraphrases a set of criteria for living polymerization set out by Quirk and Lee¹¹ who also critically assessed their applicability primarily in the context of living anionic polymerization.

- (a) "Living polymerizations proceed until all monomer is consumed and may continue growth if further monomer is added." This criterion paraphrases one of Szwarc's definitions of living polymerization.^{1,2} It becomes a rigorous criterion if we add "and the number of living chains remains constant".
- (b) "In a living polymerization the molecular weight increases linearly with conversion." This contrasts with observations for conventional radical polymerizations where molecular weights are initially high and decrease with conversion due to monomer depletion (Figure 9.1). However, molecular weights obtained in radical polymerizations with conventional transfer agents with C_{tt}>1 will increase with conversion and may meet this test. Expressions for the dependence of molecular weight on conversion for NMP (and similar polymerizations), ATRP and RAFT appear in Sections 9.3.1.2, 9.4.1 and 9.5.1

respectively. A plot of \overline{M}_n vs conversion will remain linear even in circumstances where there is a loss of a substantial fraction of the living chains, although in that case there will be a broadening of the molecular weight distribution.



Figure 9.1 Predicted evolution of molecular weight (arbitrary units) with monomer conversion for a conventional radical polymerization with a constant rate of initiation (--) and a living polymerization (--).

- (c) "In a living polymerization the concentration of active species remains constant." A plot of $\ln([M]_o/[M]_t) vs$ time should be linear. In many conventional radical polymerizations a steady state is established such that, over a wide conversion range, the concentration of active chains remains approximately constant. Thus, these polymerizations will meet this test. Conversely, some living polymerizations with reversible deactivation will not meet this test (Section 9.3.1.3). A rigorous criterion that also covers these cases is that the total concentration of active and dormant chains should remain constant. However, this is more difficult to establish from kinetic measurements alone.
- (d) "Living polymerizations provide narrow molecular weight distributions." This is a more qualitative test. What constitutes low dispersity? Theoretically, a dispersity (X̄_w / X̄_n) of 1.5 is the narrowest achievable in a conventional radical polymerization with termination by combination for long chains (Section 5.2.1.3). An ideal living polymerization can provide a Poisson molecular weight distribution and X̄_w / X̄_n=1+1/X̄_n; X̄_w / X̄_n=1.01 for X̄_n=100 (Figure 9.2). The better living radical systems produce X̄_w / X̄_n in the range 1.05-1.2. Errors associated with measuring the dispersity can be significant and most cause an underestimate of the actual value. A low dispersity alone does not imply the absence of side reactions.



Figure 9.2 Calculated (a) number and (b) GPC distributions for three polymers each with $\overline{X}_n = 100$. The number distributions of chains formed by conventional radical polymerization with termination by disproportionation or chain transfer $(--, \sum_{n_i=1,0} \overline{X}_w / \overline{X}_n = 2.0)$ or termination by combination (-----, $\sum_{n_i=1,0} n_i = 1.0$, $\overline{X}_w / \overline{X}_n = 1.5$) were calculated as discussed in Section 5.2.1.3. The number distribution of chains formed in an ideal living polymerization (----, $\sum_{n_i=1,0} n_i = 1.0$, $\overline{X}_w / \overline{X}_n = 1.01$) was calculated using a Poisson distribution function.

- (e) "Block copolymers can be prepared by sequential addition of monomers." This is a special case of (a) above.
- (f) "End groups are retained allowing end-functional polymers to be obtained in quantitative yield." Assessment of the fraction of living chains can provide a quantitative measure of the quality of a living polymerization. Currently, the most used methods for end group determination are NMR and mass spectrometry. Some discussion on these techniques is provided in Sections 3.5.3.2 and 3.5.3.4 respectively.

Quirk and Lee concluded "there is no single criterion which is satisfactory for determination of whether a given polymerization is living or not."¹¹ Most of the radical polymerizations discussed in this chapter meet one or more of these criteria.

9.2 Agents Providing Reversible Deactivation

The kinetics and mechanism of living radical polymerization have been reviewed by Fischer,²¹ Fukuda *et al.*,²² and Goto and Fukuda.²³ In conventional radical polymerization, new chains are continually formed through initiation while existing chains are destroyed by radical-radical termination. The steady state concentration of propagating radicals is $\sim 10^{-7}$ M and an individual chain will have a lifetime of only 1-10 s before termination within a total reaction time that is

typically greater than 10000 s. A consequence is that long chains are formed early in the process and (in the absence of other influences) molecular weights decrease with monomer conversion due to the depletion of monomer (Figure 9.1). In conventional (classical anionic^{1,2}) living polymerization all chains are initiated at the beginning of the reaction and grow until all monomer is consumed. As a consequence, molecular weight increases linearly with conversion and the molecular weight distribution is narrow.

The propensity of radicals to undergo self-reaction thus precludes the use of the simple strategy applied in anionic polymerization in developing a living radical polymerization. Radical polymerizations can display the characteristics normally associated with living polymerization in the presence of species that reversibly deactivate or terminate chains. These reagents control the concentration of active propagating species by maintaining a majority of chains in a dormant form. In homogeneous radical polymerization the rate of radical-radical termination is proportional to the square of the radical concentration ($R_1 \propto [P_n \bullet]^2$). Thus, the incidence of termination can be reduced relative to propagation ($R_p \propto [P_n \bullet]$) by reducing the radical concentration.

In living radical polymerization, the concentration of propagating radicals is usually similar to or lower than that in conventional radical polymerization (*i.e.* $<10^{-7}$ M). For control, and to retain a high fraction of living chains, the lifetime of chains in their active state must be significantly less than in the conventional process (<<1-10 s). A rapid equilibration between active and dormant forms then ensures that all propagating species have equal opportunity for chain growth. All chains grow intermittently.

It is not necessary that living radical polymerizations be slow. However, it follows from the above discussion that, for a high fraction of living chains, either the final degree of polymerization must be significantly lower than that in an otherwise similar conventional process or that conditions must be chosen such that the rate of polymerization is substantially lower.

Heterogeneous polymerization processes (emulsion, miniemulsion, nonaqueous dispersion) offer another possibility for reducing the rate of termination through what are known as compartmentalization effects. In emulsion polymerization, it is believed that the mechanism for chain stoppage within the particles is not radical-radical termination but transfer to monomer (Section 5.2.1.5). These possibilities have provided impetus for the development of living heterogeneous polymerization (Sections 9.3.6.6, 9.4.3.2, 9.5.3.6).

We can distinguish several sub-classes of activation-deactivation processes according to their mechanism. These are shown in Scheme 9.1-Scheme 9.3.

(a) Those giving deactivation by reversible coupling and involving a unimolecular activation process as shown in Scheme 9.1. P_n^{\bullet} is a propagating radical (an active chain). The deactivator (X) is usually, though not always, a stable radical. However, X may also be an even electron (diamagnetic) species, for example, diphenylethylene (Section 9.3.5). In this case P_n -X would be a

persistent radical, or a transition metal complex, for example, a low spin cobalt (II) complex (Section 9.3.9). These systems are discussed in Section 9.3. Possibly the best known process is nitroxide-mediated polymerization (NMP) (Section 9.3.6).



(b) Those giving deactivation by reversible atom or group transfer and involving a bimolecular activation process (Scheme 9.2). For the systems described, the deactivator (X-Y) is a transition metal complex where Y is the metal in a higher oxidation state. Y• is then the metal in a lower oxidation state. Y• is inert with respect to monomer. Y• can be considered as a catalyst for the process shown in Scheme 9.1 and many aspects of the kinetics are similar. The best known example is atom transfer radical polymerization (ATRP - Section 9.4) where the deactivator X-Y is, for example, a copper(II) halide.



(c) Those giving simultaneous deactivation and activation by reversible (degenerate) chain transfer (Scheme 9.3). These systems are discussed in Section 9.5. The best known of this class is RAFT (Reversible Addition-Fragmentation chain Transfer) with thiocarbonylthio compounds (Section 9.5.3). In this case, the chain transfer step involves formation of an intermediate adduct. Other examples thought to involve a transfer by homolytic substitution are iodine transfer polymerization (Section 9.5.4) and TERP (telluride-mediated polymerization, Section 9.5.5).



Scheme 9.3

456

The polymerizations (a) and (b) owe their success to what has become known as the persistent radical effect.²¹ Simply stated: when a transient radical and a persistent radical are simultaneously generated, the cross reaction between the transient and persistent radicals will be favored over self-reaction of the transient radical. Self-reaction of the transient radicals leads to a build up in the concentration of the persistent species which favors cross termination with the persistent radical over homotermination. The homotermination reaction is thus self-suppressing. The effect can be generalized to a persistent species effect to embrace ATRP and other mechanisms mentioned in Sections 9.3 and 9.4. Many aspects of the kinetics of the processes discussed under (a) and (b) are similar,²¹ the difference being that (b) involves a bimolecular activation process.

The reversible chain transfer process (c) is different in that ideally radicals are neither destroyed nor formed in the activation-deactivation equilibrium. This is simply a process for equilibrating living and dormant species. Radicals to maintain the process must be generated by an added initiator.

Though there is still debate about detailed mechanism, in each of the processes (a-c) the propagating species is believed to be a conventional propagating radical. Thus, termination by radical-radical reaction is not eliminated, though, as we shall see, with appropriate choice of reaction conditions, the significance of this process can be markedly reduced.

9.3 Deactivation by Reversible Coupling and Unimolecular Activation

Most polymerizations in this section can be categorized as stable (free) radicalmediated polymerizations (sometimes abbreviated as SFRMP). In the following discussion systems have been classed according to the type of stable radical involved, which usually correlates with the type of bond homolyzed in the activation process. Those described include systems where the stable radical is a sulfur-centered radical (Section 9.3.2), a selenium-centered radical (Section 9.3.3), a carbon-centered radical (Sections 9.3.4 and 9.3.5), an oxygen-centered radical (Sections 9.3.6, 9.3.7), or a nitrogen-centered radical (Section 9.3.8). We also consider polymerization mediated by cobalt complexes (Section 9.3.9) and certain 'monomers' (Section 9.3.5).

9.3.1 Kinetics and Mechanism

9.3.1.1 Initiators, iniferters, initers

In each of the sections below, we will consider the initiation process separately. For each system, various initiation methods have been applied. In some cases the initiator is a low molecular weight analog of the propagating species, in other cases it is a method of generating such a species. The initiators first used in this form of living radical polymerization were called iniferters (*initiator* - trans*fer* agent - chain *ter*minator) or initers (*initiator* - chain *ter*minator).

These terms were coined by Otsu and Yoshida²⁴ based on the similar terminology introduced by Kennedy²⁵ to cover analogous cationic systems. Except for the case of the dithiuram disulfides and related species (Section 9.3.2.1), these expressions have now fallen from favor and are no longer used as a generic terminology. In this chapter, we use the term initiator to denote alkoxyamines in NMP and halocompounds in ATRP despite the confusion this can create, especially when the process also involves added conventional initiators.

In order for the characteristics of living polymerization to be displayed, initiators should possess the following attributes:

- (a) One (in some cases, both) of the radicals formed on initiator decomposition is persistent or long-lived and unable (or slow) to initiate polymerization.
- (b) Primary radical termination (or transfer to initiator) should be the only significant mechanism for the interruption of chain growth. Primary radical termination should occur exclusively by combination. Transfer to initiator, when involved, should occur exclusively by group transfer to give a product analogous to that formed by termination by combination.
- (c) The bond to the end group (X) formed by these mechanisms must be thermally or photochemically labile under the reaction conditions such that reversible homolysis regenerates the propagating radical.
- (d) The initiator must be consumed rapidly with respect to the rate of polymerization.

9.3.1.2 Molecular weights and distributions

The initiator or iniferter determines the number of growing chains. Several methods of initiation are used. Only three will be considered here. The first involves direct use of a species 1-X (*e.g.* a dithiocarbamate ester - Section 9.3.2 or an alkoxyamine - Section 9.3.6) as shown in Scheme 9.4. Ideally, the degree of polymerization is given by eq. 1 and the molecular weight by eq. 2.

$$I = X \xrightarrow{k'_{act}} I^{\bullet} + X$$

$$k'_{deact} \xrightarrow{k_i} monomer$$

$$P_1 = X \xrightarrow{k_{act}} P_1^{\bullet} + X$$

$$monomer$$

Scheme 9.4

$$\overline{X}_{q} = \frac{\left([M]_{q} - [M]_{r}\right)}{[IX]_{q}} = \frac{[M]_{q}}{[IX]_{q}}c$$
(1)

Living Radical Polymerization

$$\overline{M}_{n} = \frac{\left([\mathbf{M}]_{o} - [\mathbf{M}]_{i}\right)}{[\mathbf{IX}]_{o}} m_{\mathbf{IM}} + m_{\mathbf{IX}}$$
(2)

where $([M]_o-[M]_t)$ is the amount of monomer consumed, m_M and m_{tx} are the molecular weights of the monomer and the initiator (IX) respectively, and *c* is the monomer conversion. For a slow decomposing initiator, the term in the denominator should be $([IX]_o-[IX]_t) = [IX](1-\exp(-k_{act}t); i.e.$ the amount of initiator consumed. An efficiency term f' that has the usual definition (eq 3) can be introduced which allows for side reactions during the decomposition of IX or in the formation of P₁•. The species I• often has different reactivity and specificity for reaction with monomer than the propagating species (P_n•). Side reactions involving I• cause the molecular weight to be higher than expected.

$$f' = \frac{[\text{chains initiated}]}{[\text{IX}]_{o}}$$
(3)

For a polymerization with initiation by the process shown in Scheme 9.4 with $k'_{act}=k_{act}$ and $k'_{deact}=k_{deact}$, the dispersity is given by eq. 4

$$\frac{X_{\rm w}}{\overline{X}_{\rm n}} = 1 + \frac{1}{\overline{X}_{\rm n}} + \left(\frac{2-c}{c}\right) \frac{k_{\rm p} [\rm IX]}{k_{\rm deact}}$$
(4)

where c is the monomer conversion. The dispersity depends on the molecular weight, the monomer conversion, and the ratio k_p/k_{deact} . This ratio governs the number of propagation steps per activation cycle and should be large for a narrow molecular weight distribution.

A second process involves use of a conventional initiator (I₂; *e.g.* AIBN, BPO) in the presence of X (*e.g.* a nitroxide) to generate a species IX *in situ* as shown in Scheme 9.5.



Scheme 9.5

The degree of polymerization will usually be determined by the concentration of X. Some X may be lost in side reactions during the formation of IX. In some

cases, I• must undergo at least one propagation step before combination with X is likely (*e.g.* in NMP with BPO as initiator). Any processes that irreversibly consume X will raise the molecular weight. Any process that provides additional chains will lower the molecular weight (*e.g.* thermal initiation in S polymerizations or an additional thermal initiator).

A third process involves use of the species (X-X) to generate the 'stable radical' in pairs and relies on the stable radical being able to react with monomer, albeit slowly, to generate P₁X (Scheme 9.6). Polymerizations with dithiuram and other disulfides (Section 9.3.2.1) and hexasubstituted ethanes (Section 9.3.4) belong to this class.



Scheme 9.6

Other variations and combinations of these processes are also possible and are described in the following sections.

9.3.1.3 Polymerization kinetics

General features of the polymerization kinetics for polymerizations with deactivation by reversible coupling have already been mentioned. Detailed treatments appear in reviews by Fischer,²¹ Fukuda *et al.*,²² and Goto and Fukuda²³ and will not be repeated here.

In conventional radical polymerization the rate of polymerization is described by eq. 5 (Section 5.2.1). As long as the rate of initiation remains constant, a plot of $\ln([M]_{t})$ vs time should provide a straight line.

$$\ln \frac{[\mathbf{M}]_{\rm c}}{[\mathbf{M}]_{\rm r}} = k_{\rm p} \left(\frac{R_{\rm i}}{k_{\rm t}}\right)^{1/2} t \tag{5}$$

For polymerizations where initiation is described by Scheme 9.4, the rate of polymerization is given by eq. $6.^{21}$

$$\ln \frac{[M]_{o}}{[M]_{t}} = \frac{3}{2} k_{p} \left(\frac{K[IX]_{o}}{3k_{t}} \right)^{1/3} t^{2/3}$$
(6)

where $K = k_{act}/k_{deact}$. The derivation of this equation requires that $[X]_0$ is zero and that there is no initiation source other than IX. Note that the relationship between

460

 $\ln([M]_o/[M]_t)$ and time is *not* anticipated to be linear. Under these circumstances, the rate of polymerization is controlled by the value of the activation-deactivation equilibrium constant *K*.

If there is an external source of free radicals (*e.g.* from thermal initiation in S polymerization or from an added conventional initiator) eq. 5 may again apply. The rate of polymerization becomes independent of the concentration of IX and, as long as the number of radicals generated remains small with respect to [IX]_o, a high fraction of living chains and low dispersities is still possible. The validity of these equations has been confirmed for NMP and with appropriate modification has also been shown to apply in the case of ATRP.²³

9.3.2 Sulfur-Centered Radical-Mediated Polymerization

The carbon sulfur bond of suitably constructed *N*,*N*-dialkyldithiocarbamates and related compounds undergoes reversible homolysis under irradiation with UV light of appropriate wavelength (Scheme 9.7) allowing monomer insertion into the C-S bond. The *N*.*N*-dialkyldithiocarbamyl radical is persistent and reacts with monomers only slowly. This form of polymerization has been comprehensively reviewed by Ameduri,²⁶ Sebenik²⁷ and Otsu and Matsumoto.²⁸ The process should be distinguished from RAFT which can involve similar thiocarbonylthio compounds but does not usually involve sulfur-centered radicals as intermediates (Section 9.5.3).



9.3.2.1 Disulfide initiators

The first detailed study of dithiuram disulfides as initiators in polymerizations of MMA and S was reported by Werrington and Tobolsky in 1955.²⁹ They observed that the transfer constant to the disulfide was relatively high and also found significant retardation. The potential of this and other disulfides as initiators of living radical polymerization was recognized by Otsu and Yoshida in 1982.²⁴ A wide range of disulfides has now been investigated in this context with varying degrees of success. These include diaryl disulfides *e.g.* diphenyl disulfide (1)],^{30,31} dibenzoyl disulfide (2),²⁴ dithiuram disulfides [*e.g.* tetraethyldithiuram disulfide (4)],^{24,32,33} and xanthogen disulfides [*e.g.* bis(isopropylxanthogen) disulfide (5)];³⁴ with the dithiuram disulfides being the most studied in this context.



The proposed mechanism of initiation with the dithiuram disulfide 4 is shown in Scheme 9.8. The dithiuram disulfide decomposes thermally or photochemically to give dithiocarbamyl radicals 6. These radicals 6 add monomer only slowly and relatively high reaction temperatures (typically $>80^{\circ}$ C) appear necessary even when the initiator is decomposed photochemically.

initiation



transfer to initiator

$$P_{n}^{\bullet} \xrightarrow[Et]{N} \xrightarrow{S} \stackrel{Et}{\searrow} \stackrel{N}{\longrightarrow} \stackrel{N$$

reversible primary radical termination



Scheme 9.8

Transfer to the dithiuram disulfide by transfer of the dithiocarbamyl group, probably by addition-fragmentation, is an important mechanism for the termination of polymer chains during the early stages of polymerization. The transfer constant of **3** is reported to be *ca* 0.5 in both S and MMA polymerizations.^{35,36} The end groups **8** formed by transfer to the dithiuram disulfide are indistinguishable from those **8** formed by primary radical termination with dithiocarbamyl radicals (**6**, refer Scheme 9.8). While the formation of the end groups **8** is reversible under the

photopolymerization conditions (Section 9.3.2.2), the primary dithiocarbamate end groups 7 formed by addition of 6 to monomer are relatively stable to photolysis.

Since the dithiocarbamyl end groups **8** are thermally stable but photochemically labile at usual polymerization temperatures, only photo-initiated polymerizations have the potential to show living characteristics. However, various disulfides, for example, **9** and **10**, have been used to prepare end-functional polymers³⁷ and block copolymers³⁸ by irreversible chain transfer in non-living thermally-initiated polymerization (Section 7.5.1).



Aliphatic disulfides are not thought to be effective as initiators in this context. However, Endo *et al.*³⁹ have described the use of the cyclic 1,2-disulfides 11 and 12 as initiators in a controlled radical polymerization. Polymerization of S at 120 °C gave a linear increase in molecular weight with conversion and the PS formed was used as a macroinitiator to form PS-*block*-PMMA. The precise mechanism of the process has not been elucidated.



The use of the disulfide (13), which can dissociate thermally to give a sulfur analog of TEMPO (Section 9.3.6.1), has also been explored for controlling S polymerization though poor results were obtained.⁴⁰

9.3.2.2 Monosulfide initiators

Certain *N*,*N*-dialkyl dithiocarbamates [*e.g.* benzyl *N*,*N*-diethyl dithiocarbamate (14)] and xanthates have been used as photoinitiators. Photodissociation of the C–S bond of these compounds yields a reactive alkyl radical (to initiate polymerization) and a less reactive sulfur-centered radical (to undergo primary radical termination) as shown in Scheme 9.9.^{30,41,42}

Since the experiment is no longer reliant on the dithiocarbamyl radical to both initiate and terminate chains (*cf.* Section 9.3.2.1), lower reaction temperatures may be used (where the dithiocarbamyl radical is slower or unable to add monomer) and better control over the polymerization process can be obtained. The transfer constants for the benzyl dithiocarbamates in polymerization of acrylic and styrenic

monomers are very low, thus primary radical termination is the predominant chain termination mechanism.

initiation



transfer to initiator



reversible primary radical termination



Scheme 9.9

The processes described in this section should be contrasted with RAFT polymerization (Section 9.5.3), which can involve the use of similar thiocarbonylthio compounds. *N*,*N*-dialkyl dithiocarbamates have very low transfer constants in polymerizations of S and (meth)acrylates and are not effective in RAFT polymerization of these monomers. However, *N*,*N*-dialkyl dithiocarbamates have been successfully used in RAFT polymerization of VAc. Certain *O*-alkyl xanthates have been successfully used to control RAFT polymerizations of VAc, acrylates and S. The failure of the earlier experiments using these reagents and monomers to provide narrow molecular weight distributions by a RAFT mechanism can be attributed to the use of non-ideal reaction conditions and reagent choice. A two part photo-initiator system comprising a mixture of a benzyl dithiocarbamate and a dithiuram disulfide has also been described and provides better control (narrower molecular weight distributions).⁴³

The use of mono-, di- and multifunctional initiators provides scope for designing polymer architectures. The use of 14, 18 and 19 in the production of block or star polymers has been demonstrated.^{41,44,45} Homopolymers of 20 or copolymers of 20 with S or MMA have been successfully used in photoinitiated

graft polymerization of S or MMA.⁴⁶⁻⁴⁸ The analogous xanthate has also been used in this context. Compounds 20^{49} and 22^{50} have also been used to make hyperbranched polymers. The monomer 20 was reported to have reactivity ratios similar to those of S. It is reported⁵¹ that the xanthate 23 does not copolymerize with MMA, it acts only as a photoiniferter in MMA polymerization and provides a polymer with a relatively narrow molecular weight distribution. In S polymerization 23 also acts as a comonomer.



9.3.2.3 Monomers, mechanism, side reactions

The outcome of the polymerization depends strongly on the particular monomer. Polymerizations of S, MMA, MA, VAc and some derivatives have been reported. Studies on model compounds indicate that the primary or secondary dithiocarbamate end groups are much less susceptible to photodissociation than benzyl or tertiary derivatives.

Dithiocarbamate 16 has been used to prepare low dispersity PMAA $(M_w/M_p\sim 1.2)$.⁵² Photopolymerization of S in the presence of dithiocarbamate 16 also displays some living characteristics (molecular weights that increase with conversion, ability to make block copolymer). However, 17 appears to behave as a conventional initiator in S polymerization.⁵³ The difference in behavior was attributed to the relatively poor leaving group ability of the 2-carboxyprop-2-yl radical. This hypothesis is supported by MO calculations. Dithiocarbamate 17 was used to control polymerizations of MMA,⁵⁴ HEMA⁵⁴ and NIPAM.⁵⁵

Chain ends formed with monosubstituted monomers, other than S, appear resistant to photolysis and polymerizations of MA and VAc do not show living characteristics. Most polymerizations involve methacrylate esters or S.

Various side reactions that are likely to lead to a slow loss of "living" ends have been described. With disulfide initiators, one (initiation by the dithiocarbamyl radical) is unavoidable since the experiment relies on the same radical species to both initiate polymerization and terminate chains.

Other side reactions that have been reported are cleavage of the carbonnitrogen bond to form 24 and an aminyl radical 25 or scission of the thiocarbonylsulfur bond to form a thiyl radical 26 and 27 (Scheme 9.10).^{33,63,64} Thiocarbonylsulfur bond cleavage may be a preferred pathway in the case of primary dithiocarbamates.



9.3.3 Selenium-Centered Radical-Mediated Polymerization

Kwon and coworkers have reported the use of diphenyl diselenide $28^{65.66}$ and a variety of benzylic selenides (*e.g.* 29,^{67,68} 30,⁶⁹ $31^{70,71}$ and 32^{72}) as photoiniferters for polymerization of S, MMA and some derivatives. Very narrow dispersities were not obtained (M_{π} / M_{π} typically 2-2.5). However, it was possible to prepare block copolymers.^{69,71,73} A related visible light photoinitiation system has recently been reported comprising 1-(phenylseleno)ethylbenzene and *t*-butyl(diphenyl)(phenylseleno)silane.^{74,75}

The polymerization mechanisms proposed are similar to those discussed for the sulfur compounds described in Sections 9.3.2.1 and 9.3.2.2 and the results obtained are also generically similar. The transfer constant of benzyl selenide (**29**) (C_{tr} is 1.04 in S polymerization at 60 °C) is substantially higher than that of sulfide photoiniferters (Section 9.3.2.2). The value suggests that the incidence of reversible chain transfer should be of significance and that development of a thermal process involving reversible chain transfer may be possible. The transfer constants of diphenyl diselenide **28** are also high (C_{tr} is 1.43 in MMA⁶⁶ and 28 in S polymerization⁷⁶ at 60 °C). Various methods have been explored for end group transformation and to remove the selenide end group from the final product. These

466

include reduction with tri-*n*-butylstannane and oxidative elimination via reaction with hydrogen peroxide.⁷⁶



9.3.4 Carbon-Centered Radical-Mediated Polymerization

Stable carbon-centered radicals, in particular, substituted diphenylmethyl and triphenylmethyl radicals, couple reversibly with propagating radicals (Scheme 9.11). With the carbon-centered radical-mediated polymerization systems described to date, the propagating radical should be tertiary (e.g. methacrylate ester) to give reasonable rates of activation.



Scheme 9.11

The first use of sterically hindered hexasubstituted ethanes [e.g. 33] as initiators of polymerization was reported by Bledzki *et al.*^{77,78} The use of related initiators based on silvlated pinacols [e.g. 34, 35] has been reported by Crivello *et al.*,⁷⁹⁻⁸² Santos *et al.*,⁸³ and Roussel and Boutevin.^{84,85} Other initiators of this class include 36^{86,87} and 37.⁸⁸ The rates of decomposition of hexasubstituted ethanes and the derived macroinitiators are known to vary according to the degree of steric

crowding about the C–C bond undergoing homolysis,⁸⁹ though few rate constants have been reported.



The proposed polymerization mechanism is shown in Scheme 9.12. Thermal decomposition of the hexasubstituted ethane derivative yields hindered tertiary radicals that can initiate polymerization or combine with propagating species (primary radical termination) to form an oligomeric macroinitiator. The addition of the diphenylalkyl radicals to monomer is slow (*e.g.* k_i for **34** is reported as $10^{-4} \text{ M}^{-1} \text{ s}^{-1}$ at 80 °C⁸⁴) and the polymerization is characterized by an inhibition period during which the initiator is consumed and an oligomeric macroinitiator is formed. The bond to the CH₂ formed by addition to monomer is comparatively thermally stable.

initiation



Otsu and Tazaki⁹⁰ have reported on the use of triphenylmethylazobenzene (**39**) as an initiator. In this case, phenyl radical initiates polymerization and the triphenylmethyl radical reacts mainly by primary radical termination to form a macroinitiator. The early report⁹¹ that triphenylmethyl radical does not initiate MMA polymerization may only indicate a very low rate of polymerization. The addition of triphenylmethyl radical to MMA has been demonstrated in radical trapping experiments.⁹²

468



Triphenylmethyl terminated polymers (41) are formed in polymerizations conducted in the presence of triphenylmethyl thiol (40).⁹³ Transfer constants for 40 are similar to other thiols (17.8 for S, 0.7 for MMA, compare Section 6.2.2.1). When the polymers (41) are heated in the presence of added monomer it is presumed that the S-CPh₃ bond is cleaved and triphenylmethyl-mediated polymerization according to Scheme 9.11 can then ensue to yield chain extended or block polymers (42).

It is of interest to speculate on the precise structure of the macroinitiator species in these polymerizations. The work of Engel *et al.*⁹⁴ suggests the likelihood of a quinonoid intermediate (*e.g.* **45**, Scheme 9.13), at least for the polymerizations involving triphenylmethyl radical (44).



9.3.4.1 Monomers, mechanism, side reactions

The hindered carbon-centered radicals are most suited as mediators in the polymerization of 1,1-disubstituted monomers (*a.g.* MMA,^{78,95} other methacrylates and MAA,⁹⁶ and AMS⁹⁷). Polymerizations of monosubstituted monomers are not thought to be living. Dead end polymerization is observed with S at polymerization temperatures <100°C.⁹⁸ Monosubstituted monomers may be used in the second stage of AB block copolymer synthesis (formation of the B block).⁹⁵ However the non-living nature of the polymerization limits the length of the B block that can be formed. Low dispersities are generally not achieved.

There will be a gradual loss of stable radical with these systems as the di- or triarylmethyl radicals produced from the macroinitiator can add monomer, albeit slowly.^{99,100} This side reaction provides a mechanism for mopping up the excess stable radical formed as a consequence of termination between propagating radicals and may be essential to maintaining polymerization rates.

A further problem with these iniferters is loss of "living" ends through primary radical termination by disproportionation. The ratio of k_{td}/k_{tc} reported for the cross

reaction between 43 and triphenylmethyl radicals (44) and at 110° C is 0.61 (Scheme 9.13).⁹⁴

9.3.5 Reversible Addition-Fragmentation

470

Certain monomers may be able to act as reversible deactivators by a reversible addition-fragmentation mechanism. The monomers are 1,1-disubstituted and generate radicals that are unable or extremely slow to propagate or undergo combination or disproportionation. For these polymerizations the dormant species is a radical and the persistent species is the 1,1-disubstituted monomer.

Thus propagating radicals were initially proposed to add reversibly to diphenylethylene as shown in Scheme 9.14.¹⁰¹





It was subsequently shown that the polymers contain semi-quinonoid structures 47 proposed to arise from α -*p* coupling of radicals 46 as shown in Scheme 9.15.¹⁰²⁻¹⁰⁴ It was also suggested that 47 could be subject to radical-induced decomposition by an addition-fragmentation process.



Polymerization in the presence of capto-dative substituted monomers has been proposed¹⁰⁵ to follow a related mechanism (Scheme 9.16) in which the concentration of the radical adduct **48** is additionally controlled by a reversible coupling reaction.



Scheme 9.16

To date, the degree of control realized with these methods is poor with respect to those achieved with NMP, ATRP or RAFT.

9.3.6 Nitroxide-Mediated Polymerization

The literature on Nitroxide-Mediated Polymerization (NMP) through 2001 was reviewed by Hawker *et al.*^{106,107} More recently the subject has been reviewed by Studer and Schulte¹⁰⁸ and Solomon.¹⁰⁹ NMP is also discussed by Fischer¹¹⁰ and Goto and Fukuda²³ in their reviews of the kinetics of living radical polymerization and is mentioned in most reviews on living radical polymerization. A simplified mechanism of NMP is shown in Scheme 9.17.



Scheme 9.17

Prior to the development of NMP, nitroxides were well known as inhibitors of polymerization (Section 5.3.1). They and various derivatives were (and still are) widely used in polymer stabilization. Both applications are based on the property of nitroxides to efficiently scavenge carbon-centered radicals by combining with them at near diffusion-controlled rates to form alkoxyamines. This property also saw nitroxides exploited as trapping agents to define initiation mechanisms (Section 3.5.2.4).

The exploitation of alkoxyamines as polymerization initiators and the use of NMP for producing block and end-functional polymers was first described in a patent application by Solomon *et al.* in 1985.¹¹¹ In this work NMP was described as a method of living radical polymerization. This work was mentioned in a communication¹¹² in 1987 and a conference paper¹¹³ in 1991. In 1990, Johnson *et al.*¹¹⁴ described what is now known as the persistent radical effect¹¹⁵ and showed that NMP, with appropriate selection of alkoxyamine and control of reaction conditions could, in principle, provide narrow dispersity polymers. These early papers focused on NMP of acrylates. However, the method only received significant attention in the wider literature following the demonstration by Georges *et al.*¹¹⁶ in 1993 that NMP could be used to prepare PS with a narrow molecular weight distribution. Since that time the literature on NMP has greatly expanded and, along with ATRP and RAFT, NMP is now one of the most cited methods for living radical polymerization.

9.3.6.1 Nitroxides

A wide range of nitroxides and derived alkoxyamines has now been explored for application in NMP. Experimental work and theoretical studies have been carried out to establish structure-property correlations and provide further understanding of the kinetics and mechanism. Important parameters are the value of the activation-deactivation equilibrium constant K and the values of k_{act} and k_{deact} (Scheme 9.17), the combination:disproportionation ratio for the reaction of the nitroxide with the propagating radical (Section 9.3.6.3) and the intrinsic stability of the nitroxide and the alkoxyamine under the polymerization conditions (Section 9.3.6.4). The values of K, k_{act} and k_{deact} are influenced by several factors.^{113,117-119}

- (a) The degree of steric compression around the C-O bond.¹¹⁸
- (b) The stabilities of the radicals formed.¹¹⁸ Higher radical stability lowers k_{act} and raises k_{deact} .
- (c) Polar factors.¹¹⁸ Electron-donating groups on the nitroxide lower k_{act} and raise k_{deact} . Electron-withdrawing groups have the inverse effect.
- (d) Hydrogen bonding.^{120,121} Hydroxyl substituents on the alkoxyamine (or on the monomer/solvent) lower k_{act} .

The rates thus depend on the structure of both the reactive radical (initiating radical, propagating radical) and the nitroxide fragment. The structures of some nitroxides used in NMP are shown in Table 9.1-Table 9.4. For structurally related nitroxides K and k_{act} are found to increase in the series five-membered ring (*e.g.* **49**, Table 9.1)< six-membered ring (*e.g.* **67**, Table 9.2) < open chain (*e.g.* **83**, Table 9.3) < seven-membered ring (*e.g.* **92**, Table 9.4).¹¹⁸ Within each series, the incorporation of bulky substituents adjacent to the nitroxide nitrogen increases k_{act} . Thus k_{act} for **58** is less than that for **59**; the value of k_{act} increases in the series **60** < **62** < **64**. In general, factors which increase k_{act} cause k_{deact} to decrease.

These major trends in k_{act} can be qualitatively predicted using semi-empirical molecular orbital calculations.^{118,122,123} However, the methods fail to adequately predict some electronic effects, remote substituent effects and the influence of hydrogen bonding. Higher level *ab initio* or DFT calculations provide a better indication of trends in these circumstances.

Another important factor is the stability of the nitroxide. Some degree of instability appears beneficial. This can compensate for the buildup of nitroxide that would occur as a consequence of radical-radical termination and which might otherwise inhibit polymerization.

A number of NMP processes have been reported where the nitroxide is formed *in situ*. Nitrones¹²⁴⁻¹²⁷ and nitroso-compounds¹²⁸ have been used as nitroxide precursors. Control of methacrylate polymerization by mixtures of nitric oxide and nitrogen dioxide has also been attributed to *in situ* formation of a nitroxide.^{129,130}

472

| Nitroxide | Structure | Nitroxide | Structure | Nitroxide | Structure |
|------------------------------|---|--------------------------------|---|----------------------------------|------------------|
| 49 ^{118,131} | •0 N | 50 ¹³¹⁻¹³⁴ | •o-N Ph | 51 ¹³¹ | •0-N |
| 52 ^{a,131} | •0-N | 53 ¹³⁴ | •0-N YPh | 54 ¹³⁴ | O-N Ph |
| 55 ¹³⁵ | O-N Ph | 56 ¹³⁴ | O-N Ph | | |
| 57 ¹³⁶ | •O-N | 58 ^{a,118,120} | •0-N | 59 111.118.120.122.137 | •o-N |
| 60 ¹³⁸ | • × × × × × × × × × × × × × × × × × × × | 61 ¹³⁸ | •0-N KN | | |
| 62 ¹³⁸ | •0-N N Bu | 63 ¹³⁸ | •o-N N ^N CH ₂ Ph | 64 ^{138,139} | •0-N +0 • CH3 |
| 65 ^{a.134} | | 66 ^{a,123} | •0-N | | |

 Table 9.1 Five-Membered Ring Nitroxides for NMP

a These nitroxides were ineffective in NMP under the conditions reported.

| <u> N</u> itroxide | Structure | Nitroxide | Structure | Nitroxide | Structure |
|-----------------------------|-----------------------|------------------------------|-----------|------------------------------|---------------------------|
| 67 TEMPO | •0-N | 68 | •o-N = 0 | 69 ¹⁴¹ | •о N — ОН |
| 70 ¹⁴³ | •0-N | | N N | 71 ¹⁴⁴⁻¹⁴⁶ | •O-N -OCH3 |
| 72 | •0 | N of | | 73 | •о-N ОРО3Н |
| 74 ¹⁴⁷ | • × v • × v × v | 75 ¹⁴⁷ | •0-N N ← | 76 | •O-N OAc |
| 77''' | •0-N | 78 ^{142,148} | | 79 ¹⁴⁹ | •o-N =0 |
| 80 ¹⁴⁹ | •о-м-он | 81 ¹⁴⁹ | •о-м -он | 82 ¹⁵⁰ | •O-N N-CH ₂ Ph |

 Table 9.2 Six-Membered Ring Nitroxides for NMP)

| Nitroxide | Structure | Nitroxide | Structure | Nitroxide | Structure |
|--|-----------|------------------------------|------------------|--------------------------|-------------|
| 83 DTBN 111,118,120,151 | •0-N | 84 ¹²² | •o-N | 85 ¹⁵² | •O−N →Ph |
| 86 ^{120.142,153,154} | | 87 ¹⁵⁵ | •O-N Ph | 88 ¹²¹ | |
| 89 SG1 ^{120,156-} 158 | | 90 ^{159,160} | •0-N -0-N | 91 ¹⁶¹ | |

Table 9.3 Open-Chain Nitroxides for NMP

Table 9.4 Seven- and Eight-Membered Ring Nitroxides for NMP

| Nitroxide | Structure | Nitroxide | Structure | Nitroxide | Structure |
|--------------------------|-----------|------------------------------|-----------|--------------------------|-----------|
| 92 ¹¹⁸ | •0-N | 93 ^{142,162} | •0-N 0 | 94 ¹⁶² | •O-N OH |
| 95 ¹⁶³ | | 96 ¹⁶³ | | 97 ¹⁶² | •0-N =0 |
| 98 ¹⁶⁴ | •0-N 0 | 99 ¹⁶⁴ | •O-N OH | | |

9.3.6.2 Initiation

Two basic strategies have been applied to initiate NMP. In the first method, the initiator is a low molecular weight alkoxyamine (Scheme 9.4). This approach was used in the original work of Solomon and coworkers.¹¹¹⁻¹¹³ Later, Hawker and coworkers^{140,165} also exploited this method and coined the term 'unimer' to describe these initiators.

In the second approach, the alkoxyamine is formed *in situ* typically from the nitroxide and radicals generated using a conventional initiator (Scheme 9.5). The initiator used in the early work of Georges *et al*,¹¹⁶ was BPO (Scheme 9.18). The yield of alkoxyamine based on BPO is not quantitative and various side reactions are known to accompany alkoxyamine formation (Section 3.5.2.4). When the

alkoxyamine is formed *in situ* the initiator efficiency must be known in order to predict molecular weights or rates of polymerization.



In principle, no added conventional initiator is required for S polymerization within the temperature range 100-130 $^{\circ}C$,¹⁶⁶ since radicals formed from monomer through thermal initiation by the Mayo mechanism generate alkoxyamine initiators (Section 3.3.6.1). However, this method is seldom used in practice because the alkoxyamine generation step constitutes a very long inhibition period (~24 hours depending on reaction temperature and nitroxide concentration).

Catala and coworkers^{167,168} made the discovery that the rate of TEMPOmediated polymerization of S is independent of the concentration of the alkoxyamine. This initially surprising result was soon confirmed by others.^{23,169} Gretza and Matyjaszewski¹⁶⁹ showed that the rate of NMP is controlled by the rate of thermal initiation. With faster decomposing alkoxyamines (those based on the open-chain nitroxides) at lower polymerization temperatures, the rate of thermal initiation is lower such that the rate of polymerization becomes dependent on the alkoxyamine concentration. Irrespective of whether the alkoxyamine initiator is preformed or formed *in situ*, low dispersities require that the alkoxyamine initiator should have a short lifetime. The rate of initiation should be as fast as or faster than propagation under the polymerization conditions and lifetimes of the alkoxyamine initiators should be as short as or shorter than individual polymeric alkoxyamines.

Various methods have been used to form low molecular weight alkoxyamine initiators for NMP. Most involve forming an appropriate carbon-centered radical in the presence of a nitroxide. Initiators that generate carbon-centered radicals may be thermally decomposed in the presence of a nitroxide. For example, alkoxyamine **100** is formed by decomposition of AIBN in the presence of TEMPO (Scheme 9.19).¹¹¹ Carbon-centered radicals may also be generated photochemically.¹⁷⁰



Scheme 9.19

Living Radical Polymerization

Another strategy involves decomposition of a peroxide or other initiator in the presence of a monomer. Conditions can be chosen such that only one unit of monomer is consumed. Thus, decomposition of DBPOX in S in the presence of DTBN provides **101** (Scheme 9.20).¹¹¹ The monomer initiator and/or combination should be chosen with care to obtain high yield of effective alkoxyamines. Many oxygen-centered radicals react with monomer by multiple pathways. Specificities shown by oxygen-centered radicals in their reaction with monomers have been studied extensively and are discussed in Section 3.4.2. Hydrogen abstraction, often by a source of *t*-butoxy radicals at low temperature [*e.g.* (*t*BuO)₂/hv,¹⁷⁰ DBPOX,^{111,171,172} *t*BuOOH/Co(II)¹⁷³], in the presence of a nitroxide is another common method for generating benzylic and other alkoxyamines.



Scheme 9.20

ATRP catalysts may be used to generate radicals and thus alkoxyamines can be produced from alkyl halides in high yield (Scheme 9.21).¹⁷⁴ The alkoxyamine **102** was obtained in 92% yield ¹⁷⁴ whereas reaction of TEMPO with PMMA• under ATRP conditions is reported to provide a macromonomer (Section 9.7.2.1).



The Manganese(V) catalyzed oxidation of S derivatives in the presence of a nitroxide provides excellent yields of phenylethyl alkoxyamines (Scheme 9.22).^{175,176} Alkoxyamines can also be prepared from acrylates by oxymercuration.¹⁷⁷


9.3.6.3 Side reactions

The nitroxides appear completely inert towards most monomers under normal polymerization conditions. Nitroxides, in general, do not directly initiate radical polymerization. Alkoxyamines are susceptible to induced decomposition under typical reaction conditions. DTBN can undergo β -scission with formation of a nitroso compound at high temperatures (Scheme 9.23). TEMPO and other cyclic nitroxides appear intrinsically stable under polymerization conditions because of the much higher likelihood of ring closure to reform the nitroxide. The open chain nitroxides (**85-90**) are thought to show greater instability because of the presence of an α -hydrogen.



Scheme 9.23

While nitroxides give overwhelmingly combination in their reaction with carbon-centered radicals, the amount of disproportionation is finite (Scheme 9.24). Disproportionation cannot always be rigorously distinguished from elimination and it is possible that both reactions occur. The combination: disproportionation ratio (or extent of elimination) depends on the nitroxide and radical structure and within a series of structurally related systems appears to increase as k_{act} increases.^{122,178}



Scheme 9.24

The thermal decomposition of the phenylethyl alkoxyamine with TEMPO and the fraction of living ends in TEMPO-mediated S polymerization has been studied by Priddy and coworkers.^{143,179} They concluded that to achieve >90% living ends conversions and/or nitroxide concentrations should be chosen to give M_n less than 10000.¹⁴³ However, disproportionation or elimination is most important during polymerizations of methacrylates and accounts for NMP being less successful with

these monomers (Scheme 9.25).¹²² The process also provides a method of macromonomer synthesis (Sections 9.7.1.1 and 9.7.2.1).



Scheme 9.25

9.3.6.4 Rate enhancement

Various strategies have been used to enhance the rate of NMP and, in particular, that mediated by TEMPO. The effects of some of these strategies on polymerization kinetics have been considered by Souaille and Fischer.¹⁸⁰ Most are based on the use of reagents that directly or indirectly consume and regulate the excess nitroxide that is formed continuously during polymerization as a consequence of radical-radical termination between propagating radicals. The amount of free nitroxide required to significantly retard polymerization is very small (~ 10^{-4} M). Reagents used include the following.

- (a) Anhydrides.^{153,181-183}
- (b) Sulfonic acids (*e.g.* camphorsulfonic acid,^{184,185} sulfoethyl methacrylate¹⁸⁵), and their salts.^{186,187} The sulfonic acid accelerants also inhibit thermal initiation of S polymerization by consuming the intermediate Diels-Alder dimer (Section 3.3.6.1).¹⁸⁵ It has been established that k_p for S is unaffected by sulfonic acid.
- (c) Reducing agents (including ascorbic acid).¹⁸⁸ Added ascorbic acid is used to facilitate miniemulsion NMP of acrylates with TEMPO.^{189,190} Reduction provides a hydroxylamine which can react as a transfer agent to reform nitroxide.
- (d) Additional (conventional) initiators.¹⁹¹⁻¹⁹⁵ This initiator is chosen to decompose slowly so as to generate a low concentration of additional radicals continuously throughout the experiment. The initiator-derived radicals consume the excess nitroxide but also generate additional polymer chains. The initiator concentration used is thus critical.

Another strategy is to use a nitroxide that is intrinsically unstable. Part of the success of the open chain nitroxides that have an α -hydrogen (86-90) has been attributed to this factor.

The Chemistry of Radical Polymerization

9.3.6.5 Monomers

Alkoxyamine C-O bond homolysis rates have been shown to increase where propagating radical is:^{23,118}

$$\begin{array}{cccc} H & H & CH_3 & CH_3 \\ \text{wwwCH}_2\dot{C}, \bullet < & \text{wwwCH}_2\dot{C}, \bullet < & \text{wwwCH}_2\dot{C}, \bullet < & \text{wwwCH}_2\dot{C}, \bullet \\ & CO_2CH_3 & Ph & CO_2CH_3 & Ph \end{array}$$

NMP has mainly been used for S polymerization (9.3.6.5.1) and, to a lesser extent, acrylate (9.3.6.5.2) polymerization. The early and much current work has focused on the use of TEMPO and derivatives. The open chain nitroxides **86-91** (Table 9.3) provide broader though still restricted utility. Some of the previously 'difficult' monomers that have recently been tackled successfully include HEA,¹⁹⁶ DMAM¹⁹⁷ and AA^{198,199} with nitroxide **89**.

9.3.6.5.1 Styrene, vinyl aromatics

NMP is most commonly used for S polymerization. For S polymerizations carried out at temperatures greater than 100 °C, thermal initiation provides some rate enhancement and a mechanism for controlling the excess of nitroxide that is formed as a consequence of radical-radical termination and the persistent radical effect.^{23,169}



Various substituted styrenes have been also polymerized by NMP. These include 103-107, *p*-chloromethylstyrene (108), *p*-halostyrenes, and *p*-acetoxystyrene. Vinyl pyridines (*e.g.* 109) are amenable to NMP²⁰⁰ and may be quaternized post-polymerization to provide water-soluble polymers.

9.3.6.5.2 Acrylates

NMP with acrylates and acrylamides with TEMPO provides only very low conversions. Very low limiting conversions and broad dispersities were reported.²⁰¹ Better results were obtained with DTBN (83),^{111,151} imidazoline (61-64)¹³⁸ and isoindoline (59) nitroxides.¹¹¹ However, limiting conversions were still observed. The self-regulation provided in S polymerization by thermal initiation is absent and, as a consequence, polymerization proceeds until inhibited by the build-up of nitroxide. The final product is an alkoxyamine and NMP can be continued

following polymer isolation and purification. The use of additives and reaction conditions to control excess nitroxide concentrations also allows higher conversions to be obtained. $^{151,188-190}$

Much better control is obtained with the open chain nitroxides, in particular 86^{153} and 89,^{156,158} where much lower reaction temperatures can be used and high conversions are achieved.

Molecular weights may also be limited by the occurrence of backbiting and fragmentation when high reaction temperatures are used. Backbiting without fragmentation was observed for BA polymerization at 112 °C with **89** (no unsaturated end groups observed by ¹H NMR).¹⁵⁸ However, macromonomer chain ends are clearly evident in the ¹H NMR of PtBA prepared with DTBN (**83**) at 120 °C.¹⁵¹ For a system showing limiting conversion behavior side reactions of the propagating radical, such as backbiting-fragmentation or disproportionation, have much greater significance as their rate is not slowed as propagation is slowed by nitroxide build-up through the persistent radical effect.

9.3.6.5.3 Methacrylates

NMP with methacrylates is generally recognized as being difficult. It is possible to make PMMA by NMP¹²² and examples of PMMA and PMMA block copolymers are provided in the first NMP patent.¹¹¹ However, in attempts to obtain high molecular weight polymers, limiting conversion behavior is observed and the product is a macromonomer.^{111,122,202} Even though these high conversion polymerizations yield 'dead' polymer, a very close correspondence of found and ealculated molecular weights is observed.¹²² This demonstrates that the polymer that is produced is formed as a consequence of NMP and that there is little chain transfer or other mechanisms for initiation.

9.3.6.5.4 Diene monomers

Of the major methods for living radical polymerization, NMP appears the most successful for polymerization of the diene monomers. There are a number of reports on the use of NMP of diene monomers (B, I) with TEMPO,^{188,203} **86**^{154,204} and other nitroxides.¹²⁷ High reaction temperatures (120-135 °C) were employed in all cases. The ratio of 1,2-:1,4-cis:1,4-trans structures obtained is similar to that observed in conventional radical polymerization (Section 4.3.2).

9.3.6.6 Heterogeneous polymerization

NMP of S in heterogeneous media is discussed in reviews by Qiu *et al.*,²⁰⁵ Cunningham,^{206,207} and Schork *et al.*²⁰⁸ There have been several theoretical studies dealing with NMP and other living radical procedures in emulsion and miniemulsion.²⁰⁹⁻²¹³ Butte *et al.*^{210,214} concluded that NMP (and ATRP) should be subject to marked retardation as a consequence of the persistent radical effect. Charleux²⁰⁹ predicted enhanced polymerization rates for miniemulsion with small

(50-100 nm) particles when the persistent radical can be desorbed from the particle phase. Ma *et al.*²¹¹⁻²¹³ also concluded that the distribution of nitroxide between the aqueous and organic phases was critical to maintaining livingness and achieving acceptable polymerization rates.

The early attempts at NMP of S in emulsion used TEMPO and related nitroxides and needed to be carried out at high temperatures (100-130 °C) necessitating a pressure reactor. Problems with colloidal stability and molecular weight control and limiting conversions were reported.²¹⁵⁻²¹⁷

Successful NMP in emulsion requires use of conditions where there is no discrete monomer droplet phase and a mechanism to remove any excess nitroxide formed in the particle phase as a consequence of the persistent radical effect. Szkurhan and Georges²¹⁸ precipitated an acetone solution of a low molecular weight TEMPO-terminated PS into an aqueous solution of PVA to form emulsion particles. These were swollen with monomer and polymerized at 135 °C to yield very low dispersity PS and a stable latex. Nicolas *et al.*²¹⁹ performed emulsion NMP of BA at 90 °C making use of the water-soluble alkoxyamine **110** or the corresponding sodium salt both of which are based on the open-chain nitroxide **89**. They obtained PBA with narrow molecular weight distribution as a stable latex at a relatively high solids level (26%). A low dispersity PBA-*block*-PS was also prepared.



NMP in miniemulsion has been more successful. In miniemulsion polymerization nucleation takes place directly in the monomer droplets that become the polymer particles. Particle sizes are small (<100 nm). Most work has used TEMPO and high reaction temperatures (120-140 °C) with S or BA as monomer.

Various initiation strategies and surfactant/cosurfactant systems have been used. Early work involved *in situ* alkoxyamine formation with either oil soluble (BPO)^{220,221} or water soluble initiators (persulfate) and traditional surfactant and hydrophobic cosurfactants. Later work established that preformed polymer could perform the role of the cosurfactant and surfactant-free systems with persulfate initiation were also developed.^{190,222,223} Oil soluble (PS capped with TEMPO,²²¹ **111**,²²⁴ PBA capped with **89**) and water soluble alkoxyamines (**110**, sodium salt²²⁴) have also been used as initiators. Addition of ascorbic acid, which reduces the nitroxide which exits the particles to the corresponding hydroxylamine, gave enhanced rates and improved conversions in miniemulsion polymerization with TEMPO.²²⁵ Ascorbic acid is localized in the aqueous phase by solubility.

Living Radical Polymerization

9.3.7 Other Oxygen-Centered Radical-Mediated Polymerizations

A number of other chemistries which involve C-O bond cleavage have been reported.^{226,227} Druliner²²⁶ has reported on systems where NCO•, **112**, **113** or related species is the persistent radical. Homolysis rates for these systems were stated to be suitable for MMA polymerization at ambient temperature. The use of NCO• has also been studied by Grande *et al.*,²²⁸⁻²³⁰ most recently for AA polymerization.²³⁰ Although control during AA homopolymerization was poor the process yielded NCO- terminated PAA that could be used to make PAA-block-PMMA.²³⁰



Chung and coworkers have reported on the use of stable borinate or boroxyl radicals (*e.g.* **114**) to mediate radical polymerization.^{231,232} Methacrylates (MMA) and acrylates (trifluoroethyl acrylate) have been polymerized at ambient temperature to yield polymers with relatively narrow molecular weight distributions.²³¹⁻²³³ The method has been used to prepare block copolymers and polyolefin graft copolymers.²³⁴⁻²³⁷

A living radical polymerization mechanism was proposed for the polymerization of $MMA^{238-240}$ and VAc^{241} initiated by certain aluminum complexes in the presence of nitroxides. It was originally thought that a carbonaluminum bond was formed in a reversible termination step. However, a more recent study found the results difficult to reproduce and the mechanism to be complex.²⁴²

9.3.8 Nitrogen-Centered Radical-Mediated Polymerization

A few studies have appeared on systems based on persistent nitrogen-centered radicals. Yamada *et al.*²²⁷ examined the synthesis of block polymers of S and MMA initiated by derivatives of the triphenylverdazyl radical **115**. Klapper and coworkers²⁴³ have reported on the use of triazolinyl radicals (*e.g.* **116** and **117**). The triazolinyl radicals have been used to control S, methacrylate and acrylate polymerization and for the synthesis of block copolymers based on these monomers [S,²⁴³⁻²⁴⁵ tBA,²⁴³ MMA,²⁴³⁻²⁴⁵ BMA,²⁴⁵ DMAEMA,²⁴⁶ TMSEMA,²⁴⁷ (DMAEMA-*block*-MMA),²⁴⁶ (DMAEMA-*block*-S)²⁴⁶ and (TMSEMA-*block*-S)²⁴⁷]. Reaction conditions in these experiments were similar to those used for NMP. The triazolinyl radicals show no tendency to give disproportionation with methacrylate propagating radicals. Dispersities reported are typically in the range 1.4-1.8.^{243,246}



The triazolinyl radical 116 is thermally unstable with a half-life of ~ 20 min at 95 °C. The compound 117 is stable under similar conditions. The decomposition mechanism involves loss of a phenyl radical and formation of a stable aromatic triazene (Scheme 9.26).²⁴³ This provides a mechanism for self regulation of the stable radical concentration during polymerization and a supplemental source of initiating radicals.



9.3.9 **Metal Complex-Mediated Radical Polymerization**

Metal complexes may also act as initiators in stable radical-mediated polymerization with the metal complex performing the role of the stable radical. There are reports of titanocene,^{248,249} cobalt,²⁵⁰⁻²⁵³ chromium, iron and molybdenum²⁵⁴ complexes in this context. Oganova *et al.*²⁵⁵⁻²⁵⁸ observed that certain cobalt (II) porphyrin complexes

reversibly inhibit BA polymerization presumably with formation of a cobalt (III) intermediate as shown in Scheme 9.27. Thus, it seemed reasonable to propose these species may function as initiators in living radical polymerization.^{250,259}



Wayland et al. reported the use of tetramesitylporphyrin complexes (CoTMP),

including 118^{250} and 119^{251} in the synthesis of high molecular weight PMA with very low dispersities (1.1-1.3). Arvanitopoulos et al.²⁶⁰ have reported similar chemistry with alkylcobaloximes (120) as photoinitiators at low temperatures.



The most important side reactions are disproportionation between the cobalt(II) complex and the propagating species and/or β -climination of an alkene from the cobalt(III) intermediate. Both pathways appear unimportant in the case of acrylate ester polymerizations mediated by Co^{II}TMP but are of major importance with methacrylate esters and S. This chemistry, while precluding living polymerization, has led to the development of cobalt complexes for use in catalytic chain transfer (Section 6.2.5).

It is also known that alkyl cobaloximes are subject to radical-induced decomposition.²⁵⁷ This suggests an alternative to the mechanism shown in Scheme 9.28 involving reversible chain transfer (Section 9.5).

initiation



reversible primary radical termination

 P_n^{\bullet} Co^{II}TMP \longrightarrow P_n^{-} Co^{III}TMP Scheme 9.28

It has also been shown that the alkyl cobalt (III) initiator can be generated *in* situ²⁵² by adding a fast-decomposing azo-initiator [2,2'-azo-bis(4-methoxy-2,4dimethyl valeronitrile] to a solution of the cobalt (II) complex in monomer. Very narrow dispersity PMA and PMA-*block*-PBA were prepared. In a very recent development, Debuigne *et al.*²⁵³ have reported polymerization

In a very recent development, Debuigne *et al.*²⁵⁵ have reported polymerization of vinyl acetate at 30 °C mediated by Co^{II}(acac)₂ (**121**). They obtained predictable molecular weights up to \overline{M}_n =100000 and dispersities < 1.3 and proposed a polymerization mechanism analogous to that shown in Scheme 9.27. The complex offered no control over BA polymerization and the porphyrin complexes inhibited VAc polymerization.

9.4 Atom Transfer Radical Polymerization

The addition of halocarbons (RX) across alkene double bonds in a radical chain process, the Kharasch reaction (Scheme 9.29),²⁶¹ has been known to organic chemistry since 1932. The overall process can be catalyzed by transition metal complexes (Mt^n -X); it is then called Atom Transfer Radical Addition (ATRA) (Scheme 9.30).²⁶²

Polymer formation during the Kharasch reaction or ATRA can occur if trapping of the radical (123), by halocarbon or metal complex respectively, is sufficiently slow such that multiple monomer additions can occur. Efficient polymer synthesis additionally requires that the trapping reaction is reversible and that both the activation and deactivation steps are facile.



Scheme 9.29 Kharasch Reaction



Scheme 9.30 Atom Transfer Radical Addition (ATRA)

The first purposeful use of ATRA in polymer synthesis was in the production of telomers.²⁶³ In this early work, comparatively poor control over the polymerization was achieved and little attempt was made to explore the wider utility of the process. Some analogies may also be drawn with the work of Bamford *et al.* and others on transition metal/organic halide redox initiation (Sections 3.3.5.1 and 7.6.2).²⁶⁴

The first reports of ATRP (Atom Transfer Radical Polymerization), which clearly displayed the characteristics of living polymerization, appeared in 1995 from the laboratories of Sawamoto,²⁶⁵ Matyjaszewski²⁶⁶ and Percec.²⁶⁷ The literature on ATRP is now so vast that a comprehensive review cannot be

Living Radical Polymerization

presented here. A number of reviews on ATRP have appeared. Most informative on the scope of the process are those by Matyjaszewski and Xia^{268,269} and Kajimoto *et al.*^{270,271} The kinetics of ATRP are considered in reviews by Fischer¹¹⁰ and Goto and Fukuda.²³ ATRP is sometimes also called transition metal-mediated radical polymerization. We use this latter term for radical polymerizations where control is achieved by a reversible coupling mechanism (Section 9.3.9).



A much-simplified mechanism for reversible activation-deactivation of polymer chains during ATRP is shown in Scheme 9.31. In the deactivation process, propagating radicals are trapped by atom or group transfer [most commonly a halogen (Cl, Br, I) although other groups (*e.g.* SCN) are known] from a metal complex in its higher oxidation state. The activation process involves a redox reaction between the polymer end group and the metal complex in its reduced form.

 P_n^{\bullet} + X-Mtⁿ⁺¹ \longrightarrow $P_n^{-}X^{\bullet}$ + Mtⁿ⁺¹ \longrightarrow $P_n^{-}X$ + Mtⁿ

Scheme 9.32

The atom transfer reaction is generally thought to involve inner sphere electron transfer (ISET) with concerted transfer of the halogen from initiator to the metal complex and various kinetic and other data support this view for most of the common initiator/catalyst/monomer combinations. However, it is possible to write the process as two steps, the first being an outer sphere electron transfer (OSET) process to provide an intermediate radical anion (Scheme 9.32).²⁶⁸ The living polymerization of vinyl chloride with alkyl iodide initiators and nascent Cu(0) catalyst is considered to involve an OSET process.^{272,273} OSET does not require a transition metal catalyst and can involve other single electron reducing agents such as dithionite.²⁷⁴ For this case it is also possible that the chain equilibration step is, in part, similar to that discussed under iodine transfer polymerization (Section 9.5.4).²⁷⁴

Ideally, the metal complex is a catalyst and, in principle, is only required in very small quantities. However, the kinetics of initiation for the systems described to date dictate that relatively large amounts are used and catalyst:initiator ratios are typically in the range 1:1 to 1:10. The most commonly used catalysts are metal

The Chemistry of Radical Polymerization

complexes based on Cu and Ru. However, a wide range of metals and ligands has been used (Section 9.4.2). Conditions and catalysts have been found such that most monomers polymerizable by a radical mechanism can be used in ATRP. Difficult monomers are vinyl acetate and simple olefins (in homopolymerization) and monomers that coordinate strongly with metal centers. It is extremely important to select the initiator, catalyst and reaction conditions for the particular monomer.

There has been some discussion on whether ATRP is a 'free' radical polymerization.^{275,276} Are the reactions of initiating and propagating species produced in ATRP influenced by the presence of the metal complex? Reports^{275,276} that reactivity ratios in copolymerization by ATRP differ from those observed in conventional radical polymerization appear to be an effect of chain length (Section 9.6). There is no doubt that the rate of polymerization in ATRP can be dramatically affected by the reaction medium but this can in large part be attributed to changes in the activation/deactivation equilibrium. The current general consensus is that the common forms of ATRP are radical processes and the propagating radicals behave as 'free' propagating radicals under the reaction conditions. The polymerization kinetics can be interpreted on this basis and radical-radical termination occurs to the extent expected given the radical concentration,

Notwithstanding the occurrence of any side reactions, a successful ATRP experiment will generally yield a polymer with halogen end groups. These endgroups are potentially labile and may impair polymer stability. Moreover, corrosive by-products (hydrohalic acids) can be formed by thermal elimination. However, the end groups are also precursors to a wide range of other functionality. It is possible to transform them into groups that are chemically inert or to useful functionalities (Section 9.7.2.1). They also render the polymers useful as precursors to block, star, comb and more complex architectures (Sections 9.8-9.9.3.2).

9.4.1 Initiators

The initiator in ATRP is usually a low molecular weight activated organic halide (RX, R=activated alkyl, X=chlorine, bromine, iodine). However, organic pseudohalides (*e,g*, X=thiocyanate, azide) and compounds with weak N-X (*e.g.* N-bromosuccinimide²⁷⁷) or S-X (*e.g.* sulfonyl halides - see below) have been used.

The first reported initiators were polyhalogeno-compounds (*e.g.* CCl_4 , $CHCl_3$, CCl_3CH_2OH , CCl_3Br). Trichloromethane derivatives and tetrachloromethane appear effective initiators. Mono- and dichloromethane derivatives are inefficient initiators. Tetrachloromethane may act as a diffunctional initiator.

In choosing an initiator the strength of the R-X bond in both the initiator and the dormant propagating species formed should be considered. It is common practice to use a compound such that the radical generated is a monomeric or low molecular weight species structurally analogous to the propagating radical. Thus, α -bromoisopropionates (124) are used for acrylates, α -bromoisobutyrates (125) are used to initiate polymerization of MMA and other methacrylates (Scheme 9.33), and benzyl bromide (126) or phenylethyl bromide (127) is used to initiate polymerization of S and derivatives. Initiator activity is discussed further in Section 9.4.1.3.



initiation



reversible deactivation

 P_n^{\bullet} $Cu^{11}Br/L \longrightarrow P_n^{-}Br Cu^{1}/L$ Scheme 9.33

Other important classes of initiator are the organic sulfonyl chlorides (*e.g.* **129**, **130**)^{267,278-280} and bromides (*e.g.* **131**).²⁸¹ These are very effective when used in conjunction with copper catalysts with bpy or dNbpy ligands. Functional sulfonyl chloride initiators have also been reported (Section 9.7.2.2). Rates of radical generation are high with respect to propagation such that they can be used with methacrylates, styrenes and acrylates. In some circumstances, initiator efficiencies observed with sulfonyl halide initiators may be lowered by side reactions involving the sulfonyl radicals.^{282,283} These side reaction include reaction of sulfonyl radical with the ligand (PMDETA) by hydrogen abstraction.²⁸² This pathway is not important with bipyridyl ligands (bpy, dNbpy). With ruthenium catalysts that use a Al(OiPr)₃ cocatalyst, the cocatalyst may react with the sulfonyl chloride to cause a decrease in the initiator efficiency.²⁸⁴

9.4.1.1 Molecular weights and distributions

In ATRP, the initiator (RX) determines the number of growing chains. Ideally, the degree of polymerization is given by eq. 7 and the molecular weight by eq. 8. Note the appearance of the initiator efficiency (f') in the numerator of these expressions. In practice, the molecular weight is often higher than anticipated because the initiator efficiency is decreased by side reactions. In some cases, these take the form of heterolytic decomposition or elimination reactions. Further redox chemistry of the initially formed radicals is also known. The initiator efficiencies are dependent on the particular catalyst employed.

$$\overline{X}_{n} = \frac{([\mathbf{M}]_{o} - [\mathbf{M}]_{t})f'}{[\mathbf{R}\mathbf{X}]_{o}} = \frac{[\mathbf{M}]_{o}f'}{[\mathbf{R}\mathbf{X}]_{o}}c$$
(7)

$$\overline{M}_{n} = \frac{\left(\left[\mathbf{M}\right]_{0} - \left[\mathbf{M}\right]_{1}\right)f'}{\left[\mathbf{R}\mathbf{X}\right]_{0}}m_{\mathrm{M}} + m_{\mathrm{RX}}$$
(8)

where $([M]_o-[M]_t)$ is the concentration of monomer consumed m_M and m_{RX} are the molecular weights of the monomer and the initiator (RX) respectively, and *c* is the monomer conversion.

It is assumed in the derivation of eq. 7 that RX is completely consumed. In order to obtain good control (low dispersities, molecular weights according to eq. 7) it is critical that initiation is rapid with respect to propagation such that RX is consumed before there is any substantial conversion of monomer. Slow usage of RX will give a post-tailing or bimodal molecular weight distribution.

In S polymerization, thermal initiation will be a source of extra chains. Additional chain formation processes will cause the molecular weight to be lower than anticipated by eq. 7. Sometimes conventional thermal initiators are added with similar effect (see also eq. 12). A pre-tailing molecular weight distribution may result.

In ideal circumstances, with polymerization described by Scheme 9.31 and rate of activation of RX equal to that of P_nX , the dispersity is given by eq. 9.²³

$$\frac{\overline{X}_{n}}{\overline{X}_{n}} = 1 + \frac{1}{\overline{X}_{n}} + \left(\frac{2-c}{c}\right) \frac{k_{p}[\mathbf{RX}]}{k_{\text{deat}}[\mathbf{Mt}^{n+1}\mathbf{X}]}$$
(9)

where c is the monomer conversion.

The rate of polymerization is given by eq. 10.

$$R_{\rm p} = k_{\rm p} K \frac{[\rm RX][\rm Mt^n]}{[\rm Mt^{n+1}X]} [\rm M]$$
(10)

The ATRP experiment is usually commenced with all of the catalyst in its lower oxidation state. The number of propagation events per activation cycle is

dependent on the concentration of catalyst in its higher oxidation state. For low dispersities it is important that this number is small. As indicated by eq. 9 dispersity is inversely proportional to the concentration of the deactivator $(Mt^{n+1}X)$. Thus, just as in NMP, where it is desirable to have a very low concentration of free nitroxide in the polymerization medium, in ATRP it can be important to have a proportion of the catalyst in its higher oxidation state. However, as implied by eq. 10, a concentration of deactivator that is too high can cause retardation or even inhibition of polymerization.

9.4.1.2 Reverse ATRP

So-called reverse ATRP has been described where a conventional radical initiator (*e.g.* AIBN) and a transition metal complex in its higher oxidation state are used.²⁸⁵⁻²⁸⁸ One of the first systems explored was CuBr₂/133/AIBN/MMA. It is important that the initiator is completely consumed early in the polymerization. The use of peroxide initiators in reverse ATRP can be problematical depending on the catalyst used and the reaction temperature.^{286,289} The system CuBr₂/133/BPO/MMA at 60°C was found to provide no control.²⁸⁶ In ATRP at lower temperatures (40 °C), the system CuCl/133/BPO/MMA was successful though dispersities obtained were relatively broad,²⁸⁹ Radicals are produced from the redox reaction between the catalyst in its reduced form and BPO.

The molecular weight in reverse ATRP will depend on the concentration of the initiator (I_2) and the initiator efficiency (f) and ideally is given by eq. 11. Side reactions between the catalyst and the initiator and the radicals formed from the initiator may lead to efficiencies being lower than those observed in conventional radical polymerization.

$$\overline{X}_{u} = \frac{[\mathbf{M}]_{o}}{[\mathbf{I}_{2}]_{o}f}c \tag{11}$$

Experiments have been described where a combination of direct and reverse ATRP is used.²⁹⁰ In this case eq. 12 should apply.

$$X_n = \frac{[\mathbf{M}]_o}{[\mathbf{I}_2]_o f + [\mathbf{RX}]_o} c \tag{12}$$

In combination ATRP, the catalyst is again present in its more stable oxidized form. A slow decomposing conventional initiator (*e.g.* AIBN) is used together with a normal ATRP initiator. Initiator concentrations and rate of radical generation are chosen such that most chains are initiated by the ATRP initiator so dispersities can be very narrow.²⁹⁰ The conventional initiator is responsible for generating the activator *in situ* and prevents build up of deactivator due to the persistent radical effect. Reverse or combination ATRP are the preferred modes of initiation for ATRP in emulsion or miniemulsion (Section 9.4.3.2).^{290,291}

9.4.1.3 Initiator activity

The activity of initiators in ATRP is often judged qualitatively from the dispersity of the polymer product, the precision of molecular weight control and the observed rates of polymerization. Rates of initiator consumption are dependent on the value of the activation-deactivation equilibrium constant (K) and not simply on the activation rate constant (k_{act}). Rate constants and activation parameters are becoming available and some valuable trends for the dependence of these on initiator structure have been established.²⁹²⁻²⁹⁷

- (a) For compounds with a similar activating group, tertiary halides are substantially more active than secondary halides, which, in turn, are more active than primary halides. Thus activity increases in the series: 126 < 127 < 128; and 124 < 125.
- (b) In the case of alkyl halide initiators >C(R)-X, activity is reported to decrease in the series where the activating group R is CN>C(O)R>C(O)OR>Ph>>Cl> OCOCH3>Me.²⁶⁸ Note, this order does not reflect the carbon-halogen bond dissociation energies or the product radical stability. This parallels the trend in activation rate constants for propagating radicals.²⁶⁸

$$\begin{array}{cccc} CH_3 & H & H & H \\ \text{vmm}CH_2 \acute{C} \bullet & > & \text{vmm}CH_2 \acute{C} \bullet & > & \text{vmm}CH_2 \acute{C} \bullet \\ CO_2 CH_3 & Ph & CO_2 CH_3 & OCOCH_3 \end{array}$$

(c) As in other radical processes (and activation NMP and RAFT), penultimate unit effects are important in determining the rate constant for activation.²⁹⁶ Dimeric (and higher) species are more active than monomeric species particularly in the case of tertiary radicals.

While the above trends appear generic, initiator activity is strongly dependent on the specific catalyst used (Section 9.4.2).

9.4.2 Catalysts

Transition metal catalysts are characterized by their redox chemistry (catalysts can be considered as one electron oxidants/reductants). They may also be categorized by their halogen affinity. While in the initial reports on ATRP (and in most subsequent work) copper^{266,267} or ruthenium complexes²⁶⁵ were used, a wide range of transition metal complexes have been used as catalysts in ATRP.

- (a) Group 6: molybdenum (Mo^{IV}-Mo^V).^{254,298,299}
- (b) Group 7: manganese $(Mn^{II}-Mn^{III})$, ^{300,301} rhenium $(Re^{V}-Re^{VI})$. ^{302,303}
- (c) Group 8: iron (Fe^{II} - Fe^{III}), (Fe^{I} - Fe^{II}) (Section 9.4.2.3), ruthenium (Ru^{II} - Ru^{III}) (Section 9.4.2.2)
- (d) Group 9: cobalt (Co^0-Co^1) ,³⁰⁴ rhodium (Rh^1-Rh^{II}) ,³⁰⁵⁻³⁰⁷
- (e) Group 10: nickel (Ni^{II}-Ni^{III}) (Section 9.4.2.4), and palladium (Pd^{II}-Pd^{III}).³⁰⁸
- (f) Group 11: copper ($Cu^{I}-Cu^{II}$) (Section 9.4.2.1)

Most are proposed to involve the general ATRP mechanism. However, it should also be noted that the detailed mechanism has not been elucidated in all cases and not all need be radical processes in the conventional sense. Moreover, in many polymerizations, the active catalyst is formed *in situ* and its exact nature is not rigorously established.

An issue with ATRP is the residual metal catalyst and its removal from the polymer post-polymerization. Many papers have been written on catalyst removal and recycling.³⁰⁹

9.4.2.1 Copper complexes

The most common catalysts for ATRP are complexes based on a copper(I) halide and nitrogen based ligand(s). Various ligands have been employed and those most frequently encountered are summarized in Table 9.5. Typically, four nitrogens coordinate to copper. The bidentate bipyridyl (bpy) ligands **132-133** are known to form a 2:1 complex. The tetradentate ligands are expected to form a 1:1 complex.

The first ATRP experiments were conducted with a complex presumed to be of the form $[Cu^{l}(bpy)_{2}]^{t}X^{-}$ as catalyst and either alkyl halide²⁶⁶ or sulfonyl chloride initiators.²⁶⁷ The complexes were formed *in situ* and the experimental process involved mixing Cu^{ll} halide and the ligand in the reaction medium. The reactions with 2,2'-bipyridine (bpy, 132) are generally heterogeneous and the precise structure of the active catalyst in solution was not known. The bpy derivatives with long chain alkyl groups (134, 133) were introduced to provide greater solubility for the copper complex and allow a more homogeneous polymerization and therefore improved control over polymerization. Many studies probing the solution and solid-state structures of bipyridine and other complexes have now been carried out.³¹⁰

Certain multidentate ligands also provide for better solubility. Cu¹ complexes formed with tetramethylethylenediamine (TMEDA), N,N,N',N',N''pentamethyldiethylenetriamine (PMDETA, 140) and 1,1,4,7,10,10hexamethyltriethylenetetramine (HMTETA, 144) and Mc₆TREN (145) have been found effective.³¹¹ Transfer to ligand during MMA polymerization has been reported as a side reaction when PMDETA is used.^{312,313}

Haddleton and coworkers³¹⁴ reported the use of Cu¹ complexes based on the methanimine ligands (*e.g.* **136-138**) and have demonstrated their efficacy in the polymerization of methacrylates. The ligands can be prepared *in situ* from the appropriate amine and 2-pyridine carboxaldehyde (Scheme 9.34).

Guidelines for predicting the activity of complexes formed with various ligands have been formulated.^{268,269} The activity goes up according to the number of nitrogens coordinated to copper and with the electron donating ability of the nitrogens. Tetradentate ligands appear more effective than tri- or bidentate ligands. Some correlation between k_{act} and k_{deact} and the redox potential of the complex has been observed.^{315,316} A lower redox potential results in a higher k_{act}

and a lower k_{deact} . However, there appears no direct correlation with structural features of the complex such as Cu-Br bond lengths.³¹⁰



Scheme 9.34





Percec and coworkers^{327,328} reported *in situ* formation of active CuCl/CuCl₂ catalyst from the initiator, Cu₂O, Cu(0) and combinations of these in conjunction with ligand (bpy) and various polyethers or ethylene glycol and suggested that improved control was obtained under these conditions.

Supported copper catalysts have also been described.³²⁹⁻³⁴⁰ The main impetus for the development of supported ATRP catalysts has been to facilitate catalyst removal and, in some cases, to allow for catalyst recycling.

9.4.2.2 Ruthenium complexes

In contrast to the situation with copper-based catalysts, most studies on ruthenium-based catalysts have made use of preformed metal complexes. The first reports of ruthenium-mediated polymerization by Sawamoto and coworkers appeared in 1995.²⁶⁵ In the early work, the square pyramidal ruthenium (II) halide **146** was used in combination with a cocatalyst (usually aluminum isopropoxide).

| Structure | Monomer | Structure | Monomer |
|--|--|--|-------------------------------------|
| PPh ₃ Cl [.] ·Ru [‼] ., _{PPh3} Ph₃P Cl 146 | MMA ^{265,341,342} EMA ³⁴³ BMA ³⁴³ DMAM ³⁴⁴ S ³⁴⁵ | CI ^{-Ru‼} .,PPh ₃ PPh ₃ 147 | MMA ³⁴² |
| CI ^{-Rull} , PPh ₃ PPh ₃ 148 | MMA ³⁴² S ³⁴² | CI ^{-Ru!!} ,PPh ₃ PPh ₃ 149 | $\frac{MMA^{346}}{MA^{346}}S^{346}$ |
| CI - Ru ^{II} , PPh ₃ PPh ₃ 150 | MMA ³⁴⁷ MA ³⁴⁷ S ³⁴⁷ | CI Rulling PPh2 Me2N | MMA ³⁴⁸ |
| CI Ru ^{II} PPh ₂ PPh ₂ SO ₃ ⁻ Na ⁺ SO ₃ ⁻ Na ⁺ 152 | HEMA ³⁴⁹ | | |

 Table 9.6
 Ruthenium Complexes Used as ATRP Catalysts

There has been substantial work on catalyst development with the aim of finding more active catalysts and catalysts appropriate for different monomers and reaction media.^{270,271,348} The complexes **149-151** (Table 9.6) appear to be some of the more active catalysts.

The Chemistry of Radical Polymerization

9.4.2.3 Iron complexes

The catalysts **153-155** shown in Table 9.7 have been used for polymerizations of acrylates and methacrylates and S. The catalyst **155** used in conjunction with an iodo compound initiator has also been employed for VAc polymerization.³⁵⁰ Catalytic chain transfer (Section 6.2.5) occurs in competition with halogen atom transfer with some catalysts.

| Catalyst | Structure | Monomer | Catalyst | Structure | Monomer |
|----------|-------------------------|------------------------------|----------|-----------|------------------|
| 153 | Br Fe‼,"PPh₃ PPh₃ | MMA ³⁵¹ | 154 | | S ³⁵² |
| 155 | | S VAc ^{350,353,354} | | | |

 Table 9.7 Iron Complexes Used as ATRP Catalysts

Polymerizations of S and MMA with *in situ* catalyst formation have also been carried out. Matyjaszewski *et al.*³⁵⁵ reported on the use of FeBr₂ together with various ligands such as $P(C_4H_9)_3$, $N(C_4H_9)_3$ and **133** alone or in combination. The use of dicarboxylic acid (iminodiacetic acid, isophthalic acid)³⁵⁶ and methanimine ligands^{357,358} for MMA polymerization has also been reported.

9.4.2.4 Nickel complexes

Nickel complexes (156-159) used as ATRP catalysts for polymerization of (meth)acrylates are shown in Table 9.8.

| Catalyst | Structure | Monomer | Catalyst | Structure | Monomer |
|----------|---|---------------------------|----------|--|-----------------------------|
| 156 | Br Br Ni ,PPh ₃ PPh ₃ | MMA ³⁵⁹ | 157 | $(C_4H_9)_3P_{(C_4H_9)_3}$ Br $Br P(C_4H_9)_3$ | MMA MA BA ³⁶⁰ |
| 158 | N | MMA BMA ³⁶¹ | 159 | PPh ₃ Ph ₃ P ^{-Ni^{II}PPh₃ PPh₃} | MMA ³⁶² |

 Table 9.8
 Nickel Complexes Used as ATRP Catalysts

The complex 157 is more soluble than 156 in organic solvents; it is more thermally stable and can be used at higher temperatures. Moreover, it can be used without the $\Lambda l((OiPr)_3 \text{ cocatalyst that is required with 156})^{360}$

9.4.3 Monomers and Reaction Conditions

ATRP has been widely used for the polymerization of methacrylates. However, a very wide range of monomers, including most of those amenable to conventional radical polymerization, has been used in ATRP. ATRP has also been used in cyclopolymerization (*e.g.* of $160^{363,364}$) and ring opening polymerization or copolymerization (*e.g.* of $161^{365,366}$ and 162^{367}).

The selection of reaction conditions for ATRP is dependent on many factors including the particular monomer, initiator and catalyst.



9.4.3.1 Solution polymerization

ATRP is usually performed in solution. Many solvents can be used with the proviso that they do not interact adversely with the catalyst. Common solvents include ketones (butanone, acetone) and alcohols (2-propanol). Solvents such as anisole and diphenyl ether are frequently used for polymerizations of S and other less polar monomers to provide greater catalyst solubility.

ATRP of various monomers including HEMA,³⁶⁸ MAA,³⁶⁹ α methoxypoly(ethylene oxide) methacrylate,^{370,371} DMAEMA^{368,372} and 2-(trimethylammonium)ethyl methacrylate salts^{368,373} has been carried out in aqueous media. Rates of ATRP in water can be substantially higher than in organic solvents such that polymerization can be carried out at ambient temperature. This has been attributed to competitive complexation of water and ligand providing a more active catalyst,³⁷⁴ to a higher equilibrium concentration of propagating radicals, to solvent effects on k_p^{371} and to removal of the deactivator by precipitation or hydrolysis. Use of higher reaction temperatures (>60 °C) can lead to loss of catalyst activity.^{371,372,374}

9.4.3.2 Heterogeneous polymerization

ATRP in heterogeneous media has been reviewed by Qiu *et al.*²⁰⁵ Cunningham²⁰⁶ and Schork *et al.*²⁰⁸ and is also mentioned in general reviews on ATRP.²⁶⁸

The Chemistry of Radical Polymerization

Many suspension polymerization recipes have been reported.³⁷⁵ Some of the more successful that yield polymers of low dispersity are for MMA with 146,³⁷⁶ S, BA, MA, tBA and copolymers with 154,³⁷⁷ and BMA with 138.³²¹ Important considerations are a catalyst that is both hydrophobic (to limit partitioning into the aqueous phase) and hydrolytically stable.

Emulsion polymerization has proved more difficult.^{287,288,378} Many of the issues discussed under NMP (Section 9.3.6.6) also apply to ATRP in emulsion. The system is made more complex by both activation and deactivation steps being bimolecular. There is both an activator (Mt^n) and a deactivator (Mt^{n+1}) that may partition into the aqueous phase, although the deactivator is generally more water-soluble than the activator because of its higher oxidation state. Like NMP, successful emulsion ATRP requires conditions where there is no discrete monomer droplet phase and a mechanism to remove excess deactivator built up in the particle phase as a consequence of the persistent radical effect.^{210,214} Reverse ATRP (Section 9.4.1.2) with water soluble dialkyl diazenes is the preferred initiation method.^{287,288}

ATRP polymerization in miniemulsion has recently attracted more attention and met with greater success. Some difficulties with conventional initiation were attributed to catalyst oxidation during the homogenization/sonication step particularly when more active, less oxidatively stable, catalysts are used. This problem was solved using reverse ATRP or combinations of reverse and normal ATRP^{290,291} that meant the catalyst could be added in its oxidized form (Section 9.4.1.2). Better results again were obtained using a conventional ATRP initiation and *in situ* catalyst (CuBr₂/BPMODA) reduction by AGET (Activator Generated by Electron Transfer).³⁷⁹ In this case water soluble ascorbic acid was used as the reducing agent and it was presumed that catalyst reduction occurs in the aqueous phase.

9.5 Reversible Chain Transfer

Radical polymerizations which involve a reversible chain transfer step for chain equilibration and which displayed the characteristics of living polymerizations were first reported in 1995.^{380,381} The mechanism of the reversible chain transfer step may involve homolytic substitution (Scheme 9.35) or addition-fragmentation (RAFT) (Scheme 9.36). An essential feature is that the product of chain transfer is also a chain transfer agent with similar activity to the precursor transfer agent. The process has also been termed degenerate or degenerative chain transfer since the polymeric starting materials and products have equivalent properties and differ only in molecular weight.

Living Radical Polymerization



Scheme 9.35

Polymerization of S and certain fluoro-monomers in the presence of alkyl iodides provided the first example of the reversible homolytic substitution process (Scheme 9.35). This process is also known as iodine transfer polymerization (Section 9.5.4).³⁸¹ Other examples of reversible homolytic substitution are polymerizations conducted in the presence of certain alkyl tellurides or stibines (Sections 9.5.5 and 9.5.6 respectively).

Polymerizations of methacrylic monomers in the presence of methacrylic macromonomers under monomer-starved conditions display many of the characteristics of living polymerization (Scheme 9.36). These systems involve RAFT (Section 9.5.2). However, RAFT with appropriate thiocarbonylthio compounds is the most well known process of this class (Section 9.5.3). It is also the most versatile having been shown to be compatible with most monomer types and a very wide range of reaction conditions.³⁸²



Scheme 9.36

9.5.1 Molecular weights and distributions

As with other forms of living radical polymerization, the degree of polymerization and the molecular weight can be estimated from the concentration of monomer and reagents as shown in eqs. 13 and 14 respectively.³⁸³

$$\overline{X}_{n} = \frac{[M]_{0} - [M]_{t}}{[T]_{0} + df([I_{2}]_{0} - [I_{2}])_{t}}$$
(13)

The Chemistry of Radical Polymerization

$$\overline{M}_{n} = \frac{[M]_{o} - [M]_{t}}{[T]_{o} + df([I_{2}]_{o} - [I_{2}])_{t}} m_{M} + m_{T}$$
(14)

where $m_{\rm M}$ and $m_{\rm T}$ are the molecular weights of the monomer (M) and the transfer agent (T) respectively, d is the number of chains produced in a radical-radical termination event ($d \sim 1.67$ for MMA polymerization and ~ 1.0 for S polymerization) and f is the initiator efficiency. The form of this term in the denominator is suitable for initiators such as AIBN that produce radicals in pairs but will change for other types of initiator.

Reaction conditions should usually be chosen such that the fraction of initiator-derived chains (should be greater than or equal to the number of chains formed by radical-radical termination) is negligible. The expressions for number average degree of polymerization and molecular weight (eqs. 13 and 14) then simplify to eqs. 15 and 16:

$$\overline{X}_{n} = \frac{[\mathbf{M}]_{o} - [\mathbf{M}]_{t}}{[\mathbf{T}]_{o}}$$
(15)

$$\overline{M}_{n} = \frac{[M]_{o} - [M]_{L}}{[T]_{o}} m_{M} + m_{T}$$
(16)

These equations suggest that a plot of M_n vs conversion should be linear. A positive deviation from the line predicted by eq. 16 indicates incomplete usage of transfer agent (T) while a negative deviation indicates that other sources of polymer chains are significant (e.g. the initiator).

Analytical expressions have been derived for calculating dispersities of polymers formed by polymerization with reversible chain transfer. The expression (eq. 17) applies in circumstances where the contributions to the molecular weight distribution by termination between propagating radicals, external initiation, and differential activity of the initial transfer agent are negligible.^{23,384}

$$\frac{\overline{X}_{u}}{\overline{X}_{n}} = 1 + \frac{1}{\overline{X}_{n}} + \left(\frac{2-c}{c}\right)\frac{1}{C_{u}}$$
(17)

where c is the fractional conversion of monomer.

The transfer constant governs the number of propagation steps per activation cycle and should be small for a narrow molecular weight distribution. Rearrangement of eq. 17 to eq. 18 suggests a method of estimating transfer constants on the basis of measurements of the conversion, molecular weight and dispersity.²³

$$\left(\frac{\overline{X}_{w}}{\overline{X}_{u}} - 1 - \frac{1}{\overline{X}_{u}}\right)^{-1} = C_{tr}\left(\frac{c}{2-c}\right)$$
(18)

In more complex cases, kinetic simulation has been used to predict the time/conversion dependence of the polydispersity. Moad *et al.*¹²² first published on kinetic simulation of the RAFT process in 1998. Many papers have now been written on this subject. Zhang and Ray³⁸⁵ and also Wang and Zhu^{386,387} applied a method of moments to obtain molecular weights and dispersities. Peklak *et al.*³⁸⁸ used a coarse-graining approach while Shipp and Matyjaszewski,³⁸⁹ and Barner-Kowollik and coworkers³⁹⁰⁻³⁹³ used a commercial software package (PrediciTM) to evaluate complete molecular weight distributions. Moad *et al.*^{122,384,394} applied a hybrid scheme in which the differential equations are solved directly to give the complete molecular weight distribution to a finite limit ($\bar{X}_n < 500$) and a method of moments is then used to provide closure to the equations, accurate molecular weights and polydispersities. Much of the research in this area has been carried out with a view to understanding the factors that influence retardation. The main difficulty in modeling RAFT lies in choosing values for the various rate constants.

9.5.2 Macromonomer RAFT

Chain transfer to methacrylate and similar macromonomers has been discussed in Section 6.2.3.4. The first papers on the use of this process to achieve some of the characteristics of living polymerization appeared in 1995.³⁸⁰ The structure of macromonomer RAFT agents (163) is shown in Figure 9.3. An idealized reaction scheme for the case of a MMA terminated macromonomer is shown in Scheme 9.36.



Figure 9.3 General description of macromonomer and thiocarbonylthio RAFT agents.

Macromonomer RAFT polymerization is most effective with methacrylate monomers (Table 9.9).^{380,395} With monosubstituted monomers (*e.g.* S, acrylates) graft copolymerization is a significant side reaction which can be mitigated but not eliminated by the use of higher reaction temperatures.

| Macro ^{a,b} | \overline{M}_n | $M_{ m ar}$ | Monomer ^a | Solvent ^a | Temp. °C | \overline{M}_n | M_w |
|----------------------|------------------|------------------|----------------------|------------------------|----------|------------------|------------------|
| | | $\overline{M_n}$ | | | | | $\overline{M_n}$ |
| MAA | 950 | - | MMA | emulsion | 80 | 3000 | 1.4 |
| MMA | 3500 | 1.6 | BMA | emulsion | 80 | 28000 | 1.4 |
| MMA | 2050 | 1.7 | EHMA | emulsion | 80 | 11800 | 1.3 |
| tBMA | 2400 | 2.1 | BMA | emulsion | 80 | 5800 | 1.3 |
| PhMA | 1100 | 2.2 | BMA | emulsion | 80 | 14500 | 2.3 |
| HEMA | 1550 | - | MMA | H ₂ O/iPrOH | 80 | 3600 | 1.8 |
| ΒΜΔ | 1050 | 2.0 | S | BuAc | 125 | 4700 | 2.4° |
| MMA-MAA | 1030 | 1.5 | BA | BuAc | 125 | 2700 | 1.8^{d} |
| 4.1.1 | 34 0 4 | 41 1 4 | 101.1.4.4 | 1 1 1 | ~ | 1 | 10 ATT |

 Table 9.9 Block Copolymers Prepared by Macromonomer RAFT Polymerization under Starved-Feed Conditions.^{380,395}

a Abbreviations: MeS 4-methylstyrene, PhMA phenyl methacrylate, BuAc butyl acetate, iPrOH 2-propanol. Other abbreviations can be found in the Glossary. b Macromonomer 163 made from monomer shown by catalytic chain transfer process. c After subtraction of residual macromonomer. d Contains graft copolymer impurity.

Transfer constants of the macromonomers are typically low (~0.5, Section 6.2.3.4) and it is necessary to use starved feed conditions to achieve low dispersities and to make block copolymers. Best results have been achieved using emulsion polymerization^{380,395} where rates of termination are lowered by compartmentalization effects. A 'one-pot' process where macromonomers were made by catalytic chain transfer was developed.^{380,395} Molecular weights up to 28000 that increase linearly with conversion as predicted by eq. 16, dispersities that decrease with conversion down to $\overline{M}_w/\overline{M}_n$ <1.3 and block purities >90% can be achieved.^{380,395} Surfactant-free emulsion polymerizations were made possible by use of a MAA macromonomer as the initial RAFT agent to create 'self-stabilizing lattices'.

9.5.3 Thiocarbonylthio RAFT

Although the term RAFT (an acronym for Reversible Addition-Fragmentation chain Transfer)³⁸² is sometimes used in a more general sense, it was coined to describe, and is most closely associated with, the reaction when it involves thiocarbonylthio compounds. RAFT polymerization, involving the use of xanthates, is also sometimes called MADIX (Macromolecular Design by Interchange of Xanthate).³⁹⁶ The process has been reviewed by Rizzardo *et al.*,³⁹⁷ Chiefari and Rizzardo,³⁹⁸ Barner-Kowollik *et al.*,³⁹⁹ McCormick *et al.*,⁴⁰⁰ and Moad *et al.*⁴⁰¹

Organic chemists have been aware of reversible addition-fragmentation involving xanthate esters in organic chemistry for some time. It is the basis of the Barton-McCombie process for deoxygenation of alcohols (Scheme 9.37).⁴⁰²⁻⁴⁰⁴



Scheme 9.37 Barton-McCombie deoxygenation reaction

In 1988 a paper by Zard and coworkers⁴⁰⁵ reported that xanthates were a convenient source of alkyl radicals by reversible addition-fragmentation and used the chemistry for the synthesis of a monoadduct to monomer (a maleimide). Many applications of the chemistry in organic synthesis have now been described in papers and reviews by the Zard group.^{406,407}

Living radical polymerization using thiocarbonylthio RAFT agents (including dithioesters, trithiocarbonates and xanthates) was first described in a patent published in 1998.⁴⁰⁸ The first paper describing the process also appeared in 1998.³⁸² Other patents and papers soon followed. Papers on this method, along with NMP and ATRP, now dominate the literature on radical polymerization.

9.5.3.1 Mechanism

A key feature of the mechanism of RAFT polymerization is the sequence of addition-fragmentation equilibria shown in Scheme 9.38.382 Initiation and radicalradical termination occur as in conventional radical polymerization. In the early stages of the polymerization, addition of a propagating radical (P_n) to the thiocarbonylthio compound 164 followed by fragmentation of the intermediate radical 165 gives rise to a polymeric thioearbonylthio compound (166) and a new radical (R'). Reaction of the radical (R') with monomer forms a new propagating radical (P_m). A rapid equilibrium between the active propagating radicals (P_n and P_m) and the dormant polymeric thiocarbonylthic compounds (166) provides equal probability for all chains to grow and allows for the production of narrow dispersity polymers. With appropriate attention to the reaction conditions, the vast majority of chains will retain the thiocarbonylthio end group when the polymerization is complete (or stopped). Radicals are neither formed nor destroyed in the chain equilibration process. Thus once the equilibria are established, rates of polymerization should be similar to those in conventional radical polymerization. This is borne out by experimental data, which show that, with some RAFT agents, RAFT polymerization is half order in initiator and zero order in the RAFT agent over a wide range of initiator and RAFT agent concentrations.

initiation

initiator
$$\longrightarrow$$
 I $\stackrel{\bullet}{\longrightarrow} M \stackrel{\bullet}{\longrightarrow} P_n^{\bullet}$

reversible chain transfer/propagation

| $\left(\begin{array}{c} \mathbf{P}_{n}^{\bullet} + \mathbf{k} \\ \mathbf{M} \end{array} \right)_{k_{n}}$ | S <mark>↓</mark> S-R Z | k _{add} | Pn ^{-S} S-R | $\frac{k_{\beta}}{k_{-\beta}}$ | P _n −S S Z | + R [•] |
|---|---------------------------|------------------|----------------------|--------------------------------|--------------------------------------|------------------|
| F | 164 | | 165 | | 166a | |

reinitiation

$$R^{\bullet} \xrightarrow{M} R^{-}M^{\bullet} \xrightarrow{M} M^{\bullet} P^{\bullet}_{m}$$

reversible (degenerate) chain transfer/propagation

$$\begin{array}{c} P_{m}^{\bullet} + S \\ (M)_{k_{p}} & Z \\ 166a \end{array} \begin{array}{c} P_{m}^{-}S \\ Z \\ 166b \end{array} \begin{array}{c} P_{m}^{-}S \\ Z \\ 166b \end{array} \begin{array}{c} P_{m}^{-}S \\ Z \\ 166b \end{array} \begin{array}{c} P_{m}^{-}S \\ M \\ 166b \end{array}$$

termination

$$P_n^{\bullet} + P_m^{\bullet} \xrightarrow{k_t} \text{dead polymer}$$

Scheme 9.38

For very active RAFT agents, the RAFT agent derived radical (R') may partition between adding to monomer and reacting with the transfer agent (polymeric or initial). In these circumstances, the transfer constant measured according to the Mayo or related methods will appear to be dependent on the transfer agent concentration and on the monomer conversion. A reverse transfer constant can be defined as follows (eq. 19)

$$C_{\rm tr} = \frac{k_{\rm tr}}{k_{\rm fr}} \tag{19}$$

and the rate of RAFT agent consumption is then given by eq. 20.394

$$-\frac{d[150]}{d[M]} \approx C_{\rm tr} \frac{[150]}{[M] + C_{\rm tr}[150] + C_{\rm -tr}[152]}$$
(20)

For addition-fragmentation chain transfer, the rate constants for the forward and reverse reactions are defined as shown in eqs. 21 and 22 respectively.

$$k_{\rm tr} = k_{\rm add} \frac{k_{\beta}}{k_{\rm add} + k_{\beta}} \tag{21}$$

Living Radical Polymerization

$$k_{\rm or} = k_{\beta} \frac{k_{\rm add}}{k_{\rm add} + k_{\beta}}$$
(22)

RAFT polymerization provides the characteristics usually associated with living polymerization. The overall process results in monomer units being inserted into the RAFT agent structure as shown in Scheme 9.38. Expressions (eqs. 13-16) for estimating number average degree of polymerization and molecular weight in RAFT polymerization are provided in section 9.5.1. Dispersities will depend on the chain transfer constants associated with both the initial and the polymeric RAFT agent. The reaction conditions should be chosen such that the initial RAFT agent is rapidly consumed during the initial stages of the polymerization.

9.5.3.2 RAFT agents

Many thiocarbonylthio RAFT agents (164) have now been described. Transfer constants are strongly dependent on the Z and R substituents. For an efficient RAFT polymerization (refer Scheme 9.38 and Figure 9.3):

- (a) Both the initial (164) and polymeric RAFT agents (166) should have a reactive C=S double bond (high k_{add}).
- (b) The intermediate radicals 165 and 167 should fragment rapidly (high k_{β} , weak S-R bond) and give no side reactions.
- (c) The intermediate 165 should partition in favor of products ($k_{\beta} \ge k_{-add}$).
- (d) The expelled radicals (R•) should efficiently re-initiate polymerization.

The dependence of the transfer constant on the Z substituent, summarized in Figure 9.4, is largely based on studies of the apparent transfer constants of benzyl and cyanoisopropyl RAFT agents in S polymerization^{384,409} and qualitative observations of other polymerizations.³⁹⁷





Early reports focused on the dithiobenzoate RAFT agents (Z=Ph; *e.g.* **171-180**, Table 9.10).^{382,410} Cumyl dithiobenzoate (**175**) shows utility with S and (meth)acrylic monomers.³⁸² However, retardation is an issue with the acrylates

and when high concentrations of RAFT agent are used. For MMA and S, cyanoisopropyl dithiobenzoate (176) gives less retardation than 175.⁴⁰⁹ The trithiocarbonates (Z–S-alkyl; *e.g.* 219-232, Table 9.15 and Table 9.16) are also effective with S and (meth)acrylic monomers and give substantially less retardation than the corresponding dithiobenzoates under similar conditions. Dithioacetate and other RAFT agents with Z=alkyl or aralkyl (*e.g.* 212-218, Table 9.14) also give less retardation but have lower transfer constants and do not give narrow dispersities with methacrylates.

The trend in relative effectiveness of RAFT agents with varying Z is rationalized in terms of interaction of Z with the C=S double bond to activate or deactivate that group towards free radical addition. Substituents that facilitate addition generally retard fragmentation. O-Alkyl xanthates (Z=O-alkyl, Table 9.17) are generally not effective with methacrylates and give relatively broad dispersities with S and acrylates. N_1 -dialkyl dithiocarbamates (Z-N-alkyl₂, Table 9.18) are not effective with S and (meth)acrylic monomers. This is rationalized in terms of the importance of zwitterionic canonical forms as shown in Figure 9.5. Substituents which make the lone pair less available for delocalization with the thiocarbonyl group (C=S) activate the RAFT agent.^{384,411-413} Thus, xanthates and dithiocarbamates where the oxygen or nitrogen lone pair is part of an aromatic ring (e.g. where Z is pyrole or imidazole) or possesses an adjacent electronwithdrawing (e.g. C=O) or conjugating (e.g. Ph) substituent are substantially more For examples see Table 9.17 (xanthates) or Table 9.18 effective. (dithiocarbamates). Electron withdrawing substituents also improve the effectiveness (i.e. give polymers with lower dispersity) of dithiobenzoate RAFT agents in MMA polymerization.⁴¹⁴



Figure 9.5 Canonical forms of thiocarbonylthio compounds.

O-Alkyl xanthates and *N*-aryl-*N*-alkyl dithiocarbamates are effective with vinyl acetate.³⁹⁷ Dithioesters and trithiocarbonates give severe retardation or even inhibition which is attributed to slow fragmentation of the adduct radical.

The choice of R substituent is also extremely important in determining the activity of RAFT agents. The radical R• needs to be a good free radical leaving group with respect to the propagating radical. The order of relative effectiveness shown in Figure 9.6 is largely based on studies of the apparent transfer constants of dithiobenzoate RAFT agents in polymerizations of S and MMA.^{394,409} However, the trends appear to be general.

R is made a better leaving group by electrophilic substituents (*e.g.* CN), by groups which stabilize the incipient radical, and by bulky subsituents. Penultimate unit effects are important.^{394,409} Thus, the 2-carboalkoxy-2-propyl radical [(CH₃)₂(CO₂R)C•] is a poor leaving group with respect to PMMA•. The *t*-butyl radical is a poor leaving group with respect to isooctyl radical.



Figure 9.6 Effect of R substituent on effectiveness of RAFT agents **164**. Dashed line implies limited effectiveness with a particular monomer (broad molecular weight distribution or severe retardation).⁴⁰¹

The dependence of RAFT agent activity on the substituents R and Z can be qualitatively predicted using low level molecular orbital calculations and these also provide a guide to the relative importance of the various factors.^{384,394,415} There also appear to be good prospects for more quantitative predictions using higher level *ab initio* and density functional theory (DFT) calculations.^{384,414,416,417} The molecular orbital calculations provide insight into the origin of substituent effects and should prove extremely useful in RAFT agent design. However, this work is still in its infancy and the use of these methods to predict absolute values of rate constants or equilibrium constants associated with RAFT must still be treated with caution.

A non-exhaustive tabulation of RAFT agents and the monomers they have been examined with is provided in Table 9.10-Table 9.18. Listing of a monomer or RAFT agent does not mean that that combination provides good results. Combinations shown in parentheses give less than ideal results (dispersity > 1.4 and/or poor molecular weight control) for the reaction conditions used. Even though many RAFT agents have been described, most polymerizations can be performed with just two RAFT agents: one for styrenic and (meth)acrylic monomers (S, AA, MA, MAA, MMA, NIPAM, DMAM, *etc.*) and another for vinyl monomers (VAc, NVP *etc.*). Specific requirements for end group functionality, architecture, case of RAFT agent synthesis and other considerations may dictate other choices.⁴⁰¹

| RAFT Agent | Monomers ^a | RAFT Agent | Monomers ^a |
|---------------------|--|------------|--|
| s=√S=↓ ⊂N 171 | MMA ⁴¹⁴ | | MMA ^{394,397,408} |
| s= 173 | S ^{397,408} DMAEMA ³⁸² AM ⁴¹⁸ | s | SSO3Na ^{397,408,419} AMPS ^{419,420} |
| s= 175 | S ^{394,397,409} MA ^{394,421-423} BA ^{394,409} (AN) ⁴²⁴ MMA ^{382,383,394,408,425} BzMA ⁴²⁵ DMAEMA ³⁹⁷ XMA ^{426,427} AM ⁴¹⁸ DMAM ^{397,428} NIPAM ^{397,429} 2VP ⁴³⁰ 4VP ⁴³⁰ (S) ^{409,423} MMA ⁴⁰⁸ BMA ^{382,308} | s | S ^{384,394,408,409} AA ³⁹⁷ MA ^{394,431} AN ⁴³² MMA ^{394,397,408,414} XMA ^{426,427,433} VBz ^{382,408} MMA ⁴³⁴ BMA ⁴³⁴ EHMA ⁴³⁴ |
| s 177 | S ^{394,408} (MMA) ^{394,408} | s 178 | (MMA) ³⁹⁷ |
| s→ 179 | S ^{394,408} (MMA) ^{394,397} | s 180 | (MMA) ³⁹⁴ AA ⁴³⁵ |

Table 9.10 Tertiary Dithiobenzoate RAFT Agents

a Abbreviations: AMPS sodium 2-acrylamido-2-methylpropane-1-sulfonate, BMDO 5,6-dibenzo-2-methylene-1,3-dioxepan, SSO₃Na sodium styrene-4-sulfonate, 2VP 2-vinylpyridine, 4VP 4vinylpyridine, XMA functional methacrylate: 2-(acetoacetoxy)ethyl methacrylate;⁴²⁷ 3-[tris(trimethylsilyloxy)silyl]propyl methacrylate;⁴²⁶ 6[4-(4'-methoxyphenyl)phenoxy]hexyl methacrylate.⁴³³ For other monomer abbreviations see Glossary. Monomers shown in parentheses give less than ideal results (dispersity > 1.4 and/or poor molecular weight control and/or marked retardation) for the reaction conditions reported. Monomers in italics were polymerized by emulsion or miniemulsion polymerization.

| RAFT Agent | M ^a | RAFT Agent | M ^a | RAFT Agent | M ^a |
|---|--------------------|---------------------|--|----------------|--------------------|
| | MA ⁴²¹ | | MMA ⁴⁰⁸ | | MMA ⁴⁰⁸ |
| s= | MMA ⁴¹⁴ | | S ⁴³⁶ MMA ^{414,437} GMA ⁴³⁸ | s=s+cn 186 | MMA ⁴³⁹ |
| $S \xrightarrow{F}_{F} CN$ $F \xrightarrow{F}_{F} F$ $I 87$ | MMA ⁴¹⁴ | s CN 188 | MMA ⁴¹⁴ | s= F 189 | MMA ⁴¹⁴ |
| s сі 190 | MMA ⁴¹⁴ | s с∩ осн₃ 191 | MMA ⁴¹⁴ | | |

 Table 9.11
 Other Aromatic Dithioester RAFT Agents

a Monomer. See footnote a of Table 9.10.

| RAFT Agent | Monomers ^a | RAFT Agent | Monomers ^a |
|-------------------------------|--|--|---|
| CH ₃ O S 192 | S, ⁴⁴⁰ MA, ⁴⁴⁰ MMΛ, ⁴⁴⁰ DMΛM ⁴⁴⁰ | (C ₂ H ₅₎₂ N S 193 | S, ⁴⁴¹ MA, ⁴⁴¹ DMAM ⁴⁴¹ |
| s 194 | | s 195 | AN ⁴²⁴ |
| s 196 | BMDO ⁴⁴³ | s 197 | S ⁴¹⁵ BA ⁴¹⁵ (MMA) ⁴¹⁵ |
| S- 198 | DMAM ⁴²⁸ | s= 199 | S ^{384,394,408,425} MA ³⁹⁴ BA ^{382,394} (MMA) ³⁹⁴ DMAM ^{408,428} NIPAM S ^{408,409} <i>MMA</i> ⁴⁰⁹ |
| s→ ^{CO2H} 200 | (S) ^{408,415} SAc ⁴⁴⁴ (MA) ⁴⁰⁸ BΛ ⁴¹⁵ (MMΛ) ⁴¹⁵ | s 201 | (S) ⁴¹⁵ BA ⁴¹⁵ (MMA) ⁴¹⁵ |
| s= 202 | DMAM ⁴²⁸ | s= 203 | MMA ^{394,408} BA ^{394,408} |

 Table 9.12
 Primary and Secondary Dithiobenzoate RAFT Agents

a See footnote a of Table 9.10.



Table 9.13 Bis-RAFT Agents



a Monomer. See footnote a of Table 9.10.

Table 9.14 Dithioacetate and Dithiophenylacetate RAFT Agents



a See footnote a of Table 9.10.

 M^{a}



Table 9.15 Symmetrical Trithiocarbonate RAFT Agents

a Monomers. See footnote a of Table 9.10.

 Table 9.16
 Non-Symmetrical Trithiocarbonate RAFT Agents





Table 9.17 Xanthate RAFT Agents

a Monomers. See footnote a of Table 9.10. b Some reports relate to the corresponding methyl xanthate.
| RAFT Agent | M ^a | RAFT Agent | M ^a | RAFT Agent | M ^a |
|--|---|--|--|---|---|
| | NIPAM ⁴⁷⁴ | s=(N] 255 | S MA ⁴⁶⁹ MMA ³⁸⁴ | s≓ s= 256 | S ^{384,397,469} MA ^{397,411,469} NIPAM ⁴⁷⁴ |
| s= N 257 | S ⁴¹¹ MA ^{411.469} | s s 0 258 | $\frac{S^{412}}{(MMA)^{412}}$ VAc ⁴¹² | $s \neq \begin{bmatrix} CO_2C_2H_5 \\ CO_2C_2H_5 \\ CO_2C_2H_5 \\ 0 \neq \end{bmatrix}$ | EA ⁴¹² |
| s ~ ^S ^{(CN} o ~ 260 | AN ⁴⁰¹ MA ⁴⁰¹ AA ^{475,476} | s o 261 | (S) ^{401,469} MA ^{401,469} | | S ^{401,469} (MA) ^{401,469} |
| $S \xrightarrow{CO_2C_2H_5} S \xrightarrow{CO_2C(CH_3)_3} Z \xrightarrow{CO_2C(CH_3)_3} Z \xrightarrow{CO_2C_2H_5} Z \xrightarrow{CO_2C_2C_2H_5} Z \xrightarrow{CO_2C_2C_2H_5} Z \xrightarrow{CO_2C_2C_2H_5} Z \xrightarrow{CO_2C_2C_2H_5} Z \xrightarrow{CO_2C_2C_2H_5} Z \xrightarrow{CO_2C_2C_2C_2H_5} Z \xrightarrow{CO_2C_2C_2C_2H_5} Z \xrightarrow{CO_2C_2C_2C_2H_5} Z \xrightarrow{CO_2C_2C_2C_2H_5} Z \xrightarrow{CO_2C_2C_2C_2C_2H_5} Z CO_2C_2C_2C_2C_2C_2C_2C_2C_2C_2C_2C_2C_2C_$ | EA ⁴¹² | $\overset{CO_2C_2H_5}{\underset{N \ CO_2C_2H_8}{\overset{CO_2C_2H_8}{\overset{N \ CO_2C_2H_8}{\overset{O}}}}}$ | S ⁴¹² EA ⁴¹² VAc ⁴¹² | $\overbrace{{}}^{CO_2C_2H_5}_{N}$ | EA ⁴¹² (VAc) ⁴¹² |
| s≓ ^{CN} s≓ ^{N−CH} 3 266 | VAc ³⁹⁷ | $s \xrightarrow{N-C_2H_5} CO_2C_2H_5$ | VAc ³⁹⁷ | $s = \underbrace{c_{0_2C_2H_5}}_{\substack{N=C_2H_5\\C_2H_5}} z68$ | (EA) ⁴¹² (VAc) ⁴¹² |
| s c ₂ H ₅ 269 | (S) ^{384,397} | | | | |

Table 9.18 Dithiocarbamate RAFT Agents

a Monomers. See footnote a of Table 9.10.

9.5.3.3 RAFT agent synthesis

Currently, few RAFT agents are commercially available. However, RAFT agents are available in moderate to excellent yields by a variety of methods and syntheses are generally straightforward.

Some of the methods exploited in recent work include:

(a) The reaction of a carbodithioate salt with an alkylating agent.^{384,394,409,415,440,450,477,478} Often this will involve sequential treatment of an anionic species with carbon disulfide and an alkylating agent in a one-pot reaction. For example, the process was used to prepare benzyl dithiobenzoate (199) from phenyl Grignard reagent (Scheme 9.39),³⁸⁴ Yields are lower when this method is used to prepare RAFT agents such as 2-(ethoxycarbonyl)prop-2-yl dithiobenzoate (177)³⁸⁴ and 2-cyanoprop-2-yl dithiobenzoate (176)⁴⁰⁹ from the corresponding tertiary halides.



Scheme 9.39

A similar approach has been used to prepare dithiocarbamates, xanthates and unsymmetrical trithiocarbonates.⁴⁷⁸ Thus, unsymmetrical primary and secondary trithiocarbonates are readily prepared in a 'one pot' reaction by treating a thiol with carbon disulfide in the presence of triethylamine to form a carbotrithioate salt and then adding the appropriate alkylating agent.^{457,478} The process is shown in Scheme 9.40 for **231**.⁴⁶³



Scheme 9.40

(b) Addition of a dithioacid across the double bond of an electron-rich olefin (S, AMS, isooctene and VAc).^{394,409,479,480} This procedure has been used to prepare cumyl dithiobenzoate (175) from AMS (Scheme 9.41)⁴⁰⁹ and isooctyl dithiobenzoate (179) from 2,2,4-trimethylpentene.³⁹⁴



Addition of dithioacids to electron-deficient monomers (MA, MMA, AN) proceeds by Michael addition to put sulfur at the unsubstituted end of the double bond.⁴⁷⁹

(c) Radical-induced decomposition of a bis(thioacyl) disulfide.^{384,450,481-483} This is probably the most used method for the synthesis of RAFT agents requiring tertiary R groups. The method was used in preparation of the unsymmetrical trithiocarbonate **226** (Scheme 9.42).⁴⁵⁰ It is also possible to use this chemistry to generate a RAFT agent *in situ* during polymerization.



Scheme 9.42

(d) Sulfuration of a thiolocster , or a mixture of a carboxylic acid with a halide, olefin, or alcohol, with Lawesson reagent (Scheme 9.43), Davey reagent or $P_4S_{10}^{394,484}$





(c) Radical-induced ester exchange.^{384,394,442,485} For example, the cyanoisopropyl radical generated from AIBN can replace the cumyl group of cumyl dithiobenzoate (Scheme 9.44). For this method to be most effective the R group of the precursor RAFT agent should be a good free radical leaving group with respect to that of the product RAFT agent.

Living Radical Polymerization



9.5.3.4 Side reactions

Various side reactions may complicate RAFT polymerization. Transfer to solvents, monomer and initiator occur as in conventional radical polymerization. Other potential side reactions involve the intermediate radicals 165 and 167. These radicals may couple with another radical (Q•) to form 271 or disproportionate with Q• to form 270. They may also react with oxygen. The intermediate radicals 165 and 167 are not known to add monomer.

Scheme 9.45 (Q• is an initiator-derived radical or a propagating radical)

Retardation is sometimes observed in RAFT polymerizations when high concentrations of RAFT agent are used and/or with inappropriate choice of RAFT agent. Some decrease in polymerization rate is clearly attributable to a mitigation of the gel (or Norrish-Trommsdorf) effect.^{384,394} However, it is also clear that other effects are important.

For example, there is significant retardation in the polymerization of acrylate esters in the presence of dithiobenzoate esters.^{392,394,409,431,486,488} With benzyl dithiobenzoate and cyanoisopropyl dithiobenzoate retardation is observed from the onset of polymerization and is not directly related to consumption of the initial RAFT agent which appears to be extremely rapid.^{394,409,486} The aliphatic dithioesters (*e.g.* dithioacetate, dithiophenylacetate) and trithiocarbonates give substantially less retardation.^{394,409,431,486} Quinn *et al.*⁴⁵³ observed that dithiophenylacetate RAFT agents enable polymerization of acrylates at ambient temperature whereas cumyl dithiobenzoate (175) gives inhibition under these conditions. McCleary *et al.*⁴⁸⁸ used cumyl dithiophenylacetate (212) and cumyl dithiobenzoate and found an inhibition period corresponding to the time taken to consume the RAFT agent. They called this the initialization step and assigned this to slow reinitiation by cumyl radicals. Moad *et al.* attributed the inhibition period seen with cumyl dithioesters not to slow reinitiation by itself, but to the importance of the back reaction of cumyl radicals with the polymeric RAFT agent.⁴⁰¹

Retardation has also been observed in polymerizations of S and methacrylates and is pronounced when high concentrations of dithiobenzoate RAFT agent are

used.^{383,391,409,442,486,489-492} With lower concentrations of RAFT agent, rates of polymerization are little different from those expected in the absence of RAFT agent.^{383,409,486} The extent of retardation is markedly dependent on which initial RAFT agent is used and may be manifested as an inhibition period corresponding to the time taken to convert that RAFT agent to the polymeric RAFT agent.^{383,409,493} Inconsistencies in reported rates of polymerization suggests that, in some cases, lower rates may in part be attributed to extraneous factors such as impurities in the RAFT^{473,494} agent or incomplete degassing.^{401,486}

9.5.3.5 Reaction conditions

RAFT polymerization can be performed simply by adding a chosen quantity of an appropriate RAFT agent to an otherwise conventional radical polymerization. Generally, the same monomers, initiators, solvents and temperatures are used. The only commonly encountered functionalities that appear incompatible with RAFT agents are primary and secondary amines and thiols.

Since radicals are neither formed nor destroyed during reversible chain transfer, RAFT polymerization must, like conventional radical polymerization, be initiated by a source of free radicals as shown in Scheme 9.38. RAFT polymerization is usually carried out with conventional radical initiators. Most often thermal initiators (*e.g.* AIBN, ACP, BPO, $K_2S_2O_8$) are used. S polymerization may be initiated thermally between 100-130°C. Polymerizations initiated with UV irradiation,^{495,496} a gamma source⁴⁹⁷⁻⁵⁰³ or a plasma field⁵⁰⁴ have been reported. In these polymerizations, radicals generated directly from the RAFT agent may be responsible for initiation. It was initially suggested by Pan and coworkers that the mechanism for molecular weight control in UV⁴⁹⁶ and γ -initiated⁵⁰² processes might involve reversible coupling and be similar to that seen with dithiocarbamate photoiniferters (Section 9.3.2). However, Quinn *et al.*^{495,497,498} demonstrated that the living behavior observed in these polymerizations could be attributed to the standard RAFT mechanism (Scheme 9.38).

The RAFT process is compatible with a wide range of reaction media including protic solvents such as alcohols and water^{382,400,419,505-507} and less conventional solvents such as ionic liquids⁵⁰⁸ and supercritical carbon dioxide.^{509,510} Even though RAFT polymerization has been successfully carried out in aqueous media, care should be taken because certain RAFT agents show some hydrolytic sensitivity particularly in alkaline media.^{400,507,511} Rates of hydrolysis depend on R and Z and roughly correlate with RAFT agent activity (*e.g.* dithiobenzoates>trithiocarbonates~aliphatic dithioesters). RAFT agents used in aqueous media include 174, 219 and 228.

There have been no comprehensive studies of the effect of temperature on the course of RAFT polymerization. Temperatures reported for RAFT polymerization range from ambient to 140 °C. There is evidence with dithiobenzoates that at higher temperatures there is less retardation and also data that suggest narrower

molecular weight distributions can be achieved.^{398,453} For MMA polymerization with trithiocarbonate **226** there appears to be no dramatic effect of temperature on the molecular weight distribution achieved at a given conversion (Figure 9.7).⁴⁰¹ It should be noted, however, that higher temperatures do offer higher rates of polymerization and allow a given conversion to be achieved in a shorter reaction time.



Figure 9.7 Evolution of molecular weight (---) and dispersity (---) with conversion for MMA polymerizations in the presence of RAFT agent **226** (0.0112 M) and (a) MMA (7.0 M) with AIBN (0.0061 M) at 60 °C (filled symbols) (b) MMA (6.55 M) with 1,1'-azobis(1-cyclohexanenitrile) (0.0018 M) at 90 °C (open symbols).⁴⁰¹

RAFT polymerizations under very high pressure (5 kbar) have been reported.^{509,512,513} At high pressures, radical-radical termination is slowed and this allows the formation of much higher molecular weight polymers and higher rates of polymerization than are achievable at ambient pressure.

RAFT polymerization can be conducted in the presence of Lewis acids. There are reports of attempts to control the tacticity of homopolymers^{451,514-516} (to enable the synthesis of stereoblock copolymers⁵¹⁷) and the alternating tendency for copolymerizations^{518,519} through the use of Lewis acids as additives. For MMA polymerization, the addition of scandium triflate Sc(OTf)₃ increases the fraction of isotactic triads and enhances the rate of polymerization in conventional radical (Chapter 8) and RAFT processes.^{451,514,516,517} Polymerizations with dithiobenzoate in the presence of Sc(OTf)₃ and with dithiobenzoate RAFT agents 175^{451,514,516,517} or 176⁵¹⁴ gave comparatively poor control over molecular weight and dispersity. NMR studies show⁵¹⁴ that the poor results can be attributed to the Lewis acid causing degradation of the dithiobenzoate group. Polymerizations with the trithiocarbonate RAFT agent 225 provided polymer with narrow molecular weight

distributions and molecular weights as anticipated for the RAFT process, as well as the expected effect on tacticity.⁵¹⁴



Figure 9.8 Comparison of molecular weight distributions for a conventional and RAFT polymerization. Data shown are GPC distributions (upper trace) for PS prepared by thermal polymerization of S at 110°C for 16 h (\overline{M}_n 324000, $\overline{M}_w / \overline{M}_n$ 1.74, 72% conversion) and (lower trace) with cumyl dithiobenzoate (**175**) (0.0029 M) (\overline{M}_n 14400, $\overline{M}_w / \overline{M}_n$ 1.04, 55% conversion).^{401,409}

9.5.3.6 Heterogeneous polymerization

Much has been written on RAFT polymerization under emulsion and miniemulsion conditions. Most work has focused on S polymerization,^{409,520,521} although polymerizations of BA,^{461,522} methacrylates^{382,409} and VAc^{471,472} have also been reported. The first communication on RAFT polymerization briefly mentioned the successful semi-batch emulsion polymerization of BMA with cumyl dithiobenzoate (175) to provide a polymer with a narrow molecular weight distribution.³⁸² Additional examples and discussion of some of the important factors for successful use of RAFT polymerization in emulsion and miniemulsion were provided in a subsequent paper.⁴⁰⁹ Much research has shown that the success in RAFT emulsion polymerization depends strongly on the choice of RAFT agent and polymerization conditions.^{214,409,520-527}

The early emulsion recipes^{382,409} were feed processes in which conversion of monomer to polymer was maintained at a very high level (often > 90%). In a first step a low molecular weight polymeric RAFT agent was prepared *ab initio*. Control during this stage was not always good. However, poor dispersity obtained in this step need not substantially affect control exerted during the later stages of polymerization.

The use of cumyl dithiobenzoate (175) and other dithiobenzoates as RAFT agent either in *ab initio* or in semi-batch emulsion polymerization of S is not recommended.⁴⁰⁹ Much has been written on failings of these systems and they will not be detailed here. Xanthates have also been recommended⁴⁶⁵ over dithioester RAFT agents for *ab initio* batch emulsion polymerization of S because the kinetics more closely approximate those of conventional emulsion polymerization. Substantially better control over S polymerization is also observed with RAFT agents such as trithiocarbonates, dithioacetates and these reagents also offer narrow molecular weight distributions. Dithiobenzoates have been successfully used in RAFT emulsion polymerization of methacrylates to produce low dispersity polymers where again transfer constants are lower.³⁸²

Some of the issues associated with RAFT emulsion polymerization have been attributed to an effect of chain length-dependent termination.⁵²⁸ In conventional emulsion polymerization, most termination is between a long radical and a short radical. For RAFT polymerization at low conversion most chains are short thus the rate of termination is enhanced. Conversely, at high conversion most chains are long and the rate of termination is reduced.

A novel approach to RAFT emulsion polymerization has recently been reported.^{461,529} In a first step, a water-soluble monomer (AA) was polymerized in the aqueous phase to a low degree of polymerization to form a macro RAFT agent. A hydrophobic monomer (BA) was then added under controlled feed to give amphiphilic oligomers that form micelles. These constitute a RAFT-containing seed. Continued controlled feed of hydrophobic monomer may be used to continue the emulsion polymerization. The process appears directly analogous to the 'self-stabilizing lattices' approach previously used in macromonomer RAFT polymerization (Section 9.5.2). Both processes allow emulsion polymerization without added surfactant.

RAFT in miniemulsion has also been reported^{210,409,423,462,530-532} and is more readily used to produce polymers with a narrow molecular weight distribution. Moad *et al.*⁴⁰⁹ used RAFT in miniemulsion to provide narrow dispersity PS in a batch process. Significant retardation was observed with the dithiobenzoate RAFT agent used. However, this is markedly reduced when aliphatic dithioesters⁴²³ or trithiocarbonate RAFT agents are used.⁴⁶² One of the issues with traditional miniemulsion polymerization is the high level of surfactant and co-stabilizer that is typically employed. Pham *et al.*⁴⁶² have recently described surfactant-free miniemulsion polymerization. Amphipathic macro RAFT agents synthesized *in situ* by polymerization of AA were used as the sole stabilizers. This process eliminated secondary nucleation of new particles and lead to a latex with no labile surfactant and good particle size control.

9.5.4 Iodine-Transfer Polymerization

The history of iodine transfer polymerization may be traced back to telomerization experiments carried out in the 1940's.^{26,533} Iodine-transfer

polymerization as a method of living radical polymerization was reported by Tatemoto in 1992.⁵³⁴ The process involves conducting a polymerization with a conventional initiator (AIBN, BPO) in the presence of an activated alkyl iodide. Iodine-transfer polymerization has been used for S, ^{381,535,536} acrylates, ⁵³⁵ VAc ^{537,538} and various fluoro-olefins. ^{534,539} Narrow dispersity PS was not obtained and this can be attributed to the transfer constant (C_{tr} ~3.6 at 80 °C).

Side reactions observed in VAc polymerization include head addition during propagation (Scheme 9.46) (Section 4.3.1.1).⁵³⁸ The primary alkyl iodide (273) is much less effective as a transfer agent than the secondary iodide (272) derived from the normal propagating radical. Thus, formation of 273 constitutes a chain termination reaction. Another side reaction is the formation of an aldehyde end group by acid catalyzed decomposition of end group 272.⁵³⁸ Despite these side reactions relatively narrow dispersities <1.4 are observed for molecular weights less than 20000. Use of higher transfer agent concentrations gives slower rates but better control over polymer dispersity.



Scheme 9.46

9.5.5 Telluride-Mediated Polymerization

Telluride-mediated polymerization (TERP) has been described.^{23,540-542} The importance of chain transfer to the organic chalcogenides Z-X-R where R is a free radical leaving group and Z is an activating group (Figure 9.9) increases in the series where X is O<S<Se<Te. In this series, only the alkyl tellurides appear effective in lending living characteristics to a thermally initiated polymerization. The application of alkyl sulfides and sclenides in photoinitiated polymerizations has already been discussed in Sections 9.3.2 and 9.3.3 respectively. It is believed that these agents control polymerization by a reversible coupling mechanism with the sulfur or selenium-centered radical as the mediating agent. When alkyl tellurides are used as control agents, it is possible that reversible activation/deactivation by reversible coupling and reversible chain transfer mechanisms are simultaneously operative (Scheme 9.47). However, the reversible chain transfer by homolytic substitution appears to be the dominant mechanism . The kinetics and mechanism of radical polymerizations in the presence of the tellurides has been studied by Goto *et al.*^{23,540}

reversible activation/deactivation



reversible chain transfer



Scheme 9.47

Alkyl tellurides appear very effective in controlling thermally initiated polymerization of a very wide range of monomers (Table 9.19).⁵⁴⁰⁻⁵⁴² In the first experiments the telluride was both a thermal initiator and a reversible chain transfer agent. This required reaction temperatures of 80-100 °C.^{541,542} In later work AIBN was used as coinitiator to enable the use of lower reaction temperatures (60 °C).⁵⁴⁰ Narrowest molecular weight distributions are obtained with methyl tellurides **276-278**. The phenyl telluride **280** and the methyl benzyl telluride **279** give poorer control. In polymerization of methacrylates, narrow dispersities are only obtained in the presence of added ditelluride (**274** or **275**).^{540,542} This may reflect the monomeric radical being a much poorer leaving group than the propagating radical as has been seen in RAFT polymerization. Polymerizations can also be carried out with AIBN as initiator in the presence of dimethyl ditelluride (**274**) to form the dormant species *in situ*.⁵⁴³



| Telluride | Monomers | Telluride | Monomers | Telluride | Monomers |
|------------------------|---|--|----------------------------------|-----------|------------------|
| сл 276 | S, BA, (MMA), AN, NIPAM ⁵⁴⁰ | ^{Te} , CH ₃ CO ₂ CH ₃ 277 | MA, tBA, (MMA) ⁵⁴² | | |
| СН ₃ 278 | S ⁵⁴¹ MA, BA, DMAEA, DMAM, AN ⁵⁴² | 279 | S ⁵⁴¹ | | S ⁵⁴¹ |

Table 9.19 Initiators for Telluride-Mediated Polymerization^a

a Dispersities <1.2 except for systems shown in parentheses.



Figure 9.9 General description of organochalcogenide transfer agents

9.5.6 Stibine-Mediated Polymerization

Very recently stibine-mediated polymerization has been reported by Yamago and coworkers^{544,545} The living characteristics are thought to be imparted by a reversible chain transfer mechanism similar to that involved with the tellurides (Section 9.5.5). Thus far only one organostibine transfer agent (281) has been reported.⁵⁴⁴ However, a class of reagents as shown in Figure 9.10 can be envisaged. Narrow molecular weight distributions (dispersity<1.3, with most <1.2) and predictable molecular weights were obtained with a remarkably wide range of monomers including S and (meth)acrylics (BA, MMA, NIPAM and AN) and vinyl monomers (NVP and VAc). Polymerizations were carried out at 60 °C with unusually large concentrations of AIBN (up to 0.5 molar equivalents with respect to 281).



Figure 9.10 General description of organostibine transfer agents

9.6 Living Radical Copolymerization

One of the major advantages of radical polymerization over most other forms of polymerization, (anionic, cationic, coordination) is that statistical copolymers can be prepared from a very wide range of monomer types that can contain various unprotected functionalities. Radical copolymerization and the factors that influence copolymer structure have been discussed in Chapter 7. Copolymerization of macromonomers by NMP, ATRP and RAFT is discussed in Section 9.10.1.

An issue in living radical copolymerization is that the conditions for dormant chain activation can vary substantially according to the particular propagating radical. The problem may be mitigated by two factors.

- (a) In copolymerization the steady state concentration of the propagating radical from the slower propagating monomer at the chain end will be higher than that of the faster propagating monomer. Deactivation events, which proceed at close to diffusion-controlled rates, should preferentially involve the species that is present in highest concentration.
- (b) For many monomer pairs the reactivity ratios are both less than unity and cross propagation is substantially faster than homopropagation.

9.6.1 Reactivity Ratios

Although, there are reports on differences in reactivity ratios observed for conventional radical copolymerization vs living radical copolymerization (ATRP^{275,276,546-548} or RAFT⁵⁴⁸), most research suggests that reactivity ratios are identical^{398,549} and any discrepancies in composition should be attributed to other factors.

In comparing observed reactivity ratios between various polymerization systems, it is important to take into account the possible effect of molecular weight on copolymer composition.^{547,549} In conventional radical copolymerization, the specificity shown in the initiation and termination steps can have a significant effect on the composition of low molecular weight copolymers (usually <10 units). These effects are discussed in Section 7.4.5. In a living polymerization molecular weights are low at low conversion and increase with conversion. In these

circumstances, the overall copolymer composition will also depend on conversion. The usual methods of determining reactivity ratios, which involve the evaluation of copolymer composition or sequence distribution for low conversion samples, are not directly applicable in these circumstances. Either, molecular weights must be sufficiently high for statistical averaging of the composition to take place, or the effects of specificity in initiation and termination steps must be explicitly included in any calculations.

One might also anticipate that the influence of 'bootstrap effects' (Section 8.3.1.2) would be quite different in living and non-living processes.²⁶⁸ A comprehensive study of reactivity ratios in living and conventional radical polymerization may provide a test of the various hypotheses for the origin of this effect.

9.6.2 Gradient Copolymers

Copolymers produced by living polymerization processes differ from those produced by conventional polymerization in one important aspect. Living polymerization processes produce gradient or tapered copolymers. Such copolymers are known from anionic living polymerization.

Disparate reactivity ratios cause unequal rates of monomer consumption and a drift in the composition of the monomer feed with conversion. In conventional radical copolymerization this means that the copolymer macromolecules formed at the beginning of the experiment will be different from those formed at higher monomer conversion; the high conversion product will be a polymer blend. In a living polymerization process, any compositional drift is captured within each chain. Such copolymers will have a blocky character with the degree of blockiness depending on the values of the reactivity ratios and the monomer feed ratio. For example, copolymerization of a 1:0.91 (mole ratio) mixture of MMA and BA (r_{MMA} 1.7 and r_{BA} 0.2) in the presence of cumyl dithiobenzoate (175) provides a narrow polydispersity copolymer with a gradient in composition of [MMA]:[BA] from *ca* 1:0.45 at the initiated end to *ca* 2:1 at the RAFT agent end.^{398,425} The overall composition of the copolymer was the same as that of a copolymer prepared in the absence of 175.

If reactivity ratios are particularly disparate then it is possible to form a block copolymer from a batch polymerization. Thus the copolymerization of MAH with S by NMP⁵⁵⁰ or RAFT^{551,552} with excess S provides P(MAH-*alt*-S)-*block*-PS. There is a similar outcome in other copolymerizations which show a strong alternating tendency such as S with maleimides (*e.g.* NPMI^{204,401}) or AN. The copolymerization of tBA with VAc by RAFT provides P(tBA-*co*-VAc)-*block*-VAc.⁴⁴⁹ Similarly, that of MA with VAc provides P(MA-*co*-VAc)-*block*-VAc.⁴⁰¹ The copolymerization of S with VAc or NVP by NMP is also reported to give a blocky copolymer but the process becomes non-living once the S is exhausted.⁵⁵³

9.6.3 NMP

A generic scheme for nitroxide-mediated copolymerization is shown in Scheme 9.48. The literature through 2001 has been summarized by Davis and Matyjaszewski.554 A non-exhaustive summary of nitroxide-mediated copolymerizations is provided in Table 9.20; most involve S or isoprene (I).

initiation

propagation deactivation termination

 $P_n A + P_n A$ $P_nA + P_nB$ dead polymer $P_n B^{\dagger} + P_n B^{\dagger}$

Scheme 9.48 (For NMP: X—I is an alkoxyamine and X is a nitroxide, A and B are specific monomers, M is any monomer, P_n is a copolymer chain; note that $P_nB\bullet$, $P_nAB\bullet$, $P_nBB\bullet$ and $P_nA\bullet$, $P_nBA\bullet$, $P_nAA\bullet$ are not distinguished)

Monomers not amenable to direct homopolymerization using a particular reagent can sometimes be copolymerized. For example, NMP often fails with methacrylates (e.g. MMA, BMA), yet copolymerizations of these monomers with S are possible even when the monomer mix is predominantly composed of the methacrylate monomer.¹⁵³ This is attributed to the facility of cross propagation and the relatively low steady state concentration of propagating radicals with a terminal MMA (Section 7.4.3.1). MMA can also be copolymerized with S or acrylates at low temperature (60 °C).¹¹¹ Under these conditions, only deactivation of propagating radicals with a terminal MMA unit is reversible, deactivation of chains with a terminal S or acrylate unit is irreversible. Molecular weights should then be controlled by the reactivity ratios and the comonomer concentration rather than by the nitroxide/alkoxyamine concentration.

| Monomers ^a | Nitroxide ^b | Monomers ^a | Nitroxide ^b |
|-----------------------|--|-----------------------|--------------------------|
| S-SMe | 83 ⁵⁵⁵ | I-S | 86 ¹⁵⁴ |
| S-SMeCl | TEMPO ⁵⁵⁶ | I-SMcCl | 86 ¹⁵⁴ |
| S-SAc | 83 ⁵⁵⁷ | I-SAc | 86 ¹⁵⁴ |
| S-SOMe | 69 ⁵⁵⁸ | I-BA | 86 ¹⁵⁴ |
| S-SOBu | 69 ^{194,558} | I-AA | 86 ¹⁵⁴ |
| S-SOCOBu | 86 ¹⁵³ | I-NVP | 86 ¹⁵⁴ |
| S-MMA | TEMPO, ⁵⁵⁶ 86 ¹⁵³ | I-MMA | 86 ¹⁵⁴ |
| S-BMA | TEMPO ^{194,559} | 1-HEMA | 86 ¹⁵⁴ |
| S-MA | TEMPO ⁵⁶⁰ | | |
| S-EA | TEMPO ⁵⁶⁰ | | |
| S-BA | TEMPO, ⁵⁵⁶ 86, ¹⁵³ 89 ⁵⁶¹ | | |
| S-AN | TEMPO ^{132,138,560,562} 61, 63, 64 ¹³⁸ | | |
| S-4VP | TEMPO ⁵⁶³ | | |
| S-VCz | TEMPO ^{193,560,564} | | |
| S-MAH | ТЕМРО ^{с,550} 86 ⁵⁵⁰ | | |

 Table 9.20
 Statistical/Gradient Copolymers Synthesized by NMP

a Abbreviations: SAc 4-acetoxystyrene, SMe 4-methylstyrene, SMeCl 4-chloromethystyrene, SOMe 4-methoxystyrene, SOBu 4-*t*-butoxystyrene, SOCOBu 4-(*t*-butoxycarbonyloxy)styrene, VCz *N*-vinylcarbazole, 4VP 4-vinylpyridine. Other abbreviations can be found in the Glossary. b Nitroxide structures in Table 9.1-Table 9.4. c Poor control/non living behavior observed.

9.6.4 ATRP

Atom transfer radical copolymerization can be described by a scheme similar to that shown in Scheme 9.48 except that bimolecular activation steps must be added (Section 9.4). Copolymerization by ATRP through 2001 has been reviewed by Kelly and Matyjaszewski.⁵⁵⁴ A summary of ATRP copolymerizations appears in Table 9.21.

Lewis acids (diethylaluminum chloride, ethyl aluminum sesquichloride) have been used in conjunction with ATRP to provide greater alternating tendency in S-MMA copolymerization.⁵¹⁹ However, poor control was obtained because of interaction between the catalyst (CuCl/dNbpy) and the Lewis acid. Better results were obtained by RAFT polymerization.⁵¹⁹ Copper catalysts, in particular Cu(II)Br/PMDETA, have been shown to coordinate monomer but this has negligible influence on the outcome of copolymerization.⁵⁶⁵

As with NMP there are examples of copolymerizations providing good control where homopolymerization is unsuccessful. Copolymerization of MA with small amounts of 1-octene is thought to provide control^{283,566} because the propagating radical with a terminal 1-octene unit undergoes rapid cross propagation. It has been established that the ATRP catalyst is unable to efficiently activate the polymeric bromo-compound with a terminal 1-octene unit.⁵⁶⁶

| Monomers ^a | Catalyst/Ligand | Monomers ^a | Catalyst/Ligand |
|-----------------------|--|------------------------|---------------------------|
| S-BA | 133 ^{b.567} | tBA-ODMA | 133 ^{b.568} |
| S-MMA | 146 ⁵⁶⁹ | tBA-ODA | 133 ^{b.568} |
| ΒΛ-ΜΜΛ | 145, ^{c.570} 144 ⁵⁷¹ 140, ⁵⁴⁷ | MMA-BMDO ^{d.} | 140 ³⁶⁶ |
| MA-O | 133 , ²⁸³ 140 ⁵⁶⁶ | MA-NFH | 140 ⁵⁷² |
| S-SAc | 1 32 ⁵⁷³ | MMA-BMA | 138 ⁵⁴⁶ |
| MMA-282 | 140 ³⁶⁷ | MMA-TBAEMA | 138 ²⁷⁵ |
| BA-iB | 133,140 ⁵⁷⁴ | MMA-DEAEMA | 138 ²⁷⁵ |
| S-AN | 132 , ⁵⁷⁵ 140 ⁵⁷⁵ | MMA-DMAEMA | 138 ²⁷⁵ |

Table 9.21 Statistical/Gradient Copolymers Synthesized by ATRP

a Abbreviations: iB isobutylene, BMDO 5,6-benzo-2-methylene-1,3-dioxepane, DEAEMA *N.N*diethylaminoethyl methacrylate, NFH 3,3,4,4,5,5,6,6,6-nonafluoro-1-hexene, O 1-octene, TBAEMA *t*-butylaminoethyl methacrylate. Other abbreviations can be found in the Glossary. b Catalyst formed in situ with CuBr and ligand indicated. c Hybrid catalyst system. d Ringopening copolymerization.

9.6.5 RAFT

The reaction scheme for RAFT copolymerization is relatively complex (Scheme 9.49) when considered alongside that for NMP or ATRP (Scheme 9.48). A summary of RAFT copolymerizations is provided in Table 9.22. An advantage of RAFT over other methods is its greater compatibility with monomers containing protic functionality though as yet few have taken advantage of this in the synthesis of functional copolymers.

RAFT of MMA with benzyl dithiobenzoate provides very poor control³⁹⁴ yet copolymerization of S with MMA with this RAFT agent provides low dispersities with as little as 5% S in the monomer feed.

Table 9.22 Statistical/Gradient Copolymers Synthesized by RAFT Polymerization

| Monomers ^a | RAFT Agent ^b | Monomers | RAFT Agent ^b |
|-----------------------|---|-----------|---------------------------|
| S-MMA | 175 ⁴⁰⁸ 216 ^{c.576} | tBA-VAc | 248 ⁴⁴⁹ |
| S-AN | 175 ^{382,408} | NIPAM-XMA | 176 ⁵⁷⁷ |
| S-MAH | 176 ⁵⁵¹ 199 ^{552,578-580} | AMBS-AMPS | 174 ⁴²⁰ |
| AMS-MAH | 199 ^{d,579} | MMA-HEMA | 175 ^{382,408} |
| | | MMA-BA | 175 ³⁹⁸ |

a Abbreviations: AMBS sodium 2-acrylamido-3-methylbutanoate, AMPS sodium 2-acrylamido-2-methylpropane-1-sulfonate, XMA *N*-hydroxysuccinimide methacrylate.⁵⁷⁷. Other abbreviations can be found in the Glossary. b Structures in Table 9.10-Table 9.18 c Miniemulsion copolymerization. d Poor control. initiation

initiator
$$\longrightarrow$$
 I^{\bullet} I^{\bullet} I^{\bullet} M A $P_{n}A^{\bullet}$
 B IB^{\bullet} M B $P_{n}B^{\bullet}$

reversible chain transfer/propagation

$$\begin{array}{c} (A) \\ P_{n}A^{\bullet} + S \\ B \\ \downarrow A \\ Z \\ \end{array} \begin{array}{c} P_{n}A^{\bullet} + S \\ B \\ \downarrow A \\ Z \end{array} \begin{array}{c} S^{-R} \\ Z \\ \end{array} \begin{array}{c} P_{n}A^{-S} \\ Z \\ \end{array} \begin{array}{c} S^{-R} \\ Z \\ \end{array} \begin{array}{c} P_{n}B^{-S} \\ Z \\ \end{array} \begin{array}{c} S^{-R} \\ Z \\ \end{array} \begin{array}{c} P_{n}B^{-S} \\ Z \\ \end{array} \begin{array}{c} S^{-R} \\ Z \\ \end{array} \begin{array}{c} P_{n}B^{-S} \\ Z \\ \end{array} \begin{array}{c} S^{-R} \\ Z \\ \end{array} \begin{array}{c} P_{n}B^{-S} \\ Z \\ \end{array} \begin{array}{c} S^{-R} \\ Z \\ \end{array} \begin{array}{c} P_{n}B^{-S} \\ Z \\ \end{array} \begin{array}{c} S^{-R} \\ Z \\ \end{array} \begin{array}{c} P_{n}B^{-S} \\ Z \\ \end{array} \begin{array}{c} S^{-R} \\ Z \\ \end{array} \begin{array}{c} P_{n}B^{-S} \\ Z \\ \end{array} \begin{array}{c} S^{-R} \\ Z \\ \end{array} \begin{array}{c} P_{n}B^{-S} \\ Z \\ \end{array} \begin{array}{c} S^{-R} \\ Z \\ \end{array} \begin{array}{c} P_{n}B^{-S} \\ Z \\ \end{array} \begin{array}{c} S^{-R} \\ Z \\ \end{array} \begin{array}{c} P_{n}B^{-S} \\ Z \\ \end{array} \begin{array}{c} S^{-R} \\ Z \\ \end{array}$$

reinitiation

 \sim

$$R^{\bullet} \xrightarrow{A} RA^{\bullet} \xrightarrow{n M} \xrightarrow{A} P_{n}A^{\bullet}$$

$$R^{\bullet} \xrightarrow{B} RB^{\bullet} \xrightarrow{n M} \xrightarrow{B} P_{n}B^{\bullet}$$

reversible chain transfer/propagation

$$\begin{array}{c} (A) \\ P_{n}A^{\bullet} + S + S^{-AP_{n}} & \longrightarrow P_{n}A^{-S} + S^{-AP_{n}} & \longrightarrow P_{n}A^{-S} + P_{n}A^{\bullet} \\ B \downarrow A & Z & Z & Z & Z \\ P_{n}B^{\bullet} + S + S^{-AP_{n}} & \longrightarrow P_{n}B^{-S} + S^{-AP_{n}} & \longrightarrow P_{n}B^{-S} + P_{n}A^{\bullet} \\ B & Z & Z & Z & Z & Z \\ \hline (A) \\ P_{n}A^{\bullet} + S + S^{-BP_{n}} & \longrightarrow P_{n}A^{-S} + S^{-BP_{n}} & \longrightarrow P_{n}A^{-S} + P_{n}B^{\bullet} \\ B \downarrow A & Z & Z & Z & Z & Z \\ P_{n}B^{\bullet} + S + S^{-BP_{n}} & \longrightarrow P_{n}B^{-S} + S^{-BP_{n}} & \longrightarrow P_{n}B^{-S} + P_{n}B^{\bullet} \\ \hline (B) & Z & Z & Z & Z & Z & Z \\ P_{n}B^{\bullet} + S + S^{-BP_{n}} & \longrightarrow P_{n}B^{-S} + S^{-BP_{n}} & \longrightarrow P_{n}B^{-S} + P_{n}B^{\bullet} \\ \hline (B) & Z & Z & Z & Z & Z & Z & Z \\ \hline termination \\ P_{n}A^{\bullet} & + P_{n}B^{\bullet} & \longrightarrow dead polymer \\ P_{n}B^{\bullet} & + P_{n}B^{\bullet} \end{array}$$

Scheme 9.49 (A and B are specific monomers, M is any monomer (A or B), P_n is a copolymer chain; note that P_nB•, P_nAB•, P_nBB• and P_nA•, P_nBA•, P_nAA• are not distinguished)

530

9.7 End-Functional Polymers

Most reviews on living radical polymerization mention the application of these methods in the synthesis of end-functional polymers. In that ideally all chain ends are retained, and no new chains are formed (Section 9.1.2), living polymerization processes are particularly suited to the synthesis of end-functional polymers. Living radical processes are no exception in this regard. We distinguish two main processes for the synthesis of end-functional polymers.

- (a) The α -functionalization approach makes use of a functional initiator (alkoxyamine, halo-compound) or transfer agent (RAFT agent) to generate a functional initiating radical. All chains should then possess this functionality. The level of functionality will be reduced if there are other processes for initiation (*e.g.* thermal initiation in the case of S polymerization at high temperatures) and by reinitiation after chain transfer to monomer, solvent or other species present in the polymerization medium. It may be increased by the incidence of chain termination by combination.
- (b) The ω -functionalization route involves chemical transformation of the dormant chain end in a post-polymerization reaction. It is also possible to introduce ω -functionality by building it in to the nitroxide fragment of an alkoxyamine NMP initiator or the 'Z' activating group of a RAFT agent (164). The level of functionality will generally equate to the fraction of living (dormant) chain ends and will be reduced by chain termination by radical-radical reaction and further reduced by any chain transfer to monomer, solvent or other species present in the polymerization medium.

There are additional factors that may reduce functionality which are specific to the various polymerization processes and the particular chemistries used for end group transformation. These are mentioned in the following sections. This section also details methods for removing dormant chain ends from polymers formed by NMP, ATRP and RAFT. This is sometimes necessary since the dormant chainend often constitutes a "weak link" that can lead to impaired thermal or photochemical stability (Sections 8.2.1 and 8.2.2). Block copolymers, which may be considered as a form of end-functional polymer, and the use of end-functional polymers in the synthesis of block copolymers are considered in Section 9.8. The use of end functional polymers in forming star and graft polymers is dealt with in Sections 9.9.2 and 9.10.3 respectively.

9.7.1 NMP

9.7.1.1 w-Functionalization

Two methods for cleaving the nitroxide functionality from polymers made by NMP are summarized in Table 9.23. Transfer agents such as thiols¹¹¹ or dithiuram disulfides (Scheme 9.50)⁵⁸¹ can be used for end group replacement and lead to the

nitroxide moiety being substituted by a transfer agent-derived group (hydrogenatom or dithiocarbamyl respectively). The reaction shown in Scheme 9.50 is a method for preparing functional dithiocarbamates⁵⁸¹ and might reasonably be applied to synthesize functional RAFT agents allowing conversion between NMP and RAFT polymerization (Section 9.8.2).

Table 9.23 Methods for End Group Transformation of Polymers Formed by NMP

| | Reaction | | Monomer/Nitroxide |
|-----------|--------------------------------------|-----|-----------------------------|
| R | ARSH | ₩H | S/ 83 ¹¹¹ |
| ∽O−Ń R | Zn/CH ₃ CO ₂ H | ₩OH | MA/59 ¹¹¹ |

A method for ω -functionalization involves polymerization in the presence of a comonomer that does not propagate under the reaction conditions. Monomers that have been used include MAH and maleimide derivatives such as NPMI (Scheme 9.51).⁵⁸² In these cases, elimination of hydroxylamine under the reaction conditions provides an unsaturated end group.

When these methodologies involving the use of a non-propagating monomer or a transfer agent are applied *in situ* during polymerization, the comonomer/transfer agent concentration and the respective reactivity ratios or transfer constants control molecular weights.



Scheme 9.51

A side reaction in NMP is loss of nitroxide functionality by thermal elimination. This may occur by disproportionation of the propagating radical with nitroxide or direct elimination of hydroxylamine as discussed in Section 9.3.6.3. In the case of methacrylate polymerization this leaves an unsaturated end group.¹¹¹ The chemistry has also been used to prepare macromonomers from PMMA prepared by ATRP (Section 9.7.2.1).

Heating an alkoxyamine in the presence of another nitroxide provides nitroxide exchange^{111,118,583} and a process for ω -functionalization.⁵⁸⁴ The product distribution will be determined by the relative stability of the alkoxyamines and the excess of nitroxide. Exchange is also observed when two alkoxyamines are heated together.^{585,586}

9.7.1.2 *α-Functionalization*

Functional alkoxyamines used as initiators for NMP include **283-287**. The functional alkoxyamines can be formed *in situ* by use of a functional azo compound or peroxide. NMP has been shown to be compatible with hydroxy, epoxy, amide and tertiary amine groups in the initiator. Carboxylic acid groups can cause problems but may be tolerated in some circumstances.¹⁰⁶



9.7.2 ATRP

The literature on synthesis of end-functional polymers by ATRP through 2000 is discussed in a review by Coessens and Matyjaszewski.⁵⁹² The topic also has coverage in more general reviews on ATRP.^{268,269}

9.7.2.1 *w*-Functionalization

Polymers formed by ATRP should retain a halogen (typically bromine) on the dormant chain end and this is confirmed by analysis for many polymerizations.

Transformation of the end group may be required to confer greater stability or to introduce new functionality. The various methods include reactions with addition-fragmentation transfer agents or non-propagating monomers (Table 9.24) added at the end of the polymerization. The extent of functionalization will depend on the efficiency of the particular reaction. Those with addition-fragmentation chain transfer agents and MAH appear highly effective with yields >95%. Processes involving the less active non-propagating monomers are prone to side reactions.⁵⁹³ An unusual non-propagating 'monomer' is buckminsterfullerene (C₆₀).^{594,595} For example, P(MAA-*block*-DMAEMA)-C₆₀ was prepared from P(MAA-*block*-DMAEMA)-Cl with CuCl/144 in the presence of C₆₀.⁵⁹⁴

 Table 9.24
 Methods for End Group Transformation of Polymers Formed by ATRP by Addition or Addition-Fragmentation.

| | Reaction | Monomer/Catalyst(Ligand) |
|----------------------|---|--|
| | CO ₂ R Br CO ₂ R | MMA/ 138 , ^{596,597} MA/ 132 ⁵⁹⁸ |
| | Ph O ^{-SiMe₃} Ph | MMA/ 146⁵⁹⁹ MMA/138 ^{a,597} PBA/140 ^{a,593} |
| mBr → m • | SnBu ₃ | MA/132 ⁶⁰⁰ |
| | OH Br OH | MA/140 ⁶⁰⁰ PBA/140 ⁵⁹³ |
| | | MA/132 ⁶⁰⁰ |
| | MAH , where the second | MMA/ 138 ^{597,601} |

a 4-trimethylsilyloxy derivative used to give phenoxy functional polymer after deprotection.

Addition of TEMPO post-polymerization to a methacrylate polymerization provides an unsaturated chain end (Scheme 9.52)^{597,599} presumably by disproportionation of the PMMA propagating radical with the nitroxide. For polymers based on monosubstituted monomers (PS,⁶⁰² PBA^{593,602}) the alkoxyamine is formed in high yield. A functional nitroxide (*e.g.* **69**⁵⁹³) can be used to yield an end-functional polymer.

Living Radical Polymerization



The chain end functionality may be reduced by the incidence of various side reactions. In that ATRP is a radical process, we should expect an amount of radical-radical termination consistent with the concentration of propagating radicals and the reaction time. Radical-radical termination cannot be eliminated, however, it can be minimized through choice of polymerization conditions. The incidence of other side reactions depends on the particular initiator, monomer(s) and catalyst used. During the (co)polymerization of S^{603,604} a slow elimination of HBr from the initiator or dormant species occurs to yield an unsaturated end group. The reaction is catalyzed by Cu(II) and limits the molecular weight of PS that can be prepared with high end group functionality to ~10000.^{603,604} For ATRP with Cu(I) and aliphatic amine ligands (*e.g.* 140), chain transfer to the ligand occurs to yield a saturated chain end.³¹²

| | Reaction ^c | Polymer |
|-----|--|---|
| ~~X | Bu ₃ SnH → ∞H | РМА-Br, РМА-Cl, РММА-Br, PS-Br ^{a,605} |
| ≁Br | $\xrightarrow{\text{PPh}_3} \xrightarrow{\text{restruction}} NaN_3 \xrightarrow{\text{restruction}} NaN_3 \xrightarrow{\text{LiAIH}_4} \xrightarrow{\text{restruction}} NaN_3 $ | PMA-Br ⁶⁰⁶ |
| | WNH ₂ | PS-Br ⁶⁰⁷ |
| ≁Br | | PS-Br ⁶⁰⁸ |
| ≁Br | $H_2N \longrightarrow OH H H H H H H H H H H H H H H H H H $ | PMA-Br, PBA-Br, PS-Br, ^{b.593.609-611} |
| мвr | HS OH DABCO ~S OH HO | PBA-Br ⁵⁹³ |
| ∞Br | | PBA-Br ⁶¹² |

 Table 9.25
 End Group Transformations for Polymers Formed by ATRP

a May be carried out as a 'one-pot' reaction. b Various amino-alcohols have been used. c Abbreviations: DABCO, 1,4-diazabicyclo[2.2.2]octane, DBU 1,8-Diazabicyclo[5.4.0]undec-7cne.

The nucleophilic displacement reactions with azide, primary amines, thiols and carboxylate salts are reported to be highly efficient giving high (>95%) yields of the displacement product (Table 9.25). The latter two reactions are carried out in the presence of a base (DBU, DABCO). Radical-induced reduction with tin hydrides is quantitative. The displacement reaction with phenolates,⁶¹³ phosphines,⁶¹⁴ and potassium phthalimide⁶⁰⁸ gives elimination of HBr as a side reaction.

9.7.2.2 α-Functionalization

Initiators containing a wide range of functional groups have been applied in ATRP (*e.g.* **288-313**).²⁶⁸ These include olefin (**293**,⁶¹⁵ **296**,⁶¹⁶ **297**,⁶¹⁵) hydroxy (**298**,⁶¹⁰ **302**,^{611,617,618} **311**,⁶¹³), tertiary amine (**303**,⁶¹⁷ **308**,⁶¹⁷), epoxy (**299**,⁶¹⁵), oxazoline (**312**,⁶¹³), *t*-butyl ester (**300**,⁶¹⁹ precursor to carboxylic acid), amide (**304**,⁶¹⁷ **309**,⁶¹⁷ **310**,⁶¹⁷) and lactone (**295**,⁶¹⁵). Unprotected acid functionality and primary and secondary amine groups are an issue as these groups may interfere with the stability of the metal complex⁶¹⁵ though, with appropriate catalyst/initiator design, even these groups may be tolerated.^{268,619,620} The ATRP process is tolerant of aromatic amine and carboxylic acid groups in initiators **288c**, **288f**,⁶²⁰ **291** and **313**.⁶¹³

N-Bromomethylphthalimide $(294)^{608}$ and 2-bromopropanenitrile $(301)^{621}$ and the tBOC derivative $(305)^{312,313}$ have been used as initiators in the synthesis of PS with primary amine functionality (deprotection involves hydrazinolysis, LiAlH₄ reduction or treatment with CF₃COOH at room temperature respectively). The initiator **306** contains a protected thiol functionality.⁶²²



536

Living Radical Polymerization



Telechelic polymers can be produced by a combination of α - and ω -functionalization⁶¹¹ or by ω -functionalization of a polymer produced using a bisfunctional initiator. Another method is to couple α -functionalized chains. Atom transfer radical coupling (ATRC) has been used to couple α -functional PS-Br made by ATRP and produce telechelic PS (Scheme 9.53).⁶¹⁸ This approach requires an appropriate rate of radical generation and should only be applied to systems where the propagating radicals undergo termination predominantly by combination. The telechelic purity is limited by the ratio of combination to disproportionation (greater than 85:15 in the case of PS• - Section 5.2.2). The technique can be applied to other polymers with the addition of small amounts of S to form propagating radicals with a terminal S *in situ*.⁶¹⁸



Scheme 9.53

9.7.3 RAFT

9.7.3.1 w-Functionalization

The thiocarbonylthio group can be transformed post-polymerization in a variety of ways to produce end-functional polymers or it can be removed. The presence of the thiocarbonylthio groups also means that the polymers synthesized by RAFT polymerization are usually colored and they possess a labile end group that may decompose to produce sometimes odorous byproducts. Even though the color and other issues may be modified by appropriate selection of the initial RAFT agent, these issues have provided further incentive to develop effective methods for treatment of RAFT-synthesized polymer to transform the thiocarbonylthio groups post-polymerization.

It is well known that thiocarbonylthio groups can be transformed into thiols by reaction with nucleophiles that include pyridines, primary and secondary amines, ammonia, other thiols and hydroxide. The kinetics and mechanism of the reaction of compounds containing thiocarbonyl groups with nucleophiles has been reviewed by Castro.⁶²³ They may also be reduced to thiols with hydride reducing agents such as sodium borohydride, lithium aluminum hydride and zinc in acetic acid. The thiocarbonylthio groups in RAFT-synthesized polymers are subject to the same reactions (Table 9.26).³⁸² Oxidation to the disulfide to form an impurity of twice the molecular weight is a complication in aminolysis that can be minimized by careful degassing or through use of dithionite.⁴⁵⁵ RAFT-synthesized thiols have been used to make protein conjugates.^{475,624}

Radical-induced reduction with, for example, tri-*n*-butylstannane can be used to replace the thiocarbonylthio group with hydrogen. Other transfer agents offer the possibility of introducing different functionality by group transfer. The RAFT end group is also light sensitive and can be removed under UV irradiation and it may be oxidized with reagents such as peroxides or sodium hypochlorite.^{382,551}

RAFT end groups are known to be unstable at very high temperatures (>200 °C). Thermal elimination has been used as a means of trithiocarbonate end group removal. For PS^{450,464} direct elimination is observed (Scheme 9.54). For poly(butyl acrylate)⁴⁶⁴ the major product suggests a homolysis/backbiting/ β -scission reaction is involved (Scheme 9.55).



Scheme 9.54



 Table 9.26 Methods for End Group Removal from Polymers Formed by RAFT

 Polymerization

a PAc = polyacenaphthalene, NaPSS = poly(sodium 4-styrenesulfonate). For other abbreviations see Glossary.



Scheme 9.55 (B=CO₂C₄H₉)

9.7.3.2 α-Functionalization

One significant advantage of the RAFT process is its compatibility with a wide range of functionality present in the monomer or the RAFT agent. This makes the technique eminently suitable for the synthesis of end functional polymers by incorporating the functionality into the Z or R groups of the RAFT agent. RAFT agents with unprotected functionality that have been used successfully include: 172, 178, 221 (-OH); 173, 219, 226, 227, 229 (-CO₂H); 174 (CO₂Na).

Polymers with primary or secondary amine functionality cannot be prepared directly by RAFT polymerization; these groups undergo facile reaction with thiocarbonylthic compounds. Such polymers can be prepared indirectly using RAFT agents with latent amine functionality, such as the phthalimido group in RAFT agents (211, 224, 232), which can be subsequently deprotected by hydrazinolysis.⁴⁵⁰

9.8 Block Copolymers

Block copolymers are composed of two or more covalently connected segments of differing composition. The simplest case is an AB diblock, which consists of two segments. These may be extended to form ABA or BAB triblocks and further extended to form higher-order $(AB)_n$ multi blocks. Introduction of a third block type creates ABC triblocks. A wide range of block copolymer architectures is possible including radial or star-blocks and graft copolymers with block copolymer arms. These structures are mentioned in the sections devoted to the synthesis of star and graft copolymers (Sections 9.9 and 9.9.3.2 respectively).

Living polymerization processes immediately lend themselves to block copolymer synthesis and the advent of techniques for living radical polymerization has lead to a massive upsurge in the availability of block copolymers. Block copolymer synthesis forms a significant part of most reviews on living polymerization processes. This section focuses on NMP,¹⁰⁶ ATRP,^{268,270} and RAFT.³⁹⁷ Each of these methods has been adapted to block copolymer synthesis and a substantial part of the literature on each technique relates to block synthesis.

Four processes for block copolymer synthesis can be distinguished.

- (a) sequential addition of monomers to a living chain end (9.8.1).
- (b) batch copolymerization of monomers with disparate reactivity ratios to form a gradient block copolymer (9.6.2).
- (c) use of a functional polymer prepared by another process as an initiator (NMP, ATRP) or transfer agent (RAFT) (9.8.2).
- (d) joining of pre-prepared blocks in a post-polymerization coupling reaction.

Block copolymers have a wide range of applications from surfactants and dispersants to compatibilizers and thermoplastic elastomers and are found in areas as diverse as biomaterials, drug delivery, nanocomposites and electronics. Many applications depend on the propensity of block copolymers to self assemble into micelles and more complex supramolecular structures.⁶²⁹ Any detailed discussion of applications is, however, beyond the scope of this book.

Some comment should be made on block copolymer purities. The usual and often the only method of assessment is GPC. For the usual case, where the molecular weight of the block is 2-5 times higher than that of the precursor, baseline resolution between block and precursor will seldom be obtained unless dispersities are very low. A complicating factor is that in a GPC trace signal intensity is proportional to molecular weight squared. This has the effect of emphasizing the block copolymer with respect to any first block impurity. Tailing to low molecular weight is deemphasized. There are additional issues that relate to the composition dependence of the refractive index and the elution behavior. A

consequence is that simple inspection of a GPC trace may not be a particularly good indicator of block purity and quantitative assessment is problematical.

9.8.1 Direct Diblock Synthesis

The most direct method for synthesizing block copolymers involves the sequential addition of two monomers in a polymerization reaction. Isolation and purification of the first block may sometimes be desirable. An advantage of living radical methods over classical (anionic) polymerization is that the product of polymerization is a dormant polymer that is usually sufficiently stable that it can be isolated and purified before being used in another polymerization process. This is important since a disadvantage of living radical methods is that it is seldom desirable to operate at very high conversion because, irrespective of method, the likelihood of side reactions is high under these conditions.

9.8.1.1 NMP

$$R=O-N \xrightarrow{\text{monomer } A} R \xrightarrow{\text{monomer } B} R \xrightarrow{\text{monomer } B} R \xrightarrow{\text{monomer } B} O-N$$

Scheme 9.56

The process for block synthesis by NMP with sequential monomer addition is shown in Scheme 9.56. Block synthesis is generally subject to the same limitations as polymer synthesis. Optimal conditions for NMP depend strongly on the particular monomer(s) and this should be taken into account when designing syntheses of block copolymers. TEMPO and similar nitroxides are most suited to controlling polymerizations of styrenic monomers and a majority of reported block copolymers prepared by NMP with TEMPO and TEMPO derivatives have a first block and often a second block based on S or a S derivative [*e.g.* PS-*block*-PBA²⁰¹, PS-*block*-P(S-*co*-BMA),⁵⁵⁹ PS-*block*-PBMA,²⁰² PS-*block*-PB,²⁰³ and PS-*block*-PI²⁰³]. S derivatives include **103** (protected 4-aminostyrene),^{630,631} **104**,⁶³² **108**,⁶³⁴ **105**,⁶³⁴ **106**⁶³⁴ and **107**.⁶³⁴ Polymers containing 4-chloromethylstyrene (**108**) and 4-vinylpyridine (**109**) often serve as precursors to other structures.⁶³⁵ However, with the use of other nitroxides, and lower reaction temperatures, a much wider range of block copolymers is possible including PS-*block*-PtBA with nitroxide **89**.⁶³⁶

An issue when making the second (and subsequent) blocks from styrenic monomers is that thermal initiation or an added initiator will provide a homopolymer impurity.

9.8.1.2 ATRP

Although, ATRP appears most suited to polymerization of methacrylate monomers, a very wide range of monomers can and have been used as is

542

illustrated by Table 9.27. A survey of block syntheses by ATRP is provided in the review by Davis and Matyjaszewski.⁵⁵⁴ A general reaction scheme for block synthesis by ATRP is shown in Scheme 9.57.





Optimal conditions for ATRP depend strongly on the particular monomer(s) to be polymerized. This is mainly due to the strong dependence of the activation-deactivation equilibrium constant (K), and hence the rate of initiation, on the type of propagating radical (Section 9.4.1.3). When using monomers of different types, polymer isolation and changes in the catalyst are frequently necessary before making the second block

For example, when using an macroinitiator based on a monosubstituted monomer (*e.g.* PMA-Br, PS-Br) and Cu(I)Br/L catalyst to initiate polymerization of a methacrylate (MMA) the rate of initiation (cross-propagation) is slow with respect to the rate of propagation of the second monomer and reinitiation from the new macroinitiator (PMMA-Br). The result can be a broad or bimodal molecular weight distribution. The process known as halogen exchange can be used to adjust the rate of initiation.⁵⁷¹ This involves use of a Cu(1)Cl/L catalyst such that a less active macroinitiator is formed following propagation (PMMA-Cl) (Scheme 9.58). Several examples of where halogen exchange has been used to prepare low dispersity block copolymers are provided in Table 9.27.



Scheme 9.58

| Macroinitiator ^a | Monomer 2 ^a | Catalyst /Ligand | Solvent | Temp. °C |
|-----------------------------|------------------------|-----------------------|---|----------|
| PMMA-Cl | DMAEMA ⁶³⁷ | CuCl/144 | $o-C_6H_4Cl_2$ | 90 |
| PMA-Br | DMAEMA ⁶³⁷ | CuCl/144 ^b | o-C ₆ H ₄ Cl ₂ | 90 |
| PMMA-Cl | 4VP | CuCl/145 | 2-C ₃ H ₇ OH | 40 |
| P(SAN)-Br | MMA ⁵⁷⁵ | CuCl/132 ^b | butanone | 80 |
| P(SAN)-Br | GA ⁵⁷⁵ | CuBr/132 | anisole | 80 |
| P(SAN)-Br | tBA, BA ⁵⁷⁵ | CuBr/140 | acetone | 60 |

Table 9.27 Diblock Copolymers Prepared by ATRP

a Abbreviations: 4VP 4-vinylpyridine, PSAN P(S-co-AN). b Halogen exchange process used.

9.8.1.3 RAFT

The synthesis of block copolymers by macromonomer RAFT polymerization has been discussed in Section 9.5.2 and examples are provide in Table 9.9. RAFT polymerization with thiocarbonylthic compounds has been used to make a wide variety of block copolymers and examples are provided below in Table 9.28. The process of block formation is shown in Scheme 9.59. Of considerable interest is the ability to make hydrophilic-hydrophobic block copolymers directly with monomers such as AA, DMA, NIPAM and DMAEMA. Doubly hydrophilic blocks have also been prepared.^{476.638} The big advantage of RAFT polymerization is its tolerance of unprotected functionality.

 Table 9.28 Diblock Copolymers Prepared by RAFT Polymerization^a

| Macro-RAFT ^{b,c} | \overline{M}_{n} | \overline{M}_{n} | Monomer ^c | Solvent | Т | \overline{M}_{η} | $\overline{M}_{ m m}$ |
|--|--------------------|--------------------|-----------------------|---------|-------------|-----------------------|-----------------------|
| | | \overline{M}_n | | | $^{\circ}C$ | | \overline{M}_n |
| S-199 ^{408,425} | 20300 | 1.15 | SMe | benzene | 60 | 25400 | 1.19 |
| S-199 ^{408,425} | 20300 | 1.15 | DMAM | benzene | 60 | 43000 | 1.24 |
| S-199 ³⁹⁷ | 13200 | 1.22 | MA | bulk | 60 | 53300 | 1.19 |
| SNHMe ₂ Cl-174 ⁶³⁹ | 6700 | 1.12 | DMAM | water | 80 | 11300 | 1.12 |
| MMA-175 ^{408,425} | 17400 | 1.20 | S | bulk | 60 | 35000 | 1.24 |
| MMA-255 ⁴¹¹ | 6700 | 1.27 | S | bulk | 60 | 25600 | 1.15 |
| MMA-175 ^{408,425} | 3200 | 1.17 | MAA | DMF | 60 | 4700 | 1.18 |
| BzMA-175 ⁴²⁵ | 1800 | 1.13 | DMAEMA | EtAc | 60 | 3500 | 1.06 |
| MA-194 ^{408,425} | 24100 | 1.07 | BA | benzene | 60 | 30900 | 1.10 |
| BA-194 ^{408,425} | 33600 | 1.13 | AA | DMF | 60 | 52000 | 1.19 |
| AA-260 ⁴⁷⁶ | 7900 | 1.19 | NIPAM | CH3OH | 60 | 13600 | - |
| AMPS-174 ⁴²⁰ | 16100 | 1.17 | AMBA | water | 70 | 24200 | 1.10 |
| DMAM-174 ⁶³⁹ | 4900 | 1.17 | SNHMe ₂ Cl | water | 70 | 14900 | 1.17 |

a For other examples and further details of reaction conditions see references cited. b Monomerinitial RAFT agent used in synthesis of Macro-RAFT agent. c Abbreviations: AMBA 3acrylamido-3-methylbutanoate, AMPS 2-acrylamido-3-methylbutanoate, SNHMe₂Cl N,Ndimethylvinylbenzylammonium chloride, SCO₂Na sodium 4-styrenecarboxylate, SSO₃Na sodium 4-styrenesulfonate, EtAc ethyl acetate





Scheme 9.59

In RAFT polymerization, the order of constructing the blocks of a block copolymer can be very important.^{394,425} The propagating radical for the first formed block must be a good homolytic leaving group with respect to that of the second block. For example, in the synthesis of a methacrylate-acrylate or methacrylate-S diblock, the methacrylate block should be prepared first.^{425,442} The S or acrylate propagating radicals are poor leaving groups with respect to methacrylate propagating radicals.

The problem of macro-RAFT agents with low transfer constants is mitigated by use of a starved-feed polymerization protocol to maximize the concentration of [RAFT agent]:monomer. It is then important to use a RAFT agent that gives minimal retardation (*e.g.* a dithioacetate or trithiocarbonate rather than a dithiobenzoate).⁴⁰⁹ Use of emulsion polymerization conditions is also beneficial. This strategy is also used in block copolymer synthesis when using macromonomer RAFT agents (Section 9.5.2)

9.8.2 Transformation Reactions

Many block and graft copolymer syntheses involving 'transformation reactions' have been described. These involve preparation of polymeric species by a mechanism that leaves a terminal functionality that allows polymerization to be continued by another mechanism. Such processes are discussed in Section 7.6.2 for cases where one of the steps involves conventional radical polymerization. In this section, we consider cases where at least one of the steps involves living radical polymerization. Numerous examples of converting a preformed end-functional polymer to a macroinitiator for NMP or ATRP or a macro-RAFT agent have been reported.⁵⁵⁴ The overall process, when it involves RAFT polymerization, is shown in Scheme 9.60.



Scheme 9.60

The alternative strategy of using a polymer prepared by one of the living radical methods as a precursor to using another (non-radical) polymerization technique is also frequently encountered. Methods for synthesizing the end-functional polymers used in such experiments are described in Section 9.7. Techniques for the interconversion of halo end-groups (formed by ATRP) alkoxyamine end-groups (formed by NMP) and thiocarbonylthio groups (formed

by RAFT polymerization) have also been devised and are also mentioned in Section 9.7. This enables one living radical method to be followed by another so that use can be made of the beneficial features of each method. In all cases triblocks by use of a bis-functional precursor and stars and grafts can be prepared from precursors with a greater number of functional groups (Sections 9.9 and 9.10).

9.8.2.1 Second step NMP

Commercial end functional polymers have been converted to alkoxyamines and used to prepare PEO-*block*-PS.⁶⁴⁰ The hydroxyl group of alkoxyamine **284** was used to initiate ring-opening polymerization of caprolactone catalyzed by aluminum tris(isopropoxide) and the product subsequently was used to initiate S polymerization by NMP thus forming polycaprolactone-*block*-PS.⁶⁴¹ The alternate strategy of forming PS by NMP and using the hydroxyl chain end of the product to initiate polymerization of caprolactone was also used.

Kobetaki *et al.*^{589,642} have examined the combination of conventional free radical and NMP to prepare PBMA-*block*-PS and the combination of anionic and NMP to prepare PB-*block*-PS.

Other block copolymers prepared using similar strategies include PEO (anionic) with second block poly(4-vinylpyridine).⁶⁴³

4-hydroxyTEMPO (69) has been used to initiate polymerization of caprolactone *via* the hydroxy group and the polymeric nitroxide formed used in NMP to give polycaprolactone-*block*-poly(4-vinylpyridine).⁶⁴⁴ The polymerization process can be described by Scheme 9.61.



Scheme 9.61

9.8.2.2 Second step ATRP

Many examples exist where a polymerization has been continued by ATRP.⁵⁵⁴ Often the procedure involves functionalization of a hydroxy-terminated polymer with bromoisobutyroyl (BriBBr, **314**) or bromoisopropionoyl (BriPBr, **315**) bromide. Examples include poly(ethylene oxide)^{645,646} and poly(propylene oxide).⁶⁴⁶



Poly(dimethyl siloxane) with vinyl or hydrosilane (Si-H) chain ends have been converted to ATRP initiator ends (*e.g.* Scheme 9.62) by hydrosilylation. Bis-functional dimethyl siloxane polymers prepared in this way were used in polymerizations of S, MA, isobornyl acrylate and BA to form ABA triblock copolymers.



Scheme 9.62

Ring-opening metathesis polymerization (ROMP) of 1,4-cyclooctadiene was used to prepare poly(1,4-B) terminated with halo end groups.⁶⁴⁷ This was then used as a macroinitiator of ATRP with heterogeneous Cu bpy catalysts to form PSblock-poly(1,4-B)-block-PS and PMMA-block-poly(1,4-B)-block-PMMA.

Polymers prepared with the trichloromethyl-functional initiators⁶⁴⁸ or with chloroform or carbon tetrachloride as a transfer agent⁶⁴⁹ have been used as macroinitiators for ATRP. The method has been used to make PVAc-*block*-PS.^{649,650}

9.8.2.3 Second step RAFT

RAFT polymerization has been used to prepare poly(ethylene oxide)-*block*-PS from commercially available hydroxy end-functional poly(ethylene oxide).^{425,449} Other block copolymers that have been prepared using similar strategies include poly(ethylene-*co*-butylene)-*block*-poly(S-*co*-MAH),⁵⁵¹ poly(ethylene oxide)-*block*-poly(MMA),⁴⁴⁰ poly(ethylene oxide)-*block*-poly(N-vinyl formamide),⁶⁵¹ poly(ethylene oxide)-*block*-poly(NIPAM),⁶⁵² poly(ethylene oxide)-*block*-poly(1,1,2,2-tetrahydroperfluorodecyl acrylate),⁶⁵³ poly(lactic acid)-*block*-poly(MMA)⁴⁴⁰ and poly(lactic acid)-*block*-poly(NIPAM).

Low molecular weight or polymeric ATRP initiators have been converted to dithiobenzoate RAFT agents by reaction with phenylethyl dithiobenzoate RAFT agent^{442,655} or by reaction with bis(thiobenzoyl) disulfides under ATRP conditions.⁴⁸³ It is likely that ATRP initiators can be transformed to other forms of RAFT agent by similar methods.

9.8.3 Triblock Copolymers

Triblock copolymers can be prepared from diblock copolymers by a third monomer addition. They can also be prepared using a bis-functional NMP or ATRP initiator or a bis-RAFT agent (for examples, see Table 9.13). Symmetrical trithiocarbonates (Table 9.15) should also be considered as bis-RAFT agents in

546

this context. For NMP and RAFT there are two limiting strategies for triblock synthesis that lead to the dormant group being in the center or at the ends of the triblock copolymer (shown in Scheme 9.63 and Scheme 9.64 respectively for the case of RAFT polymerization). The two methods are then subject to the same limitations as star polymer synthesis (triblocks may be considered as two arm stars) and these are discussed in Section 9.9.



9.8.4 Segmented Block Copolymers

NMP and RAFT polymerization can be used to prepare segmented or multiblock copolymers directly. Polymer with in-chain alkoxyamine functionality such as **316** or **317** can be heated in S to form segmented block copolymers containing PS blocks by NMP.^{636,657} Heating a mixture of the polyester (**316**) and polyurethane (**317**) provides a polymer containing novel polyester-urethane units (**318**) by a chain reorganization involving alkoxyamine exchange.^{585,586} The exchange process can be followed by NMR.



Scheme 9.65

Multi-RAFT agents have also been used to prepare segmented block copolymers.^{448,658-660} The molecular weight distributions obtained in these experiments are broad when compared to those obtained using analogous mono- or bis-RAFT agents.



Segmented copolymers can also be prepared by polymerization in the presence of appropriate cyclic trithiocarbonates as RAFT agents.⁶⁶¹

9.9 Star Polymers



Figure 9.11 Star Architectures

Several basic approaches to star polymer synthesis leading to architectures as shown in Figure 9.11 will be considered in this section.

- (a) The core first approach to star copolymers requires a core containing the appropriate functionality such that the arms can be grown from the core (Section 9.9.1). The number of arms is dictated by the functionality of the core.
- (b) The arm-first approach where the arms are grown then aggregated to form a star (*e.g.* **322**, Section 9.9.2).
- (c) Self-condensing vinyl polymerization to provide a hyperbranched polymer (Section 9.9.3.1).
- (d) The synthesis of dendritic polymers (e.g. 323) by an iterative approach (Section 9.9.3.2).

The generic features of these approaches are known from experience in anionic polymerization. However, radical polymerization brings some issues and some advantages. Combinations of strategies (a-d) are also known. Following star formation and with appropriate experimental design to ensure dormant chain end functionality is retained, the arms may be chain extended to give star block copolymers (321). In other cases the dormant functionality can be retained in the core in a manner that allows synthesis of mikto-arm stars (324).

A comment should be made on the dispersity of star polymers. If the arms each have a 'most probable' distribution $(\overline{M}_w/\overline{M}_n-2)$, dispersity of the star polymers is expected to be ~ 1+1/a, where a is the number of arms of the star polymer, simply as a consequence of statistical averaging.⁶⁶² This explains why polymers formed by conventional radical polymerization with termination by combination (*i.e.* 2 arms) have $\overline{M}_w/\overline{M}_n = 1.5$. When we additionally take into account the fact that living polymerizations are capable of producing arms of much lower dispersity, we should anticipate that low dispersities are the norm for multi-armed star polymers.

9.9.1 Core-first Star Synthesis

The possibility of attaching appropriate functionality to a multifunctional core to grow star and dendritic polymers was recognized and evaluated early during the development of each form of living radical polymerization. Thus, Ostu *et al.*⁴⁵ used the tetrakis(dithiocarbamate) photoiniferter **19** to form a four armed star (Section 9.3.2.2). A partially soluble product indicating some crosslinking was observed.

Precursors for stars by NMP (Table 9.29), ATRP (Table 9.30) and RAFT (Table 9.31) are shown below. Hawker *et al.*⁶⁶³ used NMP with **325** to form a three-armed star. Matyjaszewski *et al.*⁶⁶⁴ used ATRP with **328** to form a six-arm star. Chen *et al.*⁶⁶⁵ formed a six-arm star based on the organometallic RAFT agent **337**. Barner *et al.*⁶⁶⁶ prepared crosslinked poly(divinylbenzene) microspheres by precipitation polymerization in the presence of phenylethyl dithiobenzoate (**194**) and used these to form particles with dithiobenzoate terminated PS chains. There exist numerous other examples which make use of NMP, ATRP,⁶⁶⁷⁻⁶⁷² RAFT and other techniques using a wide range of cores. The core may be organic, inorganic or organometallic, it may be a dendrimer(ATRP,^{673,674} RAFT^{675,677}), a hyperbranched polymer (RAFT⁶⁷⁸), a (poly)saccharide (ATRP,^{670,679,680} RAFT^{681,682}, a polymer particle (NMP,^{683,684} ATRP,^{685,686} RAFT⁶⁶⁶), a macromolecular species, or indeed, any moiety possessing multiple thiocarbonylthio groups (though here the distinction between star and graft copolymers becomes blurred; graft copolymers are discussed further in Section 9.10).
Table 9.29 Star Precursors for NMP



 Table 9.30
 Star Precursors for ATRP



The method of polymerization needs to be chosen for compatibility with functionality in the cores and the monomers to be used. Star block copolymers have also been reported. Multi(bromo-compounds) may be used directly as ATRP initiators or they can be converted to RAFT agents. One of the most common

methods of core synthesis involves functionalization of an appropriate polyhydroxy compound. $^{478,669}_{\rm }$

For the case of NMP and RAFT, there exist two basic ways of growing star copolymers (this discussion also applies to block and graft copolymer synthesis).

(a) In the first approach, the polymer chains remain directly attached to the core and chain grow occurs at the periphery (Scheme 9.66). Examples of precursors are 325 (NMP) and 333-338 (RAFT). ATRP star syntheses with halo-compound initiators will always involve this approach.

| Agent | Structure | Agent | Structure |
|-------------------------------|--|---------------------------|---|
| 333 ⁴⁴⁵ | | 334 ⁴²⁵ | |
| 335 ^{425,689} | | 336 ⁶⁹⁰ | $\begin{array}{c} \begin{array}{c} \begin{array}{c} \begin{array}{c} \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \end{array} \\ \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} \\ \end{array} \\ \end{array} \\ \begin{array}{c} \end{array} \\ \end{array} $ |
| 337 ⁶⁶⁵ | | 338 ⁶²⁶ | |
| 339 ⁶²⁶ | $\left(\begin{array}{c} & & & \\ & & \\ & & \\ & & \\ & & \\ & & \\ & & \\ & & \\ & & \\ \end{array} \right) \xrightarrow{s} \xrightarrow{s} \xrightarrow{s} \xrightarrow{s} \xrightarrow{s} \xrightarrow{s} \xrightarrow{s} \xrightarrow{s}$ | | |

 Table 9.31
 Star Precursors for RAFT Polymerization



Scheme 9.66

(b) In the second approach, the polymer chains dissociate from the core during each activation-deactivation cycle and grow as linear chains (Scheme 9.67). An example of a precursor is **339** (RAFT).



Scheme 9.67

The two strategies for star synthesis each have advantages and limitations. Star-star coupling only occurs with strategy method (a). The propagating radicals remain attached to the core as shown in Scheme 9.68 for the case of a RAFT polymerization and an example is shown in Figure 9.12a.⁶²⁶



Scheme 9.68 (Q• is an initiator-derived or a propagating radical)



Figure 9.12. GPC distributions obtained during bulk thermal polymerization of S at 110 °C (a) with tetrafunctional RAFT agent **339** (0.0074M) at 6 h, 25% conversion (----), $\overline{M}_n = 25550$, $\overline{M}_w / \overline{M}_n = 1.2$; at 20h, 63% conversion (---), $\overline{M}_n = 63850$, $\overline{M}_w / \overline{M}_n = 1.1$; at 64 h, 96% conversion (----), $\overline{M}_n = 92100$, $\overline{M}_w / \overline{M}_n = 1.2$) and (b) tetrafunctional RAFT agent **338** (0.0074M) at 6 h, 24% conversion (----), $\overline{M}_n = 24300$, $\overline{M}_w / \overline{M}_n = 1.1$; at 48 h, 96% conversion(----), $\overline{M}_n = 70700$, $\overline{M}_w / \overline{M}_n = 1.2$).

In method (b) the propagating radicals are never attached to the core. Star-star coupling by combination of propagating radicals is not possible. The process is illustrated in Scheme 9.69 for RAFT polymerization and an example is shown in Figure 9.12b.⁶²⁶ Termination products are from arm-arm reaction and are always of lower molecular weight than the star. A potential disadvantage of strategy (b) is that the products are intrinsically unstable because there is a weak C-ON (NMP) or C-S (RAFT) bond attaching the polymer chains to the core. This may be used to advantage in some applications including polymer-supported synthesis.

It has been suggested that because the RAFT functionality remains at the core with method (b) at higher conversions as the arms grow longer they may shield the RAFT functionality from the propagating radicals and chain growth may be limited (PVAc).⁶⁹¹ Studies by Mayadunne *et al.*⁶²⁶ for the case of a 4-armed star based on precursor **339** found excellent agreement between found and calculated arm lengths to high conversion and suggest that this limit is not reached with an arm molecular weight \overline{M}_n 30000 (PMA) or \overline{M}_n 18750 (PS).

The Chemistry of Radical Polymerization



Scheme 9.69 (Q• is an initiator-derived or a propagating radical)

9.9.2 Arm-first Star Synthesis

In the arm-first approach the arms are prepared and then self-assembled to form the core. There are two main variants that will be considered.

(a) In-situ microgel formation by polymerization or copolymerization of a nonconjugated diene or a divinyl benzene initiated by an ATRP or a NMP macroinitiator, or carried out in the presence of a macroRAFT agent (Scheme 9.70).



(b) Self-assembly of diblock copolymers to form a micelle that is then crosslinked to form a stable structure. Core-crosslinked micelles (Scheme 9.71) and shell cross-linked micelles and other variants have been reported.



Scheme 9.71

554

The arm-first synthesis of star microgels by initiating polymerization or copolymerization of a divinyl monomer such as divinylbenzene or a bis-maleimide with a polystyryl alkoxyamine was pioneered by Solomon and coworkers.^{692,693} The general approach had previously been used in anionic polymerization. The method has now been exploited in conjunction with NMP,⁶⁹²⁻⁶⁹⁶ ATRP⁶⁹⁷⁻⁷⁰⁰ and RAFT.^{449,701,702} The product contains dormant functionality in the core. This can be used as a core for subsequent polymerization of a monoene monomer to yield a mikto-arm star (NMP,⁷⁰³ ATRP⁷⁰⁴).

The shell-crosslinking of self assembled micelles based on block copolymers made by NMP or ATRP has been exploited extensively by Wooley and coworkers⁷⁰⁵⁻⁷⁰⁷ and others⁷⁰⁸⁻⁷¹⁰ to make a variety of structures. RAFT has also been used both in this context^{449,654} and to make core crosslinked structures.^{420,449} A difficulty with this route to microgels is that the crosslinking step must typically be carried out in very dilute solution to avoid network formation and gelation. Armes and coworkers^{709,710} have found that this problem is mitigated when crosslinking the central block of micelles formed from ABC triblocks.

9.9.3 Hyperbranched Polymers

Hyperbranched or dendritic polymers have recently attracted significant attention.⁷¹¹⁻⁷¹⁴ The possibility of generating highly branched soluble polymers by polymerization of AB_x monomers was first brought to the attention of the scientific community by Flory⁷¹⁵ in a theoretical paper. Early work in this field focused on the synthesis of dendrimers by iterative approaches. These methods suffer practical disadvantages in that the syntheses are both labor and purification intensive. As a consequence, other more viable routes to the generation of branched polymers have been sought, leading to the formation of hyperbranched polymers that are polydisperse systems both in terms of molecular weight and branching distribution. These include the self-condensing vinyl polymerization of so-called AB* monomers such as **340-343** which contain monomer and initiator functionality in the one molecule.

9.9.3.1 Self-condensing vinyl polymerization

With appropriate choice of reaction conditions, hyperbranched polymers can be formed by self-condensing vinyl polymerization of monomers that additionally contain the appropriate initiator (NMP, ATRP), when the compounds are called inimers, or RAFT agent functionality. Monomers used in this process include **340**,⁷¹⁶ **341**⁷¹⁷ and **342**⁶⁴⁰ (for NMP), **108**^{718,719} and **344** and related monomers⁷²⁰⁻⁷²³ (for ATRP) and **343**⁴⁰⁸ (for RAFT). Careful control of reaction conditions is required to avoid network formation.



9.9.3.2 Dendritic polymers

The use of dendritic cores in star polymer synthesis by NMP, ATRP and RAFT polymerization was mentioned in Section 9.9.1. In this section we describe the synthesis of multi-generation dendritic polymers by an iterative approach. Percec *et al.*^{688,724} developed what they termed the TERMINI approach to

Percec *et al.*^{688,724} developed what they termed the TERMINI approach to dendritic polymers. TERMINI is an acronym for TERminator Multifunctional INItiator. Polymerization of MMA by ATRP initiated by **332** with CuO/bpy catalyst provided a three armed star in the first generation which was multiplied by the TERMINI process with **345** to provide six and twelve arms in the second and third generation respectively. The first few steps of the process are shown in Scheme 9.72. The TERMINI agent **345** is an addition-fragmentation chain transfer agent. The thiocarbamate groups are converted to sulfonyl chloride groups to initiate further ATRP by treatment with chlorine in formic acid.



Living Radical Polymerization



Scheme 9.72 ($M = CO_2CH_3$, TERMINI agent is 345)

The first steps of a second process for divergent synthesis of dendritic polymers by ATRP are shown in Scheme 9.73.⁷²⁵ In this case, a calixarene core was used.



Scheme 9.73 (calix = 4-*t*-butylcalix[6]arene, BriBBr = bromoisobutyroyl bromide (314))

558 The Chemistry of Radical Polymerization

9.10 Graft Copolymers/Polymer Brushes

Graft polymerizations involving living radical procedures use the same basic approaches as have been described for conventional radical polymerization (Section 7.6). Thus we consider in turn graft copolymer synthesis by "grafting through" - the copolymerization of macromonomers, grafting from - the use of macroinitiators, and grafting to - the attachment of functional polymers to a surface. In this section, as in the preceding section on block copolymers, there is a focus on NMP, ATRP and RAFT though most of the other methods mentioned in this chapter can and have been explored with reference to the synthesis of graft copolymers.

Graft copolymers made by living polymerization processes are often called polymer brushes because of the uniformity in graft length that is possible. The basic approaches to graft copolymers also have some analogies with those used in making block and star copolymers.

9.10.1 Grafting Through - Copolymerization of Macromonomers

The 'grafting through' approach involves copolymerization of macromonomers. NMP, ATRP and RAFT have each been used in this context. The polymerizations are subject to the same constraints as conventional radical polymerizations that involve macromonomers (Section 7.6.5). However, living radical copolymerization offers greater product uniformity and the possibility of blocks, gradients and other architectures.

NMP of S with **346** as initiator gave PS with pendant oxazoline groups. Cationic polymerization of this macromonomer gave a polyoxazoline with PS grafts that retained the alkoxyamine functionality.⁷²⁶ ATRP with **156** as catalyst and ethyl 2-bromoisobutyrate (**125a**) as initiator has been used to prepare terpolymers of MMA, HEMA and **349**.^{727,728} The terpolymer was then used to initiate ring-opening coordination polymerization of caprolactone or (*L*,*L*)-lactide with stannous octoate catalyst.





ATRP has also been used to synthesize macromonomers subsequently used to make graft copolymers by conventional radical polymerization. Thus, low molecular weight PBA formed by ATRP was converted in near quantitative yield to the methacrylate ester (351) or the corresponding acrylate ester.⁶¹²

There have been several studies on the use of RAFT to form polymer brushes by polymerization or copolymerization of macromonomers **348-350**.^{548,729-735} Systems examined include copolymerizations of **349** with MMA using RAFT agent **175**,^{548,731} **348** with AA using **220**⁷³⁰ and **348** with **352** using **176**.⁷³⁴ The latter copolymerization created a precursor for a grafting from reaction by ATRP using the bromoisobutyrate group as an initiator.⁷³⁴

A number of reviews on dendronized polymers and their synthesis by various methods including radical polymerization of dendron macromonomers have appeared.⁷³⁶⁻⁷³⁸ This macromonomer strategy for the synthesis of dendronized polymers is seen to have an advantage over other strategies that involve a postpolymerization reaction to attach dendrons to a polymer chain. The polymerization of dendron macromonomers when designed with a propensity to self assemble can show some living characteristics even when polymerized with a conventional initiator such as AIBN (Section 8.3.7). The steric demand of the dendrons has a large effect on the polymerization kinetics and the rate of polymerization is extremely sensitive to the monomer concentration. The polymerization of dendron macromonomers by $\Lambda TRP^{739,740}$ (*e.g.* 355-357 with CuBr/145 catalyst and 124a as initiator)⁷⁴¹ and RAFT polymerization (353 and 354 with RAFT agent 176)⁷²⁹ has been described.



9.10.2 Grafting From - Surface Initiated Polymerization

The preparation of polymer brushes by controlled radical polymerization from appropriately functionalized polymer chains, surfaces or particles by a grafting from approach has recently attracted a lot of attention.^{742,743} The advantages of growing a polymer brush directly on a surface include well-defined grafts, when the polymerization kinetics exhibit living character, and stability due to covalent attachment of the polymer chains to the surface. Most work has used ATRP or NMP, though papers on the use of RAFT polymerization in this context also have begun to appear.

Several routes have been reported for preparation of the required functional polymer/surface. Most methods are analogous to techniques used to form grafts by conventional radical polymerization (Section 7.6). However the living processes allow control over graft length, architecture and composition and a means of avoiding or limiting the concomitant formation of non-grafted polymer.

9.10.2.1 Grafting from polymer surfaces

Several techniques have been applied in attaching the appropriate functionality to the polymer surface. For example, copolymerization of a monomer containing functionality (alkoxyamine *e.g.* **358** or **359**,⁷⁴⁴ ATRP initiator, *e.g.* **352**,⁷³⁴ RAFT

agent, *e.g.* **360**) appropriate for producing a copolymer with pendant groups that subsequently can be used to initiate graft copolymerization. The copolymerization is carried out under conditions where said functionality is inert, otherwise the likely product is a hyperbranched polymer or a network as discussed in Section 9.9.3.1. These conditions correspond to the use of low reaction temperatures in the case of alkoxyamines **358** or **359**, the absence of catalyst in the case of ATRP initiator **352**, or a copolymerization with MMA where transfer is negligible in the case of RAFT agent **360**.

The monomer **359** has been formed *in situ* by decomposing the initiator (AIBN) in the presence of the corresponding nitroxide in a solution of S or 2-ethoxyethyl acrylate.⁷⁴⁴ The kinetics dictate that alkoxyamine formation, by coupling of the nitroxide with cyanoisopropyl radicals, will take place before copolymerization.



A second approach is to graft functionality onto a pre-existing polymer by generating radicals on the polymer surface in the presence of a nitroxide or a RAFT agent. Radicals may be formed on the surface by a number of methods including abstraction by radicals generated from a peroxide, decomposition of initiator groups on the surface, γ -irradiation or exposure to a plasma field. Alkoxyamine functionality can be attached to pre-existing polymers by generating *t*-butoxy radicals in the presence of a polymer and a nitroxide. This procedure was applied to PB and poly(isobutyl methacrylate) and the resultant polymeric alkoxyamines formed were used to initiate MA and EA polymerization respectively.¹¹¹ Recent papers describe RAFT polymerization from plasma-treated Teflon surfaces⁵⁰⁴ and ozonolyzed polyimide films.⁷⁴⁵

A third technique involves reaction of a functional alkoxyamine, ATRP initiator or RAFT agent with a functional surface. An early example involved the reaction of the sodium salt of **284** with poly(S-*co*-chloromethylstyrene) to provide multi-alkoxyamines which were used for the synthesis of a variety of graft copolymers.⁶⁶³ Interchain coupling reactions, evidenced by broadening of the molecular weight distribution, became significant when there were greater than six alkoxyamine functions per chain.⁶⁶³ The hydroxyl functionalities of ethylene-vinyl alcohol films⁷⁴⁶ were esterified with BriBBr (**314**). Acid functionality of ethylene-acrylic acid copolymer films was transformed to hydroxyl functionality and then esterified with BriBBr.⁷⁴⁷ Perrier and coworkers^{440,748} attached RAFT moieties to cellulose (cotton) in order to form PS, PMA or PMMA grafts. This

involved derivatization of the cellulosic OH groups with thiocarbonylthio functionality.

The very small number of growing polymer chains, when compared to the monomer concentration results in a very low overall concentration of free control agent and leads to inefficient capping of chain ends. One solution to this problem is the addition of a free or unbound control agent to the polymerization medium. This can take the form of a low molecular weight alkoxyamine, ATRP initiator, RAFT agent or, alternatively, free deactivator such as nitroxide or Cu(II). This species is often called a sacrificial agent. This solution also leads to the formation of free polymer that must ultimately be removed from the brush.

9.10.2.2 Grafting from inorganic surfaces

Grafting from silica particles, silicon wafers, and related surfaces usually involves attaching a chlorosilane or alkoxysilane derivative. Thus alkoxyamines $(e.g, 361,^{744,749} 362^{750})$ and a wide variety of ATRP initiators $(e.g. 363^{751})$ have been attached directly to surfaces and used to initiate "grafting from" processes.

Wu *et al.*⁷⁵² developed a technique called MAPA (acronym for Mechanically Assisted Polymer Assembly) to produce polymer brushes on a cross-linked polydimethylsiloxane (PDMS) surface. The technique involves stretching the PDMS substrate, then generating surface silanol groups by ozonolysis. The functional surface was then treated with a trichlorosilane-based ATRP initiator. PAM brushes were then grown from the surface by ATRP. The strain was then released, allowing the PDMS substrate to return to its former size thereby producing densely grafted polymer brushes. By altering the amount by which the PDMS substrate was stretched the grafting density could be controlled.

The application of RAFT polymerization in grafting to surfaces was first investigated by Tsujii *et al.*⁶⁵⁵ and Brittain and coworkers.^{753,754} The approach used in these and other more recent studies^{755,756} was to immobilize the initiator functionality on the surface (*e.g.* an ATRP initiator **364**⁶⁵⁵ or a conventional initiator **365**^{753,754}) and use this to initiate polymerization in the presence of a dithioester RAFT agent. Tsujii *et al.*⁶⁵⁵ reported that some difficulties arise in using RAFT for grafting from particles which they attributed to an abnormally high rate of radical-radical termination caused by the locally high concentration of the RAFT functionality.



562



Treatment of a gold surface with disulfide **367** left bromoisobutyrate groups on the surface to initiate ATRP of various methacrylate esters⁷⁵⁷ including HEMA.^{758,759} Skaff and Emrick⁷⁶⁰ bound RAFT agent functionality to cadmium selenide nanoparticles by a ligand exchange process and grew various narrow MWD polymers (PS, PMA, PBA, PS-*co*-MA, PS-*co*-AA, PS-*co*-IP, PS-*block*-PMA, PS-*block*-PBA) from these particles.

$$Br + O - (CH_2)_{11} - S - S - (CH_2)_{11} - O - H - Br$$

$$367$$

9.10.3 Grafting To - Use of End-Functional Polymers

The synthesis of end functional polymers by NMP, ATRP and RAFT has already been discussed in Section 9.7. The "grafting to" approach involves the covalent attachment of an end-functionalized polymer with reactive surface groups on the substrate. The approach is inherently limited by the crowding of chains at the surface and the limit this places on the final graft density.

RAFT polymerization lends itself to the synthesis of polymers with thiol end groups. Several groups have utilized the property of thiols and dithioesters to bind heavy metals such as gold or cadmium in preparing brushes based on gold film or nanoparticles^{628,761,762} and cadmium selenide nanoparticles.^{763,764}

9.11 Outlook for Living Radical Polymerization

Living radical polymerization currently dominates patents, publications and conferences on radical polymerization. The most popular systems, NMP, ATRP and RAFT, while offering unprecedented versatility are not without drawbacks and still have some limitations. Thus, while the progress in this field since the first edition of this book is substantial by any standard, there remains significant scope for new and improved processes. Further studies of the detailed kinetics and mechanism are also required to enable better understanding so that the full potential of the existing techniques can be realized. The complexities of NMP, ATRP and RAFT are many as this chapter illustrates.

Combining control over architecture with control over the stereochemistry of the propagation process remains a *holy grail* in the field of radical polymerization. Approaches to this end based on conventional polymerization were described in Chapter 8. The development of living polymerization processes has yet to substantially advance this cause.

The development of living radical polymerization has provided the capability for the polymer chemist to synthesize a wide range of novel and well-defined structures. The transformation of this capability into commercial outcomes and novel products has only just commenced.

9.12 References

- 1. Szware, M. Nature 1956, 178, 1168.
- 2. Szware, M. J. Polym. Sci., Part A: Polym. Chem. 1998, 36, ix.
- Moad, G.; Solomon, D.H. The Chemistry of Free Radical Polymerization; Pergamon: Oxford, 1995.
- 4. Matyjaszewski, K., Ed. ACS Symposium Series, Controlled Radical Polymerization; American Chemical Society: Washington, 1998; Vol. 685.
- Matyjaszcwski, K., Ed. ACS Symposium Series. Controlled/Living Radical Polymerization: Progress in ATRP. NMP and RAFT; American Chemical Society: Washington DC, 2000; Vol. 768.
- Matyjaszcwski, K., Ed. ACS Symposium Series, Advances in Controlled/Living Radical Polymerization; American Chemical Society: Washington DC, 2003; Vol. 854.
- 7. German, A.L., Ed. Macromol. Symp., Free Radical Polymerization: Kinetics and Mechanism; Wiley-VCH: Weinheim, 1996; Vol. 111.
- 8. Buback, M.; German, A.L., Eds. *Macromol. Symp., Free Radical Polymerization: Kinetics and Mechanism*; Wiley-VCH: Weinheim, 2002; Vol. 182.
- Matyjaszewski, K. In Handbook of Radical Polymerization; Davis, T.P.; Matyjaszewski, K., Eds.; John Wiley & Sons: Hoboken, 2002; p 361.
- 10. Davis, T.P.; Matyjaszewski, K., Eds. *Handbook of Radical Polymerization*; John Wiley & Sons: Hoboken, 2002.
- 11. Quirk, R.P.; Lee, B. Polym. Int. 1992, 27, 359.
- 12. Matyjaszewski, K.; Mueller, A.H.E. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem.) 1997, 38(1), 6.
- 13. Ivan, B. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem.) 2000, 41(2), 6a.
- Darling, T.R.; Davis, T.P.; Fryd, M.; Gridnev, A.A.; Haddleton, D.M.; Ittel, S.D.; Matheson, R.R., Jr.; Moad, G.; Rizzardo, E. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 1706.
- Darling, T.R.; Davis, T.P.; Fryd, M.; Gridnev, A.A.; Haddleton, D.M.; Ittel, S.D.; Matheson, R.R., Jr.; Moad, G.; Rizzardo, E. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 1709.
- 16. Penczek, S. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 1665.
- 17. Matyjaszcwski, K. Macromolecules 1993, 26, 1787.
- 18. Matyjaszewski, K. J. Phys. Org. Chem. 1995, 8, 197.
- 19. Penczek, S. Polimery 1995, 40, 384.
- 20. Russell, G.T. Aust. J. Chem. 2002, 55, 399.
- 21. Fischer, H. Chem. Rev. 2001, 101, 3581.

- 22. Fukuda, T.; Goto, A.; Tsujii, M. In *Handbook of Radical Polymerization*; Davis, T.P.; Matyjaszewski, K., Eds.; John Wiley & Sons: Hoboken, 2002; p 407.
- 23. Goto, A.; Fukuda, T. Prog. Polym. Sci. 2004, 29, 329.
- 24. Otsu, T.; Yoshida, M. Makromol. Chem., Rapid Commun. 1982, 3, 127.
- 25. Kennedy, J.P. J. Macromol. Sci., Chem. 1979, A13, 695.
- 26. Ameduri, B.; Boutevin, B.; Gramain, P. Adv. Polvm. Sci. 1997, 127, 87.
- 27. Sebenik, A. Prog. Polym. Sci. 1998, 23, 875.
- 28. Otsu, T.; Matsumoto, A. Adv. Polym. Sci. 1998, 136, 75.
- 29. Ferington, T.E.; Tobolsky, A.V. J. Am. Chem. Soc. 1955, 77, 4510.
- 30. Otsu, T.; Kuriyama, A. J. Macromol. Sci., Chem. 1984, A21, 961.
- Shefer, A.; Grodzinsky, A.J.; Prime, K.L.; Busnel, J.P. Macromolecules 1993, 26, 2240.
- 32. Otsu, T.; Yoshida, M. Polym. Bull. 1982, 7, 197.
- 33. Turner, S.R.; Blevina, R.W. Macromolecules 1990, 23, 1856.
- 34. Niwa, M.; Matsumoto, T.; Izumi, H. J. Macromol. Sci., Chem. 1987, A24, 567.
- 35. Nair, C.P.R.; Clouet, G.; Chaumont, P. J. Polym. Sci., Polym. Chem. Ed. 1982, 27, 1795.
- 36. Staudner, E.; Kysela, G.; Beniska, J.; Mikolaj, D. Eur. Polym. J. 1978, 14, 1067.
- 37. Nair, C.P.R.; Clouct, G. J. Macromol. Sci., Rev. Macromol. Chem. Phys. 1991, C31, 311.
- 38. Nair, C.P.R.; Chaumont, P.; Clouet, G. J. Macromol. Sci., Chem 1990, A27, 791.
- 39. Endo, T.; Shiroi, T.; Murata, K. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 145.
- 40. Bricklebank, N.; Pryke, A. J. Chem. Soc., Perkin Trans. 1 2002, 2048.
- 41. Otsu, T.; Kuriyama, A. Polym. J. 1985, 17, 97.
- 42. Otsu, T.; Matsunaga, T.; Kuriyama, A.; Yoshioka, M. Eur. Polym. J. 1989, 25, 643.
- 43. Doi, T.; Matsumoto, A.; Otsu, T. J. Polym. Sci., Part A: Polym. Chem. 1994, 32, 2911.
- 44. Otsu, T.; Kuriyama, A. Polym. Bull. 1984, 11, 135.
- 45. Kuriyama, A.; Otsu, T. Polym. J. 1984, 16, 511.
- 46. Otsu, T.; Yamashita, K.; Tsuda, K. Macromolecules 1986, 19, 287.
- 47. Yamashita, K.; Ito, K.; Tsuboi, H.; Takahama, S.; Tsuda, K.; Otsu, T. J. Appl. Polym. Sci. 1990, 40, 1445.
- 48. Yamashita, K.; Kanamori, T.; Tsuda, K. J. Macromol. Sci., Chem. 1990, A27, 897.
- 49. Ishizu, K.; Ohta, Y.; Kawauchi, S. Macromolecules 2002, 35, 3781.
- 50. Ishizu, K.; Shibuya, T.; Kawauchi, S. Macromolecules 2003, 36, 3505.
- 51. Ajayaghosh, A.; Francis, R. J. Am. Chem. Soc. 1999, 121, 6599.
- 52. Ishizu, K.; Katsuhara, H.; Itoya, K. J. Polym. Sci., Part A: Polym. Chem. 2005, 43, 230.
- 53. Ishizu, K.; Katsuhara, H.; Kawauchi, S.; Furo, M. J. Appl. Polym. Sci. 2005, 95, 413.
- 54. Ishizu, K.; Khan, R.A.; Ohta, Y.; Furo, M. J. Polym. Sci., Part A: Polym. Chem. 2004, 42, 76.
- 55. Ishizu, K.; Khan, R.A.; Furukawa, T.; Furo, M. J. Appl. Polym. Sci. 2004, 91, 3233.
- 56. Nakayama, Y.; Matsuda, T. *Macromolecules* **1996**, *29*, 8622.
- 57. Nakayama, Y.; Matsuda, T. Langmuir 1999, 15, 5560.
- 58. Higashi, J.; Nakayama, Y.; Marchant, R.E.; Matsuda, T. Langmuir 1999, 15, 2080.
- 59. Luo, N.; Hutchison, J.B.; Anseth, K.S.; Bowman, C.N. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 1885.
- 60. Francis, R.; Ajayaghosh, A. Polymer 1995, 36, 1091.
- 61. Ajayaghosh, A.; Francis, R. Macromolecules 1998, 31, 1436.
- 62. Francis, R.; Ajayaghosh, A. Macromolecules 2000, 33, 4699.
- 63. Lambrinos, P.; Tardi, M.; Polton, A.; Sigwalt, P. Eur. Polym. J. 1990, 26, 1125.

- 64. Doi, T.; Matsumoto, A.; Otsu, T. J. Polym. Sci., Part A: Polym. Chem. 1994, 32, 2241.
- 65. Kwon, T.S.; Kumazawa, S.; Yokoi, T.; Kondo, S.; Kunisada, H.; Yuki, Y. J. Macromol. Sci., Chem. **1997**, A34, 1553.
- 66. Kwon, T.S.; Suzuki, K.; Takagi, K.; Kunisada, H.; Yuki, Y. J. Macromol. Sci., Chem. 2001, 38, 591.
- Kwon, T.S.; Ochiai, H.; Kondo, S.; Takagi, K.; Kunisada, H.; Yuki, Y. Polym. J. 1999, 31, 411.
- 68. Kwon, T.S.; Kondo, S.; Kunisada, H.; Yuki, Y. Polym. J. 1998, 30, 559.
- 69. Kwon, T.S.; Takagi, K.; Kunisada, H.; Yuki, Y. J. Macromol. Sci., Chem. 2000, 37, 1461.
- Kwon, T.S.; Kumazawa, S.; Kondo, S.; Takagi, K.; Kunisada, H.; Yuki, Y.J. Macromol. Sci., Chem. 1998, A35, 1895.
- 71. Kwon, T.S.; Takagi, K.; Kunisada, H.; Yuki, Y. J. Macromol. Sci., Chem. 2001, *A38*, 605.
- 72. Kwon, T.S.; Takagi, K.; Kunisada, H.; Yuki, Y. Eur. Polym. J. 2003, 39, 1437.
- 73. Kwon, T.S.; Takagi, K.; Kunisada, H.; Yuki, Y. J. Macromol. Sci., Chem. 2002, A39, 991.
- 74. Patwa, A.N.; Tomer, N.S.; Singh, R.P. J. Mater. Sci. 2004, 39, 1047.
- Rathore, K.; Reddy, K.R.; Tomer, N.S.; Desai, S.M.; Singh, R.P. J. Appl. Polym. Sci. 2004, 93, 348.
- 76. Kwon, T.S.; Kondo, S.; Kunisada, H.; Yuki, Y. Eur. Polym. J. 1999, 35, 727.
- 77. Bledzki, A.; Braun, D.; Titzschkau, K. Makromol. Chem. 1983, 184, 745.
- 78. Bledzki, A.; Braun, D. Makromol. Chem. 1981, 182, 1047.
- 79. Crivello, J.V.; Lee, J.L.; Conlon, D.A. In *Advances in Elastomers and Rubber Elasticity*; Lal, J., Ed.; Plenum: New York, 1986; p 157.
- Crivello, J.V.; Lee, J.L.; Conlon, D.A. J. Polym. Sci., Part A: Polym. Chem. 1986, 24, 1251.
- 81. Crivello, J.V.; Lee, J.L.; Conlon, D.A. Polym. Bull. 1986, 16, 95.
- Crivello, J.V.; Lee, J.L.; Conlon, D.A. J. Polym. Sci., Part A: Polym. Chem. 1986, 24, 1197.
- 83. Santos, R.G.; Chaumont, P.R.; Herz, J.E.; Beinert, G.J. *Eur. Polym. J.* **1994**, *30*, 851.
- 84. Roussel, J.; Boutevin, B. Polym. Int. 2001, 50, 1029.
- 85. Roussel, J.; Boutevin, B. J. Fluor. Chem. 2001, 108, 37.
- 86. Braun, D.; SteinhauerBeisser, S. Eur. Polym. J. 1997, 33, 7.
- 87. Braun, D.; SteinhauerBeisser, S. Angew. Makromol. Chem. 1996, 239, 43.
- Qin, S.H.; Qiu, K.Y.; Swift, G.; Westmoreland, D.G.; Wu, S.G. J. Polym. Sci., Part A: Polym. Chem. 1999, 37, 4610.
- 89. Rüchardt, C. Top. Curr. Chem. 1980, 88, 1.
- 90. Otsu, T.; Tazaki, T. Polym. Bull. 1986, 16, 277.
- 91. McElvain, S.M.; Aldridge, C.L. J. Am. Chem. Soc. 1953, 75, 3987.
- 92. Moad, G.; Rizzardo, E.; Solomon, D.H. Aust. J. Chem. 1983, 36, 1573.
- 93. Demircioglu, P.; Acar, M.H.; Yagci, Y. J. Appl. Polym. Sci. 1992, 46, 1639.
- 94. Engel, P.S.; Chen, Y.; Wang, C. J. Org. Chem. 1991, 56, 3073.
- 95. Otsu, T.; Matsumoto, A.; Tazaki, T. Polym. Bull. 1987, 17, 323.
- Bledzki, A.; Braun, D.; Menzel, W.; Titzschkau, K. Makromol. Chem. 1983, 184, 287.
- 97. Bledzki, A.; Braun, D. Polym. Bull. 1986, 16, 19.
- 98. Bledzki, A.; Braun, D. Makromol. Chem. 1986, 187, 2599.
- 99. Moad, G.; Rizzardo, E.; Solomon, D.H. Macromolecules 1982, 15, 909.
- 100. Marvel, C.S.; Dec, J.; Corner, J.O. J. Am. Chem. Soc. 1945, 67, 1855.
- 101. Wieland, P.C.; Raether, B.; Nuyken, O. Macromol. Rapid Commun. 2001, 22, 700.

- 102. Wieland, P.C.; Nuyken, O.; Heischkel, Y.; Raether, B.; Strissel, C. ACS Symp. Ser. 2003, 854, 619.
- 103. Viala, S.; Antonietti, M.; Tauer, K.; Bremser, W. Polymer 2003, 44, 1339.
- 104. Viala, S.; Tauer, K.; Antonietti, M.; Kruger, R.P.; Bremser, W. Polymer 2002, 43, 7231.
- 105. Tanaka, H. Prog. Polym. Sci. 2003, 28, 1171.
- 106. Hawker, C.J.; Bosman, A.W.; Harth, E. Chem. Rev. 2001, 101, 3661.
- Hawker, C.J. In Handbook of Radical Polymerization; Davis, T.P.; Matyjaszewski, K., Eds.; John Wiley & Sons: Hoboken, 2002; p 463.
- 108. Studer, A.; Schulte, T. Chem. Rec. 2005, 5, 27.
- 109. Solomon, D.H. J. Polym. Sci., Part A: Polym. Chem. 2005, in press.
- 110. Fischer, H.; Radom, L. Angew. Chem., Int. Ed. Engl. 2001, 40, 1340.
- Solomon, D.H.; Rizzardo, E.; Cacioli, P. US 4581429, 1986 (Chem. Abstr. 1985, 102, 221335q).
- 112. Rizzardo, E. Chem. Aust. 1987, 54, 32.
- Rizzardo, E.; Chong, Y.K. In 2nd Pacific Polymer Conference, Preprints; Pacific Polymer Federation: Tokyo, 1991; p 26.
- 114. Johnson, C.H.J.; Moad, G.; Solomon, D.H.; Spurling, T.H.; Vcaring, D.J. Aust. J. Chem. 1990, 43, 1215.
- 115. Fischer, H. Macromolecules 1997, 30, 5666.
- Georges, M.K.; Veregin, R.P.N.; Kazmaier, P.M.; Hamer, G.K. Macromolecules 1993, 26, 2987.
- Moad, G.; Rizzardo, E. In *Pacific Polymer Conference Preprints*; Polymer Division, Royal Australian Chemical Institute: Brisbane, 1993; Vol. 3, p 651.
- 118. Moad, G.; Rizzardo, E. Macromolecules 1995, 28, 8722.
- Georges, M.K.; Veregin, R.P.N.; Kazmaier, P.M.; Hamer, G.K. *Trends Polym. Sci.* 1994, 2, 66.
- 120. Marque, S.; Le Mercier, C.; Tordo, P.; Fischer, H. Macromolecules 2000, 33, 4403.
- 121. Marque, S.; Fischer, H.; Baier, E.; Studer, A. J. Org. Chem. 2001, 66, 1146.
- 122. Moad, G.; Ercole, F.; Johnson, C.H.; Krstina, J.; Moad, C.L.; Rizzardo, E.; Spurling, T.H.; Thang, S.H.; Anderson, A.G. ACS Symp. Ser. 1998, 685, 332.
- 123. Kazmaier, P.M.; Moffat, K.A.; Georges, M.K.; Veregin, R.P.N.; Hamer, G.K. Macromolecules 1995, 28, 1841.
- Grishin, D.F.; Semenycheva, L.L.; Kolyakina, E.V. Russ. J. Appl. Chem. 2001, 74, 494.
- Golubev, V.B.; Zaremski, M.Y.; Orlova, A.P.; Olenin, A.V. Polym. Sci. 2004, 46, 295.
- 126. Zarcmski, M.Y.; Orlova, A.P.; Garina, E.S.; Olenin, A.V.; Lachinov, M.B.; Golubev, V.B. Polym. Sci. 2003, 45, 502.
- Detrembleur, C.; Sciannamea, V.; Koulic, C.; Claes, M.; Hoebeke, M.; Jerome, R. Macromolecules 2002, 35, 7214.
- 128. Catala, J.M.; Jousset, S.; Lamps, J.P. Macromolecules 2001, 34, 8654.
- 129. Detrembleur, C.; Claes, M.; Jerome, R. ACS Symp. Ser. 2003, 854, 496.
- 130. Detrembleur, C.; Teyssie, P.; Jerome, R. Macromolecules 2002, 35, 1611.
- Cameron, N.R.; Reid, A.J.; Span, P.; Bon, S.A.F.; van Es, J.; German, A.L. Macromol. Chem. Phys. 2000, 201, 2510.
- 132. Brinkman-Rengel, S.; Niessner, N. ACS Symp. Ser. 2000, 768, 394.
- 133. Cameron, N.R., Reid, A.J. Macromolecules 2002, 35, 9890.
- 134. Yamada, B.; Miura, Y.; Nobukane, Y.; Aota, M. ACS Symp. Ser. 1998, 685, 200.
- 135. Puts, R.D.; Sogah, D.Y. Macromolecules 1997, 30, 3323.
- Veregin, R.P.N.; Georges, M.K.; Hamer, G.K.; Kazmaier, P.M. Macromolecules 1995, 28, 4391.

- Cresidio, S.P.; Aldabbagh, F.; Busfield, W.K.; Jenkins, I.D.; Thang, S.H.; Zayas-Holdsworth, C.; Zetterlund, P.B. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 1232.
- 138. Chong, Y.K.; Ercole, F.; Moad, G.; Rizzardo, E.; Thang, S.H.; Anderson, A.G. Macromolecules 1999, 32, 6895.
- 139. Dervan, P.; Aldabbagh, F.; Zetterlund, P.B.; Yamada, B. J. Polym. Sci., Part A: Polym. Chem. 2003, 41, 327.
- 140. Hawker, C.J. J. Am. Chem. Soc. 1994, 116, 11185.
- Skene, W.G.; Belt, S.T.; Connolly, T.J.; Hahn, P.; Scaiano, J.C. *Macromolecules* 1998, 31, 9103.
- 142. Schulte, T.; Knoop, C.A.; Studer, A. J. Polym. Sci., Part A: Polym. Chem. 2004, 42, 3342.
- 143. Zhu, Y.; Li, I.O.; Howell, B.; Priddy, D.B. ACS Symp. Ser. 1997, 685, 214.
- 144. Yoshida, E. J. Polym. Sci., Part A: Polym. Chem. 1996, 34, 2937.
- 145. Yoshida, E.; Fuji, T. J. Polym. Sci., Part A: Polym. Chem. 1997, 35, 2371.
- 146. Yoshida, E.; Fujii, T. J. Polym. Sci., Part A: Polym. Chem. 1998, 36, 269.
- 147. Marque, S.; Sobek, J.; Fischer, II.; Kramer, A.; Nesvadba, P.; Wunderlich, W. Macromolecules 2003, 36, 3440.
- 148. Knoop, C.A.; Studer, A. J. Am. Chem. Soc. 2003, 125, 16327.
- 149. Miura, Y.; Nakamura, N.; Taniguchi, I. Macromolecules 2001, 34, 447.
- Aldabbagh, F.; Dervan, P.; Phelan, M.; Gilligan, K.; Cunningham, D.; McArdle, P.; Zetterlund, P.B.; Yamada, B. J. Polym. Sci., Part A: Polym. Chem. 2003, 41, 3892.
- 151. Goto, A.; Fukuda, T. Macromolecules 1999, 32, 618.
- 152. Grubbs, R.B.; Wegrzyn, J.K.; Xia, Q. Chem. Commun. 2005, 80.
- 153. Benoit, D.; Chaplinski, V.; Braslau, R.; Hawker, C.J. J. Am. Chem. Soc 1999, 121, 3904.
- 154. Benoit, D.; Harth, E.; Fox, P.; Waymouth, R.M.; Hawker, C.J. *Macromolecules* **2000**, *33*, 363.
- 155. Studer, A.; Harms, K.; Knoop, C.; Muller, C.; Schulte, T. *Macromolecules* **2004**, 37, 27.
- 156. Benoit, D.; Grimaldi, S.; Robin, S.; Finet, J.P.; Tordo, P.; Gnanou, Y. J. Am. Chem. Soc 2000, 122, 5929.
- 157. Benoit, D.; Grimaldi, S.; Finet, J.P.; Tordo, P.; Fontanille, M.; Gnanou, Y. ACS Symp. Ser. **1998**, 685, 225.
- 158. Farcet, C.; Belleney, J.; Charleux, B.; Pirri, R. Macromolecules 2002, 35, 4912.
- 159. Drockenmuller, E.; Lamps, J.P.; Catala, J.M. Macromolecules 2004, 37, 2076.
- 160. Drockenmuller, E.; Catala, J.M. Macromolecules 2002, 35, 2461.
- 161. Nesvadba, P.; Bugnon, L.; Sift, R. Polym. Int. 2004, 53, 1066.
- 162. Schulte, T.; Studer, A. Macromolecules 2003, 36, 3078.
- 163. Nesvadba, P.; Bugnon, L.; Sift, R. J. Polym. Sci., Part A: Polym. Chem. 2004, 42, 3332.
- Wetter, C.; Gierlich, J.; Knoop, C.A.; Muller, C.; Schulte, T.; Studer, A. Chem. Eur. J. 2004, 10, 1156.
- Hawker, C.J.; Barclay, G.G.; Orellana, A.; Dao, J.; Devonport, W. Macromolecules 1996, 29, 5245.
- 166. Georges, M.K.; Kee, R.A.; Veregin, R.P.N.; Hamer, G.K.; Kazmaier, P.M. J. Phys. Org. Chem. 1995, 8, 301.
- 167. Catala, J.M.; Bubel, F.; Hammouch, S.O. Macromolecules 1995, 28, 8441.
- 168. Hammouch, S.O.; Catala, J.M. Macromol. Rapid Commun. 1996, 17, 683.
- 169. Greszta, D.; Matyjaszewski, K. Macromolecules 1996, 29, 5239.
- 170. Connolly, T.J.; Baldovi, M.V.; Mohtat, N.; Scaiano, J.C. Tetrahedron Lett. 1996, 37, 4919.
- 171. Miura, Y.; Ilirota, K.; Moto, II.; Kobetake, S. Macromolecules `1998, 37, 4051.

568

- 172. Li, I.Q.; Knauss, D.M.; Priddy, D.B.; Howell, B.A. Polym. Int. 2003, 52, 805.
- 173. Sugimoto, N.; Narumi, Λ.; Satoh, T.; Kaga, H.; Kakuchi, T. Polym. Bull. 2003, 49, 337.
- Matyjaszewski, K.; Woodworth, B.E.; Zhang, X.; Gaynor, S.G.; Metzner, Z. Macromolecules 1998, 31, 5955.
- 175. Dao, J.; Benoit, D.; Hawker, C.J. J. Polym. Sci., Part A: Polym. Chem. 1998, 36, 2161.
- 176. Bothe, M.; Schmidt-Naake, G. Macromol. Rapid Commun. 2003, 24, 609.
- 177. Lukkarila, J.L.; Hamer, G.K.; Georges, M.K. Tetrahedron Lett. 2004, 45, 5317.
- 178. Gridnev, A.A. Macromolecules 1997, 30, 7651.
- Li, I.; Howell, B.; Matyjasewski, K.; Shigemoto, T.; Smith, P.B.; Priddy, D.B. Macromolecules 1995, 28, 6692.
- 180. Souaille, M.; Fischer, H. Macromolecules 2002, 35, 248.
- 181. Malmstrom, E.; Miller, R.D.; Hawker, C.J. Tetrahedron 1997, 53, 15225.
- 182. Goto, A.; Tsujii, Y.; Fukuda, T. Chem, Lett. 2000, 788.
- 183. Baumann, M.; Schmidt-Naake, G. Macromol. Chem. Phys. 2001, 202, 2727.
- Georges, M.K.; Veregin, R.P.N.; Kazmaier, P.M.; Hamer, G.K.; Saban, M. Macromolecules 1994, 27, 7228.
- 185. Matthews, B.R.; Pike, W.; Rego, J.M.; Kuch, P.D.; Priddy, D.B. J. Appl. Polym. Sci. 2003, 87, 869.
- Odell, P.G.; Veregin, R.P.N.; Michalak, L.M.; Brousmiche, D.; Georges, M.K. Macromolecules 1995, 28, 8453.
- Odell, P.G.; Veregin, R.P.N.; Michalak, L.M.; Georges, M.K. *Macromolecules* 1997, 30, 2232.
- Keoshkerian, B.; Georges, M.; Quinlan, M.; Veregin, R.; Goodbrand, B. Macromolecules 1998, 31, 7559.
- 189. Georges, M.K.; Lukkarila, J.L.; Szkurhan, A.R. Macromolecules 2004, 37, 1297.
- 190. Keoshkerian, B.; Szkurhan, A.R.; Georges, M.K. Macromolecules 2001, 34, 6531.
- 191. Goto, A.; Fukuda, T. Macromolecules 1997, 30, 4272.
- Greszta, D.; Matyjaszewski, K. J. Polym. Sci., Part A: Polym. Chem. 1997, 35, 1857.
- 193. Baethge, H.; Butz, S.; Han, C.-H.; Schmidt-Naake, G. Angew. Makromol. Chem. 1999, 267, 52.
- 194. Butz, S.; Baethge, II.; Schmidt-Naake, G. Angew. Makromol. Chem. 1999, 270, 42.
- 195. He, J.; Chen, J.; Li, L.; Pan, J.; Li, C.; Cao, J.; Tao, Y.; Hua, F.; Yang, Y.; McKee, G.E.; Brinkmann, S. *Polymer* 2000, 41, 4573.
- 196. Bian, K.; Cunningham, M.F. Macromolecules 2005, 38, 695.
- Diaz, T.; Fischer, A.; Jonquieres, A.; Brembilla, A.; Lochon, P. Macromolecules 2003, 36, 2235.
- Couvreur, L.; Lefay, C.; Belleney, J.; Charleux, B.; Guerret, O.; Magnet, S. Macromolecules 2003, 36, 8260.
- Lefay, C.; Belleney, J.; Charleux, B.; Guerret, O.; Magnet, S. Macromol. Rapid Commun. 2004, 25, 1215.
- 200. Bohrisch, J.; Wendler, U.; Jaeger, W. Macromol. Rapid Commun. 1997, 18, 975.
- Listigovers, N.A.; Georges, M.K.; Odell, P.G.; Keoshkerian, B. Macromolecules 1996, 29, 8992.
- Burguiere, C.; Dourges, M.A.; Charleux, B.; Vairon, J.P. Macromolecules 1999, 32, 3883.
- 203. Georges, M.K.; Hamer, G.K.; Listigovers, N.A. Macromolecules 1998, 31, 9087.
- 204. Benoit, D.; Harth, E.; Helms, B.; Rees, I.; Vestberg, R.; Rodlert, M.; Hawker, C.J. ACS Symp. Ser. 2000, 768, 123.
- 205. Qiu, J.; Charleux, B.; Matyjaszewski, K. Prog. Polym. Sci. 2001, 26, 2083.
- 206. Cunningham, M.F. Prog. Polym. Sci. 2002, 27, 1039.
- 207. Cunningham, M.F. Compt. Rend. Chim. 2003, 6, 1351.

- Schork, F.J.; Y., L.; Smulders, W.; Russum, J.P.; Butte, A.; Fontenot, K. Adv. Polym. Sci. 2005, 175, 129.
- 209. Charleux, B. Macromolecules 2000, 33, 5358.
- 210. Butte, A.; Storti, G.; Morbidelli, M. Macromolecules 2001, 34, 5885.
- 211. Ma, J.W.; Cunningham, M.F.; McAuley, K.B.; Keoshkerian, B.; Georges, M. Chem. Eng. Sci. 2003, 58, 1177.
- Ma, J.W.; Cunningham, M.F.; McAuley, K.B.; Keoshkerian, B.; Georges, M.K. Macromol. Theory Simul. 2003, 12, 72.
- 213. Ma, J.W.; Smith, J.A.; McAuley, K.B.; Cunningham, M.F.; Keoshkerian, B.; Georges, M.K. *Chem. Eng. Sci.* 2003, 58, 1163.
- Butte, A.; Storti, G.; Morbidelli, M. In DECHEMA Monograph., 2001; Vol. 137, p 273.
- 215. Bon, S.A.F.; Bosveld, M.; Klumperman, B.; German, A.L. *Macromolecules* **1997**, 30, 324.
- 216. Cao, J.Z.; He, J.P.; Li, C.M.; Yang, Y.L. Polym. J. 2001, 33, 75.
- 217. Marestin, C.; Noel, C.; Guyot, A.; Claverie, J. Macromolecules 1998, 31, 4041.
- 218. Szkurhan, A.R.; Georges, M.K. Macromolecules 2004, 37, 4776.
- Nicolas, J.; Charleux, B.; Guerret, O.; Magnet, S.P. Angew. Chem. Int. Ed. Engl. 2004, 43, 6186.
- 220. Prodpran, T.; Dimonie, V.L.; Sudol, E.D.; El-Aasser, M.S. *Macromol. Symp.* 2000, 155, 1.
- 221. Pan, G.F.; Sudol, E.D.; Dimonie, V.L.; El-Aasser, M.S. *Macromolecules* **2001**, *34*, 481.
- 222. MacLeod, P.J.; Barber, R.; Odell, P.G.; Keoshkerian, B.; Georges, M.K. Macromol. Symp. 2000, 155, 31.
- 223. Cunningham, M.F.; Tortosa, K.; Ma, J.W.; McAuley, K.B.; Keoshkerian, B.; Georges, M.K. Macromol. Symp. 2002, 182, 273.
- 224. Nicolas, J.; Charleux, B.; Guerret, O.; Magnet, S. Macromolecules 2004, 37, 4453.
- 225. Cunningham, M.; Lin, M.; Buragina, C.; Milton, S.; Ng, D.; Hsu, C.C.; Keoshkerian, B. *Polymer* **2005**, *46*, 1025.
- 226. Druliner, J.D. Macromolecules 1991, 24, 6079.
- 227. Yamada, B.; Tanaka, H.; Konishi, K.; Otsu, T. J. Macromol. Sci., Chem. 1994, A31, 351.
- Grande, D.; Baskaran, S.; Baskaran, C.; Gnanou, Y.; Chaikof, E.L. Macromolecules 2000, 33, 1123.
- 229. Grande, D.; Baskaran, S.; Chaikof, E.L. Macromolecules 2001, 34, 1640.
- 230. Grande, D.; Guerrero, R.; Gnanou, Y. J. Polym. Sci., Part A: Polym. Chem. 2005, 43, 519.
- 231. Chung, T.C.; Janvikul, W.; Lu, H.L. J. Am. Chem. Soc. 1996, 118, 705.
- 232. Chung, T.C.; Hong, H. ACS Symp. Ser. 2003, 854, 481.
- 233. Hong, H.; Chung, T.C. Macromolecules 2004, 37, 6260.
- 234. Chung, T.C.; Lu, H.L.; Janvikul, W. Polymer 1997, 38, 1495.
- 235. Lu, B.; Chung, T.C. Macromolecules 1998, 31, 5943.
- 236. Chung, T.C.; Xu, G.; Lu, Y.Y.; Hu, Y.L. Macromolecules 2001, 34, 8040.
- 237. Chung, T.C. Israel J. Chem. 2002, 42, 307.
- Dimonie, M.; Mardare, D.; Matyjaszewski, K.; Coca, S.; Dragutan, V.; Ghiviriga, J. Macromol. Rapid Commun. 1992, 13, 283.
- 239. Mardare, D.; Matyjaszewski, K.; Coca, S. Macromol. Rapid Commun. 1994, 15, 37.
- 240. Mardare, D.; Matyjaszewski, K. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem) 1993, 34(2), 566.
- 241. Mardare, D.; Matyjaszewski, K. Macromolecules 1994, 27, 645.
- Granel, C.; Jerome, R.; Teyssie, P.; Jasieczek, C.B.; Shooter, A.J.; Haddleton, D.M.; Hastings, J.J.; Gigmes, D.; Grimaldi, S.; Tordo, P.; Greszta, D.; Matyjaszewski, K. *Macromolecules* 1998, 31, 7133.

- 243. Klapper, M.; Brand, T.; Steenbock, M.; Müllen, K. ACS Symp. Ser. 2000, 768, 152.
- 244. Steenbock, M.; Klapper, M.; Muellen, K.; Bauer, C.; Hubrich, M. Macromolecules 1998, 31, 5223.
- 245. Steenbock, M.; Klapper, M.; Muellen, K. Macromol. Chem. Phys. 1998, 199, 763.
- 246. Khelfallah, N.S.; Peretolchin, M.; Klapper, M.; Mullen, K. Polym. Bull. 2005, 53, 295.
- 247. Dasgupta, A.; Brand, T.; Klapper, M.; Mullen, K.R. Polym. Bull. 2001, 46, 131.
- 248. Grishin, D.F.; Ignatov, S.K.; Shchepalov, A.A.; Razuvaev, A.G. Appl. Organometal. Chem. 2004, 18, 271.
- 249. Asandei, A.D.; Moran, I.W. J. Am. Chem. Soc. 2004, 126, 15932.
- 250. Wayland, B.B.; Poszmik, G.; Mukerjee, S.L.; Fryd, M. J. Am. Chem. Soc. 1994, 116, 7943.
- 251. Wayland, B.B.; Basickes, L.; Mukerjee, S.; Wei, M.; Fryd, M. *Macromolecules* **1997**, *30*, 8109.
- 252. Lu, Z.; Fryd, M.; Wayland, B.B. Macromolecules 2004, 37, 2686.
- 253. Debuigne, A.; Caille, J.-R.; Jerome, R. Angew. Chem. Int. Ed. Engl. 2005, 44, 1101 -1104.
- 254. Le Grognec, E.; Claveric, R.; Poli, R. J. Am. Chem. Soc. 2001, 123, 9513.
- 255. Oganova, A.G.; Smirnov, B.R.; Ioffe, N.T.; Enikolopyan, N.S. *Bull. Acad. Sci.* USSR **1983**, 1837.
- 256. Oganova, A.G.; Smirnov, B.R.; Ioffe, N.T.; Kim, I.P. Bull. Acad. Sci. USSR 1984, 1154.
- 257. Oganova, A.G.; Smirnov, B.R.; Ioffe, N.T.; Enikopyan, N.S. Doklady Akad. Nauk SSR (Engl. Transl.) 1983, 268, 66.
- Morozova, I.S.; Oganova, A.G.; Nosova, V.S.; Novikov, D.D.; Smirnov, B.R. Bull. Acad. Sci. USSR 1987, 2628.
- 259. Smirnov, V.R. Polym. Sci. USSR (Engl. Transl.) 1990, 32, 524.
- Arvanitopoulos, L.D.; Greuel, M.P.; Harwood, H.J. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem.) 1994, 35(2), 549.
- 261. Kharasch, M.S.; Jensen, E.V.; Urry, W.H. Science 1945, 102, 128.
- 262. Minisci, F. Acc. Chem. Res. 1975, 8, 165.
- 263. Ameduri, B.; Boutevin, B. Macromolecules 1990, 23, 2433.
- 264. Bamford, C.H. In *Comprehensive Polymer Science*; Eastmond, G.C.; Ledwith, A.; Russo, S.; Sigwalt, P., Eds.; Pergamon: Oxford, 1989; Vol. 3, p 123.
- Kato, M.; Kamigaito, M.; Sawamoto, M.; Higashamura, T. Macromolecules 1995, 28, 1721.
- 266. Wang, J.-S.; Matyjaszewski, K. Macromolecules 1995, 28, 7901.
- 267. Percec, V.; Barboiu, B. Macromolecules 1995, 28, 7970.
- 268. Matyjaszewski, K.; Xia, J. Chem. Rev. 2001, 101, 2921.
- 269. Matyjaszewski, K.; Xia, J. In *Handbook of Radical Polymerization*; Davis, T.P.; Matyjaszewski, K., Eds.; John Wiley & Sons: Hoboken, 2002; p 523.
- 270. Kamigaito, M.; Ando, T.; Sawamoto, M. Chem. Rev. 2001, 101, 3689.
- 271. Kamigaito, M.; Ando, T.; Sawamoto, M. Chem. Rec. 2004, 4, 159.
- 272. Percec, V.; Popov, A.V.; Ramirez-Castillo, E.; Monteiro, M.; Barboiu, B.;
- Weichold, O.; Asandei, A.D.; Mitchell, C.M. J. Am. Chem. Soc. 2002, 124, 4940.
- 273. Percec, V.; Popov, A.V.; Ramirez-Castillo, E.; Weichold, O. J. Polym. Sci., Part A: Polym. Chem. 2003, 41, 3283.
- 274. Percee, V.; Popov, A.V.; Ramirez-Castillo, E.; Coelho, J.F.J. J. Polym. Sci., Part A: Polym. Chem. 2005, 43, 773.
- 275. Lad, J.; Harrisson, S.; Haddleton, D.M. ACS Symp. Ser. 2003, 854, 148.
- 276. Lad, J.; Harrisson, S.; Mantovani, G.; Haddleton, D.M. Dalton Trans. 2003, 4175.
- 277. Jiang, J.G.; Zhang, K.D.; Zhou, H. J. Polym. Sci., Part A: Polym. Chem. 2004, 42, 5811.
- 278. Percec, V.; Barboiu, B.; Kim, H.J. J. Am. Chem. Soc 1998, 120, 305.

- 279. Percec, V.; Kim, H.J.; Barboiu, B. Macromolecules 1997, 30, 8526.
- 280. Percec, V.; Barboiu, B.; Bera, T.K.; van der Sluis, M.; Grubbs, R.B.; Frechet, J.M.J. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 4776.
- 281. Grigoras, C.; Percee, V. J. Polvm. Sci., Part A: Polym. Chem. 2005, 43, 319.
- 282. Gurr, P.A.; Mills, M.F.; Qiao, G.G.; Solomon, D.H. Polymer 2005, 46, 2097.
- 283. Venkatesh, R.; Harrisson, S.; Haddleton, D.M.; Klumperman, B. Macromolecules **200**4, *37*, 4406.
- 284. Matsuyama, M.; Kamigaito, M.; Sawamoto, M. J. Polym. Sci., Part A: Polym. Chem. 1996, 34, 3585.
- 285. Wang, J.-S.; Gaynor, S.; Matyjaszewski, K. Macromolecules 1995, 28, 7572.
- 286. Xia, J.H.; Matyjaszewski, K. Macromolecules 1999, 32, 5199.
- 287. Qiu, J.; Gaynor, S.G.; Matyjaszewski, K. Macromolecules 1999, 32, 2872.
- 288. Qiu, J.; Pintauer, T.; Gaynor, S.G.; Matyjaszewski, K.; Charleux, B.; Vairon, J.P. Macromolecules 2000, 33, 7310.
- 289. Wang, W.; Yan, D. ACS Symp. Ser. 2000, 768, 263.
- 290. Gromada, J.; Matyjaszewski, K. Macromolecules 2001, 34, 7664.
- 291. Li, M.; Min, K.; Matyjaszewski, K. Macromolecules 2004, 37, 2106.
- 292. Matyjaszewski, K.; Paik, H.J.; Zhou, P.; Diamanti, S.J. Macromolecules 2001, 34, 5125.
- 293. Pintauer, T.; Zhou, P.; Matyjaszewski, K. J. Am. Chem. Soc. 2002, 124, 8196.
- 294. Nanda, A.K.; Matyjaszewski, K. Macromolecules 2003, 36, 599.
- 295. Nanda, A.K.; Matyjaszewski, K. Macromolecules 2003, 36, 1487.
- 296. Nanda, A.K.; Matyjaszcwski, K. Macromolecules 2003, 36, 8222.
- 297. Pintauer, T.; Braunecker, W.; Collange, E.; Poli, R.; Matyjaszewski, K. Macromolecules 2004, 37, 2679.
- 298. Brandts, J.A.M.; van de Geijn, P.; van Faassen, E.E.; Boersma, J.; van Koten, G. J. Organometal. Chem. 1999, 584, 246.
- 299. Stoffelbach, F.; Haddleton, D.M.; Poli, R. Eur. Polym. J. 2003, 39, 2099.

- 300. Endo, K.; Yachi, A. Polym. Bull. 2001, 46, 363.
 301. Endo, K.; Yachi, A. Polym. J. 2002, 34, 320.
 302. Kotani, Y.; Kamigaito, M.; Sawamoto, M. Macromolecules 2000, 33, 6746.
- 303. Kotani, Y.; Kamigaito, M.; Sawamoto, M. Macromolecules 1999, 32, 2420.
- 304. Wang, B.Q.; Zhuang, Y.; Luo, X.X.; Xu, S.S.; Zhou, X.Z. Macromolecules 2003, 36, 9684.
- 305. Moineau, G.; Granel, C.; Dubois, P.; Jerome, R.; Teyssie, P. Macromolecules 1998, 31, 542
- 306. Opstal, T.; Zednik, J.; Sedlacek, J.; Svoboda, J.; Vohlidal, J.; Verpoort, F. Collect. Czech. Chem. Commun. 2002, 67, 1858.
- 307. Percec, V.; Barboiu, B.; A., N. Macromolecules 1996, 29, 3665.
- 308. Lecomte, P.; Drapier, I.; Dubois, P.; Teyssie, P.; Jerome, R. Macromolecules 1997, 30, 7631.
- 309. Shen, Y.Q.; Tang, H.D.; Ding, S.J. Prog. Polym. Sci. 2004, 29, 1053.
- 310. Pintauer, T.; Matyjasewski, K. Coord. Chem. Rev. 2005, 249, 1155.
- 311. Xia, J.; Matyjaszewski, K. Macromolecules 1997, 30, 7697.
- 312. Sadhu, V.B.; Pionteck, J.; Voigt, D.; Komber, H.; Fischer, D.; Voit, B. Macromol. Chem. Phys. 2004, 205, 2356.
- 313. Sadhu, V.B.; Pionteck, J.; Voigt, D.; Komber, H.; Voit, B. Macromol. Symp. 2004, 210, 147.
- 314. Haddleton, D.M.; Jasieczek, C.B.; Hannon, M.J.; Shooter, A., J. Macromolecules 1997, 30, 2190.
- 315. Qiu, J.; Matyjaszewski, K.; Thouin, L.; Amatore, C. Macromol. Chem. Phys. 2000, 201, 1625.
- 316. Xia, J.H.; Matyjaszewski, K. Macromolecules 1997, 30, 7697.
- 317. Matyjaszewski, K.; Patten, T.E.; Xia, J. J. Am. Chem. Soc. 1997, 119, 674.

- Ohno, K.; Goto, A.; Fukuda, T.; Xia, J.H.; Matyjaszewski, K. *Macromolecules* 1998, 31, 2699.
- Destarac, M.; Bessiere, J.M.; Boutevin, B. Macromo. Rapid Commun. 1997, 18, 967.
- Wang, X.S.; Malet, F.L.G.; Armes, S.P.; Haddleton, D.M.; Perrier, S. Macromolecules 2001, 34, 162.
- 321. Limer, A.; Heming, A.; Shirley, I.; Haddleton, D. Eur. Polym. J. 2005, 41, 805.
- 322. Zhang, X.; Xia, J.H.; Matyjaszewski, K. Macromolecules 1998, 31, 5167.
- 323. Kickelbick, G.; Matyjaszewski, K. Macromol. Rapid Commun. 1999, 20, 341.
- 324. Xia, J.H.; Matyjaszewski, K. Macromolecules 1999, 32, 2434.
- 325. Xia, J.; Zhang, K.; Matyjasewski, K. ACS Symp. Ser. 2000, 760, 207.
- 326. Xia, J.H.; Gaynor, S.G.; Matyjaszewski, K. Macromolecules 1998, 31, 5958.
- 327. Percec, V.; Barboiu, B.; van der Sluis, M. Macromolecules 1998, 31, 4053.
- 328. van der Sluis, M.; Barboiu, B.; Pesa, N.; Percee, V. Macromolecules 1998, 31, 9409.
- 329. Haddleton, D.M.; Kukulj, D.; Radigue, A.P. Chem. Commun. 1999, 99.
- 330. Kotre, T.; Nuyken, O.; Weberskirch, R. Macromol. Chem. Phys. 2004, 205, 1187.
- 331. Faucher, S.; Zhu, S.P. Macromol. Rapid Commun. 2004, 25, 991.
- 332. Kumar, K.R.; Kizhakkedathu, J.N.; Brooks, D.E. Macromol. Chem. Phys. 2004, 205, 567.
- 333. Nguyen, J.V.; Jones, C.W. J. Polym. Sci., Part A: Polym. Chem. 2004, 42, 1367.
- 334. Honigfort, M.E.; Brittain, W.J. Macromolecules 2003, 36, 3111.
- 335. Hong, S.C.; Matyjaszcwski, K. Macromolecules 2002, 35, 7592.
- 336. Shen, Y.Q.; Zhu, S.P. Macromolecules 2001, 34, 8603.
- 337. Hong, S.C.; Paik, H.J.; Matyjaszewski, K. Macromolecules 2001, 34, 5099.
- 338. Shen, Y.Q.; Zhu, S.P.; Pelton, R. Macromolecules 2001, 34, 3182.
- Liou, S.; Rademacher, J.T.; Malaba, D.; Pallack, M.E.; Brittain, W.J. Macromolecules 2000, 33, 4295.
- Honigfort, M.E.; Liou, S.; Rademacher, J.; Malaba, D.; Bosanac, T.; Wilcox, C.S.; Brittain, W.J. ACS Symp. Ser. 2003, 854, 250.
- Nishikawa, T.; Ando, T.; Kamigaito, M.; Sawamoto, M. Macromolecules 1997, 30, 2244.
- Takahashi, H.; Ando, T.; Kamigaito, M.; Sawamoto, M. Macromolecules 1999, 32, 3820.
- 343. Kotani, Y.; Kato, M.; Kamigaito, M.; Sawamoto, M. *Macromolecules* **1996**, *29*, 6979.
- 344. Senoo, M.; Kotani, Y.; Kamigaito, M.; Sawamoto, M. Macromolecules 1999, 32, 8005.
- 345. Kotani, Y.; Kamigaito, M.; Sawamoto, M. ACS Symp. Ser. 2000, 768, 168.
- 346. Kamigaito, M.; Watanabe, Y.; Ando, T.; Sawamoto, M. J. Am. Chem. Soc. 2002, 124, 9994.
- Watanabe, Y.; Ando, T.; Kamigaito, M.; Sawamoto, M. Macromolecules 2001, 34, 4370.
- 348. Kamigaito, M.; Ando, T.; Sawamoto, M. ACS Symp. Ser. 2003, 854, 102.
- 349. Fuji, Y.; Watanabe, K.; Baek, K.Y.; Ando, T.; Kamigaito, M.; Sawamoto, M. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 2055.
- Wakioka, M.; Baek, K.Y.; Ando, T.; Kamigaito, M.; Sawamoto, M. Macromolecules 2002, 35, 330.
- 351. Ando, T.; Kamigaito, M.; Sawamoto, M. Macromolecules 1997, 30, 4507.
- 352. Kotani, Y.; Kamigaito, M.; Sawamoto, M. Macromolecules 1999, 32, 6877.
- 353. Kotani, Y.; Kamigaito, M.; Sawamoto, M. Macromolecules 2000, 33, 3543.
- 354. Kamigaito, M.; Onishi, I.; Kimura, S.; Kotani, Y.; Sawamoto, M. Chem. Commun. 2002, 2694.

- Matyjaszewski, K.; Wei, M.L.; Xia, J.H.; McDermott, N.E. Macromolecules 1997, 30, 8161.
- 356. Zhu, S.; Yan, D.; Van Beylen, M. ACS Symp. Ser. 2003, 854, 221.
- 357. Zhang, H.Q.; Schubert, U.S. J. Polym. Sci., Part A: Polym. Chem. 2004, 42, 4882.
- 358. Zhang, H.Q.; Schubert, U.S. Chem. Commun. 2004, 858.
- 359. Uegaki, H.; Kotani, Y.; Kamigaito, M.; Sawamoto, M. Macromolecules 1997, 30, 2249.
- 360. Uegaki, H.; Kotani, Y.; Kamigaito, M.; Sawamoto, M. Macromolecules 1998, 31, 6756.
- 361. Granel, C.; Dubois, P.; Jerome, R.; Teyssie, P. Macromolecules 1996, 29, 8576.
- Uegaki, H.; Kamigaito, M.; Sawamoto, M. J. Polym. Sci., Part A: Polym. Chem. 1999, 37, 3003.
- 363. Tsuji, M.; Sakai, R.; Satoh, T.; Kaga, H.; Kakuchi, T. *Macromolecules* **2002**, *35*, 8255.
- 364. Kakuchi, T.; Tsuji, M.; Satoh, T. ACS Symp. Ser. 2003, 854, 206.
- 365. Pan, C.Y.; Lou, X.D. Macromol. Chem. Phys. 2000, 201, 1115.
- 366. Wickel, H.; Agarwal, S.; Greiner, A. Macromolecules 2003, 36, 2397.
- 367. Chung, I.S.; Matyjaszcwski, K. Macromolecules 2003, 36, 2995.
- 368. Tsarevsky, N.V.; Pintauer, T.; Matyjaszewski, K. Macromolecules 2004, 37, 9768.
- Ashford, E.J.; Naldi, V.; O'Dell, R.; Billingham, N.C.; Armes, S.P. Chem. Commun. 1999, 1285.
- 370. Perrier, S.; Armes, S.P.; Wang, X.S.; Malet, F.; Haddleton, D.M. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 1696.
- Coullerez, G.; Carlmark, A.; Malmstrom, E.; Jonsson, M. J. Phys. Chem. A 2004, 108, 7129.
- 372. Lee, S.B.; Russell, A.J.; Matyjaszewski, K. Biomacromolecules 2003, 4, 1386.
- 373. Li, Y.T.; Armes, S.P.; Jin, X.P.; Zhu, S.P. Macromolecules 2003, 36, 8268.
- 374. Perrier, S.; Haddleton, D.M. Macromol. Symp. 2002, 182, 261.
- 375. Zhu, C.Y.; Sun, F.; Zhang, M.; Jin, R. Polymer 2004, 45, 1141.
- 376. Nishikawa, T.; Kamigaito, M.; Sawamoto, M. Macromolecules 1999, 32, 2204.
- 377. Fuji, Y.; Ando, T.; Kamigaito, M.; Sawamoto, M. Macromolecules 2002, 35, 2949.
- 378. Gaynor, S.G.; Qiu, J.; Matyjaszewski, K. Macromolecules 1998, 31, 5951.
- 379. Min, K.; Gao, H.F.; Matyjaszewski, K. J. Am. Chem. Soc 2005, 127, 3825.
- Krstina, J.; Moad, G.; Rizzardo, E.; Winzor, C.L.; Berge, C.T.; Fryd, M. Macromolecules 1995, 28, 5381.
- 381. Matyjaszewski, K.; Gaynor, S.; Wang, J.-S. Macromolecules 1995, 28, 2093.
- Chiefari, J.; Chong, Y.K.; Ercole, F.; Krstina, J.; Jeffery, J.; Le, T.P.T.; Mayadunne, R.T.A.; Meijs, G.F.; Moad, C.L.; Moad, G.; Rizzardo, E.; Thang, S.H. Macromolecules 1998, 31, 5559.
- 383. Moad, G.; Chiefari, J.; Moad, C.L.; Postma, A.; Mayadunne, R.T.A.; Rizzardo, E.; Thang, S.H. Macromol. Symp. 2002, 182, 65.
- Chiefari, J.; Mayadunne, R.T.A.; Moad, C.L.; Moad, G.; Rizzardo, E.; Postma, A.; Skidmore, M.A.; Thang, S.H. *Macromolecules* 2003, 36, 2273.
- 385. Zhang, M.; Ray, W.II. Ind. Eng. Chem. Res. 2001, 40, 4336.
- 386. Wang, A.R.; Zhu, S.P. J. Polym. Sci., Part A: Polym. Chem. 2003, 41, 1553.
- 387. Wang, A.R.; Zhu, S. Macromol. Theory Simul. 2003, 12, 196.
- 388. Peklak, A.D.; Butte, A.; Storti, G.; Morbidelli, M. Macromol. Symp. 2004, 206, 481.
- 389. Shipp, D.A.; Matyjaszewski, K. Macromolecules 1999, 32, 2948.
- 390. Barner-Kowollik, C. Aust. J. Chem. 2001, 54, 343.
- 391. Barner-Kowollik, C.; Quinn, J.F.; Nguyen, T.L.U.; Heuts, J.P.A.; Davis, T.P. Macromolecules 2001, 34, 7849.
- 392. Vana, P.; Davis, T.P.; Barner-Kowollik, C. Macromol. Theory Simul. 2002, 11, 823.

- 393. Wulkow, M.; Busch, M.; Davis, T.P.; Barner-Kowollik, C. J. Polym. Sci., Part A: Polym. Chem. 2004, 42, 1441.
- 394. Chong, Y.K.; Krstina, J.; Le, T.P.T.; Moad, G.; Postma, A.; Rizzardo, E.; Thang, S.H. *Macromolecules* 2003, *36*, 2256.
- 395. Krstina, J.; Moad, C.L.; Moad, G.; Rizzardo, E.; Berge, C.T.; Fryd, M. Macromol. Symp. 1996, 111, 13.
- 396. Charmot, D.; Corpart, P.; Adam, H.; Zard, S.Z.; Biadatti, T.; Bouhadir, G. Macromol. Symp. 2000, 150, 23.
- 397. Rizzardo, E.; Chiefari, J.; Mayadunne, R.T.A.; Moad, G.; Thang, S.H. ACS Symp. Ser. 2000, 768, 278.
- 398. Chiefari, J.; Rizzardo, E. In *Handbook of Radical Polymerization*; Davis, T.P.; Matyjaszewski, K., Eds.; John Wiley & Sons: Hoboken, NY, 2002; p 263.
- 399. Barner-Kowollik, C.; Davis, T.P.; Heuts, J.P.A.; Stenzel, M.H.; Vana, P.; Whittaker, M. J. Polym. Sci., Part A: Polym. Chem. 2003, 41, 365.
- 400. McCormick, C.L.; Lowe, A.B. Acc. Chem. Res. 2004, 37, 312.
- 401. Moad, G.; Rizzardo, E.; Thang, S. Aust. J. Chem. 2005, 58, 379.
- 402. Barton, D.H.R.; McCombie, S.W. J. Chem. Soc., Perkin Trans. 1 1975, 1574.
- 403. Barton, D.H.R.; Parckh, S.I.; Tsc, C.L. Tetrahedron Lett. 1993, 34, 2733.
- 404. Forbes, J.E.; Zard, S.Z. Tetrahedron Lett. 1989, 30, 4367.
- 405. Delduc, P.; Tailhan, C.; Zard, S.Z. J. Chem. Soc., Chem. Commun. 1988, 308.
- 406. Quiclet-Sire, B.; Zard, S.Z. Pure Appl. Chem. 1997, 69, 645.
- 407. Zard, S.Z. Angew. Chem. Int. Ed. Engl. 1997, 36, 672.
- 408. Lc, T.P.; Moad, G.; Rizzardo, E.; Thang, S.H. Int. Patent Appl. WO 9801478, 1998 (*Chem. Abstr.* (1997) 128: 115390).
- 409. Moad, G.; Chiefari, J.; Chong, Y.K.; Krstina, J.; Postma, A.; Mayadunne, R.T.A.; Rizzardo, E.; Thang, S.H. Polym. Int. 2000, 49, 933.
- 410. Rizzardo, E.; Chiefari, J.; Chong, Y.K.; Ercole, F.; Krstina, J.; Jeffery, J.; Le, T.P.T.; Mayadunne, R.T.A.; Meijs, G.F.; Moad, C.L.; Moad, G.; Thang, S.H. *Macromol. Symp.* **1999**, *143*, 291.
- 411. Mayadunne, R.T.A.; Rizzardo, E.; Chiefari, J.; Chong, Y.K.; Moad, G.; Thang, S.H. Macromolecules 1999, 32, 6977.
- 412. Destarae, M.; Charmot, D.; Franck, X.; Zard, S.Z. Macromol. Rapid Commun. 2000, 21, 1035.
- 413. Destarac, M.; Bzducha, W.; Taton, D.; Gauthier-Gillaizeau, I.; Zard, S.Z. Macromol. Rapid Commun. 2002, 23, 1049.
- 414. Benaglia, M.; Rizzardo, E.; Alberti, A.; Guerra, M. Macromolecules 2005, 38, 3129.
- 415. Farmer, S.C.; Patten, T.E. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 555.
- 416. Coote, M.L. Macromolecules 2004, 37, 5023.
- 417. Coote, M.L.; Radom, L. J. Am. Chem. Soc. 2003, 125, 1490.
- 418. Thomas, D.B.; Convertine, A.J.; Myrick, L.J.; Scales, C.W.; Smith, A.E.; Lowe, A.B.; Vasilieva, Y.A.; Ayres, N.; McCormick, C.L. *Macromolecules* 2004, 37, 8941.
- Mertoglu, M.; Laschewsky, A.; Skrabania, K.; Wieland, C. *Macromolecules* 2005, 38, 3601.
- 420. Sumerlin, B.S.; Lowe, A.B.; Thomas, D.B.; McCormick, C.L. *Macromolecules* 2003, *36*, 5982.
- 421. Vana, P.; Albertin, L.; Barner, L.; Davis, T.P.; Barner-Kowollik, C. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 4032.
- 422. Ah Toy, A.; Vana, P.; Davis, T.P.; Barner-Kowollik, C. Macromolecules 2004, 37, 744.
- 423. Lansalot, M.; Davis, T.P.; Heuts, J.P.A. Macromolecules 2002, 35, 7582.
- 424. Tang, C.B.; Kowalewski, T.; Matyjaszewski, K. Macromolecules 2003, 36, 8587.

- 425. Chong, Y.K.; Le, T.P.T.; Moad, G.; Rizzardo, E.; Thang, S.H. *Macromolecules* **1999**, *32*, 2071.
- 426. Saricilar, S.; Knott, R.; Barner-Kowollik, C.; Davis, T.P.; Heuts, J.P.A. Polymer 2003, 44, 5169,
- 427. Krasia, T.; Soula, R.; Boerner, H.G.; Schlaad, H. Chem. Commun. 2003, 538.
- 428. Donovan, M.S.; Lowe, A.B.; Sumerlin, B.S.; McCormick, C.L. Macromolecules 2002, 35, 4123.
- 429. Ganachaud, F.; Monteiro, M.J.; Gilbert, R.G.; Dourges, M.A.; Thang, S.H.; Rizzardo, E. Macromolecules 2000, 33, 6738.
- 430. Convertine, A.J.; Sumerlin, B.S.; Thomas, D.B.; Lowe, A.B.; McCormick, C.L. Macromolecules 2003, 36, 4679.
- 431. Perrier, S.; Barner-Kowollik, C.; Quinn, J.F.; Vana, P.; Davis, T.P. Macromolecules 2002, 35, 8300.
- 432. An, Q.F.; Qian, J.W.; Yu, L.Y.; Luo, Y.W.; Liu, X.Z. J. Polym. Sci., Part A: Polym. Chem. 2005, 43, 1973.
- 433. Hao, X.J.; Heuts, J.P.A.; Barner-Kowollik, C.; Davis, T.P.; Evans, E. J. Polym. Sci., Part A: Polym. Chem. 2003, 41, 2949.
- 434. de Brouwer, H.; Tsavalas, J.G.; Schork, F.J.; Monteiro, M.J. Macromolecules 2000, 33, 9239.
- 435. Ladaviere, C.; Doerr, N.; Claverie, J.P. Macromolecules 2001, 34, 5370.
- 436. Zhu, J.; Zhu, X.; Zhou, D.; Chen, J. e-Polymers 2003, [43].
- 437. Zhu, J.; Zhu, X.L.; Cheng, Z.P.; Liu, F.; Lu, J.M. Polymer 2002, 43, 7037.
- 438. Zhu, J.; Zhou, D.; Zhu, X.L.; Chen, G.J. J. Polym. Sci., Part A: Polym Chem 2004, 42. 2558.
- 439. Zhu, J.; Zhu, M.L.; Zhou, D.; Chen, J.Y.; Wang, X.Y. Eur. Polym. J. 2004, 40, 743.
- 440. Perrier, S.; Takolpuckdee, P.; Westwood, J.; Lewis, D.M. Macromolecules 2004, 37, 2709.
- 441. Takolpuckdee, P.; Mars, C.A.; Perrier, S.; Archibald, S.J. Macromolecules 2005, 38, 1057.
- 442. Goto, A.; Sato, K.; Tsujii, Y.; Fukuda, T.; Moad, G.; Rizzardo, E.; Thang, S.H. Macromolecules 2001, 34, 402.
- 443. He, T.; Zou, Y.F.; Pan, C.Y. Polym. J. 2002, 34, 138.
- 444. Kanagasabapathy, S.; Sudalai, A.; Benicewicz, B.C. Macromol. Rapid Commun. 2001, 22, 1076.
- 445. Dureault, A.; Taton, D.; Destarac, M.; Leising, F.; Gnanou, Y. Macromolecules 2004, 37, 5513.
- 446. Donovan, M.S.; Lowe, A.B.; Sanford, T.A.; McCormick, C.L. J. Polym. Sci., Part A: Polym. Chem. 2003, 41, 1262.
- 447. Taton, D.; Wilczewska, A.Z.; Destarae, M. Macromol. Rapid Commun. 2001, 22, 1497.
- 448. Bussels, R.; Bergman-Gottgens, C.; Meuldijk, J.; Koning, C. Macromolecules 2004, 37, 9299.
- 449. Moad, G.; Mayadunne, R.T.A.; Rizzardo, E.; Skidmore, M.; Thang, S. Macromol. Symp. 2003, 192, 1.
- 450. Moad, G.; Chong, Y.K.; Rizzardo, E.; Postma, A.; Thang, S.H. Polymer 2005, 46, 8458-8468.
- 451. Ray, B.; Isobe, Y.; Matsumoto, K.; Habaue, S.; Okamoto, Y.; Kamigaito, M.; Sawamoto, M. Macromolecules 2004, 37, 1702.
- 452. Theis, A.; Feldermann, A.; Charton, N.; Stenzel, M.H.; Davis, T.P.; Barner-Kowollik, C. Macromolecules 2005, 38, 2595.
- 453. Quinn, J.F.; Rizzardo, E.; Davis, T.P. Chem. Commun. 2001, 1044.
- 454. Lai, J.T.; Filla, D.; Shea, R. Macromolecules 2002, 35, 6754. 455. Lima, V.; Jiang, X.L.; Brokken-Zijp, J.; Schoenmakers, P.J.; Klumperman, B.; Van Der Linde, R. J. Polym. Sci., Part A: Polym. Chem. 2005, 43, 959.

- 456. Convertine, A.J.; Lokitz, B.S.; Lowe, A.B.; Scales, C.W.; Myrick, L.J.; McCormick, C.L. Macromol. Rapid Commun. 2005, 26, 791.
- 457. Mayadunne, R.T.A.; Rizzardo, E.; Chiefari, J.; Krstina, J.; Moad, G.; Postma, A.; Thang, S.H. *Macromolecules* **2000**, *33*, 243.
- 458. You, Y.Z.; Hong, C.Y.; Wang, W.P.; Lu, W.Q.; Pan, C.Y. *Macromolecules* **2004**, *37*, 9761.
- 459. Loiseau, J.; Doeerr, N.; Suau, J.M.; Egraz, J.B.; Llauro, M.F.; Ladaviere, C.; Claveric, J. *Macromolecules* **2003**, *36*, 3066.
- 460. Convertine, A.J.; Ayres, N.; Scales, C.W.; Lowe, A.B.; McCormick, C.L. Biomacromolecules 2004, 5, 1177.
- 461. Ferguson, C.J.; Hughes, R.J.; Pham, B.T.T.; Hawkett, B.S.; Gilbert, R.G.; Serelis, A.K.; Such, C.H. *Macromolecules* 2002, 35, 9243.
- Pham, B.T.T.; Nguyen, D.; Ferguson, C.J.; Hawkett, B.S.; Serelis, A.K.; Such, C.H. Macromolecules 2003, 36, 8907.
- 463. Moad, G.; Li, G.; Rizzardo, E.; S.H., T.; Pfaendner, R.; Wertmer, H. Polym. Prepr. (Am. Chem. Soc., Div. Polym. Chem.) 2005, 46(2), in press.
- 464. Postma, A.; Davis, T.P.; Moad, G.; O'Shea, M. Macromolecules 2005, 38, 5371.
- 465. Monteiro, M.J.; Adamy, M.M.; Leeuwen, B.J.; van Herk, A.M.; Destarae, M. Macromolecules 2005, 38, 1538.
- 466. Adamy, M.; van Herk, A.M.; Destarac, M.; Monteiro, M.J. *Macromolecules* **2003**, *36*, 2293.
- 467. Destarac, M.; Brochon, C.; Catala, J.-M.; Wilczewska, A.; Zard, S.Z. Macromol. Chem. Phys. 2002, 203, 2281.
- 468. Monteiro, M.J.; de Barbeyrac, J. Macromol. Rapid Commun. 2002, 23, 370.
- 469. Chiefari, J.; Mayadunne, R.T.; Moad, G.; Rizzardo, E.; Thang, S.H. PCT Int. Appl. WO 9931144, 1999 (*Chem. Abstr.* 131:45250).
- 470. Stenzel, M.H.; Cummins, L.; Roberts, G.E.; Davis, T.P.; Vana, P.; Barner-Kowollik, C. Macromol. Chem. Phys. 2003, 204, 1160.
- 471. Simms, R.W.; Davis, T.P.; Cunningham, M.F. Macromol. Rapid Commun. 2005, 26, 592.
- 472. Russum, J.P.; Barbre, N.D.; Jones, C.W.; Schork, F.J. J. Polym. Sci., Part A: Polym. Chem. 2005, 43, 2188.
- 473. Favier, A.; Barner-Kowollik, C.; Davis, T.P.; Stenzel, M.H. Macromol. Chem. Phys. 2004, 205, 925.
- 474. Schilli, C.; Lanzendoerfer, M.G.; Mueller, A.II.E. Macromolecules 2002, 35, 6819.
- 475. Schilli, C.M.; Muller, A.H.E.; Rizzardo, E.; Thang, S.H.; Chong, Y.K. ACS Symp. Ser. 2003, 854, 603.
- 476. Schilli, C.M.; Zhang, M.F.; Rizzardo, E.; Thang, S.H.; Chong, Y.K.; Edwards, K.; Karlsson, G.; Muller, A.H.E. *Macromolecules* 2004, *37*, 7861.
- 477. Meijer, J.; Vermeer, P.; Brandsma, L. Recueil 1973, 92, 601.
- 478. Mayadunne, R.A.; Moad, G.; Rizzardo, E. Tetrahedron Lett. 2002, 43, 6811.
- 479. Oae, S.; Yagihara, T.; Okabe, T. Tetrahedron 1972, 28, 3203.
- 480. Kanagasabapathy, S.; Sudalai, A.; Benicewicz, B.C. *Tetrahedron Lett.* 2001, 42, 3791.
- 481. Thang, S.H.; Chong, Y.K.; Mayadunne, R.T.A.; Moad, G.; Rizzardo, E. *Tetrahedron Lett.* **1999**, *40*, 2435.
- 482. Bouhadir, G.; Legrand, N.; Quiclet-Sire, B.; Zard, S.Z. Tetrahedron Lett. 1999, 40, 277.
- 483. Wager, C.M.; Haddleton, D.M.; Bon, S.A.F. Eur. Polym. J. 2004, 40, 641.
- 484. Sudalai, A.; Kanagasabapathy, S.; Benicewicz, B.C. Org. Lett. 2000, 2, 3213.
- 485. Perrier, S.; Takolpuckdee, P.; Mars, C.A. Macromolecules 2005, 38, 2033.
- 486. Moad, G.; Mayadunne, R.T.A.; Rizzardo, E.; Skidmore, M.; Thang, S. ACS Symp. Ser. 2003, 854, 520.

- 487. Chernikova, E.; Morozov, A.; Leonova, E.; Garina, E.; Golubev, V.; Bui, C.O.; Charleux, B. *Macromolecules* 2004, 37, 6329.
- 488. McLeary, J.B.; Calitz, F.M.; McKenzie, J.M.; Tonge, M.P.; Sanderson, R.D.; Klumperman, B. *Macromolecules* **2005**, *38*, 3151.
- 489. Monteiro, M.J.; de Brouwer, II. Macromolecules 2001, 34, 349.
- 490. Barner-Kowollik, C.; Davis, T.P. Macromol. Theory Simul. 2001, 10, 255.
- 491. Kwak, Y.; Goto, A.; Fukuda, T. Macromolecules 2004, 37, 1219.
- 492. Kwak, Y.; Goto, A.; Tsujii, Y.; Murata, Y.; Komatsu, K.; Fukuda, T. Macromolecules 2002, 38, 3026.
- 493. McLeary, J.B.; Calitz, F.M.; McKenzie, J.M.; Tonge, M.P.; Sanderson, R.D.; Klumperman, B. *Macromolecules* **2004**, *37*, 2383.
- 494. Plummer, R.; Goh, Y.-K.; Whittaker, A.K.; Monteiro, M.J. Macromolecules 2005, 38, 5352
- 495. Quinn, J.F.; Barner, L.; Barner-Kowollik, C.; Rizzardo, E.; Davis, T.P. Macromolecules 2002, 35, 7620.
- 496. You, Y.Z.; Hong, C.Y.; Bai, R.K.; Pan, C.Y.; Wang, J. Macromol. Chem. Phys. 2002, 203, 477.
- 497. Quinn, J.F.; Barner, L.; Davis, T.P.; Thang, S.H.; Rizzardo, E. Macromol. Rapid Commun. 2002, 23, 717.
- 498. Quinn, J.F.; Barner, L.; Rizzardo, E.; Davis, T.P. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 19.
- 499. Barner, L.; Quinn, J.F.; Barner-Kowollik, C.; Vana, P.; Davis, T.P. Eur. Polym. J. 2003, 39, 449.
- 500. Bai, R.K.; You, Y.Z.; Zhong, P.; Pan, C.Y. Macromol. Chem. Phys. 2001, 202, 1970.
- 501. Hong, C.Y.; You, Y.Z.; Bai, R.K.; Pan, C.Y.; Borjihan, G. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 3934.
- 502. Bai, R.K.; You, Y.Z.; Pan, C.Y. Macromol. Rapid Commun. 2001, 22, 315.
- 503. You, Y.Z.; Bai, R.K.; Pan, C.Y. Macromol. Chem. Phys. 2001, 202, 1980.
- 504. Chen, G.; Zhu, X.; Zhu, J.; Cheng, Z. Macromol. Rapid Commun. 2004, 25, 818.
- 505. Lowe, A.B.; Sumerlin, B.S.; Donovan, M.S.; Thomas, D.B.; Hennauxz, P.; McCormick, C.L. ACS Symp. Ser. 2003, 854, 586.
- 506. Lowe, A.B.; McCormick, C.L. Aust. J. Chem. 2002, 55, 367.
- 507. Baussard, J.F.; Habib-Jiwan, J.L.; Laschewsky, A.; Mertoglu, M.; Storsberg, J. *Polymer* **2004**, *45*, 3615.
- 508. Perrier, S.; Davis, T.P.; Carmichael, A.J.; Haddleton, D.M. Chem. Commun. 2002, 2226.
- 509. Arita, T.; Buback, M.; Janssen, O.; Vana, P. Macromol. Rapid Commun. 2004, 25, 1376.
- 510. Arita, T.; Beuermann, S.; Buback, M.; Vana, P. Macromol. Mater. Eng. 2005, 290, 283.
- 511. Thomas, D.B.; Convertine, A.J.; Hester, R.D.; Lowe, A.B.; McCormick, C.L. *Macromolecules* 2004, 37, 1735.
- 512. Monteiro, M.J.; Bussels, R.; Beuermann, S.; Buback, M. Aust. J. Chem. 2002, 55, 433.
- 513. Rzayev, J.; Penelle, J. Angew. Chem. Int. Ed. Engl. 2004, 43, 1691.
- 514. Chong, Y.K.; Moad, G.; Rizzardo, M.; Skidmore, M.A.; Thang, S. In 27th Australian Polymer Symposium; RACI Polymer Division: Adelaide, SA, 2004; p C1/3.
- 515. Ray, B.; Isobe, Y.; Morioka, K.; Habaue, S.; Okamoto, Y.; Kamigaito, M.; Sawamoto, M. *Macromolecules* **2003**, *36*, 543.
- 516. Lutz, J.F.; Jakubowski, W.; Matyjaszewski, K. Macromol. Rapid Commun. 2004, 25, 486.
- 517. Lutz, J.F.; Neugebauer, D.; Matyjaszewski, K. J. Am. Chem. Soc. 2003, 125, 6986.

- 518. Kirci, B.; Lutz, J.F.; Matyjaszewski, K. Macromolecules 2002, 35, 2448.
- 519. Lutz, J.F.; Kirci, B.; Matyjaszewski, K. Macromolecules 2003, 36, 3136.
- 520. Monteiro, M.J.; Hodgson, M.; De Brouwer, H. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 3864.
- 521. Monteiro, M.J.; de Barbeyrac, J. Macromolecules 2001, 34, 4416.
- 522. Monteiro, M.J.; Sjoberg, M.; van der Vlist, J.; Gottgens, C.M. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 4206.
- 523. Uzulina, I.; Kanagasabapathy, S.; Claverie, J. Macromol. Symp. 2000, 150, 33.
- 524. Smulders, W.; Gilbert, R.G.; Monteiro, M.J. Macromolecules 2003, 36, 4309.
- 525. Prescott, S.W.; Ballard, M.J.; Rizzardo, E.; Gilbert, R.G. Aust. J. Chem. 2002, 55, 415.
- 526. Prescott, S.W.; Ballard, M.J.; Rizzardo, E.; Gilbert, R.G. Macromolecules 2002, 35, 5417.
- 527. Nozari, S.; Tauer, K. Polymer 2005, 46, 1033.
- 528. Prescott, S.W. Macromolecules 2003, 36, 9608.
- 529. Such, C.H.; Rizzardo, E.; Serelis, A.K.; Hawkett, B.S.; Gilbert, R.G.; Ferguson, C.J.; Hughes, R.J. WO 03055919, 2003 (*Chem. Abstr.* 139, 101540v).
- 530. Lansalot, M.; Farcet, C.; Charleux, B.; Vairon, J.P.; Pirri, R. Macromolecules 1999, 32, 7354.
- 531. Russum, J.P.; Jones, C.W.; Schork, F.J. *Macromol. Rapid Commun.* 2004, 25, 1064.
- 532. Tsavalas, J.G.; Schork, F.J.; dc Brouwer, H.; Monteiro, M.J. *Macromolecules* 2001, 34, 3938.
- 533. Haszeldine, R.N. J. Chem. Soc. 1949, 2859.
- 534. Tatemoto, M. Kobunshi Ronbunshu 1992, 49, 765.
- 535. Gaynor, S.; Wang, J.-S.; Matyjaszewski, K. Macromolecules 1995, 28, 8051.
- 536. Goto, A.; Ohno, K.; Fukuda, T. Macromolecules 1998, 31, 2809.
- 537. Ueda, N.; Kamigaito, M.; Sawamoto, M. Polym. Prepr. Japan 1996, 45, E622.
- 538. Iovu, M.C.; Matyjaszewski, K. Macromolecules 2003, 36, 9346.
- 539. Ameduri, B.; Boutevin, B. J. Fluorine Chem. 1999, 100, 97.
- 540. Goto, A.; Kwak, Y.; Fukuda, T.; Yamago, S.; Iida, K.; Nakajima, M.; Yoshida, J. J. Am. Chem. Soc. 2003, 125, 8720.
- 541. Yamago, S.; Iida, K.; Yoshida, J. J. Am. Chem. Soc. 2002, 124, 2874.
- 542. Yamago, S.; Iida, K.; Yoshida, J.-i. J. Am. Chem. Soc. 2002, 124, 13666.
- 543. Yamago, S.; Iida, K.; Nakajima, M.; Yoshida, J. Macromolecules 2003, 36, 3793.
- 544. Yamago, S.; Ray, B.; Iida, K.; Yoshida, J.; Tada, T.; Yoshizawa, K.; Kwak, Y.; Goto, A.; Fukuda, T. J. Am. Chem. Soc. 2004, 126, 13908.
- 545. Kwak, Y.W.; Goto, A.; Fukuda, T.; Yamago, S.; Ray, B. Z. Phys. Chem. 2005, 219, 283.
- 546. Haddleton, D.M.; Crossman, M.C.; Hunt, K.H.; Topping, C.; Waterson, C.; Suddaby, K.G. *Macromolecules* **1997**, *30*, 3992.
- 547. Klumperman, B.; Chambard, G.; Brinkhuls, R.H.G. ACS Symp. Ser. 2003, 854, 180.
- 548. Shinoda, H.; Matyjaszewski, K.; Okrasa, L.; Mierzwa, M.; Pakula, T. Macromolecules 2003, 36, 4772.
- 549. Feldermann, A.; Toy, A.A.; Phan, H.; Stenzel, M.H.; Davis, T.P.; Barner-Kowollik, C. Polymer 2004, 45, 3997.
- 550. Benoit, D.; Hawker, C.J.; Huang, E.E.; Lin, Z.Q.; Russell, T.P. *Macromolecules* **2000**, *33*, 1505.
- 551. De Brouwer, H.; Schellekens, M.A.J.; Klumperman, B.; Monteiro, M.J.; German, A.L. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 3596.
- 552. Du, F.S.; Zhu, M.Q.; Guo, H.Q.; Li, Z.C.; Li, F.M.; Kamachi, M.; Kajiwara, A. Macromolecules 2002, 35, 6739.
- Zaremski, M.Y.; Plutalova, A.V.; Lachinov, M.B.; Golubev, V.B. Macromolecules 2000, 33, 4365.

- 554. Davis, K.A.; Matyjasewski, K. Adv. Polym. Sci. 2002, 159, 2.
- 555. Gray, M.K.; Zhou, H.Y.; Nguyen, S.T.; Torkelson, J.M. Polymer 2004, 45, 4777.
- 556. Hawker, C.J.; Elce, E.; Dao, J.; Volksen, W.; Russell, T.P.; Barclay, G.G. Macromolecules 1996, 29, 2686.
- 557. Gray, M.K.; Zhou, H.Y.; Nguyen, S.T.; Torkelson, J.M. Macromolecules 2004, 37, 5586.
- 558. Yoshida, E.; Takiguchi, Y. Polym. J. 1999, 31, 429.
- 559. Butz, S.; Baethge, II.; Schmidt-Naake, G. Macromol. Rapid Commun. 1997, 18, 1049.
- Fukuda, T.; Terauchi, T.; Goto, A.; Tsujii, Y.; Miyamoto, T. *Macromolecules* 1996, 29, 3050.
- Cuervo-Rodriguez, R.; Bordege, V.; Fernandez-Monreal, M.C.; Fernandez-Garcia, M.; Madruga, E.L. J. Polym. Sci., Part A: Polym. Chem. 2004, 42, 4168.
- 562. Baumert, M.; Mülhaupt, R. Macromol. Rapid Commun. 1997, 18, 787.
- Taube, C.; Garcia, M.F.; Schmidt-Naake, G.; Fischer, H. Macromol. Chem. Phys. 2002, 203, 2665.
- Baethge, H.; Butz, S.; Schmidt-Naake, G. Macromol. Rapid Commun. 1997, 18, 911.
- Braunecker, W.A.; Tsarevsky, N.V.; Pintauer, T.; Gil, R.R.; Matyjaszewski, K. Macromolecules 2005, 38, 4081.
- 566. Venkatesh, R.; Vergouwen, F.; Klumperman, B. Macromol. Chem. Phys. 2005, 206, 547.
- 567. Arehart, S.V.; Matyjaszewski, K. Macromolecules 1999, 32, 2221.
- 568. Qin, S.H.; Saget, J.; Pyun, J.R.; Jia, S.J.; Kowalewski, T.; Matyjaszewski, K. Macromolecules 2003, 36, 8969.
- 569. Kotani, Y.; Kamigaito, M.; Sawamoto, M. Macromolecules 1998, 31, 5582.
- 570. Hong, S.C.; Lutz, J.F.; Inoue, Y.; Strissel, C.; Nuyken, O.; Matyjaszewski, K. Macromolecules 2003, 36, 1075.
- 571. Matyjaszewski, K.; Shipp, D.A.; McMurtry, G.P.; Gaynor, S.G.; Pakula, T. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 2023.
- 572. Borkar, S.; Scn, A. Macromolecules 2005, 38, 3029.
- 573. Gao, B.; Chen, X.; Ivan, B.; Kops, J.; Batsberg, W. Polym. Bull. 1997, 39(5), 559.
- 574. Lutz, J.F.; Pakula, T.; Matyjaszcwski, K. ACS Symp. Ser. 2003, 854, 268.
- 575. Tsarevsky, N.V.; Sarbu, T.; Gobelt, B.; Matyjaszewski, K. Macromolecules 2002, 35, 6142.
- 576. Luo, Y.W.; Liu, X.Z. J. Polym. Sci., Part A: Polym Chem 2004, 42, 6248.
- 577. Savariar, E.N.; Thayumanavan, S. J. Polym. Sci., Part A: Polym Chem 2004, 42, 6340.
- 578. Chernikova, E.; Terpugova, P.; Bui, C.; Charleux, B. Polymer 2003, 44, 4101.
- 579. Davies, M.C.; Dawkins, J.V.; Hourston, D.J. Polymer 2005, 46, 1739.
- 580. Zhu, M.Q.; Wei, L.H.; Li, M.; Jiang, L.; Du, F.S.; Li, Z.C.; Li, F.M. Chem. Commun. 2001, 365.
- Beyou, E.; Chaumont, P.; Chauvin, F.; Devaux, C.; Zydowicz, N. Macromolecules 1998, 31, 6828.
- 582. Harth, E.; Hawker, C.J.; Fan, W.; Waymouth, R.M. *Macromolecules* **2001**, *34*, 3856.
- 583. Bon, S.A.F.; Chambard, G.; German, A.L. Macromolecules 1999, 32, 8269.
- 584. Turro, N.J.; Lem, G.; Zavarine, I.S. Macromolecules 2000, 33, 9782.
- 585. Otsuka, H.; Aotani, K.; Higaki, Y.; Takahara, A. J. Am. Chem. Soc. 2003, 125, 4064.
- 586. Higaki, Y.; Otsuka, H.; Takahara, A. Macromolecules 2004, 37, 1696.
- 587. Chen, X.; Gao, B.; Kops, J.; Batsberg, W. Polymer 1998, 39, 911.
- 588. Hawker, C.J.; Malmstroem, E.E.; Frechet, J.M.J.; Leduc, M.R.; Grubbs, R.B.; Barclay, G.G. ACS Symp. Ser. 1998, 685, 433.

- 589. Kobatake, S.; Harwood, H.J.; Quirk, R.P.; Priddy, D.B. Macromolecules 1997, 30, 4238.
- 590. Mather, B.D.; Lizotte, J.R.; Long, T.E. Macromolecules 2004, 37, 9331.
- 591. Miura, Y.; Yoshida, Y. Macromol. Chem. Phys. 2002, 203, 879.
- 592. Coessens, V.; Pintauer, T.; Matyjaszewski, K. Prog. Polym. Sci. 2001, 26, 337.
- 593. Snijder, A.; Klumperman, B.; Van der Linde, R. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 2350.
- 594. Tcoh, S.K.; Ravi, P.; Dai, S.; Tam, K.C. J. Phys. Chem. B 2005, 109, 4431.
- 595. Zhou, P.; Chen, G.Q.; Hong, H.; Du, F.S.; Li, Z.C.; Li, F.M. *Macromolecules* **2000**, *33*, 1948.
- Bon, S.A.F.; Morsley, S.R.; Waterson, C.; Haddleton, D.M. Macromolecules 2000, 33, 5819.
- 597. Bon, S.A.F.; Steward, A.G.; Haddleton, D.M. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 2678.
- 598. Bielawski, C.W.; Jethmalani, J.M.; Grubbs, R.H. Polymer 2003, 44, 3721.
- 599. Ando, T.; Kamigaito, M.; Sawamoto, M. Macromolecules 1998, 31, 6708.
- 600. Coessens, V.; Pyun, J.; Miller, P.J.; Gaynor, S.G.; Matyjaszewski, K. Macromol. Rapid Commun. 2000, 21, 103.
- 601. Koulouri, E.G.; Kallitsis, J.K.; Hadziioannou, G. Macromolecules 1999, 32, 6242.
- 602. Chambard, G.; Klumperman, B.; German, A.L. Macromolecules 2000, 33, 4417.
- 603. Lutz, J.F.; Matyjaszewski, K. J. Polym. Sci., Part A: Polym. Chem. 2005, 43, 897.
- 604. Lutz, J.F.; Matyjaszcwski, K. Macromol. Chem. Phys. 2002, 203, 1385.
- 605. Coessens, V.; Matyjaszewski, K. Macromol. Rapid Commun. 1999, 20, 66.
- 606. Coessens, V.; Nakagawa, Y.; Matyjaszewski, K. Polym. Bull. 1998, 40, 135.
- 607. Matyjaszcwski, K.; Nakagawa, Y.; Gaynor, S.G. Macromol. Rapid Commun. 1997, 18, 1057.
- 608. Postma, A.; Moad, G.; Davis, T.P.; O'Shea, M. React. Funct. Polym. 2005, in press.
- 609. Coessens, V.; Matyjaszewski, K. J. Macromol. Sci. Chem. 1999, A36, 811.
- 610. Coessens, V.; Matyjaszewski, K. Macromol. Rapid Commun. 1999, 20, 127.
- 611. Zhang, H.; Jiang, X.; van der Linde, R. Polymer 2004, 45, 1455.
- 612. Muchlebach, A.; Rime, F. J. Polym. Sci., Part A: Polym. Chem. 2003, 41, 3425.
- 613. Malz, H.; Komber, H.; Voigt, D.; Hopfe, I.; Pionteck, J. Macromol. Chem. Phys. 1999, 200, 642.
- 614. Coessens, V.; Matyjaszewski, K. J. Macromol. Sci. Chem. 1999, A36, 653.
- 615. Matyjaszewski, K. ACS Symp. Ser. 2000, 768, 2.
- 616. Matyjaszewski, K.; Beers, K.L.; Kern, A.; Gaynor, S.G. J. Polym. Sci., Part A: Polym. Chem. 1998, 36, 823.
- 617. Baek, K.Y.; Kamigaito, M.; Sawamoto, M. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 1937.
- 618. Sarbu, T.; Lin, K.Y.; Spanswick, J.; Gil, R.R.; Siegwart, D.J.; Matyjaszewski, K. *Macromolecules* **2004**, *37*, 9694.
- 619. Zhang, X.; Matyjaszcwski, K. Macromolecules 1999, 32, 7349.
- 620. Haddleton, D.M.; Waterson, C. Macromolecules 1999, 32, 8732.
- 621. Ji, S.; Hoye, T.R.; Macosko, C.W. Macromolecules 2005, 38, 4679
- 622. Carrot, G.; Hilborn, J.; Hedrick, J.L.; Trollsas, M. Macromolecules 1999, 32, 5171.
- 623. Castro, E.A. Chem. Rev. 1999, 99, 3505.
- 624. Kulkarni, S.; Schilli, C.; Muller, A.H.E.; Hoffman, A.S.; Stayton, P.S. *Bioconjugate Chemistry* **2004**, *15*, 747.
- 625. Chen, M.; Ghiggino, K.P.; Thang, S.H.; White, J.; Wilson, G.J. J. Org. Chem 2005, 70, 1844.
- 626. Mayadunne, R.T.A.; Jeffery, J.; Moad, G.; Rizzardo, E. Macromolecules 2003, 36, 1505.
- 627. Wang, Z.M.; He, J.P.; Tao, Y.F.; Yang, L.; Jiang, H.J.; Yang, Y.L. *Macromolecules* **2003**, *36*, 7446.

- 628. Sumerlin, B.S.; Lowe, A.B.; Stroud, P.A.; Zhang, P.; Urban, M.W.; McCormick, C.L. *Langmuir* **2003**, *19*, 5559.
- 629. Riess, G. Prog. Polym. Sci. 2003, 28, 1107.
- 630. Mariani, M.; Lelli, M.; Sparnacci, K.; Laus, M. J. Polym. Sci., Part A: Polym. Chem. 1999, 37, 1237.
- 631. Ohno, K.; Izu, Y.; Tsujii, Y.; Fukuda, T.; Kitano, H. Eur. Polym. J. 2004, 40, 81.
- 632. Jousset, S.; Hammouch, S.O.; Catala, J.M. Macromolecules 1997, 30, 6685.
- 633. Lacroix-Desmazes, P.; Delair, T.; Pichot, C.; Boutevin, B. J. Polym. Sci. Pol. Chem. 2000, 38, 3845.
- 634. Gabaston, L.I.; Furlong, S.A.; Jackson, R.A.; Armes, S.P. Polymer 1999, 40, 4505.
- 635. Stancik, C.M.; Lavoie, A.R.; Schutz, J.; Achurra, P.A.; Lindner, P.; Gast, A.P.; Waymouth, R.M. *Langmuir* **2004**, *20*, 596.
- 636. Burguiere, C.; Pascual, S.; Bui, C.; Vairon, J.P.; Charleux, B.; Davis, K.A.; Matyjaszewski, K.; Betremieux, I. *Macromolecules* 2001, 34, 4439.
- 637. Zhang, X.; Matyjaszewski, K. Macromolecules 1999, 32, 1763.
- 638. Mitsukami, Y.; Donovan, M.S.; Lowe, A.B.; McCormick, C.L. *Macromolecules* 2001, *34*, 2248.
- 639. Sumerlin, B.S.; Lowe, A.B.; Thomas, D.B.; Convertine, A.J.; Donovan, M.S.; McCormick, C.L. J. Polym. Sci., Part A: Polym. Chem. 2004, 42, 1724.
- 640. Chen, X.Y.; Gao, B.; Kops, J.; Batsberg, W. Polymer 1998, 39, 911.
- 641. Hawker, C.J.; Hedrick, J.L.; Malmstroem, E.E.; Trollss, M.; Mecerreyes, D.; Moineau, G.; Dubois, P.; Jerome, R. *Macromolecules* **1998**, *31*, 213.
- 642. Kobatake, S.; Harwood, H.J.; Quirk, R.P.; Priddy, D.B. Macromolecules 1999, 32, 10.
- 643. Lu, G.Q.; Jia, Z.F.; Yi, W.; Huang, J.L. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 4404.
- 644. Li, Z.Y.; Lu, G.Q.; Huang, J.L. J. Appl. Polym. Sci. 2004, 94, 2280.
- 645. Jankova, K.; Chen, X.; J., k.; Batsberg, W. Macromolecules 1998, 31, 538.
- 646. Save, M.; Weaver, J.V.M.; Armes, S.P.; McKenna, P. Macromolecules 2002, 35, 1152.
- 647. Biclawski, C.W.; Morita, T.; Grubbs, R.H. Macromolecules 2000, 33, 678.
- 648. Destarac, M.; Boutevin, B. Macromol. Rapid Commun. 1999, 20, 641.
- 649. Destarac, M.; Pees, B.; Boutevin, B. Macromol. Chem. Phys. 2000, 201, 1189.
- 650. Semsarzadeh, M.A.; Mirzaei, A.; Vasheghani-Farahani, E.; Haghighi, M.N. Eur. Polym. J. 2003, 39, 2193.
- 651. Shi, L.J.; Chapman, T.M.; Beckman, E.J. Macromolecules 2003, 36, 2563.
- 652. Hong, C.Y.; You, Y.Z.; Pan, C.Y. J. Polym. Sci., Part A: Polym Chem 2004, 42, 4873.
- 653. Ma, Z.; Lacroix-Desmazes, P. Polymer 2004, 45, 6789.
- 654. Hales, M.; Barner-Kowollik, C.; Davis, T.P.; Stenzel, M.H. *Langmuir* **2004**, *20*, 10809.
- 655. Tsujii, Y.; Ejaz, M.; Sato, K.; Goto, A.; Fukuda, T. *Macromolecules* **2001**, *34*, 8872.
- 656. Higaki, Y.; Otsuka, H.; Endo, T.; Takahara, A. Macromolecules 2003, 36, 1494.
- 657. Higaki, Y.; Otsuka, H.; Takahara, A. Polymer 2003, 44, 7095.
- 658. Motokucho, S.; Sudo, A.; Sanda, F.; Endo, T. Chem. Commun. 2002, 1946.
- 659. You, Y.Z.; Hong, C.Y.; Pan, C.Y. Chem. Commun. 2002, 2800.
- 660. You, Y.Z.; Hong, C.Y.; Wang, P.H.; Wang, W.P.; Lu, W.Q.; Pan, C.Y. Polymer 2004, 45, 4647.
- 661. Hong, J.; Wang, Q.; Lin, Y.Z.; Fan, Z.Q. Macromolecules 2005, 38, 2691.
- 662. Schaefgen, J.R.; Flory, J. J. Am. Chem. Soc. 1948, 70, 2709.
- 663. Hawker, C.J. Angew. Chem., Int. Ed. Engl. 1995, 34, 1456.
- Matyjaszewski, K.; Miller, P.J.; Pyun, J.; Kickelbick, G.; Diamanti, S. Macromolecules 1999, 32, 6526.

- 665. Chen, M.; Ghiggino, K.P.; Launikonis, A.; Mau, A.W.H.; Rizzardo, E.; Sasse, W.H.F.; Thang, S.H.; Wilson, G.J. J. Mater. Chem. 2003, 13, 2696.
- 666. Barner, L.; Li, Č.; Hao, X.J.; Stenzel, M.H.; Barner-Kowollik, C.; Davis, T.P. J. Polym. Sci., Part A: Polym. Chem. 2004, 42, 5067.
- 667. Ueda, J.; Matsuyama, M.; Kamigaito, M.; Sawamoto, M. Macromolecules 1998, 31, 557.
- 668. Matyjaszewski, K.; Miller, P.J.; Fossum, E.; Nakagawa, Y. Appl. Organomet. Chem. 1998, 12, 667.
- 669. Lecolley, F.; Waterson, C.; Carmichael, A.J.; Mantovani, G.; Harrisson, S.; Chappell, H.; Limer, A.; Williams, P.; Ohno, K.; Haddleton, D.M. J. Mater. Chem. 2003, 13, 2689.
- 670. Haddleton, D.M.; Edmonds, R.; Henning, A.M.; Kelly, E.J.; Kukulj, D. New J. Chem. **1999**, 23, 477.
- 671. Angot, S.; Murthy, K.S.; Taton, D.; Gnanou, Y. Macromolecules 1998, 31, 7218.
- 672. Wang, J.-S.; Greszta, D.; Matyjaszewski, K. Polym. Mater. Sci. Eng. 1995, 73, 416.
- 673. Hedrick, J.L.; Trollsas, M.; Hawker, C.J.; Atthoff, B.; Claesson, H.; Heise, A.; Miller, R.D.; Mecerreyes, D.; Jerome, R.; Dubois, P. *Macromolecules* 1998, 31, 8691.
- 674. Zhao, Y.L.; Jiang, J.; Liu, H.W.; Chen, C.F.; Xi, F. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 3960.
- 675. Darcos, V.; Dureault, A.; Taton, D.; Gnanou, Y.; Marchand, P.; Caminade, A.M.; Majoral, J.P.; Destarae, M.; Leising, F. Chem. Commun. 2004, 2110.
- 676. Hao, X.J.; Nilsson, C.; Jesberger, M.; Stenzel, M.II.; Malmstrom, E.; Davis, T.P.; Ostmark, E.; Barner-Kowollik, C. J. Polym. Sci., Part A, Polym Chem 2004, 42, 5877.
- 677. You, Y.Z.; Hong, C.Y.; Pan, C.Y.; Wang, P.H. Adv. Mater. 2004, 16, 1953.
- 678. Jesberger, M.; Barner, L.; Stenzel, M.H.; Malmstrom, E.; Davis, T.P.; Barner-Kowollik, C. J. Polym. Sci., Part A: Polym. Chem. 2003, 41, 3847.
- 679. Ohno, K.; Wong, B.; Haddleton, D.M. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 2206.
- 680. Haddleton, D.M.; Ohno, K. Biomacromolecules 2000, 1, 152.
- 681. Stenzel-Rosenbaum, M.H.; Davis, T.P.; Chen, V.K.; Fane, Λ.G. *Macromolecules* 2001, *34*, 5433.
- 682. Stenzel, M.H.; Davis, T.P. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 4498.
- 683. Bian, K.J.; Cunningham, M.F. J. Polym. Sci., Part A: Polym. Chem. 2005, 43, 2145.
- 684. Hodges, J.C.; Harikrishnan, L.S.; Ault-Justus, S. J. Comb. Chem. 2000, 2, 80.
- 685. Ayres, N.; Haddleton, D.M.; Shooter, A.J.; Pears, D.A. Macromolecules 2002, 35, 3849.
- 686. Angot, S.; Ayres, N.; Bon, S.A.F.; Haddleton, D.M. Macromolecules 2001, 34, 768.
- 687. Narrainen, A.P.; Pascual, S.; Haddleton, D.M. J. Polym. Sci., Part A: Polym. Chem. 2002, 40, 439.
- 688. Percee, V.; Barboiu, B.; Grigoras, C.; Bera, T.K. J. Am. Chem. Soc. 2003, 125, 6503.
- 689. Stenzel-Rosenbaum, M.; Davis, T.P.; Chen, V.; Fane, A.G. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 2777.
- 690. Stenzel, M.IL; Davis, T.P.; Barner-Kowollik, C. Chem. Commun. 2004, 1546.
- 691. Barner, L.; Barner-Kowollik, C.; Davis, T.P.; Stenzel, M.H. Aust. J. Chem. 2004, 57, 19.
- 692. Abrol, S.; Kambouris, P.A.; Looney, M.G.; Solomon, D.H. Macromol. Rapid Commun. 1997, 18, 755.
- 693. Abrol, S.; Caulfield, M.J.; Qiao, G.G.; Solomon, D.H. Polymer 2001, 42, 5987.
- 694. Bosman, A.W.; Heumann, A.; Klaerner, G.; Benoit, D.; Frechet, J.M.J.; Hawker, C.J. J. Am. Chem. Soc. 2001, 123, 6461.

- 695. Bosman, A.W.; Vestberg, R.; Heumann, A.; Frechet, J.M.J.; Hawker, C.J. J. Am. Chem. Soc. 2003, 125, 715.
- 696. Pasquale, A.J.; Long, T.E. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 216.
- 697. Xia, J.H.; Zhang, X.; Matyjaszewski, K. Macromolecules 1999, 32, 4482.
- 698. Zhang, X.; Xia, J.H.; Matyjaszcwski, K. Macromolecules 2000, 33, 2340.
- 699. Gurr, P.A.; Qiao, G.G.; Solomon, D.H.; Harton, S.E.; Spontak, R.J. Macromolecules 2003, 36, 5650.
- 700. Connal, L.A.; Gurr, P.A.; Qiao, G.G.; Solomon, D.H. J. Mater. Chem. 2005, 15, 1286.
- 701. Lord, H.T.; Quinn, J.F.; Angus, S.D.; Whittaker, M.R.; Stenzel, M.H.; Davis, T.P. J. Mater. Chem. 2003, 13, 2819.
- 702. Zheng, G.H.; Pan, C.Y. Polymer 2005, 46, 2802.
- 703. Tsoukatos, T.; Pispas, S.; Hadjichristidis, N. J. Polym. Sci., Part A: Polym. Chem. 2001, 39, 320.
- 704. Matyjaszewski, K. Polym. Int. 2003, 52, 1559.
- 705. Wooley, K.L. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 1397.
- 706. Becker, M.L.; Liu, J.Q.; Wooley, K.L. Biomacromolecules 2005, 6, 220.
- 707. Becker, M.L.; Liu, J.Q.; Wooley, K.L. Chem. Commun. 2003, 180.
- 708. Butun, V.; Lowe, A.B.; Billingham, N.C.; Armes, S.P. J. Am. Chem. Soc 1999, 121, 4288.
- 709. Fujii, S.; Cai, Y.L.; Weaver, J.V.M.; Armes, S.P. J. Am. Chem. Soc 2005, 127, 7304.
- 710. Liu, S.Y.; Weaver, J.V.M.; Save, M.; Armes, S.P. Langmuir 2002, 18, 8350.
- 711. Yates, C.R.; Hayes, W. Eur. Polym. J. 2004, 40, 1257.
- 712. Gao, C.; Yan, D. Prog. Polym. Sci. 2004, 29, 183.
- 713. Inoue, K. Prog. Polym. Sci. 2000, 25, 453.
- 714. Voit, B. J. Polym. Sci., Part A: Polym. Chem. 2000, 38, 2505.
- 715. Flory, P.J. J. Am. Chem. Soc 1952, 74, 2718.
- 716. Hawker, C.J.; Frechet, J.M.J.; Grubbs, R.B.; Dao, J. J. Am. Chem. Soc. 1995, 117, 10763.
- 717. Ignatova, M.; Voccia, S.; Gilbert, B.; Markova, N.; Mercuri, P.S.; Galleni, M.; Sciannamea, V.; Lenoir, S.; Cossement, D.; Gouttebaron, R.; Jerome, R.; Jerome, C. Langmuir 2004, 20, 10718.
- 718. Gaynor, S.G.; Edelman, S.; Matyjaszewski, K. Macromolecules 1996, 29, 1079.
- 719. Weimer, M.W.; Frechet, J.M.J.; Gitsov, I. J. Polym. Sci., Part A: Polym. Chem. 1998, 36, 955.
- 720. Matyjaszewski, K.; Gaynor, S.G.; Kulfan, A.; Podwika, M. *Macromolecules* **1997**, *30*, 5192.
- 721. Matyjaszewski, K.; Gaynor, S.G.; Muller, A.H.E. Macromolecules 1997, 30, 7034.
- 722. Matyjaszewski, K.; Gaynor, S.G. Macromolecules 1997, 30, 7042.
- 723. Matyjaszewski, K.; Pyun, J.; Gaynor, S.G. Macromol. Rapid Commun. 1998, 19, 665.
- 724. Percec, V.; Grigoras, C.; Kim, H.J. J. Polym. Sci., Part A: Polym. Chem. 2004, 42, 505.
- Lepoittevin, N.; Matmour, R.; Francis, R.; Taton, D.; Gnanou, Y. Macromolecules 2005, 38, 3120.
- 726. Puts, R.D.; Sogah, D.Y. Macromolecules 1997, 30, 7050.
- 727. Ydens, I.; Degee, P.; Libiszowski, J.; Duda, A.; Penezek, S.; Dubois, P. ACS Symp. Ser. 2003, 854, 283.
- 728. Ydens, I.; Degee, P.; Dubois, P.; Libiszowski, J.; Duda, A.; Penczek, S. Macromol. Chem. Phys. 2003, 204, 171.
- 729. Zhang, A.; Wei, L.H.; Schluter, A.D. Macromol. Rapid Commun. 2004, 25, 799.
- 730. Khousakoun, E.; Gohy, J.F.; Jerome, R. Polymer 2004, 45, 8303.
- 731. Shinoda, H.; Matyjaszewski, K. Macromol. Rapid Commun. 2001, 22, 1176.

- 732. Lutz, J.F.; Jahed, N.; Matyjaszewski, K. J. Polym. Sci., Part A: Polym. Chem. 2004, 42, 1939.
- 733. Sprong, E.; De Wet-Roos, D.; Tonge, M.P.; Sanderson, R.D. J. Polym. Sci., Part A: Polym. Chem. 2003, 41, 223.
- 734. Venkatesh, R.; Yajjou, L.; Koning, C.E.; Klumperman, B. Macromol. Chem. Phys. 2004, 205, 2161.
- 735. Li, Y.G.; Shi, P.J.; Zhou, Y.S.; Pan, C.Y. Polym. Int. 2004, 53, 349.
- 736. Frauenrath, H. Prog. Polym. Sci. 2005, 30, 325.
- 737. Frey, H. Angew. Chem. Int. Ed. Engl. 1998, 37, 2193.
- 738. Schluter, A.D.; Rabe, J.P. Angew. Chem. Int. Ed. Engl. 2000, 39, 864.
- 739. Malkoch, M.; Carlmark, A.; Wodegiorgis, A.; Hult, A.; Malmstrom, E.E. Macromolecules 2004, 37, 322.
- 740. Cheng, C.X.; Tang, R.P.; Zhao, Y.L.; Xi, F. J. Appl. Polym. Sci. 2004, 91, 2733.
- 741. Carlmark, A.; Malmstrom, E.E. Macromolecules 2004, 37, 7491.
- 742. Edmondson, S.; Osborne, V.L.; Huck, W.T.S. Chem. Soc. Rev. 2004, 33, 14.
- 743. Pyun, J.; Kowalewski, T.; Matyjaszewski, K. Macromol. Rapid Commun. 2003, 24, 1043.
- 744. Appelt, M.; Schmidt-Naake, G. Macromol. Mater. Eng. 2004, 289, 245.
- 745. Fu, G.D.; Zong, B.Y.; Kang, E.T.; Ncoh, K.G. Ind. Eng. Chem. Res. 2004, 43, 6723.
- 746. Luo, N.; Husson, S.M.; Hirt, D.E.; Schwark, D.W. ACS Symp. Ser. 2003, 854, 352.
- 747. Luo, N.; Husson, S.M.; Hirt, D.E.; Schwark, D.W. J. Appl. Polym. Sci. 2004, 92, 1589.
- 748. Takolpuckdee, P.; Westwood, J.; Lewis, D.M.; Perrier, S. Macromol. Symp. 2004, 216, 23.
- 749. Husseman, M.; Malmstrom, E.E.; McNamara, M.; Mate, M.; Mccerreyes, D.; Benoit, D.G.; Hedrick, J.L.; Mansky, P.; Huang, E.; Russell, T.P.; Hawker, C.J. *Macromolecules* 1999, 32, 1424.
- 750. Parvole, J.; Laruelle, G.; Khoukh, A.; Billon, L. Macromol. Chem. Phys. 2005, 206, 372.
- 751. Ohno, K.; Morinaga, T.; Koh, K.; Tsujii, Y.; Fukuda, T. *Macromolecules* **2005**, *38*, 2137.
- 752. Wu, T.; Efimenko, K.; Genzer, J. Macromolecules 2001, 34, 684.
- 753. Baum, M.; Brittain, W.J. Macromolecules 2002, 35, 610.
- 754. Boyes, S.G.; Granville, A.M.; Baum, M.; Akgun, B.; Mirous, B.K.; Brittain, W.J. Surface Science 2004, 570, 1.
- 755. Yu, W.H.; Kang, E.T.; Neoh, K.G. Ind. Eng. Chem. Res. 2004, 43, 5194.
- 756. Zhai, G.Q.; Yu, W.H.; Kang, E.T.; Neoh, K.G.; Huang, C.C.; Liaw, D.J. Ind. Eng. Chem. Res. 2004, 43, 1673.
- 757. Shah, R.R.; Merreceyes, D.; Husemann, M.; Rees, I.; Abbott, N.L.; Hawker, C.J.; Hedrick, J.L. *Macromolecules* **2000**, *33*, 597.
- 758. Huang, W.X.; Kim, J.B.; Bruening, M.L.; Baker, G.L. Macromolecules 2002, 35, 1175.
- 759. Brantley, E.L.; Jennings, G.K. Macromolecules 2004, 37, 1476.
- 760. Skaff, H.; Emrick, T. Angew. Chem. Int. Ed. Engl. 2004, 43, 5383.
- 761. Lowe, A.B.; Sumerlin, B.S.; Donovan, M.S.; McCormick, C.L. J. Am. Chem. Soc 2002, 124, 11562.
- 762. Shan, J.; Nuopponen, M.; Jiang, H.; Kauppinen, E.; Tenhu, H. *Macromolecules* 2003, *36*, 4526.
- 763. Matsumoto, K.; Tsuji, R.; Yonemushi, Y.; Yoshida, T. Chem. Lett. 2004, 33, 1256.
- 764. Matsumoto, K.; Tsuji, R.; Yoncmushi, Y.; Yoshida, T. J. Nanoparticle Res. 2004, 6, 649.
This Page is Intentionally Left Blank

Abbreviations

| AA | acrylic acid | C_{M} | t |
|--------|---|------------------|-------------|
| АМ | acrylamide | $C_{\mathbb{P}}$ | t |
| АСР | azocyanovaleric acid, 4,4'- azobis(4-cyanopentanoic acid) | $C_{\rm S}$ | t a |
| AFM | atomic force microscopy | CT | t a |
| AIBN | azobisisobutyronitrile, 2,2'- azobis(2-cyanopropane) | $C_{\rm tr}$ | t |
| AIBMe | azobis(methyl isobutyrate), 2,2'-azobis(methyl 2- methylpropionate) | DFT | ((|
| AMS | α-methylstyrene | EA | e |
| AN | acrylonitrile | EMA | е |
| ATRP | atom transfer radical polymerization | EPR | e r a |
| α- | initial position, attached to | D | Ċ |
| В | butadiene | | r |
| BA | n-butyl acrylate | | (|
| tBΛ | t-butyl acrylate | DBPOX | Ċ |
| BMA | <i>n</i> -butyl methacrylate | DMAEMA | 12 r |
| tBMA | <i>t</i> -butyl methacrylate | DMAM | 1 |
| BbB | <i>t</i> -butyl perbenzoate | DMF | 1 |
| вро | benzoyl peroxide | DMSO | ¢ |
| BriBBr | bromoisobutyroyl bromide | DPPH | c |
| Bu | <i>n</i> -butyl | DTBP | ć |
| tBu | <i>r</i> -butyl | Е | c |
| β- | adjacent position, next to α | ЕНМА | 2 |
| С | conversion | ЕР | ŗ |
| C_1 | transfer constant to initiator | E+ | |

| C_{M} | transfer constant to monomer |
|------------------|--|
| $C_{\mathbb{P}}$ | transfer constant to polymer |
| C_8 | transfer constant to solvent or added transfer agent |
| C_{T} | transfer constant to transfer agent |
| $C_{\rm tr}$ | transfer constant (k_{tr}/k_p) |
| C-tr | reverse transfer constant $(=k_{-tr}/k_i)$ |
| DFT | density functional theory |
| EA | ethyl acrylate |
| EMA | ethyl methacrylate |
| EPR | electron paramagnetic resonance (spectroscopy), also ESR |
| D | dispersity/polydispersity of a molecular weight distribution ($\overline{M}_w/\overline{M}_n$) |
| DBPOX | di-t-butyl peroxyoxalate |
| DMAEMA | 2-(dimethylamino)ethyl methaerylate |
| DMAM | N,N-dimethylacrylamide |
| DMF | N,N-dimethylformamide |
| DMSO | dimethylsulfoxide |
| DPPH | diphenylpicrylhydrazyl |
| DTBP | di-t-butyl peroxide |
| Е | ethylene |
| ЕНМА | 2-ethylhexyl methacrylate |
| ЕР | poly(ethylene-co-propylene) |
| Et | ethyl |

The Chemistry of Radical Polymerization

| EtAc | ethyl acetate | $m{k}_{ m i}$ | rate constant for initiator- | |
|--------------------|---|--|--|--|
| f | initiator efficiency | derived radical adding to monomer | | |
| fx | instantaneous mole fraction of monomer X in monomer | k _p | rate constant for propagation | |
| | feed during copolymerization | $k_{\rm prt}$ | rate constant for primary radical termination | |
| F _X | instantaneous mole fraction of monomer X in a copolymer | k_{T} | rate constant for tail addition to monomer | |
| γ- | next to β | <i>k</i> t | rate constant for radical- | |
| GPC | gel permeation chromatography | k _{ic} | radical termination rate constant for radical- radical termination by combination | |
| h | hour(s) | | | |
| HEA | 2-hydroxyethyl acrylate | k. | rate constant for radical- | |
| IIDPE | high density polyethylene | eria. | radical termination by | |
| HEMA | 2-hydroxyethyl methacrylate | | disproportionation | |
| HPMA | 2-hydroxypropyl methacrylate | $k_{ m tr}$ | rate constant for reaction with chain transfer agent | |
| $\Delta H_{ m p}$ | enthalpy of polymerization | $k_{ m trl}$ | rate constant for chain transfer to initiator | |
| 1 | isoprene | k _{trM} | rate constant for chain | |
| l_2 | symmetrical initiator | | transfer to monomer | |
| J∙ | initiator-derived radical | | rate constant for chain | |
| IR | infra-red | _ | transfer to polymer | |
| К | degrees Kelvin | k _{trS} | rate constant for chain transfer to polymer | |
| kact | rate constant for activation | k – | rate constant for shein | |
| $k_{ m add}$ | rate constant for addition | constant for addition $\kappa_{\rm trT}$ | | |
| k_{β} | rate constant for β -scission | | agent T | |
| | (fragmentation) | k_z | rate constant for reaction | |
| K_{eq} | propagation/ depropagation equilibrium constant | LDPF | low density polyethylene | |
| k ₄ | rate constant for initiator | LUDPE | linear low density | |
| | decomposition | | polyethylene | |
| k_{deact} | rate constant for deactivation | LPO | lauroyl (dodecanoyl) | |
| k_{11} | rate constant for head | | peroxide | |
| | addition to monomer | m | minutes | |
| | | М | monomer | |

Abbreviations

| m- | meta- | ₽ _i • | propagating radical of length i (i is an integer) |
|--------------------------|--|-----------------------------|---|
| [M] _{eq} | equilibrium monomer concentration | Pi ^H | saturated disproportionation |
| ΜΑ | methyl acrylate | · | product of length i (i is an integer) |
| MAA | methacrylic acid | P ; [−] | unsaturated |
| MALDI | matrix assisted laser desorption ionization | . 1 | disproportionation product of length i (i is an integer) |
| MAM | methacrylamide | $\mathbf{P_i}^{\mathrm{T}}$ | product from chain transfer |
| МЛН | maleic anhydride | | of length i (i is an integer) |
| MAN | methaerylonitrile | Ph | phenyl |
| Me | methyl | PP | polypropylene |
| MMA | methyl methacrylate | Рт | propyl |
| MMAM | N-methyl methacrylamide | PX | poly(X) |
| MPK | methyl isopropenyl ketone | PX• | poly(X) propagating radical |
| MVK | methyl vinyl ketone | $P_X\bullet$ | propagating radical ending in monomer X |
| $M_{_{ m n}}$ | number average molecular weight | <i>p</i> - | para- |
| $\overline{M}_{\tt w}$ | weight average molecular | r _l j | terminal model reactivity ratio |
| \overline{M} | weight | r _{UK} | penultimate model monomer reactivity ratio |
| V | weight | RAFT | reversible addition- |
| \overline{M}_{λ} | Z average molecular weight | | |
| n- | normal- | S | second(s) |
| NIPAM | N-isopropyl acrylamide | <i>S</i> - | secondary- |
| NMP | nitroxide-mediated | S | styrene |
| | polymerization | S1 | penultimate model radical |
| NMR | nuclear magnetic resonance (spectroscopy) | $\Delta S_{\rm p}$ | entropy of polymerization |
| NVP | <i>N</i> -vinylpyrrolidone | SFRMP | stable free radical mediated |
| 0- | ortho- | | polymerization |
| OTf | triflate, | Σ | summation |
| | trifluoromethanesulfonate | Т | transfer agent |
| Pi | polymer chain of length i (i is an integer) | Т | temperature |
| | | t- | tertiary- |

| 590 | The Chemistry of Radical Polymerization |
|--------------------|---|
| Τ• | transfer agent-derived radical |
| $T_{\rm c}$ | ceiling temperature |
| tBA | <i>t</i> -butyl acrylate |
| ТВАЕМА | 2-(<i>t</i> -butylamino)ethyl methaerylate |
| tBMA | <i>t</i> -butyl methacrylate |
| tBu | t-butyl |
| TEMPO | 2,2,6,6-tetramethylpiperidin- N-oxyl |
| THF | tetrahydrofuran |
| ΤΜSΕΜΛ | trimethylsilyloxyethyl methacrylate |
| UV | ultraviolet |
| VA | vinyl alcohol |
| VAc | vinyl acetate |
| VC | vinyl chloride |
| VF | vinyl fluoride |
| VF2 | vinylidene fluoride |
| VF3 | trifluoroethylene |
| ω- | terminal (remote chain end) position |
| X | ratio of monomers in feed (f_A/f_B) |
| \overline{X}_n | number average degree of polymerization |
| \overline{X}_{w} | weight average degree of polymerization |
| \overline{X}_{z} | z average degree of polymerization |
| У | ratio of monomer units in copolymer $(F_{\rm A}/F_{\rm B})$ |

AA see acrylic acid abstraction see hydrogen atom transfer abstraction vs addition and nucleophilicity 35 by alkoxy radicals 34-5, 124-5, 392 by alkoxycarbonyloxy radicals 103, 127-8 by alkyl radicals 34 5, 113, 116 by t-amyloxy radicals 124 by arenethiyl radicals 132 by aryl radicals 35, 118 by benzoyloxy radicals 35, 53, 120, 126 with MMA 53, 120 by t-butoxy radicals 35, 53, 55, 124 solvent effects 54, 55, 123 with alkenes 122-3 with allyl acrylates 122 with AMS 120, 123 with BMA 53, 123 with isopropenvl acetate 121 with MA 120 with MAN 121 with MMA 53, 55, 120, 419 with VAc 121 with vinyl ethers 123 by carbon-centered radicals 34-5, 113, 116 by cumyloxy radicals 120 by cyanoisopropyl radicals 116 by heteroatom-centered radicals 35, 131-133 by hydroxy radicals 35, 103 with AMS 120, 128 with MA 120 with MMA 120, 128 by isopropoxycarbonyloxy radicals 103, 127-8 with MMA 103 by oxygen-centered radicals 34, 35, 118-131 by sulfate radical anion 129-30 prediction from bond dissociation energies 34-5 from FMO theory 35 from radical polarity 35 solvent effects 55, 123 abstraction-fragmentation chain transfer 309 acceptor monomers copolymerization with donor monomers 351 interaction with Lewis acids 435-6 list 351

thermal initiation 110-1 acetone, chain transfer to 295 p-acetoxystyrene, NMP 480 acetyl radicals, structure 13 acrylamide (AM) polymerization enzyme-mediated polymerization 440 head vs tail addition 182 NMP 480 tacticity 174-5 effect of Lewis acids 434 acrylate esters see also allyl acrylate, butyl acrylate; methyl acrylate reaction with radicals 120, 122-3 relative rate constants, substituent effects 18-19 acrylate ester polymerization see also butyl acrylate polymerization; dodecyl acrylate polymerization; ethyl acrylate polymerization, methyl acrylate polymerization; trifluoroethyl acrylate polymerization; vinyl acrylate polymerization backbiting 211-2, 481 chain transfer to polymer 322 combination vs disproportionation 262 head vs tail addition 182 in supercritical CO2 432 iodine transfer polymerization 522 NMP 480 template polymerization 439 thermal initiation 109-10 acrylic acid (AA), reaction with carbon-centered radicals, rate constants 114 acrylic acid (AA) copolymerization Q-e values 365 with I 528 with NMP 528 with S, thermal initiation 110 acrylic acid (AA) polymerization amphipathic macro RAFT agent 521 block copolymers 521, 543 k_0 solvent effects 426 polymer brushes 559 acticity 173 template polymerization 438 thermodynamics 215 with NMP 480

with RAFT 507-508, 510, 512-514, 521, 543, 559 with SFRMP 483 acrylic monomers, polymerization, head vs tail addition 182 acrylic polymers, head-to-head linkages 182 acrylonitrile (AN) induced decomposition of BPO 86 reaction with alkyl radicals, penultimate unit effects 345 reaction with dithioacids 516 reaction with carbon-centered radicals, rate constants 115 reaction with heteroatom-centered radicals, rate constants 131 reaction with oxygen-centered radicals rate constants 119 specificity 121 acrylonitrile (AN) copolymerization Q-e values 365 reactivity ratios 339 with AMS 353 with E 209 with S ATRP 529, 543 bootstrap effect 431 gradient copolymers 526 NMP 528 **RAFT 529** solvent effects 429-30 thermal initiation 110 acrylonitrile (AN) polymerization chain transfer to halocarbons 293 to polymer 320 to solvent 295 to thiols 290 combination vs disproportionation 262 head vs tail addition 182 inhibition constants 265 $k_{\rm p} \, 221$ effect of Lewis acid 433 solvent effects 426 tacticity 175 template polymerization 438 thermodynamics 215 with RAFT 508, 510, 514 with TERP 524 with stibine-mediated polymerization 524 activation energy for hydrogen-atom transfer 30 for initiator decomposition azo-compounds 70, 71 peroxides 81 for propagation 218 in copolymerization 349, 350

for radical addition 20, 26 calculation of 26 for termination 234, 254 for radical-radical reaction 36 activation entropy see Arrhenius A factor activation-deactivation processes equilibrium constant 461 in ATRP 492 in living radical polymerization 455-7 active species, in living radical polymerization 6, 453, 455 457 acyl peroxides see diacyl peroxides; dibenzoyl peroxide; didecanoyl peroxide; dilauroyl peroxide acyl phosphine oxides as photoinitiators 98, 101-2, 132 radicals from 101, 132 acyl radicals 117-8 decarbonylation rate constants 118 from hydrogen abstraction from aldehydes 118 nucleophilicity 118 acyloxy radicals 125-6 see also benzoyloxy radicals aliphatic fragmentation to alkyl radicals 83, 112, 126 from diacyl peroxides 83, 86 decarboxylation 83 initiator efficiency 83 from a-acylperoxydiazenes 97 from diacyl peroxides 82 from hyponitrites 78 from diaroyl peroxides 82, 215 from hyponitrites 78 from peroxyesters 88, 125 acylphosphine oxides, radical generation 117 addition see radical addition addition-abstraction polymerization 208, 212 addition-fragmentation chain transfer 296-309 see also reversible addition-fragmentation chain transfer (RAFT) mechanisms 287, 296-7 reverse transfer constant 288-9 reviews 296 to allyl halides 299-300, 302-303 to allyl peroxides 303 5 to ally I phosphonates 299, 303 to allyl silanes 299, 303 to allyl sulfides 299 300, 377 8 to allyl sulfonates 299-302 to ally I sulfones 299 to ally I sulfoxides 302 to benzyl vinyl ethers 298-9, 377 to macromonomers 305-8 to methacrylate macromonomer 252, 305-8, 322, 400-1, 419, 420, 501-502

to TERMINI agent 556-7 to thiohydroxamic esters 308, 309 to thionoesters 308, 309 to VC 180, 296 to vinyl ethers 298-9 transfer constants 299-300, 302-4, 307, 309 AIBMe see azobis(methyl isobutyrate) AIBN see azobisisobutyronitrile aldehydes, acyl radicals from 118 alkanethiyl radicals from allv1 sulfides 300 from disulfides 291-2 from thiols 290, 291 polarity 290 reaction with monomers 132 alkenyl radicals evelization 23 3-butenvl radicals 23, 197 6-heptenyl radicals 23, 193 5-hexenyl radicals 5, 54, 187, 192 alkoxy radicals 118-25 see also t-amyloxy radicals; t-butoxy radicals; cumyloxy radicals abstraction vs addition 34, 35, 124-5 combination vs disproportionation 41 fragmentation to alkyl radicals 51-7, 66-7, 88-9, 91-3, 105, 112, 123-5 rate constant 124 solvent effects 123-4 substituent effects 124 from alkyl hydroperoxides 92 from dialkyl hyponitrites 78 from dialkyl peroxides 90 from dialkyl peroxydicarbonates 87 from peroxides 391-2 from peroxyesters 88 from peroxyketals 91 isopropoxy radicals 79, 87, 125, 127-128, 138 methoxy radicals 35, 125 polarity 27, 30-1, 35, 122 primary 35, 41, 125 reviews 119 secondary 35, 41, 125 solvent effects, hydrogen atom transfer 34 temperature dependence on reactivity 56 tertiary 35 119-125 alkoxyamines as initiators of NMP 458, 471, 475-477, 544, 560 - 2functional 533, 558, 561 water-soluble 482 formation/synthesis 138 9, 471, 475 6 induced decomposition 478 radical trapping 138-9 thermal stability 140 alkoxycarbonyloxy radicals 125, 127-8

abstraction vs addition 127-8 fragmentation to alkoxy radicals 127 from dialkyl peroxydicarbonates 87, 125 α-alkoxymethacrylate polymerization ceiling temperature 216 tacticity, effect of Lewis acids 434 alkoxysilane derivatives, grafting on to silicon particles and wafers 561 a-alkyl benzoin derivatives, radical generation 100 alkyl cobaloximes as chain transfer catalysts 485 radical-induced decomposition 485 alkyl halides formation of alkoxyamines 477 reduction by stannyl or germyl radicals 137 alkyl hydroperoxides see also t-amyl hydroperoxide ; t-butyl hydroperoxide as initiators 56-7, 80, 88, 92-3 as source of alkoxy and hydroxy radicals 92 as transfer agents 93 induced decomposition 93 kinetic data for decomposition 80-1 non-radical decomposition 93 reaction with transition metals 93 alkyl iodides, as control agents 499, 522 alkyl pyridinium salts, as ionic liquids 432 alkyl radicals 112-13 see also a-aminoalkyl radicals; benzyl radicals; t-butyl radicals; 2-carboalkoxy-2propyl radicals; α -cyanoalkyl radicals; cyanoisopropyl radicals; cyclohexyl radicals; ethyl radicals; methyl radicals; undecyl radicals abstraction vs addition 34-5 combination vs disproportionation 40, 42 disproportionation pathways 38 fluorine substitution 14 from diacyl peroxides 83 from dialkyldiazenes 68 from fragmentation of acyl radicals 117-8 of acyloxy radicals 51-4, 66-7, 82-5, 88, 112, 117, 126-7, 146 rate constant 126-7 of alkoxy radicals 51-7, 66-7, 88-9, 91-3, 105, 112, 123-5 rate constant 124 of N, N-dimethylaniline radical cation from reduction of alkyl halides 137 from xanthates 502-3 head vs tail addition 112-3 hydrogen abstraction by 116 isopropyl radicals 35 nucleophilicity 13, 21, 31-2

The Chemistry of Radical Polymerization

polarity 35, 113 rate constants for addition 26, 114-5 table of 114-5 reaction with fumarodinitrile 25 reaction with inhibitors 268-73 nitroxides 138, 266 oxygen 56 7, 130, 266 rate constants 266 reaction with methyl a-chloroacrylate 25 reaction with monomers 112-3 rate constants 112-3 penultimate unit effects 345 reaction with oxygen 50, 116, 130 reviews 112 stability 14 structure 12 alkyl stibines, as control agents 499 alkyl tellurides, as control agents 499, 523 O-alkyl xanthates see xanthates alkylcobaloximes as initiators of SFRMP 484 as chain transfer catalysts 314-5 radical-induced decomposition 314-5, 485 3-alkyl-1-methylimidazolium salts, as ionic liquids 432 alkylperoxy radicals 130-1 epoxidation by 130-1 from alkyl hydroperoxides 93, 130 from alkyl radicals and oxygen 56, 130 in autoxidation processes 130 reaction with monomers 130 allyl acetate polymerization, chain transfer to monomer 319 allyl acrylamide, cyclopolymerization 189 allyl acrylate cyclopolymerization 189 reaction with t-butoxy radicals 122 allyl amines, polymerization, chain transfer to monomer 191, 319 allyl esters polymerization chain transfer to monomer 319 head vs tail addition 181-2 allyl ethers, polymerization, chain transfer to monomer 319 allyl halides chain transfer constants 303 chain transfer to 299-300, 302, 317 allyl methacrylamide, cyclopolymerization 189 allyl methacrylate cyclopolymerization 189 reaction with /-butoxy radicals 122 allyl monomers see also allyl acetate; allyl amines; allyl esters; polymerization chain transfer to monomer 191, 319-20

head vs tail addition 181-2, 188 allyl peroxides chain transfer constants 304 chain transfer to 303-5 allyl phosphonates chain transfer constants 303 chain transfer to 299 allyl polymers, head-to-head linkages 181-2, 188 allyl radical, structure 13 allyl silanes as comonomers 301 chain transfer constants 303 chain transfer to 299, 301 allyl sulfides chain transfer constants 300 chain transfer to 292, 299-300 synthesis of end-functional polymers 377-8 allyl sulfonates chain transfer constants 302 chain transfer to 299-301 allyl sulfones, chain transfer to 299 allyl sulfoxides, chain transfer constants 302 alternating copolymerization 333 cyclocopolymerization of divinyl ether with MAH 170 Lewis acid induced 435 of MMA with S 435, 436 of donor-acceptor monomer pairs 350-351 of MAII with S 340, 350, 395 thiol-ene polymerization 378-9 alternating copolymers, definition 333 aluminum isopropoxide cocatalysts, in ATRP 489, 495 AM see acrylamide α -aminoalkyl radicals 86, 102 from amine-peroxide redox couples AMS see α -methylstyrene /-amyl hydroperoxide, as initiator 92 /-amyloxy radicals abstraction vs addition 124 fragmentation to ethyl radicals 124 t-amyl peroxides, peroxyesters source of ethyl radicals 124 anhydrides, rate enhancement of NMP 479 anilinomethyl radical, formation 86 anionic polymerization transformation to radical 387-8, 545 tests for living 452-4 comparison with radical 455, 525, 541 arenethiyl radicals dimethylamino substituted, zwitterionic quinonoid form 26 from disulfides 291-2 from thiols 290 polarity 132 reaction with monomers 132

MMA 132 S 132 arm-first star synthesis 548, 554-5 cross-linked micelles 554-5 in-situ microgel formation 554-5 self-assembly of diblock copolymers to form micelles 554 5 through ATRP 554-5 through NMP 554 5 through RAFT 554-5 aromatic substitution by benzoyloxy radicals of benzene 127 of PS 127 of S 52-54, 127 reversibility 126 by chlorine atoms, of benzene 34 by isopropoxycarbonyloxy radicals, of S 128 by hydroxy radicals of S 128 of AMS 128 by phenyl radicals, of S 52, 117 aroyloxy radicals 125-7 fragmentation into aryl radicals 117 Arrhenius A factor see also activation energy for addition vs abstraction 56 for backbiting 208 for hydrogen atom transfer 32 for initiator decomposition azo-compounds 70-1 peroxides 81 for propagation 218, 221 in copolymerization 349 for radical addition 20-1, 24, 56, 221 calculation of 26 aryl ethers, unsuitability as photoinitiators 100 aryl peroxyesters, unsuitability as initiators 88 arvl radicals 117 see also phenyl radicals abstraction vs addition 35 p-bromophenyl radicals 35 from fragmentation of aroyloxy radicals 117 head vs tail addition 117 p-nitrophenyl radicals 35 polarity 35, 117 reaction with monomers rate constants 117 specificity 117 ascorbic acid, rate enhancement of NMP 479, 482 atom transfer radical coupling (ATRC), telechelic polymer synthesis 537 atom transfer radical polymerization (ATRP) 7,

250, 456 7, 486 98

activation-deactivation equilibrium constant 492 alkoxyamine formation 477 block copolymer synthesis 541-3 diblock copolymers prepared by 543 halogen exchange 542 mechanisms 542 optimal conditions 542 transformation reactions 544-6 triblock copolymers 546 catalysts 487-9, 492-7, 528, 535 copper complexes 489, 492, 492-5 iron complexes 492, 496 nickel complexes 492, 496 ruthenium complexes 492, 495 cocatalysts 481, 492 combination ATRP 491 copolymerization 528-9 effect of Lewis acids 528 mechanism 528 gradient eopolymers 529 end-functional polymer synthesis 544 α -functionalization 536-7 dormant chain end removal 531 ω-functionalization 533-6 graft copolymer synthesis by grafting from inorganic surfaces 562 by grafting through 558-9 grafting from polymer surfaces 560-1 heterogeneous polymerization 497-8 activator generated by electron transfer (AGET) 498 emulsion 491, 498 miniemulsion 491, 498 reverse ATRP 491, 498 reviews 497 suspension 498 initiators 458, 488-92, 536-7 bromoisobutyrates 489, 536-7, 545, 557-9, 563 initiator efficiency 489-91 initiator activity 492 functional 536-7 sulfonyl halides 488-9, 493, 550, 556-7 molecular weights and distributions 490-1 degree of polymerization 490 dispersity 490-1 mechanisms 486-7 inner sphere electron transfer 487 outer sphere electron transfer 487 monomers BA 495-6, 498 BMA 495 6, 498 **ДМАЕМА 497** DMAM 495 HEMA 495, 497

The Chemistry of Radical Polymerization

MA 495 6, 498 MAA 497 MMA 489, 491, 495-6, 498 α -methoxypoly(ethylene oxide) methaerv late S 495-6 VAc 496 reaction conditions 497-8 polymerization kinetics 461, 487-8, 490 reactivity ratios, copolymerization 488 reverse ATRP 491, 498 AIBN 491 peroxide initiators 491 reviews 487, 497 side reactions 488-9, 534-5 star synthesis mechanisms 551-2 precursors 549-50, 554 microgel formation 555 self-condensing vinyl polymerization 555-6 shell-crosslinking of micelles 555 dendritic cores 556 7 azeotropic copolymer composition in terpolymerization 359 penultimate model 343 terminal model 340-1 azo-compounds see also azobisisobutyronitrile; azobis(methylisobutyrate); azo-t-butane; azoisooctane; azonitriles; azoperoxides; dialkyl diazenes; dialkyl hyponitrites; dialkyl diazenes; triphenylmethylazobenzene as initiators 64-79 azobisisobutyronitrile (AIBN) ¹³C labeled 146–8, 260–1, 263 and transition metal complex, in reverse ATRP 491 as initiator 68-9, 442-3, 459, 518, 522, 524 cage reaction 60-1, 76 cage return 68, 73 chain transfer to 77 decomposition 68, 113, 476 decomposition mechanism 60-1, 68, 76, 258 decomposition products 60-1, 76 diazenyl radicals from 68 effect of viscosity on k_d 73 ketenimine from 76, 257, 258 kinetic data for decomposition 70-1 MAN from 60-1, 77 solvent effect on decomposition 68 synthesis of end-functional polymers 376 toxicity 77 azobis(methyl isobutyrate) (AIBMe) ¹³C labeled 146, 148–9, 436 advantages 77 as initiator 68-9, 73, 77, 148-9, 436

chain transfer to 77 decomposition rate 77 kinetic data for decomposition 70-1, 73 solvent effect on k_d 73 synthesis of end-functional polymers 376 zero-conversion initiator efficiency 76 azo-t-butane high temperature initiator 69 kinetic data for decomposition 70-1 azoisoctane 69 high temperature initiator 69 kinetic data for decomposition 70-1 azonitriles see also azobisisobutyronitrile as initiators 68-9 4,4'-azobis(4-cyanopentanoic acid) 72, 518 1,1'-azobis(1-cyclohexanenitrile) 69 2,2'-azobis(2-methylbutanenitrile) 69 decomposition 113 decomposition kinetics 77 azoperoxides decomposition mechanism 97 synthesis of block and graft copolymers 386 backbiting 32, 208-12 addition-abstraction polymerization 208, 212 chain transfer to polymer 208, 210, 320 effect of monomer concentration 209 effect of temperature 209, 211-2 in acrylate ester polymerization 211-2 in E polymerization 208-10 in monosubstituted monomers 211-2 in NMP with acrylates 481 in polymerization of vinyl polymers 211 in S polymerization 211 in VAc polymerization 208-10 in VC polymerization 208 mechanism 168 transition state 209 backbiting-fragmentation in acrylate polymerization 212 macromonomer synthesis 212 Barton-McCombie reaction 296, 502-3 Beer-Lambert law 59 benzene chain transfer to 295 solvent effect on VAc polymerization 324-5 benzil, as photoredox initiator 102 benzil monooxime, fragmentation 99 benzoin derivatives acyl phosphine oxides 101-2, 132 as photoinitiators 99-102 benzil monooxime 99 benzoin esters 100 benzoin ethers α -ether 100 a-sulfonvl 101

as photoinitiators 61-2, 98, 100 shelf life 100 polymerizable 102 radical generation 99-100, 117 p-benzoquinone copolymerization 271 inhibition by 266 radical addition, polar effects 271 benzoyl radical, from benzoin derivatives 100 benzoyloxy radicals 126-7 abstraction vs addition 35 aromatic substitution of benzene 127 of PS 127 of \$ 52, 127 fragmentation to phenyl radicals 52, 54 rate constants 127 from BPO 51, 82, 84, 86 head vs tail addition 126 polarity 35, 126 reaction with monomers AN 126 MA 18, 19 MC 18 MMA 18-9, 21, 52, 126 rate constants 119, 127 \$ 5, 52, 126, 415 specificity 52, 120-1, 126 VAc 126 α-benzovloxystyrene, chain transfer to 298 benzoylthiyl radicals 132 benzyl bromide, as ATRP initiator 489 benzyl dithiobenzoate, retardation in RAFT 517 benzyl dithiocarbamates as RAFT agents 514 in SFMRP 463-4 transfer constants 463-4 benzyl radicals combination pathways 37, 254 from benzoin derivatives 100 from toluene 55 radical-radical reactions 254-5 rate constants for addition 114 5 reaction with nitroxides 138 benzyl selenide, as initiator 466 benzyl thionobenzoate, chain transfer to 308 benzyl vinyl ethers chain transfer constants 299 chain transfer to 298-9, 377 synthesis of end-functional polymers 377 B-scission see fragmentation bicyclo[n,1,0]alkanes, reaction with t-butoxy radicals 33 bicyclobutanes, ring-opening polymerization 195 bicyclo[2,2,1]heptadiene derivatives, cyclopolymerization 193

binary copolymerization see copolymerization biomolecular termination 234 2,2'-bipyridine ligands (bpy, dNbpy) ATRP catalysts 493-4 in situ formation 494 bis-diazenes as multifunctional initiators 97 decomposition 97 bis-methacrylates, cyclopolymerization 424 bis-RAFT agents 511, 546 block copolymer synthesis 384-5 by coupling reaction 540 by sequential monomer addition 540-3 using ATRP 541-3 using NMP 541 using RAFT 502, 543-4 by living radical polymerization 291, 454, 463, 465, 483, 526, 540-8 by transformation reactions 387-9, 544-6 to ATRP 545 6 to NMP 545 to RAFT 546 direct diblock synthesis 541-4 from end-functional polymers 374, 377, 540 segmented block copolymers 547-8 triblock copolymer synthesis 546-7 use of multifunctional initiators 98, 252, 386, 387 block copolymers 540-1 applications 540 composition 540 definition 334 analysis of by GPC 540-1 BMA see n-butyl methacrylate bond dissociation energies abstraction vs addition 34-5 and fragmentation of t-butoxy radicals 124 and hydrogen atom transfer 30, 34-5 and radical addition 17, 23 and radical stability 14, 17 C-H bond 15 fluorine substitution effects 14, 23 substituent effects C-C bond 23, 34 C-H bond 30, 34 C-O bond 23, 34 O-H bond 34 bootstrap effect in copolymerization 357, 431-2, 526 definition 431 in living radical copolymerization 526 boroxyl radicals, in SFRMP 483 BPB see t-butyl perbenzoate BPO see dibenzoyl peroxide branched copolymers, definition 334-5

α-bromoisobutyrates, as ATRP initiators 489, 536 7, 545, 557-9, 563 α -bromoisoproprionates, as ATRP initiators 489 N-bromomethylphthalimide, as ATRP initiator 536 2-bromopropanenitrile, as ATRP initiator 536 butadiene (B) polymerization 1,2- vs 1,4-addition 183 4 kinetic parameters 219 $k_{\rm p} \, 217$ NMP with TEMPO 481 butadiene (B) copolymerization, Q-e values 365 t-butoxy radicals abstraction vs addition 35, 53, 55, 120-3 fragmentation 51-7, 66-7, 88-9, 91-3, 105, 112, 123-5 rate constant 124 radical clock 54 from DBPOX 51, 89, 119 from dialkyl peroxides 90, 119 from di-t-butyl hyponitrites 78, 119 Hammett parameters for abstraction from substituted toluenes 22 for addition to substituted styrenes 22 head vs tail addition fluoro-olefins 16,17, 28, 122 MA 120 olefins 122 VAc 121 VF 17, 23 VF2 16 8, 23, 28, 122 VF3 17 hydrogen atom transfer 32-3 polarity 22, 35, 122 reaction with bicyclo[n,1,0]alkanes 33 reaction with ethers 32-3 reaction with monomers alkenes 122-3 allyl acrylate 122 allyl methacrylate 122 AMS 120 AN 121 BMA 123 E 17 fluoro-olefins 16-8, 22-23, 28, 122 isopropenyl acetate 121 MA 18-9, 120 MAN 121 MC 18 MMA 18-9, 52-3, 55-6, 118, 120 rate constants 114-5, 119 \$ 52-3, 120, 134 specificity 120 I VAc 121 VF 17, 23 VF2 16-8, 23, 122

VE3 17 vinyl ethers 123 reaction with spiro[2,n]alkanes 33 reaction with toluene 55 specificity of hydrogen transfer 33 n-butyl acrylate (BA) copolymerization O-e values 365 with AMS 353 with MMA 526 with VAc 526 n-butyl acrylate (BA) polymerization backbiting 481 chain transfer to allyl sulfonates 302 to halocarbons 293 inhibition by cobalt (II) porphyrin complexes 484 kinetic parameters 219 $k_{\rm p} \, 247$ k_i , chain length dependence 247 emulsion polymerization NMP 482 **RAFT 520** stibine-mediated polymerization 524 with ATRP 495-6, 498 t-butyl hydroperoxide as initiator 92 radicals from 113-14 n-butyl methacrylate (BMA), reaction with tbutoxy radicals 53, 123 n-butyl methacrylate (BMA) copolymerization Q-e values 365 with MAN, combination vs disproportionation 374 with MMA, combination vs disproportionation 374 with S, using NMP 527 i-butyl methacrylate (iBMA) polymerization, kinetic parameters 219 n-butyl methacrylate (BMA) polymerization combination vs disproportionation 255, 262 thermodynamics 215 kinetic parameters 219 with ATRP 495 6, 498 t-butyl methacrylate (tBMA) polymerization, tacticity, solvent effects 428 t-butyl perbenzoate (BPB) as initiator 80 kinetic data for decomposition 80-1 t-butyl peroxybenzoate, as initiator 88 *i*-butyl peroxyesters, variation in efficiency 88 *i*-butyl peroxypentanoate, ethylene polymerization 88 2-butyl radicals, disproportionation pathways 38 t-butyl radicals abstraction vs addition 35

combination vs disproportionation 41 effect of temperature and solvent 42, 43 Hammett parameters for abstraction from substituted toluenes 22 for addition to substituted styrenes 22 polarity 35 rate constants for addition 26, 114-5 table of 114-5 reaction with fumarodinitrile 25 reaction with methyl a-chloroacrylate 25 reaction with oxygen 130 t-butylperoxy radicals 130 Hammett parameters for abstraction from substituted toluenes 22 for addition to substituted styrenes 22 cage reaction by-products from 60-1 AIBN 60, 76 BPO 83 dialkyl hyponitrites 78 definition 60 effect of initiator structure 65 effect of magnetic field 61 effect of solvents 73, 89 effect of viscosity 60 1, 75, 78, 84, 89 AIBN 75 DBPOX 89 diacyl peroxides 84 dialkyl hyponitrites 78 dialkyldiazenes 74, 75 effect on initiator efficiency 57-8, 60, 76, 84 effect on k_d 60, 84 effect on product distribution 42-3 initiator decomposition AIBMe 75 AIBN 60-1, 68, 76 alkyl hydroperoxides 93 BPO 60, 83, 84 DBPOX 89 diacyl peroxides 63, 83-5 dialkyl hyponitrites 78 dialkyl peroxydicarbonates 87 dialkyldiazenes 72 5 di-t-butyl peroxides 92-3 dicumyl peroxides 92 DTBP 60, 91-2 hyponitrites 78-9 LPO 82-3 peroxyesters 88 photodecomposition, BPO 83-4 vs encounter reactions 42-3 cage return AIBN decomposition 68, 73 BPO decomposition 60, 82 diacyl peroxides 82, 84

dialkyldiazenes 68 dilauroy1 peroxide 83 DTBP decomposition 60 effect on initiator efficiency 60 camphoroquinone, as photoredox initiator 102 camphorsulfonic acid, rate enhancement of NMP 479 caprolactone block copolymers 545, 558 captodative olefins copolymerization 270 inhibition by 269-70 reversible addition-fragmentation 470 2-carboalkoxy-2-propyl radicals as leaving group in RAFT agent 507 2-carbomethoxy-2-propyl radicals from AIBMe 436 reaction with monomers 436 radical-radical reactions 255 self-reaction 256 carbon-centered radical-mediated polymerization 467 70 monomer effects 469 carbon-centered radicals 12-13, 112-18 see also acyl radicals; alkyl radicals; aryl radicals. carbon monoxide, copolymerization with depropagation 353 carbonyl compound-tertiary amine systems as photoinitiators 102-3 mechanisms 102-3 catalytic chain transfer 310 advantages over conventional agents 310 catalysts chromium complexes 315 cobalt (II) cobaloximes 313-5 cobalt (III) cobaloximes 314-5 cobalt porphyrins 313 iron complexes 315 molybdenum complexes 315 phthalocyanine complexes 313 mechanism 310-12 reaction condition limitations 315 6 reviews 310 to cobalt complexes 310-6 to PMMA macromonomer 322 cationic polymerization of exazeline groups 558 transformation to radical polymerization 389 ceiling temperature 213-4 in copolymerization 353-4 in homopolymerization 214, 216 in ring-opening polymerization 196 table 215 ceric ions as trap for carbon-centered radicals 106 reaction with 1,2-diols 105 6

in redox initiation 104-6, 386 cetyltrimethylammonium 4-vinylbenzoate, micelle formation 442 chain length, and copolymer composition 381 chain length dependence and diffusion mechanisms, living radical polymerization 251 combination vs disproportionation for PE 258 for PMAN 256 for PS 253 of chain transfer constant 282-4, 294 of chain transfer to polymer 320 PE 321 of copolymer composition 431-2 of k_o 213, 218-21 of kt 234, 245 9 common monomers 247 chain length distribution 240-1 chain transfer see also abstraction-fragmentation chain transfer; addition-fragmentation chain transfer: backbiting; catalytic chain transfer; hydrogen atom transfer by atom or group transfer 280, 289-95 control of, defect groups 418-19 definition 234 homolytic substitution 289-95 in thiol-ene polymerization 378 in VAc polymerization 178-9 mechanisms 279-80 penultimate unit effects 282 3, 291, 294 rate constant for reinitiation 280 reversible 288 9 reviews 280 synthesis of end-functional polymers 211-12, 279, 291, 295, 298-9, 377-8 termination by 234, 279 combination vs disproportionation 283-4 rate constant 236 to disulfides 291-2 polar effects 292 substituent effects 292 to functional transfer agents 377-8 to halocarbons 283, 289, 293-4 catalysis 294 penultimate unit effects 294 to initiator see chain transfer to initiator to monomer see chain transfer to monomer to polymer see chain transfer to polymer to solvents 294-5 to sulfides 292-3 to thiols 289 91, 382 3 penultimate unit effects 291 polar effects 290 substituent effects 290

uses 279 chain transfer constants chain length dependence 282-4 definition 281 effect of conversion 281-2 for allyl halides 302 for allyl peroxides 304 for allyl phosphonates 302 for allyl silanes 302 for allyl sulfides 302 for allyl sulfonates 302 for allyl sulfoxides 302 for cobalt complexes 316 for disulfides 292 for macromonomers 307 for monomers 317 for polymers 320 for thiohydroxamic esters 309 for thiols 290 for thionoesters 309 for vinyl ethers 299 ideal 282 Mayo equation 282-3 measurement 283 7 integrated forms of Mayo equation 286 log CLD method 283-4 Mayo equation 283, 285 through evaluation of usage of transfer agent and monomer conversion 285-6 prediction 282, 283 reversible 288 temperature effects 282 chain transfer to initiator 58, 62-3, 325 AIBN 77 aliphatic diacyl peroxides 82-3, 126 alkyl hydroperoxides 56-7, 93 BPO in S polymerization 63, 85, 127, 415 in VAc polymerization 85 in VC polymerization 85 mechanisms 63 diacyl peroxides 63, 85 dialkyl peroxides 92 dialkyl peroxydicarbonates 87 dialkyldiazenes 75, 77 disulfides 103, 291-2 dithiuram disulfides 103 effect of conversion 85 effect on initiator efficiency 63 effect on kd 63, 82-3 a-hydroperoxy diazenes 97 ketenimine from AIBN 76 peroxyesters 89 persulfate 94 chain transfer to monomer 316-17 allyl acetate 317

allyl amines 191, 319 allyl chloride 317 allyl esters 319 allyl ethers 319 allyl monomers 191, 319-20 AN 317 MA 317 mechanisms 316-17 MMA 316-7 PMMA macromonomer 252, 305-8 S 316 7 transfer constants 317 VAc 316-8 VC 179-80, 296. 316-9 chain transfer to polymer 320-1 AN 320 backbiting 208, 210, 320 E 320, 321 MA 320 mechanisms 320 MMA 320-2 molecular weight dependence 321 PE 320-1 PMMA 321-2 PMMA macromonomer 321-2 poly(alkyl acrylate) 322 poly(alkyl methacrylate) 321-2 PVA 323 4 PVAc 323-5 PVC 325 **PVF 325** S 320 transfer constants 320 VAc 320 VC 320, 325 chemical control model, of termination in copolymerization 366-8 chemical methods, for measurement of end groups 144-5 chlorine atoms 1.2-chlorine shifts 179-80 aromatic substitution 34 hydrogen atom transfer solvent effects 34 specificity 32 in VC polymerization 179-80 polarity 29, 31 chlorocyclohexadienyl radical 34 p-chloromethylstyrene, polymerized by NMP 480.541 chloroprene polymerization, 1,2- vs 1,4-addition 183, 184 5 chlorosilane derivatives, grafting on to silicon particles and wafers 562 chromium complexes as chain transfer catalysts 315

in SFMRP 484 cobaloximes as chain transfer catalysts 310-6 chain transfer constants 316 in situ generation 314-5 cobalt complexes see also alkyl cobaloximes; catalytic chain transfer; cobaloximes; cobalt porphyrin complexes as chain transfer catalysts 310-6 as initiators in SFMRP 484 as catalysts in ATRP 492 catalytic chain transfer 310-6 chain transfer constants 316 control of propagation 423 in SFMRP of acrylates 484-5 of VAc 485 cobalt porphyrin complexes as chain transfer catalysts 313 cobalt tetramesitylporphyrin (CoTMP) 484 in SFMRP of acrylates 484-5 combination 36, 37-8, 233-4 definition 251 pathways for benzyl radicals 37, 254 for cyanoalkyl radicals 37-8, 116, 257-8, 262 for PAN 262 for PMAN models 256-7 for PS models 253 for triphenylmethyl radicals 37 combination vs disproportionation for alkoxy radicals 41, 78 for alkyl radicals 40, 42 for 2-carboalkoxy-2-propyl radicals 255 for cumyl radicals 40, 253, 255 for cyanoisopropyl radicals 116-17, 256-7 for ethyl radicals 41-2 for fluoromethyl radicals 41 for nitroxides 478 with carbon-centered radicals 138 for PAMS, model studies 253-4 for PAN 262 for PBMA 262 model studies 255 with PMAN 257 with PMMA 255 for PE chain length dependence 258 model studies 258 with PMAN 257 for PEMA 262 model studies 255 for 1-phenylalkyl radicals 253, 255 for PMA 262

The Chemistry of Radical Polymerization

for PMAN 262 chain length dependence 256 model studies 256-8 with PS 257 for PMMA 252, 261-2 effect of temperature 255, 261-2 model studies 255-6 for PS 260-1 effect of temperature 254, 260-1 model studies 253-5 for PVAc 263 for PVC 263 for t-butyl radicals 41, 42 guidelines 43 4, 264 in copolymerization 252, 255-7 in primary radical termination 61 in radical-radical reactions 39-43 in radical-radical terminations 251-64 measurement end group determination 259 Gelation technique 258-9 mass spectrometry 259 model studies 252-3 molecular weight distribution 259 molecular weight measurement 259 molecular weight distributions 241-2 oligostyryl radicals 253-4 polar effects 41 reaction condition effects 42-3 statistical factors 39-40 stereoclectronic factors 41-2 steric effects 40-1 substituent effects 40, 42 summary 263 synthesis of end-functional polymers 376-7 temperature dependence 254, 260-2 complex dissociation model, copolymerization 352 - 3complex participation model, copolymerization 352 compositional heterogeneity HEA:BA:S copolymer 382-3 in functional copolymers 336, 381-4 computer modeling of chain end functionality 377 of molecular weight distribution 217 concentration initiator, effect on kd 82, 87, 94 monomer effect on backbiting 209, 211 effect on ceiling temperature 215 effect on ring-opening polymerization 196 conformationally constrained compounds, hydrogen atom abstraction 33 controlled living polymerization 452

controlled radical polymerization see living radical polymerization conversion effect on branching in PVAc 324 effect on chain transfer constants 281-2 effect on initiation 55 effect on initiator efficiency 237, 244 AIBN in S polymerization 75, 76 BPO in MMA polymerization 84 effect on polymer composition 336 effect on radical-radical termination low conversion 244-8 medium to high conversion 248-9 very high conversion 244 copolymerization 333-5 binary kinetic models 354-7 penultimate model 355-6 terminal model 354-5 by ATRP, reactivity ratios 488 chemical control model 366-8 combination and disproportionation 370-1 composition drift 336 compositional heterogeneity 336, 381-4 cross-termination radical precursors 72 depropagation model 353-4 AMS 353 carbon monoxide 353 ceiling temperature 353-4 sulfur dioxide 353 diffusion control models 368-70 donor-acceptor monomer pairs, model studies 353 effect of conversion 360-1 effect of Lewis acids 435-6 effect of molecular weight 431-2 effect of solvent 336, 357, 361, 429-32 models for complex dissociation model 352-3 complex participation model 352 discrimination 348, 350 by composition data 361 monomer complex models 350-3 penultimate model 342-51, 355-6, 429-30 terminal model 337-42, 366, 431-2 monomer sequence distribution 336 effect of solvent 357, 430-1 effect of tacticity 356-7 other models 364 penultimate model 355-6 terminal model 354-5 of AN with S 345 of BA with AMS 380 of BMA with MAN 374 of BMA with MMA 374 of E 209-10

with MAN 374 of macromonomers 400-1, 558-60 of MAH with S 351 of MAN with S 373 of MAN from AIBN 77 of MMA with S chemical control model 368 combination vs disproportionation 371-3 penultimate unit effects 346-9 of quinones 271 of VAc with S 269 penultimate model 342 binary copolymerization 355-6 instantaneous composition equation 343 model description 348-51 model discrimination 348 reactivity ratios 347-8 remote substituent effects 344-7 solvent effects 429-30 polar effects 21 prediction 287 reactivity ratios 363 estimation 359-63 Patterns of Reactivity scheme 365-6 *Q-e* scheme 363–5 radical-radical termination combination vs disproportionation 252, 255, 257, 263 model studies 255, 257 ring-opening 195, 199, 206 thermal initiation 351 template effects 438 terminal model 337-42, 366 assumptions 337-9 azeotropic composition 340-1 binary copolymerization 354-5 bootstrap effect 357, 431-2 implications 340-1 Mayo-Lewis equation 338 overall rate constant for propagation 341-2 reactivity ratios for common monomers 339 termination 366 chemical control model 366-8 combination and disproportionation 370-4 diffusion control models 368-70 terpolymerization 356-7 thermal initiation 110-11 es living radical copolymerization 525 copolymers see also block copolymers; graft copolymers alternating 333 branched 334-5 chain statistics 354-9 compositional heterogeneity 336, 381-4 depiction 335 functional 381-4

gradient 334, 527 monomer sequence distribution 336, 354-7 segmented 334 statistical 333 terminology 333 5 copper complexes as catalysts for ATRP 488, 492-5, 535 activity guidelines 493-4 halogen exchange mechanism 542 in copolymerization 528 ligands 493-5 structures 494 core crosslinked microgels 548 core-first star synthesis 548-54 importance of compatibility and method of polymerization 550 mechanisms 551-2 advantages and limitations 552 4 ATRP 551, 552 NMP 551, 552 RAFT 551, 552, 553-4 multi(bromo-compounds) as ATRP initiators 550 precursors for ATRP 549, 550 for NMP 549, 550 for RAFT 549, 551 through functionalization of polyhydroxy compounds 551 cross termination ethyl and fluoromethyl radicals 41 in copolymerization PBMA• with PMMA• 255, 257 PE• with PMAN• 257 PMAN• with PS• 257 cumene hydroperoxide, as initiator 92 cumyl dithiobenzoate see dithiobenzoate RAFT agents cumyl dithiophenylacetate see dithioacetate RAFT agents cumyl radicals from α -methylstyrene dimer 140 combination vs disproportionation 40, 253. 255 cumyloxy radicals abstraction from isopropylbenzene 56 addition to styrene 56 β-seission to methyl radicals 56 fragmentation rate constants 125 solvent effects 125 from dialkyl peroxides 90 from dieumyl hyponitrites 78 reaction with monomers rate constants 119 specificity 120, 125

temperature dependence of rate constants 56 cupric salts, radical trapping 134, 136 a-cyanoalkyl radicals see also cyanoisopropyl radicals adding monomer via nitrogen 195 from dialkyldiazenes 68, 113 pathways for combination 37-8, 116, 257 polarity 116 reaction with monomers 116 rate constants 113-15 cyanoisopropyl dithiobenzoate, retardation in **RAFT 517** cyanoisopropyl radicals 13 abstraction vs addition 116 combination pathways 37 combination vs disproportionation 116-7, 256 - 7from AIBN 51 ketenimine formation from 76, 116, 257 mechanism of formation from AIBN 60-1 primary radical termination, combination and disproportionation pathways 61-2 radical-radical reactions 256 reaction with monomers 116 MMA 45, 52, 116 rate constants 113-16 S 52, 116 specificity 116 styrene 52 VAc 116 reaction with oxygen 56-7, 116 structure 13 cyclic 1,2-disulfides, as initiators 463 evelization see also cyclopolymerization 3-butenyl radicals 23, 197 6-heptenyl radicals 23, 193 5-hexenyl radicals 5 6, 23, 54, 192 radical clock 54 substituent effects 24, 187-8, 192 ω-alkenyl radicals 23 4-pentenyl radicals 23 stereoelectronic el'fects 23-4 cyclobutylmethyl radicals, ring-opening 198 cyclocopolymerization, of divinyl ether with maleic anhydride 194 cyclohexadienyl radicals structure 13 trapped by disproportionation 140 cyclohexyl radicals disproportionation pathways 38-9 Hammett parameters for abstraction from substituted toluenes 22 for addition to substituted styrenes 22 reaction with fumarodinitrile 25 reaction with MA 18, 19

reaction with MC 18 reaction with methyl a-chloroacrylate 25 reaction with MMA 17-9, 21 reaction with monomers 114-15 cyclohexyImethyl radicals, ring-opening 198 1,4-cyclooctadiene, ring-opening metathesis polymerization 546 cyclopentadiene, reaction with t-butoxy radicals 123 cyclopentylmethyl radicals, ring-opening 198 cyclopolymerization 185-94 1,4-dienes 192 divinvl ether 192-3 1.5-dienes 192-3 bicyclo[2,2,1]heptadiene derivatives 193 o-divinylbenzene 192-3 vinyl aerylate 192-3 vinyl methacrylate 192-3 1,6-dienes 5, 6, 16, 186-2 allyl acrylate 189 allyl methacrylate 189 bis-acrylamides 189 bis-methacrylamides 189 diacrylic anhydride 188 diallyl monomers 5, 187, 188 diallylammonium salts 186 dimethacrylic anhydride 188 dimethacrylic imides 188 dimethallyl monomers 188 o-isopropenylstyrene 189 kinetic vs thermodynamic control 187 propagation kinetics 189 ring size 5, 186-7, 190-1 stereochemistry of ring closure 187-8 stereoelectronic effects 186 substituent effects 187-8 symmetrical 190-1 unsymmetrical 189 1,7-dienes 193 bis-acryloylhydrazine 193 4 ethylene glycol divinyl ether 193-4 1,8-dienes, methylene-bis-acrylamide 193 1,9-dienes, o-dimethacryloylbenzene 194 1,10-dienes, 2,4-pentanediol dimethaerylate 194 1,11-dienes, diallyl phthalate 193 4 ATRP in 497 bis-methacrylates 424 bis-styrene derivatives 424 diene monomers, review 185, 191 triene monomers 191-2 double ring closure 192 triallylamine 191-2 cyclopropyl radical, structure 12 cyclopropylmethyl radicals

ring-opening 196-7 rate constants 196 reversibility 196 cyclopropylstyrene, ring-opening polymerization 196 DBPOX see di-t-butyl peroxyoxalate deactivation by reversible chain transfer and biomolecular activation 456 atom transfer radical polymerization 7, 250, 456, 457, 458, 461, 486-98 deactivation by reversible coupling and unimolecular activation 455-6, 457-86 carbon-centered radical-mediated polymerization 467-70 initiators, inferters and initers 457-8 metal complex-mediated radical polymerization 484 molecular weights and distributions 458-60 nitrogen-centered radical-mediated polymerization 483-4 nitroxide-mediated polymerization 471-82 oxygen-centered radical-mediated polymerizations 483 polymerization kinetics 460 1 reversible addition-fragmentation 470-1 selenium-centered radical-mediated polymerization 466-7 sulfur-centered radical-mediated polymerization 461-6 dead-end polymerization definition 375 synthesis of end-function polymers 375 with disubstituted monomers 469 defect groups 3 see also structural irregularities control 413 effect on polymer properties 3, 50, 414 from radical-radical termination 252 in PMMA 417-20 anionic vs radical initiation 2, 417-8 from addition-fragmentation chain transfer 419 - 20from t-butoxy radical initiation 53, 419 from radical-radical termination 417-18 head-to-head linkages 417, 418 unsaturated chain ends 53, 417-20 in PS 53, 414-17 anionic vs radical initiation 414 benzoate end groups 415-6 peroxide linkages 414-5 prepared with AIBN initiator 416 prepared with BPO initiator 415 unsaturated end groups 415 in PVA/PVAc 323-4 in PVC 2, 420-1

anionic vs radical initiation 421 formation 179 degradative chain transfer definition 234 to allyl monomers 190, 319 dendrimer synthesis 555 dendritic polymer synthesis 548, 556-7 by ATRP 556-7 by NMP 556 by RAFT 556 TERMINI approach 556-7 use of calixarene core 557 dendritic star 548 dendron macromonomers 443, 559 depropagation ceiling temperature 213-4 in copolymerization 353-4 temperature dependence 213-4 diacetylenes, thiol-ene polymerization 379 diacyl peroxides see also diaroyl peroxide; dibenzoyl peroxide; dilauroyl peroxide acyloxy radicals from 82-3 alkyl radicals from 83 as initiators 65, 66 7, 79 80, 86, 421 cage decomposition products 83 cage return 82, 84 chain transfer to 63, 85 concerted decomposition 82 decomposition diacetyl peroxide 82 didecanoyl peroxide 82 explosive decomposition 83 induced decomposition 63, 83, 85, 139 initiator efficiency 84-5 kinetic data for decomposition 80-1 non-radical decomposition 85 photochemical decomposition 83-4 reaction with nitroxides 85, 139 redox reactions 85-7 substituent effects on k_d 82 synthesis of end-functional polymers 377 thermal decomposition 82-3 rate constants (k_1) 80-81 Arrhenius parameters 80-81 effect of solvent on k_d 83 effect of viscosity on k_d 83 transfer to initiator 63, 85 N,N-dialkv1 dithiocarbamates see dithiocarbamates dialkyl hyponitrites as initiators 66-7, 78-9 cage reactions 78 induced decomposition 79 kinetic data for decomposition 70-1 triplet sensitized decomposition 78

viscosity effects 78 dialkyl ketone peroxides 78 as initiators 66-7 dialkyl peroxides alkoxy radicals from 90 as initiators 66-7, 79-80, 90-2 decomposition mechanisms 91 formation 78 induced decomposition 91 kinetic data for decomposition 80 1 solvent dependence 91 dialkyl peroxydicarbonates abstraction of α-hydrogen 87 alkoxy radicals from 87 alkoxycarbonyloxy radicals from 87, 125 as initiators 66-7, 79-80, 87-8, 421 effect of concentration 87 effect of solvent on k_d 87 induced decomposition 87 kinetic data for decomposition 80-1 synthesis of end-functional polymers 376 dialkyl peroxyketals as initiators 90-91 decomposition mechanism 91, 97 initiator efficiency 91 dialkyl peroxyoxalates see also di-t-butyl peroxyoxalate as initiators 66 7 as multifunctional initiators 97 dialkylamino radicals, polarity 31 dialkyldiazenes see also azobisisobutyronitrile; azobis(methyl isobutyrate); azonitriles as initiators 65, 66 78, 376 7 cage return 68 cis-trans isomerization 74 commercially available 68-9, 72 decomposition by-products 76-7 decomposition mechanism 68 initiator efficiency 74-7 initial hydrogen abstraction 75 solvent effects 75 tautomerization 74 temperature effects 75-6 viscosity effects 75 kinetic data for decomposition 70-1 photodecomposition 74 photolability 74 rate constants for decomposition 68 selection 68 solvent effects 73 substituent effects 72-3 symmetrical, decomposition 68 synthesis of end-functional polymers 376-7 thermal decomposition 72-4 by Lewis acids 73

by transition metal salts 73 steric effects 73 thermolysis rates (k_d) 72–3 transfer to initiator 77 unsymmetrical 68 as high temperature initiators 72 as hydroxy radical sources 72 as initiators for quasi-living polymerization 72 for enhanced solubility in organic solvents 72 water soluble 72 diallyl phthalate, cyclopolymerization 193, 194 diallyl trimethylolmethane, thiol-ene polymerization 379 diallylammonium salts, cyclopolymerization 186 diaroyl peroxides see also dibenzoyl peroxide as initiators 66 7, 82 thermal decomposition 82-3, 125 diaryl disulfides, as initiators 461 diazenyl radicals 68 dibenzoyl disulfide as initiator 461 radicals from 132 dibenzoyl peroxide (BPO) ¹³C labeled 146-7, 260, 415 ¹⁹F labeled 261 as initiator 80 2, 459 60, 518, 522 cage reaction 82 cage return 83 chain transfer to 63, 85, 127 decomposition mechanism photochemical 83-4 redox 85-6 thermal 83 in alkoxyamine formation 475-6 in living radical polymerization 459-60 induced decomposition 85 by nitroxides 85, 139 initiator efficiency 83-4 initiator of MMA polymerization 51, 84, 141 initiator of S polymerization 51, 53, 63, 85, 141, 415-16 initiator of VAc polymerization 85 initiator of VC polymerization 85 kinetic data for decomposition 80-1 radiolabeled 145-6 redox initiation systems with N,N-dimethylaniline 86 with transition metals 85-6 diblock copolymers definition 334 synthesis 541 using ATRP 541-3 using NMP 451

using RAFT 543 4 3,5-di-t-butyl catechol, inhibition by 270 di-t-butyl hyponitrites as initiators 78 as source of t-butoxy radicals 78 kinetic data for decomposition 70-1 di-t-butyl methyl radical, persistent radical 40 di-t-butyl nitroxide (DTBN) alkoxyamines from 477 in NMP 475, 480 decomposition 478 di-t-butyl peroxide (DTBP) as initiator 80, 90 from DBPOX decomposition 89 induced decomposition 91-2 initiator efficiency 91 S polymerization 92 kinetic data for decomposition 80-1 radical yield 92 solvent dependence 91 1,1-di-t-butyl peroxycyclohexane, decomposition rate 90 di-t-butyl peroxyoxalate (DBPOX) decomposition 89 in presence of DTBN 477 initiator of MMA polymerization 51, 55 initiator of S polymerization 51 kinetic data for decomposition 80-1 dicumyl hyponitrite as initiator 78 as source of cumyloxy radicals 78 cage reaction 78 kinetic data for decomposition 70-1 dicumyl peroxide as initiator 90 radical yield 92 stability 78 didodecanoyl peroxide see dilauroyl peroxide (LPO) Diels-Alder reaction of S 108 11, 317 diene monomers see also butadiene; chloroprene; cyclopolymerization; isoprene diene monomer polymerization head vs tail addition 182-5 1,2- vs 1,4-addition 182-5 diethylaluminum chloride, in ATRP copolymerization 528 diethylene glycol diacrylate, thiol-ene polymerization 379 diffusion control of radical-radical termination in copolymerization 368-70 in hompolymerization 234, 242-8 conversion dependence 244-9 at very high conversion 244 mechanisms for reaction diffusion 243, 248

reptation 243, 248 segmental motion 243, 248, 251 translational diffusion 251 diffusion controlled limit, for radical-radical reaction 36 1,1-difluoroethylene see vinylidene fluoride diisopropyl peroxydicarbonate t-amine couple 87-8 radicals from 87, 125 reaction with tertiary amines 87-8 a-diketones as photoredox initiator 102 visible light systems 103 β-diketones, in enzyme-mediated polymerization 440dilauroyl peroxide (LPO) as initiator 80, 82 cage return 83 induced decomposition 82-3 initiator efficiency 83 kinetic data for decomposition 80-1 dimethacrylate polymerization formation of head-to-head linkages 193-4 kinetic parameters 219 $k_{\rm p}$ 217, 247 $k_{\rm t}$, chain length dependence 247 o-dimethacryloylbenzene, cyclopolymerization 194dimethoxybenzyl radical, from benzoin ether photoinitiation 62 NN-dimethylacrylamide (DMAM) polymerization ATRP 495 block copolymers 543 NMP 480 RAFT 507-8, 510-12, 539, 543 tacticity 174 5 TERP 524 N,N-dimethylamino radicals Hammett parameters for abstraction from substituted toluenes 22 for addition to substituted styrenes 22 2-(dimethylamino)ethyl methacrylate (DMAEMA) polymerization ATRP 497, 534, 543 block copolymers 483, 534, 543 RAFT 508, 543 triazolinyl radical mediated 483 2-(dimethylamino)ethyl methacrylate (DMAEMA) copolymerization ATRP 529 N.N-dimethylaniline, as photoredox initiator 102 N.N-dimethylaniline/BPO couple, radical production 86 dimethylglyoxime, catalytic chain transfer 314-5 2-dioxolanyl radicals, ring-opening 201

diphenyl diselenide, as initiator 466 diphenyl disulfide, as initiator 461 diphenyl ether, for S polymerizaation 497 diphenylcyclobutanes, from S polymerization 107, 109 diphenylethylene as deactivator 455, 470 inhibition by 269 diphenylmethyl radicals 467 diphenylpicrylhydrazyl (DPPH) inhibition by 76, 268 stability 14 direct detection of end groups 141-5 chemical methods 144-5 EPR spectroscopy 143 IR spectroscopy 141 mass spectrometry 143-4 NMR spectroscopy 142-3 UV spectroscopy 141 diselenides, as initiators 466-7 dispersity and molecular weight distribution 239-42, 461 in ATRP 490-1 in living radical polymerization 459 in RAFT 519 in reversible chain transfer 500-1 in telluride-mediated polymerization 523 star polymers 549 disproportionation 36, 38-9 see also combination vs disproportionation between cobalt complexes and propagating species 485 cross termination, copolymerization 370 definition 251 end groups from PMAN 256-7 from PMMA 256, 262, 418 homotermination, copolymerization 370 of alkoxy radicals 78 of alkyl radicals 38 of but-2-yl radicals 38 of cyclohexyl radicals 38-9 of triazolinyl radicals 483 specificity of hydrogen atom transfer 38, 256-7,262 stereoelectronic control 39 termination by 233 4, 469 rate constant 236 transition state 41 dissociation energies see bond dissociation energies disulfides aliphatic, as initiators 463 as initiators 66 7, 291, 460, 461 3, 465 as photoinitiators 103-4 chain transfer constants 292

chain transfer to 291-2 in living radical polymerization 103, 291, 460-3.465 in synthesis of end-functional polymers 377 in synthesis of telechelies 103, 291 radicals from 132 transfer to initiator 103 dithioacetate RAFT agents 505-506, 511, 517, 521cumyl dithiophenylacetate 511, 517 benzyl dithioacetate 511 dithiophenylacetates 505 506, 511, 517 in emulsion polymerization 521 dithioacids, for RAFT agent synthesis 515-16 dithiobenzoate RAFT agents 505-510, 517, 519, 546 benzyl 510, 515, 529, 543 cumyl 505-508, 515-517, 519-521, 526, 529, 543.559 eyanoisopropyl 506-508, 515, 517, 519, 529, 559 in emulsion polymerization 521 primary and secondary alkyl 510 tertiary alkyl 508 retardation 517 synthesis 515-517 dithiocarbamates see also benzyl dithiocarbamate as RAFT agents 506, 514 as photoinitiators 103-4, 458, 461-3, 465-6, 549 dithiocarbamyl radicals 462-3 dithiocarbonates see xanthates dithioester RAFT agents 505-511 see also dithiobenzoate RAFT agents: dithioacetate RAFT agents dithiols, in thiol-ene polymerization 379 dithiophenylacetate RAFT agents see dithioacetate RAFT agents dithiuram disulfides as photoinitiators 103-4, 461-3 as transfer agents 103, 292, 377 in NMP 531 2 in living radical polymerization 460 initiation mechanism 462-3 synthesis of end-functional polymers 377 divinylbenzene cyclopolymerization 192-3 thiol-enc polymerization 378-9 divinvl ether cyclo-copolymerization with MAH 194 cyclopolymerization 192-3 DMAEMA see 2-(dimethylamino)ethyl methacry late dodecyl acrylate polymerization kinetic parameters 219

k_n 247 $k_{\rm b}$ chain length dependence 247 dodecyl methacrylate, polymerization, k, 247 donor monomers copolymerization with acceptor monomers interaction with Lewis acids 435-6 list 351 thermal initiation 110-1 dormant species, in living radical polymerization 6.455-457 double-ring opening polymerization see ringopening polymerization DPPH see diphenylpicrylhydrazyl DTBN see di-t-butyl nitroxide DTBP see di-t-butyl peroxide dye-partition method, to determine end groups 144 electron paramagnetic resonance spectroscopy see EPR spectroscopy electron spin resonance spectroscopy see EPR spectroscopy electron transfer pathway, for radical-radical reactions 36 7 electrophilicity see also polarity haloalkyl radicals 21 Hammett correlation 21 oxygen-centered radicals 21 electrospray ionization (ESI) see mass spectrometry EMA see ethyl methacrylate emulsion polymerization see also miniemulsion polymerization catalytic chain transfer 316 compartmentalization effects 455, 502 distinction from mini- and microemulsion 64 entry 62 63 initiation 51 initiation kinetics 64 inorganic peroxides as initiators 94 of BA with ATRP 498 with NMP 482 with RAFT 520-521 of BMA with ATRP 498 with RAFT 520 of methacrylates with macromonomer RAFT 502 with thiocarbonylthio RAFT 520 of S with ATRP 498 with RAFT 520-521 of VAc, with RAFT 520 particle size 63-64

particle phase termination rates 249 reverse or combination ATRP 491 surfactant-free RAFT with MAA macromonomer RAFT agent 502 with PAA amphipathic RAFT agent 521 termination kinetics 249-50 with ATRP 491, 497-498 with NMP 481-2 with RAFT 502, 520-1, 544 zero-one kinetics 250 s-mer 63 end-functional polymers 384 synthesis α -functionalization 531, 533, 536-7, 539 40 methods 384 ω-functionalization 531-6, 538-9 with functional inhibitors 381 with functional initiators 375-7, 533-4, 536-7 with functional monomers 379-80 with functional transfer agents 289, 377-8 with halogen transfer agents 293 synthesis by ATRP 544 α-functionalization 536-7 dormant chain end removal 531 ω-functionalization 533-6 synthesis by living radical polymerization 454, 463, 531-40 synthesis by NMP 544 α-functionalization 533 dormant chain end removal 531 ω-functionalization 531-3 synthesis by RAFT 545, 563 dormant chain end removal 531 ω-functionalization 538-9 end groups determination, combination vs disproportionation 259 direct detection 141-5 by chemical methods 144-5 by EPR spectroscopy 143 by IR spectroscopy 141 by mass spectrometry 143-4 by NMR spectroscopy 142-3 by UV-visible spectroscopy 141 ester from aliphatic diacyl peroxides 126 formation 4 by initiation 49 in persulfate initiated PS 129 in polymers initiated with dialkyl peroxides 91 in PVAc 178, 324 initiator-derived 53 peroxy groups 91 solvent-derived 55

entropy of activation see Arrhenius A factor entropy of polymerization 213, 216 steric effects 215 table 215 enzyme-mediated polymerization 437, 440-1 EPR spectroscopy 15, 84 initiation mechanisms 134-5, 143 kinetic studies 133 measurement of *k*_p 217 of k_t 238 spin trapping 134-5 ESI mass spectrometry see mass spectrometry ESR spectroscopy see EPR spectroscopy ethyl acetate, chain transfer to 295 ethyl acrylate (EA) polymerization, transfer constants, to macromonomers 307 ethyl methacrylate (EMA) polymerization combination vs disproportionation 255, 262 kinetic parameters 219 tacticity, solvent effects 428 thermodynamics 215 ethyl radicals Arrhenius A factors 24 combination vs disproportionation 41-2 disproportionation with fluoromethyl radicals 41 ethylene (E) reaction with t-butoxy radicals 17 reaction with methyl radicals 17 reaction with trichloromethyl radicals 17 reaction with trifluoromethyl radicals 17 ethylene-acrylic acid copolymer films, acid functionality 561 ethylene (E) copolymerization backbiting 209-10 with AN 209 with MAN, combination vs disproportionation 374 with (meth)acrylate esters 209 with VAc 209 solvent effects 429 ethylene glycol divinyl ether, cyclopolymerization 193-4 ethylene (E) polymerization backbiting 208-10 chain transfer, to polymer 320-1 combination vs disproportionation 258 $k_0 221$ $k_{\rm p}$ solvent effects 25 t-butyl peroxypentanoate efficiency 88 2-ethylhexyl methacrylate (EHMA), polymerization, kinetic parameters 219 Evans-Polanyi equation 30

Fenton's Reagent, reaction of organic substrates with 96 ferric salts, radical trapping 134, 136 fluorescence spectrophotometry 15-16 fluorine substituents effect on alkyl radicals 14 effect on bond dissociation energies 23 fluoro-olefins polar factors role 21 rate constants and regiospecificities for radical addition 17 18, 22 reaction with t-butoxy radicals 22, 122 reaction with radicals 22, 122 synthesis via iodine-transfer polymerization 521 use in radiation-induced grafting 390 fluoromethyl radicals combination vs disproportionation 41 disproportionation with ethyl radicals 41 FMO theory see frontier molecular orbital theory fragmentation (β -scission) initiator-derived radicals 54 of acyloxy to alkyl radicals 82-3, 112 rate constants 127 of t-alkoxy to alkyl radicals 56, 91 rate constants 124 substituent effects 123-5 of alkoxycarbonyloxy to alkoxy radicals 127 of benzoyloxy to phenyl radicals 52, 54, 84, 127 as radical clock 127 photogenerated 84 rate constants 127 of t-butoxy to methyl radicals 54, 123-4 as radical clock 54 solvent effects 123 of cumyloxy to methyl radicals 125 of 6-heptenoyloxy to 5-hexenyl radical 54 of isopropoxycarbonyloxy to isopropoxy radicals 87 of photoinitiators 98-9 Fremy's Salt, inhibition by 268 frequency factor see Arrhenius A factor Frontier Molecular Orbital (FMO) theory prediction of abstraction vs addition 35 prediction of head vs tail addition 27 functional copolymers, compositional heterogeneity 381-4 functional inhibitors 381 functional initiators 375-7, 536-7 alkoxyamines 533, 558, 561-2 benzoin derivatives 101 dialkyldiazenes 68, 376-7 disulfides 103 nitroxides 534 peroxides 376-7

functional monomers 379-80 functional transfer agents chain transfer 377-8 synthesis end-functional polymers 289, 377-8 telechelic polymers 289 α-functionalization, end-functional polymer synthesis 531 ATRP initiators 536-7 methods 537 telechelic polymers 537 NMP 533 RAFT 539-40 RAFT agents used 539 ω-functionalization, end-functional polymer synthesis 531 ATRP 533-6 end group transformation 535 nucleophilic displacement reactions 536 side reactions 534-5 NMP 531-3 end group transformation 531-2 RAFT 538-9 light sensitive end groups 538 methods for end group removal 538-9 galvinoxyl radical diacyl peroxides decomposition 85 inhibition by 76, 268 stability 14 gel effect see Norrish-Trommsdorf effect gelation technique 258-9 germyl hydrides, as radical traps 137 glycidyl methacrylate polymerization kinetic parameters 219 grafting onto polyolefins 397-8 gold surface ATRP initiation, graft copolymer synthesis 563 RAFT polymerization, synthesis of endfunctional polymers 563 GPC distributions 241-2 purities, block copolymers 540-1 with tetrafunctional RAFT agents 553 gradient copolymers 334, 526 compositional drift 526 degree of blockiness 525 reactivity ratios 526 synthesized by ATRP 529 by NMP 528 by RAFT 529 graft copolymer synthesis 90, 384-5 by living radical polymerization 558-63

grafting from - surface initiated polymerization 560 advantages 560 ATRP initiator use 560-2 grafting from inorganic surfaces 562-3 grafting from polymer surfaces, mechanisms 560-2 NMP initiator use 560 RAFT use 560-3 grafting through - copolymerization of macromonomers 558-60 ATRP use 558-9 NMP use 558 RAFT use 558-9 grafting to - use of end-functional polymers 374, 563 with ceric ion redox initiation 105 with living radical polymerization 483 with macromonomers 400-1 with multifunctional initiators 98, 386-7 with photoinitiation 98 with transformation reactions 387-9, 544-6 graft copolymerization as side reaction with macromonomer RAFT polymerization 501 categorization by method of formation 385 graft copolymers alternative names 385 definition 334-5 termination by combination, Gelation technique 258-9 graft polymerization of MMA 464-5 of S 464-5 graft polyolefins maleate ester 396-7 maleic anhydride 392-6 maleimide 396-7 (meth)acrylate 397-9 styrene 399 vinyloxazoline 400 vinylsilane 399-400 grafting from inorganic surfaces 562-3 from polymer surfaces 560-2 radiation-induced 389-90 radical-induced 390-400 group transfer polymerization, transformation to radical polymerization 309 Group VI hydrides, radical trapping 134, 137 guidelines for initiator selection 66-7 for predicting, combination vs disproportionation ratio 43-4, 264 for predicting outcomes of hydrogen atom abstraction reactions 36

of radical addition 28-9

half-lives for initiator decomposition azo-compounds 71 peroxides 81 typical values 64-5 haloalkyl radicals electrophilicity 21 from halocarbons 293 halocarbons as initiators 488 9 chain transfer to 283, 289, 293-4 effect of chain length on C_{tr} 294 in preparation of telomers 293 polarity 293 α-haloketones, fragmentation 98-9 halo-olefins see also fluoro-olefins rate constants and regiospecificities for radical addition 17-18 reaction with t-butoxy radicals 122 reaction with radicals 122 p-halostyrenes, polymerized by NMP 480 Hammett parameters and hydrogen atom transfer reactions 31 and radical polarity 21 for abstraction from substituted toluenes 22 for addition to substituted styrenes 21-2 for radical reactions 21 Patterns of reactivity scheme 365 Hammond postulate, radical addition 20 head addition, definition 5, 16 head-to-head linkages from captodative olefins 270 from head addition 167, 176, 179, 183 from radical-radical reaction 36, 37 from radical-radical termination 176, 217, 417-18 in acrylic polymers 182 in allyI polymers 181-2 in cyclopolymers 5 in dimethacrylate polymers 193-4 in fluoro-olefin polymers 181 in PAM 182 in PAN 182 in PMA 182 in PMMA 217, 417-8 effect on thermal stability 418 in PVAc 324 in PVF 5, 181 head vs tail addition definition 176 effect of temperature 116 initiation by alkyl radicals 112-13

by anyl radicals 117 by benzoyloxy radicals 52, 120-1, 126, 415 by t-butoxy radicals 120, 122 by cyanoisopropyl radicals 116 by hydroxy radicals 120, 128 with acrylate esters 18-19, 120 with halo-olefins 18, 122 with methyl crotonate 18-9 measurement, NMR spectroscopy 178-9 mechanism 16 of t-butoxy radicals with fluoro-olefins 22 polar effects 21-2, 122, 178, 181 propagation 4-5, 167, 176-85, 421 acrylic monomers 182 allyl monomers 181-2 diene monomers 182 5 fluoro-olefins 180-1 in cyclopolymerization 6, 185-94 monoene polymers 176-82 VAc 178-9, 522 VC 179-80 steric effects 16, 21, 176-8 heat of polymerization 213, 216 steric effects 215 table 215 HEMA see hydroxyethyl methaerylate 6-heptenyl radicals, cyclization 23, 193 heteroatom-centered radicals 131-3 abstraction vs addition 35, 132 polar effects in hydrogen atom transfer 31 heterogeneous polymerization see also emulsion polymerization; miniemulsion polymerizatiom compartmentalization effects 455 in ATRP 497-8 in NMP 481-2 in RAFT 520-1 initiation in 63-4 termination in 249-50 hexasubstituted ethanes in living radical polymerization 460, 467-8 in synthesis of end-functional polymers 377 polymerization mechanism 468 5-hexenyl radicals cyclization 5-6, 23, 24, 54, 192 as radical clock 54 substituent effects 24, 187-8, 192 from 6-heptenoyloxy radical fragmentation 54 reaction with monomers 114-15 n-hexyl radicals Hammett parameters for abstraction from substituted toluenes 22 for addition to substituted styrenes 22 reaction with a-chloroacrylate 25 reaction with fumarodinitrile 25

history initiation 49-50 living radical polymerization 6 polymer structure 1/2 radical polymerization 1-2 radical reactions 11-12 homolytic substitution chain transfer agents 289-90 alkyl iodides 499, 522 alkyl stibines 499 alkyl tellurides 499, 523 disulfides 291-2 monosultides 292-3 thiols 290-1 horseradish peroxidase, in enzyme-mediated polymerization 440 hydrogen abstraction see abstraction vs addition; hydrogen atom transfer hydrogen atom transfer 11, 29-36, 289 see also abstraction vs addition activation energy 29 bond dissociation energies 30, 34-5 effect of adjacent lone pair 33 effect of radical structure on site of attack 32 Evans-Polanyi equation 30 from acctone 295 from benzene 295 from cobalt hydride 311 from initiator alkyl hydroperoxides 93, 130 dialkyl hyponitrites 78 dialkyl peroxydicarbonates 87 dialkyldiazenes 75 ketenimines 77 from metal hydrides 137 from polymer 320 poly(alkyl acrylates) 322 PVAc 323 PVC 325 from solvent 294-5 from toluene 113, 295 guidelines for prediction of outcome 36 Hammett relationship 31 in disproportionation 264 from PMAN 256-7 from PMMA 256, 262 intramolecular 32, 208-12 mechanism 29 order of reactivity of X-H compounds 30 polar effects 30-2 reaction conditions 33-4 reactivity-selectivity principle 30 specificity of in disproportionation 38 intramolecular hydrogen abstraction 32 stereoelectronic effects 32-3

steric effects 30-1 summary 36 to t-alkoxy radicals 124-5 to alkyl radicals 116 to alkylperoxy radicals 56-7 to aryl radicals 117 to benzoyloxy radicals 52, 120 to t-butoxy radicals 32-3 from alkenes 123 from allyl acrylates 122 from amines 33 from AMS 120, 123 from axial vs equatorial positions 33 from BMA 123 from conformationally constrained compounds 33 from ethers 32 3 from MA 120 from MAN 121 from MMA 52, 55, 120 from toluene 55 from VAc 121 from vinyl ethers 123 with BMA 53 to carbon-centered radicals, polar effects 36 to chlorine atoms, specificity 32 to cumyloxy radicals 120 to cyanoalkyl radicals 116 to a-cyanoisopropyl radicals 116 to heteroatom-centered radicals, polar effects 31 to hydroxy radicals 120, 128, 130 specificity 32 to methyl radicals 31 specificity 32 to nitroxides 140 to oxygen-centered radicals, polar effects 36 to phenyl radicals 117 specificity 32 to sulfate radical anion 130 transition state for 29 hydrogen peroxide as initiator 93, 96 as source of hydroperoxy radicals 96 in enzyme-mediated polymerization 440 photodecomposition 96 reaction with transition metal ions 96 redox system 96 synthesis of end-functional polymers 377 thermal decomposition 96 hydroperoxides, as initiators 66-7, 79 hydroperoxy radicals, from hydrogen peroxide 96 α -hydroperoxydiazenes as multifunctional initiators 72, 97 as source of hydroxy radicals 97

decomposition mechanism 97 induced decomposition 97 hydroperoxyketals, decomposition mechanism 97 hydroquinone, inhibition by 270-1 hydrosilylation, of poly(dimethyl siloxane) 546 hydroxy radicals 128 abstraction vs addition 35, 128 from alkyl hydroperoxides 92 from hydrogen peroxide 96 from α-hydroperoxy diazenes 72, 97 from sulfate radical anion 130 polarity 35, 128 reaction with aliphatic esters 32 reaction with monomers rate constants 119 specificity 128 α-hydroxyalkyl radicals, from ceric ion initiation 105 hydroxyethyl methacrylate (HEMA) polymerization kinetic parameters 219 by ATRP 497 with dithiocarbamate photoiniferter 465 hyperbranched polymers 548, 555-7 dendritic polymers 548, 556-7 self-condensing vinyl polymerization 548, 555-6 hyperconjugation 13 hyponitrites see dialkyl hyponitrites 'ideal' inhibitors 264 induced decomposition of initiator alkyl hydroperoxides 93, 130 diacy1 peroxides 63, 83, 85, 139 dialkyl hyponitrites 79 dialkyl peroxides 91-2 dialkyl peroxydicarbonates 87 dialkyldiazenes 75 di-t-butyl peroxide (DTBP) 91-2 dilauroyl peroxide 82-3 α-hydroperoxy diazenes 97 isopropyl hyponitrite 79 ketenimines 77 peroxyesters 89 persulfate 95 inhibition definition 233 kinetics 265-7 mechanisms 266 inhibitors 233-4, 264-7 aromatic nitro-compounds 272-3 captodative olefins 269-70 definition 264 1,1-diphenylethylene 269 DPPH 14, 76, 268

Fremy's Salt 268 functional, synthesis of end-functional polymers 381 galvinoxyl radical 14, 76, 268 'ideal' 264 Koelsch radical 268 nitro compounds 272-3 nitrones and nitroso-compounds 134-5, 272 nitroxides 14, 76, 138-40, 266, 268, 381 oxygen 234, 266, 268-9 phenols 234, 270-1 phenothiazine 272 quinones 271-2 'stable' radicals 267-8 TEMPO 14, 266, 268 transition metal salts 136, 265 6, 273, 381 triphenylmethyl radical 268 triphenylverdazyl radical 76, 268 iniferters see also photoiniferters living radical polymerization 457, 465, 469 loss of 'living' ends through primary radical termination by disproportionation 469-70 inimers 555 initers, living radical polymerization 457 initiation cage reaction and initiator-derived by-products 60 1 chain transfer to initiator 62-3 definition 49 history 49-50 in heterogeneous polymerization 63-4 initiator-derived radicals formation 49-51 fragmentation 54 reaction with monomer 51-3 reaction with oxygen 56-7 solvent effects 55 photoinitiation 58-60, 74, 78, 90 primary radical termination 61-2 reaction with oxygen 56-7 reaction with solvents, additives or impurities 55 structural irregularities from 3-4, 49-50 temperature effects and reaction medium on radical reactivity 55-6 initiator efficiency definition 57 effect of 75-6 effect of cage reaction 57-8, 60, 76, 84 effect of cage return 60 effect of chain transfer to initiator 62-3 effect of conversion 84, 237, 244 effect of initiator structure 65 effect of primary radical termination 61-2 effect of radicals formed 60

effect of temperature 75-6 effect of viscosity 60-1, 75 in thermal initiation 57-8 measurement 145 of AIBMe 76 of AIBN 76 of BPO 83-4 photochemical decomposition 84 of BPO/dimethylaniline redox couple 86 of DBPOX 89 of diacyl peroxides 84-5 of dialkyldiazenes 74-7 of di-t-butyl peroxide 92 of LPO 83 of persulfate redox system 95-6 solvent effects 75 tautomerization 74 initiators 64-7 see also functional initiators; multifunctional initiators; photoinitiators; redox initiators ¹³C labeled 146-8 aqueous phase, particle formation 63-4 atom transfer mediated polymerizations 348 azo-compounds 64, 66 8 dialkyl hyponitrites 66-7, 78-9 dialkyldiazenes 65-78, 376-7 chain transfer to 62-3 disulfides 66-7, 103, 291, 460-3, 465, 488 half lives azo-compounds 71 peroxides 81 typical values 64-5 halo-compounds 458, 488-9 high temperature 72 organometallics 104-5, 423 peroxides 64, 66-7, 79-96, 376 properties 66-7 radiolabeled 145 6 rate constants for decomposition azo-compounds 71 peroxides 81 typical values 64 review 65 selection 53, 65-7 stable (free) radical-mediated polymerizations 457 - 8thermal decomposition rates 57-8, 65 thermal initiation 106-11 inorganic peroxides see also hydrogen peroxide; persulfate as initiators 79-80, 93-6 kinetic data for decomposition 80-1 intramolecular atom transfer see backbiting iodine transfer polymerization 456, 499, 521-2 mechanism 522 of acrylates 522

of fluoro-olefins 522 of S 522 of VAc 522 side reactions 522 ionic liquids, in polymerization 432-3 IR spectroscopy measurement of end groups 141 tacticity 173 iron complexes as catalysts for ATRP 492, 496 as initiators 484 catalytic chain transfer 315 in MMA polymerization 496 in S polymerization 496 in VAc polymerization 496 isobutylene, abstraction vs addition 123 isoprene (1) polymerization 1,2- vs 1,4-addition 183, 185 NMP with TEMPO 481 isopropenyl acetate reaction with carbon-centered radicals, rate constants 115 reaction with oxygen-centered radicals, specificity 121 isopropoxycarbonyloxy radicals 110 aromatic substitution by 128 fragmentation 87 from diisopropyl peroxydicarbonate 87, 108 reaction with monomers MMA 118 S 128 specificity 120 N-isopropyl acrylamide (NIPAM) copolymerization by RAFT 529 N-isopropyl acrylamide (NIPAM) polymerization block copolymers 543, 546 with dithiocarbamate photoinitiator 465 with RAFT 507-508, 510-512, 514, 539, 543, 546 with stibine-mediated polymerization 524 with TERP 524 tacticity 174, 175 Lewis acid effects 435 IUPAC recommendations copolymer depiction 335 living polymerization 452 polymer structure 2

Jablonski diagram 58, 59

ketene acetals

ring-opening copolymerization 195, 379–80 ring-opening polymerization 199–203 ring size effects 199–200

substituent effects 199-201 synthesis of end-functional polymers 195, 379 - 80ketenimine from AIBN decomposition 76-7 from combination of cyanoalkyl radicals 37-9 from cyanoisopropyl radicals 76, 116, 257 from dimeric PMAN 257-8 thermal stability 257 α -ketoalkyl radicals, combination pathways 37-8 ketyl radicals 102-3 Kharasch reaction 486 kinetic vs thermodynamic control cyclization of hexenyl radicals 5-6 in 1,6-dienes cyclopolymerization 187 of radical ring-opening 201 of ring-opening polymerization 196 radical addition 4, 12, 16-17, 50 radical reactions 50 Koelsch radical, inhibition by 268 labeling techniques 145-9 radiolabeling 145 6 NMR 146-9 ¹³C labeled initiators 146-8 ¹⁹F labeled initiators 146-8 ladder polymers from covalently bonded templates 438-40 from methacryloyl derivatives of poly(hydroxy compounds) 194 Lewis acids control of propagation 425, 433-6 head vs tail addition 434 k_p 432–3 tacticity 434-5 control of copolymerization 435-6 with ATRP 436, 528 with RAFT 436, 519, 528 control of tacticity 174-176, 434-435 with RAFT 519 effect on decomposition of azo-compounds 73 effect on RAFT agents 519 ethyl aluminum sesquichloride, in ATRP 528 ligands, for copper based ATRP catalysts 493-4 linear low density polyethylene (LLDPE) see polyolefins living/controlled radical polymerization see living radical polymerization living radical copolymerization 525-30 atom transfer radical polymerization 528-9 bootstrap effects 526 cross propagation 525, 527 end-functional polymer synthesis, by RAFT 538 40 gradient copolymers 526 major advantages of 525

molecular weight distribution 525 nitroxide-mediated polymerization 527-8 reactivity ratios 525-6 reversible addition-fragmentation chain transfer (RAFT) 529-30 steady state composition of propagating radical 525 vs conventional radical copolymerization 525 living radical polymerization see also atom transfer radical polymerization (ATRP); iodine transfer polymerization; nitroxide-mediated polymerization (NMP); metal complex-mediated radical polymerization; reversible addition fragmentation chain transfer (RAFT); tellurium-mediated radical polymerization (TERP) activation-deactivation processes deactivation by reversible coupling and unimolecular activation 455-86 deactivation by reversible atom or group transfer and biomolecular activation 456, 486-98 reversible chain transfer 456, 498-525 active species 6, 455-6 agents providing reversible deactivation 454-7 ATRP 7. 250, 456-7, 486-98 block copolymer synthesis 291, 454, 463, 465, 483, 526, 540-8 by ATRP 541 3, 545 6 by NMP 541, 545-7 by RAFT 543-4, 546-7 segmented block copolymers 547 8 transformation reactions 544 6 triblock copolymers 546-7 carbon-centered radical-mediated polymerization 467-70 chain length and conversion 250-1 and diffusion mechanisms 251 criteria for 452-4 definition 451 dormant species 6, 455-6 labile functionality 416-17, 420 chain end removal 531 end-functional polymer synthesis 454, 463, 531-40 by ATRP 533 7 by NMP 531-3 by RAFT 538-40 α-functionalization 531, 533, 536-7, 539-40 ω-functionalization 531 6, 538 9 graft copolymer synthesis 558-63 grafting from 560-3 grafting through 558-60

grafting to 563 history 6 hyperbranched polymers 555-7 telechelic polymers 103, 537 initiators for NMP 457 60 AIBN 459 alkoxvamines 458 BPO 459, 460 characteristics 458 SFMRP characteristics of 458 disclenides 466-7 disulfides 103, 291, 460-3 dithiocarbamates 458, 463-6 dithiuram disulfides 461-3 iniferters, initers 457-8 hexasubstituted ethanes 460, 467-8 triphenylmethylazobenzene 72, 468 iodine transfer polymerization 456, 499, 521 2 Lewis acids 436 metal complex-mediated radical polymerization 484-6 molecular weight distributions 251, 453-4, 458-60, 490-1, 499-501 molecular weight conversion dependence 452-3,455 nitrogen-centered radical-mediated polymerization 483 4 NMP 250, 416, 456, 458, 461, 471 82 outlook 563-4 oxygen-centered radical mediated polymerizations 483 persistent radical effect 457 publication rate 7 RAFT 470-1, 501-21 selenium-centered radical-mediated polymerization 466 7 stable (free) radical-mediated polymerizations 457 60 star polymers 547 57 arm-first approach 548, 554-5 core-first approach 548, 549-54 dendritie polymers 548, 556-7 self-condensing vinyl polymerization 548, 555-6 stibine-mediated polymerization 499, 524-5 sulfur-centered radical-mediated polymerization 461-6 TERP 456, 499, 522-4 termination kinetics 250-1 terminology 451-2 long chain branches by transfer to polymer 320 in PE 231

in PVA 324 in PVAc 323-4 in PVC 325 in PVF 325 low density polyethylene (LDPE) see polyethylene; polyoletins LPO see dilauroyl peroxide MA see methyl acrylate MAA see methacrylic acid macromonomer RAFT agents 305-8, 501-2 see also macromonomers; methacrylic block copolymers synthesis 502 emulsion polymerization 502 transfer constants 307, 502 macromonomers acrylate esters 410, 481 addition-fragmentation chain transfer, mechanism 305 chain transfer constants 307 copolymerization 400-1, 558-60 definition 400 dendron 443, 559 methacrylic see also macromonomer RAFT agents addition-fragmentation chain transfer 252, 305-8, 322, 400-1, 501-2 copolymerization 400 polymerization 401 solvent effects 428 styrenic 400 synthesis by ATRP 477, 559 by catalytic chain transfer 311-2 by NMP 481 by thermal decomposition of alkoxyamines 477, 479, 481, 559 with addition fragmentation chain transfer agents 296-7, 299-303, 305-8 MADIX (Macromolecular Design by Interchange of Xanthate) see reversible addition-fragmentation chain transfer (RAFT) polymerization magnetic field, effect on cage reaction 61 MAH see maleic anhydride MALDI-TOF mass spectrometry see mass spectrometry maleate ester graft polyolefins 396-7 maleic anhydride (MAH) copolymerization reactivity ratios 339 with \$351, 526 bootstrap effect 431 thermal initiation 110 ω-functionalization, NMP 532 cyclocopolymerization with divinyl ether 194

maleic anhydride (MAH) graft polyolefins 392-6

factors affecting grafting yields 394-6 maleimide graft polyolefins 396-7 MAM see methacrylamide MAN see methacrylonitrile manganese complexes ATRP catalysts 492 as initiators 104 mass spectrometry end group determination initiation 143 4 combination vs disproportionation 259 molecular weight determination 143-4 ESI 143-4 MALDI-TOF 143-4, 259 Mayo equation 282-3 Mayo-Lewis equation, in copolymerization 338 MC see methyl crotonate mechanically assisted polymer assembly (MAPA) 562 mediated radical polymerization see living radical polymerization mercaptoethanol as transfer agent 291, 377 synthesis of end-functional polymers 377 mercuric hydrides, radical trapping 134, 137 metal complex-mediated radical polymerization 484-6 side reactions 485 metal complex-organic halide redox systems 104 5 metal hydrides, radical trapping 134, 137 methacrylamide polymerization, tacticity, effect of Lewis acids 434-5 methacrylate esters copolymerization see also butyl methacrylate copolymerization; methyl methacrylate copolymerization with ethylene 209 methacrylate esters polymerization see also butyl methacrylate polymerization; ethyl methacrylate polymerization; hydroxyethyl methacrylate polymerization; methyl methacrylate polymerization by emulsion RAFT 520 by NMP 481 by stibine-mediated polymerization 524 catalytic chain transfer 311 combination vs disproportionation 255-6, 261-2,478 head vs tail addition 182 isotacticity 174 template 439 thermal initiation 109-10 methacrylate graft polyolefins 397-9 methacrylate macromonomers chain transfer constants 307 chain transfer to 305-8

methacrylic acid (MAA) chain transfer to 305 reaction with carbon-centered radicals, rate constants 114 methaerylic acid (MAA) copolymerization Q-e values 365 with MMA solvent effects 429 30 template effects 438 with S, bootstrap effect 431 methacrylic acid (MAA) polymerization kinetic parameters 219 $k_{\rm p}$ solvent effects 426 tacticity amine effects 428-9 solvent effects 428 thermodynamics 215 with ATRP in solution 497 methaerylonitrile (MAN) by-product from AIBN 60-1, 77 from disproportionation of cyanoisopropyl radicals 62 reaction with carbon-centered radicals, rate constants 115 reaction with heteroatom-centered radicals, rate constants 131 reaction with oxygen-centered radicals rate constants 119 specificity 121 methacrylonitrile (MAN) copolymerization 77 AIBN by-product 60 Q-e values 365 with S bootstrap effect 431 combination vs disproportionation 373 solvent effects 429 methacrylonitrile (MAN) polymerization chain transfer to allyl sulfides 300 combination vs disproportionation 256-8, 262 kinetic parameters 219 k_p 221 tacticity 173, 175 thermodynamics 215 α -methoxypoly(ethylene oxide) methacrylate polymerization, with ATRP in solution 497 methyl aerylate (MA) rate of addition to hexenyl radicals 26 reaction with benzoyloxy radicals 18-9 reaction with t-butoxy radicals 19, 122 reaction with carbon-centered radicals, rate constants 114 reaction with cyclohexyl radicals 18-9 reaction with heteroatom-centered radicals, rate constants 131 reaction with 5-hexenyl radicals, solvent effects 26

reaction with oxygen-centered radicals rate constants 119 specificity 120 reaction with phenyl radicals 18-9 methyl acrylate (MA) copolymerization O-e values 365 reactivity ratios 339 methyl acrylate (MA) polymerization catalytic chain transfer 316 chain transfer to allyl halides 303 to ally sulfides 300 to cobalt complexes 316 to disulfides 292 to polymer 320 to solvent 295 to thiohydroxamic esters 309 to thiols 283, 290 to thionoesters 309 to vinvl ethers 299 combination vs disproportionation 262 inhibition constants 265 kinetic parameters 219 kp 221, 247 k_1 , chain length dependence 247 tacticity 175 ionic liquids effect 433 thermodynamics 215 with ATRP 495-6, 498 with SFRMP CoTMP initiator 484 dithiocarbamate photoiniferters 465 methyl α -chloroacrylate, temperature effect on alkyl radical addition 25 methyl crotonate reaction with benzoyloxy radicals 18 reaction with cyclohexyl radicals 18-9 reaction with phenyl radicals 18 reaction with *t*-butoxy radicals 18 methyl ethacrylate (MEA) polymerization, thermodynamics 215-216 N-methyl methacrylamide (MMAM) polymerization, tacticity, effect of Lewis acid 435 methyl methacrylate (MMA) reaction with alkyl radicals, penultimate unit effects 346 reaction with arenethiyl radicals 132 reaction with benzoyloxy radicals 18-9, 21, 52,120 reaction with t-butoxy radicals 19, 52-3, 55, 118, 120 solvent and temperature effects 55-6 reaction with carbon-centered radicals, rate constants 114

reaction with cumyloxy radicals 120, 125

reaction with cyanoisopropyl radicals 52 reaction with cyclohexyl radicals 18-9, 21 reaction with heteroatom-centered radicals, rate constants 131 reaction with hydroxy radicals 120, 128 reaction with isopropoxycarbonyloxy radicals 118, 120 reaction with methyl radicals 52 reaction with oxygen-centered radicals rate constants 119 specificity 120 reaction with phenyl radicals 19, 52 solvent effects 26 reaction with radicals 18 methyl methacrylate (MMA) copolymerization Q-e values 365 reactivity ratios 339 with AMS 353 with BA 526 with MAA effect of solvent 429 30 template copolymerization 438 with S bootstrap effect 31 combination vs disproportionation 371-3 effect of Lewis acid 435-6 effect of solvent 429 in ionic fiquids 433 penultimate unit effects 346, 347-9 using NMP 527 with VAc, solvent effects 429 methyl methacrylate (MMA) polymerization catalytic chain transfer 310-12, 314-6 catalytic inhibition 311 chain transfer to ally I halides 303 to ally I phosphonates 303 to ally I silanes 303 to allyl sulfides 300 to allyl sulfonates 302 to allyl sulfoxides 302 to cobaloximes 316 to disulfides 292 to halocarbons 293 to macromonomers 307 to polymer 320-2 to solvent 295 to thiohydroxamic esters 309 to thiols 290 to thionoesters 309 to vinyl ethers 299 combination vs disproportionation 255, 258, 261 - 2enzyme-mediated polymerization 440 head vs tail addition 182 inhibition constants 265

in ionic liquids 433 in supercritical CO2 432 kinetic parameters 219, 244 $k_{\rm p}$ 217, 221, 247 effect of Lewis acids 434 effect of solvent 427 $k_{\rm t}$, chain length dependence 246–7 tacticity 173-4 effects of Lewis acid 435, 519 effect of solvent 428 template polymerization 437-8 thermal initiation 109-10 thermodynamics 215 with AIBN initiator 51, 75-6 with ATRP 489, 491, 495-6, 498 with BPO initiator 51, 84, 141 with DBPOX initiator 51, 55 with NMP 481 with RAFT 519 with SFMRP disulfide initiators 462-3 dithiocarbamate photoiniferters 465 with stibine-mediated polymerization 524 methyl radicals abstraction vs addition 35 addition to E 20 Arrhenius frequency factors 24 from fragmentation of t-butoxy radicals 54 Hammelt parameters for abstraction from substituted toluenes 22 for addition to substituted styrenes 22 hydrogen atom transfer 31-2 polarity 22, 31-2, 35 reaction with aliphatic esters 32 reaction with E-17 reaction with fluoro-olefins 17-8, 22 reaction with monomers 114-15 MMA 52, 114 rate constants 114-15 \$ 52, 114 reaction with propionic acid 32 methylene-bis-acrylamide, cyclopolymerization 193 2-methylene-1,3-dioxolanes, ring-opening polymerization 200-3 4-methylene-1,3-dioxolanes, ring-opening polymerization 202-3, 440 2-methylenetetrahydrofurans, ring-opening polymerization 204 2-methylenetetrahydropyrans, ring-opening polymerization 204 α -methylstyrene (AMS) aromatic substitution, by hydroxy radicals 128 reaction with arenethiyl radicals rate constants 131 solvent effects 26

reaction with t-butoxy radicals rate constants 119 solvent effects 123 specificity 120 reaction with carbon-centered radicals, rate constants 114 reaction with heteroatom-centered radicals 132 reaction with hydroxy radicals rate constants 119 specificity 120, 128 a-methylstyrene (AMS) copolymerization Q-e values 365 with depropagation 353 α -methylstyrene dimer chain transfer to 305 mechanism 141 radical trapping 140-1 reaction with oxygen-centered radicals 140-1 α-methylstyrene (AMS) polymerization combination vs disproportionation 253-4 depropagation 214 thermodynamics 215-6 a-methylvinyl monomers see also methyl methacrylate; methacrylonitrile; α -methylstyrene; methacry late esters abstraction vs addition, solvent effects 55 ene reaction with nitroso-compounds 134-5 polymerization combination vs disproportionation 264 thermodynamics 216 Michler's ketone, as photoredox initiator 102 micro-emulsion polymerization 64, 250 distinction from emulsion and miniemulsion 64,250 mikto-arm star 548-549 miniemulsion polymerization 64, 250 compartmentalization effects 455 distinction from emulsion and microemulsion 64,250 reverse or combination ATRP 491 with ATRP 491, 498 with NMP 481 482 with RAFT 520 521 MMA see methyl methacrylate MMAM see N-methyl methacrylamide molecular mechanics calculations, on radical addition 26 molecular orbital calculations 16, 472 on radical addition 27 RAFT agent activity 507 molecular weight 238-40 see also chain length dependence control 413 by template polymerization 438-40

with transfer agents 279-2 with living radical polymerization 251, 453-4, 458-60, 490-1, 519-20 effect on copolymerization 431-2 evaluation after ultrasonication 259-60 predicted evolution conventional vs living radical polymerization 452-3 molecular weight averages 238-40, 283 molecular weight moments 239-40 molecular weight distributions 239-42, 282, 458-61, 490-1, 500-1, 519, 523 see also dispersity and dispersity chain length distribution 240-1 combination vs disproportionation 241-2, 259 in ATRP 490-1 in living radical polymerization 251, 453-4, 458-60, 490-1, 519-20 in RAFT 519-20 in polymerization with reversible chain transfer 499 501 in polymerization with conventional chain transfer 240, 282 in stibine-mediated polymerization 524 in telluride-mediated polymerizations 523 living radical copolymerization 525 most probable distribution 240 Schultz-Flory distribution 240 molybdenum complexes as catalysts for ATRP 492 as chain transfer catalysts 315 as initiators 484 monoene polymers, propagation, head vs tail addition 176 82 monomers, reaction with 51-3 monomethylhydroquinone, inhibition by 270 monosulfides, as initiators 463-5 muconic acid esters, topological polymerization 441 multi-diazenes, as multifunctional initiators 97 multiblock copolymers 334 from NMP and RAFT 547 multifunctional initiators 96-7 applications 98, 386-7 azo-peroxides 386 bis- and multi-diazenes 97 concerted decomposition 97 dialkyl peroxyketals 97 a-hydroperoxy diazenes 97 hydroperoxyketals 97 non-concerted decomposition 97-8 peroxyesters 97 peroxyoxalate esters 97 synthesis of block and graft copolymers 98, 252,386-7

multimethacrylate, template polymerization 439-40

neoprene see polychloroprene network polymers, from thiol-ene polymerization 379 nickel complexes as catalysts for ATRP 492, 496-7 in meth(acrylate) polymerization 496 NIPAM see N-isopropyl acrylamide nitro-compounds diacyl peroxide decomposition 86 inhibition by 272-3 nitrogen-centered radical-mediated polymerization 483-4 nitrones as nitroxide precursors 472 as spin traps 134-5 inhibition by 272 nitroso-compounds as nitroxide precursors 472 as spin traps 134-5 diacyl peroxide decomposition 86 inhibition by 234, 272 nitroxide-mediated polymerization (NMP) 6-7. 144, 250, 416, 420, 422, 3, 451, 7, 471, 82 activation-deactivation equilibrium constant 472 $k_{\rm act}$, prediction, molecular orbital calculations 472 alkoxyamine initiators 458, 471-2, 475, 560-1 block copolymer synthesis 541 diblock copolymers 541 segmented block copolymers 547 transformation reactions 544-5 triblock copolymers 546-7 copolymerization 525-9 of MMA 527-8 of MAH with S 526 mechanism 528 gradient copolymers 526-8 table 528 end-functional polymer synthesis 531-3, 544-5 dormant chain end removal 531 α-functionalization 533 ω-functionalization 531-3 in block copolymer synthesis graft copolymer synthesis grafting from 560 grafting through 558-9 heterogeneous NMP 481-2 emulsion 481-2 miniemulsion 481-2 surfactant/cosurfactants 482

initiation 475-7
macromonomer synthesis 478-9 mechanism 471 monomers 480-1 acrylates 480-1 diene monomers 481 methacrylates 481 S 480-1 vinvl pyridines 480 nitroxides for 472 5 eight-membered ring 475 five-membered ring 473 open-chain 475, 478, 480 seven-membered ring 475 six-membered ring 474 persistent radical effect 471 polymerization kinetics 461 product stability 416, 420, 478-9 rate enhancement 479, 482 reviews 471 side reactions 478-9, 533 nitroxide decomposition 478 disproportionation/elimination 478, 533 star synthesis mechanisms 52, 551 precursors 549, 550, 554-7 microgel formation 555 self-condensing vinyl polymerization 555-6 shell-crosslinking of micelles 555 dendritie cores 556-7 nitroxides see also 2,2,6,6-tetramethylpiperidin-N-oxyl (TEMPO) alkoxyamine homolysis 472 combination vs disproportionation 478 exchange, in NMP 533 fluorescence quenching 139 formation from nitrones 135 from nitroso-compounds 135, 472 in macromonomer synthesis by ATRP 534 by NMP 478-9 inhibition by 14, 76, 138-40, 233, 266, 471, 472 radical trapping 134, 138-40 reaction with diacyl peroxides 85, 139 reaction with radicals 138-9 combination vs disproportionation 138, 478 - 9rate constants 138 stability 472, 478 structures eight-membered ring, for NMP 475 five-membered ring, for NMP 473 open-chain, for NMP 475, 478, 480-2 seven-membered ring, for NMP 475

six-membered ring, for NMP 474 synthesis of end-functional polymers 381 NMP see nitroxide-mediated polymerization NMR spectroscopy and ¹³C labeled initiators 146-9 measurement of end groups 142-3, 146-9 head vs tail addition 178-9 monomer sequence distribution 362-3 tacticity 173 monomer reactivity correlation 365 non-aqueous dispersion polymerization, compartmentalization effects 455 non-covalently bonded templates, template polymerization 437-8 Norrish-Trommsdorff effect and k₁ 244, 248 and template polymerization 438 control of with transfer agents 279 in acrylate polymerization 322 in RAFT 517 nucleophilicity see also polarity acyl radicals 118 alkyl radicals 13, 21, 31 and abstraction vs addition 35 Hammett correlation 21 oxygen-centered radicals 35 oligostyryl radicals, combination vs disproportionation 253, 254 organochalcogenides selenium-centered radical-mediated polymerization 466-7 stibine-mediated polymerization 524-5 sulfur-centered radical-mediated polymerization 461-6 TERP 522-4 organostibine transfer agents 524-5 outer sphere electron transfer (OSET), in atom transfer reaction 487 oxygen as chain transfer facilitator 116, 269 as comonomer 269 effect on initiation 56-7 inhibition by 234, 266, 268 9 reaction with alkyl radicals 50, 56, 116, 130 reaction with cyanoisopropyl radicals 56-7, 116 reaction with polymeric anions 387-8 oxygen-centered radical mediated polymerizations 483 oxygen-centered radicals 118-31 see also acyloxy radicals; alkoxy radicals; alkoxycarbonyloxy radicals; benzoyloxy radicals; t-butoxy radicals; hydroxy radicals;

methoxy radicals, isopropoxycarbonyloxy radicals; sulfate radical anion abstraction vs addition 35 polarity 21, 35 rate constants 118-9 specificity 120-1, 477 oxygen radical anion abstraction vs addition 35 polarity 35 PAA see poly(acrylic acid) PAM see polyacrylamide PAN see polyacrylonitrile palladium complexes, as eatalysts for ATRP 492 Patterns of Reactivity scheme 21, 26, 31 for prediction of reactivity ratios 365-6 for prediction of transfer constants 287 PB see polybutadiene PE see polyethylene pentafluorostyrene, thermal initiation 109 1,2,3,3,3-pentafluorovinylcyclopropane, ringopening polymerization 197 2,4-pentanediol dimethacrylate, cyclopolymerization 194 4-pentenyl radicals, cyclization 23 penultimate model copolymerization 348, 355-6 MMA-S copolymerization 347-9 model description 342-4 reactivity ratios 347-8 remote substituent effects on radical addition 344 7 solvent effect on reactivity ratios 429-30 penultimate unit effects see also chain length dependence in copolymerization 342-4, 347-8 origin 349-50 on chain transfer 282-3 to halocarbons 294 to thiols 291 to RAFT agents 507 on propagation backbiting 211 $k_{\rm p}\,220$ tacticity 171-2, 346-7 on radical addition 344-7 on radical-radical reactions 253 on termination 253 ATRP initiation 492 RAFT agent activity 507 peresters see peroxyesters peroxide linkages formation 56-7, 269, 387-8 in PS 414-5 peroxides

see also alkyl hydroperoxides; diacyl peroxides; dialkyl peroxides; dialkyl peroxyketals; peroxydicarbonates; hydrogen peroxide; peroxyesters; persulfate as component in redox system 79 as initiators 64, 66-7, 79-96, 391-2 as photoinitiators 58, 79 kinetic data for decomposition 80-1 peroxymonosulfate 93 a-peroxyalkoxy radical, from peroxyketals 91 peroxydicarbonates see dialkyl peroxydicarbonates peroxydiphosphate, as initiator 93 peroxydisulfate see persulfate peroxyesters as alkoxy and acyloxy radical source 88 as initiators 66-7, 79-80, 88-90 as multifunctional initiators 97 chain transfer to 305 kinetic data for decomposition 80-1 non-radical decomposition 89-90 photochemical decomposition 90, 125 thermal decomposition 88-90 transfer to initiator 89 peroxyketals see dialkyl peroxyketals peroxyoxalates see dialkyl peroxyoxalates persistent radical effect 471 persistent radicals see also stable free radical mediated polymerization (SFRMP) di-t-butylmethyl radical 40 triisopropylmethyl radical 40 persulfate as initiator 66-7, 79, 93-6, 518 in aqueous and organic media 94-5 crown ether complexes 94 decomposition mechanisms 94 effect of reaction conditions on k_d 94 non-radical decomposition 94 photodecomposition 95 reaction with transition metals ions 95 redox initiation 95-6 phenacyl radicals carbon monoxide loss from 118 from photodecomposition of initiators 117 9,10-phenanthrene quinone, as photoredox initiator 102 phenols, inhibition by 234, 270-1 phenothiazine, inhibition by 272 phenyl radicals abstraction vs addition 35 aromatic substitution of S 52 from BPO 82 from fragmentation of benzoyloxy radicals 52, 54, 127 from triphenylmethylazobenzene hydrogen atom transfer 32

The Chemistry of Radical Polymerization

polarity 35, 117 reaction with monomers 114-5 MA 18-9 MC 18 MMA 18-9, 52 S 52 reaction with tolyl chloride 32 solvent effects on relative reactivity towards S and MMA 26 structure 13 1-phenylethyl radicals combination vs disproportionation 255 reaction with monomers 114-15 self reaction 253 phenylseleno radicals 132 1-(phenylseleno)ethylbenzene 466 phosphinyl radicals from acyl phosphine oxides 101-2, 132 from transfer to phosphines 132 polarity 132-3 reaction with monomers 132 phosphorus-centered radicals 132-3 rate constants 131 photoiniferters 465-6 dithiocarbamates 465-6 dithiuram disulfides 103-4 xanthates 463-5 photoinitiation 58-60, 98 Jablonski diagram 58-9 quantum yield 59-60 rate and efficiency 59 rate of radical generation 60 reviews 98 photoinitiators 98-104 see also photoiniferters; photoredox initiators acyl phosphine oxides 98, 101 acyl phosphonates 101 aromatic carbonyl compounds 98-9 azo-compounds 58, 74, 78 benzil monooxime 99 benzoin esters 100 benzoin ethers 61-2, 98, 100 BPO 83-4 characteristics 58 diacyl peroxides 83-4 dialkyldiazenes 74 dialkyI hyponitrites 78 dialkyldiazenes 74 disclenides 466-7 disulfides 102-3, 458, 460-3 dithiocarbamates 102-3, 461-6 dithiuram disulfides 102-3, 458, 460-3 hydrogen peroxide 96 peroxides 58, 79 peroxyesters 90 persulfate 95

polymerizable 102 selenides 466-7 sulfur compounds 102-3 visible light 103 xanthates 463-5 xanthogen disulfides 461-2 photoredox initiators carbonyl compound/tertiary amine 102-3 metal complex/organic halide 104-5, 388-9 in block/graft copolymer synthesis 388-9 phthalocyanine complexes, chain transfer to 313 Pl see polyisoprene PMA see poly(methyl acrylate) PMAA see poly(methacrylic acid) PMAN see poly(methacrylonitrile) PMMA see poly(methyl methacrylate) polar effects 21-2 in hydrogen atom transfer 31-2, 290 on chain transfer to disulfides 292 to halocarbons 294 to thiols 290 on fragmentation of t-alkoxy radials 124 on radical addition 117 copolymerization 364 head vs tail addition 21-2, 122, 178, 181 rate 271 to p-benzoquinone 271 to fluoro-olefins 122, 181 on radical reactions 16 on radical-radical reactions 41 combination 257 combination vs disproportionation 255 polarity see also electrophilicity; nucleophilicity Hammett parameters 21 in hydrogen atom transfer 30 of alkanethiyl radicals 290 of alkoxy radicals 27, 30, 35 of alkyl radicals 35, 113 of arenethiyl radicals 132 of aryl radicals 35, 117 of benzoyloxy radicals 35, 126 of t-butoxy radicals 22, 35, 122 of t-butyl radicals 35 of cyanoalkyl radicals 116 of halocarbons 293 of hydroxy radicals 35, 128 of methyl radicals 22, 35 of phenyl radicals 35, 117 of phosphinyl radicals 132-3 of radicals, Hammett parameters 21 of selenium-centered radicals 132 of silicon-centered radicals 131 of sulfur-centered radicals 132 of thiols 290

of trichloromethyl radicals 22 of trifluoromethyl radicals 22 polyacrylamide (PAM), head-to-head linkages 182 poly(acrylic acid), tacticity 173 polyacrylonitrile (PAN) combination vs disproportionation 262 head-to-head linkages 182 tacticity 175 poly(alkyl acrylates), combination vs disproportionation 262 poly(alkyl methacrylates) chain transfer to polymer 321-2 combination vs disproportionation 255-6, 261 - 2poly(allyl esters), head-to-head linkages 182 polyamides, from ring-opening polymerization 199 polybutadiene (PB) alkoxyamine functionality attached to 561 microstructure 183-4 poly(butyl acrylate) (PBA) block-PS, prepared by NMP 482 homolysis/backbiting/β-scission reaction, RAFT end group synthesis 538-9 poly(butyl methacrylate-co-methacrylonitrile), combination vs disproportionation 374 poly(butyl methacrylate-co-methyl methacrylate), combination vs disproportionation 374 polycaprolactone-block-poly(4-vinylpyridine) synthesis 545 polycaprolactone-block-PS synthesis 545 polychloroprene, microstructure 183-5 poly(dimethyl siloxane) (PDMS) conversion to ATRP initiator ends by hydrosilylation 546 polymer brush synthesis from 562 polydispersity of molecular weight distribution see dispersity polyesters, from ring-opening polymerization 195, 199 polyethers, from ring-opening polymerization 197 polyethylene (PE) combination vs disproportionation 258 long chain branches 321 short chain branches 208-10 poly(ethylene oxide)-block-PS synthesis from alkoxyamines, using NMP 545 using RAFT 546 poly(ethylene-co-methacrylonitrile), combination vs disproportionation 374 poly(isobutyl methacrylate), attachment of

alkoxyamine functionality to 561 polyisoprene (PI), microstructure 183, 185 polyketones, from ring-opening polymerization 195, 202 - 3polymer brushes see graft copolymer polymer structure see also defect groups: structural irregularities control 413, 421-2 copolymers 333-5 monomer sequence distribution 354-7, 430 1 end groups from chain transfer 279, 282, 287, 418, 420, 421 from radical-radical termination 233, 415, 417-8,421 initiator-derived 49, 63, 414-5, 420-1 general formula 1-4, 49 history 1-2 in chain groups head-to-head linkages 167, 176, 178-9, 184, 252, 434 rings 186-7, 193 short-chain branches 208-10, 211 tacticity 434-5 tail-to-tail linkages 167, 176, 178-9 unsaturation 182-4, 195, 206, 251-2 **IUPAC** recommendations 1 repeat units 1, 4 Staudinger concept 1-2 tacticity 167 9, 175 polymeric initiators, synthesis of block copolymers 386 polymerization thermodynamics 213-16 autoacceleration phenomenon 281 degree of polymerization 281-2 reviews 213 poly(methacrylic acid) (PMAA), template polymer 438 poly(methacrylonitrile) (PMAN) combination vs disproportionation 256-8, 262 tacticity 173, 175 poly(methacrylonitrile-co-styrene), combination vs disproportionation 373 poly(methyl acrylate) (PMA), tacticity 175 poly(methyl acrylate)-block-PBA synthesis 485 poly(methyl methacrylate) (PMMA) defect groups 417-20 control 418-19 head-to-head linkages 182, 252, 417-18 unsaturated chain ends 53, 252, 417-20 end groups determination 142-144, 146-148 from benzoyloxy radicals 52-3, 120 from t-butoxy radicals 52-3, 120, 144 from cyanoisopropyl radicals 52-3, 144, 147 - 148from phenyl radicals 120

tacticity 173, 175, 424, 438 thermal stability 2, 252, 417-20 prepared with NMP 420 poly(methyl methacrylate-co-styrene), combination vs disproportionation 371-3 polyol-redox system, as multifunctional initiators 386 polyolefins (PO) radiation-induced grafting 390 melt-phase grafting to 392-400 monomers for grafting to maleate ester 396 7 maleic anhydride 392 6 maleimide 396-7 (meth)acrylate 397-9 styrene 399 vinyl oxazoline 400 vinvlsilane 399-400 polypropylene (PP) see polyolefins polystyrene (PS) aromatic substitution, by benzoyloxy radicals 127combination vs disproportionation 260-1 model studies 253 5 defect groups 414-7 benzoate end groups 53, 415-6 peroxide linkages 414-5 unsaturated end groups 415 end groups determination 141-144, 146-148 from benzoyloxy radicals 52-3, 120, 139, 415 6 from t-butoxy radicals 52-3, 120, 144 from cyanoisopropyl radicals 52-3, 144, 147 148 from phenyl radicals 120 tacticity 175 thermal stability anionic vs free-radical 414 prepared with AIBN 416 prepared with BPO 53, 85, 415-16 produced by ATRP and RAFT 416, 417 produced by NMP with TEMPO 416 weathering 53 polythioesters, from ring-opening polymerization 199 poly(trifluoroethylene), head-to-head linkages 181 poly(vinyl acetate) (PVAc) chain transfer to polymer 323-5 combination vs disproportionation 263 emulsion particle formation 482 end groups, FAB-MS analysis 178 head-to-head linkages 178-9 long-chain branches 323-4

effect of reaction conditions 324 measurement 324 short-chain branches 211 tacticity 175 poly(vinyl alcohol) (PVA) ceric ion initiated graft copolymerization 105-6 chain transfer to polymer 323-4 long-chain branches 324 tacticity 173, 175 poly(vinyl chloride) (PVC) combination vs disproportionation 263 defect groups 420-1 long-chain branches 325 propagation 179-80 short chain branches 211 structural irregularities 179 tacticity 173, 175 thermal stability 2, 179, 420-1 poly(vinyl fluoride) (PVF) head-to-head linkages 5, 181 long-chain branches 325 tacticity 173 poly(vinylidene fluoride) (PVF2), head-to-head linkages 181 poly(4-vinylpyridine) block polymer, from PEO 545 potassium persulfate, as initiator 518 pressure effects backbiting in E polymerization 209 on chain transfer to PVF 325 on radical-radical reactions 43 on RAFT process 519 primary alkoxy radicals atom abstraction 35 reactions with monomers 125 primary radical termination 58, 61-2 benzoin ethers 61-2, 100 by sulfur-centered radicals 103, 463-4 combination vs disproportionation 61 cyanoisopropyl radicals 61-2 definition 61, 233 effect on initiator efficiency 61 in living radical polymerization 458 in telechelic synthesis 62 of PBMA, with cyanoisopropyl radicals 257 of PE, with cyanoisopropyl radicals 257 of PS with benzoyloxy radicals 415 with cyanoisopropyl radicals 116, 257, 376 reversible 62 primary radicals definition 49 from azo-compound decomposition 112 propagation 167-8 control 421-43

with nitroxide 422-3 with organometallic reagent 422-3 with titanocene dichloride 424 effect of solvent 425-32 copolymerization 429-32 homopolymerization 426-9 mechanisms 426 head vs tail addition 5, 167, 176-85 kineties 216-21 kp 216-21 chain length dependence 213, 218-21 conversion dependence 218 measurement 216-18 monomer structure dependence 218-20 values 216 polymerization thermodynamics 213-16 regiosequence isomerism 167, 176-85, 421 acrylic polymers 182 allyl polymers 181-2 diene polymers 182-5 fluoro-olefin polymers 180-1 monoene polymers 176-82 PAM 182 PAN 182 PVAc 178-9 PVC 179-80 terminology 176 stereosequence isomerism/tacticity 167-76, 421 structural irregularities 3, 167 structural isomerism 167, 185-212, 421 addition-abstraction polymerization 208 backbiting 208-12 cyclopolymerization 185-94 ring-opening polymerization 194 208 PS see polystyrene pseudo-living polymerization see living radical polymerization pulsed laser photolysis (PLP) measurement of $k_p 217$ measurement of $k_t 238$ PVA see poly(vinyl alcohol) PVAc see poly(vinyl acetate) PVC see poly(vinyl chloride) PVF see poly(vinyl fluoride) 2-pyridine carboxaldehyde, methanimine ligands from 493 Q-e scheme 21, 26, 31, 287 for prediction of reactivity ratios 363-5 *O-e* values 364-5 quantum yield, definition 59. quasi-living polymerization see living radical polymerization quinones

copolymerization 271

inhibition by 234, 271 2 quinonoid intermediates from combination of benzyl radicals 37 from reaction of triphenylmethyl radicals 469 radial block copolymer 548 radiation-induced grafting processes 389-90 radical addition and product radical stability 17 application of FMO theory 27 bond strength effects 22-3 carbon-carbon double bonds 11, 16-29 entropic considerations 24 factors affecting specificity 16-17, 19 guidelines 28-9 Hammond postulate 20 kinetic vs thermodynamic control 4, 12, 16-17,50 mechanism 16 polar effects 21/2 reaction condition effects 24 6 regiospecificity 17-19 solvent effects 25 6 stability vs resonance factors 19-20 stereoelectronic effects 23-4 steric effects 19-21 substituent effects 20, 22 acrylate esters 18 halo-olefins 17-18 summary 28-9 temperature effects 24-5 theoretical studies 20-1, 26-8 transition state for 17, 20, 23 radical clock cyclization of 5-hexenyl radicals 54 fragmentation of benzoyloxy radicals 127 fragmentation of t-butoxy radicals 54 reaction of carbon-centered radicals with metal hydrides 137 with nitroxides 138 radical cyclization see cyclization radical detection 14-16 EPR 15 fluorescence 15-16 molecular orbital calculations 16 UV 15-6 radical-induced grafting 390-9 maleic anhydride graft polyolefins 392-6 melt phase grafting 390 side reactions 390-1 radical polymerization benefits of 1 general mechanism 2 history 1-2 publication rate 7 radical properties 12-16

radical-radical reactions 11, 36-44 see also combination pathways; disproportionation pathways; primary radical termination cage vs encounter reactions 43-4, 252 combination pathways 36, 37-8 combination vs disproportionation 39-43, 251-2 cross termination 255, 257 diffusion control 36 disproportionation pathways 36, 38-9 electron transfer pathway 36-7 head-to-head coupling 36-8 polar effects 41 pressure effects 43 rate and specificity steric effects 254-5 temperature effects 254 rate constants 36 solvent effects 42-3 statistical factors 39-40 stereoelectronic effects 41-2 steric effects 37, 40-1 summary 43 4 techniques 252 temperature effects 42-3 termination, in living radical polymerization 455.457 transition states 39 viscosity effects 43 radical-radical termination activation energy 234 at low conversion 244-8 at medium to high conversion 244, 248-9 at very high conversion 244 by chain transfer, rate constant 236 by combination, rate constant 236 by disproportionation, rate constant 236 chain length dependence of combination vs disproportionation 253, 256, 258 of rate constant 234, 245-8 classical kinetics 235-8 combination vs disproportionation 251-64 model studies 252-8 polymerizations 258-63 summary 263 4 diffusion controlled 242-3 low conversions 244-8 medium to high conversions 244, 248-9 very high conversions 244 in copolymerization, combination vs disproportionation 255, 257 kineties, reviews 235 model studies 252-3 pathways for 233-4

rate constant 234-8 definition 235-6 diffusion mechanisms 242-51 effect of conversion 244-9 gel or Norrish-Trommsdorff effect 244, 248 molecular weight distributions 240-2 molecular weights and molecular weight averages 238-40 prediction 244-5 terminology 234 structural irregularities from 176, 252 radical reactions addition see radical addition; abstraction vs addition historical beliefs 11-12 hydrogen atom transfer see hydrogen atom transfer; abstraction vs addition kinetic vs thermodynamic control 4, 12, 50 radical-radical reaction see radical-radical reactions; combination pathways; disproportionation pathways; primary radical termination steps in radical polymerization 11 stereoselectivity 12 temperature dependence 55-6 medium effects 55-6 radical stability 14 and hyperconjugation 13 effect on regioselectivity of radical addition 17 for calculating thermolysis rates 73 of primary, secondary and tertiary radicals 17 radical structure 12-13 see also specific radicals, e.g. acetyl radical pi-radicals 12 sigma-radicals 12-13 radical trapping 133-41 agents for 132-3 metal hydrides 134, 137 α -methylstyrene dimer 134, 140–1 mercuric hydrides 137 nitrones 134-6 nitroso-compounds 134-6 nitroxides 134, 138-40 spin traps 134-6 transition metal salts 134, 136 tri-n-butylstannane 137 radicals carbon-centered see carbon-centered radicals; alkyl radicals; aryl radicals heteroatom-centered 131-3 initiator-derived, classification 53 oxygen-centered see oxygen-centered radicals; acyloxy radicals; alkoxy radicals phosphorus-centered 132-3 polarity 21-2 properties see radical properties

sclenium-centered 132 silicon-centered 131 structure see radical structure sulfur-centered 132 terminology 12 trapping see radical trapping radiolabeling 145-6 RAFT see reversible addition-fragmentation chain transfer (RAFT) polymerization RAFT agents see dithiocarbamates; dithioesters; macromonomer RAFT agents; thiocarbonylthio RAFT agents; reversible additionfragmentation chain transfer (RAFT) polymerization; trithiocarbonates; xanthates random copolymers 335 definition 333 rate constants see also activation-deactivation equilibrium constant for abstraction, effect of bond dissociation energies 34-5 for addition effect of bond dissociation energies 34-5 to acrylate esters 18-9 to halo-olefins 17 8, 22 for β-scission, t-alkoxy radicals 124 for decarbonylation, acyl radicals 118 for fragmentation of benzoyloxy radicals 127 of cumyloxy radicals 125 for inhibitor-radical reaction (k_z) inhibitors and carbon-centered radicals 266 nitroxides and carbon-centered radicals 138, 266 for initiation (k) 236 see also for addition for initiator decomposition (k_d) azo-compounds 71 effect of cage return 60 effect of chain transfer to initiator 63 peroxides 81 typical values 64 for propagation (k_0) 236 chain length dependence 213, 218-21 conversion dependence 218 effect of Lewis acids 433-4 measurement 216-18 monomer structure dependence 218-20 solvent effects 25, 426-8 for radical addition (k_i) aryl radicals 117 carbon-centered radicals 113-5, 221 cyanoisopropyl radicals 116 heteroatom-centered radicals 131 oxygen-centered radicals 118-9 solvent effects 25-6

temperature effects 25 for radical-radical reaction typical values 36 for radical-radical termination (k_t) by combination 236 by disproportionation 236 chain length dependence 234, 245-8 definition 235-6 effect of conversion 244-9 non-steady state conditions 238 prediction 244-5 steady state approximation 236-7 typical values 238 for reinitiation (k_{iT}, k_{iM}) 235 for ring opening cyclobutylmethyl radicals 198 cyclopropylmethyl radicals 196 in ring-opening polymerization 195-6 temperature dependence, cumyloxy radicals 56 use of radical clocks in calibration 54 rate of polymerization, deactivation by reversible coupling 460-1 reaction conditions 24 see also pressure effects; solvent effects; temperature effects; viscosity effects effect on hydrogen abstraction 33 effect on radical-radical reactions 42-3 influence on combination vs disproportionation 42-3 solvent effects 25-6 temperature effects 24-5 reaction diffusion 243, 248-9, 251 reactivity ratios definition penultimate model 342 terminal model 338 estimation from composition data 360-1 from monomer sequence distribution 362-3 for common monomers 339 for gradient copolymers 526 implicit penultimate model 347-8 in ATRP 488 prediction 363 NMR chemical shifts 364 *O-e* scheme 363 5 solvent effects 361, 364, 429-32 substituent effects 344-7 template effects 438 terpolymerization 357 reactivity-selectivity principle and hydrogen atom transfer 30 and radical addition 24-5 rearrangement of radicals during polymerization 185-212

cyclopolymerization 185-94 intramolecular atom transfer 208-12 ring-opening polymerization 194-208 redox initiators 104 see also photoredox initiators metal complex/organic halide 104-5 transition metal salts effects 85-6, 95-6 with alkyl hydroperoxides 93 with ceric ions 104-6 with diacyl peroxides 85-6 with hydrogen peroxide 96 with inorganic peroxides 94-6 with N.N-dimethylaniline/BPO 86 7 with persulfate 95-6 regiosequence isomerism see head vs tail addition reptation 243, 248-9 residual termination 243, 249 retardation definition 234 in RAFT 517-8 kinetics 266-7 $k_{\rm p}$ effects 280 mechanisms 266 with transfer agents 279 addition-fragmentation chain transfer 297 solvents 294 thiols 290 VAc polymerization 294 retarders 264 7 definition 264 reverse transfer constant in RAFT process 288 9, 504 5 reversible addition-fragmentation with 1,1-diphenylethylene 455, 470 with captodative monomers 470 reversible addition-fragmentation chain transfer (RAFT) polymerization 7, 250-1, 288, 297, 456, 470-1, 499, 501-21 block copolymer synthesis 543-4 diblock copolymers prepared by 543 hydrophilic-hydrophobic blocks 543 macromonomer RAFT 502, 543 mechanism 544 order of constructing blocks 544 segmented block copolymers 547-8 transformation reactions 544, 546 triblock copolymers 546 7 copolymerization 529-30 mechanisms 530 statistical/gradient copolymers 529 dispersities 505, 519 effect of Lewis acids 519 effect of pressure 519 effect of solvents 518 effect of temperature 518-9

emulsion polymerization 520-1, 544 end-functional polymer synthesis 545 α-functionalization 539-40 dormant chain end removal 531 ω-functionalization 538-9 gel effect 517 graft copolymer synthesis 558-63 heterogeneous polymerization 520-1 kinetic simulation 501 macromonomer RAFT agents 501-2 mechanism 498 9, 503 5 miniemulsion polymerization 520-1 molecular weight distributions 519-21 of acrylates, by xanthates 464 of MAH with S 526 of VAc with dithiocarbamates 464 with xanthates 502-3, 506, 513, 521 RAFT agent synthesis 515-7 rates of polymerization 503 reaction conditions 518-20 retardation 517-8 and inhibition period 517-8 methacrylate polymerization 517-8 S polymerization 517-8 reverse transfer constants 504 5 review 502 side reactions 517-18 star synthesis GPC distributions 553 mechanisms 551-4 precursors 549, 551 microgel formation 554-5 self-condensing vinyl polymerization 555-6 shell-crosslinking of micelles 555 dendritic cores 556-7 thiocarbonylthio RAFT agents 464, 501-2, 505 - 14transfer constants 504 xanthates 502-3, 506, 513, 521 reversible chain transfer 288-9, 456, 485, 498-525 see also reversible addition-fragmentation chain transfer (RAFT) polymerization by homolytic substitution 498-9 by RAFT 498 9 degenerative chain transfer 498 effect of reaction conditions 500 in iodine transfer polymerization 456, 499, 521 2 in stibine-mediated polymerization 499 in telluride-mediated polymerization 456, 499 molecular weights and distributions 499-501 with methacrylic macromonomers 499 ring-opening of cyclobutylmethyl radicals 198

of cyclopropylmethyl radicals 196-7 of dioxolan-2-yl radicals 201 ring-opening copolymerization ATRP in 497 of ketene acetals 195 of methylenecyclohexadiene spiro compounds 199 of spiroorthocarbonates 206 of spiroorthoesters 206 synthesis of end-functional polymers 195 ring-opening polymerization 194-208 double-ring opening 205-8 effect of concentration 196 effect of temperature 196, 200, 202-3 of bicyclobutanes 195 of caprolactone 545 of cyclic allyl sulfides 204 5 of cyclopropylstyrene 196 of ketene acetals 199-203, 379-80 effect of temperature 199 polyester synthesis 199 reversibility 201 substituent effects 199-201 of methylene substituted cyclic compounds 199-205 of 2-methylene-1,3-dioxolanes 200-3 of 4-methylene-1,3-dioxolanes 202-3 effect of temperature 202-3 polyketone synthesis 202-3 substituent effects 202-3 of 2-methylenetetrahydrofurans 204 of 2-methylenetetrahydropyrans 204 of ring cyclic allyl sulfides 204-5 of spirodithioacetal rings 108, 206 of spiroorthocarbonates 206-7 of spiroorthoesters 206-7 of vinyleycloalkanes 196-9 ring size effects 198 of vinylcyclobutanes 197-8 of vinylcyclohexane 198 of vinylcyclopentane 198 of vinylcyclopropanes 195, 206-7 rate constant for ring opening 196 reversibility 196 substituent effects 196-7 of vinyloxiranes 197-8 of vinvlsulfones 198-9 rate constant for ring-opening 195-6 synthesis of polyamides 199 synthesis of polyesters 195, 199 synthesis of polyketones 195, 202-3 template polymerization 440 volume expansion 194-5, 205 ruthenium complexes as catalysts for ATRP 488, 489, 492, 495

scandium triflate, effect on RAFT process 519 Schultz-Flory distribution 240 β-seission see fragmentation secondary radicals definition 54 from peroxides by β-scission 112 segmented block copolymers 547-8 definition 334 synthesis by NMP 547 by RAFT 547 8 mechanisms 547-8 selenides, as initiators 466-7 selenium-centered radical-mediated polymerization 466-7 selenium-centered radicals 132, 522 phenylseleno radicals 132 polarity 132 rate constants 131 self-condensing vinyl polymerization 548, 555-6 with ATRP initiators 555-6 with NMP initiators 555-6 with RAFT agents 555-6 self-initiated initiation see thermal initiation self-reaction of carbon-centered radicals 11 SFRMP see stable (free) radical-mediated polymerizations SG1 nitroxide, in NMP 475 short chain branches, by backbiting 320 side reactions in ATRP 488-9, 534-5 in iodine transfer polymerization 522 in metal complex-mediated radical polymerization 485 in NMP 478-9, 533 in radical-induced grafting 390-1 in RAFT polymerization 517-18 in sulfur-centered radical-mediated polymerization 466 silicon-centered radicals 131 polarity 131 rate constants 131 solvent effects see also cage reaction on chain transfer, to PVAc 324-5 on copolymerization 25, 336, 357, 429-32 bootstrap effect 357, 431-2 model studies 431 monomer sequence distribution 357 of macromonomers 401 reactivity ratios 361, 429-32 on fragmentation benzoyloxy radicals 127 cumyloxy radicals 125 /-butoxy radicals 54, 56, 123 on hydrogen atom transfer

to alkoxy radicals 34 to t-butoxy radicals 55, 123 to chlorine atoms 34 on initiation 55 on initiator decomposition alkoxyamines 140 DBPOX 89 diacyl peroxides 83 di-t-alkyl peroxides 91 dialkyl peroxydicarbonates 87 dialkyldiazenes 73 on initiator efficiency 75 on polymerization 25, 425-9, 497 t-BMA 428 EMA 428 k_v 426–8 MAA 428 9 mechanisms 426 MMA 427 8 molecular weight 428 propagation rate constants 25 \$427 tacticity 174 175, 428 9 VAc 179, 324-5, 427-8 on radical addition 25-6, 426-9 to arenethiyl radicals 26 on radical reactivity *i*-butoxy radicals 56, 123-4 mechanisms 426 phosphinyl radicals 132 on radical-radical reactions 42 combination vs disproportionation 43, 255 on radical-radical termination, PMMA 262 on RAFT process 518 SOMO-HOMO orbital interactions 27 SOMO-LUMO orbital interactions 27 spin trapping initiation mechanism 134-6 limitations 135 spiroorthocarbonates, ring-opening polymerization 206-7 spiroorthoesters, ring-opening polymerization 206 - 7spontaneous initiation see thermal initiation stable (free) radical-mediated polymerizations (SFRMP) 457-86 carbon-centered radical-mediated 467-71 kinetics and mechanism initiators, iniferters and initers 457-8 molecular weight distributions 458-60 degree of polymerization 458-9 dispersity 459, 461 nitrogen-centered radical-mediated 483-4 oxygen-centered radical-mediated 471-84 nitroxide-mediated (NMP) 471-83 polymerization kinetics 460-1

selenium-centered radical-mediated 466 7 sulfur-centered radical-mediated 461-6 stable radicals. see also nitroxides; stable (free) radicalmediated polymerizations (SFRMP) as radical traps, for initiation mechanism 138-40 inhibition by 233-4, 267-8 star polymers 547-57 star block or radial block copolymer 548, 550 core crosslinked star microgel 548 dendritic polymers 548, 556-7 dispersity of 549 hyperbranched polymers 555-7 mikto-arm star 548-9 synthesis 547-57 arm-first approach 548, 554 5 core-first approach 548, 549-54 self-condensing vinyl polymerization 548, 555.6 State Correlation Diagram (SCD) approach 27-8 statistical copolymerization see copolymerization statistical copolymers see also copolymers definition 333 Staudinger concept of polymer structure 1-2 stereoelectronic effects disproportionation pathways 39 in combination vs disproportionation 41-2 on cyclopolymerization 186 on hydrogen atom transfer 32 3 backbiting 209 on radical addition 23-4 on radical cyclization 23-4 on radical-radical reactions 41-2 on ring-opening polymerization 196, 200 stereosequence isomerism see tacticity steric effects 19-21 B-strain 19 in hydrogen atom transfer 30-1 in radical reactions 16 on chain transfer 283 to halocarbons 294 on fragmentation of t-alkoxy radials 124 on initiator decomposition, dialkyldiazenes 73 on radical addition head vs tail addition 16, 21, 117, 176-7, 178 intramolecular 187, 189 polymerization thermodynamics 215-16 on radical cyclization 23 on radical-radical reactions 37, 40-1 combination 257 combination w disproportionation 254-5, 256, 263 on rate of radical addition 19-21 steric inhibition 31

stibine-mediated polymerization 499, 524-5 molecular weight distributions 524 structural irregularities control 279, 413-22 from anomalous propagation 167-8 from chain transfer 3-4, 287 from initiation 3 4, 49 50 from radical-radical termination 176 structural isomerism see backbiting; cyclopolymerization; ring-opening polymerization styrene (S) aromatic substitution by benzoyloxy radicals 5, 52, 127 by hydroxy radicals 128 by isopropoxycarbonyloxy radicals 128 by phenyl radicals 52, 117 Diels-Alder reaction 107-9, 251, 317 Hammett parameters 21-2 inhibition of VAc polymerization 265-6 reaction with alkyl radicals penultimate unit effects 345, 346 rate constants 114 reaction with arenethiyl radicals 132 reaction with benzoyloxy radicals 5, 52, 120, 127 reaction with t-butoxy radicals 52-3, 120, 134 reaction with cumyloxy radicals 56, 120, 125 reaction with cyanoisopropyl radicals 52, 114 reaction with hydroxy radicals 120 reaction with isopropoxycarbonyloxy radicals 128 reaction with methyl radicals 52, 114 reaction with oxygen-centered radicals 120 reaction with phenyl radicals 52 rate constants 114 solvent effects 26 reaction with sulfate radical anion 129 ring-opening copolymerization 195, 199, 379 styrene (S) copolymerization Q-e values 365 reactivity ratios 339 thermal initiation 110-11 with AMS 353 with AN bootstrap effect 431 solvent effects 430 with BMA, using NMP 527 with MAA, bootstrap effect 431 with MAH 526 bootstrap effect 431 with MAN bootstrap effect 431 solvent effects 429 with MMA bootstrap effect 431

ionic liquids effect 433 Lewis acid effects 435, 436 solvent effects 429 using NMP 527 styrene graft polyolefins 399 styrene (S) polymerization backbiting 211-12 biradical mechanism 107 catalytic chain transfer 311, 316 chain transfer to ally I halides 303 to ally I sulfides 300 to ally I sulfonates 302 to cobalt complexes 316 to disulfides 292 to halocarbons 283, 293 to macromonomers 307 to monomer 317 to polymer 320 to silylcyclohexadienes 309 to solvent 295 to thiohydroxamic esters 309 to thiols 283, 290 to thionoesters 309 to vinvl ethers 299 combination vs disproportionation 253 5, 258 effect of oxygen 269 enzyme-mediated polymerization 440 inhibition constants 265 initiator efficiency in 75-6, 92 kinetic parameters 219 k_p 217, 221, 247, 427 solvent effects 427 $k_{\rm t}$, chain length dependence 246–7 tacticity 175 thermal initiation 107-9, 251, 317 Mayo mechanism 107-8 thermodynamics 215 with AIBN initiator 51, 75-7, 116, 141, 416 with BPO initiator 51, 63, 85, 141, 415-16 with DBPOX initiator 51 with disulfide initiators 462-3 with dithiocarbamate initiators 465 with DTBP initiator 92 with NMP 108, 476, 480-1 with RAFT 520 with iodine transfer polymerization 522 with ATRP 496-7 with stibine-mediated polymerization 524 substituent effects fluorine, on alkyl radicals 14 on C-C bond dissociation energies 23, 34 on C-H bond dissociation energies 34 on C-O bond dissociation energies 23, 34 on ceiling temperature 215 on chain transfer

to disulfides 292 to thiols 290 on combination vs disproportionation 40, 42 for 2-carboalkoxy-2-propyl radicals 255 for phenylethyl radicals 253-4 on cyclopolymerization of 1,6-dienes 187-8 on fragmentation of alkoxy radicals 123-5 on hydrogen atom transfer from toluene 22 on $k_{\rm d}$ diacyl peroxides 83 dialkyldiazenes 73 on monomer reactivity ratios 344 7 on O-H bond dissociation energies 34 on radical addition 20 to acrylate esters 18 to halo-olefins 17-18 to styrene 22 on radical cyclization 23-4 on regiospecificity of radical addition 19 on ring-opening polymerization of 2-methylene-1,3-dioxolanes 200-1 of vinylcyclobutanes 198 of vinylcyclopropanes 196-7 of vinyloxiranes 197-8 on stereochemistry of radical addition 20 sulfate radical anion 129-30 abstraction vs addition 129-30 conversion to hydroxyl radical 130 effect of pH 129 hydrolysis to hydroxy radical 130 sulfides, chain transfer to 292-3 sulfonic acids, rate enhancement of NMP 479 sulfonyl halides, as ATRP initiators 488-9, 493, 550.556-7 and TERMINI agents 556-7 side reactions 489 star precursor 550 sulfonyl bromides 489 sulfonyl chlorides 489, 550, 556-7 sulfur-centered radical-mediated polymerization 461-6 initiators see disulfides, dithiocarbamates, dithiuram disulfides, xanthates side reactions 466 sulfur-centered radicals 132 alkanethiyl radicals 132 arenethiyl radicals 132 benzoylthiyl radicals 132 N,N-dialkyl dithiocarbamate radicals 461, 463-4 polarity 132 rate constants 131 sulfur dioxide, copolymerization with depropagation 353 supercritical carbon dioxide as solvent in polymerization 432, 518

for RAFT 518 suspension polymerization initiation 63 termination kinetics 249 with ATRP 498 tacticity 167 chain statistics 170-5 1st order Markov 171-2, 175 Bernoullian 171, 173, 175 Coleman-Fox 172 random 170-1 control **ATRP 423** by topological polymerization 441-2 effect of amines 428-9 effect of ionic liquids 433 effect of Lewis acids 425, 434, 519 effect of solvent 174-5, 425, 428-9, 433 effect of steric factors 174 effect of temperature 174-5 effect of template polymer 437-440 NMP 423 RAFT 423, 519-20 with cobalt complexes 423-4 with cyclopolymerization 424 definition 168-9 diastereoisomers 169-70 dyad composition 170-2, 421 meso169-70 racemic169 70 effect on backbiting 211 measurement of 173 of diene polymers 183 of 1,1-disubsituted monomers 174-5 of bulky methacrylate polymers 174, 423-4 in RAFT 519-20 of PAA 173 of PAM 174-5 of PAN 175 of PtBMA 428 of PDMAM 174-5 of PEMA 428 of PMA 175, 433 of PMAA 428-9 of PMAN 173, 175 of PMMA 173, 175, 424, 427, 434-5 of PNIPAM 174 5, 434 5 of polyacrylamides 174-5, 423-4, 434-5 of poly(a-alkoxymethacrylates) 434 of PS 175 of PVAc 173, 175, 428 of PVC 173, 175 of PVF 173 of vinyl polymers 174 penultimate unit effects 171 2, 346 7

reviews 168 structure atactic chain 169 heterotactic chain 169 isotactic chain 168 syndiotactic chain 169 terminology 168-72 tail addition see also head vs tail addition definition 4, 16 tail-to-tail linkages in PVAc 324 with head-to-head linkages 167, 176 tapered copolymers see gradient copolymers techniques for measurement of branching 208-10, 320 of chain transfer constants 283-7, 321 of combination vs disproportionation 259-60 end group determination 259 Gelation technique 258-9 mass spectrometry 259 model studies 252-3 molecular weight distribution 259 molecular weight measurement 259 of compositional heterogeneity 381 of initiation mechanism 133 chemical methods 144-5 direct detection of end groups 141-5 kinetic studies 133 labeling techniques 145-9 radical trapping 133-41 spectroscopic methods 133, 141-4 spin trapping 134-6 of kp 216–18 of $k_1 238$ of monomer sequence distribution 354-7, 362 - 3of reactivity ratios 359-63 of tacticity 173 telechelic polymers definition 374 synthesis 62, 375-6 by ATRP 537 by copolymerizing butadiene or acetylene derivatives 380 by living radical polymerization 103, 537 with functional transfer agents 289 telluride-mediated polymerization (TERP) 456, 499. 522-4 dispersities 523 initiators for 524 kinetics 522 mechanisms reversible activation/deactivation 522-3 reversible chain transfer by homolytic substitution 522-3

molecular weight distributions 523 substituent effects 522 telomer synthesis 293, 486 temperature effects on backbiting in E polymerization 209 on chain transfer backbiting 209 C_{tr} 282 to PVAe 324 to PVF 325 on combination vs disproportionation, for PS 254, 260-2 on copolymerization ceiling temperature 353-4 on inhibition, by nitroxides 268 on initiation 56, 123 on initiator efficiency 75-6 on propagation backbiting 209, 211-2 ceiling temperature 213 4 diene monomers 183-4 k_p 282 ring-opening polymerization 196, 200, 202 - 3VAc 179 VF 181 on radical addition 24-5 on radical-radical reactions 42 combination vs disproportionation of tbutyl radicals 43 on RAFT process 518-9 on tacticity 175 template polymerization 437-40 and gel or Norrish-Trommsdorff effect 438 covalently bonded templates 438-40 definition 437 enzyme-mediated polymerization 437 ladder polymerization 194, 439 mechanisms 437 non-covalently bonded templates 437-8 of MMA 437-8 of multimethacrylate 439-40 optical induction 439 ring-opening polymerization 440 TEMPO see 2,2,6,6-tetramethylpiperidin-N-oxyl terminal model for copolymerization 337-42, 366 binary copolymerization 354-5 chemical control model 366-8 diffusion control models 368-70 termination 4 chain transfer constants 283-4 disproportionation vs combination 251-64 during copolymerization 370-4 in statistical copolymerization 366-74 pathways for 233-4

structural irregularities 3 termination kinetics 235 classical kinetics 235-8 diffusion controlled termination 242-4 at low conversion 244-8 at medium to high conversion 244, 248-9 at very high conversion 244 heterogeneous polymerization 249-50 living radical polymerization 250-1 molecular weight averages 238-40 molecular weight distributions 239-42 molecular weights 238-40 TERMINI agents 556-7 ternary copolymerization see terpolymerization TERP see telluride-mediated polymerization terpolymerization 357-9 azeotropic compositions 359 composition equation 357-9 mechanism 357 reactivity ratios 357 tetraalkylammonium salts, as ionic liquids 432 tetraalkylphosphonium salts, as ionic liquids 432 tetrachloromethane, as ATRP initiator 488 tetraethyldithiuram disulfide, in SFRMP 461 tetrahydronaphthalene, formation of, in S polymerization 107-8 tetramethylenes, unifying mechanism, donoracceptor polymerization 111 2,2,6,6-tetramethylpiperidin-N-oxyl (TEMPO) alkoxyamine formation from AIBN decomposition 476 alkoxyamine concentration, S polymerization 476 in block copolymer synthesis 541, 545 in methacrylate polymerization, unsaturated chain end production 534-5 in S polymerization 476, 480, 482 inhibition by 14, 266, 268 rate enhancement of NMP 479 reaction with alkyl radicals, rate constants 138 reaction with benzyl radicals, rate constants 138 stability 14, 478 sulfur analog 463 thermal decomposition of phenylethyl alkoxyamine 478 tetramethylsuccinonitrile, by-product from AIBN 60-1,77 tetrathiol, thiol-ene polymerization 379 theoretical studies on 1,2-chlorine shifts 179-80 on reactivity ratios in copolymerization 365-6 radical addition 20-1, 26-8 thermal initiator decomposition and effective rate of initiation 57-8 diacyl peroxides 82-3

dialkyldiazenes 72-4 diaroyl peroxides 82-3 thermal initiation 106-11 of acrylate ester polymerization 109-10 of copolymerization 110-1, 351 of MMA polymerization 109-10 of pentafluorostyrene polymerization 109 of S polymerization 107-9 unified mechanism for 111 thermal stability of PMMA 2, 417-20 of PS 53, 85, 414-7 prepared by ATRP 416-7 prepared with AIBN initiator 416 prepared with BPO initiator 53, 85, 415-6 prepared by NMP 416-7 prepared by RAFT 416 7 of PVC 2, 179, 420-1 thiocarbonylthio RAFT agents 501-2, 505-14 see also dithioacetate RAFT agents; dithiobenzoate RAFT agents; dithiocarbamates; reversible additionfragmentation chain transfer (RAFT) polymerization; trithiocarbonate RAFT agents; xanthates activity prediction, molecular orbital calculations 507 bis-RAFT agents 511, 546 canonical forms 506 characteristics for efficient RAFT polymerization 505 end-functional polymer synthesis 539 relative effectiveness 506 synthesis of 515 6 transfer constants 505 thioglycolic acid, as transfer agent 291 thiohydroxamic esters chain transfer constants 309 chain transfer to 308-9 thiol-ene polymerization 378-9 thiols as transfer agents, in NMP 531 chain transfer constants 290 chain transfer to 289, 290-1, 382-3 in thiol-ene polymerization 378-9 polarity 290 polymeric 388 synthesis of end-functional polymers 377 protein conjugates 538 thionoesters chain transfer constants 309 chain transfer to 308-9 thiophenol, radicals from 132 titanocene dichloride as initiator 484 control of propagation 424

titanous salts, radical trapping 134-6 toluene chain transfer to 295 Hammett parameters for abstraction 22 reaction with t-butoxy radicals 55 topological radical polymerization 441-3 transfer to initiator see chain transfer to initiator transformation reactions 387 9, 544 6 anionic-radical 387-8 cationic-radical 389 group transfer-radical 389 in synthesis of block and graft copolymers. 387-9, 544-6 to ATRP 544-6 to NMP 544-5 to RAFT 544, 546 transition metal salts/complexes see also catalytic chain transfer; chromium complexes; cobalt complexes; iron complexes; molybdenum complexes catalysts for ATRP 487-9, 492-7 cobalt complexes 492 copper complexes 488, 492-5 iron complexes 492, 496 manganese complexes 492 molybdenum complexes 492 nickel complexes 492, 496-7 palladium complexes 492 rhenium complexes 492 rhodium complexes 492 ruthenium complexes 488-9, 492, 495 catalysis for chain transfer to halocarbons 294 catalysts for catalytic chain transfer chromium complexes 315 cobalt complexes 310-316 molybdenum complexes 315 iron complexes 315 inhibition by 136, 234, 265-6, 273 mediators for SFRMP chromium complexes 484 cobalt complexes 484-486 molybdenum complexes 484 titanium complexes 484 radical trapping cupric salts 136 ferric salts 136 mercuric hydride 137 tin hydrides 137 titanous salts 136 reaction with carbon-centered radicals 136 reaction with dialkyldiazenes 73 redox initiation 85-6, 93, 95-6, 104-5, 381 with alkyl hydroperoxides 93 with diacyl peroxides 85-6 with hydrogen peroxide 96 with persulfate 95

transition metal-mediated radical polymerization see atom transfer radical polymerization transition state for combination 39 for disproportionation 39, 41 for hydrogen atom transfer 30 for radical addition 17, 20 for radical cyclization 23 for radical-radical reaction 39 triaryImethyl radicals 469 see also triphenylmethyl radical triazolinyl radicals decomposition mechanism 484 disproportionation 483 in block copolymer synthesis 483 in SFRMP 483-4 stability 484 triblock copolymer synthesis 546-7 by ATRP 546 by NMP 546 by RAFT 546 trichloroacetyl isocyanate, facile reaction with protic end groups 144-5 trichloromethane derivatives, as ATRP initiators 488 trichloromethyl-functional initiators, as ATRP macroinitiators 546 trichloromethyl radicals Hammett parameters for abstraction from substituted toluenes 22 for addition to substituted styrenes 22 polarity 22 reaction with ethylene 17 reaction with fluoro-olefins 17-8, 22 triethylamine, chain transfer to 295 trifluoroethyl acrylate polymerization, with oxygen-centered radicals 483 trifluoroethylene polymerization, head vs tail addition 180-1 reaction with t-butoxy radicals 17 reaction with methyl radicals 17 reaction with trichloromethyl radicals 17 reaction with trifluoromethyl radicals 17 trifluoromethyl radicals polarity 22 reaction with ethylene 17 reaction with fluoro-olefins 17-18, 22 structure 12 2-(trimethylammonium)ethyl methacrylate salts, with ATRP in solution 497 triphenylverdazyl radical in SFRMP 483 in block copolymer synthesis 483 triphenylmethyl radicals abstraction vs addition 35 diacyl peroxides decomposition 85

from triphenylmethylazobenzene 468 in SFRMP 467-9 inhibition by 268 pathways for combination 37 quinonoid intermediate 469 stability 14 triphenylmethylazobenzene as initiator 70-2, 468 in SFRMP 72, 468 kinetic data for decomposition 70-1 triphenylverdazyl radical, inhibition by 76, 268 trithiocarbonate RAFT agents 506, 512, 515-6, 519, 548 see also reversible addition-fragmentation chain transfer (RAFT) polymerization; thiocarbonylthio RAFT agents cyclic, in segmented block copolymer synthesis 548 non-symmetrical 512 symmetrical 512, 546 synthesis of 515-6 Trommsdorf effect see Norrish-Trommsdorf effect undecyl radicals Hammett parameters for abstraction from substituted toluenes 22 for addition to substituted styrenes 22 unsaturated chain ends from addition-fragmentation chain transfer 296-7, 303, 308 from catalytic chain transfer 310-2 from transfer to monomer 311-2, 319 in PMAN, from disproportionation 257 in PMMA effect on thermal stability 418-20 from catalytic chain transfer 310-12 from disproportionation 256, 262, 418 from initiation 53 from TEMPO post-polymerization 534-5 in PS effect on thermal stability 415-6 from benzoate chain ends 415-6 from disproportionation 253-5, 260-1 UV-visible spectrophotometry 15-16 from radical-radical termination 251-2 kinetic studies 133 UV-visible spectroscopy, measurement of end groups 141 verdazyl radicals, inhibition by 76, 268 VAc see vinyl acetate VC see vinyl chloride VF see vinyl fluoride VF2 see vinylidene fluoride vinyl acetate (VAc)

abstraction vs addition 121 radical addition rate constants 114-5, 119, 131 carbon centered radicals s 114-5 t-butoxy radicals 122 heteroatom-centered radicals 131 oxygen-centered radicals 119, 121 specificity 121 vinyl acetate (VAc) copolymerization Q-e values 365 reactivity ratios 339 with tBA 526 with ethylene 209-10 solvent effects 429 with MMA, solvent effects 429 vinyl acetate (VAc) polymerization backbiting 209-11 chain transfer intramolecular 209 11 to allyl sulfides 300 to allyl sulfonates 302 to BPO 85 to disulfides 292 to halocarbons 283, 293 to initiator 85 to monomer 318 to polymer 209 11, 320, 323 5 to solvent 294-5 to thiohydroxamic esters 309 to thiols 290 to thionoesters 309 to vinyl ethers 299 combination vs disproportionation 263 effect of oxygen 269 head vs tail addition 178-9, 522 inhibition by styrene 265-6 inhibition constants 265 kinetic parameters 219 $k_{\rm p} \, 217$ retardation 294 solvent effects 25 324-5, 427 tacticity 173, 175 thermodynamics 215 with AIBN initiator 77 with ATRP 488 iron catalysts 496 with iodine transfer polymerization 522 side reactions 522 with ¹³C labeled initiator with RAFT 520 with SFMRP 465, 483, 485-6 with stibine-mediated polymerization 524 vinyl acrylate, cyclopolymerization 192-3 vinyl chloride (VC) radical addition

carbon-centered radicals 114 5 rate constants 114-5 vinyl chloride (VC) copolymerization Q-e values 365 reactivity ratios 339 vinyl chloride (VC) polymerization 1,2-chlorine shifts 179-80 backbiting 208, 211 chain transfer intramolecular 208, 211 to BPO 85 to halocarbons 283 to initiator 85 to monomer 180, 318-19 to polymer 208, 211, 320, 325 combination vs disproportionation 263 head vs tail addition 179-80 tacticity 173, 175 thermodynamics 215 with ¹³C labeled AIBN 147 vinyl ethers chain transfer to 298-9 chain transfer constants 299 vinyl fluoride (VF) radical addition t-butoxy radicals 17 methyl radicals 17 trichloromethyl radicals 17 trifluoromethyl radicals 17 vinyl fluoride (VF) polymerization chain transfer to polymer 325 head vs tail addition 180-1 tacticity 173 vinyl methacrylate, cyclopolymerization 192-3 N-vinylcarbazole, induced decomposition of BPO 86 vinvleveloalkanes ring-opening polymerization 196-9 ring size effects 198 vinyleyclobutanes, ring-opening polymerization 197 - 8vinylcyclohexane, ring-opening polymerization 198 vinylcyclopentane, ring-opening polymerization 198 vinyleyclopropanes double ring-opening polymerization 206-7 ring-opening polymerization 195-7 vinylidene cyanide, copolymerization with S, thermal initiation 110 vinylidene fluoride (VF2) polymerization, head vs tail addition 180-1 radical addition 21-2 t-butoxy radicals 16-7, 23, 122 methyl radicals 17 trichloromethyl radicals 17

trifluoromethyl radicals 17 N-vinylimidazole induced decomposition of BPO 86 template polymerization 438 N-vinyloxazoline, grafting to polyolefins 400 vinyloxiranes, ring-opening polymerization 197-8 vinylpyridine polymerization graft of 309 with ATRP 543 with NMP 480, 541 with RAFT 508 N-vinylpyrrolidone (NVP) polymerization induced decomposition of BPO 86 **RAFT 507** stibine-mediated 524 N-vinylpyrrolidone (NVP) copolymerization with NMP 526, 528 vinylsilane graft polyolefins 399-400 vinvlsulfones ring-opening polymerization 198-9 viscosity effects dialkyldiazenes 74, 75 on cage reaction 61 DBPOX 89 diacyl peroxides 83, 84 hyponitrites 78 on copolymerization of macromonomers 401 on initiator efficiency 60-1, 75 on k_e 83 on radical-radical reactions 43 on radical-radical termination 238, 242, 248 volume expansion in double ring-opening polymerization 205 in ring-opening polymerization 194-5 xanthates see also reversible additionfragmentation chain transfer (RAFT) polymerization; thiocarbonylthio RAFT agents as alkyl radical source 296, 502-3 Barton-McCombie reaction 296, 502-3 as RAFT agents 502-3, 506, 513, 521 table 513 monomers polymerized 513 emulsion polymerization 521 as photoiniferter 463-5 MADIX 502 synthesis 515 xanthogen disulfides as initiators 461 chain transfer to 292

z-mer, in emulsion polymerization 63-4

This Page is Intentionally Left Blank